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# 1 Post-fire mechanical properties of corroded grade D36 marine steel

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## 11 Abstract

12 This paper presents an experimental study on the post-fire mechanical properties of  
13 corroded grade D36 marine steel. Corrosion and high temperature are two main  
14 adverse factors that degenerate the mechanical properties and lead to decrease in  
15 strength and durability of steel. The combined influence of corrosion and high  
16 temperature on the mechanical properties of marine steel was investigated in this  
17 study. A series of tests were conducted to determine the post-fire mechanical  
18 properties of corroded marine steel. Specimens were corroded by a salt spray test  
19 which simulated the marine atmospheric environment, and then heated to 500 °C or  
20 900 °C. After exposure to corrosion and high temperature, the elastic modulus, yield  
21 stress, ultimate strength and stress-strain curves of grade D36 marine steel specimens  
22 were obtained from tensile coupon tests. Three-dimensional laser scanning was used

1 23 to evaluate the impact of corrosion. The microstructure and fracture morphology  
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3 24 analyses were carried out by scanning electron microscope (SEM) to obtain the  
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6 25 combined effects of corrosion and high temperature. It was shown that the corrosion  
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8  
9 26 and high temperature had significant influence on the microstructures and  
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11  
12 27 corresponding fracture morphology of grade D36 marine steel.  
13

14 28

17 29 **Keywords:** Corrosion, marine steel, mechanical properties, microstructure, post-fire,  
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19  
20 30 salt spray accelerated test, scanning electron microscope, three-dimensional laser  
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22  
23 31 scanning.  
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25 32

### 28 33 **1. Introduction**

31 34 Large and complex steel offshore industrial facilities such as Blue Whale 1,  
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34 35 Umm Lulu Gas Treatment Platform have been constructed because of the increasing  
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37 36 demand for marine resources [1]. Due to harsh marine environment, marine steel with  
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40 37 good durability is designed for offshore industry facilities to resist more adverse  
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43 38 factors [2], among which corrosion and fire are the two main detrimental factors.  
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46 39 Offshore industry facilities are suffered from corrosion damage, which accelerates  
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49 40 with service time. Such damage would be more severe when a catastrophe like a fire  
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51  
52 41 happens on facilities that have serviced for decades. Mechanical properties are crucial  
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55 42 indexes to evaluate the structural performance of steel structures after damage [3].  
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57  
58 43 Thus, post-fire mechanical properties of marine steel after exposure to combined  
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61 44 corrosion and high temperature were evaluated in this study, in order to determine  
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1 45 whether offshore industrial facilities could continue to be used after a fire.

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3 46 In the past few years, extensive experimental work has been performed on the  
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6 47 effects of corrosion and fire on steel separately [4-10]. On one hand, studies have  
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8  
9 48 been carried out on mechanical properties, mass changes, corrosion pits and corrosion  
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11  
12 49 mechanism of steel [11-15]. The elastic modulus, yield stress, ultimate strength and  
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14  
15 50 ultimate strain of steel influenced by corrosion have the similar variation tendency  
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18 51 which first decline and then slightly increase with corrosion duration, due to different  
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21 52 stages of corrosion. Corrosion process of steel is generally composed of two stages.  
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23 53 First, corrosion products emerge and accumulate on the surface of metal. Second,  
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26 54 corrosion products wrap on the whole surface to prevent further corrosion, which  
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28  
29 55 means that the latter stage provides protection to deeper corrosion of the metal. It has  
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31  
32 56 been proved that it is feasible to use the mass loss ratio to evaluate the degree of  
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35 57 corrosion. It is defined as the basic index in the time-dependent corrosion model of  
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38 58 steel, which is the base for complex corrosion models of steel.

39 59 Studies on the micro level of steel material provided more evidence to  
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42 60 demonstrate the influence of corrosion on steel, especially on corrosion pits and  
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45 61 corrosion mechanism [16-20]. The corrosion effect on the micro level of steel was  
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48 62 detected by X-Ray diffraction [18,21], three-dimensional laser scanning [15-16,22],  
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51 63 scanning electron microscope (SEM) [23-24], thermal analysis [18], and chemical  
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54 64 analysis [19]. These studies showed that the number and depth of the corrosion pits,  
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57 65 which led to stress concentration and reduction of the effective cross-sectional area,  
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60 66 increased with corrosion rate. Corrosion pits showed different status in various  
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1 67 corrosion conditions. Relevant results pointed out that if the status of corrosion pits  
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3 68 and corrosion mechanism were similar, mechanical properties of steel after different  
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6 69 corrosion treatments could be compared. Besides, the studies on corrosion  
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9 70 demonstrated it slightly changed the microstructure and fracture morphology of steel.

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11 71 Moreover, research work on mechanical properties of steel in fire and after a fire  
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13  
14 72 was conducted. Most studies on mechanical properties of steel in fire were applied to  
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17 73 the fire resistance design, such as European Code [25], Australian Standard [26], and  
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20 74 American Specification [27]. The post-fire mechanical properties of steel are  
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23 75 significant indexes to evaluate whether a structure could continue to be used after a  
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25  
26 76 fire. So far, studies on post-fire mechanical properties of steel were seldom reported,  
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28  
29 77 and limited investigation on marine steel was found in the literature. Tao et al. [28]  
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32 78 and Yu et al. [3] proposed prediction models for the residual mechanical properties of  
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34  
35 79 hot-rolled steel and cold-formed steel after cooling down from 600 °C to the ambient  
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37  
38 80 temperature. Chiew et al. [29], Outinen and Makelainen [30], Qiang et al. [31]  
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41 81 presented the mechanical properties of different grades of structural steel after a fire.  
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43  
44 82 Ren et al. [32] compared the differences in mechanical properties of the flat portion  
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47 83 and corner portion that cut from the same cold-formed C-section steel. In Wang et al.  
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49  
50 84 [33], the post-fire mechanical properties of Q460 high strength steel with water  
51  
52  
53 85 cooling and air cooling method were evaluated. Li et al. [34] conducted an  
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56 86 experimental study on the post-fire mechanical properties of Q690 structural steel  
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59 87 with different cooling methods. Lu et al. [9] carried out a series of experimental work  
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62 88 on the post-fire mechanical properties of cast steel. Test results indicated that cyclic  
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1 89 heating-and-cooling had no obvious effect on the post-fire mechanical properties of  
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3 90 G20Mn5N and G20Mn5QT steel. Huang and Young [35-36] presented the post-fire  
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6 91 mechanical properties of ferritic stainless steel and lean duplex stainless steel, and the  
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8  
9 92 relevant design equations were proposed. Above investigations revealed that the  
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11  
12 93 mechanical properties of steel decreased only marginally after exposure to a  
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14 94 temperature up to 600 °C, and a considerable reduction was observed when the  
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17 95 temperature was above 600 °C. In addition, the post-fire mechanical properties of steel  
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20 96 were also influenced by cooling methods.

21  
22 97 To the best knowledge of the authors, research on combined effects of corrosion  
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24  
25 98 and fire on mechanical properties of steel is limited. In marine environment, corrosion  
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28 99 is a constant process and is harmful to marine construction especially to steel  
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31 100 structures. Besides, there is a significant number of offshore industrial facilities that  
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34 101 are used for exploitation, storage and transportation of combustible oil and gas  
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37 102 resources. A leak of oil and gas is highly possible to cause fire and even explosion.  
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40 103 Fire damage on aged steel offshore facilities may be more severe and result in  
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43 104 catastrophic accidents. Han et al. [37] considered the temperature effect on the  
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46 105 corrosion behavior of 2205 stainless steel. Yu et al. [38] proposed an improved  
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49 106 numerical corrosion model for rebar steel regarding temperature and relative humidity  
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51  
52 107 as indexes. These two studies only considered temperatures range from 0 °C to 50 °C  
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54  
55 108 which are much lower than the temperature in a real fire. Li et al. [39] carried out an  
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58 109 experimental study on the post-fire mechanical properties of corroded 2205 duplex  
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60  
61 110 stainless steel, in which the corrosion and high temperature were conducted  
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1 111 simultaneously with solution-treated and water-cooling methods. Kong et al. [40]  
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3 112 examined the mechanical properties of corroded 316L stainless steel after exposed to  
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6 113 1050 °C and 1200 °C. These results showed that the multi-influential effects of  
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9 114 corrosion and fire on the mechanical properties were noteworthy. However, limited  
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11 115 studies were conducted on the post-fire mechanical properties of construction steel  
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14 116 after marine atmospheric corrosion.  
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17 117 In this study, the combined effects of corrosion and post-fire on the mechanical  
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20 118 properties of grade D36 marine steel are presented. An experimental study on the  
21  
22 119 mechanical properties of corroded marine steel after fire exposure with water cooling  
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25 120 method was conducted. A salt spray test was conducted to simulate the marine  
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28 121 atmospheric environment. The corroded specimens were heated and then water cooled  
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31 122 to the ambient temperature, in order to simulate the condition of offshore facility  
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33  
34 123 extinguished from fire. The elastic modulus, yield stress, ultimate strength and  
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36 124 ultimate strain of grade D36 marine steel after exposure to corrosion and high  
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39 125 temperatures were obtained by tensile coupon tests. Five different corrosion durations  
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42 126 and two different temperatures were selected for comparison. Three-dimensional laser  
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45 127 scanning and scanning electron microscope (SEM) were used to investigate the  
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48 128 influence on the dimensions and microstructures, respectively.  
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## 51 130 **2. Experimental program**

### 52 131 *2.1. Test devices*

53 132 Steel in offshore industrial structures above the sea level suffers from  
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1 133 atmospheric corrosion over its service life. Under this circumstance, a fire attack  
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3 134 would cause more detrimental impact. Consequently, rather than immersion corrosion,  
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6 135 a salt spray test was carried out to simulate the atmosphere corrosion environment.  
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9 136 The test was conducted in an atmosphere environment chamber which provides  
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11 137 constant neutral salt mist, stable temperature and humidity, as shown in Fig. 1(a). In  
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14 138 order to simulate different levels of corrosion, the specimens were exposed for 48  
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16  
17 139 hours, 96 hours, 192 hours, and 384 hours according to ASTM B117-16 [41].  
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20 140 The target temperature in the high temperature test was determined by Fire  
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22 141 Dynamics Simulation (FDS) of an offshore facility [42]. 500 °C and 900 °C were  
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24  
25 142 selected as when facilities were exposed to fire in open and indoor spaces,  
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27  
28 143 respectively. A small electrothermal furnace as depicted in Fig. 1(b) was used to  
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31 144 perform the high temperature test.  
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34 145 Specimens experienced corrosion and post-fire treatments were then tested by an  
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36 146 MTS universal testing machine (Fig. 1(c)), which was employed to obtain the  
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39 147 mechanical properties of the tensile coupon specimens. It should be noted that  
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42 148 specimens without experiencing corrosion and high temperature were also tested for  
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44  
45 149 comparison purpose.  
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47 150

## 50 151 *2.2. Specimen design*

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53 152 All specimens were manufactured by grade D36 marine steel with a nominal  
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56 153 yield stress of 355 MPa. The weathering steel with high durability is usually adopted  
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59 154 by ship and offshore facilities. The geometry and dimensions of the specimens were  
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1 155 determined according to ASTM B117-16 [41] and GB/T 228.1-2010 [43] (Fig. 2).  
2

3 156 Table 1 shows the corrosion duration and exposure temperature of the specimens. The  
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6 157 temperature with the symbol \* represents the measured temperature inside the  
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9 158 chamber.

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### 12 13 14 160 *2.3. Test procedure*

#### 15 16 17 161 *2.3.1. Salt spray test*

18  
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20 162 The salt spray test was conducted in an environmental chamber to accelerate the  
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22 163 corrosion condition of marine steel specimens. Generally, there were three steps for  
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24  
25 164 the accelerated corrosion. i) Pretreatment: All the specimens were numbered and  
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27  
28 165 weighted, and all their detailed properties were recorded. The mass loss ratios were  
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31 166 calculated according to Eq. (1) and are listed in Table 2. ii) Corrosion treatment: The  
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34 167 specimens were placed in the chamber on a shelf (Fig. 3) which was adopted to  
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36 168 provide a certain angle recommended by ASTM B117-16 [41]. The chamber was  
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39 169 fulfilled with salt fog generated by 5% wight NaCl solution. The temperature, relative  
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42 170 humidity (RH) and pH value inside were consistently controlled at 35°C, 100% and 7,  
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45 171 respectively. The corrosion periods were set at 48 hours, 96 hours, 192 hours, and 384  
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48 172 hours for observing the whole process of metal corrosion. iii) Post-processing: Two  
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51 173 specimens were taken out each time for different heating treatments, and the  
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53 174 remaining specimens were rotated between two top pipes every 48 hours, as  
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56 175 illustrated in Fig. 4. After taken out from the chamber, the specimen was cleaned by  
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59 176 tap water and dried by clean tissues according to ASTM specification [41]. They were  
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1 177 then weighed again to assess the corrosion and each was wrapped by a preservative  
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3 178 film to prevent extra corrosion. Fig. 5 displays the specimens after the salt spray test.

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$$\text{Mass loss ratio} = (m_o - m_1) / m_o \quad (1)$$
  
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8  
9 180 where  $m_o$  is the initial mass of the specimen;  $m_1$  is the mass of the specimen after  
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11 181 cleaning.

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17 183 *2.3.2. Heating and cooling procedures*  
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20 184 An electrothermal furnace was used to heat up the specimens, so as to evaluate  
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22 185 the post-fire behaviour of grade D36 marine steel after corrosion. During the test, the  
23  
24 186 corrosion product was not cleaned to simulate a situation closed to the engineering  
25  
26 187 practice. The specimen was placed on a small mantelpiece as shown in Fig. 6 which  
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28 188 was used to prevent the specimen from touching the furnace wall. The heating process  
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30  
31 189 was set according to ISO 834 [44]. 500°C and 900°C were defined as the target  
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33 190 temperatures according to the Fire Dynamics Simulation (FDS) [42]. The temperature  
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35 191 variation on the specimen was recorded by a thermocouple installed at the mid-length  
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37 192 of the specimen. Once the temperature reached the target value, the specimen was  
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39 193 kept in the electrothermal furnace for another 15 mins to ensure that it was uniformly  
40  
41 194 heated. Afterward, it was taken out from the electrothermal furnace and immediately  
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43 195 cooled down to the ambient temperature by water. Fig. 7 shows the specimens after  
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45 196 the salt spray test and high temperature test.  
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58 198 *2.3.3. Tensile coupon test*  
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1 199 The mechanical properties, including the elastic modulus, yield stress, ultimate  
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3 200 strength and ultimate strain, were obtained by tensile coupon tests which were  
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6 201 performed by an MTS universal testing machine. The specimens' surface was cleaned  
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9 202 by sandpapers followed by acetone and the specimens were weighed to assess the  
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11 203 mass loss. An extensometer was mounted (shown in Fig. 8) to monitor the strain  
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14 204 development. Displacement control with a rate of 1 mm/min was adopted. Fig. 9  
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17 205 presents all the specimens after fracture.  
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#### 21 207 *2.4. Three-dimensional laser scanning analysis*

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25 208 Three-dimensional laser scanning with an accuracy of 0.01 mm was conducted to  
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28 209 evaluate the effect of corrosion on the surface of the specimens. After tensile coupon  
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31 210 tests, the specimens' surface was cleaned and sprayed by eikonogen. The specimen  
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34 211 was placed in the center of the scanning table and the camera was rotated around the  
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37 212 specimen for one scan. A typical three-dimensional laser scanning result is given in  
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39 213 Fig. 10.  
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#### 42 215 *2.5. Microstructure and fracture morphology analysis*

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47 216 Microstructure and fracture morphology analysis were employed to evaluate the  
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50 217 combined effects of corrosion and high temperatures on the micro level of the  
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53 218 specimens. Considering that the effect of corrosion on the microstructure and fracture  
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56 219 morphology was limited, only the specimens 48 h-500 °C, 48 h-900 °C and 0 h-25 °C  
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59 220 were selected for analysis. There were four procedures in the pretreatment step. Cross  
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1 221 sections of the specimens were cut at 10 mm from the end and 15 mm from the  
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3 222 fracture position to analyze the microstructure and fracture morphology, respectively.  
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6 223 Sandpapers were used to grind the surface of the cut edge, and the cross-section was  
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9 224 polished to a mirror level. After etched by nitric acid alcohol liquid, the  
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11 225 microstructure and the fracture morphology of the specimen surfaces were carefully  
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14 226 observed by scanning electron microscope (SEM).  
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### 20 228 **3. Test results and discussions**

#### 22 229 *3.1. Specimen appearance*

25 230 Figs. 5, 7 and 9 show the specimens after the salt spray test, heating-cooling  
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27  
28 231 treatment, and tensile coupon tests, respectively. It is shown in Fig. 5 that color of  
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31 232 corroded specimens gradually became darker with the accumulation of corrosion  
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34 233 products. Corrosion products were partially distributed on the specimens after 48  
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36 234 hours' corrosion, while distribution of the corrosion product on the specimens  
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39 235 experienced 96 hours' and 192 hours' corrosion apparently spread. For the specimens  
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42 236 after 384 hours' corrosion, the whole surface of the specimen was uniformly covered  
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45 237 by the corrosion products. Fig. 5 suggested that the corrosion process and the  
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48 238 corrosion mechanism of metal materials were in a good agreement, which was also  
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50 239 proved by the tensile coupon tests. The corrosion products increased with the  
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53 240 corrosion duration and the color of the corrosion products gradually became darker. It  
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56 241 was found from Fig. 7 that the color of the specimens treated by 900 °C was darker  
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59 242 than that treated by 500 °C, which indicated that the reaction between steel, corrosion  
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1 243 product and air in 900 °C were more severe than that in 500 °C. Fig. 9 shows that the  
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3 244 specimens under 0 hour and 384 hours' corrosion approximately fractured at the  
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6 245 middle of the gauge length and the others failed near the end of the gauge length,  
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9 246 indicating that partially distributed corrosion had a significant influence on the failure  
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11 247 mode of the specimens.  
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### 17 249 *3.2. Mechanical properties*

#### 20 250 *3.2.1. Stress-strain curves*

22 251 The stress, strain, and related values hereinafter were all calculated based on the  
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25 252 dimensions of the specimens before corrosion. The nominal stress-strain curves  
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28 253 obtained from tensile coupon tests are plotted in Fig. 11. A noticeable difference was  
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31 254 observed from curves of the specimens after exposure to 500 °C and 900 °C. An  
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34 255 apparent yield platform was found in the stress-strain curves of the specimens cooled  
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36 256 from 500 °C to the ambient temperature, but not in those cooled from 900 °C.  
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39 257 Therefore, the yield stress was defined as the lower yield point for the specimens after  
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42 258 treatment of 500 °C, and 0.2% proof strength (stress at a strain of 0.2%) for the  
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45 259 specimens after treatment of 900 °C, respectively. Generally, the strength of the  
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48 260 specimens was significantly enhanced after exposure to 900 °C followed by water  
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51 261 cooling, but a considerable reduction was observed in terms of the ultimate strain. The  
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53 262 ultimate strength of the specimens after exposure to 900 °C was 1.7 – 2.5 times of that  
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56 263 after 500 °C. Nevertheless, the ultimate strain of the specimens cooled down from 900 °C  
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59 264 was only 20%-50% of that from 500 °C. It is also observed that the post-fire elastic  
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