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INFLUENCE OF MICRO GLASS BEADS ADDED IN A PBT MATRIX ON THE MECHANICAL PROPERTIES OF COMPOSITES

Mihail BOTAN, Constantin GEORGESCU, Lorena DELEANU

"Dunarea de Jos" University of Galati emails: mihai.botan@ugal.ro, constantin.georgescu@ugal.ro

ABSTRACT

This paper presents the influence of adding materials (micros glass spheres) in a matrix of PBT on several mechanical properties: elasticity modulus, stress at break, elongation at break, energy at break.

The mechanical properties of these composites depend on the glass beads concentration. The additivation of polybuthylen therephtalate with glass beads increases the values of the elasticity modulus, but reduces quite drastically the elongation at break. Using the SEM investigation, the authors pointed out the particular aspects of fracture surfaces: a ductile process at sample margins and a brittle one in the middle of the composite.

KEYWORDS: PBT, glass beads, composite, elasticity modulus, stress at break, elongation at break, energy at break

1. Introduction

It is a basic problem with the tests that the result will vary with the test piece geometry and the test conditions and it may not be easy to extrapolate to different conditions [7-9].

Even if there are many standards related to shortterm tensile testing, they endeavor to quantify a number of specific characteristics which relate to the strength and deformation of a material. Knowledge of these characteristics can supply the designers with the potential performances of a material and a reliable basis of comparing materials [8, 19].

The term 'short term mechanical tests' is used as a convenience to describe mechanical properties where the effects of long times and cycling are ignored. This group of tests includes hardness, tensile, compression, shear, flexing, impact and tear. The material properties require to be rated for generating design data, for quality controlling, for predicting their durability and, unfortunately but necessarily for investigating failures. The polymers have a complex behavior thus, more than any material, they have to be evaluated in a useful way by particular or adequately adapted methods [8].

Polymeric materials are non-linear and their stress-strain characteristic is not linear and hence their modulus is not a constant [1, 2, 8, 9].

The use of PBT as an engineering material is a consequence of a balance of good properties rather than of a few outstanding ones. It does not possess the toughness of polycarbonate (PC), the abrasion resistance of an aliphatic polyamide, the heat resistance of a polysulphone, polyphenylene sulphide (PPS) or polyketone, the low water absorption of a modified poly(p-phenylene oxide) (PPO).

However, PBT, when it is suitably modified by, for example, glass fiber or fire retardants, can contribute to produce very useful compounds.

The particular characteristics of PBT, as pointed out by the suppliers [9, 26, 27], include:

- high softening temperatures (especially for composites with glass or carbon fibers, challenging the use of PC and modified PPOs);

- high rigidity, exceeded only by PPS among the engineering thermoplastics;

- good electrical insulation properties for an engineering thermoplastic as compared to PC, modified PPOs, PPS and the polyether imides;

- low friction and good abrasion resistance;

- good impact strength at low temperatures and excellent creep rupture strength;

- low water absorption and good chemical resistance, including resistance to stress cracking;



- good dimensional stability, a consequence of the low water absorption but also because of a low coefficient of thermal expansion;

- good moldability (easy flow and rapid setting).

A large number of PBT grades is available, including unreinforced, glass- and carbon-fiber reinforced, mineral filler reinforced, impact modified, elastomer modified, flame retardant and various combinations of the foregoing [9, 15, 16, 24, 27, 28].

2. Glass beads as adding material in a polymeric matrix

C type glass is adequate for using in chemically aggressive environments, especially acids. This outstanding chemical resistance is a consequence of its composition, typically as following: at least 60% SiO_2 , Li and Na oxides, Yn and Ca oxides and also rare earth oxides with small amounts of Al_2O_3 , B_2O_3 , Fe_2O_3 , TiO_2 , MnO and SnO_2 [17, 18].

Adding this type of reinforcing material influences the technological properties [5, 6, 10, 25] the mechanical ones [3, 4, 11-13, 22] and the tribological ones [14, 20]. In 2010, Akiyama et al. [2] reported a modifying tendency of the mechanical properties of the composites with ceramic particles (average diameter of 150 μ m) a little greater than the glass beads introduced in PBT for this study and more ragged as compared to almost spherical shape of the glass beads (Fig. 5). Adding 60% of ceramic particles in PBT makes the elastic modulus to increase to 7500 MPa, 2.5 times greater than pure PBT, but the tensile at break decreases with 10...12% and Vickers hardness has a double value.



Fig. 1. Damaging models of the interface in composites with spherical particles [21]

Medadd and Fisa [21] have proposed a damaging model of the interface when samples were loaded with tensile forces (Fig. 1) that proved to be suitable to explain the PBT + glass beads composite behavior in a qualitative way. Break at traction of a composite with spherical particles depends very much on the chemical and mechanical nature of the interfaces between the polymer and the hard particles:

a) the resistance interface is not damaged under loading and the break is initially developed in the composite matrix;

b) an interface partially damaged, usually dependent on the elasticity modulus, the volume fraction of the adding material without damaged



interfaces and the complementary volume fraction of the adding material that is characterized by a damaged interface (the ratio between these two interface categories being very hard to be estimated);

c) a wick defective interface that is easy to be destroyed when applying the load.

The difficulty is that the sample made of such a composite is loaded, all the above-described processes could occur, with different contribution to the final fracture.

Several research works reported that even if the glass beads bearing different treatment, the mechanical properties do not have significant modifications (Fig. 1.7) [21].



b) treated by silan (SiH₄)





Dekkers and Heikens [12, 13] noticed that, for polymeric composites, the band forming mechanism at traction is fundamentally different, depending on the treatment applied to glass beads. After analyzing



a). a good adherence between matrix and glass beads, using γ -aminopropil silan

the stresses, they concluded that the generation of the bands had occurred in the zones characterized by maximum values for the main shear stress and by maximum values of the strain energy (Fig. 3).



b). without adherence between glass bead and the polymer matrix, as obtained with silicon oil

Fig. 3. Tensile fracture zones of the PC + GB composite [13]

3. Testing methodology materials

The samples were obtained by extruding the mixtures of granulated PBT and glass beads (Fig. 5), in three mass concentrations (Table 1), at ICEFS Savinesti, Romania. The polymer had been dried up at a temperature of 100°C within two hours. There were obtained bone samples with the geometry and dimensions given in Figure 4 and a thermal treatment was applied to the bone samples, as recommended by the producer [27].

The initial distance between the gauge marks on the central part of the test specimen was of 115 mm, the rate of separation of the grips of the testing machine during test 5 mm/min (the speed of testing) and the it was calculated the tensile Stress (engineering), that is the tensile force per unit area of the original cross section within the gauge length, carried by the specimen at any given moment.

The here-presented results include data for at least 5 tensile tests and the average value for each mechanical characteristic was calculated and there also were presented the scattering intervals for each one. Stress was calculated with reference to the initial values of the cross section thus, there are given the engineering stress-strain curves for each material.



Fig. 4. Bone sample Type 1A ISO 527-2

Table 1. Materials tested

Material symbol	Composition (%wt)				
PBT	grad	neat polymer, le Crastin 6130 NC010 [10]			
GB10	10% GB	glass bead + 1.52% PA + 1%			
GB20	20% GB	black carbon,			
GB30	30% GB	for technological reason			



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Fig. 5. Glass beads before being mixed with PBT

The traction tests were done with the help of the universal testing machine TESTOMETRIC M350-5AT, having a force cell of 5kN, as recommended by EN ISO 527-2 [25], in the Laboratory of Polymeric Materials Research (Faculty of Mechanical Engineering, "Dunarea de Jos" University of Galati).

The values for the mechanical properties were calculated as the average of five tests that the authors consider to have no anomalous features and in accordance with the literature [9, 24, 27, 28].

4. Experimental results

For PBT, the stress-strain curves (Fig. 7a) have the same aspect as presented in [4, 5, 16, 24]; it was noticed a typical creep zone for the thermoplastic polymers, as this polymer could be included in the class of tough materials with a yield stress lower than the failure stress [8]. The composites have the shape of the stress-strain curves typical for brittle materials (Fig. 7b, c and d).

The authors applied Einsteins's model for having a dependency of the mechanical properties on the mass concentration of glass beads,

$$E_c = E_m \left(l + V_f \right), \tag{1}$$

where E_m is the elasticity modulus of the matrix and V_f is the volumic fraction of the adding material. For the composite with 10% GB, the value of the

elasticity modulus as experimentally determined is greater with 10.8% as compared to the value given by Eq. (1) and for the composite with 20% GB, the experimentally obtained value is greater with 21.3% as compared to the theoretical value given by the same model of mixture. The elasticity modulus and the density for glass beads were taken from literature [13, 21]: E_I =70000 MPa, ρ_f = 1.6g/cm³.

Analyzing Fig. 3.4, the following remarks may be made:

- the elasticity modulus of these composites with PBT matrix increases almost linearly with the massic concentration of GB;

- stress at break for the composite with 10% GB is just 10% higher than the polymer but the composites with 10% GB and 20% GB have lower values, 85% and 63%, respectively from the values exhibits by the polymer (Fig. 6b);

- elongation at break decreases with \sim 72% for PBT + 10% GB and with \sim 92% for the composite PBT + 20% GB, as compared to the value obtained for PBT.

The composites have reduced elongation at break (Table 2 and Fig. 6c) and none of the samples not presented the typical bottle neck shape characterizing the polymer.

There is a clear tendency of decreasing the value for the strength limit only for the composite with 20% and 30% GB, respectively.

Table 2. Average values of several mechanical properties for the tested materials

Charactoristic	Material				
Characteristic	PBT	GB10	GB20	GB30	
Elasticity modulus , E [N/mm ²]	1923.458	2358.356	2848.581	4087.458	
Stress at break, $\sigma_r [N/mm^2]$	41.571	40.265	36.543	25.795	
Elongation at break, Er [mm]	9.404	2.609	0.763	0.411	
Energy at break, [N m]	17.665	2.207	0.701	0.294	



Fig. 6. Average values and scattering ranges for the discussed mechanical characteristics of the tested materials





curves stress-straina) set of five samples, after being testedFig. 7. Stress-strain curve (left) for PBT and tested samples (right)





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Fig. 8. The stress-strain curves for the *PBT*+*GB* composite (left) and the sets of samples after break (right)

Material symbol	Relation	Correlation coefficient	Standard error about the line
PBT	$ y = -0.741 + 50.492 x + 57.331 x^{2} - 124.096 x^{3} + 89.270 x^{4} 34.749 x^{5} + 8.185 x^{6} - 1.201 x^{7} + 0.107 x^{8} - 0.005 x^{9} + 0.0001 x^{10} $	0.999	0.376
	$y = \frac{a \cdot b + c \cdot x^{d}}{b + x^{d}} $ (Fig. 4) a = 2.886, b = 0.1674, c = 50.597, d = 2.255	0.996	0.667
GB10	y=-0.356+82.171x - 36.466x ²	0.999	0.304
GB20	$y=-0.871+93.729x-57.723x^2$	0.999	0.396
GB30	$y = 1.513x + 119.147 - 88.929x^2$	0.993	0.701

Table 2. Data upon the mathematical models attached to the experimental data

SEM images show that the damaging process of the interface has an intermittent nature (Fig. 8b reveals a bigger glass bead that was discontinuously detached from the matrix). For the tested composites, the 1% PA, even small, could influence the interface resistance, having a greater ductility and adherence to



a). Ductile (at the sample margin - right) and brittle (left) aspects of the fracture section

the glass beads as compared to PBT. Analyzing the SEM images from a small scale to larger ones, one may notice the existence of two types of surface aspect. At the sample margins the fracture surface has a ductile nature but towards the center of the sample the aspect becomes brittle (Figs. 7 and 8).



b). Intermittent detaching of glass beads in the middle of the fracture section

Fig. 9. Typical aspects of the fracture section for samples made of the PBT + 10% GB composite



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a). Zone of ductile fracture



b). Middle zone of the fracture, with brittle aspect

Fig. 10. Typical aspects for tensile break surface of the PBT + 20% GB composite

4. Conclusions

The additiveation of polybuthylen therephtalate with glass beads increases the values of the elasticity modulus but reduces quite drastically the elongation at break. The results obtained point out the importance of testing polymers and their composites.

The authors elaborated mathematical models for each tested material. The mathematical models could be useful in analyses with finite elements in the first step evaluation of a design.

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