

Communication

Two-Color Infrared Sensor on the PbTe: In *p-n* Junction

Jonas Gradauskas ^{1,2,*}, Bohdan Dzundza ³, Leonid Chernyak ⁴ and Zinoviy Dashevsky ⁵¹ Department of Electronic Processes, Center for Physical Sciences and Technology, 10257 Vilnius, Lithuania² Department of Physics, Vilnius Gediminas Technical University, 10223 Vilnius, Lithuania³ Department of Computer Engineering and Electronics, Vasyl Stefanyk Precarpathian National University, 76018 Ivano-Frankivsk, Ukraine; bohdan.dzundza@pnu.edu.ua⁴ Department of Physics, University of Central Florida, Orlando, FL 32816-2385, USA; Leonid.Chernyak@ucf.edu⁵ Department of Materials Engineering, Ben-Gurion University of the Negev, Beer-Sheva 84105, Israel; dashevsky.45@mail.ru

* Correspondence: jonas.gradauskas@ftmc.lt; Tel.: +370 655 06220

Abstract: A lead telluride sensor was fabricated on the base of a *p-n* PbTe junction created on a PbTe single crystal grown by the Czochralski technique, followed by the diffusion of an indium donor impurity into a crystal. The capacitance-voltage and current-voltage characteristics of the sensor were measured over the temperature range from 80 K to 150 K. A prototype of a high-temperature mid-IR sensor, a PbTe diode, with a cut-off wavelength of 4 μm , operating at temperatures up to 150 K, was demonstrated for the first time. The advantage of the sensor is that its operating temperature is high enough to be reached by a solid-state thermoelectric cooler. The sensor showed a specific detectivity value of $10^{10} \text{ cm}\cdot\text{Hz}^{1/2}/\text{W}$ at a temperature of 150 K and a wavelength of 4.2 μm . The possibility to sense pulses of long-IR radiation by means of the PbTe diode was also demonstrated over the 100–180 K temperature range. For the first time, a two-photon absorption-caused photo-voltaic effect was observed in PbTe at a wavelength of 9.5 μm at 150 K.

Keywords: infrared sensor; high-temperature PbTe photodiode; specific detectivity; two-photon absorption

Citation: Gradauskas, J.; Dzundza, B.; Chernyak, L.; Dashevsky, Z. Two-Color Infrared Sensor on the PbTe: In *p-n* Junction. *Sensors* **2021**, *21*, 1195. <https://doi.org/10.3390/s21041195>

Academic Editor: Stefan Hippler

Received: 28 December 2020

Accepted: 4 February 2021

Published: 8 February 2021

Publisher's Note: MDPI stays neutral with regard to jurisdictional claims in published maps and institutional affiliations.



Copyright: © 2021 by the author. Licensee MDPI, Basel, Switzerland.

This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC BY) license (<http://creativecommons.org/licenses/by/4.0/>).

1. Introduction

Photodiodes based on InSb compounds are traditional semiconductor sensors for detection in the midinfrared region because of their forbidden energy gap E_g close to 200 meV at operating temperature $T = 77 \text{ K}$ (cooled by liquid nitrogen) and demonstrated a specific detectivity D^* of about $10^{11} \text{ cm}\cdot\text{Hz}^{1/2}/\text{W}$ [1–3].

On the other hand, the PbTe semiconductor compound, having a practically comparable value of $E_g = 210 \text{ meV}$ ($T = 77 \text{ K}$), is also used for the fabrication of IR sensors [4,5]. The advantage of PbTe in comparison with InSb is the increase of its energy gap with growing operating temperatures. Thus, it is possible to use PbTe photodiodes up to a cooling temperature of 180 K [6]. Therefore, practically for the first time, it has become possible to cool these sensors with solid-state coolers based on multistage thermoelectric modules.

Photodiode infrared sensors can be either of the Schottky barrier [7] or of the *p-n* junction type. PbTe photodiodes are usually made from layers grown on BaF₂ substrates [8] or on Si substrates using fluoride buffer layers [9–11].

In this work, we fabricated an indium-doped PbTe *p-n* junction on a single crystal grown by the Czochralski technique [12] and investigated its electrical and detection characteristics. The estimated parameters of the IR sensor demonstrate its high photodetection

performance: $D^* = 1.5 \times 10^{10} \text{ cm}\cdot\text{Hz}^{1/2}/\text{W}$ at temperature $T = 150 \text{ K}$ and radiation wavelength $\lambda = 4.2 \text{ }\mu\text{m}$. For the first time, a two-photon absorption-caused photovoltaic effect was observed in PbTe at a wavelength of $9.5 \text{ }\mu\text{m}$ at $T = 150 \text{ K}$.

As mentioned, in contrast to classical semiconductors such as Si or GaAs, the bandgap E_g of lead telluride increases with the temperature varying from 190 meV at 0 K to 306 meV at 300 K [4] as

$$E_g(T) = 0.19 + 4.5 \times 10^{-4} T^2 / (T+50) \quad (1)$$

Correspondingly, the cut-off wavelength of PbTe calculated as $\lambda = 1.24/E_g(\text{eV}) \text{ }\mu\text{m}$ as a function of temperature is shown in Figure 1.

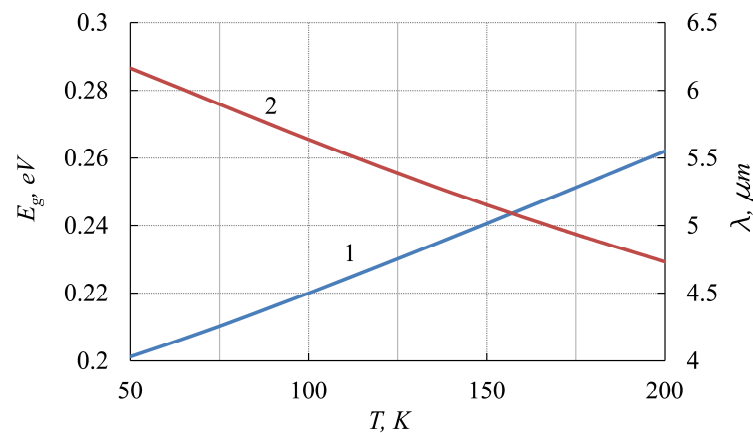


Figure 1. Bandgap of PbTe (1) and the cut-off wavelength (2) as functions of temperature.

2. Detector Fabrication

The single crystals of the PbTe semiconductor were grown by means of the Czochralski method, which allows unidirectional solidification from the melt [12]. The synthesized PbTe was loaded into a quartz crucible, which was heated in a furnace to $1200 \text{ }^\circ\text{C}$. A PbTe seed crystal of $\langle 100 \rangle$ orientation was used to pull the crystal out of the melt. A molten boron trioxide B_2O_3 layer about 1 cm thick was used for liquid encapsulation. The molten boron trioxide was inert to PbTe at the solidification temperature. The temperature gradient at the crystallization was 25 K cm^{-1} , and the crystal pull rate was 10 mm h^{-1} . The crystal and the crucible were rotated in opposite directions at an angular velocity of 1 s^{-1} .

The crystal orientation was determined by the Laue method. The $\text{Cu}_{\text{K}\alpha 1}$ radiation ($\alpha = 1.5406 \text{ \AA}$, $\Delta 2\theta = 0.005^\circ$, 2θ range $10\text{--}120^\circ$) and Bragg-Brentano geometry were employed in an X-ray diffractometer. The Hall effect measurements were carried out in a permanent magnetic field $B = 2 \text{ T}$ with an accuracy of 8%.

To form the p - n junction, one face of the p -type PbTe single crystal was polished, and then an oxide layer was grown electrochemically [13]. The required pattern was formed by ordinary photolithography. Then, the diffusion of the In dopant was carried out from the In_4Te_3 gaseous phase into the PbTe crystal [14]. In this process, the PbTe crystal was placed in a quartz ampoule, evacuated to a residual pressure of 10^{-5} mbar , and then heated in a furnace up to $500 \text{ }^\circ\text{C}$. The diffusion time was 10 h. The surface indium concentration was determined by the energy-dispersive X-ray spectroscopy (EDS) method. As a result, the p - n diodes with respective hole and electron densities $p \approx 10^{18} \text{ cm}^{-3}$ and $n = 2 \times 10^{18} \text{ cm}^{-3}$ ($T = 80 \text{ K}$) were produced.

3. Device Properties

3.1. Electrical Characteristics

The capacitance versus voltage, C - V , characteristic of the In-doped PbTe p - n junctions was measured over a temperature range of 80–150 K with an Agilent HP4280A (Agilent Technologies, Santa Clara, US) capacitance meter [6]. The diodes were placed in a He gas closed-cycle cryostat in a vacuum of about 10^{-7} Torr. The capacitance was measured through an alternate signal of 10 or 30 mV at a frequency of 1 MHz applied along with the DC voltage. The fine linearity of C^{-3} curves as a function of voltage V agrees well with the standard theory of a linearly graded p - n junction [15] (Figure 2).

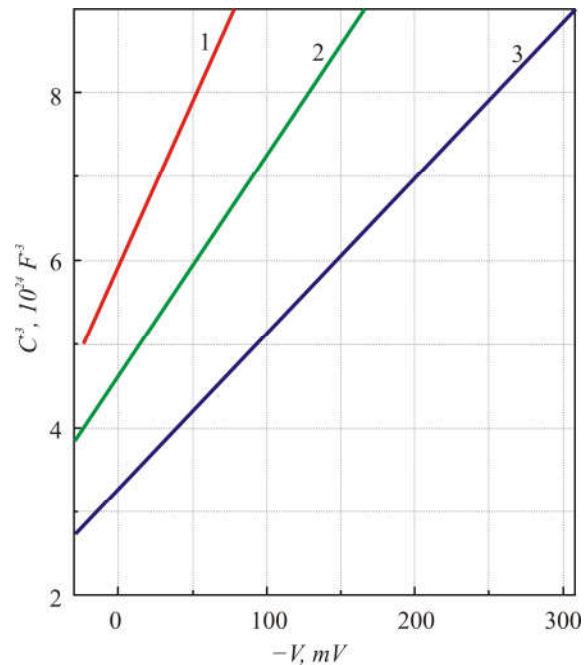


Figure 2. Capacitance-voltage characteristics of the PbTe p - n junction at different temperatures: 1—150 K; 2—110 K; 3—80 K.

The experimental I - V curves of the PbTe p - n structure are presented in Figure 3. These curves have been fitted by the Shockley equation [15]:

$$I = I_0 [\exp(eV/nkT) - 1] \quad (2)$$

where I_0 is the saturation current, and n is the ideality factor. This way, the ideality factor was evaluated to equal 1.7, 1.6, 1.7 and 1.8 at, respectively, 80 K, 100 K, 120 K and 150 K temperatures [6]. It is worth noting that, in spite of the inverse dependence of E_g on temperature, both the reverse current and the forward turn-on voltage of the PbTe p - n diodes depend on temperature in the same manner typical of diodes of the classical semiconductors, such as Si, Ge, or GaAs.

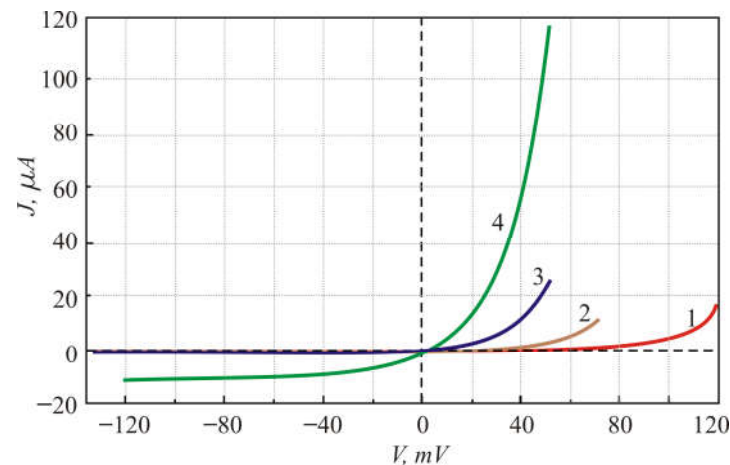


Figure 3. Current–voltage characteristics of the In-doped PbTe photodiode at different temperatures: 1—80 K; 2—100 K; 3—120 K; 4—150 K.

3.2. Sensing Characteristics

Detectivity of the In-doped PbTe p - n junction was measured at 150 K in the infrared range following the earlier described techniques [8]. A schematic view of the detector is presented in Figure 4.

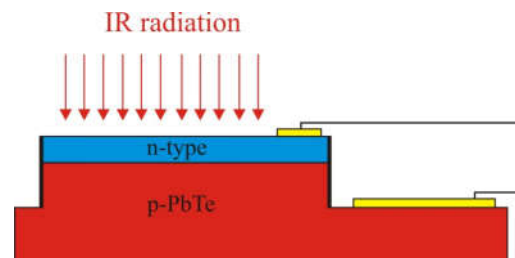


Figure 4. Schematic view of the In-doped PbTe p - n photodiode.

The radiation from a blackbody source ($T = 1000$ K) chopped at a 1000 Hz frequency was directed to the detector through a BaF₂ window in a He gas closed-cycle cryostat. The radiation intensity was $I_R = 10^{-3}$ W/cm², and the root-mean-square voltage signal V_{rms} was measured by means of a lock-in amplifier (Tektonix Differential Preamplifier ADA400A, Shanghai, China) with a bandwidth $\Delta f = 10$ Hz. The detector background noise V_n was measured at 300 K and a field of view equal to 25°. The integral detectivity D^* was calculated as [8]

$$D^* = \left(\frac{1}{I_R} \right) \sqrt{\frac{\Delta f}{A} \frac{V_{rms}}{V_n}} \quad (3)$$

where $A = 1 \times 1$ mm² is the active area of the detector.

The value of $D^* \approx 10^{10}$ cm Hz^{1/2}/W was obtained for the PbTe photodiode at 150 K. The spectral response was measured in the same setup by placing a grating monochromator between the infrared source and the PbTe detector. Figure 5 presents the spectral response characteristic of the In-doped PbTe p - n junction at $T = 150$ K. The cut-off wavelength ≈ 4.2 μ m corresponds to the PbTe energy gap E_g at $T = 150$ K.

The R_0A values taken from [6] allowed estimating the specific detectivity D_{λ}^* of the detector. It was defined as [6]

$$D_{\lambda}^* = \frac{R_{\lambda}}{\left[\frac{4kT}{R_o A_e} + 2e^2 \eta Q_B \right]^{\frac{1}{2}}} \quad (4)$$

$$R_{\lambda} = \eta e / hc \quad (5)$$

$$Q_B(\nu_c, T) = J(\nu) d\nu \quad (6)$$

$$J(\nu) = \frac{8\pi\nu^2}{2c^2 \left[\exp\left(\frac{h\nu}{kT}\right) - 1 \right]} \quad (7)$$

Here, R_{λ} is the current mode responsivity for the wavelength λ , η is the quantum efficiency, and ν_c is the cut-off frequency. For blackbody radiation at 300 K, typical values are $\eta = 0.5$ and $R_{\lambda} \approx 1.6$ A/W at a cut-off wavelength of 4 μm [6]. The estimated value of the specific detectivity D_{λ}^* was 2.2×10^{10} cm Hz^{1/2}/W at a temperature of 150 K.

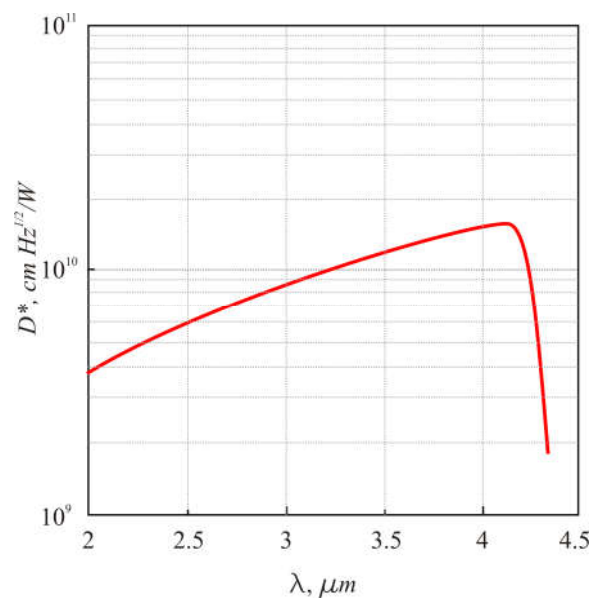


Figure 5. Spectral response of the In-doped PbTe *p-n* junction at $T = 150$ K.

As a source of long-wavelength IR radiation, a Q-switched CO₂ laser operating at a 9.5 μm wavelength with a pulse duration of 150 ns, a repetition rate of 40 Hz, and a maximum intensity of about 0.1 MW/cm² was used. The measurements were performed in the photocurrent regime, i.e., the photovoltage across a 50 Ohm load resistance was recorded by a LeCroy Wavepro 7200 (LeCroy Corporation, Chestnut Ridge, US) oscilloscope with a 2 GHz passband. The measurements were carried out in the same closed-cycle optical cryostat.

The fabricated PbTe *p-n* junction sensed the CO₂ laser radiation. The photocurrent demonstrated a polarity similar to that of the classical one originating from the electron-hole pair generation. Figure 6 presents the dependence of the photocurrent pulses induced by the CO₂ laser across the *p-n* PbTe structure as a function of time in the 100–180 K temperature range.

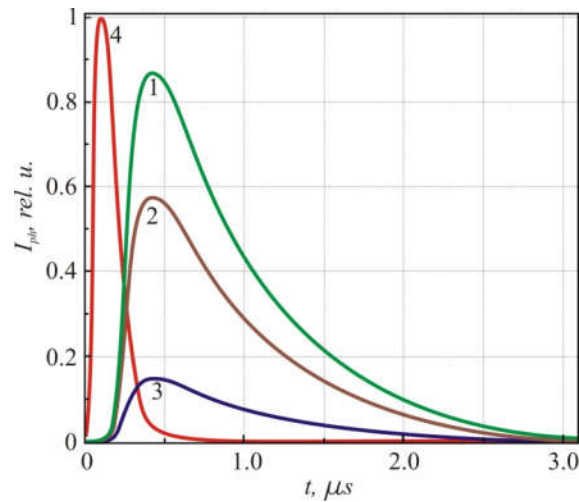


Figure 6. Photocurrent pulses across the PbTe *p-n* junction at different temperatures: 1—100 K; 2—150 K; 3—180 K. For comparison in the timescale, Trace 4 (red) represents the laser pulse.

It can be seen that the photocurrent is stronger at lower temperatures. The responsivity of the sensor to 9.5 μm radiation was 10^{-6} A/W at a 150 K temperature. The typical photocurrent pulse relaxation time was about 6×10^{-7} s. It did not depend on the temperature within the given range; thus, its origin cannot be attributed to the thermal effects, which, like thermal conductivity, should depend on the sample temperature. The dependence of the peak value of the CO₂ laser-induced photocurrent pulse on the diode temperature is shown in Figure 7. Two linear slopes can be distinguished in the photocurrent dependence on reciprocal temperature (Figure 7). These indicate the exponential dependence of the photocurrent I_{ph} :

$$I_{ph} \propto \exp\left(\frac{E_a}{kT}\right) \quad (8)$$

where E_a is the specific activation energy. One of the slopes discloses an activation energy of 100 meV, which corresponds to half of the PbTe forbidden energy gap, i.e.,

$$I_{ph} \propto \exp\left(\frac{E_g}{2kT}\right) \quad (9)$$

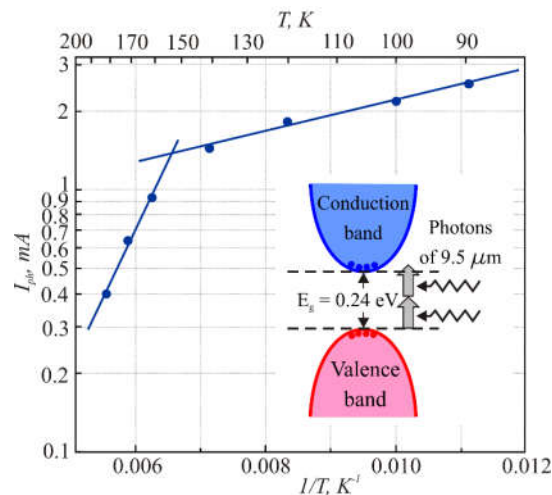


Figure 7. Dependence of the photocurrent across the PbTe *p-n* junction on temperature. Straight lines are guides for the eye of exponential ranges. In inset: schematic view of the two-photon absorption in PbTe at 150 K.

This finding supports our assumption that the photovoltage should result from the interband generation of electron and hole pairs. The CO₂ laser photon energy (130 meV) is lower than the forbidden energy gap of PbTe of 216–253 meV over the entire investigated temperature range of 100–180 K, respectively. Therefore, the only mechanism of such generation is two-photon absorption [16]. The inset in Figure 7 illustrates the scheme of the two-photon absorption process. The increase of the photocurrent peak value with lower temperatures is related to the temperature dependence of the forbidden energy gap of PbTe: the lower the temperature, the narrower the energy gap and, correspondingly, the stronger the two-photon absorption.

Another activation energy is equal to 12 meV. The question of this absorption mechanism remains open. It may be related to the impurity levels of indium E_{in} that at low temperatures are located in the conduction band close to its bottom [17]. It was shown that the indium doping of PbTe leads to the formation of long-lived electronic states. However, it is still not clear whether these states are formed by the impurity atoms or whether the doping causes the appearance of another type of defect forming quasilocal states. The model for the formation of defective states in PbTe and its alloys was considered in Refs. [18–24]. Indium impurity has a variable valency and produces a DX center, not a trap [19]. When an electron from lower state E_2 absorbs a long-wave photon, it moves into another energetic state E_1 on the In level and thus into another configuration coordinate (with respect to space and area) [18]. In this case, the long-term relaxation effect of photoconductivity appears due to the formation of an effective barrier caused by the change of two units in the impurity valency upon photoexcitation [18,19]. The two-electron nature of the ground state of the center is confirmed by the absence of a paramagnetic resonance signal of the defects. In this case, the energy level of the defect according to [18] and the corresponding analytical expression for the energy of the center that has captured n electrons ($n = 0, 1, \text{ or } 2$) is

$$\varepsilon_n = \frac{\Delta^2}{2\Delta_0} + (\varepsilon_0 - \Delta)n + U_n \quad (10)$$

where Δ is the coordinate of the configuration determined in energy units, and Δ is equal to the energy shift of one-electron defect level; Δ_0 determines the strength of the electron-phonon interaction in the center, and ε_0 is the energy position of the empty level of the center. $U_n = U \times \beta$ is the energy of the Coulomb repulsion of the two electrons in the center, and β is the Kronecker symbol [18]. The expression for U is [18]:

$$U \approx \frac{e^2}{\varepsilon r} \approx 10 \text{ meV} \quad (11)$$

Here, ε is the permittivity of PbTe [6], and $r \approx 1 \text{ nm}$ is the radius of the quasilocalized electronic state.

Therefore, the optical absorption with activation energy $E_a \approx 12 \text{ meV}$ can be associated with electron transition from the low energy level E_2 to the higher level E_1 [18].

The authors of Ref. [24] have developed a multistage thermoelectric cooler maintaining a temperature of 140–150 K on the cold side. A schematic view of this cooler is presented in Figure 8. The heat sink (7 in Figure 8) is highly efficient if based on heat pipes since the designed power is about 100 watts ($I = 5.1\text{--}6.4 \text{ A}$, $U = 12\text{--}14 \text{ V}$, electric power $P = 60\text{--}90 \text{ W}$). Such heat sinks are quite light due to their very thin fins, and the heat pipes allow placing the sensor in a convenient place. The first part of the cooler is a four-stage module based on the BiTe compound and operates within the 300–180 K temperature range. The second low-temperature two-stage part of the cooler uses n -BiSb and p -BiSbTe alloys and operates at the 180–140 K temperature range, i.e., below 150 K.

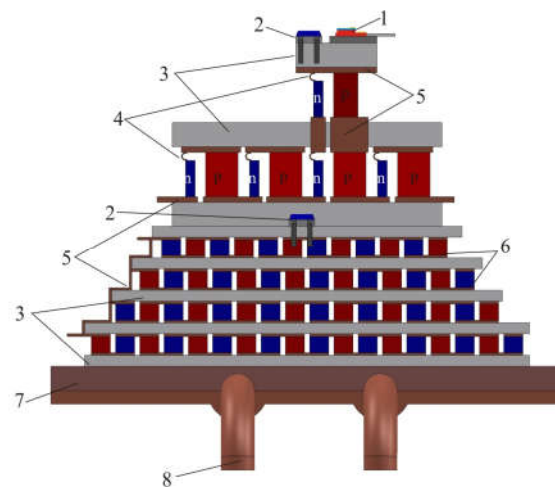


Figure 8. Schematic view of the multistage thermo electric cooler (MTEC). 1—IR detector with the In-doped PbTe *p-n* junction; 2—platinum thermistors; 3—BeO₂ ceramic; 4—copper foil tape; 5—copper busbars; 6—thermocouples of a multistage thermoelectric module; 7—heat sink; 8—heat pipe.

4. Conclusions

The PbTe *p-n* junction diode was fabricated on a PbTe single crystal grown by the Czochralski technique and followed the diffusion of an indium donor impurity into the crystal. The capacitance-voltage and current-voltage characteristics of the sensor were measured over the temperature range from 80 K to 150 K. The saturation current density was $\sim 10^{-5}$ A/cm² at $T = 90$ K, while at $T = 150$ K it was $\sim 10^{-1}$ A/cm².

The concept of an IR sensor on the base of a PbTe diode with a cut-off wavelength $\lambda_c \approx 4$ μm operating at temperatures up to 150 K was demonstrated for the first time. The estimated parameters of the IR sensor demonstrate its high photodetection performance: $R_0A = 1.5$ Ω cm², $J_0 = 3 \times 10^{-3}$ A/cm², and $D^* \approx 10^{10}$ cm Hz^{1/2}/W at temperature $T = 150$ K and wavelength $\lambda \approx 4.2$ μm .

The possibility to sense pulses of long-IR radiation with the PbTe diode was also demonstrated over the 100–180 K temperature range. For the first time, a two-photon absorption-caused photovoltaic effect was observed in PbTe for the wavelength $\lambda = 9.5$ μm at $T = 150$ K.

The advantage of the new diode lies behind its successful operation at temperatures higher than the cryogenic ones, i.e., above 120 K, which opens the perspective of its application supported (cooled) by a solid-state thermoelectric refrigerator.

Author Contributions: Conceptualization, Z.D. and B.D.; methodology, Z.D.; analysis, J.G.; investigation, J.G.; writing—original draft preparation, L.C.; writing—review and editing, Z.D. and J.G.; visualization, B.D.; design of the prototype, B.D.; supervision, Z.D. All authors have read and agreed to the published version of the manuscript.

Funding: This research received no external funding.

Institutional Review Board Statement: Not applicable.

Informed Consent Statement: Not applicable.

Conflicts of Interest: The authors declare no conflicts of interest.

References

1. Rogalski, A. Infrared detectors: Status and trends. *Prog. Quantum Electron.* **2003**, *27*, 59–10.
2. Fuchs, F.; Buerkle, L.; Hamid, R.; Herres, N.; Pletschen, W.; Sah, R.; Kiefer, R.; Schmitz, J. Optoelectronic properties of photodiodes for the mid-and far-infrared based on the InAs/GaSb/AlSb materials family. *Proc. SPIE* **2001**, *4288*, 171–182.

3. Infrared Detectors. Hamamatsu. Selection guide. Available online: https://www.hamamatsu.com/resources/pdf/ssd/infrared_kird0001e.pdf (accessed on 12 December 2020).
4. Ravich Yu, I.; Efimova, B.A.; Smirnov, I.A. *Semiconducting Lead Chalcogenides*; Plenum Press: New York, 1970.
5. Khokhlov, D. *Lead Chalcogenides. Physics and Application*; Taylor & Francis: New York, 2003.
6. Butenko, A.V.; Kahatabi, R.; Mogilko, F.; Strul, R.; Dashevsky, Z.; Kasiyan, V.; Genikov, S. Characterization of high temperature PbTe p-n junctions prepared by thermal diffusion and ion implantations. *J. Appl. Phys.* **2008**, *103*, 024506.
7. Zogg, H.; Fach, A.; Maissen, C.; Masek, J.; Blunier, S. Photovoltaic lead-chalcogenide on silicon infrared sensor arrays. *Opt. Eng.* **1994**, *33*, 1440–1449.
8. Barros, A.S.; Abramof, E.; Rappl, P.H.O. Electrical and optical properties of PbTe p-n junction infrared sensors. *J. Appl. Phys.* **2006**, *99*, 024904.
9. Yasuda, A.; Suto, K.; Takahashi, Y.; Nishizawa, J.I. Mid-infrared photoconductive properties of heavily Bi-doped PbTe p-n homojunction diode grown by liquid-phase epitaxy. *Infrared Phys. Techn.* **2014**, *67*, 609–612.
10. Fill, M.; Debernardi, P.; Felder, F.; Zogg, H. Lead-chalcogenide mid-infrared vertical external cavity surface emitting lasers with improved threshold: Theory and experiment. *Appl. Phys. Lett.* **2013**, *103*, 201120.
11. Zogg, H.; Arnold, M.; Felder, F.; Rahim, M.; Ebnetter, C.; Zaslavitskiy, I.; Quack, N.; Blunier, S.; Dual, J. Epitaxial lead chalcogenides on Si for mid-IR detectors and emitters including cavities. *J. Electron. Mater.* **2008**, *37*, 1497–1503.
12. Dariel, M.P.; Dashevsky, Z.; Jarashnely, A.; Shusterman, S.; Horowitz, A. Carrier concentration gradient generated in p-type PbTe crystals by unidirectional solidification. *J. Cryst. Growth* **2002**, *234*, 164–170.
13. Dashevsky, Z.; Kasiyan, V.; Mogilko, E.; Butenko, F. High-temperature PbTe diodes. *Thin Solid Films* **2008**, *516*, 7065–7069.
14. Dashevsky, Z.; Shusterman, S.; Dariel, M.P.; Drabkin, I. Thermoelectric efficiency in graded indium-doped crystals. *J. Appl. Phys.* **2002**, *92*, 1425–1430.
15. Sze, S.M. *Physics of semiconductor devices*. 2nd ed.; John Wiley and Sons: New York, NY, USA, 1981.
16. Ma, J.; Chiles, J.; Sharma, Y.D.; Krishna, S.; Fathpour, S. Two-photon photovoltaic effect in gallium arsenide. *Opt. Lett.* **2014**, *39*, 5297–5300.
17. Kaidanov, V.I.; Ravich, Yu.I. Deep and resonance states in A^{IV}B^{VI} semiconductors. *Sov. Phys. Usp.* **1985**, *28*, 31–53.
18. Akimov, B.A.; Dmirtiev, A.V.; Khokhlov, D.R.; Ryabova, L.I. A carrier transport and non-equilibrium phenomena in doped PbTe and related materials. *Phys. Stat. Sol.* **1993**, *137*, 9–55.
19. Belogorokhov, A.I.; Volkov, B.A.; Ivanchik, I.I.; Khokhlov, D.R. Model of DX-like Impurity Centers in PbTe (Ga). *JETP Lett.* **2000**, *72*, 123–125.
20. Volkov, B.A.; Ryabova, L.I.; Khokhlov, D.R. Mixed-valence impurities in lead telluride-based solid solutions. *Phys. Usp.* **2002**, *45*, 819–846.
21. Heremans, J.P.; Wiendlocha, B.; Chamoire, A.M. Resonant levels in bulk thermoelectric semiconductors. *Energy Environ. Sci.* **2012**, *5*, 5510–5530.
22. Wiendlocha, B. Fermi surface and electron dispersion of PbTe doped with resonant Tl impurity from KKR-CPA calculations. *Phys. Rev. B.* **2013**, *88*, 205205.
23. Parashchuk, T.; Dashevsky, Z.; Wojciechowski, K. Feasibility of a high stable PbTe:In semiconductor for thermoelectric energy applications. *J. Appl. Phys.* **2019**, *125*, 245103.
24. Parashchuk, T.; Sidorenko, N.; Sorokin, A.; Maksymuk, M.; Dzunzda, B.; Vichor, L.; Dashevsky, Z. Development of cryogenic thermoelectric cooler (CTEC). *Cryogenics* **2020**. In press.