



1 *Communication*

# 2 **Biom mineralization of Engineered Spider Silk** 3 **Protein-Based Composite Materials for Bone Tissue** 4 **Engineering**

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20 **Abstract:** Materials based on biodegradable polyesters such as poly(butylene terephthalate) (PBT)  
21 or poly(butylene terephthalate-co-poly(alkylene glycol) terephthalate) (PBTAT) have potential  
22 application as pro-regenerative scaffolds for bone tissue engineering. Herein is reported the  
23 preparation of films composed of PBT or PBTAT and an engineered spider silk protein,  
24 (eADF4(C16)), that displays multiple carboxylic acid moieties capable of binding calcium ions and  
25 facilitating their biom mineralization with calcium carbonate or calcium phosphate. Human  
26 mesenchymal stem cells cultured on films mineralized with calcium phosphate show enhanced  
27 levels of alkaline phosphatase activity suggesting that such composites have potential use for bone  
28 tissue engineering.

29 **Keywords:** spider silk; recombinant protein; biodegradable polymers; biomaterials;  
30 biom mineralization; bone tissue engineering.

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## 33 **1. Introduction**

34 Bones are composed of mixtures of inorganic material, predominantly calcium phosphate in the  
35 form of carbonated hydroxyapatite and organic material, predominantly collagen, and many  
36 different materials and manufacturing methodologies are used in the development of bone tissue  
37 scaffolds [1]. While non-biodegradable materials (e.g. metals, polyethylene and  
38 polyetheretherketone [2,3]) are commonly used to manufacture components for certain applications  
39 in bone tissue, for instance hip replacements, there are issues with these materials such as  
40 inflammation, metal sensitivity and toxicity, and solutions to these issues are the subject of ongoing  
41 research [2,3]. Biodegradable materials are of particular interest because their eventual resorption  
42 allows them to be remodelled in vivo, and biodegradable polymer-based materials and composites  
43 based thereon are popular avenues of research [4-15].

44 Poly(butylene terephthalate) (PBT) and its copolymers with poly(ethylene oxide) (e.g. PBTAT  
45 derivatives) are biodegradable polymers that are easy to process into films, fibers and foams [16-19].  
46 Scaffolds based on PBT and/or PBTAT have been demonstrated to be suitable substrates for the  
47 attachment and proliferation of chondrocytes, mammalian skeletal muscle cells [19], bone marrow  
48 stromal cells [18], and human mesenchymal stem cells [17] in vitro. Preclinical studies in various  
49 animal models showed that the degradation rate of scaffolds based on PBT and/or PBTAT were  
50 dictated by the precise composition of the polymer backbone which suggests it may be possible to  
51 tailor-make such materials for specific conditions or patients; and in mammals PBTAT-based  
52 materials encouraged bone growth, which motivates the development of PBT-/PBTAT-based  
53 scaffolds for bone regeneration [20-23].

54 Silk protein-based materials are also candidates for the generation of tissue scaffolds [24-31].  
55 The natural silk fibroin of the domesticated *Bombyx mori* silkworm is the most commonly  
56 investigated for such applications [24-32], however, recombinantly produced silk-inspired proteins  
57 represent interesting alternatives because it is possible to produce large quantities of such silks with  
58 designed primary sequences [33-37]. Silk-based composites are also widely investigated for  
59 application as tissue scaffolds [37-40], and preclinical trials in animal models are promising  
60 [35,36,41].

61 Scheibel and coworkers have developed engineered spider silks based on the two most  
62 abundant proteins found in the dragline silks of the European garden spider (*Araneus diadematus*,  
63 *A. diadematus* fibroin 3 and 4, ADF3 and ADF4 respectively); the engineered silk protein analogues  
64 (eADF3 and eADF4 respectively), can be produced by an industrially viable fermentation process in  
65 *Escherichia coli* bacteria [42-45]. The repetitive backbone sequence of eADF4 analogues displays  
66 numerous glutamic acid residues [42] enabling their chemical modification [46] or binding cations  
67 such as drugs [47].

68 This manuscript describes the preparation and characterization of composites of PBT or PBTAT  
69 with an eADF4 analogue, namely eADF4(C16), and their biocompatibility as assayed with  
70 fibroblasts (M-MSV-BALB/3T3) and human mesenchymal stem cells. Moreover, mineralization of  
71 these composites with calcium phosphate enhanced the levels of alkaline phosphatase activity of  
72 human mesenchymal stem cells cultured on the substrates, and therefore they are potentially useful  
73 for integration in biodegradable devices applied in bone tissues [48]. Such materials have prospects  
74 for application in tissue engineering and regenerative medicine, for use in various bone tissue  
75 specific niches.

## 76 2. Materials and Methods

### 77 2.1. Materials

78 Unless otherwise stated, all chemicals were of ACS grade, purchased from Sigma-Aldrich  
79 Chemie GmbH and used as supplied. Reagents for cell culture were purchased from Invitrogen  
80 (Carlsbad, CA) unless otherwise noted. Human mesenchymal stem cells (HMSCs) were purchased  
81 from Lonza Cologne GmbH (Cologne, Germany). High glucose Dulbecco's Modified Eagle Medium  
82 (DMEM) and fetal bovine serum (FBS) were purchased from Biochrom AG (Berlin, Germany). The  
83 recombinantly produced silk protein was based on the consensus motif of the repetitive core domain  
84 of one of the major ampullate silk fibroins of the garden cross spider (*A. diadematus* fibroin 4). The  
85 recombinant protein is composed of sixteen repeats of the polypeptide module C (amino acid  
86 sequence: GSSAAAAAASGPGGYGPENQGPSGPGGYGPGGP), and is referred to hereafter as  
87 eADF4(C16). Production and purification of eADF4(C16) was carried out as described previously  
88 [42].

### 89 2.2. Film preparation, thermogravimetric analysis (TGA), X-ray diffraction (XRD), Fourier transform infrared 90 (FTIR) spectroscopy, in vitro degradation studies, and in vitro fibroblast adhesion studies

91 Adapted from previously described methodology [47], for full experimental details refer to the  
92 Supporting Information.

### 93 2.3. Mineralization of films with calcium carbonate

94 Three beakers (10 mL) containing crushed ammonium carbonate were also covered with  
95 Parafilm® punched with three needle holes and placed at the bottom of a large desiccator, above  
96 which films cast in 24 well tissue culture plates were incubated in an aqueous solution (1 mL) of  
97 calcium chloride (25 mM), and covered with Parafilm® punched with three needle holes. The  
98 desiccator was sealed and the samples left for 72 hours. The samples were subsequently washed  
99 with water until the pH was neutral, and then with ethanol/water (70 % ethanol, 30% water) and  
100 allowed to dry in a sterile fume hood overnight.

### 101 2.4. Mineralization of films with calcium phosphate

102 Films cast in 24 well tissue culture plates were incubated in an aqueous solution (1 mL) of  
103 calcium chloride (200 mM) for 20 minutes, after which the solution was removed and the samples  
104 were washed with water (3 x 1 mL). Thereafter, samples were incubated in an aqueous solution (1  
105 mL) of sodium phosphate (120 mM) for 20 minutes, after which the solution was removed and the  
106 samples were washed with water (3 x 1 mL). The cycle of incubation with calcium chloride and  
107 sodium phosphate was repeated a further six times (i.e. a total of 7 cycles), after which the samples  
108 were incubated in ethanol/water (70% ethanol, 30% water) for 30 minutes and allowed to dry in a  
109 sterile fume hood overnight.

### 110 2.2.5. Scanning Electron Microscopy (SEM) and Energy Dispersive Spectroscopy (EDS)

111 Samples were mounted on metal stubs, coated with Pt/Pd or Carbon using a Cressington 208  
112 benchtop sputter coater before being observed with a Hitachi S5500 SEM equipped with an EDS  
113 probe.

### 114 2.6. Stem cell culture and qualitative and quantitative studies of alkaline phosphatase activity

115 Commercially available Nunclon® Δ surface tissue culture plates were used for control  
116 experiments. Silk films were sterilized by incubation in 70% ethanol solution followed by exposure  
117 to UV for 60 minutes. After sterilization, the samples were incubated for 30 minutes under 3 mm of  
118 HMSC growth medium. HMSC growth medium was composed of: high glucose Dulbecco's  
119 Modified Eagle Medium (DMEM, 440 mL); fetal bovine serum (50 mL); antibiotic-antimycotic (5  
120 mL); non-essential amino acids (5 mL), and 2 ng/mL basic fibroblast growth factor. Medium was  
121 aspirated and replaced prior to HMSC seeding. Cell viability before starting the experiment was  
122 determined by the Trypan Blue exclusion method, and the measured viability exceeded 95% in all  
123 cases. HMSCs were seeded at 10,000 cells/cm<sup>2</sup> under 3 mm of medium, and incubated at 37°C, 95%  
124 humidity, and a CO<sub>2</sub> content of 5%. After 3 days the medium was aspirated, the films were washed  
125 gently with PBS and replaced with osteogenic medium. Osteogenic medium was composed of: high  
126 glucose Dulbecco's Modified Eagle Medium (DMEM, 425 mL); fetal bovine serum (50 mL);  
127 antibiotic-antimycotic (5 mL); non-essential amino acids (5 mL), dexamethasone (100 nM), β-glycerol  
128 phosphate (10 mM) and ascorbic acid (50 μM). Thereafter the osteogenic medium was aspirated and  
129 replaced every 2 days until the samples were analysed. Alkaline Phosphatase (ALP) activity was  
130 visualized with a Leukocyte Alkaline Phosphatase Kit using the manufacturer's protocol. Images of  
131 stained cells were obtained using a camera AxioCam MRm attached to a Zeiss Axio Observer Z1  
132 equipped with an ApoTome unit. Images are representative of 3 samples. DNA was quantified  
133 using PicoGreen® assay (Life Technologies GmbH, Darmstadt, Germany) using a Synergy HT  
134 Multi-Mode Microplate Reader (Bio-tek Instruments GmbH, Bad Friedrichshall, Germany). ALP  
135 activity of the cell population was quantified by first scraping and breaking up the films in a buffer  
136 of 0.2% Triton X-100, and then measuring ALP activity using an ALP LiquiColor® kit (Stanbio,  
137 Boerne, TX) in accordance with the manufacturer's protocol. The sample and reagents were  
138 incubated in a 96 well plate for 1 h at 37°C and then read using a Synergy HT Multi-Mode  
139 Microplate Reader (Bio-tek Instruments GmbH, Bad Friedrichshall, Germany). Data were  
140 normalized to DNA quantity. Statistical analysis via ANOVA (null hypothesis that all groups have

141 the same true mean, P-value < 0.0001) carried out within R (<http://www.r-project.org/>), and one way  
142 ANOVA statistics were calculated and interpreted with Tukey's T-test, for which any interval that  
143 does not cross zero (the dashed line) is significant with an alpha = 0.05 [9].

### 144 3. Results and Discussion

#### 145 3.1. Film preparation and characterization

146 The compositions of the films described herein are found in Table 1. All films had thicknesses of  
147 ca. 100  $\mu\text{m}$ , and therefore would not be expected to be encapsulated inside a very thick foreign body  
148 capsule in vivo [47]. Thermogravimetric analysis revealed that "as cast" films contained residual  
149 volatiles (HFIP and water), levels of which were diminished by immersion of the films in methanol  
150 (Figures S1–S9, Supporting Information).

151 Analysis of the films by X-ray diffraction (Figures S1–S9 and Table S1, Supporting Information)  
152 was informative, confirming that the eADF4(C16) silk component of the "as cast" films was water  
153 soluble due to its  $\alpha$ -helix rich nature (XRD peaks at  $2\theta = 14.4^\circ$  and  $19.4^\circ$ ) induced by the HFIP used in  
154 the casting process [47], and that methanol treatment rendered the silk component of films insoluble  
155 in water due to induction of  $\beta$ -sheet formation (XRD peaks at  $2\theta = 16.7^\circ$ ,  $19.9^\circ$ ,  $24.0^\circ$ , and  $31.8^\circ$ , in  
156 agreement with literature data), suggesting that this process removes residual HFIP [47]. The peak  
157 positions for PBT [49,50] or PBTAT [49,50] are in line with those reported in the literature for each  
158 polymer, respectively. Interestingly, the XRD spectra of the films composed solely of PBT or PBTAT  
159 revealed that they became more crystalline after treatment with methanol, which supports our  
160 assertion that methanol treatment removes residual HFIP that solvates the polymers, thereby  
161 deterring their crystallization. XRD spectra of films composed of mixtures of eADF4(C16) and the  
162 PBT or PBTAT displayed peaks due to the combinations of the two components, however, the  
163 signals of eADF4(C16) were normally only evident as shoulders on the peaks due to the more  
164 crystalline PBT or PBTAT.

165 FTIR spectroscopy confirmed that HFIP (Figure S10, Supporting Information) was present in  
166 the "as cast" films (strong absorption at  $1105\text{ cm}^{-1}$ ), and that it could effectively be removed by  
167 methanol treatment, as the absorption was markedly diminished or absent (Figures S1–S9,  
168 Supporting Information). Furthermore, FTIR spectroscopy confirmed the silk component of the as  
169 cast films to be  $\alpha$ -helix rich (amide I and II peaks were observed at  $1656$  and  $1547\text{ cm}^{-1}$ , respectively),  
170 whereas the methanol treated films were  $\beta$ -sheet rich (amide I and II absorptions were shifted to  
171  $1625$  and  $1521\text{ cm}^{-1}$  respectively, and a peak at  $965\text{ cm}^{-1}$  assigned to polyalanine-based  $\beta$ -sheets).

172 Visual observation of the "as cast" and "methanol treated" films by photography and bright  
173 field microscopy (Figures S1–S9, Supporting Information), revealed a degree of phase separation  
174 between the eADF4(C16) and PBT or PBTAT (analogous to that observed for composites of  
175 eADF4(C16) and polycaprolactone or Pellethane 2363-80A) [47]. Differences in the optical properties  
176 of the components of the films (the silk being relatively clear, and the PBT/PBTAT being relatively  
177 opaque) enabled the assignment of the component constituting the continuous phase as reported in  
178 Table 1.

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**Table 1.** Film compositions and properties.

Film	Mass ratio protein:polymer	Continuous phase	Fibroblast adhesion relative to Nunclon® $\Delta$ surface (%)	Figure
eADF4(C16)	100:0	eADF4(C16)	72.0 $\pm$ 8.0	S1 and Ref. 47
PBT-25	75:25	eADF4(C16)	55.5 $\pm$ 5.9	S2
PBT-50	50:50	PBT	58.9 $\pm$ 8.0	S3
PBT-75	25:75	PBT	69.8 $\pm$ 10.0	S4
PBT-100	0:100	PBT	75.8 $\pm$ 3.5	S5
PBTAT-25	75:25	eADF4(C16)	76.9 $\pm$ 6.6	S6
PBTAT-50	50:50	PBTAT	104.5 $\pm$ 4.4	S7
PBTAT-75	25:75	PBTAT	76.4 $\pm$ 2.4	S8
PBTAT-100	0:100	PBTAT	69.3 $\pm$ 2.4	S9
Untreated Nunclon®	Not applicable	Not applicable	74.0 $\pm$ 6.2	S11
Nunclon® $\Delta$ Surface	Not applicable	Not applicable	100.0 $\pm$ 7.5	Ref. 47

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### 192 3.2. *In vitro* degradation studies

193 A biomaterial's performance *in vivo* is influenced by its stability and degradation profile. For  
 194 tissue engineering applications materials that degrade are attractive as they can be replaced by  
 195 native extracellular matrix, and it is useful to be able to tune the degradation behavior of  
 196 biomaterials [24,32,51]. Trypsin and elastase were chosen as biologically relevant model proteolytic  
 197 enzymes that play roles in digestion and wound healing, respectively. The *in vitro* degradation of  
 198 the films in solutions of elastase and trypsin in phosphate buffered saline (PBS) was studied over the  
 199 period of 250 hours (Figures S1–S9, Supporting Information). Spontaneous hydrolysis of  
 200 eADF4(C16), PBT and PBTAT has been reported to be negligible (<2%) as they are insoluble in water,  
 201 and hydrolysis of the amides and esters in their respective backbones is a very slow process  
 202 [22–24,47]. In the presence of elastase and trypsin the films composed solely of eADF4(C16) were  
 203 observed to degrade slowly and had sufficient structural integrity to be manipulated for over 250  
 204 hours (Figure S1, Supporting Information). Mass loss profiles recorded using the same procedure for  
 205 PBT-25 (Figure S2, Supporting Information) and PBTAT-25 (Figure S6, Supporting Information)  
 206 films showed that they degraded more swiftly, in part because their phase separated nature formed  
 207 the basis for small parts of the film separating from the bulk; their degradation profiles are included  
 208 for completeness and not representative solely of the enzymatic degradation of the silk protein. The  
 209 structural integrity of all of the other films was maintained for the duration of the experiments, and  
 210 the data are therefore representative of the enzymatic degradation of the silk protein, and mass loss  
 211 was faster from films with higher eADF4(C16) content. Clearly, it would be expected that the  
 212 degradation of the films *in vivo* would be markedly slower than that of our *in vitro* assay, in line  
 213 with the literature precedent for *Nephila clavipes* spider silk [52], *B. mori* silkworm silk [41], or the  
 214 polyesters [22,23], respectively.

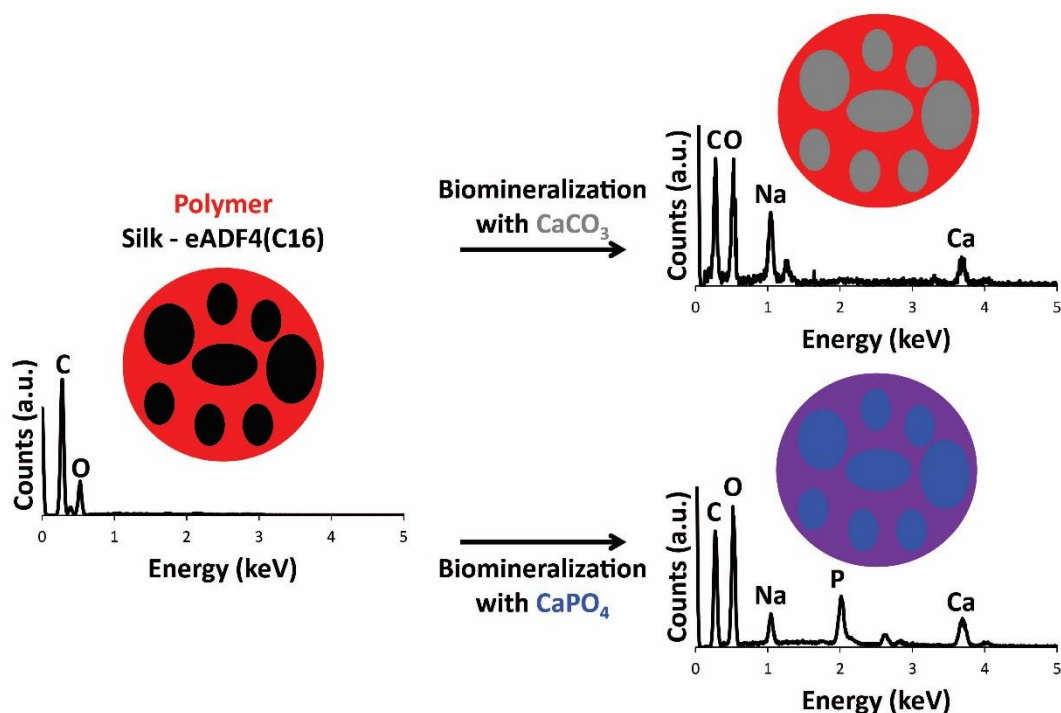
### 215 3.3. *In vitro* fibroblast adhesion studies

216 BALB/3T3 mouse fibroblast adhesion to the films was assayed using Alamar Blue, with two  
 217 commercially available surfaces as references for our studies, untreated polystyrene tissue culture  
 218 plates (Nunclon®) and plasma treated polystyrene tissue culture plates (Nunclon®  $\Delta$  Surface), and  
 219 cell adhesion is reported relative to the Nunclon®  $\Delta$  surface [46,47]. Since the cells were in a

220 quasi-steady-state situation, increasing values of fluorescence are proportional to the number of  
221 cells, observing fibroblast adhesion on all of the films (Table 1 and Supporting Information).  
222 Fibroblast adhesion to films incorporating PBT or PBTAT was in all cases better than to films  
223 composed of eADF4(C16) alone (which already have been described to be a poor surface for  
224 fibroblast adhesion), and generally comparable to levels of adhesion observed for the untreated  
225 Nunclon® tissue culture plates; interestingly, levels of cell adhesion to PBTAT-50 films were similar  
226 to that on plasma treated Nunclon®  $\Delta$  Surface tissue culture plates. Cells were clearly observable on  
227 the optically clear films of eADF4(C16) and tissue culture plates (Figure S1, S11 and [47],  
228 respectively), whereas cells on the composite films were more easily visualized after Calcein A/M  
229 staining (Figures S2-S9, Supporting Information).

### 230 3.4. Film biomineralization with calcium carbonate or calcium phosphate

231 With a view to the application of the materials as scaffolds for bone tissue engineering, the films  
232 were biomineralized [53,54] with calcium carbonate or calcium phosphate. Mineralization of the  
233 films with calcium carbonate was achieved by incubation of the films in solutions of calcium  
234 chloride in a container with ammonium carbonate, and mineralization of the films with calcium  
235 phosphate was achieved by iterative sequences of incubation of the films in solutions of calcium  
236 chloride followed by sodium phosphate. The engineered silk eADF4(C16) displays multiple  
237 carboxylic acid moieties capable of binding calcium ions facilitating their mineralization. Energy  
238 dispersive spectroscopy (EDS) analysis of the films confirmed that the surface chemistry of the films  
239 before and after mineralization was different. Peaks in the EDS spectra of the eADF4(C16) and  
240 composite films prior to mineralization have lines at 0.277, 0.525, and 1.041 keV that are the  
241 characteristic  $K\alpha$  emissions of carbon, oxygen and sodium, respectively, and the weak emission at  
242 0.392 keV is the  $K\alpha$  emission of nitrogen (Figure 1). After the mineralization, new peaks appeared in  
243 the spectra at 2.013, 2.621 and 3.690 keV which are the characteristic  $K\alpha$  emission line of  
244 phosphorous, chlorine (from the calcium chloride used as a source of  $Ca^{2+}$ ) and calcium, respectively  
245 (Figure 1). Imaging with SEM-EDS revealed that calcium carbonate was preferentially deposited in  
246 the eADF4(C16) phase of the films, as opposed to the PBT or PBTAT phases, whereas the calcium  
247 phosphate was deposited more homogeneously across the surface of the films (as depicted in  
248 schematic format in Figure 1); this is likely to be caused by differences in the concentration of  
249 calcium chloride solution in which the films were incubated, 25 mM for calcium carbonate  
250 mineralization as opposed to 200 mM for calcium phosphate deposition (examples for PBT-50 and  
251 PBTAT-50 are displayed in Figure 2).



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**Figure 1.** Schematic of biom mineralization of films with representative EDS analysis of films.

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### 3.5. *In vitro* stem cell culture

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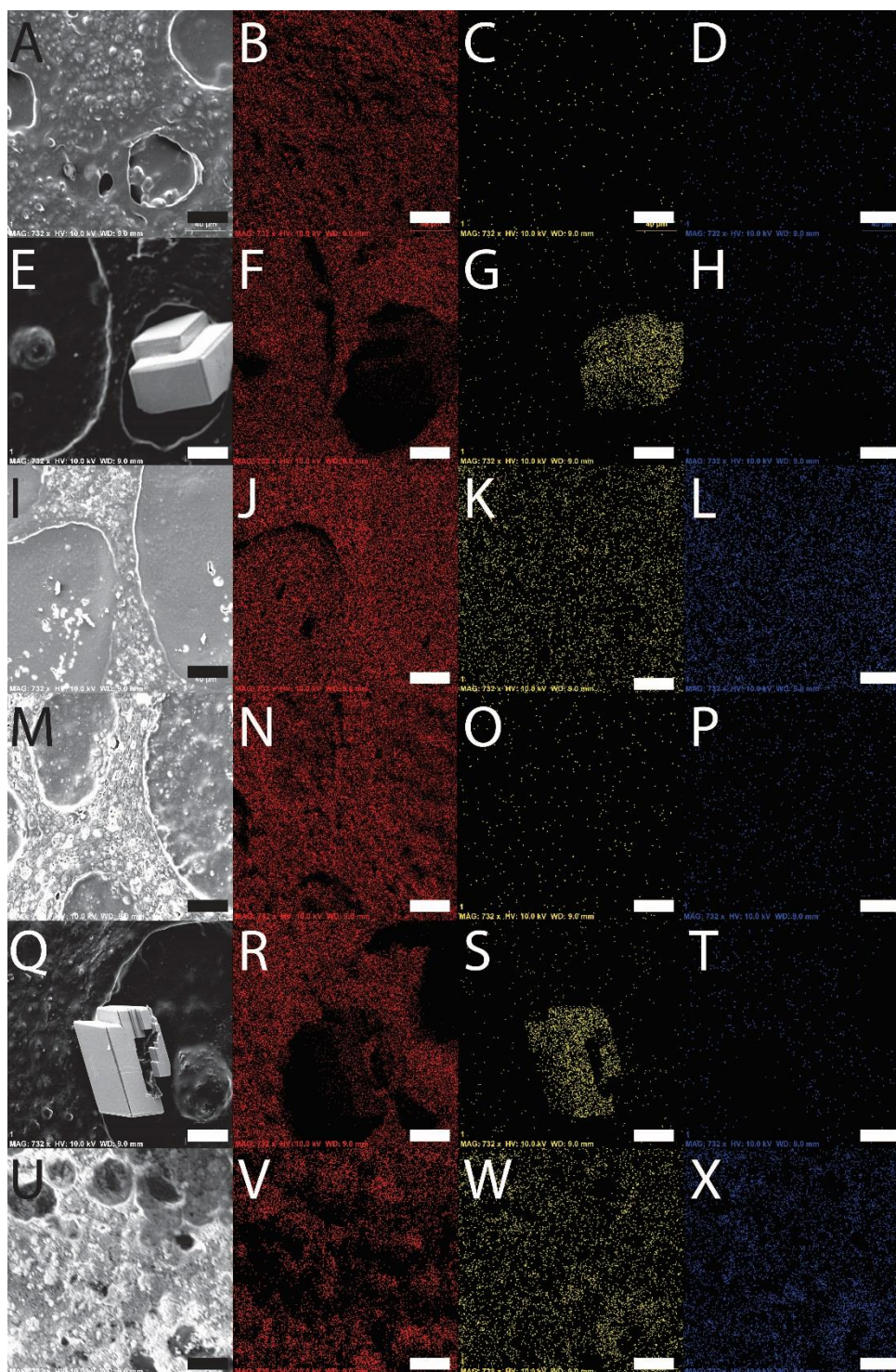
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Human mesenchymal stem cells were cultured *in vitro* for 2 weeks on calcium phosphate mineralized films. Alkaline phosphatase (ALP) activity is a hallmark of bone tissue formation, and therefore both qualitative and quantitative analyses of ALP activity were studied. Qualitative analysis of ALP activity using ALP live staining (Figure 3, A-J) showed that the cells were alive and functional on the films as seen by the patches of dark coloration that is characteristic of the precipitated stain. Quantitative analysis of ALP activity for the cells cultured on the mineralized films (Figure 4) showed that ALP activity (Figure 4A) was correlated with levels of fibroblast adhesion (Table 1). The one-way analysis of variance (ANOVA) was used to determine whether there were any significant differences in the quantitative analyses of ALP activity (Figure 4B), and the one-way ANOVA rejects the null hypothesis that all groups have the same true mean ( $P$ -value < 0.0001). Consequently, Tukey's T-test was used to compare differences between groups, where any interval that does not cross zero (the dashed line in Figure 4B) is significant with an  $\alpha = 0.05$ . Interestingly, levels of ALP activity for the cells cultured on Nunclon®  $\Delta$  were significantly different from all other films. Levels of ALP activity for the cells cultured on mineralized eADF4(C16) were not significantly different from the mineralized PBT composites, or indeed the pure PBT or PBTAT; however, statistically significant differences were observed for mineralized PBTAT-50 and PBTAT-75, wherein ALP activity for cells cultured on these materials was higher than for either of the constituents (eADF4(C16) or PBTAT) alone (and logically the PBT composites). Together, this suggests that composites of eADF4(C16) and PBTAT have some potential for bone tissue engineering.

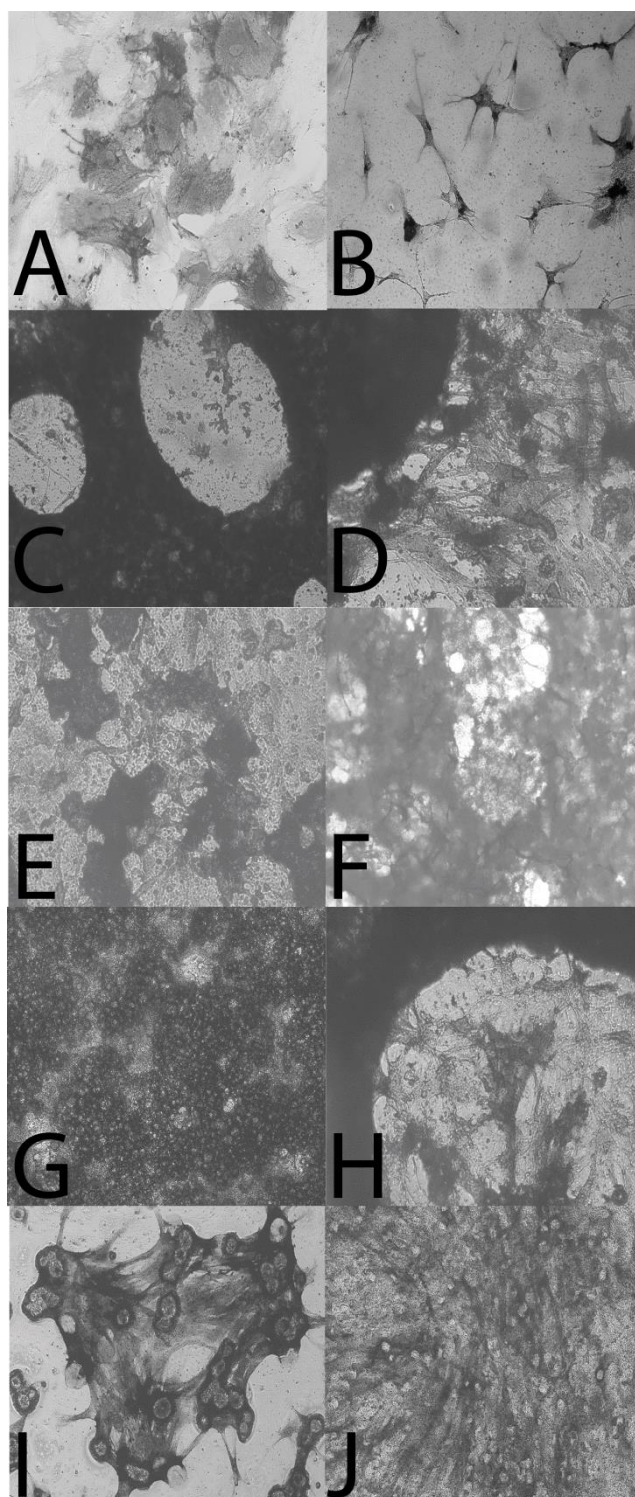




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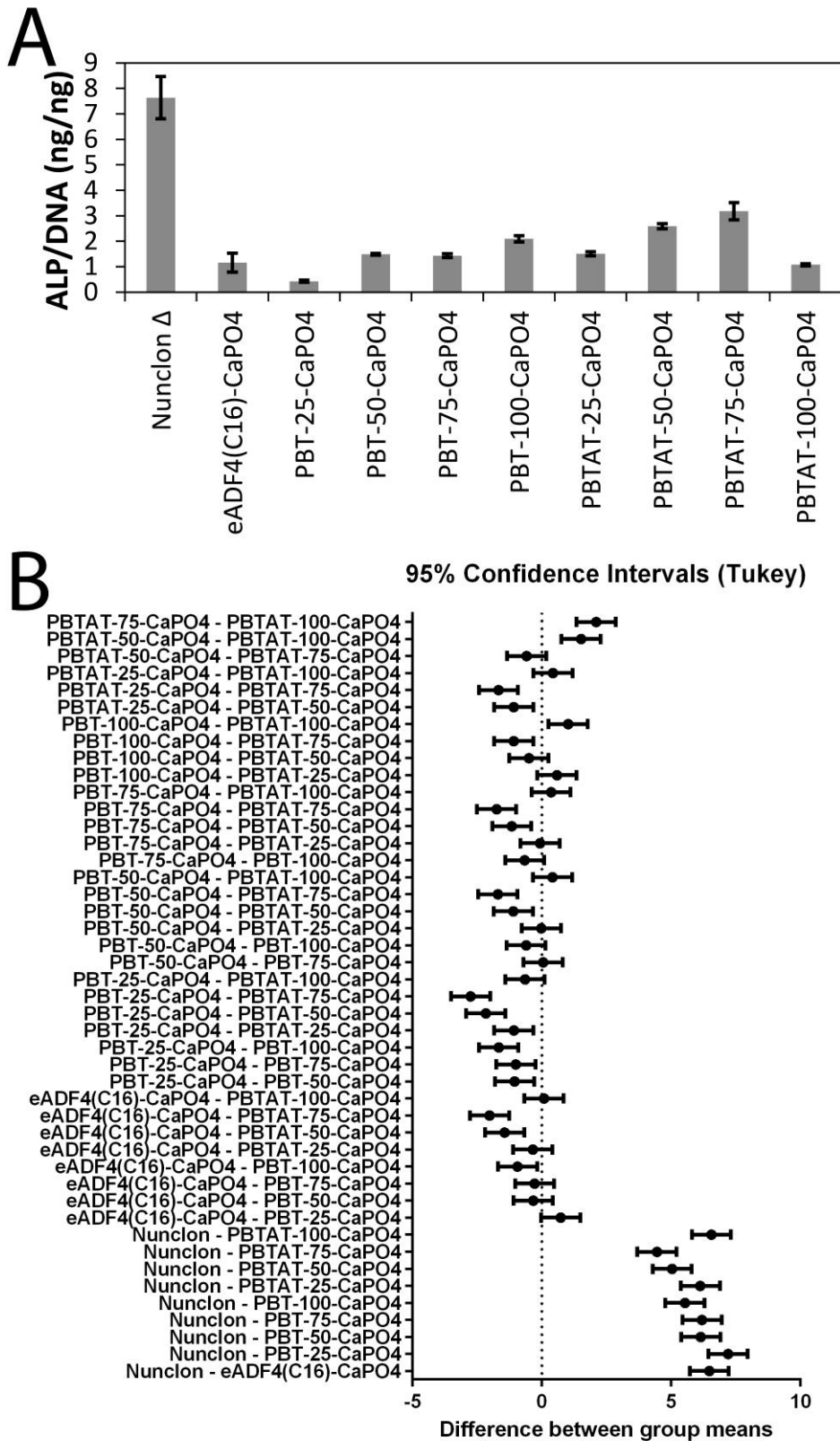
**Figure 2.** SEM-EDS analysis of films. A-D) PBT-50. E-H) PBT-50-CaCO<sub>3</sub>. I-L) PBT-50-CaPO<sub>4</sub>. M-P) PBTAT-50. Q-T) PBTAT-50-CaCO<sub>3</sub>. U-X) PBTAT-50-CaPO<sub>4</sub>. A, E, I, M, Q, U) Secondary electron SEM image. B, F, J, N, R, V) Carbon, red. C, G, K, O, S, W) Calcium, yellow. D, H, L, P, T, X) Phosphorous, blue. Scale bar represents 40 µm.





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**Figure 3.** A-J) Qualitative analysis of ALP activity of stem cells on films mineralized with calcium phosphate using bright field microscopy after ALP live staining. A) Nunclon® Δ. B) eADF4(C16)-CaPO<sub>4</sub>. C) PBT-25-CaPO<sub>4</sub>. D) PBT-50-CaPO<sub>4</sub>. E) PBT-75-CaPO<sub>4</sub>. F) PBT-100-CaPO<sub>4</sub>. G) PBTAT-25-CaPO<sub>4</sub>. H) PBTAT-50-CaPO<sub>4</sub>. I) PBTAT-75-CaPO<sub>4</sub>. J) PBTAT-100-CaPO<sub>4</sub>. Images are 900 μm wide.



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**Figure 4.** A) Quantitative analysis of ALP activity of stem cells on films mineralized with calcium phosphate. B) Statistical analysis via ANOVA (null hypothesis that all groups have the same true mean, P-value < 0.0001), and one way ANOVA statistics were calculated and interpreted with Tukey's T-test, for which any interval that does not cross zero (the dashed line) is significant with an alpha = 0.05.

#### 293 4. Conclusions

294 Films composed of natural and recombinantly produced silk proteins have been widely  
295 investigated for biomedical applications such as biocompatible coatings for biomedical implants,  
296 owing to the facility with which silk proteins can be processed into films with tunable surface  
297 properties (morphology, hydrophilicity, etc.), their biodegradability and low levels of  
298 immunogenicity in vitro/in vivo. This manuscript reports a simple method of producing films  
299 composed of a recombinantly produced spider silk inspired protein eADF4(C16) and biodegradable  
300 polymers (PBT and PBTAT), their mineralization with either calcium carbonate or calcium  
301 phosphate, and a preliminary in vitro cell culture experiment to assess their efficacy for bone tissue  
302 engineering. Interestingly, levels of ALP activity for HMSCs residing on calcium phosphate  
303 mineralized PBTAT-50 and PBTAT-75 films were elevated when compared to the other formulations  
304 investigated or indeed the constituents alone, and it is concluded that such composites have  
305 potential for the development of functional biomineralized biomaterials [55–62].

306 **Supplementary Materials:** The following are available online at [www.mdpi.com/link](http://www.mdpi.com/link), Figure S1: eADF-4(C16)  
307 films. Figure S2: PBT-25 films. Figure S3: PBT-50 films. Figure S4: PBT-75 films. Figure S5: PBT-100 films. Figure  
308 S6: PBTAT-25 films. Figure S7: PBTAT-50 films. Figure S8: PBTAT-75 films. Figure S9: PBTAT-100 films. Table  
309 S1: Positions of XRD peaks of films determined using Jade 9 XRD Pattern Processing software. Figure S10: FTIR  
310 spectrum of pure HFIP. Figure S11: Bright field microscope image of fibroblasts cultured on Nunclon® Tissue  
311 Culture Plate (scale bar represents 100 µm).

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316 preparation, Ute Kuhn for assistance with TGA, Roman Kress for assistance with X-ray diffraction (all at the  
317 University of Bayreuth). We thank Reed Harrison of the Department of Bioengineering at the University of  
318 California, Riverside, in the USA for statistical analysis.

319 **Author Contributions:** J.G.H. prepared the samples, performed characterization and analyzed the data;  
320 J.G.T.-S. carried out microscopy on the stem cells; A.L.-E. performed all experiments and analysis of data  
321 regarding fibroblasts; A.W., H.S., H.C. and T.R.S. supervised the research; all authors discussed the data and  
322 wrote the paper.

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