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# Highly active P-doped sulfided NiMo/alumina HDS catalysts from Mo-blue by using saccharose as reducing agents precursor



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#### ABSTRACT

Saccharose (SA) was used as organic additive in simultaneously impregnated P-doped NiMo/Al<sub>2</sub>O<sub>3</sub> hydrodesulfurization (HDS) catalysts (Mo, Ni and P at 12, 3, and 1.6 wt%, respectively). One-pot impregnating solutions were prepared by MoO<sub>3</sub> digestion (~353 K) in diluted aqueous H<sub>3</sub>PO<sub>4</sub>, followed by 2NiCO<sub>3</sub>·3Ni (OH)<sub>2</sub>·4H<sub>2</sub>O addition. Saccharose (SA, SA/Ni = 0.5, 1-3) was dissolved in originally emerald-green impregnating solutions which changed to cobalt blue by room-temperature aging (2-4 days, depending on SA concentration) due to Mo-blue formation by partial molybdenum species reduction. After sulfiding of samples impregnated with SA shorter MoS<sub>2</sub> slabs of enhanced stacking were observed (by HR-TEM). Ni and Mo dispersion and nickel sulfidability (as determined by XPS) increased with the amount of organic modifier. Enhanced hydrodesulfurization activity in dibenzothiophene HDS was registered for catalyst obtained from Mo-blue precursor as to that of corresponding materials obtained from conventional emerald-green NiMoP impregnating solutions (with or without SA). However, in solids at high saccharose content (SA/Ni = 3) enhanced "NiMoS" phase amount was not reflected in improved activity. Probably, excessive amount of carbonaceous deposits from SA residua decomposition during catalyst activation provoked partially plugged porous network (as determined by N<sub>2</sub> physisorption) in sulfided formulations. That fact seemed to limit accessibility of reactant molecules to surface active sites. Mo-blue precursor obtained through monosaccharides partial reduction seemed to play decisive role in obtaining HDS catalysts of improved properties. Saccharose results a highly soluble, cheap and non-toxic environmentally-friendly additive to produce catalysts of enhanced HDS activity.

#### 1. Introduction

The use of various types of organic additives during hydrodesulfurization (HDS) catalysts preparation has resulted in materials of enhanced S removal activity. For instance, different beneficial effects have been related to chelators addition those mainly being attributed to their strong chemical interaction with the promoters (Ni and Co) [1]. It is well-accepted [1,2] that those complexing organic compounds could enhance molybdenum-promoter interaction during catalyst sulfiding that resulting in materials of increased hydrodesulfurizating activity. Recently, Hensen et al. [2] reported that organic chelators mainly enhance the HDS activity of promoted catalysts more than those of Mobased unpromoted ones confirming that their main role is improving Co-Mo interaction. Chelation allows promoter sulfidation to temperatures high enough to permit efficient integration of sulfidic Co or Ni species to then already formed MoS<sub>2</sub> (or WS<sub>2</sub>), thus optimizing promoted phases ("NiMo(W)S" or "CoMo(W)S") formation. However, for CoMo/Al<sub>2</sub>O<sub>3</sub> catalysts prepared with nitrilo triacetic acid, Maugé et al. [1] found that in addition to enhanced MoS<sub>2</sub> promotion using that chelator could modify active sites nature. Probably, highly active Type II sites characterized by weakened interaction with the alumina support were formed in that case.

Regarding non-chelating ligands as HDS catalysts additive so far no general agreement about the origin of their beneficial influence has been reached. It appears that enhanced HDS activity could be originated by a number of factors [3–9]. For instance, for P-doped CoMo/ $Al_2O_3$  formulations impregnated in glycols presence it has been proposed that the interaction of the organic additive with both basic OH

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groups and coordinatively unsaturated Al<sup>3+</sup> sites on the alumina surface could partially prevent Co diphosphopentamolybdate (Strandberg heteropolyanions [10]) complexes decomposition during their deposition on the carrier [4]. Thus, close proximity of Co and Mo in those complexes could result in formation of increased amount of the socalled "CoMoS phase" during sulfiding. Others [7] found optimized promoter integration to MoS<sub>2</sub> phase in catalysts prepared in presence of glycols (di-ethyleneglycolbutylether). They observed that Mo and Co sulfidation took place at temperatures up to 473 K, this fact probably contributing to better integration of cobalt promoter to edge sites of molybdenum sulfide particles. Our group reported [3] on the beneficial influence of ethylene glycol added to solutions where the organic additive was simultaneously impregnated with Ni. Mo and P over an alumina support. Again, enhanced formation of promoted "NiMoS" phase appeared to be at the origin of enhanced activity in both model compound (dibenzothiophene) and real feedstock (straigth-run gas oil) hydrodesulfurization. Considering that reported in the pertinent literature, it seems that the positive influence of those organic species on the HDS activity of corresponding catalysts could be related to several independent phenomena. For instance, when comparing the effects of chelating and non-chelating agents on the properties of CoPMo/alumina sulfided catalysts Nikhulsin et al. [11] found that glycols better prevented the decomposition of impregnated 12-molybdophosphoric heteropolyacid  $H_3PMo_{12}O_{40}$ , as compared to a chelating agent (citric acid). They also observed that active sites turnover frequency (TOF) increased with the (Co/Mo) edge promotion degree in materials prepared with either chelators and no-chelator species or with a mixturte of both modifiers.

Other [12] have used polydopamine biopolymer to coat alumina support where molybdenum was impregnated. The presence of the organic layer was reflected in higher number of Mo adsorption sites as to those over similar material impregnated on bare alumina carrier. The corresponding CoMo-supported catalysts had improved performance for hydrotreating reactions including aromatics saturation, hydrodenitrogenation and gasoil HDS. In that case, better accessibility of reacting molecules and weakened support-impregnated phases interaction due to carbon adlayer were both invoked to explain enhanced catalytic activity in aforementioned reactions.

Saccharose (SA) has been used in the past in developing Co-based Fischer-Tropsch catalysts with alumina as carrier [13]. Using that organic additive resulted in highly dispersed supported phase in reduced catalysts obtained from Co nitrate (Co/SA = 10). It is worth mentioning that in this case saccharose was added during impregnation being then removed during calcining at various temperatures (373–673 K). Thus, no carbonaceous residua were observed, the main effect of the additive being exerted during impregnation step.

Saccharose has also been applied as precursor intended to form carbonaceous materials that could be utilized as catalyst carrier. In this line, composite supports prepared from carbothermal reduction of carbonaceous silicon xerogel obtained by sol-gel by using tetraethoxysilane and saccharose [14] proved to be effective Ru catalysts supports, regarding materials applied in ammonia synthesis.

Regarding saccharose-alumina interaction, some studies have shown that saccharide adsorption follows a typical Langmuirian (Stype) adsorption isotherm, occurring via metal-saccharide interaction and also through electrostatic interactions between surface aluminol groups on alumina and sugar hydroxyls [15]. The exothermic adsorption ( $\Delta H^{\circ} = -9.68 \text{ kcal mol}^{-1}$ ) had negative free energy change ( $\Delta G^{\circ} = -6.47 \text{ kcal mol}^{-1}$ ) pointing out to its spontaneous nature. Saccharose could be highly dispersed on  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> surface at moderate concentrations crystalline sugar being just observed in higly loaded samples ( $\sim 0.35 \text{ g}_{\text{sugar}} \text{ g}_{\text{alumina}}^{-1}$ ) [16]. The partially negatively charged oxygen atoms of hydroxyls in saccharose could establish hydrogen bonding with AlOH groups on alumina surface [15]. Also, interaction of those organic moieties bearing partial negative charge with surface Lewis sites on alumina (Al<sup>3+</sup> CUS, coordinatively unsaturated sites) could not be discarded.

In previous report [9] we found increased HDS activity (~17% DBT HDS) in alumina-supported NiMo-based catalysts simultaneously impregnated with saccharose by using freshly prepared emerald-green solutions. On the other hand, in the present communication we used corresponding aged cobalt-blue colored impregnating solutions finding much better sulfided catalyst performance. SA was simultaneously one-pot impregnated with Ni, Mo and P components as that easily scalable (for applications at commercial level production) methodology provided HDS catalysts of enhanced activity [9].

Materials were studied by using several textural, structural and surface characterization techniques. Corresponding sulfided catalysts were tested in dibenzotiophene HDS at conditions close to those used in commercial-scale hydroprocessing units aimed to ultra-low-sulfur diesel (ULSD) production.

#### 2. Experimental

#### 2.1. Catalysts synthesis

Commercial powdered boehmite (Versal 200 from Euro Support) was calcined at 773 K (5 h) to obtain high surface area alumina (see Table 1). Pore-filling simultaneous impregnation was carried out over previously dried powder support (393 K, 2 h) with an acidic solution (pH  $\sim$  2.5) prepared from digestion (at  $\sim$  353 K, in water and under vigorous stirring, 4 h) of MoO<sub>3</sub> 99.5 wt % (PQM) in presence of H<sub>3</sub>PO<sub>4</sub>, 85.3 wt% (Tecsiquim). H<sub>2</sub>O in excess (typically 250 ml of starting solution to impregnate 5 g of alumina carrier) was used to accelerate dissolution and digestion of molybdenum salt. A yellow transparent solution was obtained after approximately 2 h hydrolysis. 2NiCO3:3Ni (OH)<sub>2</sub>·4H<sub>2</sub>O (Sigma-Aldrich) was added, processing temperature being maintained for 2 h. Ni concentration corresponded to Ni/ (Ni + Mo) = 0.29 molar ratio whereas  $P_2O_5/(NiO + MoO_3) = 0.2$  mass ratio was fulfilled [17]. A transparent emerald- green solution was thus finally obtained. The solution volume was then reduced by evaporation until reaching the suitable one for pore-filling impregnation of the given mass of support. The described one-pot impregnation method was chosen as it constitutes a readily scalable methodology during catalyst preparation at commercial scale. It includes cheap industrial precursors easily attainable for commercial catalyst manufacturing. Although not commonly used at laboratory scale preparations [18,19], Ni hydroxycarbonate (soluble in acidic medium) was preferred as nickel source as usage of corresponding nitrate (highly soluble in water) must be avoided due to its explosive properties. Nominal Mo, Ni and P loadings corresponded to 12, 3 and 1.6 wt%, respectively. Phosphorus content closely corresponded to concentration used in preparation of highly active HDS catalysts (Mo/P~2.5) [20]. Saccharose (SA, J.T. Baker) at SA/Ni molar ratio of 0.5, 1, 2 and 3 was dissolved into the impregnating solutions at room temperature stirring being maintained by 1 h. No appreciable changes neither in pH nor in coloring (emerald-green) was observed after that period of time. However, after room temperature aging the original green coloring of impregnating solutions containing NiMoP-SA changed to cobalt blue. The period of time required to

Table 1

Textural properties (as determined by  $\rm N_2$  physisorption at 75 K) of alumina and SA-modified NiMo/Al\_2O\_3 sulfided catalysts at different organic additive concentration.

| Material                       | $S_{BET}$<br>$(m^2g^{-1})$ | $S_{BJHa}$<br>$(m^2g^{-1})$ | $S_{BJHd}$<br>(m <sup>2</sup> g <sup>-1</sup> ) | $V_{p}$ (cm <sup>3</sup> g <sup>-1</sup> ) | D <sub>p</sub> <sup>a</sup><br>(10 <sup>-9</sup> m) |
|--------------------------------|----------------------------|-----------------------------|---|--|---|
| Al <sub>2</sub> O <sub>3</sub> | 307                        | 375                         | 488   | 0.90                                       | 11.7  |
| SA(0)                          | 224                        | 215                         | 291   | 0.64                                       | 11.4  |
| SA(1)                          | 260                        | 256                         | 354   | 0.71                                       | 10.9  |
| SA(3)                          | 244                        | 231                         | 354   | 0.59                                       | 9.7   |
|                                |                            |                             |   |  |   |

<sup>a</sup> From  $4 \times V_p S_{BET}^{-1}$ .

observe the blue coloring was function of SA concentration being shortened in solutions of enhanced SA content [21]. For instance, for the solution of the highest organic modifier concentration (SA/Ni = 3)the coloring change took place after 48 h whereas for the one at SA/ Ni = 1 blue coloring was registered after about 96 h. No significant effect on pH of pristine impregnating solutions (pH = 2.2) was observed neither after SA addition (non-aged green solution, pH = 2.5) nor after coloring change to blue (pH = 3.4). Alumina support was impregnated by pore-filling with the cobalt-blue solutions at various SA concentrations right after the coloring change from green to blue was observed. Also, a reference matterial was prepared by depositing the SA-modified NiMoP (at SA/Ni) green-colored not aged impregnating solution over corresponding Al<sub>2</sub>O<sub>3</sub> carrier. After impregnation, materials were aged overnight at room temperature allowing intradiffusion of impregnated species through the support porous network. Materials were then dried at 393 K (2 h). Calcining was avoided to preserve organic additive integrity. Samples were identified by using the SA(x) key where "x" represented the SA/Ni ratio. Also, when pertinent, subscript "g" (representing "green") was added to previously mentioned keys to differentiate materials impregnated with NiMoP-SA as-made green solutions. A reference material with no organic additive, sample SA(0), was also synthesized. This solid was just dried at 393 K (2 h) before sulfiding.

#### 2.2. Materials characterization

Chemical state of Ni and Mo species present in impregnating solutions was studied through UV–vis spectroscopy by using a Cary 100 instrument (absorbance mode) in the 200–1000 nm wavelength range. Textural properties of support and sulfided catalysts were determined by N<sub>2</sub> physisorption (at 75 K, nitrogen saturation temperature at Mexico City barometric pressure) in a Micromeritics ASAP 2000 apparatus after ultra-high vacuum degassing at 423 K for 6 h. Surface area and pore size distribution of studied materials were determined by BET and BJH (N<sub>2</sub> adsorption and desorption branch data) methods, respectively. Thermal analyses of dried samples were carried out in the room temperature to 1250 K range with a Netzch Thermische Analize STA 409 EP apparatus under static air atmosphere.

Raman spectra of various impregnated samples were obtained using a Jobin Yvon Horiba (T64000) spectrometer, equipped with a CCD camera detector. As excitation source the 514.5 nm line of a Spectra Physics 2018 Argon/Krypton ion laser system was focused through an Olympus BX41 microscope equipped with a  $100 \times$  magnification objective. Laser power never exceeded 5 mW to avoid damaging the studied samples. Other groups [22,23] have used similar Laser power values during characterization of hydrotreating formulations.

UV-vis diffuse reflectance spectroscopy (DRS) of impregnated samples were carried out on a Perkin Elmer Lambda 35 spectrophotometer equipped with integration sphere. Spectra were taken in the wavelength range from 200 to 1100 nm. Spectralon signal was subtracted as baseline.

Supported sulfidic catalysts were obtained by submitting approximately 0.3 g of ground impregnated precursors (U.S. mesh size 80–100, circa 165 µm average particle diameter) to treatment at 673 K (heating rate 6 K min<sup>-1</sup>) under H<sub>2</sub>/H<sub>2</sub>S (Praxair) at 50 ml min<sup>-1</sup>/6 ml min<sup>-1</sup> ratio during 2 h. Sulfided catalyst were characterized by electron microscopy and X-ray photoelectron spectroscopy (XPS). High-resolution transmission electron microscopy TEM was performed in a JEM-2200FS at accelerating voltage of 200 kV. The apparatus was equipped with a Schottky-type field emission gun and an ultra-high resolution (UHR) configuration (Cs = 0.5 mm; Cc = 1.1 mm; point to point resolution, 0.19 nm) and in-column energy filter omega-type. Sulfided solids to be analyzed were ground, suspended in isopropanol at room temperature and dispersed with ultrasonic agitation. Then, some drops of obtained suspensions were deposited on a 3 mm diameter lacey carbon copper grid. GATAN<sup>™</sup> software was used during statistical determination of slab length and stacking degree in MoS<sub>2</sub> supported particles. Average length and stacking number of MoS<sub>2</sub> slabs ( $L_{avg.}$  and  $N_{avg.}$ , respectively) were statistically estimated through measurements carried out in ~ 300 MoS<sub>2</sub> slabs per sample through the following equations:

$$L_{\text{avg.}} = \frac{\sum_{i=1...i} n_i L_i}{\sum_{i=1...i} n_i}$$
(1)

$$N_{\text{avg.}} = \frac{\sum_{i=1...t} n_i N_i}{\sum_{i=1...t} n_i}$$
(2)

where  $n_i$  is the number of slabs in a given range of length or stacking (*L* and *N*, respectively) and *t* is the total number of analyzed MoS<sub>2</sub> slabs.

Dispersion of molybdenum atoms in  $MoS_2$  crystals was determined by geometrical model from Kasztelan et al. [24]. Thus,

$$D_{Mo} = \frac{Mo_e + Mo_c}{Mo_t} = \frac{\sum_{i=1...t} 6n_i - 6}{\sum_{i=1...t} 3n_i^2 - 3n_i - 1}$$
(3)

where  $D_{Mo}$  is MoS<sub>2</sub> dispersion,  $Mo_e$ ,  $Mo_c$  and  $Mo_t$  are molybdenum atoms on the edges, corners and total ones (as determined by slabs size);  $n_i$  is the number of atoms along one side of a molybdenum sulfide slab as determined from its length,  $L = 0.316 \times (2n_i-1)$ , in nm, and t is the total number of slabs analyzed from HR-TEM corresponding micrographs during the statistical counting.

XPS spectra were obtained in a THERMO-VG SCALAB 250 spectrometer equipped with  $Al_{K\alpha 1}$  X-ray source (1486.6 eV) and a hemispherical analyzer. Experimental peaks were decomposed into components using mixed Gaussian-Lorentzian functions and a non-linear squares fitting algorithm. Shirley background subtraction was applied. Binding energies were reproducible to within  $\pm$  0.2 eV and the C 1s peak (from adventitious carbon) at 284.6 eV was used as a reference. Surface elemental composition was determined by fitting and integrating the Ni (2p), Mo (3d), S (2 s), P (2p) and Al (2p) signals intensity and converting these values to atomic ratios by using theoretical sensitivity factors provided by the manufacturer of the XPS apparatus. Sulfided catalysts were kept under inert atmosphere to avoid samples oxidation by using a glove box and a special box vessel to introduce the samples into the spectrometer analysis chamber.

#### 2.3. HDS reaction test

Impregnated precursors were sulfided at 673 K (heating rate 6 K  $min^{-1}$ ) under H<sub>2</sub>/H<sub>2</sub>S (Praxair) at 50/6 (ml min<sup>-1</sup>)/(ml min<sup>-1</sup>) during 2 h. HDS activity of the obtained sulfided catalysts was studied in a triphasic slurry batch reactor (Parr 4575). The reaction mixture was prepared by dissolving  $\sim 0.3$  g of dibenzothiophene (model of organo-S species in middle-distillates) in 100 cm<sup>3</sup> of *n*-hexadecane (representing hydrocarbon matrix in straight-run gasoil) (98 mass % and 99+ mass %, respectively; both from Aldrich) and adding  $\sim 0.2$  g of sieved catalyst (80-100 U.S. mesh, ~0.165 mm average particle diameter). Operating conditions, carefully chosen to avoid external diffusion limitations [9], were P = 7.1 MPa, T = 593 K and 1000 RPM (~105 rad  $s^{-1}$ , mixing speed). Samples were taken periodically and analyzed by gas chromatography (Agilent 6890 N with flame ionization detector and Econo-Cap-5 capillary column length: 30 m, 0.53 mm diameter and film thickness of 1.2 µm, from Alltech). HDS kinetic constants were calculated assuming pseudo-first order model referred to DBT concentration and zero order with respect to excess H<sub>2</sub>:

$$k = \frac{-\ln(1-x)}{t_r} \tag{4}$$

where *x* is conversion and *t<sub>r</sub>* refers to reaction time. *k* values for various catalysts were normalized by considering reaction volume and mass of catalyst used ( $k^1$  in m<sup>3</sup> kg<sub>cat</sub><sup>--1</sup>s<sup>-1</sup>)). According to constant slope of  $-\ln(1-x)$  versus reaction time plots in the dibenzothiophene HDS over various tested materials tested (see, for instance, Fig. S1) significant

deactivation along the duration of our catalytic tests could be ruled out. Thus, in our case calculation of k (pseudo first order kinetic constant) we considered all the experimental points obtained during duration of our reaction tests. On the other hand, declining slope at progressively longer reaction times in corresponding  $-\ln(1-x)$  versus reaction time plots in the case of supported Pd-Pt catalysts tested in the HDS of sterically-hindered dibenzothiophenic species in batch reactor have been related to catalysts deactivation [25].

HDS activity of various materials was also compared on per active sites basis by affecting intrinsic activity (on per Mo mass basis) taking into account corresponding  $Mo^{4+}$  fraction and dispersion (as determined by XPS analysis and from Eq. (3), respectively) over different prepared catalysts [26].

#### 3. Results and discussion

#### 3.1. Visible spectra characterization of impregnating solutions

To discern the chemical phenomena behind color changes of impregnating NiMoP-SA solutions after aging we studied corresponding liquid samples by visible region spectroscopy. To determine if Ni promoter chelation took place in the impregnating solutions Ni hydroxycarbonate and H<sub>3</sub>PO<sub>4</sub> were dissolved at concentration equivalent to that used during catalysts preparation Due to its strong absorbance the sample was diluted with water (5:1). Characteristic green-emerald coloring imparted by hexa-aquonickel cations (Ni[(H<sub>2</sub>O)<sub>6</sub>]<sup>2+</sup>) was observed, the corresponding visible spectra showing absorptions at 657 and 723 nm due to octahedral Ni<sup>2+</sup><sub>(Oh)</sub> [27], Fig. S2. After organic additive addition (at SA/Ni = 1) no significant changes were evidenced even after several days room temperature aging (maxima at 663 and 724 nm, Fig. S3) meaning that water molecules in  $Ni[(H_2O)_6]^{2+}$  were not substituted by stronger (in spectroscopic series [28]) organic ligands (carboxylates, for instance). Nickel complexation under those conditions could be then ruled out. Although preparation of chelated iron cations with fructose has been reported [29,30] nickel complexation appeared improbable as disaccharides lack this ability [30].

In the opposite and as already mentioned, when sucrose was added to impregnating solutions containing NiMoP the original green coloring shifted to cobalt blue after room temperature aging for some days, depending on SA concentration. Visible region spectra in Fig. 1 show evolution of spectra corresponding to impregnating solutions through different preparation stages. Due to different absorbance of studied samples dilutions (with water) were required. Dilution factors used are shown in Fig. 1 caption. Pristine NiMoP solution and the as made SAmodified one (SA(1)<sub>g</sub>) showed characteristic Ni<sup>2+</sup><sub>(Oh)</sub> [27] absorptions (Fig. 1(a) and (b), respectively) whereas aged deep blue solution registered peaks related to molybdenum-blue complex (Fig. 1(c)). Similar signals with peak maximum at 865 nm and shoulder at 700 nm were considered by Shukor et al. [31] as unique fingerprint characterizing Mo-blue compounds from phosphomolybdates reduction using various carbohydrates as electron donors. Also, Yoshimura et al. [32] obtained Mo-blue species (visible spectra maxima at 890 and 710 nm) from phosphomolybdates reduction by ascorbic acid pointing out that those P-containing compounds are important intermediates in partial Mo reduction.

Cáceres et al. [33] identified Mo-blue species after X-ray irradiation in XPS equipment chamber that provoked elimination of lattice oxygen from MoO<sub>6</sub> octahedra followed by rearrangement of produced defective structures. Also, Louis and Che [34] reported air oxidation of physisorbed Mo<sub>5</sub>Cl<sub>10</sub> on silica into molybdenum-blues. In this case, however those species were eliminated after washing with ammonia solution due to their very weak interaction with the support. To the best of our knowledge, just a few papers reports on using Mo-blue as heterogeneous catalyst or catalyst precursor. For instance, Hutchings et al. [35] reported utilizing Mo-blue nanorings with Mo<sup>5+/</sup>Mo<sup>6+</sup> ratio = 1/ 4 founding excellent activity and stability in the partial oxidation of cyclohexane to cyclohexanol and cyclohexanone.

Saccharose has been added to solutions containing disolved  $MoO_3$  which were maintained at 353 K until color changed from orange to blue. According to the authors, that indicated  $Mo^{6+}$  reduction to colloidal molybdenum-sucrose complex [36]. The dark blue grains obtained after drying were carburized the final material presenting promising properties in the methane dry reforming.

Regarding hydrotreating catalysts the preparation of highly active Mo-blue based sulfided catalysts applied in thiophene HDS and pyridine hydrodenitrogenation was previously reported [37]. However and in spite of very promising results obtained no further investigations by using promoted (by Co or Ni) catalysts were further published. In mentioned paper hydrazine sulfate was used as reductant, the product showing visible spectra maximum at 775 nm. The structure was considered as that of  $[Mo_6O_{18}]^{-2}$  anion. Srinivasan et al. [38] also used hydrazine-hydrochloric acid solution although in this case Mo-blue coloring was just used as visible indicator of molybdenum diffusion through alumina pellets. However, susbstitution of hydrazine sulfate as reducing agent by citric acid for Mo-blue formation has been proposed [39] due to its extreme toxicity and carcinogenicity that makes impractical its utilization. Interestingly, Srinivasan et al. [38] failed in obtaining Mo-blue by using sucrose as reducing agent eve after heating at 373 K heating. Very probably at the high pH conditions of ammonium paramolybdate solutions they used the non-reducing disaccharide could not be properly hydrolyzed to reducing fructose and glucose. That reaction easily takes place in acidic medium [41]. It is worth mentioning that Mo-blue generically refers to partially reduced molybdenum species with average oxidation number of Mo is between 5.5 and 6 and which exact nature depends on formation conditions (type of reducig agent, pH of solution, etc.) [40]. Thus, changing the synthesis conditions could result in various Mo-blue species which could have very different characteristics as HDS catalysts precursor.



**Fig. 1.** (a) Visible region spectrum of Ni hydroxycarbonate + H<sub>3</sub>PO<sub>4</sub> + MoO<sub>3</sub> + saccharose solutions originally at concentrations similar to those used during catalysts preparation. (a): As made NiMoP solution, diluted with water 2.5:1; (b): SA-modified, non-aged (SA(1)<sub>g</sub>) impregnating solution, diluted with water 2.5:1; SA-modified impregnating solution aged until Mo-blue formation, diluted with water 500:1. (For interpretation of the references to colour in the figure text, the reader is referred to the web version of this article.)

In order to be able to act as reductant a given sugar requires the presence of hemiacetal moieties, i.e., a carbon linked to both one OHand one OR- group. In that way, the cyclic sugar could be in equilibrium with an open-chain linear isomeric form containing an aldehyde group that could be oxidized to corresponding carboxylic acid, with the consequent reduction of involved oxidizing agent (Mo<sup>6+</sup> species in our case). On the other hand, saccharose have glycosidic bonds between its anomeric carbons (acetal moieties) joining glucose and fructose rings and thus could not be converted to an open-chain isomer forming an aldehyde. However, saccharose in aqueous solution could be easily hydrolyzed into monosaccharides components under highly acidic conditions  $(pH \sim 2)$  [41] (see Scheme S1), as in the case of our NiMoP one-pot impregnating solutions (see Section 2.1 Catalysts Synthesis). Glycosidic bonds linking glucose and fructose rings could be then readily hydrolyzed converting saccharose into those monosaccharides very well-known for their reducing properties. In our case, as the openchain linear isomeric form of monosaccharides (containing an aldehyde group to be oxidized) could disappear by reacting with Mo<sup>6+</sup> species their consumption could progressively displace thermodynamic equilibrium to its fomation providing enhanced amounts of reducing agents. Through their oxidation glucose and fructose could be able to originate corresponding carboxylic acids (saccharic acid from the former, glycolic and trihydroxy-butyric ones from the latter) which could then carry out cations chelation. However, in order that phenomenon to take place pH during impregnation over alumina surface must be high enough to allow weak organic acids deprotonation to permit then Nicomplexation. Advantageously, those oxidated derivatives could probably complexate nickel promoter in our impregnating solutions.

The saccharide and its oxidized derivatives could also form carbonaceous deposits able to weaken the supported phases-carrier interaction [9]. In fact, in previous investigations we found that in SA-modified alumina-supported sulfided NiMoP formulations obtained from non-aged green solutions (where no reduced Mo-blue species existed) carbonaceous residua from the disaccharide decomposition during catalyst activation by sulfiding resulted in enhanced HDS activity in dibenzothiophene conversion. In that case, alumina surface passivation weakened carrier-deposited species interaction that being reflected in ~17% enhanced HDS activity, as to that of non-modified otherwise similar NiMoP/alumina conventional catalyst. All those mentioned effects could positively influence the properties of corresponding HDS catalysts. Although Mo-blue could be obtained through the use of other reducing agents we emphasize that this work was focused in using an additive which characteristics could allow its application in commercial level catalysts manufacturing. The use of a readily scalable preparation protocol including no additional preparation steps in our case was of crucial importance. It is worth mentioning that Mo-blue obtention in SA-modified NiMoP impregnating solutions could be accelerated by heating at 353 K where the partially reduced molybdenum species could be almost instantaneously formed. Thus, SA results a very convenient, cheap, environmentally friendly additive. Finally, the low pH of impregnating solution containing nickel and phosphopentamolybdates species provides the ideal environment for SA acid hydrolysis for in situ formation of monosaccharides actually acting as reducing agents. Again, by avoiding the direct addition of glucose, fructose or a mixture of them the proposed preparation process is simplified by utilizing a readily available disaccharide significantly cheaper than its corresponding hydrolysis products.

In the following, characterization of Mo-blue based Ni promoted catalysts was endeavoured to try to get insigth into the effect of using that precursor on HDS properties of corresponding sulfided catalysts.

#### 3.2. Textural characterization of materials

 $N_2$  physisorption isotherms (Fig. S4) of alumina and corresponding sulfided catalyst SA(0) were intermediate between types II and IV according to IUPAC classification [42] suggesting mesoporosity and perhaps some macroporosity. In SA(0) adsorbed N<sub>2</sub> volume decrease, as to that of alumina support, was proprotional to the non-porous NiMoP phases loading in sulfided catalyst (~29 wt%) ruling out partial poreplugging then suggesting well dispersed impregnated phases. Isotherm corresponding to SA(1) (Fig. S4) was similar as to that of the nonmodified SA(0). However, in SA(3) N<sub>2</sub> adsorption capacity had clearly diminished probably due to enhanced amount of carbonaceous deposits partially plugging the porous network.

Alumina and SA(0) materials had type H1 hysteresis characterizing solids with porous networks of uniform size and shape [42] where capillary condensation [43] started at ~P/P<sub>s</sub> = 0.45. On the other hand, in SA-modified solids the hystheresis shape progressively shifted to H3 type suggesting presence of slit-shaped pores of non-uniform size and/ or shape [42] that could be provoked by pore mouth narrowing [44]. In these cases, capillary condensation started at approx. ~P/P<sub>s</sub> = 0.40 suggesting smaller pores as to those of SA(0). Then, carbon deposits were suggested, mainly on the SA(3) solid.

Carrier surface area loss in SA(0) was commensurated with deposited NiMoP phases (Table 1) loading supporting the notion of well dispersed sulfidic non-porous phases. Interestingly, SA-modified samples had augmented surface area as to that of non-modified SA(0), pointing out to additional contribution from void spaces in porous carbonaceous deposits. Similar results were reported by Rinaldi et al. [45] in the case of sulfided Co-Mo/B<sub>2</sub>O<sub>3</sub>/Al<sub>2</sub>O<sub>3</sub> catalysts prepared with citric acid. In SA(3) incremental surface area could be counterbalanced by decreased pore volume due to more significant amount of carbonaceous deposits.

To determine which pore size distribution profile (PSD, through Barret-Joyner-Halenda, BJH, methodology) obtained from either adsorption or desorption data branch (Figs. S5 and S6, respectively) better represented the actual one corresponding cumulative surface area values were compared to those obtained through the BET method, Table 1. As values from adsorption branch data ( $S_{BJHa}$ ) were closer to the BET ones we considered that PSD profiles from those data were the most reliable to describe the actual porous networks.

From Fig. S5, a proportion of the smaller pores existing in alumina support network disappeared after NiMoP phases impregnation. Whereas SA(1) BJH<sub>a</sub> PSD was similar in both SA(0) and SA(1), the maximum at  $\sim 12$  nm in the latter profile decreased by 50% in the SA(3), again pointing out to partial pore-plugging by increased carbonaceous deposits. The SA-modified catalysts PSD profile from desorption data (Fig. S6) had maximum at  $\sim 3.2$  nm that was much more remarkable in SA(3). We considered that an artifact due to tensile strength effect (TSE) originated by random distribution and interconnectivity of pores [46] that commonly appears by the sudden forced closure of hystheresis at P/Ps between 0.4-0.45 and 0.4-0.5 for SA(1) and SA(3), respectively. If that PSD maximum around 3.8 nm were real it must be evident in profiles obtained from both adsorption and desorption data branches. It is worth mentioning that several investigators have been misled by aforementioned effects attributing corresponding PSD maximum to porosity creation [46]. For instance, a peak between 3-4 nm have been reported for SA-impregnated alumina at content superior to monolayer [47] that being dubiously attributed to carbonaceous deposits from crystalline SA carbonization. Taking into account that our SA-modified catalysts manifested pore network effects the PSD from adsorption branch was deemed more representative of the actual as it was not significantly affected by TSE phenomenon, as already suggested by comparing corresponding  $S_{BJHa}$  surface area data (Table 1) with corresponding  $S_{BET}$  ones.

#### 3.3. Thermal analysis (TG-DTA) of impregnated solids

Up to 366 K (water boiling temperature in Mexico City) weight losses of around 3.3% were registered in both alumina impregnated with SA and phosphate (at concentrations similar as to those in SA(1) dried formulation, Fig. S7. In the former case, phosphates were added in concentration similar to that used in impregnating solutions due to the effect of those anions on amount of saccharide adsorbed by the carrier. The electrostatic repulsion between the alumina  $AlOH_2^+$ groups of the positively charged support surface under acidic conditions due to its isoelectric point [48] and partially charged hydrogen in  $-OH^+$  groups of saccharides could be screened by the added phosphate anions then significantly increasing the amount of adsorbed organic species [15]. Also, interaction between the negatively charged oxygens  $-HO^{\delta-}$  and aluminols could be favored. Although carbohydrates are slightly acidic in aqueous solutions [15] under the low pH conditions provided by dissociated H<sub>3</sub>PO<sub>4</sub> they could not be deprotonated that precluding its interaction with coordinatively unsatured sites of surface  $Al^{3+}$  cations. Organic matter decomposition and combustion were related to additional losses (~23 wt%) up to 773 K, Fig. S7.

Physisorbed species were diminished on SA(1) and SA(3), as to that in the alumina modifed by SA and phosphates probably due to surface coverage by deposited species Weigth losses due to organic matter decomposition and combustion were notably accelerated by Ni and Mo phases acting as combustion catalysts [49]. In SA(1) increased losses related to mentioned phenomena suggested presence of species of weaker interaction with the support. In the 366–773 K weight losses in analyzed dried samples SA(1) and SA(3) were 22 and 35 wt%, respectively. According to saccharose composition about 42% of those weigth losses could be attributed to carbon species (i.e.,  $\sim$ 9 and  $\sim$ 15 wt%, respectively). Beyond  $\sim$ 1090 K MoO<sub>3</sub> sublimation [49] was reflected in pronounced weight losses in Mo-containing solids.

Endothermal signals due to melting and decomposition of saccharides were absent in profiles of SA-containing samples of Fig. 2 pointing out to adsorbed species in chemical interaction with the alumina support [47]. The signal at approximately 470 K and the wide exothermal inflexion between ~540–755 K were both related to combustion of species bonded to the carrier. Indeed, this kind of carbonaceous deposits have been related as support-like carbon which participates in the structure of the active phase as an intermediate carrier decreasing their interaction with the alumina support, as proposed by Chianelli et al. [50] from their studies on CoMo/Al<sub>2</sub>O<sub>3</sub> catalysts modifed by ethylenediamine.

Exothermal peaks at 773 and 1080 K were originated by NiO crystallization and  $Ni_2Al_{18}O_{29}$  formation, respectively, due to high-temperature migration and sintering of nickel species. The small endothermal signal at 1230 K in Mo-containing samples could be originated by MoO<sub>3</sub> sublimation [49].

The strong exothermal peak at 514 K in SA(1) profile shifted to 546 K in SA(3) showing significantly enhanced intensity in agreement with its augmented saccharide content. Those inflexions could be related to organic moieties in interaction with deposited phases. It

seemed that in SA(3) more stable metal-organic complexes could be formed. In this line the visible region signal around 400 nm previously identified in SA-containing alumina-supported NiMo formulations [9] could be rather attributed to ligand to metal charge transfer (LMCT) complexes saccharide-molybdenum. Indeed, absorptions due to LMCT complexes of thioglycolic acid-molybdenum have been identified in that region [26,51].

#### 3.4. Raman spectroscopy studies of impregnated samples

According to the Mo/P ~2.5 ratio used during our catalyst preparation (~2.5) and the pH of impregnating solutions (~2.5) Strandberg H<sub>2</sub>P<sub>2</sub>Mo<sub>5</sub>O<sub>23</sub><sup>4-</sup> anions constituted by two PO<sub>4</sub> tetrahedrons surrounded by five interconnected octahedral MoO<sub>6</sub> species could be preferentially formed in impregnating solutions obtained from MoO<sub>3</sub> digestion in diluted H<sub>3</sub>PO<sub>4</sub> [4,52]. Also, due to their stability  $H_{v}P_{2}Mo_{5}O_{2}$  (6-x)- anions could exist in different protonation states over a wide pH range [52]. Then, those species could be preserved after the slight pH changes observed after SA addition. However, after contacting the γ-Al<sub>2</sub>O<sub>3</sub> carrier surface due to increased pH that species start decomposing into phosphate and molybdates [53]. In agreement with that, deposited Mo species in dryed precursor prepared with no SA (inset in Fig. 3) showed Raman absorptions at 225, 359, 913 (shoulder) and 949 cm<sup>-1</sup> that could be assigned to Anderson-type Al  $(OH)_6 Mo_6 O_{18}^{3-}$  species [4] which formation could involve partial carrier dissolution. The former two signals could be assigned to Mo-O-Mo and O-Mo-O vibrations [54], meanwhile those at higher wavenumbers were related to Mo=O bands. The shoulder at 1085 cm<sup>-1</sup> suggested deposited polyphosphates originated by phosphomolybdates decomposition [6]. Interestingly, the M=O stretching vibration shifted to 969 cm<sup>-1</sup> in SA-containing samples strongly suggesting octahedrally-coordinated  $Mo_8O_{26}^{4-}$  or  $Mo_7O_{24}^{6-}$ -like polymolybdate species [55]. Similar signals have been associated to hydrated molybdates [56]. In our case, OH's could be partially provided by organic moieties closely interacting with deposited Mo species. Also, organic moieties could compete with heteropolymolybdateslike species present in solution during impregnation. Some organic moieties could deposit on surface OH groups and Al<sup>3+</sup> sites. Thus, interaction between diphosphopentamolybdate-like anions and the alumina support could be diminished. Through this mechanism, decomposition of those impregnated species could be prevented at some extent. Once deposited, those species could be precursors of polymolybdates observed in Raman spectra of SA-containing sample (Fig. 3). The shoulder at 985  $\text{cm}^{-1}$  in SA-containing materials (more notable in SA(3)) was related to Mo=O symmetric stretch of well-



Fig. 2. Differential thermal analysis (under static air atmosphere) of dried samples of  $Al_2O_3$  carrier, SA(0), SA(1) and SA(3) impregnated samples.



**Fig. 3.** Raman spectra of SA(0), SA(1) and SA(3) impregnated samples.Inset: Magnified  $(13.5 \times)$  SA(0) spectrum.

defined, three-dimensional molybdate clusters [57] in agreement with Mo-species in lower interaction with the carrier.

Two bands due to deposited carbonaceous species could be clearly identified in Raman spectra of SA-modified materials (Fig. 3). The D band centered around 1377 cm<sup>-1</sup> and the eco band at 1613 cm<sup>-1</sup> (D2, shoulder) could be assigned to disordered or poorly structured aromatic moieties whereas the shoulder observed between 1510-1530 cm<sup>-1</sup> could be attributed to D3 band related to structural defects of poorlyorganized aromatic domains [58]. More structured aromatic clusters originated the G band at 1580 cm<sup>-1</sup>. All mentioned signals were much more intense in SA(3) spectrum indicating increased carbonaceous deposits amount. This fact agreed with the enhanced height/width G band ratio in SA(3) as to that of SA(1) pointing out to larger carbon clusters in the former material. It should be mentioned that some carbonaceous deposits heating and graphitization could occur due to the laser power used during Raman analysis (less than 5 mW) [59]. Expectedly, those deposited carbon residua could play a determining role in the HDS catalytic activity of prepared formulations. That kind of carbonaceous deposits could be partially playing support-like role [50] as suggested by our DTA results (see Section 3.3 Thermal analysis (TG-DTA)) decreasing the deposited phases-alumina carrier interaction. However, if massive, formation of carbon deposits could also affect textural properties of corresponding catalysts as found by N2 physisorption (see Section 3.2 textural characterization).

Interestingly, it has been proposed that carbonaceous deposits could act as reservoir of molecular  $H_2$  dissociated on  $MoS_2$  phases which could then spillover onto the basal planes of graphite-like species through reaction of atomic H with *p* electrons in the basal plane of those carbon species [60]. Accumulated hydrogen stored in deposited carbon could be then delivered to "CoMoS" sites (or "NiMoS" in our case) during the course of HDS reactions significantly enhancing sulfur removal activity [61].

### 3.5. Characterization of SA-modified NiMo/Al<sub>2</sub>O<sub>3</sub> and CoMo/Al<sub>2</sub>O<sub>3</sub> impregnated solids by UV-vis spectroscopy

Comparison of visible spectra of SA-modified NiMo/Al<sub>2</sub>O<sub>3</sub> oxidic formulations impregnated with green (as made) and blue solutions (Fig. 4(a) and (b), respectively) evidenced Mo-blue species characterized by maximum at 698 nm [31] in the latter case whereas in the former absorptions due to octahedral Ni<sup>2+</sup> [27] could be identified. Formation of LMCT saccharide-Mo complex where molybdenum atoms could be partially reduced could be related to absorption at 403 nm [26] in Mo-blue containing sample. Position of bands due to LMCT complexes depends on local symmetry of Mo<sup>6+</sup> and Mo<sup>5+</sup> cations originated by their coordination and aggregation state [62].

In order to assess the feasibility of synthesizing Mo-blue precursors also in the case of alumina-supported CoMo catalysts corresponding

solutions were prepared by using similar methodology (Co/SA = 1) to that used in the case of Ni-based formulations (see Section 2.1. Catalysts Synthesis). The original wine-colored solution (CoMoP-SA) turned very dark after heating at ~ 353 K. In this case, for sake of simplicity, heating was applied to accelerate saccharose acid hydrolysis (see Scheme S1) conditions at which Mo-blue formation took just some minutes. After alumina impregnation by pore-filling, the dried material  $(SA_{CoMo}(1))$ was analyzed by UV-vis DRS. A calcined sample prepared with no SA (CoMo/Al<sub>2</sub>O<sub>3</sub>) was also studied as reference, Fig. S8. In addition to well-known signal at 250 nm corresponding to tetrahedral molybdenum species the inflexion at 337 nm in CoMo/Al<sub>2</sub>O<sub>3</sub> was related to octahedral ones [9]. That signal red-shifted to 355 nm in SA<sub>CoMo</sub>(1) spectra suggesting larger polymolybdates domains [9] in agreement with that found by Raman spectroscopy in Ni-based formulations (see Fig. 3 and Section 3.4). Similarly to the case of SA-modified NiMo/Al<sub>2</sub>O<sub>3</sub> sample impregnated with blue solution absorption at 403 nm evidenced LMCT saccharide-Mo complex [26]. Signals at 700 and 867 nm related to Moblue species [31] were clearly identified. On the other hand, on conventional non-modified CoMo/Al<sub>2</sub>O<sub>3</sub> in the visible region a triplet with maxima at 540, 590 and 642 nm characterized tetrahedral  $Co^{2+}$  species [63]. Enhanced background absorption in visible spectrum range for SA-containing sample was provoked by its dark coloring. Finally, the possibility of easily obtaining supported Mo-blue precursors also in Cobased formulations by using the used synthesis methodology was proven.

### 3.6. Sulfided catalysts characterization of by high-resolution electron microscopy (HR-TEM)

From corresponding electron micrographs (Figs. 5(a)-(e) and S9) some changes in MoS<sub>2</sub> particles morphology caused in prepared sulfided catalysts by SA addition at various contents were detected (Table 2). It was determined that average stacking degree of  $MoS_2$ fringes of around 0.62 nm interplanar d-spacing (S-Mo-S layers in (002) plane) slightly increased whereas slab length diminished in samples prepared by saccharide addition, even at the lowest modifier concentration (SA/Ni = 0.5, SA(0.5)). Average  $MoS_2$  slab length in sulfided SA(0) formulation (Table 2 and Fig. 5(a)) was larger as compared with values commonly reported in the literature for similar catalysts. However, it should be considered that our samples had rather weak interaction with the support as they were not annealed at high temperature (see Section 2.1 Catalyst synthesis). In this line, Duan et al. [64] identified MoS<sub>2</sub> slabs of 5.6 nm average length in NiMo/ $\theta$ -Al<sub>2</sub>O<sub>3</sub> materials where the low surface hydroxyl concentrations that characterizes the used alumina phase carrier precluded high deposited phases-support interaction. Even lower carrier-deposited phases interaction in SA-modified samples due to support-like carboceous remains could be reflected in more stacked MoS<sub>2</sub> slabs during catalysts sulfiding



**Fig. 4.** UV–vis DRS spectra (Schuster-Kubelká-Munk function) of P-modified NiMo/alumina prepared with saccharose (at SA/Ni = 1). (a): Impregnated with green non-aged solution, SA(1)<sub>g</sub>; (b): Impregnated with Mo-blue containing solution, SA(1). (For interpretation of the references to colour in the figure text, the reader is referred to the web version of this article.)



Fig. 5. HR-TEM micrographs of various sulfided catalysts prepared. (a): SA(0); (b) and (c): SA(1); (d) and (e): SA(3).

| Table 2   |
|---|
| Statistical distribution of stacking and length of MoS2 slabs over various sul- |
| fided catalysts prepared from corresponding HR-TEM micrographs (see Fig. 5      |
| and Fig. S9). Sulfided molybdenum dispersion $(D_{M_2})$ also included.         |

| Catalyst                                    | Stacking  | Slab length<br>(nm)  | $D_{Mo}^{a}$                         |
|---|---|--|--------------------------------------|
| SA(0)<br>SA(0.5)<br>SA(1)<br>SA(2)<br>SA(3) | $\begin{array}{r} 3.4 \ \pm \ 0.60 \\ 3 \ \pm \ 0.55 \\ 4 \ \pm \ 0.60 \\ 4 \ \pm \ 0.70 \\ 4 \ \pm \ 0.65 \end{array}$ | $6 \pm 0.50  4 \pm 0.90  4 \pm 1.10  3.5 \pm 1.10  4 \pm 1.00$ | 0.20<br>0.29<br>0.29<br>0.33<br>0.29 |

<sup>a</sup>  $Mo^{4+}$  dispersion calculated from Eq. (3).

#### (Table 2 and Figs. 5(b)-(e) and S9).

That fact suggested, again, weakened MoS<sub>2</sub>-support interaction in samples prepared with SA probably due to deposited carbon residua playing support-like role [50] in agreement with our DTA and Raman spectroscopy results (see Sections 3.3 y 3.4, respectively). As stacking of sulfided Mo particles was not affected by increased amount of carbonaceous deposits over SA(3) as to those over SA(1) preferential deposition of excess organic residua on bare alumina support was suggested. Additionally to the partial Mo<sup>6+</sup> reduction provoked by formation of LMCT complex and according to  $D_{Mo}$  data in Table 2 saccharose could contribute to significantly enhanced alumina-supported molybdenum sulfide particles dispersion. It should be taken into account, however, that information provided by conventional HR-TEM could be limited as it provides insight of imaged "edge-on" sulfided Mo particles contacting the Al<sub>2</sub>O<sub>3</sub> support through edge planes (100) and (110) perpendicular to the S-Mo-S layers where slabs appeared as lines. Details about morphology of those crystallites contacting the carrier through the (001) basal plane parallel to S-Mo-S layers could remain then undetected [65]. That could be provoked by invisibility (by HR-TEM) of basal-bonded MoS<sub>2</sub> slabs [65] that could indeed be more abundant in the catalysts of higher Mo-carrier interaction (among our formulations, the one with no organic modifier SA(0)). Note the apparent lower population of  $MoS_2$  slabs in Fig. 5(a), as to those in the rest of micrographs included.

Decreased interaction between alumina carrier and deposited  ${\rm MoS}_2$  due to "sandwiched" carbonaceous deposits could favor formation of

so-called Type II sites [66] characterized by having significantly augmented HDS activity as to that of conventional Type I sites [67].

Carbonaceous residua could also act as spacer hindering sulfided phases sintering thus improving their dispersion. In this line, Yi et al. reported [68] preparation of NiMo-based catalaysts in presence of polyethylene glycol (PEG) by using their solvent-free solid-state method. Based in their results that group claimed that when in suitable amount PEG addition rendered materials of improved sulfidability and active  $MoS_2$  dispersion the latter effect being caused by carbonaceous residua blocking Ni and Mo phases aggregation avoiding then formation of large clusters.

## 3.7. Characterization of sulfided catalysts by X-ray photoelectron spectroscopy (XPS)

Mo 3d region XPS profiles corresponding to various sulfided NiMoP/Al<sub>2</sub>O<sub>3</sub> catalysts studied are shown in Fig. 6. Decomposition of registered signals provided the data summarized in Table 3. Mo signals were fitted considering doublets (corresponding to  $3d_{5/2}$  and  $3d_{3/2}$ contributions) due to  $Mo^{6+}$  (MoO<sub>x</sub>),  $Mo^{5+}$  (MoO<sub>x</sub>S<sub>y</sub>) and  $Mo^{4+}$  (MoS<sub>2</sub>) species [69]. Oxidic molybdenum species dispersion (Mo<sup>6+</sup>/Al, column 4 of Table 3) progressively decreased by saccharide addition in SA(1) and SA(2) suggesting competition between Mo species and organic moieties for adsorptive sites on alumina surface [15]. Also, total molybdenum (oxidic + oxysulfides + sulfides, Mo<sub>T</sub>/Al, column 5 of Table 3) dispersion decreased in SA(1) as to that in SA(0) probably by aforementioned effect. Decreased Mo-carrier interaction in SA-modified samples could also contribute to lower molybdenum dispersion in agreement with that found in previous studies [9] by temperature programmed reduction where tetrahedral Mo<sup>6+</sup> species in high interaction with the alumina support were essentially absent in SA-modified NiMo/Al<sub>2</sub>O<sub>3</sub> impregnated samples prepared with non-aged as-made green solutions. At high SA concentration, however, it seemed that large amount of deposited carbonaceous residua (see Section 3.4 Raman spectroscopy) could contribute to disperse molybdenum species [70].

Oxysulfides dispersion ( $Mo^{5+}/Al$ , column 3 of Table 3) was clearly diminished in SA-containing solids. In those cases, decreased concentration of that species could be probably related to facilitated  $Mo^{4+}$  formation from partially reduced entities already extant in Mo-blue



Fig. 6. XPS envelope of Mo 3d region of sulfided P-doped NiMo/Al<sub>2</sub>O<sub>3</sub> catalysts prepared with or without saccharose. Peak decomposition to various Mo species considered also shown. (a) SA(0); (b) SA(1); (c) SA(3).

species [40] present in formulations modified by saccharide addition. Sulfided  $Mo^{4+}$  species dispersion (column 2 of Table 3) showed different trend as to that found by our HR-TEM studies (column 4 of Table 2) where that parameter augmented in SA-containing samples. That could be provoked by aforementioned invisibility (by HR-TEM) of basal-bonded  $MoS_2$  slabs [65] that could indeed be more abundant in the catalysts of higher Mo-carrier interaction (among our formulations, the one with no organic modifier SA(0)). It should also be taken into account that geometrical model proposed by Kasztelan [24] rather than considering total molybdeum sulfide dispersion emphasizes the proportion of exposed  $Mo^{4+}$  atoms located on slab edges and corners, being then potential active HDS reaction sites.

Differently to that observed for sulfided catalysts prepared with ethyleneglycol as modifier [3], Mo sulfidability [71] (column 6 of Table 3) augmented in SA(1) and SA(3), as to the sample prepared with no additive. Again, that could probably due to the presence of intermediate partially reduced  $Mo^{5+}$  species already present in solids impregnated with the saccharide (Mo-blue LMCT species) that facilitated molybdenum species sulfidation. In this regard, we found [26] enhanced Mo sulfiding degree in unpromoted alumina-supported molybdenum catalysts modified by thioglycolic acid impregnation which addition resulted in partial  $Mo^{6+}$  species reduction to  $Mo^{5+}$  (by LMCT complexes formation) previously to materials sulfidation.

However, the positive influence on  $Mo^{6^+}$  (and  $Mo^{5^+}$ , in our case) to  $Mo^{4^+}$  conversion of decreased deposited Mo-support interaction due to carbonaceous deposits from SA decomposition during catalyst activation could not be discarded. In this line, the carbon signal observed at binding energy of 288.1–288.9 eV (referred to C 1s signal at 284.6 eV from adventitious carbon) related to O–C=O (carboxyl or ester) groups from organic additive thermal decomposition [72] was registered corresponding data being expressed as C/Al area ratio (column 7 of Table 3). Incremental amount of carbonaceous deposits (that could play "support-like" role [50]) with SA content in sulfided solids was evidenced. Although the amount of those carbonaceous residua was significantly enhanced in SA(3) (see also Sections 3.3 Thermal analysis and 3.4 Raman spectroscopy) as to those in SA(1) the corresponding C/Al area ratio was not notably increased in accordance with extant larger carbonaceous domains of lower dispersion.

Phosphorus dispersion strongly diminished by SA addition (column 8 of Table 3) suggesting polyphosphates formation [6]. Competition

between phosphates and organic moieties for adsorptive sites [15] could be the reason of that fact. Due to strong interaction between phosphate anions and the alumina carrier surface after impregnation phosphomolybdates decomposes into tetrahedral molybdates ( $MOO_4^{2^-}$ ) and phosphate groups [73]. Al<sub>2</sub>O<sub>3</sub> surface passivation by organic moieties adsorption could contribute in inhibiting that phenomenon [4] preserving the integrity of original Mo species.

Large graphite-like carbonaceous domains in SA(3) (see section 3.4 Raman spectroscopy) could contribute to enhanced phosphates adsorption. In this line, it has been reported that graphitic carbon could act as effective adsorbent for adsorption/removal of phosphate from aqueous solutions [74]. XPS analysis of sulfided sample prepared at SA/ Ni = 2, SA(2) in general showed similar composition results as to those of SA(1).

Regarding Ni species (Table 4) the Ni 2p envelope of various samples (Fig. 7) was decomposed by considering three major contributions [75] namely NiS<sub>x</sub> (including Ni<sub>3</sub>S<sub>2</sub>, Ni<sub>9</sub>S<sub>8</sub> or NiS, with binding energy between 853.5 and 854.2 eV), oxidic nickel (binding energy ~856.5–857 eV) and NiMoS phase (binding energy 854.8–856 eV). After catalyst activation (Section 2.3 HDS Reaction Test) segregated sulfided Ni species (not integrated in the "NiMoS" phase) could be present forming a number of different compounds. In this line, among others, Ni<sub>3</sub>S<sub>2</sub>, Ni<sub>9</sub>S<sub>8</sub> and NiS species have been identified after submitting oxidic nickel to sulfiding treatments at similar conditions as to those we used (673 K, under H<sub>2</sub>S/H<sub>2</sub> flow) [76]. The binding energy of oxidic nickel signal strongly suggested presence of Ni<sub>2</sub>O<sub>3</sub> or Ni(OH)<sub>2</sub> [77]. As our formulations were just dried (393 K, Section 2.1 Catalysts Synthesis) presence of NiO species was ruled out considering that high temperature (673 K) calcining could be required for its formation [78].

As the energetic resolution of the XPS used could hardly allow correct decomposition of Ni signals just the mentioned species were considered. In addition to the main photoelectron peaks, our Ni XPS plots (Fig. 7) contain broad "shake-up" satellites where the degree of components overlap in corresponding envelopes was severe as those peaks are superimposed. Thus, the nickel species analysis carried out should be considered semiquantitative [79].

Sulfided NiS<sub>x</sub> dispersion increased with SA content in studied solids (column 2, Table 4). Monosaccharides from saccarose hydrolysis could be converted to chelating compounds (carboxylic acids) by oxidation during partial  $Mo^{6+}$  reduction through Mo-blue formation. For

Table 3

Surface molybdenum, carbon and phosphorous species over various P-doped NiMo/alumina sulfided catalysts prepared with or without saccharose, as determined by XPS analysis. C/Al represents the areal ratio of C signal at BE = 288.1-288.9 eV, over Al peak.

| Sample | Mo <sup>4+</sup> /Al | Mo <sup>5+</sup> /Al | Mo <sup>6+</sup> /Al | Mo <sub>T</sub> /Al | $\mathrm{Mo^{4+}/Mo_{T}}$ | C/Al  | P/Al   |
|--------|----------------------|----------------------|----------------------|---------------------|---------------------------|-------|--------|
| SA(0)  | 0.049                | 0.009                | 0.012                | 0.071               | 0.697                     | 0.00  | 0.0241 |
| SA(1)  | 0.031                | 0.0042               | 0.008                | 0.043               | 0.724                     | 0.049 | 0.0154 |
| SA(2)  | 0.031                | 0.0041               | 0.006                | 0.042               | 0.748                     | 0.052 | 0.0134 |
| SA(3)  | 0.054                | 0.007                | 0.015                | 0.076               | 0.704                     | 0.054 | 0.0750 |

 $Mo_T$  (total surface molybdenum) =  $Mo^{4+} + Mo^{5+} + Mo^{6+}$ .

#### Table 4

| vickel surface species of various P-o | oped NiMo/alumina sulfided cat | lysts prepared with or without saccharose, | as determined by XPS analysis. |
|---------------------------------------|--------------------------------|--|--------------------------------|
|---------------------------------------|--------------------------------|--|--------------------------------|

| Cat.           | NiS <sub>x</sub> /Al | NiMoS/Al        | Ni <sub>Ox</sub> /Al | Ni <sub>T</sub> /Al | Ni <sub>S</sub> /Ni <sub>T</sub> | PR                   | (NiMo) <sub>slabs</sub> | (NiMo) <sub>slabs/DMo</sub> |
|----------------|----------------------|-----------------|----------------------|---------------------|----------------------------------|----------------------|-------------------------|-----------------------------|
| SA(0)<br>SA(1) | 0.001<br>0.0027      | 0.004<br>0.0025 | 0.003<br>0.002       | 0.008<br>0.007      | 0.63<br>0.71                     | 48.8<br>34.1<br>26.7 | 0.076<br>0.081          | 0.38<br>0.28                |
| SA(2)<br>SA(3) | 0.0029               | 0.0021          | 0.003                | 0.008               | 0.84                             | 46.3                 | 0.140                   | 0.48                        |

Ni<sub>Ox</sub>, oxidic nickel; NiS, sulfidic nickel = NiS<sub>x</sub> + NiMoS; Ni<sub>T</sub> (total surface Ni) = NiS<sub>x</sub> + NiMoS + Ni<sub>Ox</sub>; NiS/NiT = 1-(Ni<sub>Ox</sub>/Ni<sub>T</sub>); PR (promotion rate) = (NiMoS/Ni<sub>T</sub>) × 100; (Ni/Mo)<sub>slabs</sub> = promoter ratio = NiMoS/Mo<sup>4+</sup>; (Ni/Mo)<sub>slabs/DMo</sub> = (promoter ratio/ $D_{Mo}$ ).  $D_{Mo}$  from Table 2, as determined by HR-TEM data.

instance, in presence of some oxidizing ions glucose could be transformed to saccharic acid [80] this species being able to effectively chelate Ni<sup>2+</sup> under pH condtions suitable for weak acid ionization. That effect could probably result in improved nickel species dispersion through interaction of organic Ni complex with the alumina carrier [9]. Also, total nickel dispersion was strongly enhanced in the sample of the highest concentration of organic modifier (column 5, Table 4). Although nickel sulfidability augmented with SA concentration in samples that was not necessarily reflected in increased proportion of NiMoS phase (see SA(1) in column 3 of Table 4) as decreased Mo<sup>4+</sup> dispersion in that sample precluded mixed species formation due to limited sites to accomodate Ni in edge-decorating positions. Due to that, promotion rate (PR, fraction of Ni engaged in forming NiMoS phase, column 7 of Table 4) actually diminished in sacharide-modified catalysts. However, the absolute amount of that highly active phase ("promoter ratio", column 8 of Table 4) was clearly enhanced in materials prepared with SA. That meant that MoS<sub>2</sub> edges could be effectively promoted in solids prepared with organic additive. Following the way of reasoning of Nicosia and Prins [6], impregnated phosphomolybdates complexated with Ni promoter atoms could be preserved on the passivated alumina surface provided by SA deposition that resulting in augmented more homogeneous NiMoS phases formation. In addition, those mixed phases could mostly belong to Type II sites of enhanced HDS activity due to weakened intearction with the alumina carrier. Similarly to the case of Mo entities (Table 3), the XPS analysis of nickel species of an additional sulfided sample prepared at SA/Ni = 2, SA(2) showed similar composition results as to that from SA(1).

Finally, column 9 of Table 4 show amounts of NiMo<sub>slabs</sub> phase adjusted by considering Mo<sup>4+</sup> in MoS<sub>2</sub> edges ((NiMo)<sub>slabs/DMo</sub>,  $D_{Mo}$  calculated from HR-TEM data, Table 2). Although that parameter diminisihed over SA(1) and SA(2) it was clearly increased in the sample of the highest SA content.

#### 3.8. Dibenzothiophene hydrodesulfurization (DBT HDS) reaction test

Catalytic activity of various sulfided materials is shown in Table 5. As previously reported [9], SA addition by using a non-aged emeraldgreen impregnating solution resulted in around 22% increase in activity ( $k^1$  basis) as to that over SA(0). However, the effect of utilizing solutions containing partially reduced Mo<sup>5+</sup> species (Mo-blue) was significantly more pronounced. HDS capability of tested catalysts increased with SA content getting maximum on SA(1) with 77% higher activity as to that of the sample prepared with no saccharose. Further enhanced amount of organic modifier was not reflected in improved HDS activity. Even more, at the highest concentration studied saccharide addition resulted clearly detrimental.

To compare the effect on HDS activity of other organic modifiers on otherwise similar catalysts additional formulations were prepared by using the one-pot impregnation and alike Ni, Mo and P loading as in SA(1) material prepared with Mo-blue solution (see Section 2.1 Catalyst preparation). Corresponding sulfided formulations were tested in the DBT HDS as already described (see Section 2.3 HDS Reaction test). Under those conditions a material simultaneously impregnated with NiMoP and ethylene glycol (EG, at EG/Ni = 1) non-chelating additive showed 17% increase in activity (on pseudo first order kinetic constant basis) as to the non-modified conventional NiMoP catalyst, alike to that previously found [3]. Also, citric acid (CA, at CA/Ni = 1) chelating additive was utilized the corresponding sulfided solid showing just marginal improvement ( $\sim 6\%$ ) in HDS properties, as compared to the reference material. Although the positive effect of using citric acid as chelator of Ni or Co promoters is well known [28,45,81] it is also true that its influence strongly depends on the preparation stage of the organic acid addition [82]. Under the low pH conditions of one-pot impregnating solutions used (see Section 2.1 Catalyst synthesis) weakly acidic CA could not be deprotonated being then unable to form citrate anions that could adequately complexate Ni promoter species. It is worth emphasizing that our investigation was focused in the application of the readily scalable one-pot preparation methodology that could be easily applied in the catalysts production at commercial level. Then, that method was the one used to compare the effect of various organic additives.

Activity trend was similar when adjusted by either total Mo or Mo<sup>4+</sup> content in formulations ( $k^2$  and  $k^3$  basis, respectively). However, when Mo<sup>4+</sup> dispersion over various catalysts was taken into account (Table 2, from Eq. (3)) ( $k^4$  basis) differences in S removal capability among various sulfided catalysts were levelled suggesting that amount of Mo<sup>4+</sup> sites could be playing very important role in HDS advantage over SA(1) and SA(2). Then, augmented number of sulfided Mo active sites seemed to be partially reponsible for the observed improved activity (on a per mass basis,  $k^1$ ) of SA-containing materials. Finally, when comparing



Fig. 7. XPS envelope of Ni 2p<sub>3/2</sub> region of sulfided P-doped NiMo/Al<sub>2</sub>O<sub>3</sub> catalysts prepared with or without saccharose. Peak decomposition to various Ni species considered also shown. (a) SA(0); (b) SA(1); (c) SA(3).

| Table | 5 |
|-------|---|
|-------|---|

| Pseudo first order kinetic constant values in DBT HDS over su | sulfided catalysts prepared with and without SA. |
|---|--|
|---|--|

| Cat.            | $k^1 \times 10^{-4}$<br>m <sup>3</sup> kg <sub>cat</sub> <sup>-1</sup> s <sup>-1</sup> | $k^2 \times 10^{-4}$<br>l g <sub>Mo</sub> <sup>-1</sup> s <sup>-1</sup> | $k^3 \times 10^{-1}$<br>l mol <sub>Mo4 +</sub> <sup>-1</sup> s <sup>-1</sup> | $k^4 \times 10^{-23}$<br>at. <sub>Mo4 + edge</sub> <sup>-1</sup> s <sup>-1</sup> | $k^5 \times 10^{-23}$<br>(NiMoS) <sub>slabs/DMo</sub> <sup>-1</sup> s <sup>-1</sup> |
|-----------------|--|---|--|--|---|
| SA(0)           | 1.35   | 11.25   | 1.55   | 1.29   | 3.38  |
| $SA(1)_{g}^{a}$ | 1.64   | 13.71   | -  | -  | -   |
| SA(0.5)         | 2.04   | 17.00   | -  | -  | -   |
| SA(1)           | 2.40   | 20.04   | 2.66   | 1.52   | 5.43  |
| SA(2)           | 2.19   | 18.27   | 2.34   | 1.18   | 5.62  |
| SA(3)           | 0.28   | 2.32  | 0.32   | 0.18   | 0.38  |
|                 |  |   |  |  |   |

 $k^2:k^1$  adjusted by Mo content in formulations.

 $k^3:k^2$  adjusted by Mo<sup>4+</sup> content (as determined by XPS analysis, Table 3) in sulfided formulations.

 $k^4 k^3$  adjusted by atomic Mo<sup>4+</sup> content at edges (as determined by HR-TEM, Table 2) in sulfided formulations.

 $k^5$ : Activity per site,  $k^4$  adjusted by (NiMoS)<sub>slabs/DMo</sub> content (as determined by XPS, Table 4) in sulfided formulations.

<sup>a</sup> Prepared from emerald-green impregnating solution.

HDS performance of various sulfided formulations by adjusting their S removal activity per amount of mixed NiMoS phases entities ( $k^5$  basis, using data from Table 4) on studied catalysts existence of more active sites over materials obtained from Mo-blue precursors seemed to be evidenced. Thus, it appeared that the notably improved DBT conversion registered over SA-modified materials could be provoked by improved dispersion of more active well-promoted Mo<sup>4+</sup> sites. The former effect originated in shorter MoS<sub>2</sub> slabs of slightly higher stacking (Table 2) whereas support-like [50] carbonaceous residua decreasing deposited species-carrier interaction could result in Type II sites of enhanced HDS intrinsic activity [83]. In this line, formation of Type I structures characterized by existence of Mo-O-Al linkages in the MoS<sub>2</sub> basal plane in direct contact with the carrier surface could be precluded by using weakly interacting supports as carbon [84]. It is worth mentioning that even for the material prepared with no SA the possibility of formation of those Mo-O-Al linkages could be rather low considering that the formulation was non-calcined (just dried at 393 K, see Section 2.1 Catalysts Synthesis). In that case, the amount of Mo species in high interaction with the alumina carrier could be diminished.

Very recently other groups [85] have addressed on the convenienence of existence of carbon overlayers on HDT catalyst supports prior to impregnation of active phases precursors. For instance, glucose has been used as carbon precursor being pore-filling impregnated at various contents over titania carrier where the resulting solids were further annealed at 773 K. Increased activity in DBT HDS was found over corresponding C-containing NiMo sulfided catalysts, that effect being however function of the carbonaceous deposits amount. In that case, the highest activity was registed for materials with around 1–2% of initial carbon whereas the catalytic properties progressively decreased by augmenting C-content in studied solids.

We also found dramatically decreased HDS activity over our SA(3) sample, in spite of containing the highest amount of (NiMo)<sub>slabs</sub> phase (column 8 of Table 4) among prepared formulations. We consider that the massive amount of carbonaceous deposits over that formulation (see Section 3.4 Raman spectroscopy of impregnated samples) could be playing decisive role on the observed behavior. Also, the strongly diminished porosity (Fig. S5) of the material prepared at the highest SA content could provoke severely hindered accesibility of reactant molecules to active sites. Even more, the corresponding N2 adsorption isotherm (Fig. S4) strongly suggested formation of inkle-bottle-shaped pores that could significantly affect diffusion/counterdifussion of reactants/products into the catalysts porous network. Additionally, we could not discard the possibility that some of NiMoS phase active sites could be detected although being covered by carbon overlayers considering that XPS probe radiation could penetrate nanometric depths. Then, thus sites could be accounted for but not being available to adsorb reactant molecules. In previous reports, we have observed that on NiMo sulfided catalysts prepared at high organic additive (chitosan) content [86] where large amount of carbon overlayers impeded

adsorption of CO probe molecule used during corresponding FTIR studies of corresponding alumina-supported sulfided phases.

Regarding selectivity to various products, DBT conversion could be carried out through direct desulfurization (DDS, to biphenyl) and hydrogenation (HYD, firstly to hydrodibenzothiophenes, HDBT's) [87] reaction pathways, Scheme S2. At our reaction conditions (see Section 2.3 HDS Reaction Test), neither hydroDBT's nor the product from complete saturation of aromatic rings (bicyclohexyl) were observed. Taking into account that BP hydrogenation to cyclohexylbenzene (CHB) could be strongly inhibited by DBT competitive adsorption under our HDS conditions [88], all CHB produced must have predominantly come from HYD route, through rapid sulfur removal of partially saturated HDBT's [88].

No important differences in selectivity in the DBT HDS (at 50% conversion) over various tested sulfided formulations were found (Fig. S10) suggesting similar reaction mechanism over active sites existent over various prepared materials. That fact suggested alike  $MoS_2$  promotion degree by nickel, as it is know that non-efficient molydenum sulfide slab edges decoration by Ni could provoke both diminished HDS activity and decreased selectivity to the product from DDS route (biphenyl) [89]. It has been claimed that when taking place on Type II sites HDS had enhanced selectivity to products from HYD route. However, that seems to be much more evident in catalysts of high Mo loading (~24%) [90]. On the other hand, those authors [90] found that selectivity to species from both reaction pathways (DDS and HYD, DBT HDS) was similar over Type I and Type II catalysts in materials of molybdenum concentration alike as to that used in the present work (14 vs. 12%, respectively)

Sacharose as additive seemed to play several roles that could contribute as a whole in obtaining materials of enhanced sulfur removal activity. However, further studies focused in several aspects that deserve deeper insigh are clearly needed. Among them, specific studies on the exact nature of LMCT complex formed during Mo partial reduction and the identification of possible formation of oxidized byproducts (carboxylic acids) that could in turn complexate Ni promoter species should be endeavored.

Through its interesting chemistry saccharose results a highly soluble, cheap and environmentally-friendly non-toxic additive to produce catalysts of enhanced HDS activity.

#### 4. Conclusions

Saccharose (SA) was used as organic additive in simultaneously impregnated P-doped NiMo/Al<sub>2</sub>O<sub>3</sub> hydrodesulfurization catalyst. Originally emerald-green one-pot impregnating solutions prepared by MoO<sub>3</sub> digestion (~353 K) in diluted aqueous H<sub>3</sub>PO<sub>4</sub> followed by nickel hydroxycarbonate addition changed to cobalt-blue by dissolving saccharose (SA, at SA/Ni = 0.5–3) and room-temperature aging (2–4 days, depending on disaccharide concentration). Molybdenum-blue species characterized by presence of partially reduced Mo<sup>6+</sup> to Mo<sup>5+</sup> were evidenced by UV-vis spectroscopy. The originally non-reducing sugar was hydrolyzed under the very acidic conditions (pH~2) of impregnating NiMoP solutions producing fructose and glucose which partially reduced Mo<sup>6+</sup> to Mo<sup>5+</sup> species. Monosaccharides oxidation could result in organic acids that could effectively complexate Ni promoter. After sulfiding of nickel-promoted Mo-blue precursors shorter MoS<sub>2</sub> slabs length of enhanced stacking were observed (by HR-TEM). Mo and Ni sulfidability and nickel dispersion (as determined by XPS) increased with the amount of organic modifier. Enhanced hydrodesulfurization activity in liquid-phase dibenzothiophene HDS was registered for catalvst obtained from Mo-blue precursors as to that of corresponding material from conventional emerald-green impregnating solutions (with or without SA). It appeared that carbonaceous deposits from saccharides residua also contributed to enhanced activity through formation of more active sites (Type II) by weakening deposited speciescarrier interaction However, in solids impregnated at high saccharose content (SA/Ni = 3) enhanced mixed "NiMoS" phase formation was not reflected in improved activity probably because excessive amount of carbonaceous deposits from SA residua decomposition during catalyst activation partially plugged the porous network (as determined by N2 physisorption) in sulfided formulations. That fact seemed to provoke limited accessibility of reactant molecules to surface active sites.

Mo-blue precursor obtained through monosaccharides partial reduction seemed to play decisive role in obtaining HDS catalysts of improved properties. Saccharose results a highly soluble, cheap and environmentally-friendly additive to produce catalysts of enhanced HDS activity.

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#### Appendix A. Supplementary data

Supplementary material related to this article can be found, in the online version, at doi:https://doi.org/10.1016/j.apcatb.2018.06.034.

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