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Wet-spun bi-component alginate based hydrogel fibers: Development and in-vitro evaluation as a potential moist wound care dressing

DOI: 10.1016/j.ijbiomac.2020.12.088

Document Version

Accepted author manuscript

Link to publication record in Manchester Research Explorer

Citation for published version (APA):

Umar, M., Ullah, A., Nawaz, H., Àreeb, T., Hashmi, M., Kharaghani, D., Kim, K. O., & Kim, I. S. (2021). Wet-spun bi-component alginate based hydrogel fibers: Development and in-vitro evaluation as a potential moist wound care dressing. *International Journal of Biological Macromolecules*, *168*, 601-610. https://doi.org/10.1016/j.ijbiomac.2020.12.088

Published in:

International Journal of Biological Macromolecules

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International Journal of Biological Macromolecules Wet-Spun Bi-component Alginate Based Hydrogel Fibers: Development and In-vitro Evaluation as a Potential Moist Wound Care Dressing. --Manuscript Draft--

Manuscript Number:	IJBIOMAC-D-20-05305R1						
Article Type:	Research Paper						
Section/Category:	Carbohydrates, Natural Polyacids and Lignins						
Keywords:	wet spinning moist wound care bio-compatibility hyaluronic acid sodium alginate						
Corresponding Author:	Icksoo Kim Shinshu University Ueda, Nagano JAPAN						
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	Kyu Oh Kim						
	Icksoo Kim						
Abstract:	In this study, bi-component alginate-hyaluronic acid (AHA) fibers were developed by using two different routes. In the first method, sodium alginate dope solution was extruded into a coagulation bath containing CaCl 2 and subsequently dip-coated with hyaluronic acid (HA) whereas, in the second method, hyaluronic acid-containing sodium alginate dope solution was directly extruded into CaCl 2 bath. The resulting AHA fibers were then dehydrated in 25-100% v/v acetone solutions and dried in air. The fibers were characterized by surface morphology, physicochemical analysis, mechanical performance, swelling percentage, and total liquid absorption (g/g), cell viability, and release behavior. The results showed that AHA fibers produced by the second method have better mechanical performance, high liquid absorption, and swelling percentage with a more controlled release of hyaluronic acid. The AHA fibers showed high biocompatibility toward NIH-3T3 cell line in in-vitro testing, and the MVTR values (650 – 800 g/m 2 /day) are in a suitable range for maintaining a moist wound surface proving to be appropriate for promoting wound healing.						
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	Wei Kai, PhD Professor, Suzhou University weikai@suda.edu.cn He is an expert in textile structures.						

Opposed Reviewers:	
Response to Reviewers:	A point to point revision file is attached for the reference.

The Editor International Journal of Biological macromolecules December 02, 2020.

Wet-Spun Bi-component Alginate Based Hydrogel Fibers: Development and *In-vitro* Evaluation as a Potential Moist Wound Care Dressing.

Dear,

It is our pleasure to submit the revision of our **manuscript titled "Wet-Spun Bi-component** Alginate Based Hydrogel Fibers: Development and In-vitro Evaluation as a Potential Moist Wound Care Dressing." to "International Journal of Biological macromolecules." This manuscript contains only original data, which has not been published elsewhere, nor is submitted, in the press, or under consideration simultaneously for publication elsewhere.

All authors fully participated in the preparation of the manuscript and accepted responsibility for the results presented. The authors also declare no conflict of interest.

We welcome the reviewers and editorial comments. As these will improve the quality of our work. We tried our best to provide data and answer all the queries of the reviewers in the limited available time and unusual closure and limited accessibility of labs due to COVID 19 pandemic situation. All the possible minor revisions have been addressed.

We hope the work we are submitting meet the Journal's standard and will bring an excellent impact at large.

Thanking you in anticipation.

Regards,

Optimie

Rising Star Professor, Ick Soo KIM, Dr. Eng. 3-15-1, Tokida, Ueda, Nagano, 386-8567, JA PAN Tel : <u>+81-268-21-5439</u>, Fax : <u>+81-268-21-5482</u> E-Mail : <u>kimicksoo@hotmail.com</u>, <u>kim@shinshu-u.ac.jp</u> <u>https://sites.google.com/site/kimlabgr/Associate</u> Editor for RSC Advances, Royal Society of Chemistry

Comments from Editor and Reviewers:

We welcome the reviewers and editorial comments as these will improve the quality of our work. We tried our best to provide data and answer all the queries of the reviewers in the limited available time and unusual closure and limited accessibility of labs due to the COVID 19 pandemic situation. All the possible major revisions have been addressed.

Note: Responses to the reviewer comments are highlighted in red color in the revised manuscript.

Reviewer #1:

1. Which sterilization method was applied before cell experiments?

The fiber mats were sterilized with 100% ethanol for 10 min and dried in the air. Afterward, UV sterilized for 24 h. the sample specimen was not subjected to PBS washing as they dissolve due to ion exchange.

New cell study with nHDF added as per the suggestion of Reviewer 2.

2. The standard error of the control group has to be given in Fig. 8.

Standard errors for control group cannot be added as it was always taken as 100%.

3. In Figure 9. SEM images are given. The figure captions of 9a, 9b,)g has to be specified. The authors declare that they represent the cell adhesion images on the fibers. However, I could not see cells on the fibers.

Also, magnification values have to be given for each image.

A new cell study with Normal human dermal fibroblast (NHDF) was conducted and added as per the suggestion of **Reviewer 2**.

Fig. 9. Caption is revised as per your suggestion.

4. The characteristic peaks of the fibers should be labeled and identified on the FTIR graph. It is revised as per suggestion.

5. Why the FTIR graph of the S7 example is absent?

S7 is the neat alginate fiber. Caption and legends are changed for clarification.

6. What kind of information we have from the results obtained with the linear density experiment. Why did the authors need the analysis, and which information about the material they had from the analysis? It should be concluded. Added as per suggestion.

Fiber linear density influence the properties of the products produced from them. Linear density is directly proportional to the air and moisture permeability of the fabric produced from it. The finer the fibers greater will be the surface area and coverage of the fabric. Thus a fabric made of finer fibers will have low MVTR and air permeability and will present better barrier performance than the fabric produce of high linear density. Furthermore, low linear density fibers have greater flexibility, soft and pliable, thus improving the final produced product's comfort.

Reviewer #2:

This paper reports the spinning and characterization of alginate-hyaluronic acid blend fibers using the wet-spinning technique. The as-spun fibers were further made into nonwovens to explore its potential in wound healing. The fibers developed in this work might have a practical application, and the manuscript is well organized and well written. My comments are as follows:

1. The material part of this paper is systematic, but the biological part is too weak to demonstrate the potential application of the fibers. The authors should at least conduct more in vitro assessments, such as cell morphology observations. The selection of cells is also questionable. It is better to use dermal cells for the biocompatibility tests.

A new cell proliferation and cell adhesion study is conducted using nHDF and results are shared.

Cell viability plays an important role in living tissues and thus is an important factor for the tissue engineering and regenerative medicine. To study the cell activity the mitochondrial activity of the nHDF cell on AHA fibers, control alginate and Dimora fibrous mats after 1, 3, and 7 days of culture was investigated by WST-1 assay, as presented in Fig. 8. In metabolically active cells, tetrazolium compound is metabolized to formazan and is secreted into the medium. The resulting change in the medium color was spectrophotometrically measured to repersent the mitochondrial activity of the viable cells. At all times of inspection the mitochondrial activity of the nHDF cells was higher in the AHA fibers in comparision to the control alginate and the commercial Dimora wound dressing. The difference in the dip coated and dope mixed AHA fibers can be attributed to the fact that the HA release from the dip coated is samples was faster in comparision to the dope mixed samples. Furthermore, as the amount of HA in the fibers increases the cell viability also increased. So, overall we can infer from the results of the WST-1 assay that the addition of HA in the alginate hydrogel fibers improves their biocompatibility.

To determine the morphological differences in the way cells attach to the alginate control and AHA fibrous mats, the cell-cell and cell-material interactions were investigated by SEM. The represented micrographs are shown in Fig. 9. Interestingly the cells were found in a cluster form on the pure alginate fibers and the dip coated AHA fibers (especially with low amount of HA). From the results it is indicated that the cell-cell interaction is stronger on the alginate and low HA containg samples rather than cell material interaction, which results in weakened attachment of nHDF to the alginate fiber surface and results in agglomeration of the cells.

On AHA fibers produced by dope mixing, more cells were found spreaded acroos the surface of the fibers which indicates that the addition of HA has improved the cell material interaction and allow for the migration of the cells acroos the fibrous mats.

2. Wet-spun alginate fibers have long been commercialized, and I understand this is the reason that hyaluronic acid has been added. However, the comparison between commercial alginate fiber and the fiber developed in this work should be included.

Comparison between AHA fibers and commercial Alginate wound dressing (Dimora) marketed by Winner Medical Co., Ltd., included.

Comparing the liquid absorption of the AHA fibers with the commercially available alginate wound dressing (Dimora), it was found that the PBS $(12.01\pm0.62 \text{ g/g})$ and Sol. A $(6.97\pm0.46 \text{ g/g})$ absorption of the Dimora dressing was similar to the pure alginate (S7) samples. The slight

difference accounted between S7 and Dimora could be due to the difference in the composition of the alginate used for making these fibers, As S7 and AHA fibers are produce using high guluronic acid content alginate and Dimora is composed of high mannuronic acid content. The composite AHA fibers showed higher liquid absorption both in PBS and Sol.A in comparison to Dimora dressing.

Dimora wound dressing having a similar GSM to the produced nonwoven dressings showed a similar MVTR (837 ± 47 g/m2/day) to S7 pure alginate fibers. The MVTR of all the composite AHA fibers was lower than the Dimora dressing commercially available.

Highlights

- HA-alginate fibers by dip coating and dope mixing were successfully prepared.
- The inclusion of HA improved the cell viability and proliferation of AHA fibers.
- AHA fiber mats show MVTR of less than 35 g/m²/h.
- AHA composite fiber showed absorption of 23 g/g in PBS.

Abstract:

In this study, bi-component alginate-hyaluronic acid (AHA) fibers were developed by using two different routes. In the first method, sodium alginate dope solution was extruded into a coagulation bath containing CaCl₂ and subsequently dip-coated with hyaluronic acid (HA) whereas, in the second method, hyaluronic acid-containing sodium alginate dope solution was directly extruded into CaCl₂ bath. The resulting AHA fibers were then dehydrated in 25-100% v/v acetone solutions and dried in air. The fibers were characterized by surface morphology, physicochemical analysis, mechanical performance, swelling percentage, and total liquid absorption (g/g), cell viability, and release behavior. The results showed that AHA fibers produced by the second method have better mechanical performance, high liquid absorption, and swelling percentage with a more controlled release of hyaluronic acid. The AHA fibers showed high biocompatibility toward NIH-3T3 cell line in *in-vitro* testing, and the MVTR values (650 – 800 g/m²/day) are in a suitable range for maintaining a moist wound surface proving to be appropriate for promoting wound healing.

Keywords: hyaluronic acid; sodium alginate; wet spinning; moist wound care; bio-compatibility

Wet-Spun Bi-component Alginate Based Hydrogel Fibers: Development and *In-vitro* Evaluation as a Potential Moist Wound Care Dressing.

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1. Introduction

In the modern world, the concept of moist wound healing demands enhancement to increase the level of life, ease of application, and fasten the healing process [1]. The moist wound healing process was introduced by George Winter in 1962, according to his theory, the moist wound heals faster than dry wounds. Many other studies have also been documented, which describes that the wet dressings are beneficial in quality of healing and reducing the time of healing [2]. Maintaining a moist environment of the wound facilitates the healing procedure. The beneficial effects of the humid wound climate are to prevent tissues or cells from death by dehydration and accelerated angiogenesis [3]. The fundamental idea in moist wound healing is that the suitable quantity exudate will provide an environment that stimulates healing by delivering a range of cells and cytokines necessary for wound repair [4]. The dermal tissues exposed in the air become dehydrated, and epidermis migrates under the dehydrated tissue where it can get sufficient moisture to live [5,6]. If the surface of the wound is kept moist by using moist wound dressings, the epidermis will quickly migrate towards the surface of the dermis for faster wound healing [7,8].

Polysaccharide based fibers and polymers have amazing applications, including nutrition, therapeutics, and wound dressings [9,10]. Polysaccharides are long-chain carbohydrates consisting of mostly repeated units of monosaccharides, which may or may not be of the same monomer [11]. These polysaccharide-based fibers possess biologically active compounds for the design of biocompatible, biodegradable, and environmentally friendly materials owing to their remarkable molecular structure [12]. In the last twenty years, a lot of research has been done to explore the biological properties of polysaccharides for their utilization in the biomedical field. Polysaccharides exist in the form of plants and animals like psyllium and alginate, which can be extracted [13].

Alginate belongs to a family of polysaccharides obtained from brown algae [14]. Alginic acid was found, produced and patented by Stanford in 1881. It has been used in a variety of sectors, such as food, textile printing, medication, paper industry, and many other novel ends uses. Alginate is a remarkable gel developing content blessed of having a lot of water. In recent times, alginate continues to be used in the wound management industry as a unique material for the development of 'moist healing' items, for example, skin gels, ointments, foams, and also fibrous nonwoven dressings which can be utilized to protect chronic wounds [15]. Alginate exhibits a chain of (1,4) linked β -D-mannuronic acid (M) and α -L-guluronic acid (G) monomers [16,17]. These sugar acids are distributed in blocks and have extensively varying

proportions and sequences. Alginate consists of homopolymeric regions of M and G, known as M and G blocks, which combine as an alternating structure. Typically, the blocks are composed of three different forms of polymer segments: consecutive G residues, consecutive M residues, and alternating MG residues resultant hydrogels [18,19].

Hyaluronic acid (HA) is a linear polysaccharide sourced from nature. The macromolecule consists of N-acetyl glucosamine as well as glucuronic acid solution. It is composed of alternating disaccharide units of a-1,4-D-glucuronic acid and b-1,3-N-acetyl-D-glucosamine [20]. Its molecular weight varies from 103 to 107. HA is a glycosaminoglycan discovered in conjunctive cells of almost any vertebrate. This polyanionic polymer is also found in connective tissues, including umbilical cord, synovial fluid, vitreous, etc. [21]. HA has been largely applied in the fields of controlled medication delivery, cellular encapsulation, and muscle regeneration due to its exclusive viscoelastic properties and excellent biocompatibility [22]. Being an extracellular matrix component, HA may impact several cellular characteristics such as attachment, migration, as well as growth. The latest biomedical applications of HA include things like scaffolds pertaining to wound therapeutics as well as cell technological innovation, and surgery therapy, osteoarthritis cure, in addition to being a part of implant materials [23,24].

Hussain et al. developed alginate/chitosan/hyaluronic acid (ACH) composite fibers by wet spinning and subsequent coating procedures. Sodium alginate was extruded into a calcium chloride bath to produce calcium alginate fibers and then passed through hydrolyzed chitosan bath and dip-coated in HA to develop new fibers. The inclusion of HA increases the swelling and absorption properties of fibers. The addition of hydrolyzed chitosan and HA also improves the tensile properties of fibers. Newly developed fibers have shown a controlled release of HA and found useful applications in wound care [19]. Yamane et al. developed chitosan polymer fibers and chitosan-based HA hybrid polymer fibers by wet spinning technique. The novel fiber of chitosan-based HA comprised of chitosan-coated HA (0.04% and 0.07%). These can be useful as a scaffold biomaterial for cartilage tissue engineering. It has been found that composite fibers have more cell adhesivity, proliferation, and the synthesis of cartilage-specific proteoglycan core protein (aggrecan) than in the chitosan fiber. The chitosan-based hyaluronic acid hybrid polymer fibers demonstrate excellent prospects as a suitable biomaterial for cartilaginous tissue scaffolds [25]. Maeda et al. manufactured chitosan-hyaluronic acid hybrid polysaccharide fibers by electrospinning of polycationic chitosan with PEO and then coated them by polyanionic hyaluronic acid. Chitosan-hyaluronic acid fibers remained stable in water, although they show a controlled release of HA in phosphate-buffered saline solution (PBS). The swelling ratio of newly developed nanofibers was increased by 30% on coating of 3-wt% hyaluronic acid. Newly developed chitosan-hyaluronic acid fibers have significant contribution in faster wound healing and controlled drug delivery along with other biomedical applications [26]. *Yamane et al.* developed three-dimensional scaffolds designed from the composite chitosan-based HA polymer fibers. The wet spinning technique was used to manufacture chitosan-based HA composite polymer fibers, followed by the coating of HA (0.07%). The tensile strength, diameter, and tension at fracture of the composite polymer fiber were 144.4 N/mm², 0.03 mm, and 9.7%, respectively [27].

The present research work deals with the development and characterization of alginatehyaluronic acid (AHA) fibers for moist wound management. It is quite challenging to achieve properties like high tenacity, swelling, absorption, and controlled release by the coating of HA on alginate fiber. It is also challenging to fabricate pure HA fibers due to its high viscosity; therefore, alginate and HA are combined to prepare a dope solution and then converted into fibers. The fibers developed by both coatings of HA on alginate fiber and mixing of HA in alginate dope to fabricate a bi-component fiber were characterized for surface morphology, physicochemical analysis, *in vitro* biocompatibility, release behavior, swelling and fluid absorption properties.

2. Experimental

2.1 Materials

Hyaluronic acid was purchased from Sigma Aldrich, Czech Republic. Fiber grade Sodium Alginate (Portnal L/F 10/60), with high guluronic acid content (65-75%), was purchased from FMC Biopolymer, Norway. Calcium chloride (CaCl2) and Acetone (99 %) of analytical grade were purchased from RDH laboratories GmbH & Co., and Merck, respectively. Laboratory grade sodium chloride (NaCl), acetic acid 100%, and hydrochloric acids (HCl) 37% were purchased from UNICHEM chemical reagents.

2.2 Fiber Production

Spinning solutions of sodium alginate and HA were prepared in double-distilled water. The dope solution was mechanically stirred overnight to ensure solution homogeneity and was allowed to stand still afterward to degas them. Fig.1 shows the two fiber production routes employed: first, the alginate dope was extruded into $CaCl_2$ (1.5% w/v) at a constant draw ration

of 5. The fibers were then washed in distilled water and then dehydrated stepwise in acetone solution at different ratios 25%, 50%, 75%, and 100%, respectively. The fibers were then airdried before dipping in HA solution (0.25%, 0.5%, and 1% w/v) for 24 h. The dip-coated fibers were again dehydrated using acetone solution stepwise. In the second route, the dope solution of alginate and HA was extruded directly into the CaCl₂ solution using the same parameter as in the first route. Table 1 shows the experimental design for AHA composite fibers production.

The AHA composite fibers and control alginate fibers were converted into nonwoven sheets using a lab-scale carding machine (Shirley + miniature spinning plant, UK), to perform specific characterization like cell study and moisture vapor transmission rate analysis. The final weight of the produced nonwoven sheets was ~200 g/m². The nonwoven sheets were only consolidate using cold calendar rollers to cause minimum change to the morphology of the as-spun fibers.

2.3 Characterization of fibers

2.3.1 Scanning electron microscopy

The scanning electron microscopy (SEM, JSM-6010LA, JEOL, Japan) was used to study the surface morphology of the produced fibers. The fibers were sputtered coated with platinum before SEM photographs were taken.

2.3.2 Physicochemical analysis

The produced AHA fibers were subjected to Fourier transform infrared (FTIR) spectroscopy for physicochemical analysis using Prestige-21, Shimadzu Co, Ltd., Japan, with a range of 4000 - 400 cm⁻¹ wavenumber.

2.3.3 Fiber linear density

After conditioning all the fiber samples at standard atmospheric conditions, the linear density of the fibers was determined according to ASTM D 1059-12. Forty specimens of 2-inch length from each fiber sample were prepared and weighed, using an analytical balance. The length was converted into meters. The linear density in "tex" was calculated by using Eq 1. Finally, the mean value of the linear density of each fiber sample was determined along with the standard deviation.

$$tex = \frac{weight in grams (g)}{1000 m}$$
(1)

2.3.4 Tensile properties

The tensile properties of the fibers were measured by using a single fiber strength testing system, M250-2.5CT (Testometric, Rochdale, England). The test method employed was BS EN ISO 5079. The machine works on the principle of a constant rate of extension (CRE). A single fiber was clamped between two points, one moveable and one fixed, 10 mm apart, and force applied was 10.0 N and set at 12 mm/min constant rate of extension. The average value of ten specimens from each fiber sample was taken.

2.3.5 Liquid absorption

Liquid absorption properties of all the developed fibers were tested using three different liquids, i.e. deionized water, saline (0.9% w/v NaCl) and solution A (0.8298 % w/v NaCl and 0.0368 % w/v CaCl2). All samples were initially soaked for 1 hour and hung in the air until no liquid droplet fell prior to taking wet weight measurements. Liquid absorption was calculated as the ratio of the wet weight of the fibers to the weight of the fiber after drying overnight at 105°C. Absorption was calculated by using the Eq 2:

Absorption (g/g) =
$$\frac{Ww-Wd}{Wd}$$
 (2)

Where W_d refers to the dry weight of fiber, and W_w is the wet weight of fiber[17].

2.3.6 Swelling behavior

Swelling behavior or change in fiber diameter of the produced fiber was observed by using a MICROS optical microscope (MC-50) with a digital camera. The change in diameter of the fibers was noted after soaking (for 4 minutes) in the above-mentioned solutions at 25°C. Eq 3 can be used to calculate % swelling of fibers:

% swelling
$$= \frac{Dw - Dd \times 100}{Dd}$$
 (3)

Where Dw is the wet fiber diameter, and Dd is dry fiber diameter [17].

2.3.7 Release study of hyaluronic acid

The release profile of HA was checked by using a UV-Vis spectrophotometer. In this process, firstly standard absorption curve was developed at 600nm for HA at different concentrations. A quadratic equation was developed from the standard absorption curve. Fig. 2 shows the standard absorption curve and the quadratic equation.

The amount of turbidity developed on the addition of cetyltrimethylammonium bromide (CTAB) into acetate buffer and specimen is proportional to the amount of HA present in solution. Acetate buffer includes 0.2M sodium acetate-acetic acid and 0.15M NaCl of PH 6. CTAB reagent includes cetyltrimethylammonium bromide (2.5gm) in 2 percent of 100ml NaOH. These two solutions were incubated at 37°C in order to synchronize the reaction temperature [19,28,29].

The developed fibers were placed separately in PBS and solution A. After 2 hours, 1ml of the specimen was taken from solutions into test tubes, then 1ml of acetate buffer and 2ml of CTAB reagent was added in the same tubes. After mixing, the contents of test tubes were transferred to cuvettes, and absorption was checked at a wavelength of 600nm [28]. The absorption values taken by UV/VIS spectrophotometer were placed in the quadratic equation (based on standard absorption curve) to calculate the release of HA in mg/liter (ppm).

2.3.8 In-vitro Cell Proliferation

According to ISO 10993-5 standards, the effect of AHA fibers on the viability of normal human dermal fibroblast (nHDF) cell line Briefly, nHDF cell were cultured in Dulbeco modified eagle medium (DMEM) accompanying 10% fetal bovine serum (FBS) in a humidified incubator at 37° C and 5% CO₂ environment. The fiber mats were sterilized with 100% ethanol for 10 min and dried in vaccum chamber at ambient temperature, and afterward UV sterilized for 24h. The sample speciemen were not subjected to PBS washing as they will dissolve due to ion exchange. The sterilized AHA fibers were placed in 96 well glass culture plates. The fiber-free cells were used as controls. The well was seeded with 1×10^4 cells per well. The well was cultured for 1, 3, and 7 days. At the predetermined times, 10μ l WST -1 was added to each well and incubated for two hours. Absorbance was measured at 440 nm using a microplate reader (Thermos Scientific, Multiskan FC instruments) as an indicator for proliferation. Readings were measured in triplicate, and averages were recorded.

2.3.9 Cell adhesion

The nHDF cell attachment on the control alginate and AHA fibrous mats were observed with SEM micrographs. nHDF cells after 48 h of culture were fixed on the mats with 4% paraformaldehyde for 4h at 4°C. The mats were dehydrated using ethanol. The mats were vacuum dried before SEM imagery. The resolution for SEM micrographs was x100 for all samples.

To evaluate the MVTR of the produced composite AHA fibers, the fibers were converted into nonwoven sheets using a lab-scale carding machine (Shirley + miniature spinning plant, UK) and afterward consolidated using cold calendar rollers at 2 bar pressure. The nonwoven webs were cut into 9 cm diameter and placed over a cup with 7 cm diameter. The containers were filled with water up to three-quarter level and weighed "W₁". The cups were placed in an incubator under 40% humidity at 37°C. The cups were removed from the incubator after 24 hours and weighed again "W₂". MVTR was calculated from the following Eq. 4.

$$MVTR = G/At \tag{4}$$

Where "G" is the weight change (W_1-W_2) , "A" is the area of the cup mouth, and "t" is the time in which "G" happened.

2.3.11 Comparison with a commercial Alginate dressing

The produced AHA fibers were compared against commercially available Dimora (Alginate Wound Dressing) marketed by Winner Medical Co., Ltd. The produced fibers were compared for MVTR, liquid absorption (g/g), and cell proliferation. The commercial alginate dressing was composed of high mannuronic acid content 50 - 70 %, the GSM of the commercial dressing was ~192 g/m², which is close to our produced nonwoven dressing.

2.3.12 Statistical Analysis

One-way analysis of variance (ANOVA) to indicate the statistical significance of the results was performed using the MINITAB-17 ® software package.

3. Results and Discussion

The statistical significance of the different concentrations of the HA on the response variables was conducted by one-way ANOVA. Table 2 gives details of the analysis. P-value (probability of null hypothesis) ≤ 0.05 indicates the statistical significance of the effect [30,31]. It is evident from Table 2 that the impact of HA concentration has a statistically significant impact on most of the responses under study, i.e., linear density, tenacity, swelling, and absorption capacity.

3.1 Surface morphology

The representative S.E.M. micrographs of the composite AHA and the control fibers are given in Fig. 3. Despite the smooth rounded holes of the spinneret, the produced fibers are partially flattened like ribbon. All fibers are characterized by surface striation. These striations were formed by the formation of an egg-box structure in the guluronic constituent of the sodium alginate and the chelating structure built by the HA upon interaction with the calcium ion in the coagulation bath. These striations are dominant in fibers produced by the second route and the control fibers. But in the fibers created by the first route have diminished striation due to the HA coating after the fibers were produced and can be seen in Fig. 3 (a', b', c', d', e', f', and g'). The surface morphology of the materials is very useful in determining their end application. Fibers with uniform and smooth surfaces provide a very soft feel, whereas fibers with striation and non-uniform surfaces provide a very harsh feel.

3.2 Physicochemical analysis

Fig. 4 shows the ATR-FTIR spectra of the sodium alginate (S7) and HA powder along with the prepared AHA composite fibers. The alginate spectra showed a few significant intense peaks. O – H stretching vibrations were seen in the range of $3400 - 3200 \text{ cm}^{-1}$, aliphatic C – H stretching was observed 2938 cm⁻¹. COO. – asymmetric and symmetric (C = O) stretching were at wavenumber 1640 cm⁻¹, and 1437 cm⁻¹. At 1324 cm⁻¹, C – O stretching vibration was recorded. The strong peaks between 1100 – 950 cm⁻¹ are attributed to C – O, C – C, and C—O, C – C vibrational stretching's [32].

In HA ATR-FTIR spectra, the – OH and – NH group vibrational peak was at 3453 cm⁻¹. C— H band stretching was observed at 2925 cm⁻¹. Two distinct peaks at 1710 cm⁻¹ and 1648 cm⁻¹ corresponds to amide and carbonyl groups. The band at 1427 cm⁻¹ represents COO vibrational stretch, which refers to the acid group of HA [33]. The absorption bands at 1139 cm⁻¹ and 1043 cm⁻¹ are attributed to ether C—O—C linkages in the polymer chain [34].

The ATR-FTIR fingerprints of the two polymers understudy, SA, and HA are quite similar to each other; that is why it is challenging to find a peak different from each other in the FTIR spectrum of the produced composite fibers. The only difference observed was in the absorbance peak of the – OH group, which appears to be shrunken due to the participation of the hydroxyl and carboxylate groups of the polymers in forming a chelating structure with the calcium ions in the coagulating bath [32]. The peak observed in the FTIR fingerprints is in according to the previously reported literature.

3.3 Linear Density and Tensile Properties

The results of linear density and tenacity of all fibers produced by two different methods are shown in Fig. 5. The results show (Fig. 5 (a)) that the addition of HA increases the fibers' linear density developed by the dip-coating method. This may be due to the coating of hyaluronic acid. But the fibers produced by using HA in dope solution have significantly reduced linear density. The reason could be due to the increase in the dope solution's viscosity upon the addition of HA (Table 1). As a result of increased viscosity, the drag (draft) on the dope solution increases while exiting the spinneret's orifice. Thus the resulting fibers are finer than the fibers produced by the dip-coating method. Fiber linear density influence the properties of the products produced from them. Linear density is directly proportional to the air and moisture permeability of the fabric produced from it. The finer the fibers greater will be the surface area and coverage of the fabric. Thus a fabric made of finer fibers will have low MVTR and air permeability and will present better barrier performance than the fabric produce of high linear density. Furthermore, low linear density fibers have greater flexibility, soft and pliable, thus improving the final produced product's comfort.

The tenacity of AHA composite fibers improves on the addition of hyaluronic acid, and it increases with an increase in concentration in Fig. 5 (b). It may be due to alginate and hyaluronic acid's similar properties, which makes them compatible with each other. The chemical resemblance of HA with alginate results in strong interaction among them and thus increases the fiber's strength. It can also be due to the interaction of cationic calcium ions with anionic HA to form calcium hyaluronate [35], which increases the tenacity of the developed fiber. The fibers produced by method B have even higher tenacity than the fiber produced by method A. It can be due to the fact that after mixing alginate and hyaluronic acid, the homogenous dope solution was extruded in a coagulation bath, and fibers were drawn during the manufacturing process. Due to the combined drawing of alginate and HA dope solution, the crystallinity of the fibers increases, and fibers produced are finer and have a more aligned and linear structure, which helps to improve the tenacity of fibers.

3.4 Swelling and Absorption Properties

Fig. 6 (a) indicates that the swelling of AHA fibers increases on increasing the concentration of HA at a constant drawing ratio. This seems to be due to the increasing number of hydroxyl and carboxylic groups by adding HA in the developed fiber, which improves the hydrophilic characteristics and thus increases the fiber swelling [36].

The fibers produced by the coating method do not show a marked difference from the control alginate fiber (S7), this may be due to the removal of the HA from the surface of the composite fibers upon ion exchange with the respective liquid under study.

In fibers produced by the second route, the overall trend of swelling behavior (%) is the greatest for PBS solution with a value of 476 %. It may be due to the formation of sodium hyaluronate, which is more soluble in water. It can also be due to more ions exchange property (calcium ions exchange by sodium ions) in PBS solution and then solution A. Still, in Solution A, swelling is relatively lower than the saline solution (250 %); this may be due to the presence of calcium ions, which also form calcium hyaluronate, which reduces solubility in water [37]. The presence of calcium also creates a hindrance in ions exchange.

High exudate absorption is a crucial aspect of wound dressing made, especially for moist wound care applications. Wound dressing ensures that the wound surface remains moist, aiding the autolytic debridement and movement of cells across the wound bed during reepithelialization without damaging the surrounding skin by exudate over saturation. Liquid absorption (g/g) properties in various liquids of the developed AHA fibers is shown in Fig. 6(b). The addition of HA increases the liquid absorption of developed fibers due to its excellent gel-forming properties. Absorbance properties of the composite AHA fiber improved as the amount of HA increased in both sets of composite fibers produced. The newly developed fibers show less absorption in solution A as compared to PBS, it may be due to the presence of calcium ions in it, which creates further hindrance in ion exchange. Most of the absorption in saline and solution A is due to the coating of hydrophilic hyaluronic acid.

Comparing the liquid absorption of the AHA fibers with the commercially available alginate wound dressing (Dimora), it was found that the PBS $(12.01\pm0.62 \text{ g/g})$ and Sol. A $(6.97\pm0.46 \text{ g/g})$ absorption of the Dimora dressing was similar to the pure alginate (S7) samples. The slight difference accounted between S7 and Dimora could be due to the difference in the composition of the alginate used for making these fibers, As S7 and AHA fibers are produce using high guluronic acid content alginate and Dimora is composed of high mannuronic acid content. The composite AHA fibers showed higher liquid absorption both in PBS and Sol.A in comparison to Dimora dressing.

Accordingly, the statistical significance of the HA (p-value 0.000 and 0.003 for swelling, and 0.000 and 0.000 for absorption) indicates that the addition of HA to the alginate fibers had

significantly improved its liquid retention properties as compared to control alginate fibers (S7).

3.5 Release study

The release of HA from the composite AHA fiber after 2 h of dipping time was studied. According to the results in Fig. 7, it is evident that the release of HA increases on increasing the concentration of hyaluronic acid. It can be due to the higher quantity of HA attached to the fiber.

Overall the trend of the release of HA is maximum in saline solution. It can be due to sodium hyaluronate formation, which is more soluble in water [37]. It can also be due to more ions exchange property (calcium ions exchange by sodium ions) in saline solution. But in Solution A, the release of HA is the lowest. This may be due to the presence of calcium ions, which form calcium hyaluronate, which reduces solubility in water [37]. The presence of calcium in solution A also creates a hindrance in ions exchange.

The release in two sets of composite fibers was quite distinct. In dip-coated fibers, the release was abrupt, which can be attributed to the fact that the HA was coated on the surface, and the ion exchange between the HA and the liquid was effortless without any hindrance thus the release was abrupt. While in dope mixed AHA fibers, the release was slow as compared to the dip-coated fiber set. The reason for this could be the complex chelating structure formed by the alginate and HA upon interaction with the calcium ions in the coagulation bath.

3.6 Cell viability and adhesion

Cell viability plays an important role in living tissues and thus is an important factor for the tissue engineering and regenerative medicine. To study the cell activity the mitochondrial activity of the nHDF cell on AHA fibers, control alginate and Dimora fibrous mats after 1, 3, and 7 days of culture was investigated by WST-1 assay, as presented in Fig. 8. In metabolically active cells, tetrazolium compound is metabolized to formazan and is secreted into the medium. The resulting change in the medium color was spectrophotometrically measured to reperesent the mitochondrial activity of the viable cells. At all times of inspection the mitochondrial activity of the nHDF cells was higher in the AHA fibers in comparision to the control alginate and the commercial Dimora wound dressing. The difference in the dip coated and dope mixed AHA fibers can be attributed to the fact that the HA release from the dip coated is samples was faster in comparision to the dope mixed samples. Furthermore, as the amount of HA in the

fibers increases the cell viability also increased. So, overall we can infer from the results of the WST-1 assay that the addition of HA in the alginate hydrogel fibers improves their biocompatibility.

To determine the morphological differences in the way cells attach to the alginate control and AHA fibrous mats, the cell-cell and cell-material interactions were investigated by SEM. The represented micrographs are shown in Fig. 9. Interestingly the cells were found in a cluster form on the pure alginate fibers and the dip coated AHA fibers (especially with low amount of HA). From the results it is indicated that the cell-cell interaction is stronger on the alginate and low HA containg samples rather than cell material interaction, which results in weakened attachment of nHDF to the alginate fiber surface and results in agglomeration of the cells. A similar result has been previously reported [38].

On AHA fibers produced by dope mixing, more cells were found spreaded acroos the surface of the fibers which indicates that the addition of HA has improved the cell material interaction and allow for the migration of the cells acroos the fibrous mats.

3.7 MVTR Analysis

In 1962, G. D. Winter found out that scab formed on the surface of the wound contains a superficial part of the dermis, which due to the dehydration of the dermis. He suggested that if the surface of the injury is deliberately kept moist, then the scab formation can be prevented. This would, in turn, help in an increased rate of re-epithelialization [3]. Wound dressings with low MVTR are considered to be able to maintain a moist wound environment. A higher MVTR value will dehydrate the wound and may result in scar formation. Healthy human skin has a trans-epidermal water loss of $4 - 9 \text{ g/m}^2/\text{h}$. In a full-thickness wound, the water loss increases up to 80 - 90 g/m²/h. Dressings that have an MVTR of less than 35 g/m²/h are considered to be moisture-retentive [39,40]. Dressing that retains the tissue fluid help retains the cellular and extracellular components of the wound healing that naturally support the healing process. The MVTR depends not only on the density of the produced nonwoven but also on the absorption properties of the materials. A lower density leads to a higher MVTR, and similarly, a higher absorption leads to a lower MVTR. The density of nonwoven mats was kept constant at 200 g/m^2 . The MVTR analysis of the control (S7) and composite AHA fibers are given in Fig. 10. The control alginate fiber (S7) shows an MVTR of 885 $g/m^2/day$. The inclusion of HA to the alginate fibers helps increase the moisture absorption of the composite fiber and hence resulted in a decrease in the MVTR of the composite AHA fibers. In both sets of fibers as the HA content increases, the MVTR decreases further with the fiber S1, S2, and S3 show an MVTR of 796, 736, and 700 g/m²/day. Similarly, the fibers S4, S5, and S6 show an MVTR of 722, 693, and 667 g/m²/day. The MVTR values of the two sets of composite fibers (S1, S2, S3) and (S4, S5, S6) along with the control alginate fibers (S7) are in the acceptable range of value as reported in the literature. Dimora wound dressing having a similar GSM to the produced nonwoven dressings showed a similar MVTR (837 ± 47 g/m²/day) to S7 pure alginate fibers. The MVTR of all the composite AHA fibers was lower than the Dimora dressing commercially availbale.

4. Conclusion

The current work utilized two routes to produce composite AHA fibers. AHA composite fibers were successfully developed by wet spinning and subsequent coating techniques. The fiber surface has distinct surface striations due to the egg box and chelating structure formed by alginate and HA constituents upon interaction with calcium ions in the coagulation bath. The FTIR spectra of alginate and HA were very similar so no significant difference can be observed in the spectra of the composite AHA fibers except the - OH group, which appears to be shrunken due to participation of the hydroxyl and carboxylate groups of the polymers in forming a chelating structure with the calcium ions in the coagulating bath. The addition of HA resulted in increased tensile strength due to strong interaction between anionic HA and cationic calcium ions. Composite fiber swelling and absorption (g/g) properties are markedly improved upon the addition of HA with maximum swelling of 476% and absorption of 23 g/g in PBS. The release of HA from the composite fiber was very different in two sets of fibers. In dipcoated fibers, the release was abrupt while in dope mixed fibers, the release was slow, this difference is due to the strong interaction of HA with calcium ions in the dope mixed AHA fibers. MVTR values $(650 - 800 \text{ g/m}^2/\text{day})$ of the composite AHA fibers are in the acceptable range for moist wound care. Cell viability and adhesion study indicates that the addition of HA in the alginate fibers has improved it overall biocompatibility and cell material interaction. Based on the results the fiber produced by dope mixing method have shown an overall better performance.

The developed fibers have low mechanical strengths and can only be produced into nonwoven structures. These fibers can be used only for treating exuding wounds as moisture is one of the significant components for their practical use.

Funding

This research did not receive any specific grant from funding agencies in the public, commercial, or not-for-profit sectors.

Declaration of Interest

None

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Fig. 10. Moisture vapor transport rate (MVTR) of AHA composite and Alginate control fiber (S7).

Sample Symbol	Sodium Alginate (%)	Hyaluronic Acid (%)	Drawing Ratio	Dope Solution Viscosity (cP)	Coagulation bath (CaCl2) (%)				
Fiber Produced by First Route									
S1	5.0	0.25	5.0	5500	1.5				
S2	5.0	0.5	5.0	5500	1.5				
S3	5.0	1.0	5.0	5500	1.5				
Fiber Produced by Second Route									
S4	5.0	0.25	5.0	6250	1.5				
S5	5.0	0.5	5.0	6900	1.5				
S6	5.0	1	5.0	7850	1.5				
S7*	5.0	0	5.0	5500	1.5				

Table 1. Experimental design for alginate-hyaluronic acid fibers production.

*control fiber

Factors	Linear density (tex)		Tenacity (cN/tex)		Swelling (%)				Absorbance (%)			
					Sol. A		PBS		Sol. A		PBS	
	P-value	R ² (%)	P-value	R ² (%)	P-value	R ² (%)	P-value	R ² (%)	P-value	R ² (%)	P-value	R ² (%)
Hyaluronic	0.000*	94.53	0.000*	98.42	0.000*	90.66	0.007*	96.04	0.000*	97.21	0.000*	95.21
Acid (dip												
coating)												
Hyaluronic	0000*	98.34	0.000*	96.78	0.003*	89.34	0.000*	87.27	0.000*	91.37	0.000*	89.93
Acid (dope												
mixture)												

Table 2. ANOVA for responses with respect to hyaluronic acid content.

*P-value < 0.05 indicating statistical significance of the factor to response under study, R^2 coefficient of determination.

Author's Contribution Statement:

Azeem Ullah: Conceptualization, Methodology, Writing - Original Draft, Investigation, Visualization, Data Curation, Writing – review & editing.

Muhammad Umar: Conceptualization, Methodology, Investigation, Data Curation.

Hifza Nawaz: Formal analysis.

Tanzeel Areeb: Visualization, Formal analysis.

Motahira Hashmi: Formal analysis

Davood Kharaghani: Resource and Formal analysis.

Kyu Oh Kim: Resource

Ick Soo Kim: Validation, Supervisor.