

1 **Photocatalytic Nitrogen Reduction by Ti₃C₂ MXene Derived Oxygen**

2 **Vacancy-rich C/TiO₂**

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1 **Abstract**

2 In this work, oxygen vacancy-rich C/TiO₂ (OV-C/TiO₂) samples were prepared by a one-step
3 calcination approach using Ti₃C₂ MXene as the precursor, and used for the photocatalytic N₂
4 reduction. The NH₃ yields of all the prepared OV-C/TiO₂ samples exceeded those achieved on
5 commercial anatase TiO₂ and P25, with both H₂O and CH₃OH as the proton sources. Among them,
6 the OV-C/TiO₂-600 offered the remarkable NH₃ synthesis rates, which are 41.00 μmol g⁻¹ h⁻¹ (with
7 H₂O as the proton source) and 84.00 μmol g⁻¹ h⁻¹ (with CH₃OH as the proton sources). The
8 photocurrent and fluorescence spectra showed that OV-C/TiO₂-600 exhibited the highest
9 generation/separation rate and longest lifetime of photocarriers among all the prepared samples.
10 ESR and TPD experiments confirmed much more efficient chemisorption of N₂ on the surface of
11 the prepared OV-C/TiO₂-600 than that on the surface of the commercial anatase TiO₂. Moreover,
12 DFT calculations further demonstrated that N₂ conversion to NH₃ through a Gibbs free energy
13 release leading alternating pathway with a low energy barriers, on the oxygen vacancy on TiO₂
14 surface.

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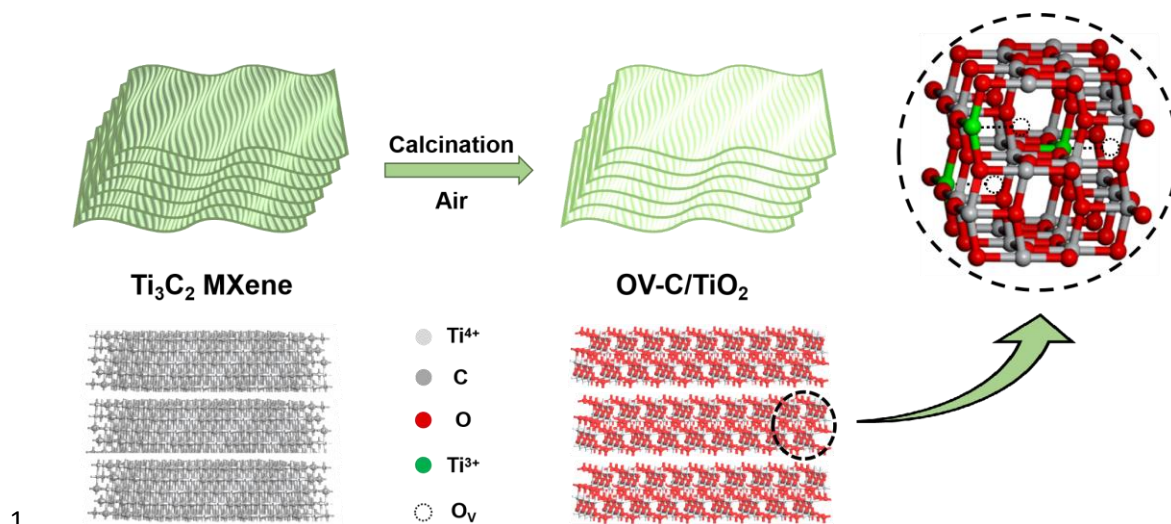
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1. Introduction

Reduction of N_2 to NH_3 is a vital process to promote the growth of the organism.¹⁻⁴ In nature, biological nitrogen fixation and light-driven nitrogen fixation are two main pathways for NH_3 synthesis. However, productions of NH_3 by the two approaches are far away from the agricultural requirements. Hence, artificial nitrogen fixation is a pressing demand for solving the global starvation problem.⁵ More than a century ago, German chemist Fritz Haber contrived Haber-Bosch process to manufacture NH_3 by N_2 and H_2 at drastically high temperature (> 673 K) and pressure (> 200 bar).⁶⁻¹⁰ About a decade ago, Greek chemists Marnellos and Stoukides synthesized NH_3 using H_2 and N_2 diluted by He in an electrolytic cell with the solid electrolyte of strontium-cerium-yttrium-perovskite porous ceramics at 843 K and atmospheric pressure, improving NH_3 yield from 15% (Haber-Bosch process) to 78%.¹¹ Notably, the reaction conditions of the two methods are still very harsh with intensive energy consumption and environmental pollution.¹² Thus a safe, clean and sustainable NH_3 synthesis processes are urgently demanded.¹³

Semiconductor photocatalysis as a clean and sustainable technology has been applied in many fields, such as water splitting, pollutant elimination, CO_2 reduction, CH_4 activation, as well as N_2 fixation.¹⁴ As early as in 1977, Schrauzer and Guth reported the groundbreaking research on photolysis of N_2 to NH_3 on TiO_2 .¹⁵ Since then, the photocatalytic N_2 fixation becomes highly attractive as a promising strategy for NH_3 synthesis under mild conditions.¹⁶⁻²² However, its development is barely satisfactory for the long term in the following decades.²³ This is due to the difficulty of breaking the highly stable $N\equiv N$ triple bond, whose bond energy is as high as 941 kJ/mol.⁴ This is still an enormous challenge for all scientists today. Until recently, some prospective semiconductors, TiO_2 ,^{4, 12, 15, 24-32} $SrTiO_3$,³³ $BaTiO_3$,³⁴ CdS ,⁶ $BiOX$,^{1, 23, 35-37} $g-C_3N_4$,³⁸⁻⁴¹ and WO_3 ,^{42, 43} have been successively used for photocatalytic N_2 fixation. Among them, TiO_2 -based semiconductors are the most stable and robust photocatalysts, but they still suffer from low efficiency. Notably, oxygen vacancies on the surface of TiO_2 ,^{4, 12} and $BiOBr$,^{1, 23} have been proved to be effective for photocatalytic N_2 fixation, due to their favorable adsorption of N_2 molecules and high reduction potential of conduction band electrons. Hence, constructing oxygen vacancies on the surface of semiconductors is a promising strategy to improve the photocatalytic efficiency of N_2 fixation.



Scheme 1 Diagram of the preparation process of OV-C/TiO₂.

3 Ti_3C_2 MXene is a new layered material without noble metal, which was firstly synthesized in
4 2011⁴⁴ and has been widely applied as co-catalysts in photocatalysis⁴⁵⁻⁵¹. Meanwhile, it is also a
5 good TiO_2 precursor, from which abundance of vacancies can be created.^{52, 53} Herein, OV-C/ TiO_2
6 samples were prepared through one calcination step by using Ti_3C_2 MXene as the precursor
7 (Scheme 1), which was synthesized by etching Al layers from the Ti_3AlC_2 in HF at room
8 temperature⁵⁴. Abundance of oxygen vacancies and C doping was observed in the as-prepared
9 TiO_2 sample. Hence, this OV-C/ TiO_2 was prospected to have a high ability to convert N_2 to NH_3
10 under sunlight irradiation. The experimental results in this work showed that, over all the prepared
11 OV-C/ TiO_2 samples under full spectrum light irradiation, NH_3 was successfully generated from N_2
12 with either H_2O and CH_3OH as the proton source. Remarkably, the efficiency of N_2 fixation was
13 achieved at the OV-C/ TiO_2 -600 sample. All the data of NH_3 yields were detected by ion
14 chromatography method to prove the effectiveness of the photocatalyst.⁵⁵ The origin of the N
15 source of the produced NH_3 was studied by the isotope labeled N_2 ($^{14}\text{N}_2$ and $^{15}\text{N}_2$), the different
16 products were identified by ^1H spectra of nuclear magnetic resonance (NMR). This excellent N_2
17 fixation was proposed due to high generation/separation rate of photocarriers with prolonged
18 lifetime, the efficient N_2 adsorption on the abundance of oxygen vacancies, as well as the
19 low-energy-barrier pathway of N_2 to NH_3 on oxygen vacancy. The mechanism of high
20 photocatalytic activity for NH_3 synthesis over OV-C/ TiO_2 we prepared were further discussed
21 based on the measurements of photoelectrochemical experiments, fluorescence spectra, electron

1 spin-resonance spectroscopy (ESR), temperature programmed desorption (TPD) and DFT
2 calculations.

3 **2. Experimental Section**

4 **2.1. Materials**

5 Ti_3AlC_2 MAX phase powders (>98 wt% purity) were purchased from Laizhou Kaixi Ceramic
6 Materials Co., Ltd. (Shandong, China). Ethanol (AR) and hydrofluoric acid were purchased from
7 Sinopharm Chemical Reagent Co., Ltd. Ammonium chloride and sodium hydroxide and sodium
8 were obtained from Shanghai Aladdin Bio-Chem Technology Co., Ltd. Ultrapure water (18.2 M Ω
9 cm) used throughout all experiment was purified through Millipore Milli-Q system. All chemical
10 reagents were used as received without further purification.

11 **2.2. Preparation of OV-C/TiO₂**

12 Ti_3C_2 MXene was synthesized by HF-etching method according to the reported literature.⁵⁴ In
13 detail, 1 g of Ti_3AlC_2 was added into 20 mL 40% HF solution with magnetic stirring at room
14 temperature for 24 hours. The resulting product was washed with deionized water and ethanol
15 several times until pH > 6. The precipitates were sonicated with the dispersion of water for 1 hour
16 and then centrifuged at 3500 rpm for 1 hour to obtain Ti_3C_2 MXene. To prepare OV-C/TiO₂, the
17 prepared Ti_3C_2 MXene was heated to different temperatures (400, 500, 600, 700 and 800°C,
18 respectively) in air at a rate of 5°C/min and kept for 2 hours. The resultant samples were denoted
19 as OV-C/TiO₂-X, herein X represents the annealing temperature, being 400, 500, 600, 700 and 800,
20 respectively.

21 **2.3. Characterizations**

22 The SEM images were collected from Hitachi SU-8010 scanning electron microscopy with
23 an operating voltage of 5 kV. The TEM and HRTEM images were performed using an FEI Tecnai
24 G2 F20 transmission electron microscope operated at 200 kV. The specific surface areas were
25 determined using a surface analyzer (Tristar II3020) through nitrogen adsorption and desorption
26 isotherms at 77 K by Brunauer-Emmett-Teller (BET) method. The XRD patterns were recorded on
27 a Bruker AXS-D8 diffractometer using Cu K α radiation. The UV-vis DRS spectra were recorded

1 by UV-vis spectrophotometer (U-3310, HITACHI) equipped with a diffuse reflectance accessory,
2 using BaSO₄ as the reflectance standard reference. The low temperature ESR spectra were
3 recorded on a Bruker EXM EPR Spectrometer at 77 K. The XPS experiments were carried out by
4 PHI 5000 VersaProbe II XPS system. All binding energies were referenced with respect to the C1s
5 peak (284.6 eV) of the surface adventitious carbon. The photoluminescence spectra decay curves
6 were obtained by using FLsp920 with EPL375 as the light source. The adsorptions of N₂ on the
7 surface of the commercial TiO₂ and prepared OV-C/TiO₂-600 were performed on a TPD
8 equipment (Maiké, Autochem II 2920, TCD detector). Specifically, 100 mg sample was pretreated
9 at 300 °C for 2 hours, and then heated to 450 °C at a rate of 10 °C/min with a gas flow rate of 50
10 mL/min, and kept for two hours.

11 **2.4. Electrochemical tests**

12 The Transient photocurrent responses was performed on a Princeton 283 electrochemical
13 workstation equipped with a standard three-electrode cell with 1M Na₂SO₄ as the electrolyte and a
14 Xe-lamp (HSX-F/UV 300) as the light source. An ITO glass electrode (20×30×0.4mm, 7-10 Ω)
15 coated with photocatalyst (drop coating), a platinum foil and a saturated calomel electrode were
16 employed as the working electrode, the counter electrode and the reference electrode respectively.

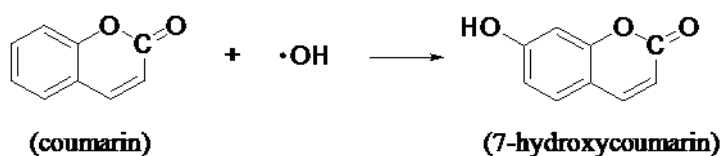
17 EIS test was performed on CHI 660E electrochemistry workstation (Chenhua Instrument)
18 with a standard three-electrode cell system. The working electrode was a glassy carbon electrode
19 (d = 3 mm, S = 0.0707 cm²), the counter electrode was a platinum foil and the reference electrode
20 was a saturated calomel electrode. For preparing working electrode, 5 mg of powder and 0.1 mL
21 of Nafion solution (5 wt %) were dispersed in 0.9 mL absolute ethanol to form a homogeneous ink,
22 and 4 μL ink was loaded onto glassy carbon electrode via drop coating technique. Electrochemical
23 impedance spectra (EIS) were conducted in the frequency range of 1 mHz to 100 kHz under
24 relatively open circuit voltage, using 0.5 M NaSO₄ solution as electrolyte. The EIS spectra were
25 collected both under light irradiation and in the dark.

26 **2.5. Photocatalytic nitrogen reduction**

27 The photocatalytic N₂ reduction experiments were carried out in N₂ atmosphere at room
28 temperature using a 300 W Xe lamp (PLS-SXE300D, 200-800 nm) as the light source, with either

1 H₂O or CH₃OH as the proton source. Typically, 50 mg of catalyst was dispersed in 50 mL of
2 ultrapure water and then transferred to the reactor equipped with water circulation in the outer
3 jacket in order to maintain at the temperature of 25 °C. The methanol concentrations in the system
4 are 0.01 M, when using methanol as the proton source. Before irradiation, the mixture was
5 continuously stirred in the dark with pure N₂ (99.999%) bubbling at a flow rate of 50 ml min⁻¹ for
6 30 minutes and then irradiated for 120 minutes. During the period of a photocatalytic reaction, 1
7 mL of the solution was sampled every 30 minutes for product analysis. The ammonia
8 concentration was detected by ion chromatography (IC, SHINE CIC-D120) equipped with an
9 Ionpac CS16 column (4 × 250 mm, Dionex), a suppressor (SHY-A/C), and a conductivity detector.
10 The eluent was 30 mM CH₄O₃S solution, the flow rate was 0.5 mL/min, and the suppress current
11 was 50 mA. Notably, ion chromatography (IC) method is much more reliable than Nessler or
12 indophenol blue (IB) method.⁵⁵ Before catalysts were removed by filtration, 500 μL 0.1 M HCl
13 solution was added to 500 μL suspension for 5 minutes. Consequently, the obtained solutions were
14 diluted by 4 mL ultrapure water for measurement. The retention time of NH₄⁺ is at about the 11th
15 minute. Absolute calibration of the method was done using ammonium chloride solutions of
16 known concentration as standards.

17 The photocatalytic generations of ·OH over OV-C/TiO₂-600 sample was measured by a
18 fluorescence method using coumarin as a chemical trap.⁵⁶ The photocatalysts were dispersed in a
19 30ml aqueous solution of 1 mM coumarin, and stirred for 30 min in the dark before UV-vis light
20 irradiation. 3 ml suspension was sampled and separated by centrifugation at a given irradiation
21 time interval (5 min). The fluorescence emission intensity evolution of 7-hydroxycoumarin
22 (7-HC), a highly fluorescent product formed by the coumarin and ·OH, was detected at about 450
23 nm under the excitation at 350 nm using a spectrofluorometer (PerkinElmer LS55).



24
25 The production of the liquid product HCHO was measured by colorimetry.⁵⁷ Specifically, 15
26 g of ammonium acetate, 0.3 mL of acetic acid, and 0.2 mL of pentane-2,4-dione were dissolved in

1 water to prepare 100 ml of a color reagent aqueous solution. After diluting 0.5 mL of the test
2 solution with 2 mL of water, mixed it with 0.5 mL of the chromogenic reagent aqueous solution
3 and maintained at room temperature for 1 hour. The absorbance at 412 nm was measured by
4 UV-Vis absorption spectroscopy.

5 The apparent quantum yields (AQY) were determined using monochromatic filter of 400 nm
6 and 420 nm with a bandwidth of 5 nm, to carried out the photocatalytic N₂ reduction. The
7 calculated as follows:¹²

$$8 \quad \text{AQY} = 100\% \times (3 \times \text{generated ammonia molecular number}) / \text{incident photon number}$$

9 **2.6. ¹⁵N₂ isotope Labeling Experiments**⁵⁸

10 An isotope labeling experiments were carried out to clarify the N source of generated NH₃,
11 by using ¹⁴N₂ and ¹⁵N₂ as the feeding gas. The ¹⁴NH₃ and ¹⁵NH₃ were distinguished by NMR
12 spectra of ¹H of NH₄⁺. As displayed in Figure S1, the NMR peak of ¹H is divided into triple peaks
13 by ¹⁴N and and double peaks by ¹⁵N isotopes, for standard ¹⁴NH₄Cl and ¹⁵NH₄Cl. The number of
14 peaks (*N*) in accordance with the theory calculations: $N = 2nI + I$. This formula means that the
15 NMR peak of ¹H will be divided to several peaks due to N isotopes, where *n* represents the
16 number of N atoms, and *I* is the spin quantum number of isotope N: *I* = 1 for ¹⁴N and *I* = 1/2 for
17 ¹⁵N. In addition, from Figure S1, the coupling constant of the triple peaks is 52.3 Hz for ¹⁴NH₄Cl,
18 while the coupling constant of double peaks is 73.2 Hz for ¹⁵NH₄Cl.

19 After the photocatalytic NH₃ synthesis by OV-C/TiO₂-600 for 2 hours, the obtained ¹⁴NH₃
20 and ¹⁵NH₃ were distinguished by ¹H nuclear magnetic resonance (NMR, 400 MHz). The typical
21 NMR signals of ¹H for ¹⁴NH₄⁺ and ¹⁵NH₄⁺ were detected using ¹⁴N₂ and ¹⁵N₂ as the feeding gas. In
22 this process, the prepared NH₃ solutions were concentrated by distillation before being added to
23 deuterated DMSO.

24 **2.7. DFT calculations**

25 The spin-polarized projector augmented wave (PAW) pseudopotentials via the Vienna Ab
26 initio Simulation Program (VASP) were used to perform the density functional theory (DFT)
27 calculations.⁵⁹⁻⁶² The exchange and correlation energy were described by the generalized gradient

1 approximation (GGA) via Perdew-Burke-Ernzerhof (PBE) functional.⁶³ The constituted atomic
 2 valence states adopted were H 1s², N 2s²2p³, O 2s²2p⁴, and Ti 3p⁶3d³4s¹. The plane wave cut-off
 3 energy was 450 eV and a 7×7×3 grid of Monkhorst-pack points were employed for geometry
 4 optimization of a 12-atom conventional cell of anatase TiO₂.⁶⁴ Geometry optimizations were
 5 obtained by minimizing the total energy using a quasi-Newton algorithm to relax ions until it
 6 converged within 10⁻⁴ eV and the forces on each ion were less than 0.05 eV/Å. During the
 7 geometry optimization, volume and shape of the cell as well as atomic positions were relaxed.⁴⁵

8 For the anatase (1 0 1) surface with an oxygen vacancy (S_{OV}), a vacuum region of at least 15
 9 Å was used, so that the slabs in neighbouring cells in the vertical direction are separated by a
 10 sufficiently large distance. During the geometry optimization, the first three O-Ti-O-layers of the
 11 atoms were allowed to relax until reaching a 10⁻⁴ eV convergence in the total energy, while the
 12 forces on each ion were less than 0.05 eV/Å with the three remaining O-Ti-O-layers being kept
 13 fixed (bulk like). To avoid the interaction among adsorbed molecules in neighboring cells,
 14 supercells with 10.3×15.2 Å units of surfaces are constructed.⁴⁵ To investigate the mechanisms of
 15 N₂ reduction on the oxygen vacancy on TiO₂ surface, we carried out the DFT-D3 method with
 16 Becke-Jonson damping.^{65, 66} We studied the most plausible N₂ adsorption configuration on the S_{OV}.
 17 The structure after the geometric relaxation is shown in Figure S2, which was drawn using the
 18 VESTA software.⁶⁷

19 The adsorption energy (E_{ads}) was calculated using the equation:

$$20 \quad \Delta E = E_{\text{ads}} = E_{\text{gas@Sov}} - E_{\text{Sov}} - E_{\text{gas}} \quad (1)$$

21 In Eq. 1, ΔE is the reaction energy, E_{Sov} is the total energy of the anatase (1 0 1) surface with an
 22 oxygen vacancy, E_{gas@Sov} is the total energy of E_{Sov} with gas adsorbed, and E_{gas} is the total energy
 23 of the isolated species (N₂, NH₃, H⁺+e⁻). In the present models, the chemical potential of a
 24 proton-electron pair, μ (H⁺)+μ (e⁻), was equal to the half of the chemical potential of one gaseous
 25 hydrogen, 1/2 μ (H₂).

26 In all fundamental reaction steps, reaction Gibbs free energy changes were calculated using
 27 formula:⁶⁸⁻⁷⁰

$$28 \quad \Delta G = \Delta E + \Delta ZPE - T\Delta S \quad (2)$$

29 In Eq. 2, ΔZPE is the difference to zero point energies due to the reaction,⁶⁶ and ΔS is the entropy

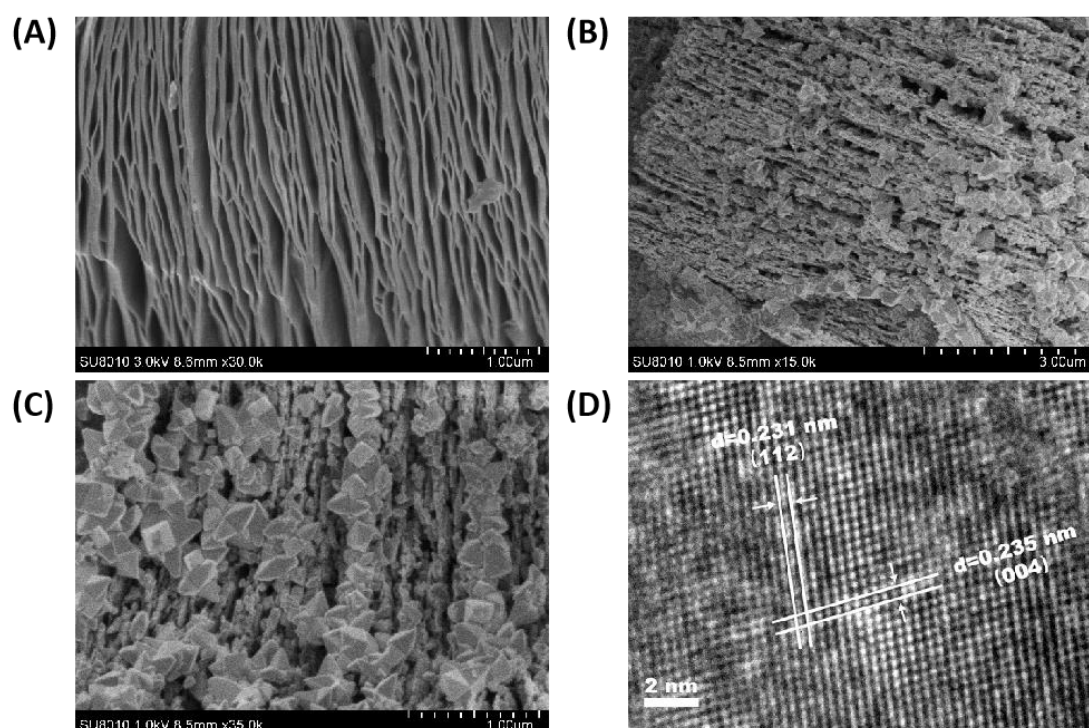
1 change, respectively. Because the temperature $T = 0$ K was used in the ΔG calculation of each
2 reaction step, Eq (1) can be written as:

3
$$\Delta G = \Delta E + \Delta ZPE \quad (3),$$

4 meaning ΔG are corrected via computing the vibrational frequencies of surface species for the
5 ΔZPE .

6 3. Results and Discussions

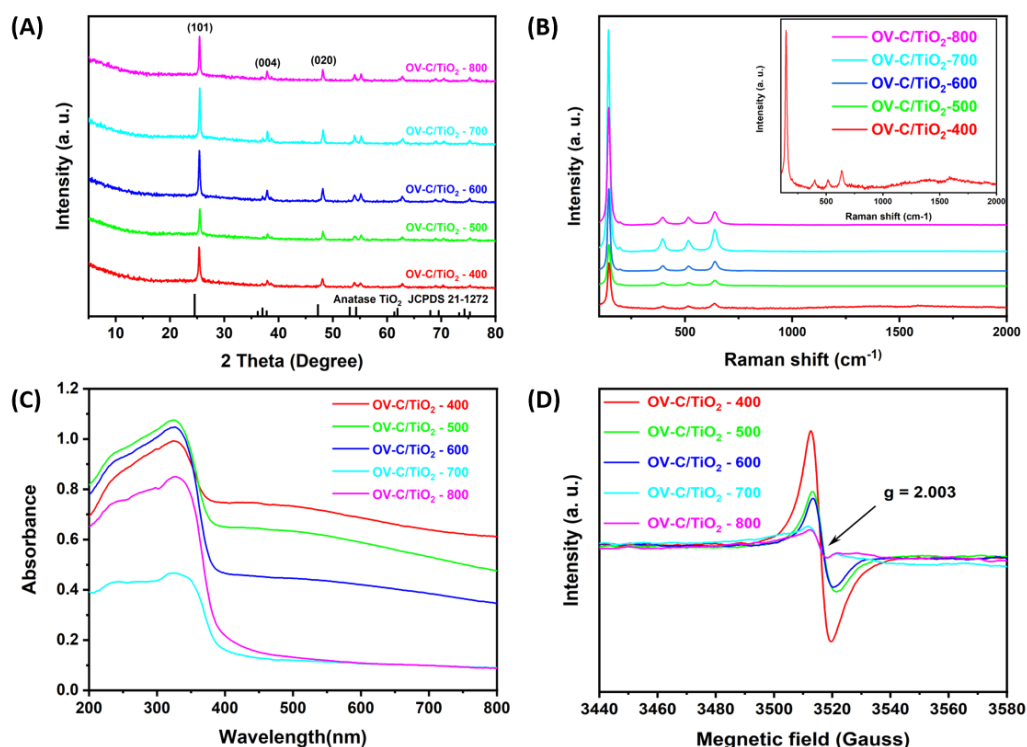
7 3.1. Characterizations



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9 **Figure 1** (A) SEM image of Ti₃C₂ MXene, (B) and (C) SEM images of OV-C/TiO₂-600, (D)
10 HRTEM image of OV-C/TiO₂-600.

11 The micromorphology and architectural structures of Ti₃C₂ MXene and the calcined
12 OV-C/TiO₂ samples were characterized by scanning electron microscopy (SEM). In Figure 1A,
13 the layered structure of Ti₃C₂ MXene is clearly observed, which comes from etching the Al layers
14 of Ti₃AlC₂ MAX phase.^{52, 53} After the oxidation at high temperature in air, its layered structure
15 remained (Figure 1B and S3-6). Meanwhile, type II adsorption isotherms in Figure S7 also suggest
16 multilayer structures of Ti₃C₂ MXene and as-prepared samples.⁷¹ The surface areas of all the
17 prepared samples were also evaluated by Brunauer-Emmett-Teller method and listed in Table S1.

1 All samples show relatively small surface areas, while OV-C/TiO₂-600 has the largest of 16.9 m²
 2 g⁻¹. As shown in Figure 1C, large amounts of small crystal particles (100 nm - 300 nm) with
 3 regular octahedral structure were formed along with the layers. The crystal structure of the
 4 particles was further identified by high-resolution transmission electron microscopy (HRTEM) in
 5 Figure 1D. Two sets of intersectant fringes with the lattice spacing of 0.231 nm and 0.235 nm can
 6 be indexed to (112) and (004) planes of anatase TiO₂,⁷² while no crystal lattice of Ti₃C₂ MXene
 7 was found.⁵²



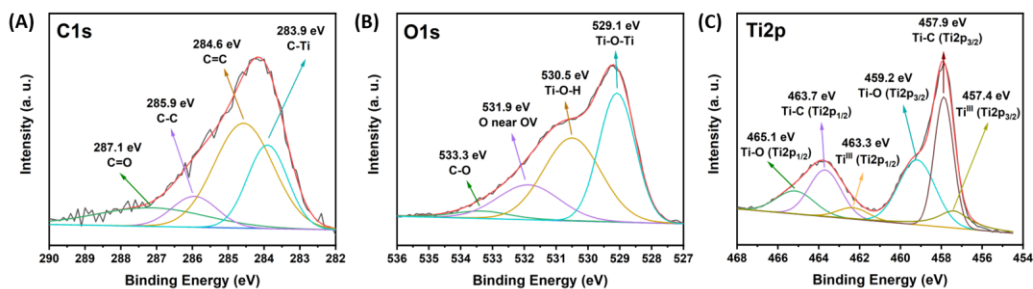
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9 **Figure 2** (A) XRD full spectra, (B) Raman spectra (inset: Raman spectra of OV-C/TiO₂-400), (C)
 10 UV-vis DRS spectra, (D) ESR spectra of OV-C/TiO₂ derived by Ti₃C₂ MXenes calcined
 11 at different temperatures.

12 The crystal structures of calcined samples were further confirmed by X-ray diffractometer
 13 spectra. The XRD patterns described in Figure 2A show good tetragonal crystal phase of anatase
 14 TiO₂, all the peaks are in accordance with the standard database (JCPDS No. 21-1272) of anatase
 15 TiO₂, while no (002) peak of Ti₃C₂ MXene crystal appears at 9°-10°.⁵² The crystal lattice
 16 parameters of OV-C/TiO₂-600 calculated by Jade 5 are listed in Table S2, which is consistent with
 17 the results of HRTEM in Figure 1D. Raman spectra (Figure 2B) was also obtained to confirm the

1 conversion of Ti_3C_2 MXene to TiO_2 . The four sharp peaks below 750 cm^{-1} for all the samples are
2 attributed to the band modes of TiO_2 ,^{45, 52} while the wide peak at the range of $1000 - 2000\text{ cm}^{-1}$ for
3 the sample of OV-C/ TiO_2 -400 (inset of Figure 2B) were attributed to the residual of Ti_3C_2
4 MXene.⁴⁵ However, the peaks derived from residual Ti_3C_2 MXene disappeared when the
5 calcination temperature is higher than 500°C , suggesting the absolute conversion of Ti_3C_2 MXene
6 to TiO_2 . Meanwhile, no peaks at 1350 cm^{-1} and 1590 cm^{-1} related to amorphous carbon⁵² are
7 observed in the Raman spectra of all the samples prepared at the temperature higher than 400°C .
8 More detailed observation in Figure S8 indicates that (101) diffraction peaks of the prepared
9 OV-C/ TiO_2 samples significantly shift to higher 2θ values compared with the standard data, but no
10 regulation with the increase of the calcination temperature. According to Bragg's law, the high 2θ
11 value of the diffraction peaks means shrinkage of the crystal lattice, while the low 2θ value means
12 an expansion of the crystal lattice.⁷¹ Therefore, C doping was proposed, which could cause the 2θ
13 angle shifts to smaller value, but be offset by oxygen vacancies.⁵⁵ The unclear trend was thus
14 believed due to the cooperative effects of the oxygen vacancies introduction and C doping.^{12, 74}

15 The introduction of oxygen vacancies can also be proved by UV-vis diffuse reflectance
16 (UV-vis DRS) and electron spin-resonance spectroscopy (ESR). As displayed in Figure 2C, all
17 samples exhibit an extended absorption in the range of $400 - 700\text{ nm}$ with a threshold at about 400
18 nm. The absorption threshold at about 400 nm can be corresponding to the intrinsic band gaps
19 absorption of TiO_2 ,⁷⁵ while the wide absorption in the range of $400 - 700\text{ nm}$ is attributed to the
20 introduction of oxygen vacancies.⁷⁶ Its lowering trend suggest the decrease of the oxygen vacancy
21 concentration with the increase of calcination temperature.⁷⁶ In addition, electron spin resonance
22 (ESR) was also used to characterize the existence of oxygen vacancies in the prepared TiO_2
23 samples. Herein, distinctive responses near g value of 2.003 in Figure 2D is related to the typical
24 signals of the unpaired electrons of oxygen vacancies,⁷⁷ which shows a decreased trend with the
25 increase of calcination temperature. These results again suggest the decreased oxygen vacancies of
26 OV-C/ TiO_2 samples with the increase of calcination temperature, being consistent with the UV-vis
27 DRS results. In brief, fewer oxygen vacancies were formed in the structure of TiO_2 at higher
28 temperatures.



1

2 **Figure 3** High-resolution XPS spectra of C1s (A), O1s (B) and Ti2p (C) spectra of
 3 OV-C/TiO₂-600.

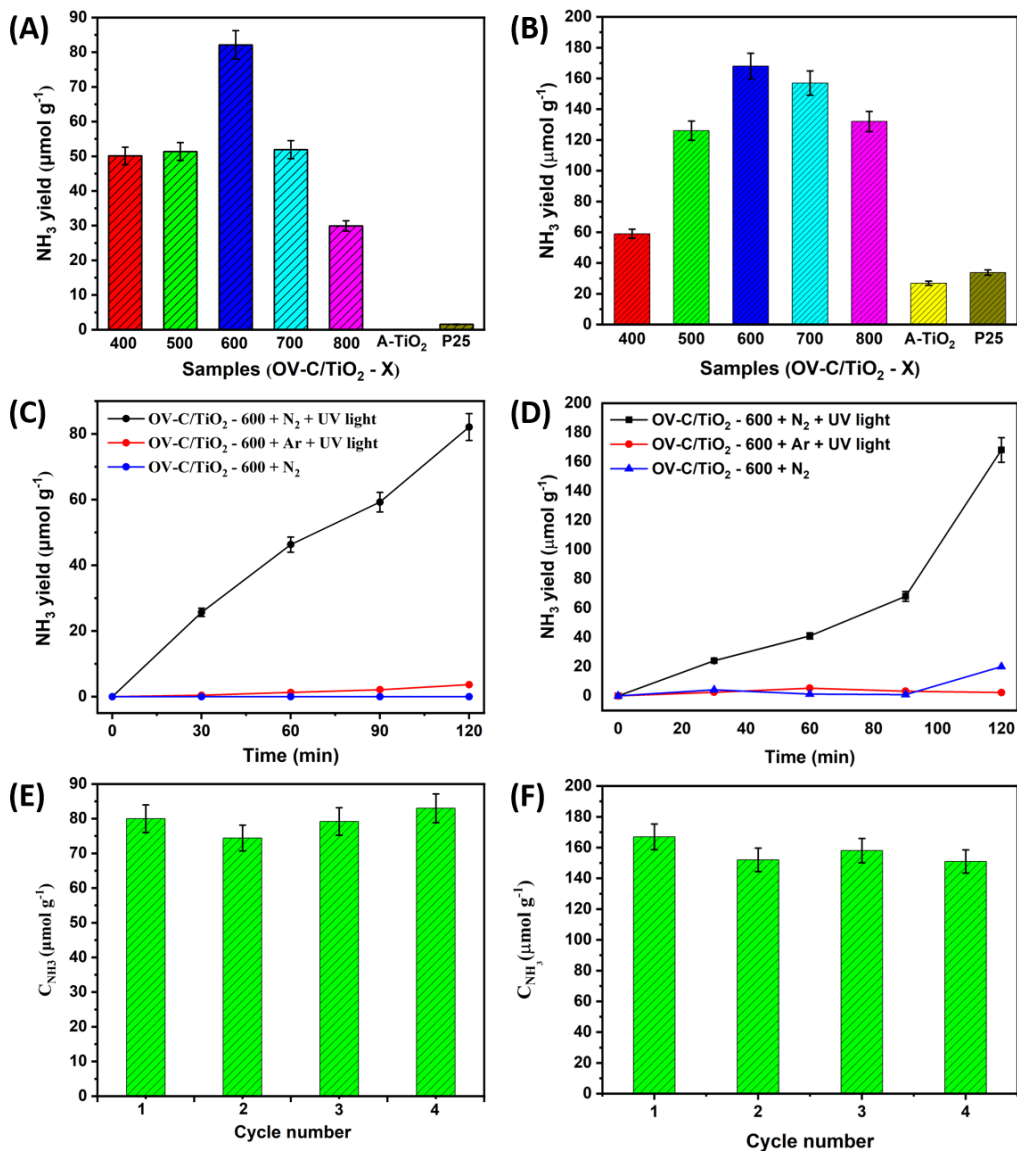
4 X-ray photoelectron spectroscopy was conducted to investigate the chemical compositions
 5 and oxygen vacancies of the prepared OV-C/TiO₂ - 600 sample. The XPS survey spectra in Figure
 6 S9 reveal the presence of Ti, O and C in the sample of OV-C/TiO₂-600. Then, the high resolution
 7 XPS spectra of C 1s, O 1s and Ti 2p are shown in Figure 3A - C, respectively. The C 1s peak in
 8 Figure 3A can be divided into 4 peaks at 287.1 eV, 285.9 eV, 284.6 eV, and 283.9 eV, indicating
 9 the carbon of C=O, C-C, C=C and C-Ti species.^{45, 52} Four types of oxygen are observed in Figure
 10 3B, and the peaks at 533.3 eV, 531.9 eV, 530.5 eV and 529.1 eV are assigned to adsorption oxygen
 11 groups, oxygen near oxygen vacancy, Ti-O-H oxygen and Ti-O-Ti oxygen, respectively.⁷⁸ In
 12 Figure 3C, two peaks located at 465.1 eV and 459.2 eV are ascribed to Ti 2p_{1/2} and Ti 2p_{3/2}
 13 typical Ti-O bond in TiO₂, while the peaks at 463.7 eV and 457.9 eV are attributed to Ti-C bond,
 14 confirming the doping of carbon in TiO₂.⁴⁵ Meanwhile, Ti^{III} vacancy peaks at 463.3 eV and 457.4
 15 eV are also observed, further suggesting the formation of oxygen vacancy.⁷⁸ The oxygen vacancies
 16 could be formed on the interface between Ti₃C₂ MXene and TiO₂ when C in the crystal of Ti₃C₂
 17 reacted with the closest O in the crystal of TiO₂, releasing CO₂.

18 Based on the results and discussions, the samples we obtained by Ti₃C₂ MXene oxidation at
 19 different temperatures are OV-C/TiO₂ nanoparticles, which can distinctly indicate the structure of
 20 the prepared catalyst.

21 **3.2. Photocatalytic fixation of N₂**

22 The prepared OV-C/TiO₂ samples in this work show excellent performances for
 23 photocatalytic NH₃ synthesis with both H₂O and CH₃OH as the proton sources. As shown in
 24 Figure 4A, all the prepared OV-C/TiO₂ photocatalysts exhibited higher activity for ammonia

1 production than the commercial anatase-TiO₂ (0 μmol g⁻¹ h⁻¹) and P25 (1.55 μmol g⁻¹ h⁻¹), when
2 we used H₂O as the proton source. Among them, OV-C/TiO₂-600 generated the most NH₃ with the
3 rate of 41 μmol g⁻¹ h⁻¹, although it has 3 times smaller surface area (16.9 m² g⁻¹) than the referred
4 P25 TiO₂ (50.0 m² g⁻¹). However, it is difficult to detect the production of O₂, the product of hole
5 oxidation with H₂O, which may be due to its over low production. Therefore, we tested the
6 production of ·OH formed by the valence hole oxidation under visible light irradiation, with the
7 coumarin method (Figure S10). Further, we used CH₃OH to replace H₂O as the proton source, the
8 photocatalytic production of NH₃ over all the sample were significantly improved (Figure 4B),
9 and the highest NH₃ production was achieved to 84 μmol g⁻¹ h⁻¹ at OV-V/TiO₂-600. Meanwhile,
10 the generation of HCHO in this system, which was produced by the oxidation of CH₃OH by the
11 valence hole, was also detected by UV-vis absorbance spectra (Figure S11). In addition, the
12 control experiments by commercial anatase TiO₂ particle were also carried out for a comparison.
13 The results in Figure 4C and D further confirm that OV-C/TiO₂-600 photocatalyst is active for
14 photocatalytic N₂ fixation, and negligible NH₃ was observed in the absence of either full spectrum
15 light or N₂. These results indicate that OV-C/TiO₂ is necessary for catalytic synthesis of NH₃, as
16 well as the light irradiation. The prepared OV-C/TiO₂ samples in this work show a comparable
17 NH₃ production rate compare to the other TiO₂-based photocatalysts over the past 5 years (Table
18 S3). This outstanding N₂ fixation performance of OV-C/TiO₂ reveals the critical role of vacancies
19 introduction. It is worth to note that the concentration of oxygen vacancy and C doping in
20 photocatalysts is very important to their photocatalytic activity. As most of the heteroatom doping,
21 oxygen vacancy and C introduction acts not only as the separation center of photogenerated
22 carriers, but also as the recombination center of photogenerated carriers. Therefore, both too much
23 or over low vacancies can not enhance the photocatalytic activity of photocatalyst. This is the
24 main reason for the highest photocatalytic activity was achieved on OV-C/TiO₂-600. In addition,
25 NH₃ generation yields almost remained steady over four photocatalytic cycling experiments,
26 indicating the good stability of OV-C/TiO₂-600 sample (Figure 4E and F). The apparent quantum
27 yields (AQY) of the photocatalytic N₂ reduction over the prepared OV-C/TiO₂-600 were also
28 measured under the homogeneous light, with H₂O as the proton source. They are 0.04% for the
29 light of 400 nm and 0.01% for the light of 420 nm, respectively (Figure S12).

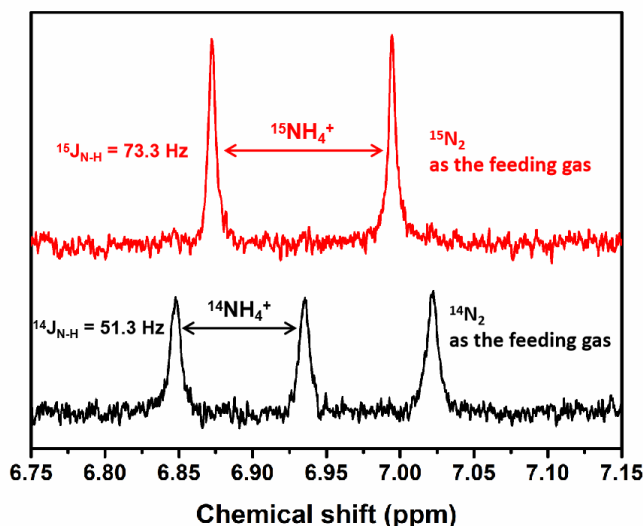


1

2 **Figure 4** (A) NH₃ generation on OV-C/TiO₂ under UV-vis illumination for 2 hours with H₂O as
 3 the proton source, (B) NH₃ generation on OV-C/TiO₂ under UV-vis illumination for 2
 4 hours with CH₃OH as the proton source, (C) Control experiments of NH₃ generations for
 5 OV-C/TiO₂-600 with H₂O as the proton source, (D) Control experiments of NH₃
 6 generations for OV-C/TiO₂-600 with CH₃OH as the proton source, (E) Cycling
 7 experiments of NH₃ synthesis in 2 hours on OV-C/TiO₂-600, with H₂O as the proton
 8 source, (F) Cycling experiments of NH₃ synthesis in 2 hours on OV-C/TiO₂-600, with
 9 CH₃OH as the proton source.

10 Isotope labeling experiments were conducted to confirm that the N source of the produced
 11 NH₃. According to the previous report,⁵⁸ the experimental results by OV-C/TiO₂-600 using ¹⁴N₂

1 and $^{15}\text{N}_2$ as the feeding gas are clearly shown in Figure 5. Being consisted with the standard data
2 in Figure S1, the isotope labeling experimental results support that the produced NH_3 is generated
3 from the direct photo-reduction of N_2 over OV-C/ TiO_2 .



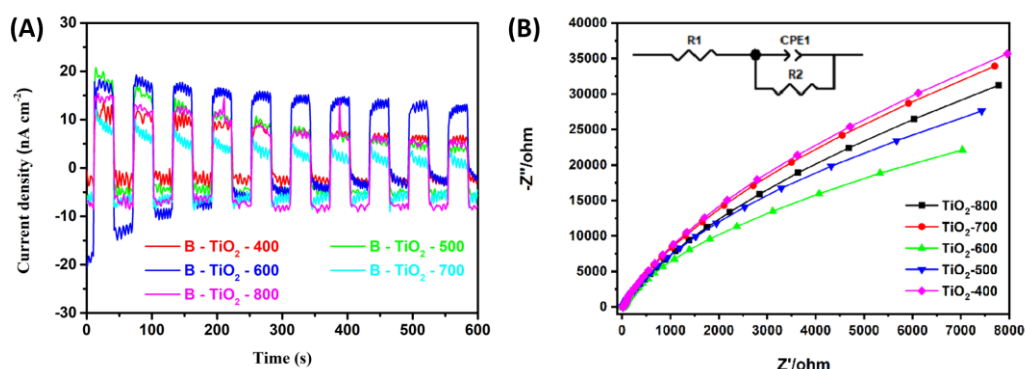
4

5 **Figure 5** ^1H NMR spectra of $^{15}\text{NH}_4^+$ and $^{14}\text{NH}_4^+$ ions produced by OV-C/ TiO_2 using $^{15}\text{N}_2$ and $^{14}\text{N}_2$
6 as the feeding gas.

7 3.3. Mechanism

8 In order to further understand the underlying mechanism on the high photocatalytic N_2
9 fixation performance over OV-C/ TiO_2 -600, some more experiments were carried out. Firstly,
10 photocurrent densities were measured to investigate the generations of electron-hole pairs. As
11 displayed in Figure 6A, the photocurrent density of OV-C/ TiO_2 -600 is slightly larger than those of
12 other samples, suggesting the highest charge generation rate.¹ Secondly, electrochemical
13 impedance spectra both under the light irradiation (Figure 6B) and in the dark (Figure S13) were
14 also performed to study the charge migration of the prepared OV-C/ TiO_2 samples. Smaller
15 diameter of the semi-circle in a Nyquist plot means a smaller charge-transfer resistance for a
16 photocatalyst.¹² The OV-C/ TiO_2 -600 shows the significantly smallest diameters among all the
17 samples both with and without light irradiation, indicating the smallest charge-transfer resistance.
18 Meanwhile, the charge resistances under light irradiation of all the samples are smaller than those
19 in the dark, which may be caused by the electron excitation. Lastly, the lifetime data of the

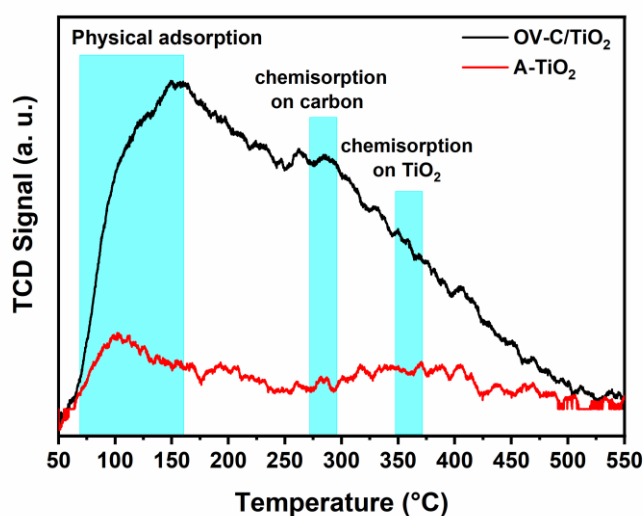
1 photocarriers collected from the time-resolved photoluminescence spectra (Table S4) show that
 2 OV-C/TiO₂-600 has the longest average decay time (5.58 ns) compared with other samples
 3 (commercial TiO₂: 4.80 ns; OV-C/TiO₂-400: 2.88 ns; OV-C/TiO₂-500: 3.35 ns; OV-C/TiO₂-700:
 4 5.44 ns; OV-C/TiO₂-800: 4.08 ns). Longer decay time implies a longer lifetime of the
 5 photogenerated charge carriers. The high generation rate, efficient charge migration and long
 6 lifetime of photogenerated electron-hole pairs on OV-C/TiO₂-600 are considered to be the main
 7 reasons of the highest photocatalytic activity for N₂ fixation, which was caused by the abundance
 8 of oxygen vacancies and carbon doping of the sample. However, oxygen vacancies and doped C
 9 can not only act as the separation center, but also work as the recombination center of the
 10 photogenerated electron-hole pairs.²³ As the optimized adjustment between these two roles,
 11 OV-C/TiO₂-600 shows the biggest photocurrent density and longest decay time.



12
 13 **Figure 6** (A) Photocurrent responses of OV-C/TiO₂ derived by Ti₃C₂ MXene calcined at different
 14 temperatures, (B) Electrochemical impedance spectra Nyquist plots of OV-C/TiO₂
 15 derived by Ti₃C₂ calcined at different temperatures under full spectrum light
 16 illuminations, inset: the equivalent circuit.

17 The adsorption of N₂ on OV-C/TiO₂-600 was next observed by ESR spectra in Ar and N₂
 18 atmosphere, respectively. As observed in Figure S14, the signal of oxygen vacancies in N₂
 19 atmosphere is much weaker than that in Ar, which attributed to the efficient adsorption of N₂ on
 20 oxygen vacancies. Actually, our previous study has also demonstrated that oxygen vacancy is an
 21 efficient chemisorption and activation point of N₂ molecular by DFT calculation, with larger
 22 combination energy release, longer N≡N bond and shorter N-Ti distances than clean TiO₂
 23 surface.⁴⁵ In view of that the chemisorption is an essential step during the photocatalytic N₂

1 reduction processes, N₂-temperature-programmed desorption (N₂-TPD) was further performed to
 2 peruse the adsorption-desorption activation of N₂ on the surface of the prepared OV-C/TiO₂.
 3 Referencing the previous report,⁷⁹ the chemisorption of N₂ usually occurs at about 280 °C (on
 4 carbon) and 360 °C (on TiO₂), while the physical adsorption of N₂ occurs at about 120 °C. As
 5 displayed in Figure 7, the remarkable higher TCD signal of OV-C/TiO₂ than that of the
 6 commercial anatase TiO₂ shows the much better chemisorption of N₂ on the surface of OV-C/TiO₂
 7 than that of the commercial anatase, which is crucial for the N₂ activation.

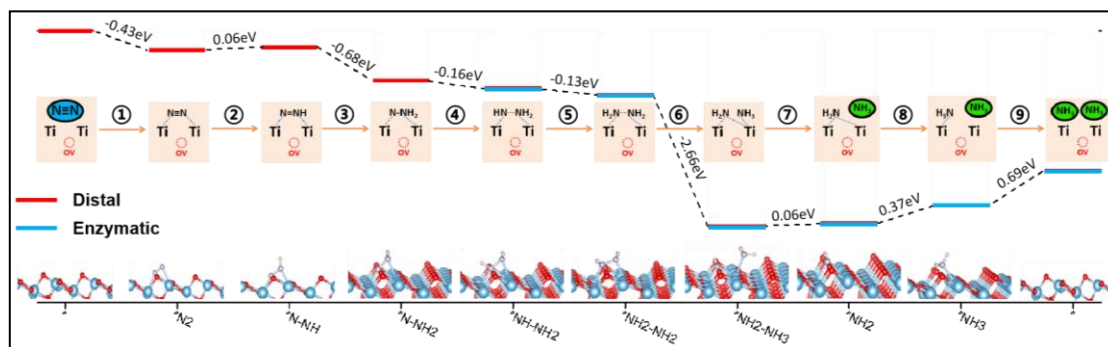


8

9 **Figure 7** N₂-TPD profiles of the commercial anatase TiO₂ and OV-C/TiO₂-600.

10 Following with the adsorption of N₂ (step ①), there are 6 steps of hydrogenation and 2 steps
 11 of NH₃ desorption, as described in Figure 8. In this work, we further calculate the free energy and
 12 bond changes of N-N & N-Ti of every step, to describe the reaction pathways of N₂ conversion to
 13 NH₃ on the oxygen vacancy on TiO₂ surface. The optimized structures of selected N₂ reduction
 14 reaction intermediates are presented in Figure S15-23. As shown in Figure 8, N₂ on the oxygen
 15 vacancy on TiO₂ surface is reduced through an “alternating” pathway, being started from “distal”
 16 and finished by “enzymatic” pathways.⁸⁰ The two N atoms in the N₂ molecules become asymmetric
 17 when it is adsorbed on the oxygen vacancy, resulting in the polarization of N≡N bond. Table S5 lists
 18 the bond distance changes of N-N and N-Ti with the hydrogenation and NH₃ desorption steps,
 19 listing the activation process of N₂ conversion to NH₃ on the oxygen site on TiO₂ surface. Detailedly,
 20 the hydrogenation steps are exothermic processes (③): -0.68 eV; (④): -0.16 eV, (⑤): -0.13 eV and (⑥):

1 -2.66 eV) except step ② to form *N=NH from *N₂ (+0.06 eV) and step ⑧ to form *NH₂ from
 2 *NH₃ (+0.37 eV), while the NH₃ desorption steps are endothermic processes (⑦): +0.06 eV and ⑨):
 3 +0.69 eV). Based on these results, the reaction of N₂ in the oxygen vacancy on TiO₂ surface
 4 reduction to NH₃ is dominated by a Gibbs free energy releasing process, and the potential
 5 determining step (PDS) is ascertained to be the last step involved in the second NH₃ desorption,
 6 which is consisted with the experimental results that the generated NH₃ can absolutely desorb into
 7 the solution only after reacting with 0.1M HCl reagent. Our DFT calculations demonstrated that
 8 oxygen vacancy can effectively polarize N≡N bond and reduce it through a Gibbs free energy
 9 release leading alternating pathway with the second NH₃ desorption as the PDS.

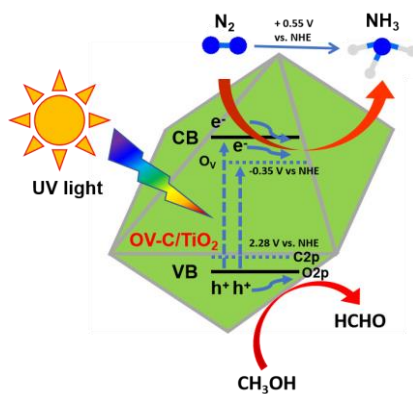


10

11 **Figure 8** Minimum energy path for the N₂ conversion into NH₃ in the site of oxygen on the
 12 surface of TiO₂.

13 Overall, based on the studies and discussions above, a schematic illustration of photocatalytic
 14 N₂ fixation over OV-C/TiO₂ with oxygen vacancies is displayed in Scheme 2. With the doping of
 15 oxygen vacancy and carbon in the crystal of TiO₂, defect energy levels were introduced in the
 16 middle of forbidden zone, forming heteroatom energy band above the valence band⁸¹ and vacancy
 17 energy band¹² under the conduction band. The specific band structure of OV-TiO₂-600 was
 18 obtained by the Tauc-plot curve (Figure S24) and Mott-schokkty plots (Figure S25). Upon the
 19 irradiation of full spectrum light, electrons on the valence band of TiO₂ are excited and populated
 20 into the conduction band or vacancy states. Then the excited electrons on both the conduction
 21 band and defect states react with N₂ adsorbed on oxygen vacancies, producing NH₃ molecular.
 22 Meanwhile, the holes will react with CH₃OH. Several reasons can explain the enhanced N₂
 23 fixation activity. Firstly, oxygen vacancies and doping carbon can broaden the range of the light
 24 absorbance, which helps to provide more electron-hole pairs and promote their separations;

1 secondly, oxygen vacancies capture the electrons on the conduction band, resulting in a prolonged
2 lifetime of photocarriers; thirdly, oxygen vacancies act as adsorption and activation sites for N_2 ,
3 enhancing and increasing the N_2 conversion to NH_3 through a Gibbs free energy decrease leading
4 alternating pathway with a low energy barriers.



5

6 **Scheme 2** Diagram of the photocatalytic N_2 fixation mechanism over OV-C/TiO₂.

7 **4. Summary**

8 In summary, OV-C/TiO₂ with the abundance of oxygen vacancies were successfully prepared
9 by calcining Ti₃C₂ MXene at different temperatures. Notably, this strategy can control the amount
10 of oxygen vacancy through a facial and reproducible process. All the prepared OV-C/TiO₂
11 catalysts show excellent photocatalytic N_2 fixation performance, which achieved the highest of 84
12 $\mu\text{mol g}^{-1} \text{h}^{-1}$ at OV-C/TiO₂ with CH₃OH as the proton source, much higher than P25 and
13 commercial A-TiO₂. The introductions of oxygen vacancies and carbon doping are the most
14 crucial factor for such significant photocatalytic N_2 fixation performance, which not only enhance
15 the generation of electron-hole pairs, reduce the recombination of electron-hole pairs, but also
16 prolong the lifetime of photocarriers and enhance the chemisorption & activation of N_2 molecular.
17 Most importantly, oxygen vacancies play efficient N_2 absorption site and active the N_2 atoms
18 trough a Gibbs free energy releasing leading pathway. The efficiency of photocatalytic N_2 fixation
19 by TiO₂ is thus lifted to a high level, although it is still too low to realize the practical applications.

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8

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