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## **CRITICAL CURRENT MEASUREMENTS ON Nb<sub>3</sub>Sn CONDUCTORS FOR THE NED PROJECT**

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### **Abstract**

The Next European Dipole (NED) project is a EU funded program aiming at the realization of a 15 T accelerator-type of Nb<sub>3</sub>Sn dipole magnet. The objective of NED phase one is developing high performance Nb<sub>3</sub>Sn conductor in collaboration with European industry. The envisaged Rutherford type of cable consists of about 40 superconducting strands exhibiting a non-copper critical current density of 1500 A/mm<sup>2</sup> at 4.2 K and 15 T. In the frame of conductor development, adequate critical current measurements in terms of accuracy and reproducibility should be ensured to qualify the ambitious strands produced. Therefore, a NED working group focuses at the establishment and implementation of appropriate and standardized procedures for sample preparation and mounting, critical current measurement, data analysis and reporting. A cross-calibration program between the institutes involved has been initiated by exchanging samples of Nb<sub>3</sub>Sn conductor from a single billet for critical current measurements, prepared at each institute. The experimental results of this program and the conclusive procedures for NED critical current measurements are presented and discussed.

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**Index Terms**—Accelerator magnets, superconducting filaments wires, Niobium compounds

## I. INTRODUCTION

The objective of NED phase one is developing high performance Nb<sub>3</sub>Sn conductor in collaboration with European industry. The envisaged NED Nb<sub>3</sub>Sn wires and cables are expected to generate a dipole field of 13-15 T at 4.3 K in a bore of about 88 mm [1]. Assuming a copper percentage in the wire cross section of 55% (Cu:non Cu ~ 1.2) the wires should exhibit a non-copper J<sub>c</sub> of about 1500 A/mm<sup>2</sup>@15T to obtain such a field. Assuming a 2-layer cosθ reference coil design the corresponding strand and cable requirements are summarized in Table I. Industrial development of monolithic conductors exhibiting such unprecedented properties is a major ambition of the NED

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strand		
strand diameter	1.25	mm
Cu:non Cu (λ)	1.15 < λ < 1.35	-
filament twist pitch	30	mm
geometric diameter Nb <sub>3</sub> Sn region in filament	< 50	μm
I <sub>c</sub> @ 4.2 K (10 μV/m criterion) at 12 T	> 1636	A
at 15 T	> 818	A
RRR after heat treatment at 675 °C	>150	-
Rutherford cable		
number of strands	40	
mid thickness	2.275	mm
width	26.00	mm
keystone angle	0.22	
allowed I <sub>c</sub> degradation due to cabling	10	%
I <sub>c</sub> @ 4.2 K (10 μV/m criterion) at 12 T	58.9	kA
at 15 T	29.4	kA
RRR after heat treatment at 675 °C	> 70	

program and calls for reliable, reproducible and unambiguous methods for the measurement of their electromagnetic properties I<sub>c</sub>(B), M(B) and RRR(B=0). A Working Group on Conductor Characterization (WGCC) with representatives from CERN, CEA-Saclay, INFN-Milano, INFN-Genova and the University of Twente is charged with the definition and development of standardized and certified procedures to measure the mentioned properties of virgin, deformed and extracted strands. In this paper the focus is on critical current measurements.

### A. The concept of 'critical current' for Nb<sub>3</sub>Sn conductors

By convention the critical current I<sub>c</sub> in a superconductor is the current where the longitudinal electrical field reaches the I<sub>c</sub> criterion E<sub>0</sub> under accurately known (B,T,ε)-conditions. For conductors like NbTi, for which I<sub>c</sub> is hardly sensitive to strain, above definition justifies a straightforward measurement of I<sub>c</sub>(B,T) ignoring the precise strain state. In contrast, the critical current of Nb<sub>3</sub>Sn and most ceramic HTC conductors shows a rather strong dependency on strain, principally excluding such an I<sub>c</sub> determination if exact knowledge of the strain state is lacking. However, I<sub>c</sub>(B,T,ε) experiments are expensive, lasting and difficult to perform and therefore unattractive as an experimental qualification procedure, especially if tenths of samples need verification [2], [3]. Besides, in tangible applications of Nb<sub>3</sub>Sn conductors precise knowledge about the strain state throughout the manufacturing process usually lacks or is hard to predict, especially without detailed knowledge of the (σ,ε)-relation throughout the whole process.

Above implies that the concept of ‘critical current’ of a strain sensitive superconductor only holds if general agreement exists on standardized experimental circumstances for a critical current measurement. Despite the valuable recommendations from the VAMAS and ITER programs, this is unfortunately still not the case [4]-[10]

### B. Conditions affecting $Nb_3Sn$ $I_c$ measurement

Table II summarizes the capabilities and experimental boundary conditions of the  $I_c$ -measurement facilities at the participating labs in the WGCC. For  $Nb_3Sn$  conductors the  $I_c$  as a function of field, temperature and strain can be fairly accurately predicted by applying phenomenological scaling relations [3]. At a constant strain and constant applied magnetic field the critical current normalized to its value at 4.20 K varies as a function of temperature nearly linearly from 15 %/K at 12 T to 25 %/K at 16 T. Assuming an uncertainty of the sample temperature of 30 mK, its influence on  $I_c$  amounts to 0.75 % maximum in this field range.

A similar linear dependency of the applied field on the critical current normalized to its value at 13 T and 4.2 K amounts to 20 %/T which varies only slightly at different normalization fields. With a typical field uncertainty at the sample of 20 mT the resulting error in  $I_c$  amounts to 0.4 %.

The strain state of the reacted  $Nb_3Sn$  in the composite sample wire is by far the most uncertain factor affecting the  $I_c$  rather severely. It is mainly determined by: 1) the tension during sample winding and wire fixation after winding, 2) the bending stress which is a function of the wire diameter and the radius of the sample holder, 3) the mechanical properties of the wire components and the sample holder during the heat treatment, 4) the temperature ramp to the reaction temperature at about 675 °C, 5) eventual sample transfer after the heat treatment to another holder and finally 6) the relative thermal contraction of the composite wire and the sample holder during heat treatment, soldering to the current leads and the cool-down to 4.2 K.

Though above factors depend also on the type of  $Nb_3Sn$  conductor, the strain response of the critical current at 12 T and 4.2 K for  $Nb_3Sn$  wire samples typically amounts to 140 %/% in terms of critical current normalized to the maximum in the  $I_c$  vs. strain curve [2], [3].

Even without knowing the final strain state of the sample a fair and sensible judgment of reproducibility and accuracy of  $I_c$  measurements on  $Nb_3Sn$  wires necessitates control of all above factors.

TABLE II  
CHARACTERISTICS  $I_c$  MEASUREMENT FACILITIES

	CEA	INFN	UT
maximum current (A)	1000	1500	1200
current uncertainty (%)	< 0.1	< 0.1	< 0.1
maximum field (T)	15	13.5	15
field uncertainty at sample (%)	0.2	0.2	0.1
magnet bore (mm)	50	60	80
sample temperature (K)	1.8 – 4.2	4.2	4.2
temperature uncertainty 4.22 K (mK)	30	30	30

## II. INTER-LABORATORY CROSS-CALIBRATION PROGRAM

Inspired by experiences of the VAMAS program and the successful development of standardized procedures for specifying and measuring the critical current of ITER conductors [10] the WGCC has performed a program of cross-calibration between the various test facilities with the purpose to: 1) reveal possible systematic differences of  $I_c(B)$  results from samples of identical wires prepared and measured at each lab, 2) get acquainted to preparation and measurement of such large, very high  $I_c$  samples shaped as round or flat rolled wire or as strand extracted from a Rutherford cable, 3) bring forward shortcomings or limitations in existing preparation or measuring procedures and agree on a roadmap to improvements, 4) enable sample exchange between the labs 5) set up procedures for dealing with conflicting results, 6) set up agreed and approved procedures for sample mounting, measuring procedures and data analysis and 7) develop standard reporting.

### A. Stage I:

As a first step each lab independently prepared a series of wire samples and measured the  $I_c(B)$  and n-value at 4.2 K (10  $\mu$ V/m criterion, n-value determined between 5 and 50  $\mu$ V/m) to the extend of their (I,B) capabilities. This first series comprised 2 samples each of a 1.06 mm LHC NbTi wire, a binary 1.26 mm  $Nb_3Sn$  wire and finally both a virgin and an extracted strand from a Rutherford cable made out of a binary 0.8 mm  $Nb_3Sn$  wire. Initially each laboratory performed heat treatment, sample preparation, measurements and report according to existing procedures. The results of this stage-1 inter-lab comparison are summarized in Table III, in which  $\Delta I_{max}$  is the maximum difference in  $I_c$  between the labs and  $\Delta I_{lab}$  is the maximum spread per lab for a single type of wire.

Regarding the uncertainty in temperature and field mentioned in Table II the differences in  $I_c$  of the NbTi LHC wire, referenced by very accurate measurement at CERN, are within the range of expectation. The differences for the virgin  $Nb_3Sn$  wires however are unacceptably large, most probably due to differences in sample strain as a result of differences in sample mounting and unguarded temperature evolution during the sample heat treatment. The relatively small differences for the extracted strands should be considered statistically normal but coincidentally small. Consequently a thorough comparison of sample holder lay-out, mounting and measuring procedures has been carried out.

TABLE III  
RESULTS  $I_c$  MEASUREMENTS STAGE I

	NbTi	$Nb_3Sn$ virgin	$Nb_3Sn$ virgin	$Nb_3Sn$ extracted
diameter (mm)	1.06	1.26	0.84	0.84
# samples per institute	2	2	2	1
$\Delta I_{max}$ (%)	8	15	12	4
$\Delta I_{lab}$ (%)	3	7	5	-
systematic trend		lab 3 > lab 1 > lab 2		

As a result the 3 labs agreed all to employ the standard grooved TiAlV sample holder and corresponding copper rings, referred to as the ITER barrel, and alter the cryogenic insert accordingly if necessary. After mounting and screw fixation to the copper rings the sample wire remains fixed to the sample holder during the heat treatment, soldering to the copper rings and cool-down to 4.2 K. The design of the copper rings is adapted such to enable sample exchange between the labs. Furthermore, reporting will include the monitored temperatures during the sample heat treatment and thermometers will be attached to the sample holder during the  $I_c$  measurements. At this stage details of mounting procedures have been discussed but no further restrictions on sample preparation have been imposed.

### B. Stage 2:

After the above mentioned adaptations 2 labs prepared 3 samples of the same 1.26 mm  $Nb_3Sn$  wire used in stage 1. Subsequently in stage 2-1 all labs performed an  $I_c$  measurement on all samples between 10 and 15 T at  $(4.22 \pm 0.03)$  K.

Fig. 1 displays the results of these measurements arranged per lab which prepared the samples (solid markers). A first remarkable observation is the large difference between the samples prepared at each lab in the order of 15 %. This could be an indication that the sample wire exhibits inhomogeneous properties along its length, which would make this wire rather useless as reference wire. Secondly,  $I_c$  changes after subsequent thermal cycles by 3% for samples prepared at lab 2 and 6 % for samples at lab 1. Though the direction is rather random, qualitatively this behavior has also been observed by Goodrich [8] which most conceivably results from stress redistribution along the sample wire.

Above results are still non-discriminative with respect to the measuring procedures. Therefore this stage has been extended with stage 2-2, in which 3 samples were measured again by all labs, indicated by the open markers in fig. 1. The  $I_c$  of sample NED10 shows a large decrease of about 125 A, but like the other re-tested samples NED13 and NED15 the results of the 3 labs agree within 2 %. Though after 3-4 thermal cycles full stress relaxation has been established in all 3 samples, even the absolute differences between NED13 and NED15, prepared at the same lab by the same person, still amounts to 7 % which again supports the suspicion of longitudinal in-homogeneity of the sample wire. An important conclusion at this stage however is that despite some differences in the measuring procedure, the relative differences between  $I_c$  values obtained at different labs have converged to an acceptable level of  $\pm 2\%$ .

Above results have lead to the general conclusion that a standard sample preparation prescribed in detail must be respected to reach the highest level of relative accuracy if  $I_c$  results of samples prepared at different labs are compared. The finally agreed detailed procedures, which resemble the ITER protocol to a high degree, are summarized in Table V. The major accepted differences concern the thickness of the

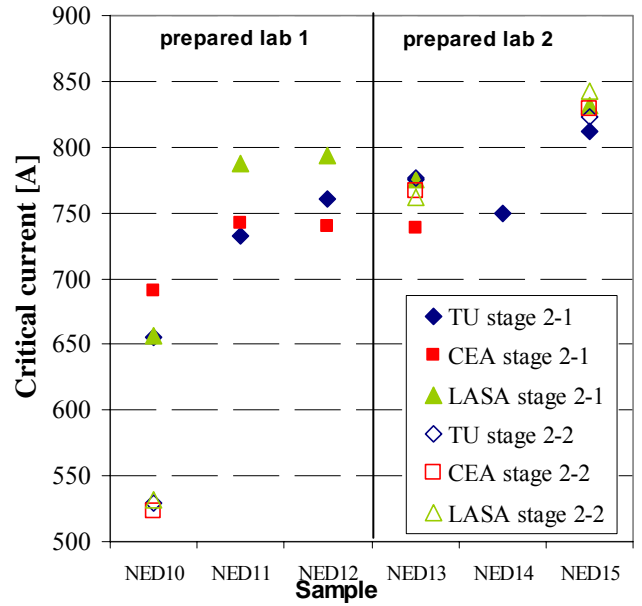


Fig. 1. Results of the  $I_c$  measurements at 12 T@4.22 K of stage 2-1 (solid markers) and stage 2-2 (open markers) on  $Nb_3Sn$  conductor samples prepared at 2 different labs. The legend indicates where the samples were measured.

TiAlV barrel (related to differences in both the fixation of the copper rings to the barrel and the connection to the current leads) and voltage monitoring during the transition either at static currents (after a waiting time of about 5 seconds after each current step) or dynamically during a slow current ramp.

### C. Stage 3:

After obeying strictly the described sample mounting procedures from Table V, the last stage 3 of this program comprised the  $I_c$  measurement of 2 samples of the previous 1.26 mm  $Nb_3Sn$  wire, a flat rolled (unsupported, single sided reduction of 0.35 mm) version of the same wire and a state-of-the-art PIT- $Nb_3Sn$  wire of 1 mm diameter, reaching a non-copper  $J_c$  at 12 T of about 2500 A/mm<sup>2</sup>. Flat rolling to a compaction comparable to what is obtained in a Rutherford cable may be a viable, efficient and controlled way to predict the possible  $I_c$  degradation of the same wire in a Rutherford cable [12]. The results in Table IV show, that for the last wire the restrictions in sample preparation indeed results in an acceptable qualification window of 2 % between the labs and less than 1 % at each lab. Though a decrease in  $I_c$  of about 40 % of the flat rolled wire are observed both  $\Delta I_{abs}$  max. and  $\Delta I_{lab}$  max. compare quite well to the differences observed for

TABLE IV  
RESULTS STAGE 3 MEASUREMENTS

	$Nb_3Sn$ virgin	$Nb_3Sn$ flat rolled	$Nb_3Sn$ virgin
diameter (mm)	1.26	0.91/>1.26	1.0
Cu:non Cu	1.25	1.25	0.82
# samples per institute	2	2	2
$\Delta I_{abs}$ max. (%)	8	4	2
$\Delta I_{lab}$ max. (%)	4	3	0.6
systematic trend		none	

the virgin wire. This confirms again the longitudinal inhomogeneity of this wire but strongly supports the convergence to the required accuracy for comparing  $I_c$  results from samples prepared and measured at different labs.

TABLE V  
NED SAMPLE MOUNTING AND  $I_c$  MEASUREMENT PROCEDURES

Before heat treatment	
TiAlV barrel dimensions ( $R_i \times R_o \times H$ ) (mm)	14.2 x 32 x 32.5 (CEA, INFN) 28 x 32 x 32 (UT)
TiAlV barrel oxidized in air at ~ 250 °C	yes
wire sample cleaned	alcohol
TiAlV barrel and copper rings cleaned	alcohol or ultrasonic soap
copper rings fixed (axially and azimuthally) to barrel	stainless steel screws (CEA, INFN), punched holes (UT)
slit in copper rings	yes (CEA), no (INFN, UT)
initial wire fixation to the first copper ring	screw tensioning the wire
wire winding tension ( $N/mm^2$ )	15
final wire fixation to second copper ring	screw tensioning the wire
During heat treatment	
reaction atmosphere	argon (CEA, INFN), vacuum (UT)
ramp to T (°C/h) unless otherwise prescribed	60
After heat treatment	
fixation copper rings to barrel replaced	yes (CEA, INFN), no (UT)
shunt Nb <sub>3</sub> Sn wire transition copper-barrel	no (CEA, INFN), yes (UT)
soldering heating sequence wire to copper solder	ring by ring PbSn (60/40)
soldering temperature (°C)	~190
cleaning after soldering copper-barrel	alcohol or ultrasonic soap
if wire movement possible, additional wire fixation (e.g. filled Stycast)	yes
curing after additional wire fixation	no (CEA, INFN), yes (UT)
gold plating of the flat contact copper rings to current leads	yes, (CEA), no (INFN), N/A (UT)
Mounting of voltage taps	
sample heating voltage tap soldering	PbSn (60/40)
voltage taps coverage from middle of barrel	3 and 5 turns (CEA)
	4 turns (INFN) 3,5 and 8 turns (UT)
overall voltage tap (including copper-barrel connection)	yes
fixation voltage wires to barrel	yes (CEA), no (INFN, UT)
thermometer mounted on sample holder	yes
sample completely uncovered by any insulating materials	no (CEA), yes (INFN, UT)
Measurement procedure	
field sequence	1 T steps $B_{high} \rightarrow B_{low}$
Lorentz force	inward
$I_c$ criterion	10 $\mu V/m$
n-determination	5 < E < 50 $\mu V/m$
voltage measurement	- during current ramp with 3 A/s (CEA, INFN),
	- at static current (UT)

### III. CONCLUSION

As part of the European NED project a 3-stage iterative inter-lab comparison program between 3 European labs has been carried out with the purpose to obtain an experimental protocol ensuring reproducible critical current measurements of thick, high current Nb<sub>3</sub>Sn conductors. Provided the sample wire is longitudinally homogeneous, this program shows that if a detailed protocol -which closely resembles the ITER protocol- on sample mounting, heat treatment and measuring procedures is respected, the inter-lab differences in  $I_c$  results of samples prepared and measured at different laboratories can be reduced to 2 % maximum, whereas the intra-lab reproducibility should be better than 1 %. Though the obtained values for the critical current lack absolute meaning due to uncertainty of the actual strain state, this relative accuracy is adequate to judge and compare the performance of industrially produced Nb<sub>3</sub>Sn conductors.

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