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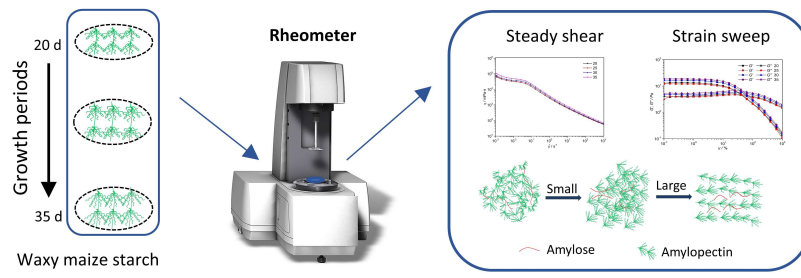
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**Zipeng Liu:** Methodology, Validation, Formal analysis, Investigation, Data Curation, Writing - Original Draft, Visualization. **Ling Chen:** Writing - Review & Editing, Conceptualization, Methodology, Resources, Supervision, Project administration, Funding acquisition. Writing - Review & Editing. **Pingping Bie:** Methodology, Writing - Review & Editing. **Fengwei Xie:** Methodology, Resources, Writing - Review & Editing, Visualization. **Bo Zheng:** Writing - Review & Editing, Conceptualization, Methodology, Resources, Supervision, Project administration.

**Graphical Abstract**

Journal Pre-proof

1 **An insight into the structural evolution of waxy maize starch**  
2 **chains during growth based on nonlinear rheology**

3  
4 Zipeng Liu <sup>a</sup>, Ling Chen <sup>a,\*\*</sup>, PingPing Bie <sup>a</sup>, Fengwei Xie <sup>b,c</sup>, Bo Zheng <sup>a,\*</sup>

5 <sup>a</sup> Ministry of Education Engineering Research Center of Starch & Protein Processing, Guangdong  
6 Province Key Laboratory for Green Processing of Natural Products and Product Safety, School of Food  
7 Science and Engineering, South China University of Technology, Guangzhou 510640, China

8 <sup>b</sup> International Institute for Nanocomposites Manufacturing (IINM), WMG, University of  
9 Warwick, Coventry CV4 7AL, United Kingdom

10 <sup>c</sup> School of Chemical Engineering, The University of Queensland, Brisbane, Qld 4072, Australia

11  
12 \*Corresponding author. Email: bzheng@scut.edu.cn; (B. Zheng);

13 \*\*Corresponding author. Email: felchen@scut.edu.cn; (L. Chen)

14  
15 **Abbreviations**

16 WMS, waxy maize starch; NMR, nuclear magnetic resonance; GPC, gel permeation  
17 chromatography; LS, light scattering; LAOS, large amplitude oscillating shear; SAOS, small  
18 amplitude oscillating shear; LVR, linear viscoelastic region; NVR, nonlinear viscoelastic region;  
19 DP, degree of polymerization

20 **Abstract:** This work investigated the chain structure and large amplitude oscillatory shear  
21 (LAOS) rheological properties of waxy maize starch (WMS) at different growth periods and  
22 established a motion model to clarify the evolution of chain structure. During the growth period  
23 of 20-25 days, the apparent viscosity and dynamic moduli for WMS paste decreased, along with  
24 a greater degree of nonlinearity. This could be correlated with an increasing branching degree of  
25 starch due to more short-branched chains generated during growth. With the growth period  
26 reaching 25 days, the apparent viscosity, dynamic moduli and zero-strain nonlinearity ( $Q_0$ ) for  
27 WMS increased and the b value deviated from 2 seriously, indicating obvious nonlinearity  
28 accompanied by a weak stress overshoot and shear-thickening. This could be explained by the  
29 growing length of branched chains and the rising content of B<sub>3</sub>- and B<sub>4</sub>-chains. Thus, this work  
30 demonstrate LAOS rheology can be used as a new and effective method to characterize the chain  
31 structural evolution of starch through monitoring the motion pattern of different starch chains  
32 under shear conditions.

33

34 *Keywords:* Waxy maize starch; Amylopectin; Nonlinear rheology; Large amplitude oscillating  
35 shear

36

## 37 **1 Introduction**

38 Starch is the main storage carbohydrate in plants. As a natural polymer material, starch has  
39 been widely used in diverse applications such as in food, pharmaceutical and chemical industries  
40 due to its advantages such as biodegradability, biocompatibility and low-price (Kochkina &  
41 Lukin, 2020). Starch consists of amylose and amylopectin, which form a semi-crystalline  
42 structure (Cai & Shi, 2010). According to amylose content, maize starch can be subdivided into  
43 high-amylose maize starch, normal maize starch and waxy maize starch. Waxy maize starch  
44 (WMS) is almost composed of amylopectin, which has a higher molecular weight and branching  
45 degree (Bertoft, 2013). The side chains of amylopectin can be divided into the outermost  
46 branchless A-chains, the backbone C-chains with a reductive end, and the B-chains that connect  
47 A-chains and C-chains, which can be further subdivided into B<sub>1</sub>-chains, B<sub>2</sub>-chains and B<sub>3</sub>-chains  
48 depending on the degree of polymerization (Hizukuri, 1986; Madhusudhan & Tharanathan,  
49 1996; Singh, Inouchi, & Nishinari, 2006). The understanding of starch chain structure is highly  
50 important as it is closely related to starch processability (Hsieh, Liu, Whaley, & Shi, 2019; Li,  
51 Su, Shi, Wang, & Chen, 2017). However, it is still highly challenging to analyze the chain  
52 structure of natural polymers, especially branched polymers. Commonly used spectroscopy and  
53 chromatographic analysis methods such as nuclear magnetic resonance (NMR), gel permeation  
54 chromatography (GPC), and light scattering (LS), all have certain limitations. For example,  
55 NMR cannot distinguish branched chains with length of six or more carbon atoms (Usmanov,  
56 1991). GPC-LS can qualitatively explain the existence of branched chains by mean square radius  
57 of gyration, but the calculation of branching degree still needs to be based on the assumption of  
58 molecular structure (Wang, Kharchenko, Migler, & Zhu, 2004). Nowadays, there is still no direct  
59 method to determine the topological molecular structure of natural polymers. However, when the

60 molecular mass of starch branched chains exceed a critical value, chain entanglement results. In  
61 this case, these branched chains can be called rheologically long branched chains. Specifically,  
62 the rheological properties of the starch solution are sensitive to the molecular structure especially  
63 branched-chain length, and thus rheology can be used for the qualitative and quantitative  
64 characterization of starch chain structure (Liu et al., 2019b).

65 Rheological properties are important properties of starch-based food and materials, such as  
66 during hot-extrusion 3D printing (Liu, Chen, Zheng, Xie, & Chen, 2020) and starch film (Silva-  
67 Weissa, Bifani, Ihl, Sobralc, & Gómez-Guillénd, 2013), having great scientific and practical values in  
68 understanding the starch structure and guiding starch processing (Yong et al., 2019). Starch paste  
69 is a kind of pseudoplastic fluid with shear thinning behavior, as starch chains undergo  
70 disentanglement and orientational arrangement under shear, leading to decreases in flow  
71 resistance and viscosity (Ketthaisong, Suriharn, Tangwongchai, & Lertrat, 2014; Zhang, Li,  
72 Zhang, Wei, & Fang, 2019). Dynamic rheology can effectively reflect the processing properties  
73 and monitor the movement of molecular chains and the structural evolution of materials during  
74 processing (Lee, Porcar, & Rogers, 2019). When starch paste is subjected to small-amplitude  
75 shear, its solidity (elastic modulus) becomes more apparent, and its elastic modulus depends on  
76 the degree of heterogeneity (Rodríguez-Hernández, Durand, Garnier, Tecante, & Doublier,  
77 2006). Under large-strain shearing, the junctions connecting starch chains in starch paste are  
78 broken and the chains are rearranged (Carmona, Ramírez, Calero, & Muñoz, 2014). Since strain  
79 amplitude and frequency can be controlled independently under nonlinear viscoelastic  
80 measurement, more information about the structure and properties of complex materials can be  
81 obtained (Ghosh et al., 2019).

82           With the rapid development of precision instruments and computational simulations, large-  
83 amplitude oscillating shear (LAOS) rheology has drawn much attention to probe the  
84 microstructure of complex polymer systems (Ducarci, Yazar, & Kokini, 2017; Oliver, Scholten,  
85 & van Aken, 2015). Hyun et al. (2009) investigated the nonlinear rheological properties of linear  
86 and comb polystyrene and found two obvious peaks on the nonlinear viscoelasticity curve. The  
87 peaks at low and high frequency correspond to the relaxation of backbone chains and branched  
88 chains, respectively (Hyun & Wilhelm, 2009). They also reported that the relative intensity of  
89 3rd harmonics ( $I_{3/1}$ ) for linear polymers is proportional to the square of strain ( $\gamma_0^2$ ), while the  
90 slope of  $I_{3/1}$  for branched polymers is normally less than 2. Therefore, the branching degree of  
91 polymers can be measured quantitatively by the slope, and zero-strain nonlinearity can be used  
92 for defining the nonlinearity of rheological behavior (Hyun et al., 2009). Oelschlaeger et al.  
93 studied the microphase orientation/reorientation alignment behavior of two kinds of block  
94 copolymers by LAOS rheology and found that the kinetics of orientation was strongly dependent  
95 on the strain amplitude, and  $I_{3/1}$  can be described by a stretched exponential function to  
96 distinguish diblock copolymers and triblock copolymers with lamellar structure (Oelschlaeger et  
97 al., 2010). By examining the interaction between fish gelatin and Arabic gum through LAOS  
98 testing, Anvari et al. found the two gels had strong electrostatic attraction under a suitable pH  
99 condition, thus forming a compact rigid network structure, which increased viscous dissipation  
100 and led to a shift from elastic-dominated to viscous-dominated behavior (Anvari & Joyner,  
101 2018). The main analysis methods of nonlinear rheology include Fourier-transform rheology,  
102 Lissajous curves, stress decomposition, etc., and they can separate the structural characteristics  
103 and flow behavior contained in a nonlinear stress response, which is highly useful for



104 understanding the molecular topology, molecular interactions, micromorphology and  
105 crystallization of polymer fluids (Hyun et al., 2011).

106 In this study, WMS at different growth periods was studied by ion chromatography and  
107 LAOS measurements to understand its chain-length distribution and nonlinear rheological  
108 properties and to establish a correlation between them. Based on this, a motion model of starch  
109 chains under different shear conditions was established. This study demonstrates the application  
110 of nonlinear rheology in the analysis of starch chain structure and provides new knowledge  
111 regarding the starch chain structural evolution during growth.

## 112 **2 Materials and methods**

### 113 **2.1 Materials**

114 Waxy maize of inbred line L5 was planted by the Zhong Luotan Experimental Base of the  
115 Guangdong Academy of Agricultural Sciences (Guangzhou, China), and waxy maize seeds at  
116 different growth periods (20 d, 25 d, 30 d and 35 d) were collected as experimental materials.  
117 Amylose and amylopectin standards were purchased from the Heilongjiang Academy of  
118 Agricultural Sciences (Harbin, China). Isoamylase (Cat. No. I5284, activity  $\geq 3,000,000$  U/mg)  
119 from pseudomonas was purchased from Sigma-Aldrich Co., Ltd (Santa Clara, USA). Double  
120 distilled water was used in this work. All reagents were of analytical grade and were used as  
121 received without further purification.

### 122 **2.2 Sample preparations**

123 WMS was isolated from waxy maize following our previous method with minor  
124 modification (Wang, Wang, Li, Chen, & Zhang, 2017). Waxy maize seeds at different growth  
125 periods (20 d, 25 d, 30 d and 35 d) were selected and mixed with 0.45% (w/w) sodium

126 pyrosulfite at a weight ratio of 1:2. The mixture was smashed in a Waring Blender (HR 1727  
127 Philips, Zhuhai, China) for 30 s, then stood at 4 °C for 12 h. The slurry was sieved by gauze and  
128 sieve (75 µm) to eliminate fibers. Next, the slurry was centrifuged at 2800 g for 15 min and the  
129 obtained precipitate was washed with sufficient water three times. Finally, the precipitate was  
130 evenly mixed with absolute ethyl alcohol and filtered through a sieve (75 µm), and starch was  
131 dried at 35 °C for 24 h. The prepared starch was named as starch 20, starch 25, starch 30 and  
132 starch 35 according to its growth periods, respectively.

### 133 **2.3 Amylose content determination**

134 Amylose content of starch was determined using a previous method (Chen, Liang, Li,  
135 Chen, & Xie, 2016; Liu, Zhang, Chen, Li, & Zheng, 2019a). The prepared starch (100 mg, dry  
136 basis) was dispersed in 1 mol/L NaOH and then distilled water was added to achieve a 1 mg/mL  
137 solution. Next, I<sub>2</sub>/KI solution (0.0025 mol/L I<sub>2</sub> and 0.0065 mol/L KI) was used for amylose  
138 complexation and the absorbance at 620 nm of the product was measured using a  
139 spectrophotometer (Evolution 201, Thermo Scientific, USA). Amylose content was obtained  
140 from a standard curve range from 0% to 100% that established using solutions of amylose and  
141 amylopectin, and amylopectin content was equal to 100% minus amylose content.

### 142 **2.4 Amylopectin chain-length distribution**

143 The prepared starch (2 mg, dry basis) was dispersed in 0.5 mL of ethanol (95%) and 4.5  
144 mL of distilled water. Then, the mixture was heated in boiling water for 60 min. Next, 2.5 mL of  
145 the gelatinized starch paste was mixed with 125 µL of sodium acetate solution (600 mmol/L), 25  
146 µL of sodium hydrazoic acid solution (2%, w/w) and 10 µL of isoamylase. After standing at 38  
147 °C for 24 h, the pH was adjusted to 9.0 with ammonia solution (5%, w/w) and 375 µL of sodium  
148 borohydride solution (2%, w/w) was added. The sample was equally divided into 600 µL per

149 tube and dried under vacuum at 25 °C. The dried sample was dissolved in 60  $\mu$ L of NaOH  
 150 solution (1 mol/L) and diluted with 540  $\mu$ L of distilled water, then centrifugated at 12000 rpm for  
 151 10 min. All supernatant prepared was uniformly mixed into one tube for analysis.

152 The amylopectin chain-length distribution was performed using a high-performance ion  
 153 chromatographer (HPIC) (ICS-5000, Dionex, USA) coupled with a pulsed amperometric  
 154 detection. The sample was eluted from an AP1 column (Dionex, USA) at a flow rate of 0.5  
 155 mL/min at 30  $\square$  using a mixture of three eluents: 0.1 mol/L NaOH (eluent A), 0.1 mol/L NaOH  
 156 + 1 mol/L NaAc (eluent B), and water (eluent C). PAD signal was analyzed with the software  
 157 Peaknet (Dionex, USA) to obtained chain-length distribution (Kong, Bertoft, Bao, & Corke,  
 158 2008; Noda, Takahata, & Sato, 1995).

159

160

Table 1 Elution gradient (Kong, Bertoft, Bao, & Corke, 2008).

t / min	$\phi_A$ / %	$\phi_B$ / %	$\phi_C$ / %
0	55	10	35
8	40	25	35
25	35	30	35
65	20	45	35
75	55	10	35

161

## 162 2.5 Rheological measurements

163 Steady shear testing was performed on an MCR302 rheometer (Anton Paar, Austria) with a  
 164 parallel-plate geometry (25 mm diameter and 1 mm gap). WMS suspension of 4% (w/w, dry  
 165 basis) concentration was heated and stirred at 95 °C for 20 min, and then WMS paste was made  
 166 and preserved at 25  $\square$ . WMS paste was loaded and equilibrated at 25  $\square$  for 5 min. Steady shear

167 testing in a range of  $10^{-3}$ - $10^3$  s<sup>-1</sup> was conducted, and flow curves were acquired using  
168 RheoCompass 1.21 software (Fang et al., 2019).

169 Strain sweep testing was performed on an ARES-G2 rheometer (TA Instruments, USA)  
170 with a parallel-plate geometry (25 mm diameter and 1 mm gap). Similarly, 4% WMS paste was  
171 loaded and equilibrated. Dynamic strain sweep testing in a range of 0.01%-1000% at 6.28 rad/s  
172 was undertaken (Liu, Chen, Zheng, Xie, & Chen, 2020). Raw data under oscillation shear was  
173 analyzed and processed by MITlaos (MATLAB Version, USA).

## 174 **2.6 Statistical analysis**

175 All tests were conducted at least in triplicate and the experimental data were analyzed  
176 using SPSS Statistics 23.0 (IBM, USA). One-way analysis of variance was used to find the  
177 significant difference by Duncan's test ( $p \leq 0.05$ ).

## 178 **3 Results and discussion**

### 179 **3.1 Amylopectin content and amylopectin chain-length distribution**

180 Amylopectin has a higher density of branching attached by  $\alpha$ -D-(1, 6)-glucosyl linkage in  
181 the main  $\alpha$ -D-(1,4)-glucosyl chain, which is mainly composed of A-chains, B-chains and C-  
182 chains. Among them, there is only one C-chain in each amylopectin molecule, which has both a  
183 non-reductive end and a reductive end. As branched chains, both A-chains and B-chains has only  
184 non-reductive ends. Thus, enzymatic glucose chains produced by isoamylase also have multiple  
185 hydroxyl groups, and they are dissociated into the form of anions in a strongly basic buffer.  
186 Amylopectin chain-length distribution can be analyzed by measuring the retention time through  
187 HPIC. As glucose chains of different length have different retention time, the chain-length  
188 distribution can be determined according to the peak time and peak area in the chromatogram

189 (Kong et al., 2008). Generally, the chain-length distribution of starch can be divided into four  
 190 parts: A-chains (DP 6-12), B<sub>1</sub>-chains (DP 13-24), B<sub>2</sub>-chains (DP 25-36) and B<sub>3</sub>-chains (DP>36)  
 191 (Madhusudhan et al., 1996; Singh et al., 2006). As presented in Table 1, the amylopectin content  
 192 and the B<sub>2</sub>- and B<sub>3</sub>-chain lengths of WMS all increased ( $P \leq 0.05$ ) with a prolonged growth  
 193 period, while the length of B<sub>1</sub>-chain, as the main type of amylopectin branch, increased first and  
 194 then decreased. The B<sub>1</sub>-chain length of WMS increased notably during 20-25 d, and meanwhile,  
 195 amylopectin produced more short-branched chains. Nonetheless, when the growth period was  
 196 above 25 d, the degree of polymerization of branched chains further increased, leading to  
 197 increases in the B<sub>2</sub>- and B<sub>3</sub>-chain lengths.

198

199 Table 2 Amylopectin content and amylopectin chain-length distribution of WMS at different  
 200 growth periods.<sup>A</sup>

Sample	Amylopectin content / %	A-chains / %	B <sub>1</sub> -chains / %	B <sub>2</sub> -chains / %	B <sub>3</sub> -chains / %
20	98.09±0.03 <sup>c</sup>	20.6±0.2 <sup>a</sup>	42.5±0.1 <sup>b</sup>	18.9±0.1 <sup>c</sup>	18.0±0.2 <sup>b</sup>
25	98.18±0.02 <sup>c</sup>	20.2±0.2 <sup>b</sup>	42.9±0.1 <sup>a</sup>	18.8±0.3 <sup>c</sup>	18.1±0.2 <sup>b</sup>
30	98.53±0.01 <sup>b</sup>	20.0±0.2 <sup>b</sup>	41.6±0.1 <sup>c</sup>	20.0±0.2 <sup>b</sup>	18.4±0.3 <sup>a</sup>
35	98.89±0.02 <sup>a</sup>	19.5±0.3 <sup>c</sup>	41.2±0.2 <sup>d</sup>	20.8±0.3 <sup>a</sup>	18.5±0.2 <sup>a</sup>

201 <sup>A</sup> The chain-length distribution is the percentage of individual chains in the total of the four chains.  
 202 Different letters above the same column indicate a significant difference ( $P \leq 0.05$ ).

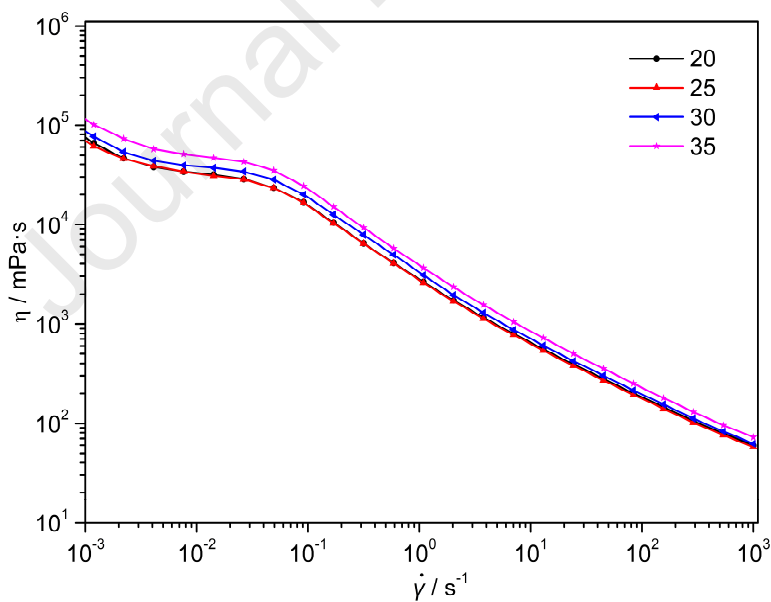
203

### 204 3.2 Steady shear characteristics

205 Fig. 1 shows the viscosity curves for WMS at different growth periods. Starch pastes are  
 206 typical pseudoplastic fluids with strong shear-thinning behavior. Regarding this, intertwined  
 207 starch chains were stretched under shear force and reoriented along the shear direction, causing  
 208 decreases in shear resistance and viscosity. In the low-shear rate region, the original structure of  
 209 WMS paste was destroyed, which was mainly caused by the unwinding of main chains, which

210 reduced the viscosity. With increasing shear rate, WMS paste flowed into the high-shear rate  
 211 region. In this case, the branched chains of starch also began to orientationally arrange under  
 212 shear, resulting in stronger shear-thinning behavior (Hyun et al., 2009). Compared with starch  
 213 20, the branching degree of starch 25 was higher with more short-branched chains, which limited  
 214 chain entanglement to some extent and thus resulted in lower viscosity of the starch paste. With  
 215 an even longer growth period, the DP of branched chains increased, contributing to chain  
 216 entanglement in the low-shear rate region and thus increasing the starch paste viscosity.  
 217 However, in the high-shear rate region, the long-branched chains ( $DP \geq 25$ ) also promoted the  
 218 unwinding and orientation of chains, and the formed structure was more sensitive to shear,  
 219 showing a strong non-Newtonian flow behavior with shear-thinning.

220



221

222

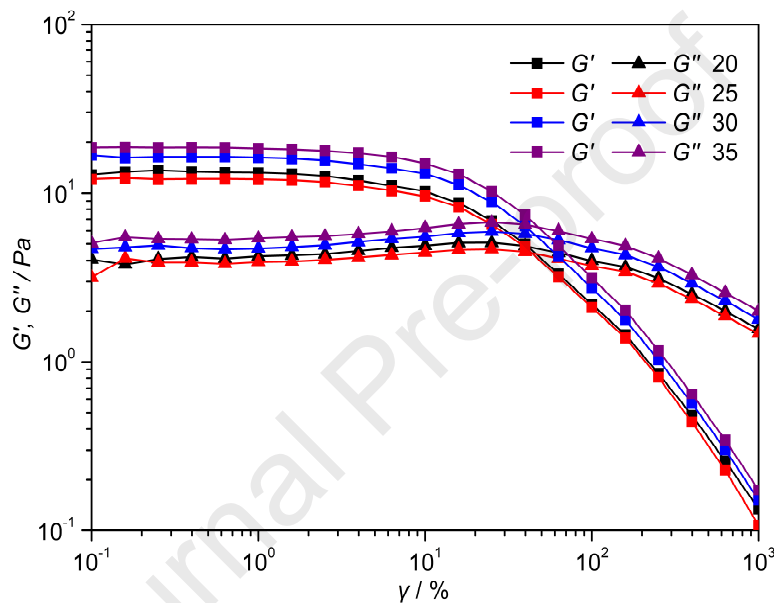
Fig. 1 Viscosity curves for WMS pastes at different growth periods.

223

### 224 3.3 Dynamic rheological properties

225 The dynamic viscoelasticity of WMS paste was determined by strain sweep testing. As  
226 shown in Fig. 2, when the strain was 0.1-1%, the  $G'$  and  $G''$  values of WMS paste did not change  
227 with strain, indicating that the WMS paste was in the LVR and its network structure deformed  
228 reversibly without destruction. Once the strain was greater than 1% and exceeded the LVR,  $G'$   
229 decreased with increasing strain, while  $G''$  first rose slightly and then decreased abruptly. Once  
230 WMS paste entered into NVR, its network structure began to collapse irreversibly, accompanied  
231 by weakened chains interactions and mechanical strength of the paste, until reaching the flow  
232 point where  $G'$  equals  $G''$ . In the strain sweep curves,  $G''$  had a pronounced maximum value but  
233  $G'$  did not, showing a typical phenomenon of weak stress overshoot (Cloitre, Borrega, & Leibler,  
234 2000; Hyun et al., 2011). This observation could indicate that the network junctions of WMS  
235 paste were in a dynamic change process in the transition region. With increasing strain, the main  
236 chains of starch were unwinding while partial long branched chains became entangled under  
237 shearing. At this point, both the rates of creation and loss of junctions were positive, leading to a  
238 balanced state. Afterwards, the creation rate was smaller than the loss rate, leading to the  
239 weakening of chains entanglement. This further destroyed the cohesiveness of the network  
240 structure, resulting in the occurrence of a yield point. From the dynamic strain sweep curves  
241 (Fig. 2), both the  $G'$  and  $G''$  values for starch 25 were smaller than those for starch 20, while the  
242  $G'$  and  $G''$  values for starch 35 were the largest with an obvious stress overshoot. This was due to  
243 the higher branching degree of starch 25 compared with that of starch 20. Nonetheless, its  
244 branched chains were still not long enough to form entanglements, which led to a looser structure  
245 of WMS paste with a larger molecular gap. In this case, the interactions between main chains  
246 were restricted to some extent and the starch paste had reduced viscosity and rigidity as shown

247 by decreases in  $G'$  and  $G''$ . With the prolonging of the growth period, the length of branched  
 248 chains rose, and entanglement between branched chains occurred creating more junctions, which  
 249 increased the cohesiveness and the energy loss of the starch paste, as shown by increases in  $G'$   
 250 and  $G''$ .  
 251



252  
 253 Fig. 2 Dynamic strain sweep curves for WMS pastes at different growth periods.

### 254 255 3.4 LAOS rheology analysis

256 Oscillatory shear testing is classified into two categories, namely small-amplitude  
 257 oscillatory shear (SAOS) and LAOS (Melito, Daubert, & Foegeding, 2012). When the applied  
 258 strain amplitude ( $\gamma_0$ ) is small enough, the stress ( $\sigma(t)$ ) responds within the linear viscoelastic  
 259 region (LVR). With increasing strain at a certain frequency, the rheological behavior transits  
 260 from linear to nonlinear gradually and falls into the nonlinear viscoelastic region (NVR)  
 261 (Ptaszek, 2017). When the strain is low in the LVR, the storage modulus ( $G'$ ) and loss modulus



262 ( $G''$ ) are independent of strain, and  $\sigma_{(t)}$  is a sine function. When the strain exceeds a certain value,  
 263 the moduli begin to change with strain, and the waveform of  $\sigma_{(t)}$  becomes distorted and  
 264 transforms into a non-sinusoidal function, in which case the common linear viscoelastic theory is  
 265 not applicable (Narsimhan, 1994). The main analysis methods for LAOS results proposed are  
 266 Fourier-transform rheology, Lissajous curves, and stress decomposition.

267 In Fourier-transform rheology, the relative strength of higher harmonics is taken as the  
 268 nonlinear metric, and the non-sinusoidal stress generated from LAOS testing is represented by  
 269 the Fourier series of elastic scaling (Eq. (1)) and viscous scaling (Eq. (2)) (Yan, Costanzo, Jeong,  
 270 Chang, & Vlassopoulos, 2016). In LAOS testing,  $\sigma_{(t)}$  is a non-sinusoidal periodic function and  
 271 higher harmonics contribution appear under the Fourier transform, thus there are obvious signal  
 272 peaks at the  $\omega t$ ,  $3\omega t$ ,  $5\omega t$  ... in the Fourier-transform rheology spectrum (Cogswell, 1991;  
 273 Wilhelm, Reinheimer, & Ortseifer, 1999). Wilhelm et al. studied the relative intensity of a high  
 274 harmonic and the first harmonic ( $I_{n/1} = I_n / I_1$ ) and found that the nonlinear rheological behavior  
 275 of complex materials with different structures can be characteristic by  $I_{3/1}$  (Wilhelm, Reinheimer,  
 276 Ortseifer, Neidhöfer, & Spiess, 2000). The nonlinear coefficient ( $Q$ ) (Eq. (3)) in Fourier-  
 277 transform rheology provides strong evidence for the transformation from LVR to NVR.  
 278 Furthermore, the zero-strain nonlinearity ( $Q_0$ ) (Eq. (5)) is defined as a constant value at relatively  
 279 small strain, which can effectively represent the inherent nonlinearity of complex fluids (Hyun et  
 280 al., 2009).

$$281 \quad \sigma(t, \omega, \gamma_0) = \gamma_0 \sum_{n=odd} [G'_n(\omega, \gamma_0) \sin(n\omega t) + G''_n(\omega, \gamma_0) \cos(n\omega t)] \quad (1)$$

$$282 \quad \sigma(t, \omega, \gamma_0) = \dot{\gamma}_0 \sum_{n=odd} [\eta'_n(\omega, \gamma_0) \sin(n\omega t) + \eta''_n(\omega, \gamma_0) \cos(n\omega t)] \quad (2)$$

$$283 \quad Q = I_{3/1} / \gamma_0^2 \quad (3)$$

$$284 \quad \log I_{3/1} = a + b \log \gamma_0 \quad (4)$$

$$285 \quad Q_0 = \lim_{\dot{\gamma}_0 \rightarrow 0} Q(\dot{\gamma}_0) = \lim_{\dot{\gamma}_0 \rightarrow 0} (I_{3/1}/\dot{\gamma}_0^2) \quad (5)$$

286 Where  $\dot{\gamma}_0$  represents the maximum strain-rate, and  $G'_n$  and  $G''_n$  represent the elastic and  
 287 viscous moduli for the Nth harmonic, and  $\eta'_n$  and  $\eta''_n$  represent the apparent viscosity in phase  
 288 and out of phase for the Nth harmonic, and  $I$ ,  $Q$  and  $Q_0$  represent the relative intensity of high  
 289 harmonic, zero-strain nonlinearity and the slope of the fitting curve respectively.

290 The stress response in the NVR is processed by Fourier-transform to obtain the  
 291 contribution of higher harmonics, and it is found that the amplitude of the 3rd harmonic is high  
 292 and available. Therefore the nonlinear viscoelasticity of WMS paste at different growth periods  
 293 can be investigated by  $I_{3/1}$  (Hyun et al., 2011). Fig. 3a shows that  $I_{3/1}$  rose from 1% to 15% with  
 294 increasing strain. In LAOS testing, the branching degree of WMS can be measured quantitatively  
 295 by the slope  $b$  and  $Q_0$ , which were calculated by fitting using Eq. (4) and (5) (Fig. 3). As a rule,  
 296 the more the  $b$  value deviated from 2, the higher was the branching degree of WMS paste; and a  
 297 larger  $Q_0$  value indicates a greater degree of the inherent nonlinearity. The prolonging of the  
 298 growth period was found to decrease  $b$  from 0.7384 to 0.4671 (Table 3), which deviated from 2  
 299 largely, while  $Q_0$  increased from 0.0017 to 0.0087 (Table 3). This indicates that the inherent  
 300 nonlinear characteristic of WMS paste became more significant. These results were ascribed to  
 301 increases in amylopectin content and the length of branched chains in WMS. In this case, there  
 302 was an unstable network structure in WMS paste, which was prone to show nonlinearity under  
 303 LAOS testing.

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Fig. 3 Curves of  $I_{3/1}$  and  $Q$  for WMS pastes at different growth periods.

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Table 3 Fitting parameters for WMS pastes at different growth periods.

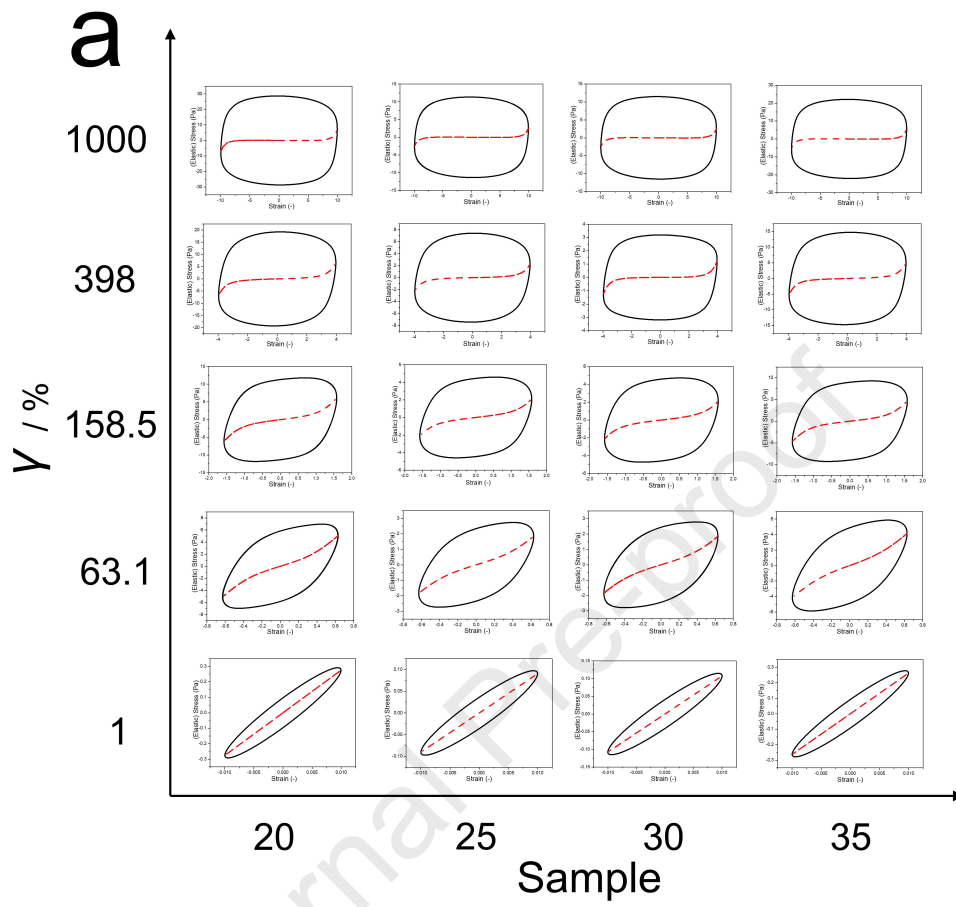
	20	25	30	35
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<i>b</i>	0.7384	0.5998	0.5913	0.4671
<i>Q<sub>0</sub></i>	0.0017	0.0036	0.0039	0.0087

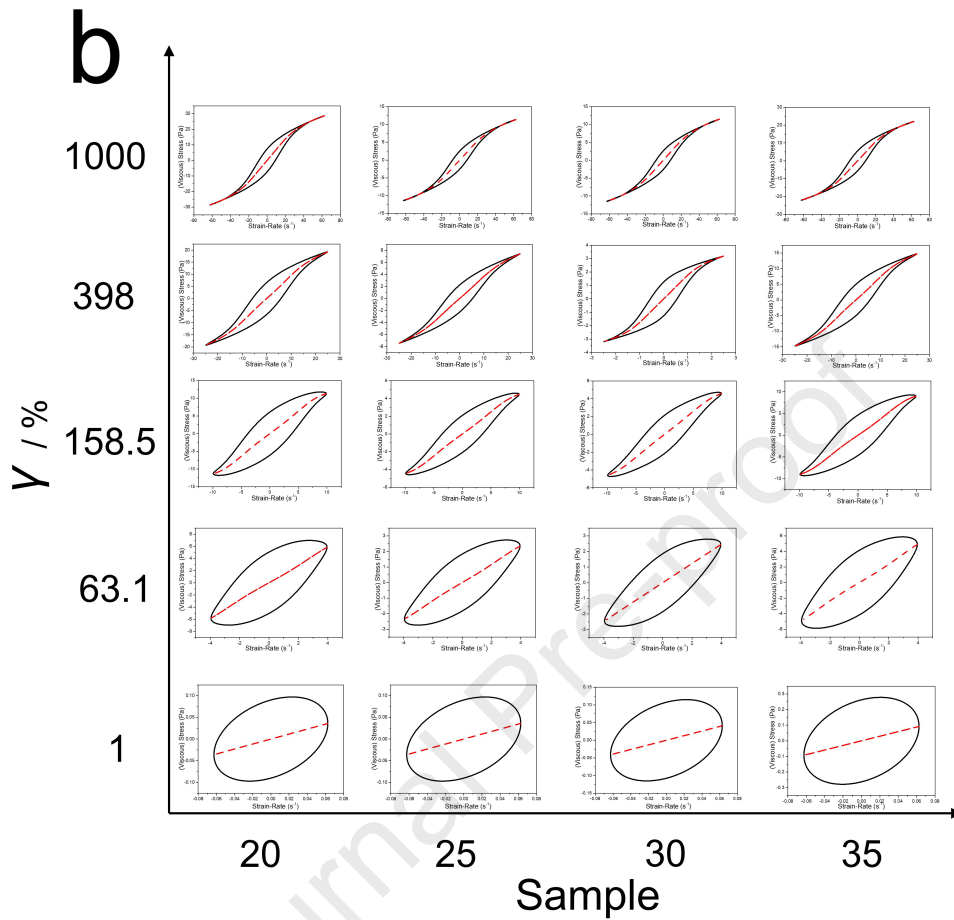
310

311 Nonlinear stress response can be probed by associated Lissajous curves, where the  
312 obtained stress curve can be easily separated into the strain function and the shear rate function  
313 to display the elastic contribution and the viscous contribution of the nonlinear response,  
314 respectively (Ptaszek, et al., 2016). A Lissajous curve eliminates the influence of time variables,  
315 and different Lissajous curves are generated with different frequency and strain. Notably, the  
316 elastic and viscous Lissajous plots can be represented as a function of strain (Fig. 4a) and strain-  
317 rate (Fig. 4b), and the dotted line indicates elastic stress and viscous stress, respectively (Anvari  
318 et al., 2018). All elastic and viscous Lissajous plots for WMS pastes at different growth periods  
319 were elliptic at 1% strain, and the elastic and viscous stress were in a straight line, indicating  
320 ideal viscoelastic behavior. With increasing strain, the Lissajous plots changed from ellipse to  
321 parallelogram with distortion, and the elastic and viscous stresses became bent, indicating  
322 nonlinear viscoelasticity with the appearance of high harmonic contribution. In the high strain  
323 range, the slope of elastic stress increased significantly, suggesting strain-stiffening behavior of  
324 the WMS paste (Fig. 4a), while the slope of viscous stress decreased with shear-thinning in the  
325 high strain-rate range (Fig. 4b).

326



327



328

329 Fig. 4 Lissajous plots for WMS pastes at different growth periods. (a) the elastic parts; (b) the  
 330 viscous parts.

331

332  $G'_L$  and  $G'_M$  are determined as the secant modulus at the maximum and zero strains.

333 Similarly,  $\eta'_L$  and  $\eta'_M$  are determined as the dynamic viscosity at the maximum and zero shear

334 rates (Ewoldt, Hosoi, & McKinley, 2008). As can be seen from Table 4, the  $G'_L$  and  $G'_M$ ,  $\eta'_L$  and

335  $\eta'_M$  values for WMS pastes at different growth periods were equal at 1% strain, which also

336 suggests that the tests were within the LVR. When the strain exceeded 1%, the  $G'_L$  and  $G'_M$ ,  $\eta'_L$

337 and  $\eta'_M$  value began to be different, and the tests were beyond the LVR, resulting in strain-

338 stiffening/softening and shear-thickening/thinning. Under the same strain, the LAOS parameters

339 for WMS paste decreased first and then increased with a longer growth period. Among them, the  
 340 parameters for starch 25 were the smallest, in agreement with the strain sweep results. Compared  
 341 with that of starch 20, the amylopectin content of starch 25 increased but its branched chains  
 342 were relatively short, thus the presence of short-branched chains without entanglements inhibited  
 343 interactions between main chains and there were fewer junctions in the network structure,  
 344 leading to lower rigidity and viscosity. When the growth period was more than 25 d, the length  
 345 of branched chains increased significantly, which led to enhanced chain interactions, resulting in  
 346 increased elastic moduli and viscosity of the starch paste.

347

348

Table 4 LAOS parameters for WMS pastes at different growth periods.

Sample	Viscoelasticity	$\gamma = 1\%$	$\gamma = 63.1\%$	$\gamma = 158.5\%$	$\gamma = 398\%$	$\gamma = 1000\%$
20	$G_M' / \text{Pa}$	27.606±0.031	4.679±0.016	1.245±0.015	0.154±0.002	-0.030±0.001
	$G_L' / \text{Pa}$	27.606±0.031	8.036±0.019	3.860±0.018	1.662±0.004	0.741±0.012
	$\eta_M' / \text{Pa}\cdot\text{s}$	1.541±0.018	1.329±0.010	1.179±0.013	0.891±0.008	0.689±0.018
	$\eta_L' / \text{Pa}\cdot\text{s}$	1.541±0.018	1.477±0.011	1.141±0.014	0.773±0.019	0.458±0.014
25	$G_M' / \text{Pa}$	9.042±0.022	1.825±0.015	0.474±0.013	0.028±0.001	-0.014±0.001
	$G_L' / \text{Pa}$	9.042±0.022	2.907±0.017	1.322±0.010	0.569±0.018	0.253±0.007
	$\eta_M' / \text{Pa}\cdot\text{s}$	0.576±0.017	0.547±0.009	0.459±0.011	0.348±0.010	0.269±0.009
	$\eta_L' / \text{Pa}\cdot\text{s}$	0.576±0.017	0.593±0.002	0.449±0.014	0.298±0.009	0.182±0.009
30	$G_M' / \text{Pa}$	10.746±0.023	1.798±0.014	0.469±0.012	0.031±0.002	-0.023±0.001
	$G_L' / \text{Pa}$	10.746±0.023	2.945±0.016	1.357±0.014	0.617±0.012	0.263±0.010
	$\eta_M' / \text{Pa}\cdot\text{s}$	0.652±0.018	0.616±0.009	0.495±0.010	0.413±0.014	0.280±0.010

$\eta_L' / \text{Pa}\cdot\text{s}$	0.652±0.018	0.617±0.013	0.463±0.008	0.319±0.014	0.184±0.005
$G_M' / \text{Pa}$	26.331±0.027	4.232±0.012	0.951±0.011	0.102±0.008	-0.024±0.001
$G_L' / \text{Pa}$	26.331±0.027	6.598±0.018	2.950±0.017	1.265±0.017	0.561±0.0013
$\eta_M' / \text{Pa}\cdot\text{s}$	1.472±0.012	1.103±0.010	0.907±0.016	0.689±0.012	0.527±0.011
$\eta_L' / \text{Pa}\cdot\text{s}$	1.472±0.012	1.237±0.013	0.906±0.017	0.592±0.012	0.353±0.007

349 To investigate the change in elastic modulus and instantaneous viscosity of materials under  
 350 shear, Ewoldt et al. (2008) proposed to use the ratios of strain-stiffening ( $S$ ) (Eq. (6)) and shear-  
 351 thickening ( $T$ ) (Eq. (7)) to further understand if it is strain-stiffening ( $S > 0$ ) or softening ( $S < 0$ )  
 352 and if it is shear-thickening ( $T > 0$ ) or thinning ( $T < 0$ ) under nonlinear intracycle shearing  
 353 (Ewoldt et al., 2008; Hyun et al., 2011; Carmona et al., 2014; Zhang et al., 2019).

$$354 \quad S = (G'_L - G'_M) / G'_L \quad (6)$$

$$355 \quad T = (\eta'_L - \eta'_M) / \eta'_L \quad (7)$$

356 As shown in Fig. 5, all the  $S$  values for WMS paste was a positive value and increased with  
 357 increasing strain, showing strong shear-stiffening behavior. During dynamic shearing, the  
 358 network structure of WMS paste was in the reversible deformation without destruction in the  
 359 LVR, while WMS paste underwent plastic deformation when the strain exceeded the yield point.  
 360 In the case, molecular chains were aligned along the strain direction, and hydrogen-bonding  
 361 interactions occurred between starch chains, which increased the ordering and deformation-  
 362 resistance. Thus, the  $S$  curve continued to rise but the slope gradually decreased. Besides, the  
 363 interior structure of WMS paste was destroyed and reorganized constantly to form more stable  
 364 and a strong network structure under shearing, further contributing to stress-hardening. On the  
 365 other hand, the  $T$  value for WMS paste increased first and then decreased and  $T$  was a positive  
 366 value only at 63.1% strain, suggesting WMS paste showed intracycle shear-thickening only at



367 63.1% strain but shear-thinning under other strain conditions. Compared with other starch  
368 samples, the S value for starch 35 was the maximum at 63.1% strain due to its relatively high  
369 content of amylopectin and long-branched chains, in agreement with the strain sweep results.  
370 Regarding this phenomenon, loose and irregular long branched chains had a shorter relaxation  
371 time than main chains so that it had enough time to move and re-entangle, resulting in increased  
372 viscosity and shear-thickening. With a continuous increase in strain, main chains and branched  
373 chains in WMS paste were rearranged orientationally under shearing and the entanglements  
374 between starch chains weakened, leading to decreased viscosity with shear-thinning behavior.  
375

376

377 Fig. 5 The S and T curves of WMS at different growth periods.

378 The total stress obtained in oscillating shear testing can be decomposed into two parts  
379 (elastic stress and viscous stress). Cho et al. used Chebyshev polynomials to perform the  
380 orthogonal decomposition of total stress (Eq. (8) and (9)), where the Chebyshev coefficient  
381 nicely explains the contribution of higher harmonics and is closely related to the elastic and

382 viscous stresses in nonlinear rheology (Cho, Hyun, Ahn, & Lee, 2005). The strain-  
 383 stiffening/softening and shear-thickening/thinning are judged by the sign of the third-order  
 384 Chebyshev coefficients, and  $e_3 > 0$  and  $v_3 > 0$  indicate intracycle strain-stiffening and shear-  
 385 thickening, respectively, whereas  $e_3 < 0$  and  $v_3 < 0$  correspond to intracycle strain-softening and  
 386 shear-thinning, which further clarified the motion of molecular chains under large-strain  
 387 conditions, including directional arrangement and reconstruction (Cho et al., 2005; Ewoldt et al.,  
 388 2008), and it is similar to the case shown by Fourier-transform rheology (Cho et al., 2005).

$$389 \quad \sigma' \left( \frac{\dot{\gamma}}{\dot{\gamma}_0} \right) = [\sigma(\gamma, \dot{\gamma}) - \sigma(-\gamma, \dot{\gamma})] / 2 = \gamma_0 \sum_{n=odd} e_n(\omega, \gamma_0) T_n \left( \frac{\dot{\gamma}}{\dot{\gamma}_0} \right) \quad (8)$$

$$390 \quad \sigma'' \left( \frac{\dot{\gamma}}{\dot{\gamma}_0} \right) = [\sigma(\gamma, \dot{\gamma}) - \sigma(\gamma, -\dot{\gamma})] / 2 = \dot{\gamma}_0 \sum_{n=odd} v_n(\omega, \gamma_0) T_n \left( \frac{\dot{\gamma}}{\dot{\gamma}_0} \right) \quad (9)$$

391 Where  $\sigma'$  and  $\sigma''$  represent elastic stress and viscous stress respectively, and the  $e_n$  and  $v_n$   
 392 represent the Chebyshev coefficients at Nth order,  $T_n$  represents the Chebyshev functions.

393 Fig. 6 provides that  $e_3$  was a positive value for all the samples, showing strong shear-  
 394 stiffening behavior. And,  $v_3$  was a positive value only at 63.1% strain but negative under other  
 395 strain conditions, which was consistent with the results in Fig. 5.

396

397

Fig. 6  $e_3$  and  $v_3$  curves for WMS pastes at different growth periods.

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### 399 **3.5 Motion model of WMS chains**

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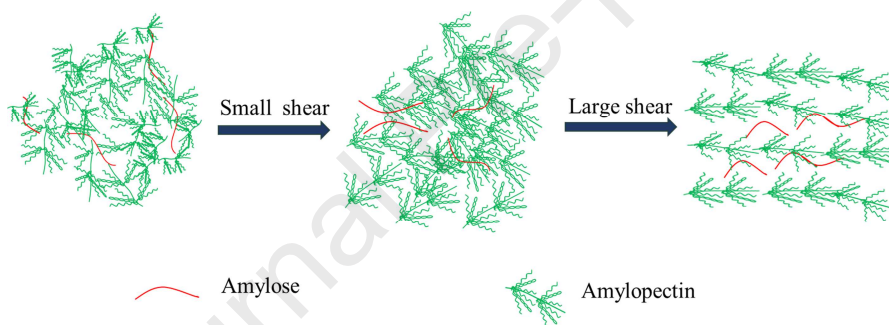
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In the steady shear testing, the apparent viscosity change of WMS paste was divided into two parts, namely the unwinding and reorientation of main chains under weak shear and the unwinding and reorientation of main chains and branched chains under intensive shear, as reflected by its strong shear-thinning behavior. In dynamic shear testing, when the strain exceeded the yield point, the WMS paste structure began to break down and reconstruct, and the test entered the NVR. Besides, from LVR to NVR, WMS paste appeared to show a weak stress overshoot and shear-thickening behavior because the long branched chains with a shorter relaxation time had enough time to re-entangle under shearing, which led to increased loss modulus. Once WMS paste was subject to strain that is large enough, main chains and branched chains of amylopectin were unwound under shearing and they changed from random coils to an orientated state following the damage and restructuring of the starch structure. This resulted in

411 the formation of a more stable and stronger network structure with intracycle strain-stiffening.  
 412 Meanwhile, the entanglement or hooking effect of starch chains was weakened, leading to  
 413 decreased instantaneous viscosity with intracycle shear-thinning behavior. Most short-branched  
 414 chains of starch 25 were not long enough to form entanglements and thus chain interactions were  
 415 restricted, resulting in decreased viscosity and moduli. When the growth period was longer than  
 416 25 d, the DP of branched chains increased and more starch chains intertwined, leading to  
 417 increased viscosity and moduli of the WMS paste. This reveals obvious nonlinearity  
 418 accompanied by a weak stress overshoot and shear-thickening behavior. Therefore, the structural  
 419 evolution of WMS chains can be clearly shown by the nonlinear rheological data.



420

421

Fig. 7 Motion model of WMS chains under shearing

## 422 4 Conclusion

423 We investigated that the amylopectin content and branched chains of WMS increased with  
 424 prolonged growth period, in which long branched chains ( $DP \geq 25$ ) improved the apparent  
 425 viscosity and dynamic moduli of the starch paste, while short branched chains did the reverse. In  
 426 the steady shear testing, compared with main chains, long branched chains underwent unwinding  
 427 and reorientation accompanied by strong shear-thinning behavior at high shear rate. In the  
 428 dynamic shear testing, long branched chains re-entangled in the strain transition region, leading  
 429 to a weak stress overshoot and shear-thickening ( $T > 0$ ). When WMS paste was subjected to

430 LAOS, it showed intracycle strain-stiffening and shear-thinning behaviors, while the nonlinearity  
431 of WMS paste became more distinct with increasing amylopectin content and branched chains  
432 length. In all, nonlinear rheology can be used as a new way to characterize the structural  
433 evolution of starch chains.

#### 434 **Conflict of interests**

435 The authors declare to have no conflict of interests.

#### 436 **Acknowledgment**

437 This research has been financially supported by the National Natural Science Foundation  
438 of China (31871751) and the Key Project of Guangzhou Science and Technology program  
439 (201804020036).

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- 555

## Highlights

- Nonlinear rheology was used in the insight into waxy maize starch chains
- Branched chains ( $DP \geq 25$ ) improved the apparent viscosity and dynamic moduli
- Amylopectin content and B-chains length increased its nonlinearity
- Motion model of WMS chains under shearing was established

**– Declaration of Interest–**

**An insight into the structural evolution of waxy maize starch chains during growth based on nonlinear rheology**

**Zipeng Liu <sup>a</sup>, Ling Chen <sup>a,\*\*</sup>, PingPing Bie <sup>a</sup>, Fengwei Xie <sup>b,c</sup>, and Bo Zheng <sup>a,\*</sup>**

*<sup>a</sup> Ministry of Education Engineering Research Center of Starch & Protein Processing, Guangdong Province Key Laboratory for Green Processing of Natural Products and Product Safety, School of Food Science and Engineering, South China University of Technology, Guangzhou 510640, China*

*<sup>b</sup> International Institute for Nanocomposites Manufacturing (IINM), WMG, University of Warwick, Coventry CV4 7AL, United Kingdom*

*<sup>c</sup> School of Chemical Engineering, The University of Queensland, Brisbane, Qld 4072, Australia*

***The authors declare that there is no conflict of interest regarding the publication of this article.***