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Citation	The 1st Symposium of International Waste Working Group - Asian Region Branch (IWWG-ARB), Sapporo, Japan, 18-21 March 2013. In the Proceedings of the 1st Symposium of International Waste Working Group - Asian Region Branch, 2013, p. article no. 8-5
Issued Date	2013
URL	http://hdl.handle.net/10722/190270
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Copper Stabilization in Beneficial Use of Waterworks Sludge and Copper-laden Electroplating Sludge for Ceramic Materials

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1. INTRODUCTION

Industrial processes such as electroplating always produce solid sludge containing significant amounts of toxic metal-laden compounds (Peng & Tian, 2010). In China, electroplating sludge discarded every year contains more than 100,000 tons of hazardous metals (Wang, 2006). A widely accepted treatment process is the stabilization/solidification (S/S) method, which can convert hazardous wastes into chemically stable solids (Sophia & Swaminathan, 2005; Chen et al., 2011). The S/S method achieves pollutant encapsulation through an interlocking framework of hydrated minerals (Sophia & Swaminathan, 2005; Zhou et al., 2006). In general, the S/S products are ultimately disposed of in secure landfills (Chen et al., 2011). Even after solidification and encapsulation, the metal-laden sludge still exhibits considerable long term metal leachability (Yousuf et al., 1995). The limited number of landfills capable of accepting S/S substances, together with their adverse environmental impacts (Malviya & Chaudhary, 2006), has made the development of effective and financially viable treatment technologies essential.

In our previous studies (Tang et al., 2010; 2011), the simulated copper-sludge were found to be able to react with the aluminum-rich precursors and form $CuAl_2O_4$ spinel phases after a 3-h sintering process. The metal leachability after thermal treatment was also significantly reduced due to the formation of spinel phases (Tang et al., 2010; 2011). As a waste-to-resource technology, the use of waste sludge resulting from water and wastewater treatment processes has attracted much attention (Babatunde & Zhao, 2007). In water treatment, for example, all processing systems generate a substantial amount of sludge with the residues of treatment chemicals used as coagulants (commonly aluminum-based), and the use of aluminum-rich sludge as raw material for ceramic products has shown promise (Vicenzi et al., 2005).

Since water treatment works in Hong Kong use aluminum-based coagulants (aluminum sulphate, aluminum chloride, poly aluminum sulphate, etc.) to facilitate particulate sedimentation. In this study, we attempted to evaluate the potential use of such waterworks sludge for immobilizing metal-laden electroplating sludge within a 3-h sintering scheme.

2. MATERIALS AND METHODS

2.1. Sample preparation

The Cu-containing electroplating sludge was collected from an industrial sludge treatment facility in Guangdong Province in China, and the collected sludge was dried at 105 °C for further sintering experiments in this study. The sludge elemental compositions detected via X-ray fluorescence spectroscopy (XRF; JEOL JSX-3201Z) was normalized by their oxide forms (Table 1) and showing copper as the predominant constituent. A waterworks sludge sample collected in Hong Kong was heated for using as an aluminum-rich ceramic precursor. The sludge was dried and fired at 900 °C for 30 min to remove the organic content and then ground into powder for elemental composition analysis by XRF. Normalization into metal oxides (Table 2) shows aluminum to be the predominant constituent. Samples for sintering experiments were prepared by mixing 900 °C calcined waterworks sludge with the dried electroplating sludge for a total dry weight of 60 g at a Cu:Al molar ratio around 1:2.5 (with excess alumina available for copper incorporation). The mixing process was carried out by ball milling the powder in water slurry for 18 h. The slurry samples were dried and homogenized by mortar grinding, pressed into 20-mm pellets at 480 MPa to ensure consistent compaction of the powder samples for the sintering process.

Components	CuO	SO_3	Al ₂ O ₃	SiO ₂	CaO	Fe ₂ O ₃	Others
Weight Percentage	64.74	19.91	4.94	3.19	3.07	2.57	0.77
(wt%)							

Table 1-Major compositions of the dehydrated copper-bearing electroplating sludge

Table 2-Ma	ior com	ositions	of incinerated	waterworks	sludge.

Components	Al ₂ O ₃	SiO ₂	P ₂ O ₅	Fe ₂ O ₃	SO ₃	K ₂ O	CaO	TiO ₂	MnO	As ₂ O ₃
Weight										
Percentage	56.53	30.16	3.81	6.17	0.55	1.28	0.48	0.31	0.66	0.05
(wt%)										

2.2. Metal incorporation

To identify the potential phases responsible for the metal incorporation mechanisms in a 3-h short sintering scheme, the pelletized powder mixture samples were fired in a high-temperature furnace (LHT 02/16 LB, LBR, Nabertherm Inc.) for a fixed 3-h dwelling time at targeted sintering temperatures ranging from 750 to 1250 °C. After sintering, the samples were air-quenched and ground into powders for XRD analysis and the leaching test. Phase transformation during sintering was monitored using the powder XRD technique. The step-scanned XRD pattern of each powder sample was recorded by a Bruker D8 Advance X-ray powder diffractometer equipped with Cu K $\alpha_{1,2}$ X-ray radiation source (40 kV, 40 mA) and a LynxEye detector. The 2 θ scanning range was 10 to 90°, and the step size was 0.02° with a scan speed of 0.8 s/step. Qualitative phase identification was executed by matching powder XRD patterns with those retrieved from the standard powder diffraction database of the International Centre for Diffraction Data (ICDD PDF-2, Release 2008).

2.3. Leaching procedure

A prolonged leaching test was employed in this study to further the discussion of the long-term leachability and leaching behavior of waste materials. As the goal of this test was to distinguish the leaching characteristics of different copper-bearing phases, single-phase samples were considered preferable in the leaching experiment. Therefore, after identifying the potential copper-containing phase(s) in the sintered products, single-phase copper samples were further fabricated by extending the sintering time. In the modified leaching procedure, 10 mL of pH 2.9 acetic acid solution was used as the leaching fluid for 0.5 g of sample powder in each leaching vial. The leaching vials were rotated end-over-end at 60 rpm for agitation periods of 0.75 to 22 days. At the end of each agitation period, the leachates were filtered with 0.2 μ m syringe filters. Their pH values were measured with a pH meter (Thermo Scientific), and their metal concentrations with an ICP-AES (Perkin-Elmer Optima 3300 DV).

3. RESULTS AND DISCUSSION

3.1. Characterization of raw materials

The XRD pattern of the 105 °C dried electroplating sludge in Figure 1 shows that the predominant crystalline Cu-containing phase was detected to be posnjakite $(Cu_4(SO_4)(OH)_6(H_2O), PDF\#83-1410)$. Other Cu-containing crystalline phases were identified as malachite $(Cu_2(OH)_2CO_3, PDF\#76-0660)$ and copper oxide (CuO, PDF#80-1268). The XRD pattern in Figure 2 further indicates that the aluminum component may exist as mullite and poor-crystalline phase(s) in the calcined sludge. A strong quartz signal was also found, which may reflect the second-largest component (silicon) detected by XRF.



Figure 1. XRD pattern of the dehydrated copper-bearing electroplating sludge. The crystalline phases are identified as: **①** posnjakite $(Cu_4(SO_4)(OH)_6(H_2O), PDF#83-1410)$; **②** gypsum $(Ca(SO_4)(H_2O)_2, PDF#70-0982)$; **③** malachite $(Cu_2(OH)_2CO_3, PDF#76-0660)$; **④** silicon oxide $(SiO_2, PDF#82-1569)$; **⑤** copper oxide (CuO, PDF#80-1268); **⑥** magnetite $(Fe_3O_4, PDF#0866)$.



Figure 2. XRD pattern of waterworks sludge calcined at 900 °C for 30 min. The standard patterns retrieved from the ICDD database include quartz (SiO₂,

PDF#79-1910) and mullite (3Al₂O₃·2SiO₂, PDF#79-1455).

3.2. Incorporation of copper during sintering process

XRD patterns in Figure 3 demonstrate the phase transformation at temperatures ranging from 650 to 1150 °C in a 3-h sintering scheme. The peaks of CuAl₂O₄ phase were first observed in the 650 °C sintered sample. The increased peak intensity was found in the sample sintered at 850 °C for 3 h. However, the substantial development of the CuAl₂O₄ spinel phase occurred when the sample was sintered at 1000 °C. The signal of CuAl₂O₄ spinel phase continued until the temperature reached 1150 °C. Further heating will cause the generation of the other copper-containing phases, i.e. CuAlO₂, Cu₂O and CuO. The decomposition of CuAl₂O₄ spinel at higher temperatures can be organized in the following way (Tang et al., 2010):

$$2\mathrm{CuAl}_{2}\mathrm{O}_{4} \rightarrow 2\mathrm{CuAl}_{2}\mathrm{O}_{2} + \mathrm{Al}_{2}\mathrm{O}_{3} + 0.5\mathrm{O}_{2} \tag{1}$$

$$2\mathrm{CuO} \rightarrow \mathrm{Cu}_2\mathrm{O} + 0.5\mathrm{O}_2 \tag{2}$$

$$Cu_2O + Al_2O_3 \rightarrow 2CuAlO_2 \tag{3}$$

When the transformation of aluminum and silicon was considered, at temperatures lower than the decomposition temperature of the CuAl₂O₄ spinel, the excess aluminum and silicon existed as mullite, together with the calcium silicate phase. But at higher temperatures, due to the decomposition of the CuAl₂O₄ spinel structure the excess aluminum existed as α -Al₂O₃, which was different from the regeneration of mullite phase when sintering CuO with kaolinite and mullite precursors (Tang et al., 2011).

According to the CuAl₂O₄ XRD pattern database (PDF#78-1605), its major peaks are located at $2\theta = 36.868$, corresponding to the diffraction planes of (3 1 1). Therefore, the 2θ range of $36.5-37.2^{\circ}$ was selected to further observe the variation of CuAl₂O₄ spinel with the elevated temperatures (Figure 4). The substantial increase of CuAl₂O₄ peak intensity occurred at 950 °C, and similar peak intensities within the temperature range of 950-1050 °C were observed. The decomposition of CuAl₂O₄ spinel was observed with the decrease of CuAl₂O₄ peak intensity due to the reaction shown in Eqs. (1-3).



Figure 3. XRD patterns of the Cu-electroplating sludge + waterworks sludge system sintered at 650 to 1150 °C for 3 h. The crystalline phases are identified as: CuO (PDF#80-1268), CuAl₂O₄ (PDF#78-1605), CaCu₂O₃ (PDF#34-0284), Cu₂O (PDF#77-0199), CuAlO₂ (PDF#75-2356), α -Al₂O₃ (PDF#10-0173), anhydrite (CaSO₄, PDF#37-1496), CaSO₄·2H₂O (PDF#70-0982) mullite (PDF#79-1455), cristobalite (PDF#76-0938), quartz (PDF#79-1910), and calcium silicate (CaSiO₄, PDF#33-0303).



Figure 4. Comparison of the growth of $CuAl_2O_4$ spinel phase by XRD patterns in 2θ between 36.6° and 37.2° for the Cu-electroplating sludge + waterworks sludge samples sintered at 650-1250 °C for 3 h.

3.3. Leachability of copper in product phase(s)

From section 3.2, the CuAl₂O₄ spinel was the predominant copper-containing product phase when sintering copper-sludge with waterworks sludge. Therefore, single-phase CuAl₂O₄ was prepared for the prolonged leaching experiments. To ensure the complete transformation into the CuAl₂O₄ without residual reactants, the Cu/Al molar ratio = 1:2 mixture pellets were sintered at 990 °C for 20 d and ball milled into powder form. The leachability of CuO, as the mainly existed form in copper-laden sludge heating without any addictives, was compared with that of CuAl₂O₄ product phase. When leaching CuO and CuAl₂O₄ samples in an acidic environment, the reaction processes can be described as:

$$CuO + 2H^+ \rightarrow Cu^{2+} + H_2O \tag{4}$$

$$CuAl_2O_4 + 8H^+ \rightarrow Cu^{2+} + 2Al^{3+} + 4H_2O$$
 (5)

The copper concentrations in both the CuO and $CuAl_2O_4$ leachates are presented in Figure 5. The leached copper in CuO leachate reached around 2000 mg/L. However, throughout the entire leaching process, the concentration of the leached copper from the CuAl_2O_4 was significantly lower than that of the CuO leachate at the end of

leaching experiment. Figure 5 and its inset show that the leached copper from the CuO sample is two orders of magnitude greater than that from the $CuAl_2O_4$ sample. The result of prolonged leaching test favorably confirmed the good stabilization achieved through incorporating copper into a $CuAl_2O_4$ structure.



Figure 5. Variation of copper concentrations in the leachates from CuO and $CuAl_2O_4$ powder samples. The leaching solutions were acetic acid solution at pH 2.9. Each leaching vial was filled with 10 mL of extraction fluid and 0.5 g of powder, and then tumbled end-over-end from 0.75 to 22 d.

4. CONCLUSIONS

When sintering the mixtures of copper-electroplating sludge and waterworks sludge precursors, the peaks of the CuAl₂O₄ phase were clearly identified, together with the disappearance of the initial copper-hosting phases (posnjakite, malachite) in the starting material. Sintering the sample mixture at 850 °C shows a distinguishable CuAl₂O₄ peak, and a substantial peak development was found at 950 °C. The maximum generation of the CuAl₂O₄ spinel was observed at around 1000 °C, and the CuAlO₂ phase was observed at temperatures higher than 1100 °C. The leaching results show explicitly that the CuAl₂O₄ product phase is superior to the copper oxide phase for immobilizing hazardous copper over longer leaching periods.

As the waterworks sludge is able to initiate the crystallization of the $CuAl_2O_4$ at an attainable sintering temperature (~1000 °C), the incorporation of copper-sludge into the fabrication process of some common ceramic products, such as those used for

construction purposes, may be an effective and economical strategy to reduce the environmental hazard of copper-laden sludge.

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