

1 Structure and stability of arsenate adsorbed on  $\alpha$ -Al<sub>2</sub>O<sub>3</sub>  
2 single-crystal surfaces investigated using  
3 grazing-incidence EXAFS measurement and DFT  
4 calculation

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12 **Abstract:** Direct characterization of contaminants on single-crystal planes is required because the  
13 specific adsorption characteristics on different exposed crystal planes constitute their actual behavior  
14 at water-mineral interfaces in aquifers. Here, the structure and stability of arsenate on  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> (0001)  
15 and (11 $\bar{2}$ 0) surfaces were characterized by using a combination of grazing-incidence extended X-ray  
16 absorption fine structure (GI-EXAFS) spectra and periodic density functional theory (DFT)  
17 calculation. The combined results indicated that arsenate was mainly adsorbed as inner-sphere  
18 monodentate and bidentate complexes on both surfaces, but the orientational polar angles on the (0001)  
19 surface were commonly 10~20° greater than that on the (11 $\bar{2}$ 0) surface. The DFT calculation showed  
20 that the large polar angle was more favorable for arsenate stabilized on the alumina surfaces. Based on  
21 the spectroscopic and computational data, the dominant bonding modes of arsenate on the two crystal  
22 planes of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> were identified as bidentate binuclear structures, and the (0001) surface displayed a  
23 stronger affinity toward arsenate.

24

25 **Keywords:** surface complex; molecular orientation; structure-stability relationship; GI-EXAFS;  
26 density functional theory

## 27 **1. Introduction**

28 The persistence of arsenic in soil and aquatic environments has been ranked as a global  
29 environmental problem due to its toxic impact on human health (Oremland and Stolz, 2003).  
30 Adsorption of arsenic at the solid-water interface is one of the dominate effects responsible for water  
31 treatment and soil decontamination (Jing et al., 2005; Mohan and Pittman, 2007; Singer et al., 2013).  
32 Detailed information about the structure and stability of arsenic on solid surfaces is essential for  
33 understanding its fate in the environment and developing high-performance adsorbents (van  
34 Genuchten et al., 2012).

35 Over the past two decades, extended X-ray absorption fine structure (EXAFS) spectroscopy has  
36 been developed as one of the most important technologies for determining the microstructure of  
37 environmental contaminants on solid surfaces (Arai et al., 2001; Duarte et al., 2012; van Genuchten et  
38 al., 2012; Waychunas et al., 1993). Most existing EXAFS studies are carried out using powder  
39 substrates, where all types of crystal planes may be mixed and expose to the adsorption process. The  
40 multiple surface terminations and extensive defects of mineral powder may produce a large number of  
41 different adsorption sites (Daniels et al., 2001). Therefore, the structural data obtained from standard  
42 powder EXAFS experiment are essentially the average over all values from different exposed crystal  
43 planes. To avoid these problems, the application of structurally well-defined single-crystal substrates  
44 is needed (Catalano et al., 2005; Singer et al., 2012; Waychunas et al., 2005). Direct characterization  
45 of the structure and bonding properties on specific crystal planes is required to conclusively unveil the  
46 interaction of arsenic species with adsorbent powders.

47 Grazing-incidence EXAFS (GI-EXAFS) is a significant development of EXAFS technique,  
48 which utilizes well-characterized single-crystal samples (Singer et al., 2012). The application of

49 crystallographically anisotropic substrate and polarized X-ray beam allows GI-EXAFS to provide  
50 more structural information (interatomic distance, coordination number and polar angle) toward the  
51 oriented adsorbates (Waychunas et al., 2005). These advantages provide the opportunity to obtain  
52 direct information about the structure of contaminant molecules on different exposed crystal planes,  
53 which may trigger wide applications of GI-EXAFS in environmental geochemistry and  
54 geo-engineering as synchrotron facilities are improving. Density functional theory (DFT) calculation  
55 can provide a detailed characterization of adsorbate molecules at solid surfaces, including structure,  
56 bonding, and energy (Duarte et al., 2012; He et al., 2011). The combination of GI-EXAFS  
57 measurement with DFT calculation enables analysis of the structure and stability characteristics of  
58 arsenic species on specific crystal planes.

59 Aluminum is one of the most abundant metals in the earth, and Alumina has been recognized as a  
60 very efficient adsorbent for the removal of arsenic from waters (Mohan and Pittman, 2007).  $\alpha$ - $\text{Al}_2\text{O}_3$  is  
61 the most stable phase of alumina with a well-determined structure, and an excellent model system for  
62 understanding the transport mechanisms of contaminants on Al-(hydr)oxides (Catalano et al., 2008;  
63 Trainor et al., 2001). Both (0001) and (11 $\bar{2}$ 0) planes are stable terminations of  $\alpha$ - $\text{Al}_2\text{O}_3$  surfaces  
64 (Lockwood et al., 2008). However, there have been few studies of surface complexation reactions on  
65 (11 $\bar{2}$ 0) plane compared to (0001) and (1 $\bar{1}$ 02) planes.

66 Here, we present a combined GI-EXAFS and DFT calculation investigation to study the structure  
67 and stability of As(V) adsorbed on the (0001) and (11 $\bar{2}$ 0) surfaces of  $\alpha$ - $\text{Al}_2\text{O}_3$ . In the GI-EXAFS  
68 experiment, two directions of the X-ray electric vector, parallel and perpendicular to the substrate,  
69 were used to determine the structure of As(V) bonded to both surfaces. Periodic DFT calculations  
70 were performed to describe the binding and energetic characteristics of arsenate on  $\alpha$ - $\text{Al}_2\text{O}_3$  surfaces.

## 71 **2. Experimental and Computational Methods**

### 72 *2.1. Sample preparation*

73 Highly polished  $25 \times 25 \times 0.43$  mm<sup>3</sup>  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> (0001) and (11 $\bar{2}$ 0) single crystals were obtained  
74 commercially from Shanghai Daheng Optics and Fine Mechanics Co., Ltd, China. The orientations  
75 were checked using High accuracy X-ray single crystal orientation instrument (DX-4A, Dandong,  
76 China) and were found to be perfectly oriented to the resolution limits of the equipment ( $\sim 2$  Å). The  
77 roughness was assessed with AFM (Veeco Dimension 3100) and was found to be  $\leq 5$  Å rms. Prior to  
78 initial use, the crystals were washed in  $10^{-4}$  mol/L nitric acid and ethanol followed by multiple rinses  
79 with MilliQ water (resistivity 18 M $\Omega$ .cm). The crystals were then equilibrated with 1 mmol/L  
80 Na<sub>2</sub>HAsO<sub>4</sub> solution at pH 7.0 for 2h, with a 0.01 mol/L NaNO<sub>3</sub> background electrolyte. The pH of the  
81 reaction system was constantly monitored and adjusted to the desired value of 7.0 with 0.1 mol/L  
82 NaOH or 0.1 mol/L HNO<sub>3</sub>. All chemicals used for solution preparation were of reagent grade quality.

### 83 *2.2. GI-EXAFS data collection and analysis*

84 Grazing-incidence EXAFS experiments were performed on the beamline BL14W1 at the Shanghai  
85 Synchrotron Radiation Facility (SSRF). A purposely built apparatus (Fig. S1 in the Supporting  
86 Information) consists of a carriage that holds the sample stage, with motor drives for positioning the  
87 sample over 5 degrees of freedom (Waychunas et al., 2005). A wet arsenate-loaded crystal sample was  
88 placed on the sample stage for GI-EXAFS measurements. The carriage can be rotated around the  
89 incident X-ray beam to set the angle of the X-ray electric vector polarization plane with respect to the  
90 sample surface. Ionization chamber detectors were mounted on both sides of the sample for  
91 transmission absorption data collection, and a 4-channel Si drift detector (SDD, Canberra Industries,  
92 Inc.) was used to collect fluorescence signals. The angle of the incident X-rays (beam diameter of  $\sim 30$

93  $\mu\text{m}$ ) to the single-crystal surfaces was set to  $0.16^\circ$ , which is less than the critical angle of  $\alpha\text{-Al}_2\text{O}_3$  at  
94 the energy position of As K-edge ( $0.19^\circ$ ) (Klockenkämper, 1997). The experimental uncertainty on the  
95 incident angle of X-rays is  $\leq 0.01^\circ$ . The GI-EXAFS measurements were carried out using two scan  
96 modes with the electric vector respectively parallel and perpendicular to the single-crystal substrate.  
97 The crystal samples were mounted in a Teflon cell and sealed with Mylar film during the GI-EXAFS  
98 experiment. During the data collection, the sample cell was continuously purged with a constant flow  
99 of water-saturated inert gas (ultrapure  $\text{N}_2$ ) to keep the crystals moist, which has been widely used in  
100 GI-EXAFS experiments (Furnare et al., 2005; Waychunas et al., 2005). An average of 3 scans was  
101 performed to achieve suitable signal/noise, and no obvious change in spectral data was observed  
102 during the 3 scans.

103 The spectral data were processed using WinXAS 3.1 software package (Ressler, 1998). A linear  
104 function fit for the pre-edge region and a second-order polynomial fit in the post-edge region were  
105 used to yield the normalized and background-corrected spectra. Subsequently, the normalized spectra  
106 were converted to frequency ( $k$ ) space using a cubic spline function and weighted by  $k^3$ . The  $k^3\chi(k)$   
107 spectra (typically range from 2.2 to  $12.4 \text{ \AA}^{-1}$ ) were Fourier-transformed (FT) to  $R$  space using a Bessel  
108 window function with smoothing parameter of 4. Theoretical phase shift and amplitude functions for  
109 single- and multiple-scattering paths were calculated by ab initio Feff 9.0 code (Ankudinov et al.,  
110 1998) using the cluster of scorodite ( $\text{FeAsO}_4 \cdot 2\text{H}_2\text{O}$ ) with the Fe atom replaced by Al atom. This  
111 method has been successfully used in the study of arsenic species adsorption on Al-(hydr)oxide  
112 surfaces (Arai et al., 2001). An amplitude reduction factor ( $S_0^2$ ) of 0.9 was used in data-fitting  
113 procedure (Grafe et al., 2008; Jing et al., 2005). We used 4, 2, and 1 as the starting values of the As-O  
114 coordination number ( $CN_{\text{As-O}}$ ), the  $CN_{\text{As-Al}}$  for binuclear complex, and the  $CN_{\text{As-Al}}$  for mononuclear

115 complex in the data fitting, to obtain estimated values for interatomic distances ( $R$ ), DW factors  $\sigma^2$ ,  
116 and  $\Delta E_0$ . Then the  $R$ ,  $\sigma^2$ , and  $\Delta E_0$  were fixed to obtain the estimated  $CN$  for each shell. The estimated  
117 values of  $CN$ ,  $R$ ,  $\sigma^2$  and  $\Delta E_0$  were then used for a sequential fitting that recorded the reduction of the  
118 residual until the best fit was obtained. A single  $\Delta E_0$  was applied to all shells and allowed to float  
119 during the fitting. Finally, all the parameters ( $CN$ ,  $R$ ,  $\sigma^2$ ) for each backscattering paths were allowed to  
120 vary. The goodness of fit is evaluated by the residual, defined as

$$121 \quad \text{Residual [\%]} = \frac{\sum_{i=1}^N |y_{\text{exp}}(i) - y_{\text{theo}}(i)|}{\sum_{i=1}^N |y_{\text{exp}}(i)|} \cdot 100$$

122 with  $N$  the number of data points,  $y_{\text{exp}}$  and  $y_{\text{theo}}$  experimental and theoretical data points,  
123 respectively (Ressler, 2009). It has been reported that reasonable EXAFS results were generally  
124 obtained for arsenate adsorption on Fe- and Al-oxides when the residuals were less than 20% (Grafe  
125 and Sparks, 2005; Tang and Reeder, 2009). The As-Al shells were respectively regarded as a single  
126 shell and two subshells, and fitted using one single As-Al scattering path and two different As-Al  
127 paths. The results showed that the residuals of single-shell fits were commonly larger than that of  
128 two-subshell fits (see Fig. S2 in the Supporting Information), which experimentally confirmed the  
129 multiple coordination structures (i.e., the coexistence of two dominant adsorption configurations) of  
130 arsenate on  $\alpha$ - $\text{Al}_2\text{O}_3$  surfaces. The experimental spectra were also fitted using the As-As scattering  
131 path and the As-O-O-As multiple scattering path (Fig. S3 and Fig. S4 in the Supporting Information).  
132 The increasing residuals indicated that the As(V)-bearing solid phase did not exist in the adsorption  
133 system considered here (1 mmol/L, pH 7.0), and no significant improvement was found by adding  
134 multiple scattering effects to the fitting. Therefore, to reduce the number of free parameters during the  
135 fitting, As-As scattering path and multiple scattering path were not employed in the final fitting.

### 136 2.3. Computational settings

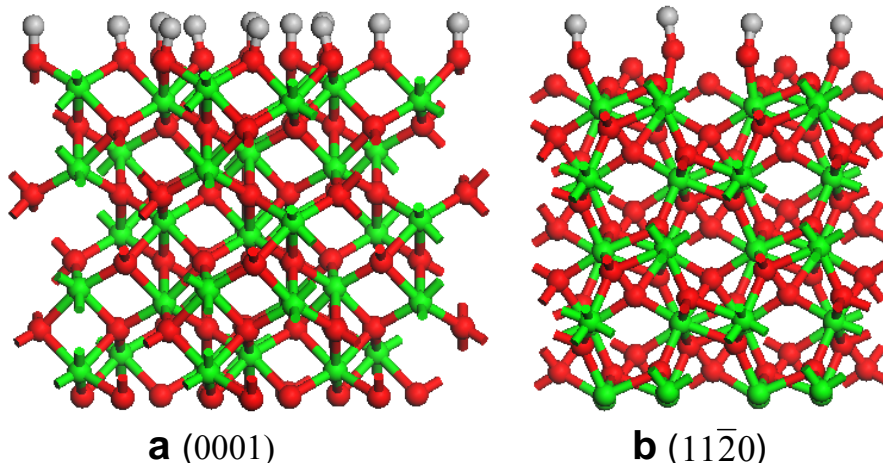
137 The calculations were carried out using the generalized gradient approximation (GGA) with the  
138 functional parameterized by Perdew, Burke and Enzerhof (PBE) (Perdew et al., 1996) as implemented  
139 in the DMol3 module of Materials Studio package (Accelrys Software Inc.) (Delley, 1990; Delley,  
140 2000). In the spin-restricted DFT calculations, the geometrical and electronic structures of Al-AsO<sub>4</sub>  
141 complexes were computed under periodic boundary conditions (PBC). The valence electron wave  
142 functions were expanded using a double numerical plus polarization (DNP) basis set which includes a  
143 polarized d-function for non-hydrogen atoms and p-function for hydrogen atoms (Delley, 2000). The  
144 size of the DNP basis set is comparable to Gaussian 6-31G (d, p), but the DNP is more accurate than a  
145 Gaussian basis set of the same size (Inada and Orita, 2008). The core electrons were described by  
146 effective core potentials (ECP) (Hay and Wadt, 1985). The global orbital cutoff was taken to be 4.8 Å  
147 (fine standard in DMol3). The conductor-like screening model (COSMO) was applied to simulate the  
148 water solvent environment (Klamt, 1995; Klamt and Schuurmann, 1993). The fine quality mesh size  
149 was employed for the numerical integration. The Brillouin zone was sampled with a 2×2×1  
150 Monkhorst-Pack k-point grid for the structural optimization. A Fermi smearing of 0.005 hartree (1  
151 hartree = 27.2114 eV = 2625.5 kJ/mol) was used to improve convergence efficiency (Delley, 1995).  
152 The geometry optimization convergence tolerances of the energy, gradient, and displacement were  
153  $1 \times 10^{-5}$  Hartree,  $2 \times 10^{-3}$  Hartree Å<sup>-1</sup>, and  $5 \times 10^{-3}$  Å, respectively.

### 154 2.4. Computational model

155 The unit cell of α-Al<sub>2</sub>O<sub>3</sub> was optimized before construction of the supercell, and the calculated  
156 lattice constants ( $a = 4.759$  Å,  $c = 12.991$  Å) agreed favorably with the experimental values ( $a = 4.766$   
157 Å,  $c = 13.010$  Å) and other theoretical works (Montanari et al., 2006). The surface in Fig. 1a is the



158 most stable among the three cleavage planes of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> (0001) surface (Mason et al., 2009; Walters et  
159 al., 2000) and was used in this study. From existing studies, a slab contained five repeated Al<sub>2</sub>O<sub>3</sub>  
160 layers generally allows a good description to the surface adsorption of adsorbate molecules (Duarte et  
161 al., 2012; Hellman and Gronbeck, 2009). Therefore, the slab consisting of five Al<sub>2</sub>O<sub>3</sub> layers was used  
162 here as the surface model (see Fig. 1). The supercell expansions and spatial dimensions of each  
163 surface slab are listed in Table 1. Surface hydroxylation of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> will occur in the presence of water,  
164 and forms the reactive surface sites (Hass et al., 1998; Wang et al., 2000). Therefore, the surface of the  
165 slab was hydroxylated in the calculation of arsenate surface complexation. During the geometrical  
166 optimization, the arsenate and the top three Al<sub>2</sub>O<sub>3</sub> layers (i.e., six atomic planes) were allowed to relax,  
167 and the bottom two Al<sub>2</sub>O<sub>3</sub> layers were fixed at equilibrium crystal lattice sites to simulate bulk  
168 conditions (Hellman and Gronbeck, 2009; Ojamae et al., 2006; Ranea et al., 2008). A vacuum  
169 thickness of 15 Å normal to the surface was used to eliminate spurious interactions between the  
170 adsorbate and the periodic image of the bottom layer of the slabs (Roscioni et al., 2013). A counter  
171 positive charge was automatically added by DMol3 package during the calculations in order to keep  
172 the neutrality of the whole model to avoid divergence in the electrostatic energy (Makov and Payne,  
173 1995).



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175 **Fig. 1.** Side views of the hydroxylated  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> (0001) (a) and ( $11\bar{2}0$ ) (b) surface models used in DFT  
 176 calculation. Green, red, and gray circles denote aluminum, oxygen, and hydrogen atoms, respectively.

177 **Table 1.** Supercell expansions and lattice parameters used to construct  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> (0001) and ( $11\bar{2}0$ )  
 178 surface models

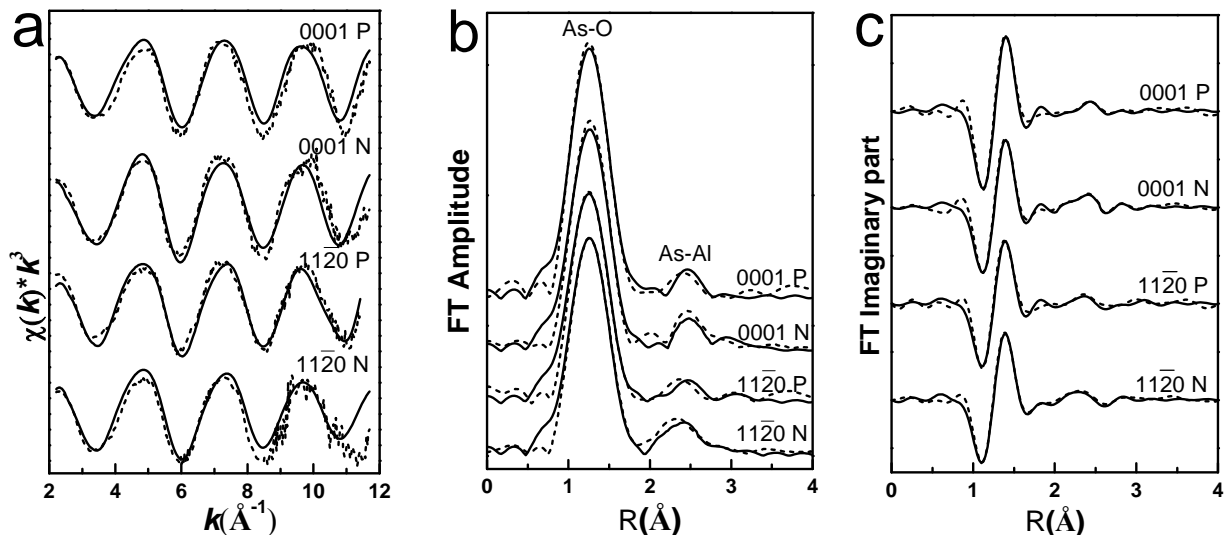
Surface	Expansion	$L_x$ (Å)	$L_y$ (Å)	$L_z$ (Å)	Area/Å <sup>2</sup>	$N_{\text{Al}_2\text{O}_3}$
(0001)	2×2	9.52	9.52	10.83	90.63	22
( $11\bar{2}0$ )	1×2	7.00	10.26	11.17	71.82	20

179 Test calculations with larger orbital cutoff (5.2 Å) and denser  $k$ -point sampling ( $3\times 3\times 2$ ) show  
 180 almost no change in the structural and energetic properties. Negligible difference to the adsorption  
 181 geometries (<0.01 Å) and binding energies (<5 kJ/mol) was found when the relaxed Al<sub>2</sub>O<sub>3</sub> layers  
 182 increased from 3 to 4 (see Table S1 in the Supporting Information), indicating that the structural  
 183 parameters and relative energies of arsenate adsorption have converged when the relaxed Al<sub>2</sub>O<sub>3</sub> layers  
 184 increased to 3 (i.e., six atomic planes). These tests verified that the present computational settings and  
 185 models were reliable for describing the properties of arsenate on  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> surfaces.

### 186 3. Results and Discussion

#### 187 3.1. GI-EXAFS analysis

188 The  $k^3$ -weighted and Fourier transform spectra of As K-edge GI-EXAFS for the adsorption samples  
189 of arsenate on  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> (0001) and (11 $\bar{2}$ 0) single-crystal surfaces are shown in Fig. 2. GI-EXAFS  
190 fitting data are presented in Table 2. Based on our fitting results, we estimated the accuracies of the  
191 As–O shell to be  $\pm 0.005$  Å for interatomic distance ( $R$ ) and  $\pm 5\%$  for coordination number ( $CN_{As-O}$ ).  
192 The estimated errors were  $\pm 0.02$  Å,  $\pm 0.03$  Å and  $\pm 20\%$  for  $R$  of As–Al first subshell,  $R$  of As–Al  
193 second subshell and  $CN_{As-Al}$ , respectively. The GI-EXAFS fitting results showed that the first  
194 coordination shell of the arsenic atom consisted of four oxygen atoms at a distance of  $\sim 1.69$  Å, which  
195 is in good agreement with the literature (He et al., 2009; Waychunas et al., 2005). As a result of the  
196 high symmetry of AsO<sub>4</sub> tetrahedron, no obvious difference in  $CN_{As-O}$  (Table 2) was observed between  
197 both polarized scans (electric vector parallel and perpendicular to the substrate). For (0001) plane, the  
198 two As–Al distances obtained from GI-EXAFS analysis ( $3.11 \pm 0.02$  and  $3.33 \pm 0.03$  Å) agreed well with  
199 DFT-calculated values of bidentate binuclear (BB) complex ( $3.07$  Å) and monodentate binuclear (MB)  
200 complex ( $3.30$  Å), respectively (Table 3). The two experimentally measured As–Al distances on (11 $\bar{2}$ 0)  
201 plane ( $3.04 \pm 0.02$  Å and  $3.20 \pm 0.03$  Å) were consistent with the theoretical values of BB ( $3.07$  Å) and  
202 monodentate mononuclear (MM,  $3.21$  Å) complexes (see Table 3).



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**Fig. 2.** (a)  $k^3$ -weighted GI-EXAFS, (b) Fourier transformed (FT) spectra, and (c) imaginary part of FT spectra of arsenate adsorption on  $\alpha$ - $\text{Al}_2\text{O}_3$  (0001) and ( $11\bar{2}0$ ) planes measured using two scan modes [electric vector parallel (P) and normal (N) to the substrate]. The dashed lines are the experimental spectra, while the solid lines are fitting curves. The peak positions are uncorrected for phase shifts.

**Table 2.** GI-EXAFS measured structural parameters of arsenate adsorption on  $\alpha$ - $\text{Al}_2\text{O}_3$  (0001) and ( $11\bar{2}0$ ) single-crystal planes\*

Shell	Type of neighbor	CN	$R(\text{\AA})$	$DW(\sigma^2)$	$\Delta E_0(\text{eV})$	Residual*** (%)
<b>(0001) surface E-vector** parallel</b>						
First	O	$4.0 \pm 0.2$	$1.68 \pm 0.005$	$0.002 \pm 0.001$	2.23	10.2
Second	Al	$1.5 \pm 0.3$	$3.11 \pm 0.02$	$0.004 \pm 0.001$		
	Al	$0.7 \pm 0.2$	$3.32 \pm 0.03$	$0.007 \pm 0.002$		
<b>(0001) surface E-vector normal</b>						
First	O	$4.0 \pm 0.2$	$1.69 \pm 0.005$	$0.003 \pm 0.001$	3.70	11.3
Second	Al	$2.9 \pm 0.6$	$3.12 \pm 0.02$	$0.005 \pm 0.001$		
	Al	$4.2 \pm 0.8$	$3.35 \pm 0.03$	$0.006 \pm 0.003$		
<b>(<math>11\bar{2}0</math>) surface E-vector parallel</b>						
First	O	$4.1 \pm 0.2$	$1.68 \pm 0.005$	$0.003 \pm 0.001$	3.45	11.4
Second	Al	$1.4 \pm 0.3$	$3.03 \pm 0.02$	$0.005 \pm 0.002$		
	Al	$0.4 \pm 0.1$	$3.19 \pm 0.03$	$0.007 \pm 0.003$		
<b>(<math>11\bar{2}0</math>) surface E-vector normal</b>						
First	O	$3.9 \pm 0.2$	$1.69 \pm 0.005$	$0.002 \pm 0.001$	4.65	11.0
Second	Al	$3.7 \pm 0.7$	$3.05 \pm 0.02$	$0.006 \pm 0.002$		

	Al	2.6± 0.6	3.22± 0.03	0.006±0.003
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210 \* The listed parameters ( $CN$ , coordination number;  $R$ , interatomic distance;  $\sigma^2$ , Debye-Waller factor) reflected the  
 211 final best fit.

212 \*\* E-vector, electric vector.

213 \*\*\* Residual, gives a measure of the agreement between experimental and theoretical FT curves. All Residuals are <  
 214 12%, indicating reliable fits.

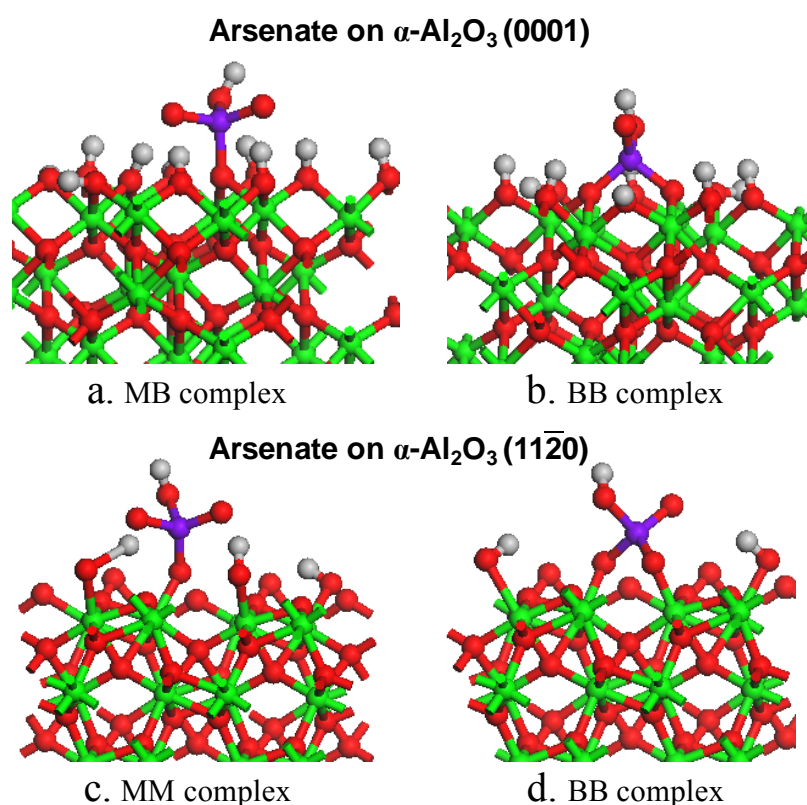
215 An obvious difference in  $CN_{As-Al}$  was observed from the same sample between the two scan modes  
 216 (see Table 2), resulting from the polarization dependence of As-Al bonds. Different to the  $CN_{iso}$   
 217 determined by standard powder EXAFS, the polarized coordination number ( $CN_{pol}$ ) obtained from  
 218 GI-EXAFS varies with the angle between the incident electric vector and the bond (Waychunas et al.,  
 219 2005). The polarization dependence allows a maximum of  $CN_{pol}$  when the bond parallel to the direction  
 220 of electric vector, and a near-zero value would appear when the bond normal to the electric vector in the  
 221 GI-EXAFS measurement. For instance, Fitts *et al.* presented that the  $CN_{pol}$  of Cu-Si bond on SiO<sub>2</sub> (0001)  
 222 surface in monodentate complex was 0.3 when the electric vector parallel to the substrate (Fitts et al.,  
 223 1999). It was also reported that a zero  $CN_{pol}$  value of Co-O bond occurred in an on-top complex of Co  
 224 (II) adsorbed on  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> (0001) surface when the bond was absolutely normal to the electric vector  
 225 (Shirai et al., 1992; Shirai et al., 1994). The difference of  $CN_{As-Al}$  between both polarized scans can be  
 226 used to deduce the polar angle ( $\phi$ , angle between the As-Al vector and the substrate normal), and to  
 227 determine the stereostructure of each surface complex. For the two scan modes (electric vector parallel  
 228 and perpendicular to the support plane), the relationships between the coordination numbers of a  
 229 polarized EXAFS measurement ( $CN_{pol}$ ) and that of isotropic EXAFS ( $CN_{iso}$ ) are  
 230  $CN_{pol}^{\perp} = CN_{iso} \times 3 \cos^2 \phi$  and  $CN_{pol}^{\parallel} = \frac{3}{2} CN_{iso} \times \sin^2 \phi$ , respectively (Manceau et al., 2002; Waychunas  
 231 et al., 2005). Based on the two equations, the polar angles ( $\phi$ ) of BB complex obtained from GI-EXAFS  
 232 fitting were  $45^{\circ} \pm 8^{\circ}$  on (0001) surface and  $36^{\circ} \pm 6^{\circ}$  on (11 $\bar{2}$ 0) surface, while the experimental  $\phi$  of MB

233 complex on (0001) surface and MM complex on (11 $\bar{2}$ 0) surface were  $33^\circ \pm 8^\circ$  and  $18^\circ \pm 11^\circ$  respectively.  
234 The discrepancies between experimental and DFT-calculated polar angles (see Table 3) were less than  
235  $10^\circ$ , indicating that a good estimation of the polar angles was obtained.

### 236 3.2. DFT calculation

237  $\text{HAsO}_4^{2-}$  is the dominant species of arsenate under pH 7.0 (i.e., the GI-EXAFS experimental condition)  
238 (Sadiq, 1997), and hence was used in the DFT calculation to determine the adsorption site preference.  
239 EXAFS signal is less influenced by the hydrogen atoms than by the heavier elements around the  
240 absorbing atoms, and thus hydrogen atoms are normally not included in analyses of EXAFS data. Here,  
241 we used DFT calculation to identify the protonation state of adsorbed arsenate. Monodentate binuclear  
242 (MB) and bidentate binuclear (BB) complexes are the two typical structures of arsenate adsorbed on the  
243 (0001) surface (see Fig. 3). On the (11 $\bar{2}$ 0) surface, there are two kinds of unsaturated oxygen sites,  
244 monodentate-terminal and bidentate-bridging O sites, which may be potentially formed the reactive sites.  
245 Arsenate would be bonded as monodentate mononuclear (MM) and bidentate binuclear (BB) structures  
246 at the terminal oxygen sites (Fig. 3), while a bidentate mononuclear (BM) adsorption mode may be  
247 potentially produced at the bridging oxygen sites (Fig. S5a). However, the calculated BM adsorption  
248 mode yielded an As-Al distance of 2.75 Å (see the Supporting Information), much smaller than the  
249 distance obtained from experimental GI-EXAFS measurement, which confirmed the fact that the  
250 twofold bridging oxygen site is less active than the high coordination-unsaturated terminal oxygen site  
251 at the  $\alpha\text{-Al}_2\text{O}_3$  (11 $\bar{2}$ 0) surface. Outer-sphere adsorption of arsenate on  $\alpha\text{-Al}_2\text{O}_3$  at pH 5 has been  
252 reported (Catalano et al., 2008). It is well known that the surface complexation of arsenate on  
253 metal-(hydr)oxides is significantly affected by pH. In our previous study of arsenate adsorption on  
254 anatase  $\text{TiO}_2$  surfaces (He et al., 2009), we also found that the as pH decreased to 5.5, outer-sphere

255 adsorption became thermodynamically favorable. However, there was no evidence indicating the  
256 occurrence of outer-sphere adsorption of arsenate on  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> surfaces under pH 7. Our DFT calculation  
257 showed that the As-Al distances would be  $\sim 5$  Å in outer-sphere Al-AsO<sub>4</sub> complexes (Fig. S5b), however,  
258 no effective signals were observed more than 4 Å away from the As atom in the GI-EXAFS spectra (see  
259 Fig. S2). Furthermore, outer-sphere H-bonded adsorption of arsenate on  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> surfaces [e.g., -251.5  
260 kJ/mol on (0001) surface] was more thermodynamically unfavorable than the inner-sphere adsorption  
261 (-434.7 and -584.3 kJ/mol, see Table 3). This result indicated that arsenate adsorbed on the  $\alpha$ -Al<sub>2</sub>O<sub>3</sub>  
262 surfaces mainly as inner-sphere complexes under the neutral pH condition. Therefore, in order to  
263 directly compare calculated results with experimental data, complexation at the bridging oxygen sites on  
264 (11 $\bar{2}$ 0) surface and outer-sphere adsorption were excluded from the theoretical analysis and discussion.



265  
266 **Fig. 3.** DFT-calculated structures of arsenate adsorbed on  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> (0001) and (11 $\bar{2}$ 0) surfaces. Purple,  
267 green, red, and gray circles denote arsenic, aluminum, oxygen, and hydrogen atoms, respectively.

268 Adsorption on the (0001) facet (-434.7 kJ/mol for monodentate complexes and -584.3 kJ/mol for  
 269 bidentate complexes) were generally energetically more favorable than that on the (11 $\bar{2}$ 0) facet  
 270 (-172.7 kJ/mol for monodentate complexes and -339.5 kJ/mol for bidentate complexes), indicating  
 271 that the (0001) facet exhibited a higher affinity toward arsenate. The spectral and computational  
 272 results showed that the polar angles of monodentate and bidentate complexes on the (0001) surface  
 273 were respectively 10~20° greater than that on the (11 $\bar{2}$ 0) surface (see Table 3). The result suggested  
 274 that a relatively large polar angle was more favorable for arsenate stabilized on  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> surfaces. It  
 275 can be also noted that the DFT-calculated As-O distances for the Al-AsO<sub>4</sub> surface complexes (1.71 Å)  
 276 were generally slightly longer than the experimentally measured values (1.69 Å). This phenomenon  
 277 was also observed in the adsorption of arsenite on Fe-(hydr)oxides (Zhang et al., 2005). This  
 278 difference between experimental and theoretical values may be due to the theoretical underestimation  
 279 of solvent effect from COSMO model (Contentin et al., 2004; Tossell, 2005). However, we expect that  
 280 this artifact does not affect the relative stability of the adsorption modes and their complexation  
 281 properties.

282 **Table 3.** DFT-calculated interatomic distances (Å), polar angles (°) and binding energies ( $\Delta E$ ) of  
 283 arsenate adsorbed on  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> surfaces

Bond	As-O(H)				Average	As-Al			Polar angle*		Average	$\Delta E^{**}$ (kJ/mol)
MB-(0001)-HAsO <sub>4</sub> <sup>2-</sup>	1.69	1.69	1.73	1.75	1.71	3.14	3.45	3.30	30	30	30	-434.7
BB-(0001)-HAsO <sub>4</sub> <sup>2-</sup>	1.69	1.71	1.72	1.74	1.71	2.98	3.17	3.07	42	40	41	-584.3
MM-(11 $\bar{2}$ 0)-HAsO <sub>4</sub> <sup>2-</sup>	1.68	1.69	1.74	1.77	1.72	3.21	—	3.21	10	—	10	-172.7
BB-(11 $\bar{2}$ 0)-HAsO <sub>4</sub> <sup>2-</sup>	1.69	1.70	1.72	1.75	1.71	3.01	3.13	3.07	33	30	32	-339.5

284 \* Polar angle is the angle between the surface normal and the interatomic As-Ti vector direction.



285 \*\*  $\Delta E$  were calculated as  $\Delta E = E_{\text{tot}}(\text{Al-AsO}_4) - [E_{\text{tot}}(\text{arsenate}) + E_{\text{tot}}(\alpha\text{-Al}_2\text{O}_3)]$ , where  $E_{\text{tot}}(\text{Al-AsO}_4)$  was the total energy  
286 of Al-AsO<sub>4</sub> adsorption complex,  $E_{\text{tot}}(\text{arsenate})$  and  $E_{\text{tot}}(\alpha\text{-Al}_2\text{O}_3)$  were the total energy of arsenate molecule and  
287  $\alpha\text{-Al}_2\text{O}_3$  cluster, respectively. The structures of non-adsorbed arsenate were optimized in a periodic box with a side  
288 length of 10 Å (Ojamae et al., 2006).

## 289 4. Conclusions

290 The spectral and computational results showed that arsenate primarily bonded as inner-sphere MB  
291 and BB complexes on the (0001) surface of  $\alpha\text{-Al}_2\text{O}_3$ , and MM and BB complexes on the (11 $\bar{2}$ 0) surface.  
292 The orientational polar angles on the (0001) surface were commonly 10~20° greater than that on the  
293 (11 $\bar{2}$ 0) surface for both monodentate and bidentate complexes. The DFT calculation showed that the  
294 (0001) surface exhibited a stronger affinity toward arsenate, suggesting that the large polar angle was  
295 more favorable for arsenate stabilization. The dominant bonding modes of arsenate on the two planes of  
296  $\alpha\text{-Al}_2\text{O}_3$  were identified as bidentate binuclear structures.

297 The combination of GI-EXAFS and DFT calculation facilitates the direct identification of adsorbed  
298 species on different exposed crystal planes of the mineral. Our results demonstrated the arsenate  
299 complexes on different crystal planes, which may be regarded as one single adsorption structure from  
300 the measurements based on powder substrates, exhibited an obvious difference in stability. This  
301 unambiguous characterization suggested a potential development to predict the fate and transport of  
302 arsenic in aluminum-bearing soils, sediments, and water treatment systems that should take into account  
303 the contributions from different crystal surfaces.

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## 309 **Appendix A. Supplementary data**

310 Supplementary data to this article can be found online at

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