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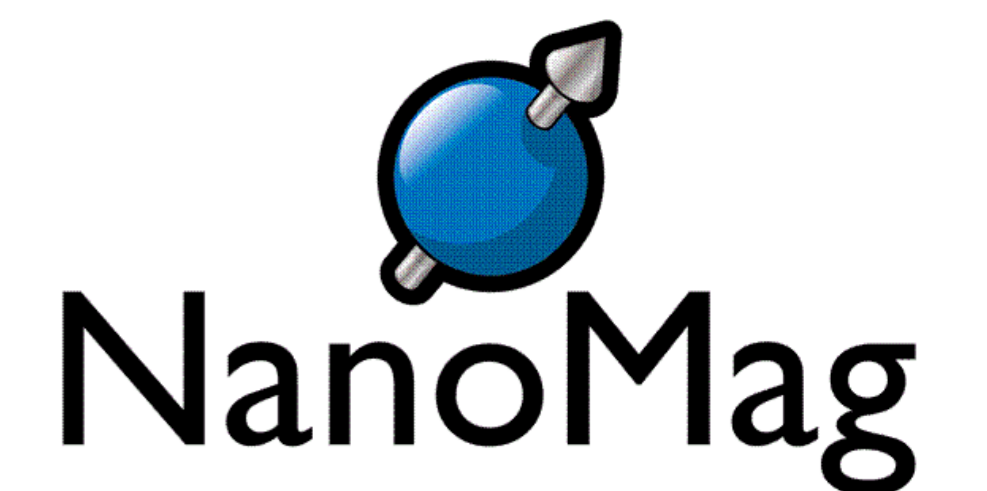
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Mössbauer, SANS and magnetic characterization of interacting iron oxide nanoparticles (IONPs)



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Introduction

Magnetization behavior of ensembles of Iron Oxide Nanoparticles (IONPs) primarily depends on their structural as well as magnetic properties + interparticle interactions.

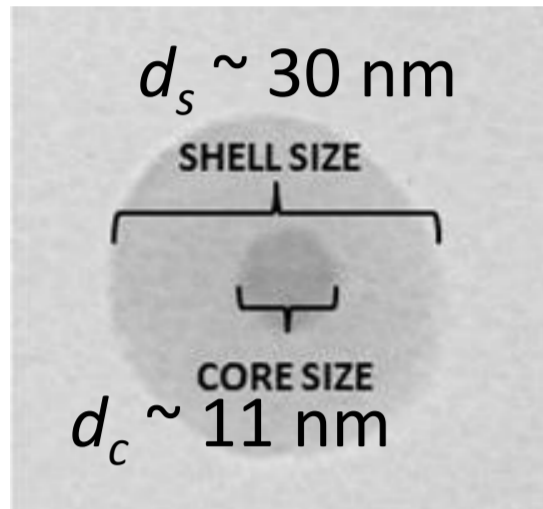
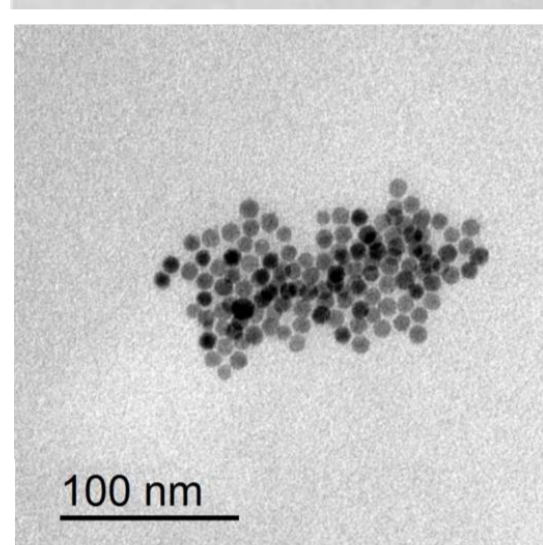
Presence of dipolar interactions has significant implications in biomedical applications [1].

Standardizing the determination of magnetic parameters (e.g. chemical composition, magnetic moment distribution, dipolar interactions) driving those applications is a need for the future and goal of the **NanoMag** project.

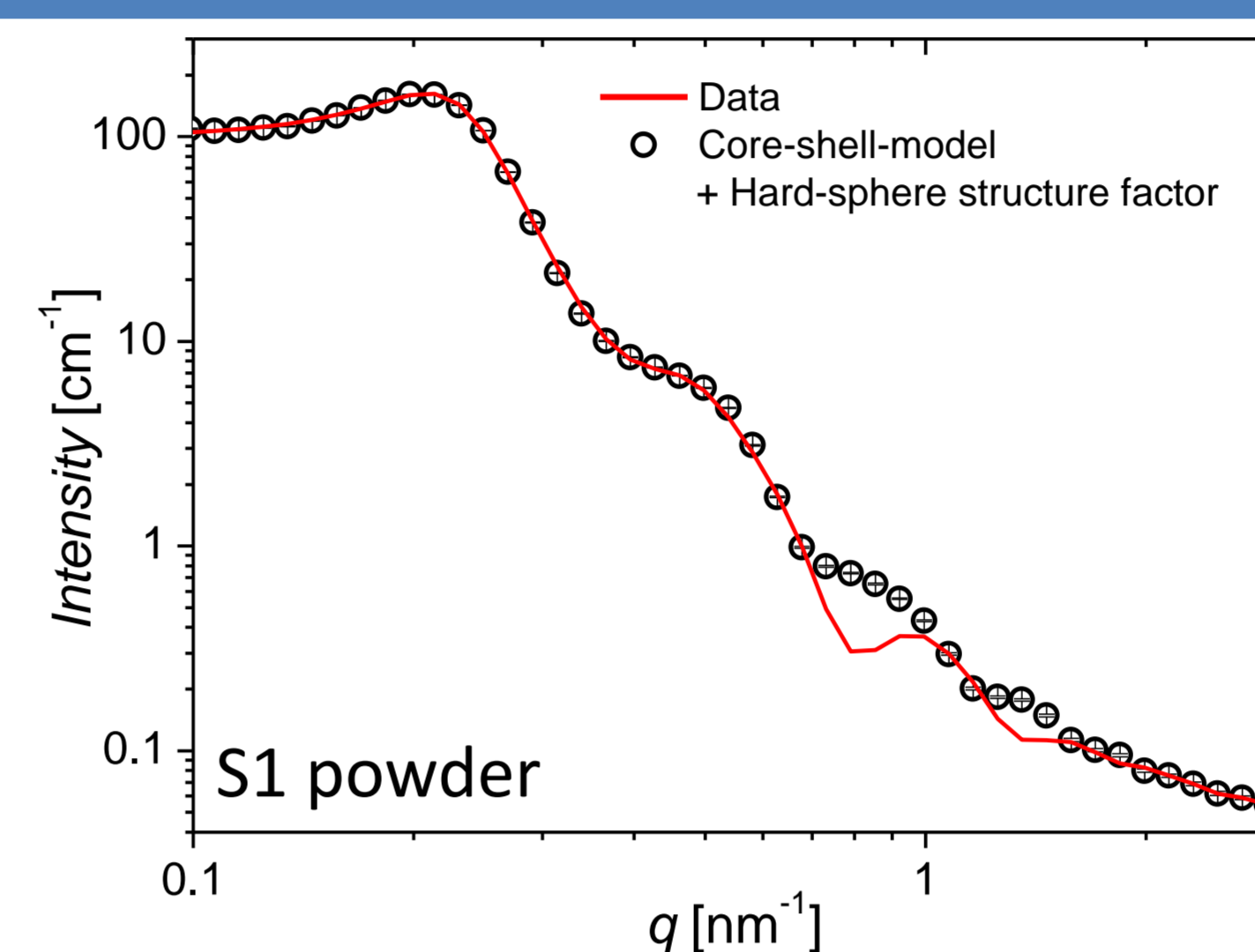
[EC FP7 NMP project, grant no. 604448]

Current study: IONPs with different amounts of dipolar interactions are characterized by Mössbauer, SANS (structural properties) and combined M(H)/ZFC-FC-measurements (magnetic properties) to detect the influence of dipolar interactions on their magnetization behavior.

Samples

- S1**
- 
- IONPs surrounded with rigid silica shell.
 $d_c = 11 \text{ nm}$, $\sigma_{dc} = 0.7 \text{ nm}$
 - Shell prevents agglomeration of IONPs.
→ Shell thickness determined by SANS.
- S2**
- 
- Comparable d_c as S1 but agglomerates.
 $d_c = 10 \text{ nm}$, $\sigma_{dc} = 0.7 \text{ nm}$
 - On average dipolar interactions should be higher than in S1 due to agglomerates.

Results: SANS



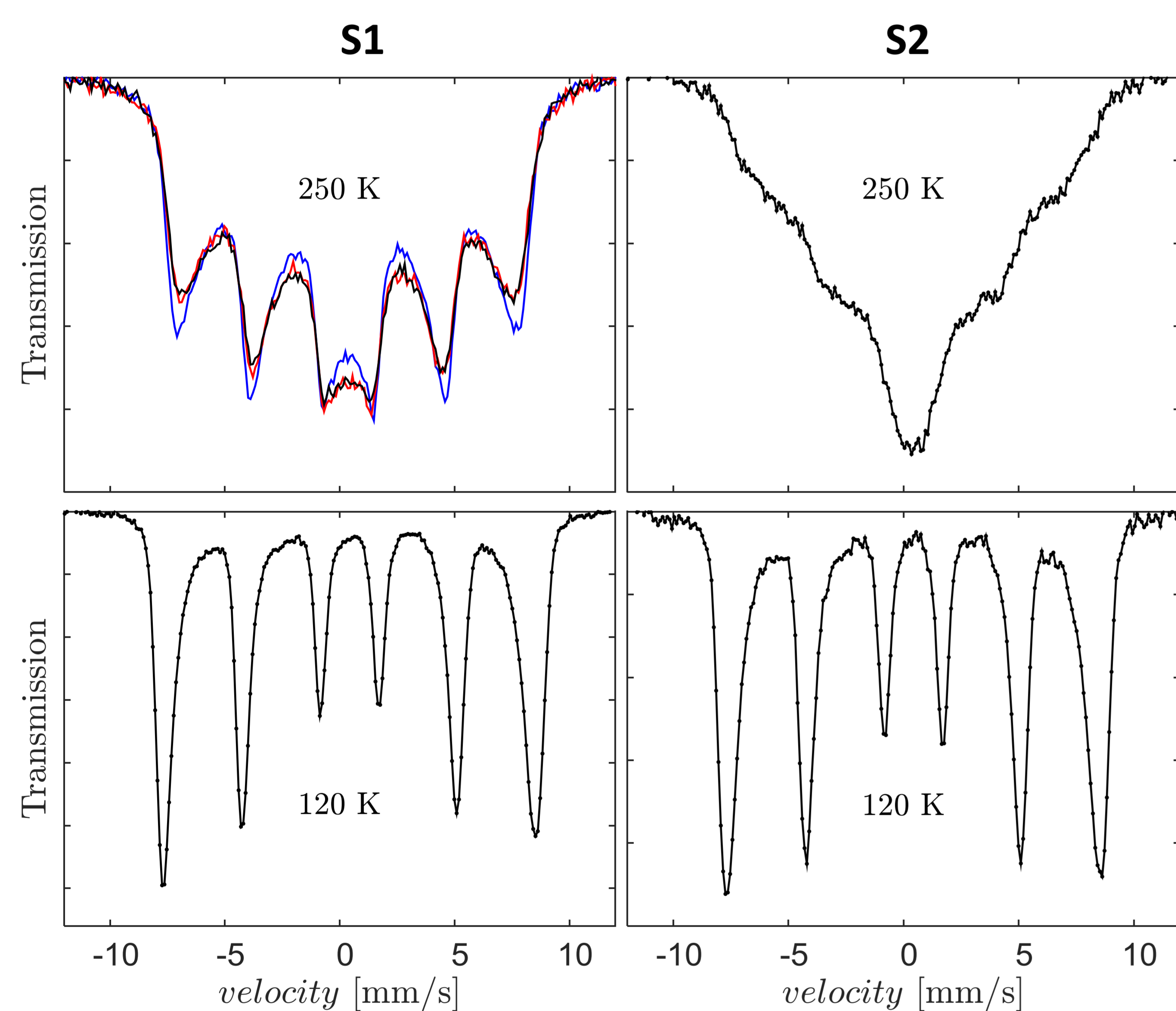
Adjusting scattering intensity with core-shell-model [2] enables exact determination of core (d_c) and shell (d_s) size distribution of S1:

$$d_c = 11 \text{ nm}, \sigma_{dc} = 1 \text{ nm}$$

$$d_s = 26 \text{ nm}, \sigma_{ds} = 7 \text{ nm}$$

Distance between magnetic cores in S1 is large enough so that dipolar interactions should be negligible!

Results: Mössbauer



i) Influence of sample preparation

S1: Mössbauer data of dispersion (black), powder (red) and gel (blue) at 250 K show no influence of sample preparation.

ii) Iron oxide composition

Maghemite/magnetite composition analysis using the mean isomer shift [3] corrected for 2nd order Doppler shift gives that main iron oxide is maghemite (at% Fe in magnetite: 25(5) for S1 and 0(5) for S2).

iii) Relaxation – size effect / dipolar interactions

S1: Using $K = 9.05 \text{ kJ/m}^3$, $V = \pi/6 \cdot d_c^3$, $\tau_0 = 1 \text{ ns}$ and a measuring time of $\tau \approx 5 \text{ ns}$, we estimate a blocking temperature $T_B = KV/[k_B \ln(\tau/\tau_0)] \approx 300 \text{ K}$. Experimentally, we find $T_B \approx 300 \text{ K}$ (data not shown) consistent with the simple estimate. A more accurate analysis is topic for future work

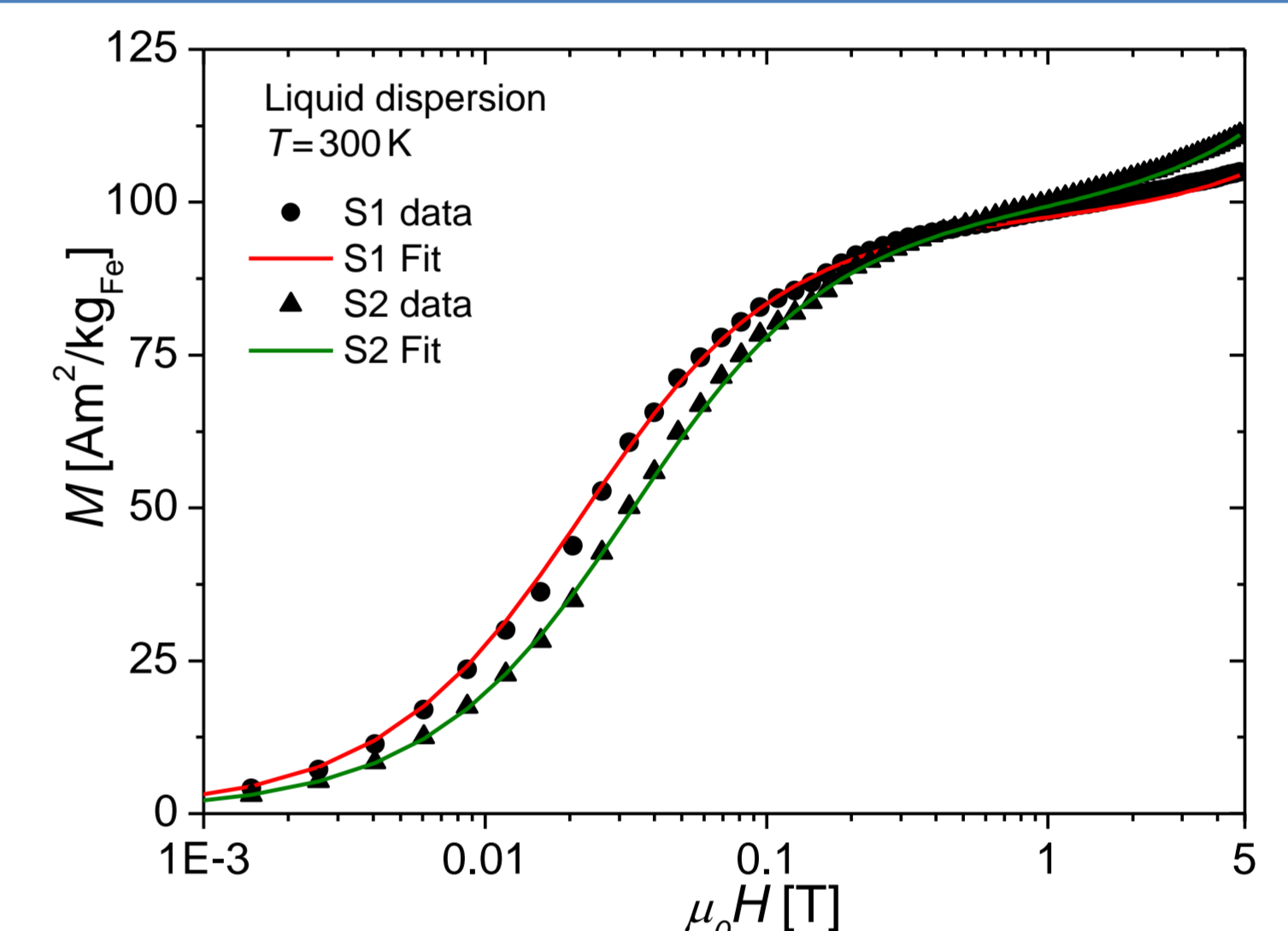
S2: We observe a lower blocking temperature than for S1, possible due to the slightly smaller size or due to **dipolar interactions** [4,5].

Results: Magnetic characterization

i) M(H) curves of dispersions

Fit with Langevin function, assuming a normal distribution of the core diameters d_c .

Fit results	S1	S2
d_c [nm]	11.1(1)	9.7(1)
σ_{dc} [nm]	2.2(1)	2.1(2)
M_S [$\text{Am}^2/\text{kg}_{\text{Fe}}$]	98.5(1)	99.5(1)

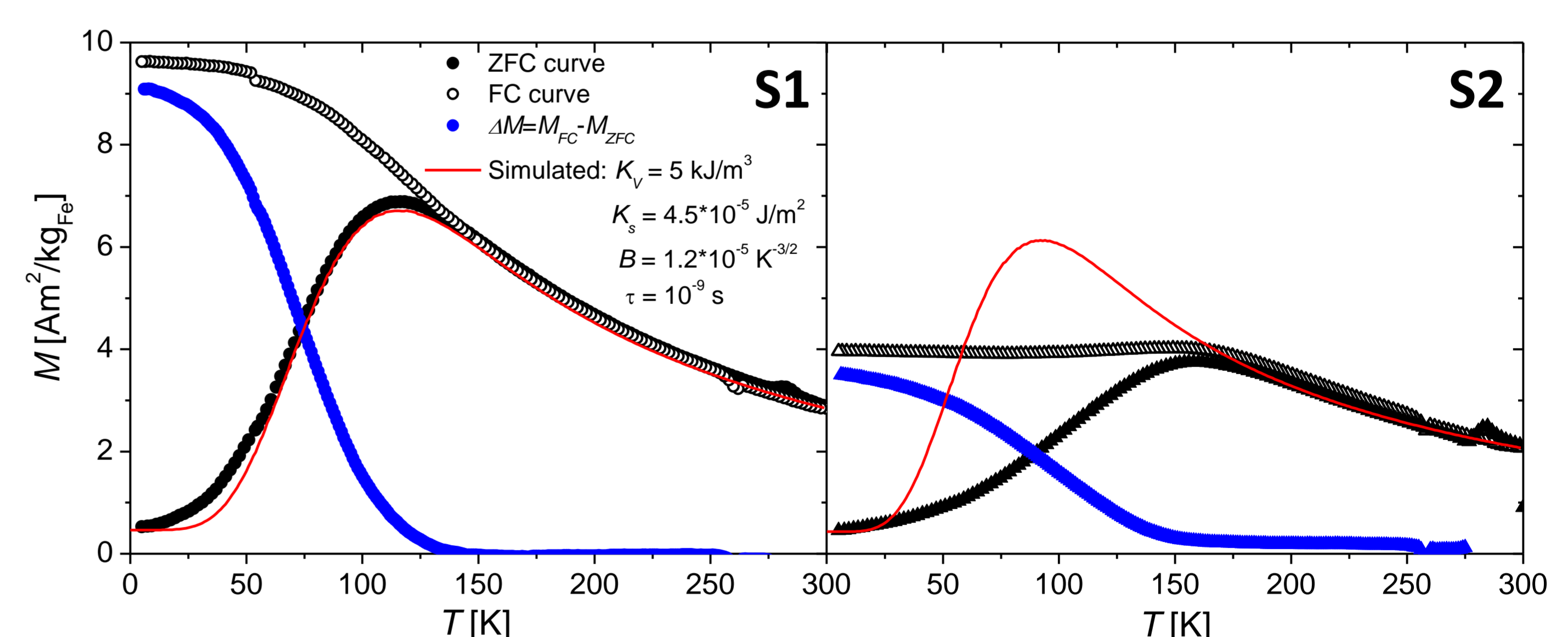


Reduced M_S values can be attributed to spin canting at **particle surface** [6].

ii) Simulation of ZFC curves including d_c distributions [7]

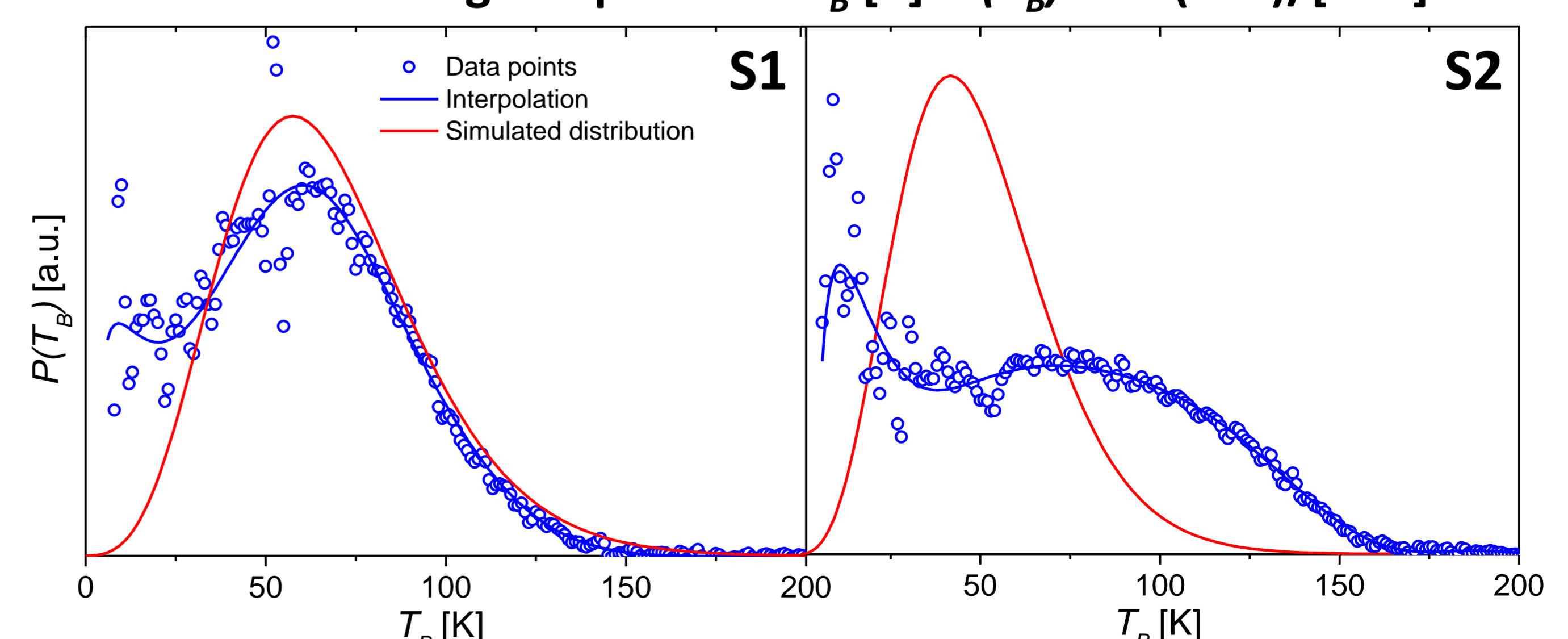
With: $M_S(T) = M_0(1 - BT^{3/2})$ [6], $K = K_V + 6K_S/d_c$ [8]

K_V : Anisotropy constant of particle volume, K_S : surface anisotropy constant



S1: Curve can be simulated by introducing a **surface anisotropy** constant K_S .
S2: Curve is shifted probably due to **dipolar interactions** [9].

iii) Distribution of blocking temperature T_B [7]: $P(T_B) \propto -d(\Delta M)/[TdT]$



S2: $P(T_B)$ is highly distorted indicating significant **dipolar interactions**.