Chapter 18 Moisture Stable Soot Coated Methylammonium Lead Iodide Perovskite Photoelectrodes for Hydrogen Production in Water

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Abstract Metal halide perovskites have triggered a quantum leap in the photovoltaic technology marked by a humongous improvement in the device performance in a matter of just a few years. Despite their promising optoelectronic properties, their use in the photovoltaic sector remains restricted due to their inherent instability towards moisture. Here, we report a simple, cost-effective and highly efficient protection strategy that enables their use as photoelectrodes for photoelectrochemical hydrogen production while being immersed in water. A uniform coating of candle soot and silica is developed as an efficient hydrophobic coating that protects the perovskite from water while allowing the photogenerated electrons to reach the counter electrode. We achieve remarkable stability with photocurrent density above 1.5 mA cm⁻² at 1 V versus saturated calomel electrode (SCE) for ~1 h under constant illumination. These results indicate an efficient route for the development of stable perovskite photoelectrodes for solar water splitting.

Keywords Photoelectrochemical water splitting \cdot Halide perovskites \cdot Hydrogen \cdot TiO₂ \cdot CH₃NH₃PbI₃

18.1 Introduction

The world has entered into an era where climate change has become an unfortunate reality [1]. Any further increase in the CO_2 levels might push the Earth's climate over a tipping point from which recovery would become impossible [2]. The unprecedented pace at which global warming is damaging the earth is evident from the warming of the oceans and the accelerated melting of the polar ice caps all of which point to the increasing level of the greenhouse gases [3]. There is thus an urgent need for the world

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to shift from the conventional carbon-based fuels to a renewable energy source to tackle global warming while also fulfilling the global energy demands. In this regard, hydrogen can be seen as the best alternative energy source as it is absolutely clean (C/H ratio is 0), freely available (most abundant element), has a very high energy per unit mass (1 kg of hydrogen provides the same power as 3.78 L of gasoline) and has a high energy storage capacity per mole [4]. Currently, hydrogen production mainly relies on methane steam reforming and electrolysis both of which cannot be considered environment-friendly and thus defeats the purpose of producing hydrogen [5]. Amongst renewable methods of hydrogen production, the photoelectrochemical splitting of water is considered to be the most promising method which uses semiconductors submerged in aqueous electrolyte as photo-absorbers for sunlight driven splitting of water [6]. However, the development of inexpensive and moisture stable visible light-absorbing materials is still a challenging task [7]. Hybrid Organometal halide perovskites offer great promise as photoelectrode materials thanks to their low cost, easy processability, high efficiency [8], tunable band-gap [9], ambipolar charge transport [10], long carrier lifetimes [11], and long charge diffusion lengths [12]. The immense potential of these materials is evident from the rapid improvement in their power conversion efficiencies from over 3.8% in 2009 to values reaching over 23% in 2018 making it the fastest growing technology in the history of photovoltaics [13]. Despite the tremendous promise shown by these materials, their wide-scale commercialisation is restricted owing to one major drawback; they are inherently unstable in water. The lattice of methylammonium lead iodide perovskite degrades even with the slightest exposure to moisture, followed by the decomposition of the material into PbI₂ [14]. Several attempts have been made to improve the stability of halide perovskites towards moisture, including protective surface coatings, use of more hydrophobic alkyl ammonium salts, atomic layer deposition or through 2D/3D hybrid structures [15]. These protection strategies are not only expensive but also fail to impart long term stability to the perovskite photoelectrodes and therefore are difficult to upscale.

Here we report a simple, cost-effective and efficient protection strategy to enable the perovskite photoelectrodes to produce hydrogen while being submerged in an aqueous electrolyte. A thin layer of silica over a uniform layer of candle soot is used to effectively protect the moisture-sensitive CH₃NH₃PbI₃ (referred to as MAPbI₃) perovskite layer from degradation in water. The combination of carbon and silica served as an effective hydrophobic layer shielding the photoelectrodes from water. The m-TiO₂IMAPbI₃IClSiO₂ photoanodes operate in water, exhibiting photocurrents of over 1.8 mA/cm² at an applied bias of 1 V in an alkaline solution (pH 13). Even more impressively, these photoelectrodes exhibit remarkable stability showing no considerable change in the current for over 1 h under constant illumination.

18.2 Experimental

18.2.1 Synthesis of Methylammonium Lead Iodide

30 mL Hydroiodic acid (57 wt.% in water, Aldrich) was added dropwise to 27.86 mL methylamine (40% in methanol, Aldrich) in a 250 mL round-bottomed flask and was subjected to constant magnetic stirring at 0 °C for 2 h. Crystallization was done by evaporating the solvents at 50 °C for 1 h. The obtained powder was purified by washing three times with diethyl ether, followed by drying at 60 °C in a vacuum oven for 24 h. The product, methylammonium iodide was saved at room temperature in a desiccator [16].

18.2.2 Solar Cell Fabrication

A compact layer of TiO₂ as an electron transport layer (ETM) was deposited onto the FTO substrates by spin coating a solution containing 0.65 ml of Ti (IV) isopropoxide (Sigma Aldrich, 97%), 0.38 ml of Acetylacetone (Sigma Aldrich, reagent grade) and 5 ml of Ethanol at 3000 rpm for 60 s. The substrates were then calcined at 500 °C for 30 min to obtain the dense TiO₂ layer. The perovskite precursor solution was prepared by mixing CH₃NH₃PbI₃ and PbI₂ (1:1 mol ratio) in γ -butyrolactone at 60 °C for 12 h with constant magnetic stirring. The resulting solution was coated onto the FTO/TiO₂ substrate at 2,000 rpm for 60 s then at 3,000 rpm for 60 s, and dried on a hot plate at 100 °C for 2 min [16]. A uniform layer of soot was immediately deposited onto the FTO/TiO₂/MAPbI₃ substrate with the help of a burning candle, followed by the spin-coating of a uniform layer of SiO₂ (0.1 g SiO₂ in 2 mL γ -butyrolactone) over it. These FTO/TiO₂/MAPbI₃/C/SiO₂ substrates (see Fig. 18.1) were developed into photoelectrodes by establishing ohmic electrical contacts using





silver paste and copper wire from the uncoated area of the conductive substrates. The area of contact was then covered with non-transparent and non-conducting epoxy-resin (Hysol, Singapore).

18.3 Results and Discussion

18.3.1 Analysis of the Crystal Structure

Figure 18.2 shows the XRD pattern of the soot coated perovskite thin film. The peaks at 14.08°, 28.36°, 31.76°, and 43.08° correspond to the tetragonal/ β phase of MAPbI₃ oriented along the (110), (220), (310,) and (330) planes respectively (JCPDS 00-021-1276). The peaks at 25.304°, 37.793° and 48.037° correspond to the (101), (004), and (200) planes respectively of the tetragonal phase of TiO₂ (JCPDS 65-5714). The peaks at 20 values of 15.194°, 16.966° and 26.544° correspond to the (110), (003) and (300) planes respectively of carbon (JCPDS 50-0927), while those at 10.892° and 40.265° correspond to the (101) and (005) planes respectively of SiO₂ (JCPDS 771414). All other peaks correspond to the FTO substrate. The XRD pattern reveals that both MAPbI₃ and TiO₂ are obtained in a highly crystalline state while also indicating the presence of carbon and SiO₂. Moreover, the coating of soot over the perovskite layer does not alter its crystalline structure suggesting that carbon did not incorporate into the crystal lattice of the perovskite.



Fig. 18.2 XRD pattern of the FTO/TiO2/MAPbI3/C/SiO2 photoelectrode

18.3.2 Optical Measurements

The optical absorbance measurements were made using a UV-Visible Spectrophotometer (Shimadzu, UV-2450, Japan) in the wavelength range of 800 to 200 nm using a UV quartz sample cell with a transmission range of 190–2500 nm. The UV-Visible spectra (Fig. 18.3a) compares the absorption characteristics of the pure perovskite (FTO/MAPbI₃) and the soot coated perovskite photoelectrode (FTO/TiO₂/MAPbI₃/C/SiO₂). The spectra show that the incorporation of TiO₂ and carbon as electron transport layer (ETL) and hole transport layer (HTL) respectively results in the shifting of the absorption towards the ultraviolet region. The band-gap energies were calculated by the absorbance data using the following equation:

$$\alpha h \upsilon = A (h \upsilon - E_g)^m \tag{18.1}$$

where α is the absorption coefficient, h υ is the photon energy in eV, and E_g is the band-gap energy in eV. A is a constant related to the effective mass of the electrons and holes and m is equal to 0.5 for allowed direct transition and 2 for an allowed indirect transition. Plots between $(\alpha h \upsilon)^2$ and h υ for the pristine perovskite and the soot coated perovskite are shown in Fig. 18.3b, c respectively. The linear nature of the graph suggests that the sample behaves as a direct band-gap material. The band-gap for the pristine perovskite sample comes out to be 2.05 eV close to the reported value of 1.56 eV [7] while for the soot coated perovskite, it comes out to be 2.49 eV. The high band gap SiO₂ layer may have resulted in the overall increase in the device band gap.



Fig. 18.3 a Absorption spectra of the $FTO/MAPbI_3$ and $FTO/TiO_2/MAPbI_3/C/SiO_2$ thin films. b Tauc plot for the pristine perovskite sample ($FTO/MAPbI_3$). c Tauc plot for the soot coated perovskite sample



18.3.3 Photoelectrochemical Measurements

The photosensitivity of the prepared electrode is analysed by investigating its current versus voltage characteristics in a three-electrode quartz cell under light and dark conditions. Figure 18.4 shows the photocurrent density of the photoelectrode as a function of the electrode potential. The FTO/TiO₂/MAPbI₃/C/SiO₂ combination yielded a photocurrent density of 1.8 mA cm⁻² at an applied bias of 1 VSCE. The excellent photo-response of the soot coated perovskite photoelectrode suggests that the hydrophobic soot/SiO₂ coating did not interfere with the photoelectrochemical response of the perovskite. The stability of the prepared photoelectrodes is investigated by measuring the current versus time characteristics in a PEC cell. Figure 18.5 shows Chronoamperometry recorded at an applied potential of 1 V versus SCE in a



0.1 M aqueous NaOH solution. The soot coated photoelectrodes showed remarkable stability with currents remaining above 1.5 mA cm⁻² for ~1 h under constant illumination. This represents a first example of halide perovskite photoelectrode, protected with an inexpensive protection layer.

18.4 Conclusion

This work focusses on solving the long-standing moisture instability issue of halide perovskites through a simple, efficient and inexpensive protection route. The FTO/TiO₂/MAPbI₃/C/SiO₂ photoelectrodes exhibited the potential to be employed as stable electrodes for water splitting in PEC hydrogen production, showing outstanding stability for ~1 h in aqueous electrolyte. The perfect energy band alignment permitted carbon to be an efficient and directional hole extraction layer. While the excited electron moves from the conduction band (CB) at -3.86 eV of the MAPbI₃ to the CB of TiO₂ (-4.00 eV), the hole in the valence band (VB) at -5.43 eV is effectively extracted to the carbon layer (work function 5.00 eV). While the candle soot acted as an effective HTL due to its intimate contact with the perovskite layer, the SiO₂ offered a more compact seal that protected the perovskite from the liquid electrolyte. The enormous stability coupled with the high photocurrent density makes these soot coated photoelectrodes an attractive device for application in a tandem photoelectrochemical cell for solar water splitting. Further research into the optimization of the electron transport layer and the use of more efficient perovskites can improve both stability and efficiency.

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