

1 Polymer-lipid hybrid membranes as a model
2 platform to drive membrane-cytochrome c
3 interaction and peroxidase-like activity

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9 ABSTRACT

10 Controllable attachment of proteins to material surfaces is very attractive for many applications
11 including biosensors, bioengineered scaffolds or drug screenings. Especially, redox proteins have
12 received considerable attention as a model system not only to understand mechanism of electron
13 transfer in biological systems but also development of novel biosensors. However, current research
14 attempts suffer from denaturation of the protein after its attachment to solid substrates. Here, we
15 present how lipid, polymer and hybrid membranes based on mixtures of lipids and copolymers on solid
16 support provide more favourable environment to drive selective and functional attachment of a model
17 redox protein, cytochrome c (cyt c). Polymer membranes provided chemical versatility to support

18 covalent attachment of cyt c, whereas lipid membranes provided flexibility and biocompatibility to
19 support insertion of cyt c through its hydrophobic part. Hybrid membranes combined the most
20 promising characteristics of both lipids and polymers and allow attachment of cyt c with both covalent
21 attachment and insertion driven by hydrophobic interactions. We then investigated the effect of
22 different attachment strategies on the accessibility and peroxidase like activity of cyt c, in presence of
23 the different membranes. The real-time combination of cyt c with the planar membranes was
24 investigated by quartz crystal microbalance with dissipation (QCM-D). It was possible to selectively drive
25 the insertion of the cyt c into a specific lipid domain of hybrid membranes. In addition, protein
26 accessibility and its functionality were dependent on the specificity of the combination strategy:
27 covalent conjugation of cyt c to polymer and hybrid membranes promoted higher accessibility and
28 supported higher peroxidase-like activity. Taking together, the combination of biomolecules with planar
29 membranes can be modulated such to improve the accessibility of the biomolecules and their resulting
30 functionality for development of efficient “active surfaces”.

31 Introduction

32 Cell membranes consist of phospholipids, glycolipids and a large variety of proteins, responsible for
33 active or passive transport and signalling¹. One strategy to understand how biological membranes are
34 associated with proteins, to support biological processes, is to utilize model membrane platforms. To
35 date, naturally occurring or synthetic phospholipids and amphiphilic block copolymers have been used
36 to create model membrane platforms²⁻⁴ in combination with a variety of different biomolecules
37 including proteins, peptides and receptors. So far, synthetic membranes have been characterized in a
38 three-dimensional assembly (e.g. liposomes, polymersomes, giant vesicles), which has been not always
39 straightforward, due to the challenges associated with characterization of vesicular model systems.
40 Therefore, supported membrane formation by single component (e.g. lipids or polymers) emerged⁵⁻⁹.

41 Deposition of planar membrane on solid support present as advantages compared vesicular
42 membranes¹⁰; (i) a higher amenability for characterization by surface sensitive analytical tools (e.g.
43 atomic force microscopy(AFM), quartz crystal microbalance with dissipation (QCM-D), and surface
44 plasmon resonance (SPR)) and thus the investigation of protein-membrane interaction kinetics in real-
45 time, *in situ* and label-free format¹¹ and (ii) the possibility to pattern the surface to monitor membrane
46 phase separation¹² and selective molecular binding¹³.

47 In addition to single component membrane platforms, phospholipids and block copolymers have been
48 mixed to form hybrid membranes^{5-6,12}. Within hybrid membranes, amphiphilic block copolymers provide
49 improved mechanical and structural stability while phospholipids can provide a better environment for
50 integration of biomolecules^{12, 14-16}. Depending on how lipid and polymer chains are distributed within
51 hybrid membranes, an enhanced control of membrane functionalization with different biomolecules
52 (e.g. proteins) can be achieved¹⁷⁻²⁰. In fact, hybrid membranes can undergo phase domain separation
53 when the fluid phase polymer and gel phase lipids are used to prepare them and this significantly affects
54 proteins-membrane interaction²¹. Additionally, phase domains in hybrid membranes have been created
55 by changing the molar ratio of each component and molecular parameters of lipids and block
56 copolymers, such as phase transition temperature and composition^{12, 22-24}. The control over the
57 membrane-protein combination is essential for the development of “active surfaces” with desired
58 properties. Although most of the hybrid membrane studies have been focused on the mixing of lipids
59 and block copolymers in vesicles, only a few studies have explored the properties of the hybrid
60 membranes on solid supports and their combinations with proteins (e.g. P-glycoprotein) and peptides
61 (e.g. gramicidin A and valinomycin)^{21, 25-27}.

62 Planar membranes on solid support are obtained by either substrate mediated vesicle fusion²⁸⁻²⁹ or
63 Langmuir Blodgett transfer method^{19, 28}, while their characterisation is achieved by a combination of
64 surface methods including QCM-D, AFM and ellipsometry¹². Due to the capability of lipids and

65 copolymers to re-arrange the membrane architecture and generate domains, hybrid membranes
66 represent ideal candidates for understanding protein-membrane interactions³⁰. These domains facilitate
67 the interaction with proteins, in a way similar to the so-called “lipid-rafts” found in cell membranes^{15, 30}.
68 For example, when the DPPC lipids (phase transition temperature of 41°C) were mixed with the PMOXA-
69 *b*-PDMS block copolymer at room temperature, proteins selectively inserted only into the polymer
70 domains because the lipid domains were in gel-phase and they did not support insertion of the proteins
71 ¹². The advantage of combining proteins with synthetic planar membranes relies in preserving protein
72 activity, which can be prohibited when a protein is directly attached to a bare substrate surface, such as
73 gold, silica or glass³¹⁻³⁵. Attachment of proteins with different solid substrates are mainly based on
74 noncovalent interactions, such as ionic or hydrogen bonds and often induces a protein denaturation. For
75 example, it is well known that functional properties of cytochrome c (cyt c) was hampered because it
76 adsorbed strongly on electrodes (e.g. Pt, Hg, Au and Ag) since this adsorption caused large
77 conformational changes and denaturation³⁶⁻³⁸. Moreover, direct electron transfer between cytochrome
78 c and unmodified electrode surface is slow due to undesired contact between the prosthetic group and
79 the electrode³⁹. So far, in order to provide a better environment for cyt c to function, different
80 electrodes which are mainly based on nanomaterials (e.g. carbon nanotubes, graphene and
81 nanoparticles) have been used⁴⁰⁻⁴⁵. Nevertheless, these attempts might not be sufficient to preserve the
82 whole cyt c activity. By forming planar membranes on solid support as a mean of electrode surface
83 modification, (i) a natural biocompatible means of cyt c-substrate interactions, and (ii) effective, highly
84 dynamic platform to host biomolecules can be achieved⁴⁶⁻⁴⁷. Such system improves the protein-
85 membrane-substrate communication for development of highly sensitive and efficient biosensors.

86 Here we combine different solid-supported membranes based on lipids, copolymers and mixtures of
87 lipids and copolymers with cytochrome c, a model protein in order to understand which molecular
88 factors are playing a crucial role in its accessibility and functionality. We attached the cyt c by two

89 different approaches, one based on insertion mediated by hydrophobic interactions and a second one
90 by covalent conjugation of the cyt c with specific functional groups of the copolymer. More specifically,
91 cyt c is a critical signalling molecule, leading to activation of enzymes in the intrinsic pathway of
92 apoptosis upon permeabilization of upper layer of mitochondrial membrane⁴⁸⁻⁵¹. It has a relatively small
93 size of 12 kDa with globular shape and it is known that its binding to the membranes can be driven by
94 either ionic interactions or small hydrophobic domains which drives the spontaneous insertion⁵¹⁻⁵².
95 Moreover, the presence of amine groups of two specific external lysines (Lys72 and Lys73) supports cyt c
96 to be covalently conjugated to the carboxylic groups of the copolymer in the membranes. To date, the
97 combination of cyt c with the biologic membranes mainly takes places through electrostatic interaction
98 with negatively charged surfaces, due to two residual lysines, partial insertion via hydrophobic
99 interaction or complete incorporation into the membrane⁵²⁻⁵³. Both electrostatic and hydrophobic
100 interactions plaid a role on cyt c combination with membranes based on anionic lipids⁵⁴⁻⁵⁶. The
101 contribution of the hydrophobic interaction on cyt c insertion has been studied by using zwitterionic
102 lipid membranes and the results indicated that cyt c induced the disruption of the membrane⁵⁷. To
103 preserve membrane integrity, another approach to combine cyt c with membranes has to be taken into
104 account. In this respect, the covalent conjugation of cyt c to carboxylated nanoparticles, provides a
105 strong binding, leading to an improved stability for biosensing applications⁵⁸. However, up to now, there
106 is no report regarding the effect of combination strategies of cyt c with planar membranes to distinguish
107 its accessibility and the resulting peroxidase-like activity at different membranes.

108 Towards this goal, we first explored the formation of lipid, polymer and hybrid monolayers at the air-
109 water interface to determine the surface pressure at which densely packed films were formed, before
110 transferring them onto silica wafers with the Langmuir-Blodgett method. Then, we characterized
111 integrity, topography and morphology of resulting supported membranes before and after the
112 combination with cyt c by AFM and confocal laser scanning microscopy (CLSM), respectively. QCM-D

113 was used to quantify the amount of cyt c combined with each type of planar membrane and how it
114 affected the viscoelastic properties thus providing insights into the accessibility of cyt c. After
115 combination with different membranes, we assessed indirectly the peroxidase-like activity of cyt c by an
116 Amplex red (AR)-based fluorimetric assay. These different approaches to combine a protein with planar
117 membranes together with the differences in the composition, morphology and properties of these
118 planar membranes serve to indicate, which are the essential factors for equipping different membranes
119 with proteins in order to produce of efficient “active “surfaces”.

120

121 Materials and methods

122 Materials

123 1,2-dipalmitoyl-sn-glycero-3-phosphocholine (DPPC) and 1,2-dipalmitoyl-sn-glycero-3-
124 phosphoethanolamine-N-(lissamine rhodamine B sulfonyl) (ammonium salt), rhodamine B labeled DPPC,
125 were purchased from Avanti Polar Lipids (Alabaster, AL). Cytochrome c from bovine heart ($\geq 95\%$) was
126 obtained from Sigma-Aldrich and reconstituted to a concentration of 500 $\mu\text{g}/\text{mL}$ with phosphate buffer
127 (PB) solution. The phosphate buffer was prepared by using the $\text{Na}_2\text{HPO}_4 \cdot 7\text{H}_2\text{O}$, and NaH_2PO_4 which
128 were purchased from Sigma-Aldrich. N-(3-Dimethylaminopropyl)-N'-ethylcarbodiimide (EDC), sulfo-N-
129 hydroxy-sulfosuccinimide sodium salt (sNHS) and all reported organic solvents were obtained from
130 Sigma-Aldrich. The silica slides (Ultrapak 100 mm thin wafer box) were purchased from Entegris Inc.
131 (Germany) and used for layer deposition.

132

133 Polymer synthesis

134 The amphiphilic diblock copolymer poly(dimethyl siloxane)₉₀-block-poly(2-methyl-2-oxazoline)₁₀ with a
135 carboxylic functional group at the end of the hydrophilic domain (PDMS₉₀-b-PMOXA₁₀-COOH) was
136 synthesized according to the following protocol: monocarbinol-functionalized PDMS-OH was

137 synthesized by anionic ring-opening polymerization of hexamethylcyclotrisiloxane and end-group
138 modification with 2-allyloxyethanol. Afterwards, PDMS-OH was activated with trifluoromethanesulfonic
139 anhydride and chain-extended by cationic ring-opening polymerization with 2-methyl-2-oxazoline
140 monomer. Quenching with triethylamine/water in order to obtain hydroxy-functionalized PDMS-b-
141 PMOXA-OH was followed by end-group modification with succinic anhydride, leading to the final
142 carboxy-functionalized PDMS-b-PMOXA-COOH.

143 The molecular weight of the copolymer was estimated as 8400 g/mol by nuclear magnetic resonance
144 (¹H-NMR) (Figure S1A). The dispersity of the copolymer was 2.09 according to gel permeation
145 chromatography (GPC) (Figure S1B). In order to confirm the presence of the COOH end group, Fourier-
146 transform infrared spectroscopy (FTIR) was performed, showing the characteristic COOH stretching at
147 1730 cm⁻¹ (Figure S1C).

148

149 Area-surface pressure isotherms

150 The area-dependent surface pressure isotherms of lipid, copolymer and mixtures of lipid and
151 copolymer were measured by Langmuir Teflon mini-trough (KSV Instruments, Finland). The trough was
152 equipped with two movable, computer-controlled Delrin barriers for variation of the area per molecule
153 and a Wilhelmy plate made of filter paper for measuring the surface pressure. The trough and barriers
154 were cleaned with high purity chloroform (HPLC grade, Sigma-Aldrich) and ethanol (≤99.8%, Fluka)
155 before each measurement, and the trough was filled with ultrapure water. A fresh Wilhelmy paper was
156 mounted and fully wetted with ultrapure water every time. The copolymer, lipid and hybrid mixtures in
157 chloroform were diluted with chloroform to a concentration of 1 mg/mL. An aliquot was spread onto
158 the ultraclean water subphase with a glass Hamilton microsyringe and left untouched for 10 min to
159 allow for chloroform evaporation. Then, the lipid, copolymer or hybrid monolayers at the air-water

160 interface were compressed with a constant rate of 10 mm min^{-1} . During all measurements, a constant
161 temperature of $23 \text{ }^\circ\text{C}$ was maintained. All reported data represent triplicate measurements.

162

163 Brewster angle microscopy

164 Brewster angle microscopy (BAM) experiments were performed with the EP3SW system
165 (Nanofilm Technologie GmbH, Göttingen, Germany) equipped with a Nd:YAG laser ($\lambda = 532 \text{ nm}$),
166 a long distance objective (Nikon, $20\times$), and a monochrome CCD camera. The size of the BAM
167 image corresponds to $220 \times 250 \text{ }\mu\text{m}^2$, with a resolution of $1 \text{ }\mu\text{m}$. The microscope was installed
168 over the KSV Langmuir trough equipped with two movable barriers that lead to a symmetrical
169 compression. The BAM measurements were done in triplicate in order to obtain the best
170 representation of the images.

171

172 Preparation of silica wafers

173 Silica wafers were cut into slides of 1 cm^2 that were rinsed with ethanol and dried. Before using, slides
174 were placed in a UV ozone cleaner (Jelight Company Inc, Irvine, USA) for 20 min to remove contaminants
175 from the surface. The cleaned slides were immediately used for layer deposition.

176

177 Langmuir-Blodgett transfer

178 The Langmuir monolayers formed at the air-water interface were deposited on the solid substrate
179 (silica wafer or silicon dioxide sensor QSX 303 SiO_2) by a vertical dipping method. The dipping speed was
180 0.5 mm/min for downstroke and upstroke, maintaining the surface pressure at 35 mN/m . The bilayer
181 films were transferred to the silica substrate by immersing the dipper downstroke for deposition of the
182 first layer and lifting it upstroke for deposition of the second layer.

183

184 Confocal laser scanning microscopy

185 For confocal laser scanning microscopy (CLSM), 1-2 volume % of rhodamine B-labeled lipid (Liss Rhod
186 PE) was added to the lipid/copolymer mixture before the LB transfer to the solid support. The supported
187 hybrid bilayer was transferred to a glass slide and imaged within 1 h after the LB deposition. CLSM
188 images were recorded on a Zeiss 880 LSM equipped with a 40x water-immersion objective (C-
189 Apochromat 40x/1.2 W Korr FCS M27). A DBSS 561-10 laser ($\lambda = 561$ nm) with laser power at 2% was
190 used for the excitation of Liss Rhod PE. The fluorescence intensities of the images were analyzed by
191 ImageJ (v. 1.52r).

192

193 Atomic force microscopy

194 Atomic force microscopy (AFM) was performed with a JPK NanoWizard 3 AFM (JPK Instruments AG).
195 AC mode topography images were obtained in air, using silicon cantilevers (Tap150 Al-G, Budget
196 Sensors) with a nominal spring constant of $10\text{--}130$ Nm^{-1} and a resonance frequency of 150 kHz. Images
197 were analyzed with the data analysis software JPK Data Processing (v. 5.0).

198

199 Quartz crystal microbalance with dissipation

200 Quartz crystal microbalance with dissipation (QCM-D) with Q-Sense E1 (Biolin Scientific, Sweden) set
201 up was employed to characterize the combination of cyt c to the membrane on silicon dioxide sensors.
202 Changes in the resonance frequency (ΔF) and energy dissipation (ΔD) of the oscillating sensor chip (QSX
203 303 SiO_2) as a function of time were simultaneously recorded at multiple odd overtones (3rd, 5th, 7th,
204 9th and 11th). All data shown represent recordings at the 7th overtone. In order to estimate the mass of
205 protein attached to the different membranes, the Sauerbrey equation was applied. This equation
206 converts the frequency shift into mass by using simple relation; $\Delta m = -C\Delta f$, where Δm is the mass, C is
207 the proportionality constant (17.7 $\text{ng cm}^{-2} \text{Hz}^{-1}$) and Δf is the frequency shift. After establishing a

208 baseline in aqueous buffer solution, QCM-D measurements were conducted under continuous flow
209 conditions. A flow rate of 50 $\mu\text{L}/\text{min}$ for protein addition and washing processes was delivered by a
210 Reglo Digital peristaltic pump (Ismatec, Glattbrugg, Switzerland). The temperature of the flow cell was
211 fixed at 24.0 ± 0.5 $^{\circ}\text{C}$.

212 Activation of the carboxylic end group of copolymers in the membranes by EDC/NHS

213 For the covalent combination of cyt c to the copolymers in the solid-supported membranes, the
214 functional carboxylic end groups were first activated to amine-reactive NHS esters by submerging the
215 membranes in 3 mL of EDC solution (10 mg/ml in PB) for 10 min under gentle agitation. After washing
216 with Phosphate buffer (PB) for 30 minutes to remove the unbound molecules, the membranes were
217 submerged into sNHS solution (10 mg/ml in PB) for 10 min and washed again with buffer.

218

219 Fluorimetry

220 Fluorimetry was performed with a Spectramax M5e microplate and cuvette reader (Molecular
221 Devices, USA) using a 10 mm light path quartz cuvette (Hellma, Germany), an excitation wavelength of
222 570 nm, and an emission wavelength of 595 nm. All specimens were measured immediately after QCM-
223 D. The silica substrate was placed standing upright in the cuvette, facing the light source. Amplex red
224 (AR) and H_2O_2 were added to a final concentration of 3.3 mM and 0.66 mM, respectively, and the final
225 volume adjusted to 3 mL with 100 mM PB.

226

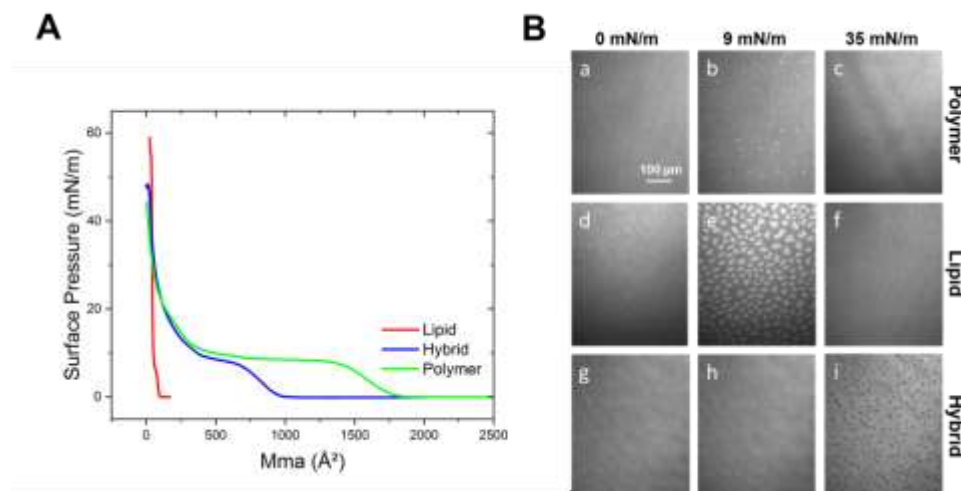
227 Results and Discussion

228 Monolayers at air-water interfaces

229 We measured the changes in the surface pressure of 1,2-dipalmitoyl-sn-glycero-3-phosphocholine
230 (DPPC) lipid, PDMS90-b-PMOXA10-COOH copolymer and the mixture of both with weight ratio 50:50 wt.
231 % as a function of the mean molecular area occupied by a molecule (Figure 1 A). Above the surface

232 pressure of 15 mN/m, all isotherms showed a transition from liquid expanded to liquid condensed
233 phase, irrespective of the composition. Isotherm of PDMS₉₀-*b*-PMOXA₁₀-COOH copolymer showed a
234 lift-off around 1750 Å, a larger plateau and a breaking point higher than 40 mN/m, indicating highly
235 flexible monolayers at the air-water interface, already observed for other PMOXA-PDMS copolymers
236 38. Isotherm of the DPPC lipid exhibited a lift-off at circa 100 Å with a steep slope and a breaking point at
237 60 mN/m. Isotherm of the mixture of the DPPC lipid and PDMS₉₀-*b*-PMOXA₁₀-COOH copolymer, called
238 hybrid, displayed an intermediate profile compared to the isotherms obtained from single components,
239 with a lift-off lower than 1000 Å and a breaking point at 49 mN/m. During the recording of the Langmuir
240 isotherms at air-water interface, the changes in morphology of the lipid, copolymer and hybrid
241 monolayers were real-time monitored by BAM (Figure 1B). During the compression of the copolymers
242 (Figure 1B (a-c)), lipids (Figure 1B (d-f)) and copolymers-lipids mixture (Figure 1B (g-i)) at three
243 representative surface pressure (0, 9 and 35 mN/m, respectively). Initially, when no surface pressure
244 was applied, no film was observed. At the surface pressure of 15 mN/m, the copolymers formed
245 micelles at the air-water interface and then they changed their micellar architecture to a homogenous
246 copolymer monolayer at the surface pressure of 35 mN/m (Figure 1B b-c), in agreement to previous
247 reports for block copolymers⁵⁹. In the case of lipids, flower like assemblies with sizes of approximately
248 10 μm were observed at a surface pressure of 9 mN/m, which transformed to planar lipid monolayers at
249 air water interface for a surface pressure of 35 mN/m (Figure 1 e-f). The lipid-copolymer mixture did not
250 show any assemblies at the intermediate surface pressure of 9 mN/m, and micron sized domains were
251 observed at the surface pressure of 35 mN/m due to separation of lipids from the copolymers (Figure 1
252 h-i). It is important to mention that polymer-lipid phase separation does not completely occur, and each
253 domain is not pure, presenting partially the other component. The lipid domains were found to be
254 embedded into the continuous polymer rich phase, as already observed elsewhere¹².

255



256

257 **Figure 1.** Langmuir isotherms of lipid, copolymer and hybrid (50:50 wt.%) (A) and BAM images of lipid,
258 polymer and hybrid (B) at surface pressure of 0, 9 and 35 mN/m.

259

260 Formation of different model membranes on solid support

261 Monolayers of polymer, lipid and hybrid mixture were transferred to solid support using Langmuir
262 Blodgett method, to create supported polymer, lipid and hybrid bilayer membranes, respectively. First,
263 the monolayers at the air-water interface were deposited to the solid support through an upstroke at
264 a constant surface pressure of 35 mN/m^{8, 17, 26, 60}, resulting in solid-supported monolayers (Figure S2).
265 AFM images showed that the solid-supported polymer monolayer was not homogenous and self-
266 assembled into micellar structures with sizes of 20 nm on solid support, which is in agreement with
267 literature reports (Figure S2A)^{48, 61}. In contrast, the hybrid monolayer was uniform on solid support, with
268 lipid and polymer domains, which had a height difference of 6-8 nm (Figure S2B). This difference was
269 caused by the molecular mismatch between lipids and block copolymers.

270 The supported bilayer membranes were prepared by LB transferring, on a silica support, of two
271 consecutive monolayers, one deposited via an upstroke and a following via a down-stroke⁶². The
272 surface morphology and topography of the resulting bilayer membranes were analyzed by AFM phase

273 and height imaging (Figure 2). Phase profile was necessary to observe the domain separation, whereas

274 the height profile showed the mismatch between the polymer and the lipid phase domains in the hybrid

275 membranes. Surface topography of supported polymer membrane was planar and homogenous (Figure

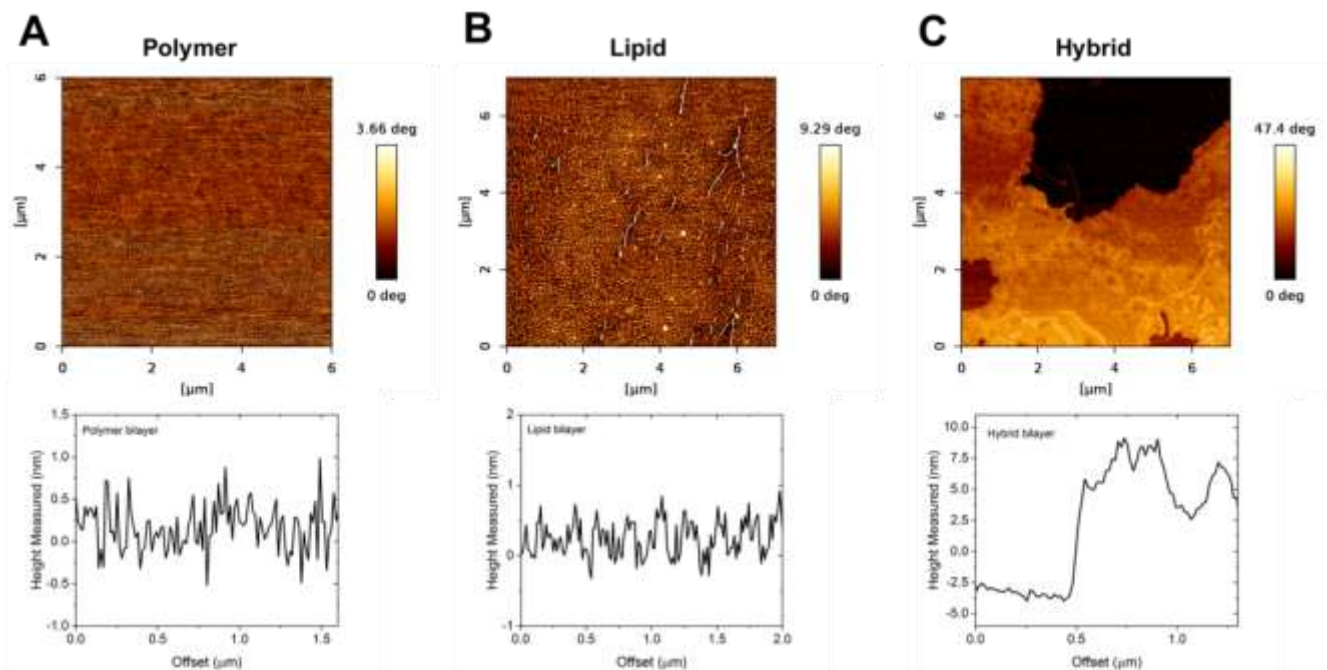
276 2A and S3A), indicating that the deposition of a second layer induced the formation of a stable planar

277 architecture. After drying, there were no micellar structures on the supported polymer membranes, in

278 contrast to the polymer monolayer). As confirmed by AFM profiles, the supported DPPC lipid

279 membranes contained several defects (Figure 2B and S3B), due to the possible overturning mechanism

280 that occurred during the deposition⁶¹. Moreover, supported hybrid membranes showed clear separation
 281 of lipid and polymer domains (Figure 2C and S3C). The height difference between lipid and polymer
 282 domains in hybrid membrane was approximately of 11 ± 1 nm. The AFM phase image shows dark that
 283 represent the lipid domains and bright spots that represent the polymer domains. Domain enrichment
 284 was also observed: small polymer clusters were present in lipid-rich phase while small lipid clusters were
 285 present in polymer-rich domains. As the size of the lipid or polymer domains are tens of micron, there is
 286 no limit for size of proteins, which are either inserted or conjugated to the membranes as long as they



287 have specific characteristics (e.g. surface charge or hydrophobic part for insertion, functional group for
 288 conjugation). Therefore, any hydrophilic peripheral proteins can be inserted or conjugated to hybrid
 289 membranes after necessary adjustments of the membrane composition. For example, extracellular
 290 matrix (ECM) proteins (e.g. collagen with molecular weight of approximately 300 kDa or fibronectin with
 291 molecular weight of approximately 220 kDa) have been attached to the polymer domains through
 292 covalent conjugation and their effect on cell adhesion, proliferation and function have been investigated
 293 for development of cell-based diagnostics, tissue engineering, medical implants and biosensors⁶³.

294

295 **Figure 2.** AFM characterization of solid-supported bilayers. Phase profile (up) and cross section (down)
296 of polymer (A), lipid (B) and hybrid (C) membranes. Corresponding height images are reported in Figure
297 S3.

298

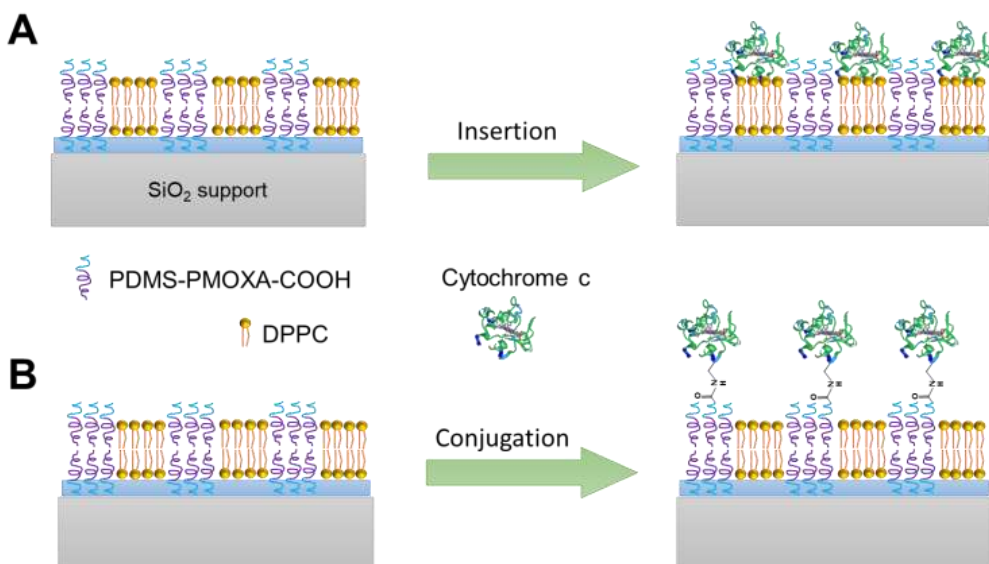
299 We then investigated the presence of lipid-rich or polymer-rich domains in hybrid bilayers, by CLSM
300 (Figure S4A). Rhodamine B labelled lipid was added (1 wt. %) into the solution of the hybrid mixture
301 before deposition onto silica support. The Rhodamine B lipid diffused only into the polymer-rich
302 domains, as indicated by red colour, because they are both in a fluid phase. The blend of a small fraction
303 of lipids with polymers into hybrid bilayers was due to similarity of the phases of the labeled lipid with
304 the polymer. On the contrary, dark domains are specific for DPPC lipids, and are similar in shape and size
305 when compared to BAM images^{62, 64}. To further explore the uniformity of supported model membranes,
306 we evaluated the adsorption of bovine serum albumin (BSA), a globular protein which has been
307 previously used to determine membrane surface coverage^{61-62, 64-65}. BSA adsorption was appreciable on
308 silicon dioxide compared to the planar bilayer membranes made of DPPC lipids or PMOXA-PDMS
309 copolymers⁶⁵. In principle, BSA adsorption is expected to be negligible on the defect-free membranes
310 deposited on silicon dioxide, whereas it increases with increasing amount of defects found in a
311 membrane, e.g. the availability of bare silicon oxide for BSA. We quantified the BSA adsorption onto
312 polymer, lipid and hybrid membranes by QCM-D and compared them with bare silicon dioxide support
313 (Figure S5). A solution of BSA ($C = 500 \mu\text{g mL}^{-1}$) was injected onto bare silica or supported membranes
314 after baselines were established in PB. BSA adsorption on silicon dioxide led to a frequency shift of $-24 \pm$
315 1 Hz whereas on supported polymer membrane of $-2 \pm 1 \text{ Hz}$ which resulted in an appreciable reduction
316 in amount of BSA adsorbed. We calculated a membrane surface coverage higher than $92 \pm 4 \%$,
317 indicating that the polymer membrane did not contain major defects. By contrast, the frequency shift

318 was -6 ± 2 Hz for the BSA adsorption on supported lipid membranes, leading to a reduction of
319 membrane coverage to 75 ± 3 %. Thus, there were more defects in supported lipid membranes
320 compared to hybrid or polymer counterparts. These results were in agreement with AFM data. In the
321 case of BSA adsorption onto supported hybrid membranes, the frequency shift to -4 ± 2 Hz, was
322 associated with a membrane surface coverage of 83 ± 4 %. The presence of defects on the membranes
323 on silica was also compensated by covering the defects with BSA adsorption, which is expected to
324 prevent nonspecific interactions of cyt c with bare silicon oxide.

325 Combination of cytochrome c with supported model membranes

326 After depositing the membranes, we monitored the real-time combination of cyt c with the lipid,
327 polymer and hybrid membranes. We employed two different strategies: i) cyt c insertion into the
328 membrane through hydrophobic interactions, and ii) covalent conjugation (Figure 3) of cyt c to the
329 membranes by binding the accessible lysine of cyt c to COOH functionalized polymer through EDS/sNHS
330 coupling chemistry^{52, 66-67}. The conjugation provides stability to cyt c, preventing the loss of its heme or
331 denaturation. Because cyt c is normally found in the inner cell membranes, the covalent bond provides
332 the needed stability for cyt c when located on the outer membrane⁶⁷⁻⁶⁸.

333



334

335 **Figure 3.** Schematic representation of hybrid supported bilayer and their combination with model
 336 protein cytochrome c through (A) insertion and (B) covalent conjugation by EDC/sNHS coupling.

337
 338 We compared the insertion of cyt c into solid supported membranes by QCM-D (Figure 4A). The QCM-
 339 D measurement baseline signals were stabilized in phosphate buffer. After 5 min of baseline
 340 stabilization, a solution of 500 $\mu\text{g/ml}$ of cyt c was injected. First, we monitored cyt c combination onto
 341 silica surface (Figure S6), resulting in a frequency shift of -16.4 ± 5.8 Hz and a cyt c mass of 290 ± 103
 342 ng/cm^2 . The combination here consisted in an irreversible and not selective process, since it was driven
 343 by either attractive electrostatic interaction between positively charged cyt c and negatively charged
 344 silica and , which could result in protein denaturation. Moreover, cyt c insertion into supported polymer
 345 and hybrid membranes led to a similar frequency shift of -8.0 ± 0.9 Hz and -7.5 ± 2.1 Hz, respectively,
 346 while it led to small frequency shift of -2.4 ± 2.3 Hz for the supported lipid membranes. The mass of the
 347 inserted cyt c was estimated by using the Sauerbrey equation (Table 1). The mass of inserted cyt c on
 348 the supported polymer bilayer was 142 ± 16 ng/cm^2 , 133 ± 38 ng/cm^2 for hybrid bilayer and only 42 ± 40
 349 ng/cm^2 for lipid bilayer. The frequency shifts obtained from cyt c adsorption in the presence of polymer-,
 350 lipid- and hybrid-based layers were lower than the one obtained from bare silica substrate because
 351 PMOXA-PDMS block copolymer has antifouling characteristics and cyt c combination was driven by
 352 specific interactions⁶⁹.

353

354 **Table 1.** Quantification of cytochrome c combined with polymer and hybrid membranes

Membrane type	Frequency Shift ^a		Mass ^a		Molecules of cyt c	
	Insertion	Conjugation	Insertion	Conjugation	Insertion	Conjugation

Polymer	-8.0 ± 0.9	-7.4 ± 0.8	142 ± 16	132 ± 18	7.1 ± 0.8	6.6 ± 0.7
Hybrid	-7.5 ± 2.1	-2.5 ± 0.7	133 ± 38	44 ± 13	6.7 ± 1.9	2.2 ± 0.6
Lipid	-2.4 ± 2.3	n.a. ^b	42 ± 40	n.a.	2.1 ± 2.0	n.a.

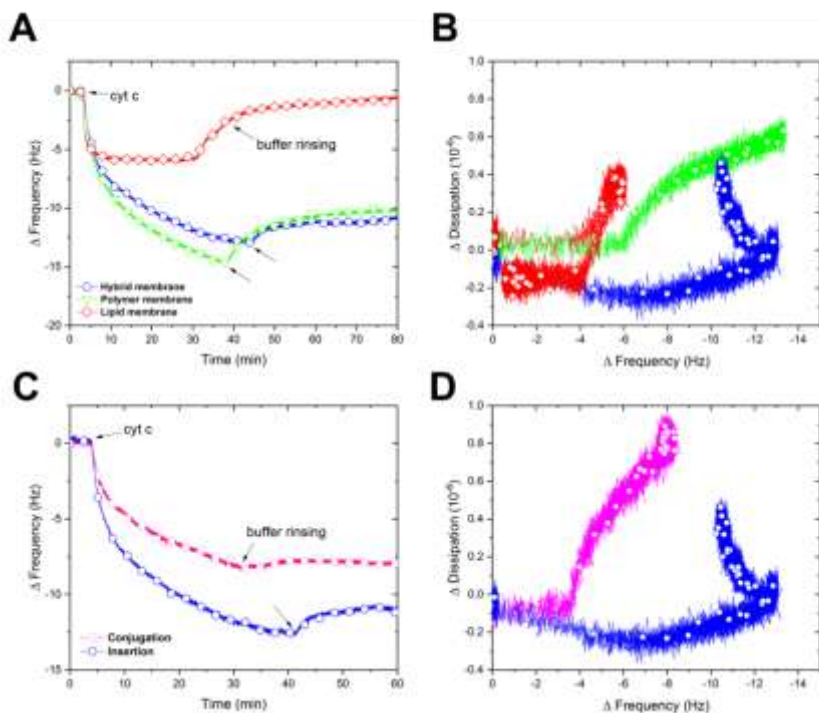
355 ^aThe Frequency is reported in Hz, the mass is reported in ng/cm^2 , the molecules of cyt c have to be
356 multiplied by 10^{12} . ^bNot available, the conjugation was not performed for the lipid because it has no
357 carboxylic end group. The quantity of cyt c combined with the two strategies are compared.

358

359 To gain more insight into the viscoelastic properties of the combined cyt c on the different supported
360 membranes, we used the correlation between dissipation and frequency (Figure 4B), which can be
361 linked to accessibility of the protein on the membrane surface⁷⁰. The dissipation value for the lipid
362 membrane was around $0.1 \cdot 10^{-6}$ and lower than the value determined for the polymer and hybrid
363 membranes ($0.7 \cdot 10^{-6}$ and $0.5 \cdot 10^{-6}$, respectively). The values for dissipation also indicate the difference
364 in the overall flexibility of the different membranes: polymer and hybrid membranes present a higher
365 degree of mobility due to their molecular structure and the resulting membrane packing, compared to
366 lipid membrane 12. This is in agreement with the fact that the lipid is in the gel phase. Secondly, the cyt
367 c was covalently conjugated to supported polymer bilayer and hybrid bilayer, based on the formation of
368 a peptidic bond between the ending functional group of the functionalized polymer and the outer lysine
369 of the cyt c (Figure S7)⁵².

370 The conjugation reaction was performed before the addition of cyt c to the bilayers. The carboxylic
371 groups of the polymers were activated by injecting EDC/sNHS solution ($C = 10 \text{ mg/ml}$) (Table S1) and
372 then rinsed with buffer, before the injection of cyt c solution ($C = 500 \text{ } \mu\text{g/ml}$). Nevertheless, during the
373 conjugation reaction a small amount of cyt c may have also inserted into the membrane. The
374 conjugation of cyt c to the supported polymer bilayer led to a frequency shift of -7.4 ± 0.8 resulting in a
375 mass of $132 \pm 18 \text{ ng}/\text{cm}^2$, while the conjugation of cyt c to the supported hybrid bilayer induced a

376 frequency shift of -2.5 ± 0.7 Hz, resulting in a mass of 44 ± 13 ng/cm² (Table 1). The lower amount of cyt
377 c conjugated to the hybrid membrane is mainly due to the lower number of carboxylic groups available
378 compared to the polymer membrane, according to the molar ratio of 50% in the hybrid. No conjugation
379 was performed for the lipid membrane, due to the absence of the carboxylic functionalization. When we
380 compared the two combination strategies of cyt c for the supported hybrid bilayers, the frequency shifts
381 for the insertion method was higher than for the conjugation (Figure 4C), while the dissipation shifts
382 revealed to be higher for the conjugation, with the highest value of $1.0 \cdot 10^{-6}$ (Figure 4D). The differences
383 in the frequency/dissipation ratio suggest that cyt c has a different conformation when it is combined
384 with the hybrid bilayers by using different combination strategies. The explanation relies on the higher
385 degree of freedom cyt c has when it is conjugated to the membrane, standing away from it, instead of
386 partially penetrating it with its hydrophobic part. Moreover, cyt c preferred a combination with the
387 hybrid membrane through insertion rather than conjugation. However, neutral DPPC lipid cannot
388 establish a strong attractive electrostatic interaction with the positive-charged cyt c. Instead, the
389 presence of anionic lipids in the hybrid membrane may increase the insertion of the protein^{48, 71-74}. The
390 isoelectric point (pI) of cyt c is in the range of 10.0-10.5 and in condition of neutrality it presents a
391 positive charge⁵², so by increasing the pH over this value it would be possible to inhibit the spontaneous
392 insertion due to repulsive interactions between the membrane and the protein⁵². We kept here the
393 system at neutral pH condition.

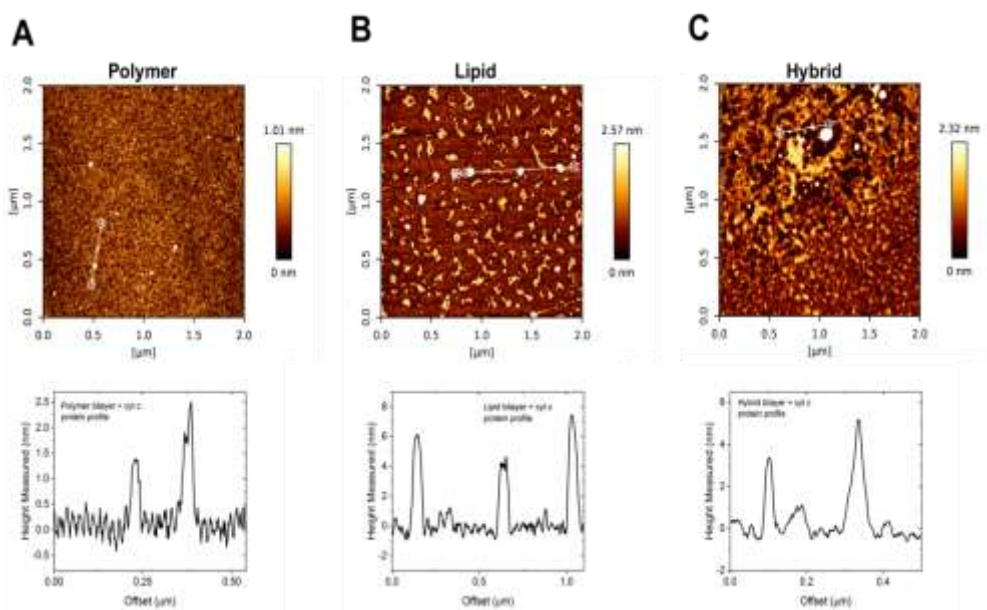


394
 395 **Figure 4.** QCM-D plots of protein combination with supported membranes: protein insertion (A);
 396 comparison of protein combination methods with hybrid membrane (C) and their corresponding ΔD vs
 397 Δf plots (B and D). Solutions used are cytochrome c in PB (500 $\mu\text{g}/\text{ml}$) and pure PB for rinsing steps.

398
 399 After insertion of cyt c within the different membranes, we also monitored the changes in membrane
 400 integrity by analysing AFM height and phase profiles. Because of the small size of cyt c (approximately 3
 401 nm) compared to the domain sizes, we assume that each domain can accommodate more than one cyt
 402 c. After the protein insertion, we observed that, in general, the protein aggregates in clusters (white
 403 spots) of different sizes and average height of 5 ± 2 nm. No modification of the polymer membrane
 404 architecture was observed after cyt c insertion: the synthetic membranes preserved their planarity and
 405 homogeneity, probably due to their robustness (Figure S8B). The low cyt c height (2 ± 1 nm) found in
 406 polymer membrane, might be due to a deeper penetration of the protein into the bilayer, as compared
 407 to lipid and hybrid membranes (Figure 5A). As the lipid bilayer was removed from the silica support,

408 similarly to other reports^{48, 57}, it resulted in a discontinuous membrane (Figure S8A) with cyt c clusters
 409 mainly located on the silica dark background (Figure 5B). Desorption of lipid bilayers from the silica after
 410 addition of cyt c was also confirmed by QCM-D (Figure 4A): the decrease in mass observed before the
 411 buffer-rinsing step indicated the removal of the membrane from the support. Interestingly, the
 412 interaction of the cyt c with the hybrid bilayer indicated no bilayer removal, as expected for the lipid
 413 phase, rather a bilayer reorganization took place, where the lipids self-assembled in the polymer matrix
 414 and constituted rafts accommodating the protein (Figure 5C and S9). The copolymer preserved the
 415 bilayer integrity due to its mechanical resistance while the lipid phase allowed the cyt c to insert due to
 416 its fluidity^{22, 75}. Therefore, the hybrid bilayer was suitable for achieving a selective combination of the
 417 protein with a specific bilayer domain: the lipid domain allowed the insertion while the polymer one
 418 served for the conjugation.

419



420

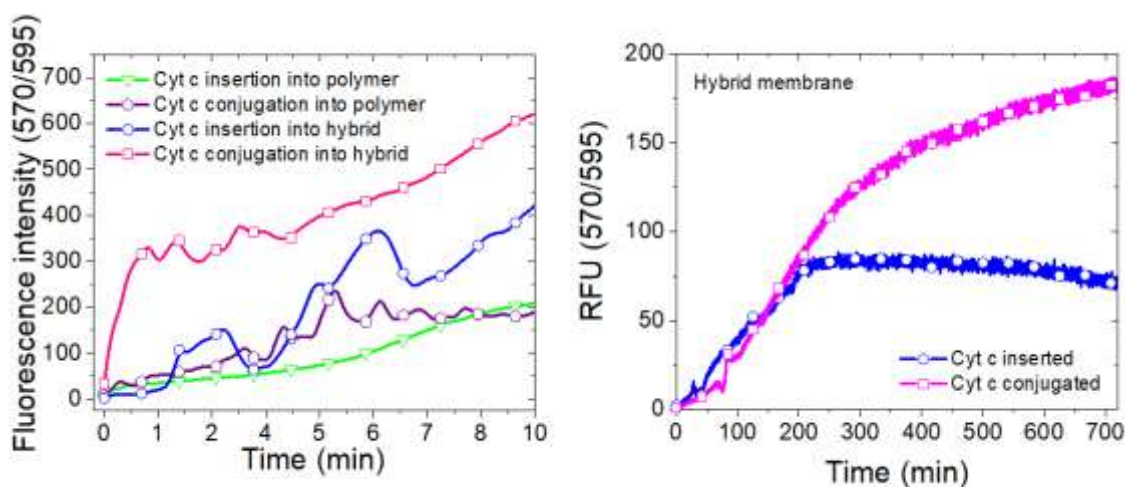
421 **Figure 5.** AFM height profile of different membranes after protein insertion: polymer (A), lipid (B) and
 422 hybrid bilayers (C).

423 Assessment of peroxidase-like activity of cyt c on supported hybrid membranes

424 Cyt c plays an important role in the elimination of hydrogen peroxide (H_2O_2) from mitochondria and
425 respiratory chain, leading to exhibited peroxidase-like activity⁷⁶. The activity of cyt c is based on the
426 reaction of its iron (III) porphyrin, which, in the presence of a cofactor, turns to an oxo-iron species that
427 rapidly oxidize different substrates⁷⁷. We evaluated the peroxidase-like activity of cyt c when combined
428 with supported polymer and hybrid bilayers either by two different combination approaches: i) insertion
429 or ii) conjugation. We did not evaluate solid-supported lipid membranes due to their instability after
430 insertion of cyt c. We assessed indirectly the cyt c activity in the presence of the cofactor H_2O_2 , with an
431 Amplex red (AR) based fluorimetric assay. This colorless, non-fluorescent substrate can be oxidized by
432 the OH^\cdot radicals, produced by cyt c, to the colored and fluorescent resorufin (Figure S10A). A first
433 indication of a successful reaction was the color change of the solution (Figure S10B and S10C). Time-
434 dependent kinetics of resorufin fluorescence intensity (in arbitrary units, a.u.) induced by the addition of
435 H_2O_2 to cyt c serves to evaluate the peroxidase-like activity of cyt c after its combination with supported
436 polymer and hybrid membranes (Figure 6). In order to eliminate the effects of amplex red auto-
437 oxidation from the spontaneous dissociation of H_2O_2 and avoid overestimating the peroxidase-like
438 activity of cyt c, we used the AR auto-oxidation for background correction. We did not expect a
439 degradation of hybrid membranes by the presence of H_2O_2 , as experiments in similar conditions have
440 been already tested for PDMS-b-PMOXA polymers, resulting that no significant membrane disturbance
441 by H_2O_2 were observed⁷⁸⁻⁷⁹. Moreover, a negligible oxidation of AR by H_2O_2 was observed by the
442 fluorometric measurements and AR here served as a scavenger and prevented a possible degradation of
443 lipid domains. The activity of cyt c when combined with polymer or hybrid membranes, was always
444 preserved in different degrees as compared with the free protein (Figure 6A), regardless from the
445 combination method adopted. As expected, cyt c has a lower activity, which can be explained by two
446 factors: i) a reduced conformational freedom of cyt c due to its interaction with the membrane, and ii)
447 the fact that the enzyme only covers a bidimensional space, to which the substrates must diffuse,

448 instead of occupying the entire volume, underlying that diffusion is a fundamental parameter⁷⁹. Besides,
449 the presence of molecules nearby the bilayer interface, might cause further oxidation of the produced
450 resorufin, before it diffuses back into the bulk of the solution, to non-fluorescent dihydroresazurin⁸⁰⁻⁸¹,
451 inducing a drop in the fluorescence signal detected in the initial non-linear kinetics. When the cyt c was
452 covalently bound to supported polymer or hybrid membranes, a higher peroxidase-like activity was
453 determined, compared to cyt c inserted into the membranes. This is due to a higher accessibility of the
454 protein for the substrates when it is conjugated. When cyt c is inserted, a greater density of cyt c on the
455 membrane than the previous case (per QCM-D measurements) does not translate into higher activity,
456 likely because such technique does not ensure an easy accessibility of the enzyme's active site, which
457 can end up buried towards the membrane, whereas the anchoring of cyt c prevents it from inserting
458 with a wrong orientation⁸². Moreover, QCM-D overestimates the density of cyt c on the membranes
459 since bound water was also taken into account when calculating the protein density. Therefore, when
460 the bound water is subtracted, actual density of cyt c on the membranes might be smaller and directly
461 correlated with activity. In order to obtain a rough estimation of the average activity of the protein,
462 inserted and conjugated into polymer and hybrid membranes, the intensity of fluorescence after 10
463 minutes was normalized by the number of cyt c molecules. It was found that cyt c inserted into the
464 polymer membrane exhibits a low activity, with a fluorescence intensity of 30 a.u. The value slightly
465 increased for the cyt c inserted into the hybrid membrane (50 a.u.). A high activity was instead found for
466 the cyt c conjugated to the hybrid membrane, with the highest value of circa 300 a.u. Remarkably, in
467 both cases the activity was increased if the protein was associated to a hybrid membrane, rather than
468 the purely polymeric one, in this case the presence of lipid rafts might play an important role, increasing
469 the flexibility of the protein and the mobility within the membrane and, consequently, improving its
470 accessibility. To study the long-term activity of our hybrid membranes, we followed the enzymatic
471 oxidation of Amplex red for 12 h, for inserted and conjugated cyt c (Figure 6B). Even though the

472 intensity profile is similar within the first 200 minutes, afterwards the activity of the conjugated cyt c
473 drifts to higher values than the inserted cyt c, confirming the sustained and long-lasting activity of the
474 protein with a proper accessibility. Further studies will evaluate in more details cyt c after insertion or
475 conjugation with the hybrid membrane.



476
477 **Figure 6.** Qualitative comparison of protein activity by fluorimetric assay after the combination with
478 different membranes for 10 minutes (A), and with hybrid membrane through different methods for 12
479 hours (B).

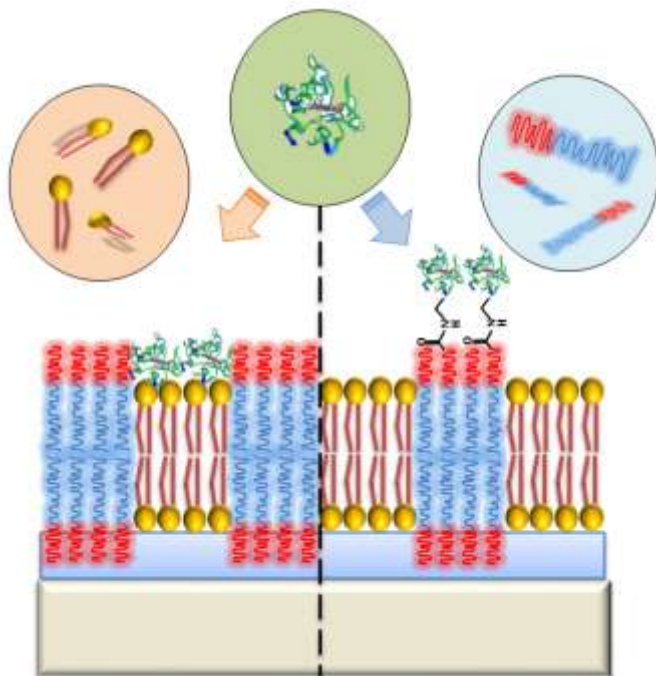
480 481 Conclusions

482 The combination of artificial planar membranes with biomolecules has high potential for development
483 of efficient active surfaces where the functionality is given by the specificity of the biomolecules.
484 Depending on the combination strategy adopted, it is possible to tune the protein distribution into a
485 specific domain of hybrid membranes (polymer- or lipid-rich phase). Here we utilised two strategies –
486 insertion and conjugation – for combining the peripheral membrane protein cyt c with a specific domain
487 of the hybrid membranes based on mixture of PDMS-*b*-PMOXA diblock copolymers and DPPC lipid. The
488 lipid in the hybrid membrane preferentially promoted the cyt c insertion into lipid domains, while the

489 copolymer allowed the conjugation with the cyt c into polymer domains through the carboxylic
490 functionalization. The comparison between the hybrid and one-component membranes (e.g. polymer or
491 lipid) revealed that the protein activity was higher in hybrid membranes, due to the combined
492 properties of lipid and polymer. Moreover, the conjugation strategy resulted as the best approach for
493 increasing the cyt c accessibility and stability, and consequently its activity. Our results open the
494 possibility to extend the combination to different proteins, as for example Horseradish peroxidase
495 (HRP), after proper adaption of the system. Taken together, our results support further development of
496 complex and versatile hybrid bio-interfaces by indicating the molecular factors that are relevant when
497 biomolecules are combined with planar synthetic membranes, especially when they are hybrid and thus
498 contain both lipid- and polymer domains.

499

500 TABLE OF CONTENT



501

502

503 ASSOCIATED CONTENT

504 Supporting Information

505 Characterization of PDMS-*b*-PMOXA diblock copolymer with NMR, FTIR and GPC; schematic
506 representation of protein conjugation strategy; AFM images of polymer mono and bilayer, hybrid and
507 lipid bilayer after protein combination; Schematic representation of protein-bilayer conjugation; QCM
508 data for quantification of EDC/sNHS conjugation for comparison of protein combination with and
509 without BSA step; and comparison of polymer and fluorimetry study for protein activity.

510

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