

Doping stability and opto-electronic performance of CVD graphene on transparent flexible substrates

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Abstract

The primary barrier to wider commercial adoption of graphene lies in reducing the sheet resistance of the transferred material without compromising its high broad-band optical transparency, ideally through the use of novel transfer techniques and doping strategies. Here, chemical vapour deposited graphene was uniformly transferred to polymer supports by thermal and UV approaches and the time-dependent evolution of the opto-electronic performance was assessed following exposure to three kinds of common graphene dopants. Doping with FeCl₃ and SnCl₂ showed minor, and notably time unstable, enhancement in the $\sigma_{\text{opt}}/\sigma_{\text{dc}}$ figure of merit, whilst AuCl₃-doping markedly reduced the sheet resistance by 91.5% to 0.29 k Ω /sq for thermally transferred samples and by 34.4% to 0.62 k Ω /sq for UV

transferred samples offering a means to fabricating viable transparent flexible conductors that near the indium tin oxide benchmark.

Keywords: graphene, chemical vapour deposition, flexible transparent electrode, doping, sheet resistance, optical transmittance, lamination

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Introduction

Graphene, a two-dimensional honeycomb of single carbon atoms, has a myriad of impressive, novel optical and electrical properties [1, 2]. As such, it has attracted much attention as a promising material to complement indium tin oxide in large area transparent electrodes for many applications including organic light emitting diodes, [3-6] touch screens, [5, 7, 8] and photovoltaic cells. [9-11] Graphene is mechanically flexible, does not readily form micro-cracks when strained and, when coupled with suitable substrates, provides an exciting and paradigm shifting technological platform for future flexible transparent electronic devices. Though mechanically and chemically exfoliated graphene, and inks based thereon, have made some progress towards truly flexible, high form factor, transparent conductors, the intrinsic disorder and sub-micron grain sizes associated with such approaches limit their

practicality. [12-15] Chemical vapour deposition (CVD) has thusly supplanted exfoliation as the fabrication procedure of choice, particularly given recent technological advances and associated cost reductions in growth allowing for truly monolayer material of near-equivalent quality to mechanically exfoliated graphene along with substantially higher yields. [16-21] Unlike exfoliated approaches, CVD graphene is large-area compatible and offers electrical continuity and optical uniformity without the use of aggressive chemical treatments. Nevertheless, for CVD graphene to be industrially viable, it must be transferred to arbitrary substrates from the optically opaque catalyst on which it was grown. Though the native properties of graphene are central to its function in a wide range of applications, the transfer method plays an equally important role in realizing useful devices. Several transfer methods have been reported with hot and cold lamination being two such methods that have shown significant promise due to their facile processing, low-cost and large-area compatibility.

Polymers such as poly-methyl methacrylate (PMMA) and polydimethylsiloxane are commonly used to mediate the transfer process. Here, the polymer supports the graphene while the catalyst is etched. [22-26] The PMMA/graphene stack is then located onto an arbitrary target substrate and the PMMA removed, typically in an acetone bath. [22-27] Though the value of this technique to the global research community is unrivalled, it lacks the fundamental commercial scalability required for an emerging material to gain industrial traction. Handling the polymer-support, on areas greater than a few centimetres square, is

challenging and requires significant training and associated expense [22, 28, 29]. Alternatives must be sought. In 2010 Bae *et al.* transferred large-area CVD graphene using thermal release tape.[5] In this study Cu-foil-catalysed CVD graphene was attached to thermal release tape and the Cu etched in aqueous ammonium persulphate ((NH₄)₂S₂O₈). The graphene was then transferred using a roll-to-roll system and the thermal release tape removed by heating to 100°C. [5] This approach allowed graphene transfer, layer-by-layer, onto arbitrary substrates. Nevertheless, this method cannot be applied to many low-cost and widely available polymeric substrates given the limited thermal budget of many such materials and the potentially high mismatch in thermal expansion coefficient between the graphene and the substrate which may induce crack formation, thereby compromising the films opto-electronic properties.

To overcome these drawbacks we have investigated alternative transfer techniques and have studied the temporal variation of the opto-electronic properties of facile and mechanically stable thermal lamination and UV-curable adhesion transferred films. At present the sheet resistance (R_s) of single-layer graphene is of the order of a few kΩ/sq; an order of magnitude too high to be an effective transparent conductive electrode in most applications. Chemical doping is considered the most appropriate means to decrease R_s without significantly compromising the optical transmission. [30-36] In this study we investigated the opto-electronic spatial uniformity and time dependent behaviour of chemically doped

polymer-supported graphene films under ambient conditions following exposure to three common chloride-based graphitic nanocarbon dopants; ferric chloride (FeCl_3), tin chloride (SnCl_2) and gold chloride (AuCl_3).

Experimental

Graphene was grown, as reported in detail elsewhere [37], using a commercially available Aixtron Black Magic Pro, hot-walled thermal CVD system on 25 μm Cu foil (99.999% Alfa Aesar) under 5 sccm CH_4 (99.5%) at 1000°C in Ar: H_2 (960 (99.9997%): 40 (99.9992%) sccm) at 25 mbar. Following 15 minutes growth, samples were quenched under 2000 sccm N_2 (99.99%) to 250°C and were removed from the reactor. Graphene grains were approximately 10-100 μm in diameter with a mean area of 400 μm^2 which coalesced to form a large area, few layer polycrystalline graphene film. As-grown graphene was inspected by selected area electron diffraction (SAED) recorded using a FEI Philips Tecnai operated at 200 keV. The SAED patterns were assessed relative to a thallium chloride calibration standard. Weak patterns from the supporting grid and residual organics were present. Six-fold symmetry, characteristic of the graphites and graphenes, was clearly noted. It is possible that the multiple, unassigned spots are associated with turbostratic alignment, though film discontinuity and polycrystallinity makes verification of this challenging. Certainly the $\bar{1}010$ to $\bar{2}110$ intensity ratio ($I_{\bar{1}010} / I_{\bar{2}110}$), deviates somewhat from that of A-A or A-B (Bernal)

stacking, suggesting a turbostratic multi-layered film [38-40].

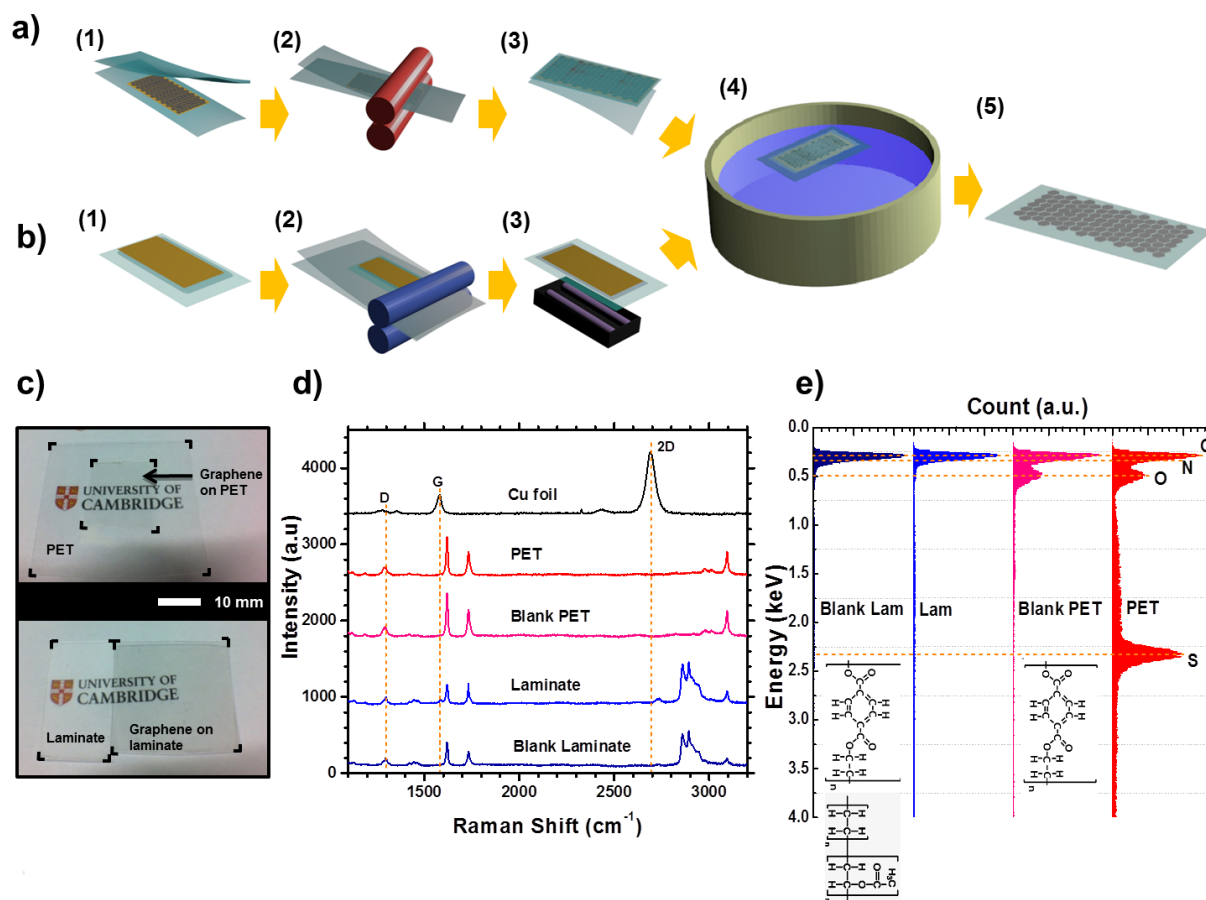


Fig. 1: Schematic depiction of graphene transfer by; a) thermal lamination, and b) UV-curable adhesive. c) Optical images of graphene transferred on PET (top) and laminate (bottom) (Scale bar: 10 mm). d) Raman spectra (457 nm) of the nascent Cu-foil catalysed graphene by thermal CVD, as-received PET and laminate substrates, and transferred graphene on PET and laminate. e) Energy-dispersive X-ray spectra of a blank laminate, a blank PET, and transferred graphene on Laminate and PET.

To hot-press laminate the graphene-on-catalyst samples, commercially available thermally-activated ethylene vinyl acetate (EVA) treated polyethylene terephthalate (PET) substrates were used (GBC Co.). This hot-press lamination process, termed ‘laminate’ hereafter, is illustrated in **Figure 1a**). First, graphene grown on Cu foil was sandwiched between two laminates (1) and passed through a dual roller laminator, heated to 120°C (2).

Following thermal lamination, the backside laminate was detached (3) and the Cu etched in $(\text{NH}_4)_2\text{S}_2\text{O}_8$ in DI water (1M) for 12 hours (4). Following the etching, samples were rinsed with DI water and dried in high-purity N_2 (5). We have transferred areas of up to 10 cm x 10 cm successfully using this method. **Figure 1b)** describes the UV-cured adhesive transfer approach. The UV-curable adhesive was first coated onto acetone cleaned PET (1). The as-grown graphene on Cu was then placed on the adhesive and pressed at around 0.2 MPa, ensuring that all air is removed from the graphene-adhesive interface (2). Plastic protective films were used to cover both sides of the Cu foil and the PET substrate. To cure the adhesive, the backside of the PET substrate was exposed to a UV optical source (365 nm, 222 Wm^{-2}) for 15 minutes (3). Following UV curing the Cu foil was etched in aqueous $(\text{NH}_4)_2\text{S}_2\text{O}_8$ for 12 hours (4) and rinsed in DI water and dried in high-purity N_2 as before (5). Both transfer methods are rapid, inexpensive, large-area compatible and provide strong, long-lasting robust adhesion between the substrate and the graphene – critical for high form factor electronics. **Figure 1c)** shows optical micrographs of transferred samples.

Results

Figure 1d) shows a typical Raman spectrum (457 nm, Renishaw InVia) of the as-grown CVD graphene. The I_D/I_G ratio was 0.12 ± 0.05 suggesting a highly crystalline, well-graphitised material [41, 42] with an I_{2D}/I_G ratio of 2.33 ± 0.6 indicating bilayers. Repeated point measurements over large-areas suggested the presence, on average, of 3-4 layers. [14,

41, 42] The full-width at half-maximum (FWHM) of the 2D peak was narrow; 45-55 cm^{-1} , further suggesting high graphitisation. [41, 42] Additional spatially resolved Raman analysis, under 532 nm and 633 nm excitation, revealed a $\langle I_{2D}/I_G \rangle$ of 1.3 - 2.2, with the number of layers per unit area being relatively uniform with 79% coverage of 3-4 graphene layers, 14% bilayer, and 7% monolayer. Raman spectra of the transferred graphene on laminate and PET showed no notable D, G or 2D peaks with the characteristic graphene spectra being indistinguishable from that of the polymer support. Nevertheless, Energy-dispersive X-ray (EDX) analysis of the blank laminate, blank PET, graphene-on-laminate (Laminate) and graphene-on-PET (PET) samples, as shown in **Figure 1e**, evidences the distinct absence of any Cu suggesting that only the transferred carbon (C) from the as-grown graphene mediates the observed conductivity enhancement. The conductive carbon allotrope is most likely graphitic given the comparatively low transfer temperatures. EDX also revealed, in the case of the graphene-on-PET, significant sulphur (S) and oxygen (O) peaks, which we believe are most likely from the $(\text{NH}_4)_2\text{S}_2\text{O}_8$ exposure during the Cu-etch. Interestingly, such peaks were not observed for the laminate samples. Sulphur is a well-known potent dopant of graphitic nanocarbons. [43, 44] Evidently the Cu etching produces significant unintentional, but nevertheless advantageous doping prior to any further chemical treatments. **As discussed in detail later, this doping degrades with time.** This is attributed to thermally activated desorption of the weakly surface bound species. [36, 45-48]

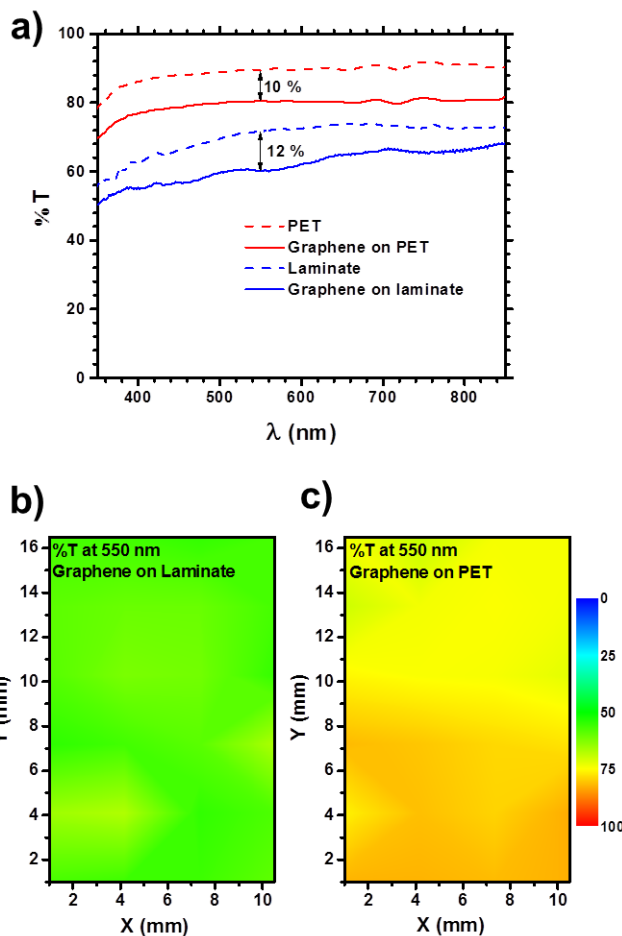


Fig. 2: a) Typical optical transmittance spectra of graphene transferred to Laminate and PET. Optical transmittance maps (550 nm) of graphene on; b) laminate and c) PET.

Figure 2a) shows the optical transmittance spectra (%T) for graphene transferred to laminate and PET substrates (ATI, Unicam UV2). The transmittance (550 nm) of graphene on PET and on laminate was 10% and 12% lower than the as-received PET and laminate, respectively. The high optical absorption suggests around four layer coverage on average on for the PET samples and five layer coverage for the laminate [49, 50], consistent with earlier Raman maps of the nascent material. As shown in **Figure 2b) and c)**, the spatially averaged

transmittance of the graphene/laminate and graphene/PET at 550 nm was $58.6\pm 3.6\%$ and $76.5\pm 3.8\%$, respectively. The PET transfer was around 18.1% more transparent than the laminate, though both transfer techniques afforded equivalent areal uniformity of $< 4.0\%$. The modest increase in absorption between the two samples is likely due to folding and wrinkling of the graphene during the transfer process, as clearly depicted in the SEM micrographs (Hitachi, S4700SEM) of **Figure 3a) and b)** and the AFM micrographs (Agilent, 5400SPM) in **Figure 3c) and d)**. Certainly hot-press lamination is more aggressive in terms of augmenting the morphology of the as-grown graphene. Indeed, the root mean square (RMS) surface roughness of the graphene on the laminates was 161 nm, some 26% higher than the PET (119 nm). The maximum perturbation for the graphene on laminate and PET was 949 nm and 551 nm, respectively. As eluded to prior, wrinkling increases the optical diffusivity but will, conversely, enhance the mechanical flexing performance of the electrodes.

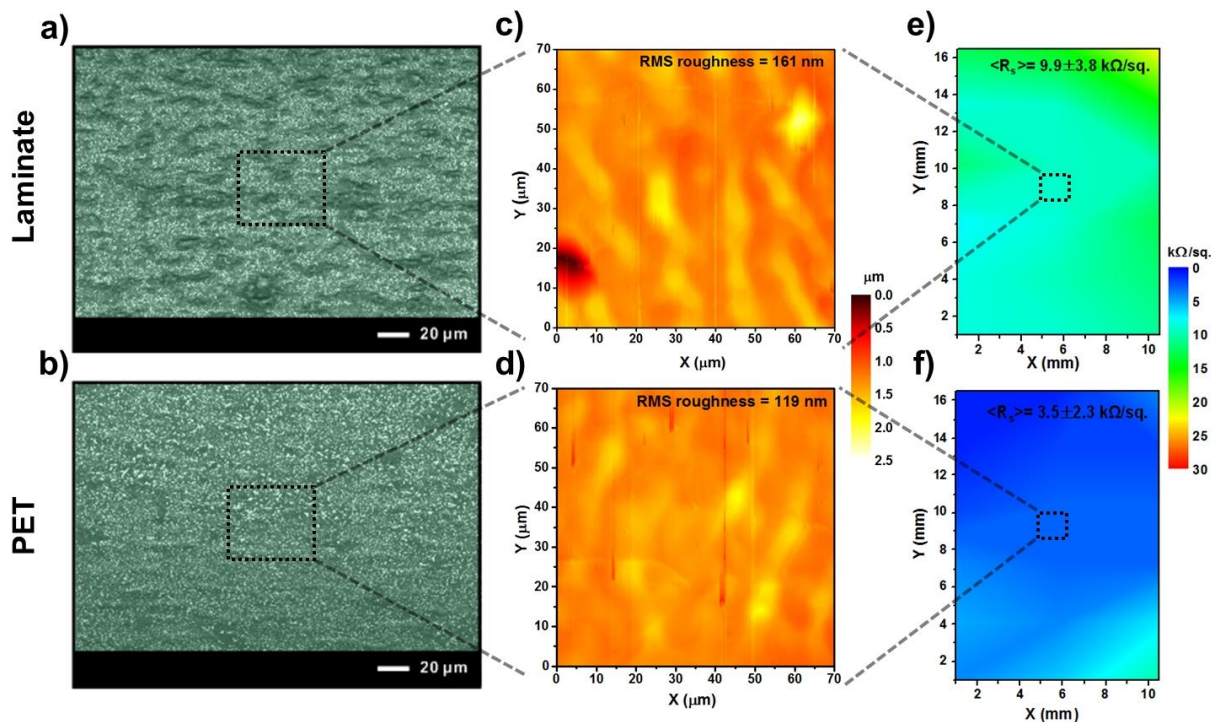


Fig. 3: Scanning electron micrographs of the transferred graphene on; a) Laminates and b) PET (Scale bar 20 μm). Atomic force micrographs of graphene on; c) Laminates and d) PET (Scale bar 10 μm). Spatially resolved sheet resistance of graphene on; e) Laminates and f) PET.

Prior to the metal-chloride doping the sheet resistance for the as-grown graphene-laminates samples was $9.9 \pm 3.8 \text{ k}\Omega/\text{sq}$, whilst for the PET it was $3.5 \pm 2.3 \text{ k}\Omega/\text{sq}$ (Figure 3e and f); PET samples showed a markedly lower R_s due to the notable sulphur doping. The growth and transfer process showed high reproducibility though some slight variation in sheet resistance and transmission between samples was observed. The sheet resistance of the as-grown graphene was assessed independently by transferring equivalent as-grown graphene to quartz substrates, using the conventional PMMA-approach, with subsequent Cr/Au Van der Pauw structures deposited. This as-grown graphene showed a sheet resistance of $5.47 \pm 1.2 \text{ k}\Omega/\text{sq}$. The laminates and UV-transferred samples showed an increase

of around 4.43 k Ω /sq and, rather interestingly, a decrease of 1.97 k Ω /sq relative to the conventional PMMA approach control sample, respectively. This decrease may be attributed to enhanced mass density and increased connectivity within the percolative network associated with significant wrinkling and augmentation of the as-grown morphology. Indeed, the transferred graphene on laminate and PET are non-contiguous. The agglomerates on the PET and laminate are approximately 5.5 ± 5.6 and 7.9 ± 3.9 μm in diameter and are most likely adhesive residues.

Although stacking graphene to form artificial multi-layer materials is one possible way to decrease the sheet resistance, it has many intrinsic limitations; time and cost being but two. Alternatively, intentional chemical doping is one simple route to decrease the sheet resistance without the necessary complicated process and severe degradation of the optical transparency. In this study, to reduce the sheet resistance to a technologically relevant value (< 0.2 k Ω /sq) chemical doping via aqueous metal-chloride exposure of the UV and thermally transferred samples was investigated.

To assess the doping time stability three common nanocarbon dopants were considered; ferric chloride (FeCl₃), tin chloride (SnCl₂) and gold chloride (AuCl₃). FeCl₃ and SnCl₂ doping solutions were dissolved in DI water at 20 mM concentration, whilst AuCl₃ was dissolved in nitromethane (20 mM). Laminate and PET samples were dipped into the dopant solutions for 5 minutes and dried with N₂ gas. For doping with AuCl₃, as elsewhere [31, 33,

36], the dopant solution was spin-coated at 2000 rpm for 1 minute. Nitric Acid (HNO_3), which has been previously reported as a p -type dopant [32, 34, 35], was also considered. However, the HNO_3 dissolved the adhesion layer in the PET samples resulting in the removal of the graphene. Adhesion in the laminate samples were unaffected by the HNO_3 and showed only a negligible decrease in R_s from 12.96 $\text{k}\Omega/\text{sq}$ to 12.5 $\text{k}\Omega/\text{sq}$, whilst the %T decreased from 71.5% to 61.8%.

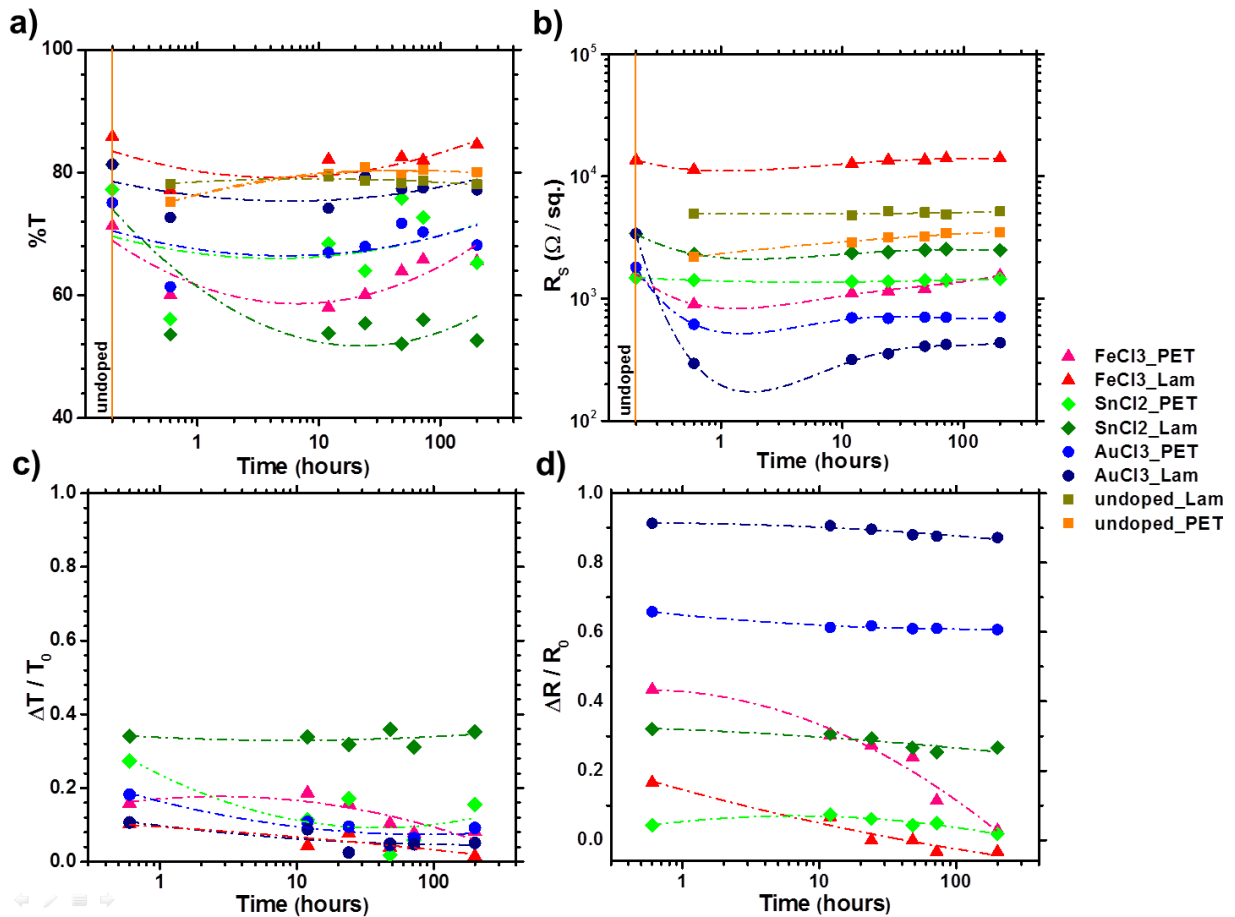


Fig. 4: Time-dependent electrical and optical properties of chemically doped graphene on laminate and PET substrates under ambient conditions: a) Optical transmittance (%T), b) sheet resistance (R_s) of doped graphene, c) Ratio of the change in transmittance (ΔT) relative to the transmittance of undoped graphene (T_0) and d) the ratio of sheet resistance relative to the sheet resistance of undoped graphene (R_0).

To assess the time evolution of the doping the optical transmittance (550 nm) and sheet resistance were measured immediately after doping and at fixed time points thereafter, as illustrated in **Figure 4a) and b)**, respectively. An ideal transparent conductor requires time and ambient invariant transmittance and sheet resistance. However, as reported elsewhere [31, 51], although dopants reduce the sheet resistance they also often reduce optical transmission, both of which notably shift deleteriously with time – an effect which is particularly exacerbated during ambient exposure. For all samples the transmittance decreased following doping. The most significantly decreased transmittance (81→54%) occurred for SnCl₃-laminated, whereas FeCl₃- and AuCl₃-doped graphene showed smaller changes (FeCl₃-laminated: 86→77%, AuCl₃-laminated: 81→73%). A recovery process was noted with the transmittance tending to increase back to the undoped state with time. The most substantial increase was observed for the FeCl₃-doped graphene (laminated: 77→85%, PET: 60→66%). This increase is presumed to be due to the time-dependent desorption of chemisorbed dopants, activated by ambient thermal excitation. [36, 45-48] Interestingly, the transmittance of the undoped graphene on PET increased over time (75→80%), whereas there was no observable change for the graphene on laminated due to the absence of any significant doping during the transfer process, in contrast to the PET samples. It is unclear at this stage as to why the PET presented a higher inclination for sulphur binding than the laminated, though in order to fabricate a time-stable flexible transparent graphene-based

electrode what is clear is that such transient binding must be prevented even though it reduces the initial sheet resistance. Indeed, the metallic element within the chloride dopant is evidently central to the effectiveness and temporal stability of the observed doping; a detailed theoretical study of the underlying electron transport and its dependence on the metal type will be presented elsewhere. The ratio of the change in transmittance to the change in transmittance of the undoped graphene ($\Delta T/T_0$) is illustrated in **Figure 4c**. $\Delta T/T_0$ for AuCl_3 and FeCl_3 was < 0.1 at most time points highlighting the maintained optical quality of the doped films.

The undoped graphene on laminate showed a stable sheet resistance over time (5.0→5.2 k Ω /sq), whilst the undoped graphene on PET increased from an initial 2.2 k Ω /sq to 3.5 k Ω /sq, after 200 hrs, again attributed to the unintentional sulphur deposited during the graphene-transfer process. The AuCl_3 -doping dramatically reduced the sheet resistance from 3.4 k Ω /sq to 0.29 k Ω /sq (laminate) and 1.8 k Ω /sq to 0.62 k Ω /sq (PET). Nevertheless, as before, the sheet resistance in the AuCl_3 case still tended to increase with time, though to a much lesser extent, suggesting the need for polymer passivation or hermetic capping layers. $\Delta R/R_0$, the ratio of the change in sheet resistance of doped graphene to undoped graphene, is the highest for the AuCl_3 -laminate, as shown in **Figure 4d**. Unlike the other dopants the reduced sheet resistance remains over time and only increased by 47% (laminate) and 15% (PET) after 200 hours. The sheet resistance was as low as 294 Ω /sq for the AuCl_3 -laminate,

which increased by only 141 Ω/sq , whilst for the $\text{AuCl}_3\text{-PET}$ it was 616 Ω/sq , and increased by only 92 Ω/sq after 200 hrs ambient exposure.

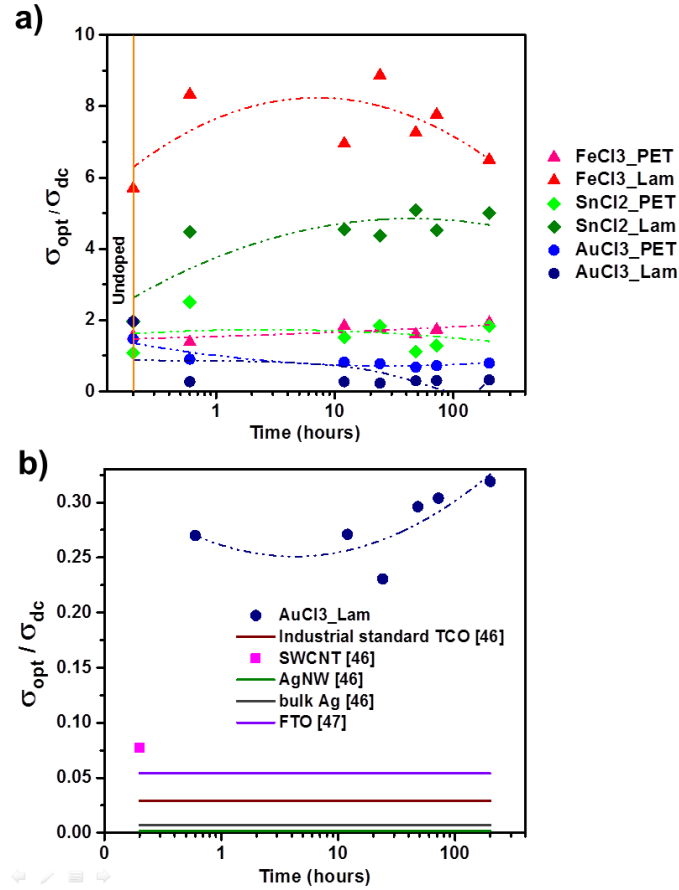


Fig. 5. a) Ratio of optical conductivity to dc electrical conductivity and b) the comparison of the ratio to other conductive transparent media. [52, 53]

The ratio of the optical conductance, σ_{opt} , to the DC electronic conductance, σ_{dc} , defines a figure of merit of their optoelectronic performance, and can be estimated from;

$$T = \left[1 + \left(\frac{tZ_0}{2} \right) \sigma_{opt} \right]^{-2} = \left[1 + 188.5 \frac{1}{R_s} \left(\frac{\sigma_{opt}}{\sigma_{dc}} \right) \right]^{-2} \quad (1)$$

Here Z_0 is the impedance of free space (377 Ω) and t is the film thickness. [53, 54] For an ideal transparent conductive electrode $\sigma_{opt}/\sigma_{dc} \rightarrow 0$; requiring a low sheet resistance and high

optical transmittance. The approximate $\sigma_{\text{opt}}/\sigma_{\text{dc}}$ values of our transferred and doped graphene and other competing transparent flexible conductors are plotted in **Figures 5a) and 5b)**, respectively. $\sigma_{\text{opt}}/\sigma_{\text{dc}}$ was < 2 for the FeCl_3 -doped and SnCl_2 -doped graphene cases and < 1 for AuCl_3 -doped graphene, during the entire measurement period (200 hours; AuCl_3 -laminate: 0.32, and AuCl_3 -PET: 0.79). The conductivity ratio of AuCl_3 -doped graphene on laminate is around ten times higher than the minimum industry standard (0.029), but was significantly lower than the undoped graphene and even lower again than that of stacked graphene (2.5). [54] AuCl_3 doping is a most efficient means to improve the opto-electronic conductivity of non-contiguous graphene on flexible and transparent substrates, and unlike other dopants, retains this state even after extended periods of time under ambient conditions.

Conclusion

Here we have detailed techniques for thermal and cold lamination of graphene onto flexible and transparent substrates. High transfer fidelity, indicated by high spatial uniformity, homogenous optical transmission and sheet resistance were noted. Cold lamination using UV-curable adhesive produced less wrinkling of the nascent graphene relative to the hot lamination with the added benefit that in the case of the cold lamination the films are doped with sulphur during the Cu-etch, reducing the sheet resistance without any additional doping processes. The reported facile transfer methods are inexpensive and reproducible. From time-dependent doping studies we showed that AuCl_3 is a most effective dopant, even for non-contiguous thin films, giving sheet resistances as low as $294 \Omega/\text{sq}$ and a $\sigma_{\text{opt}}/\sigma_{\text{dc}} \sim 0.3$ after 200 hrs ambient exposure. Such facile doped graphene / polymer films are showing increasing promise for highly transparent and highly conductive flexible electronics which may ultimately find use in OLED displays and next-generation photovoltaic devices.

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