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Thorium-doping-induced superconductivity up to 56 K in Gd1-xThxFeAsO

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Thorium-doping–induced superconductivity up to 56 K in $Gd_{1-x}Th_xFeAsO$

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PACS 74.62.Dh – Effects of crystal defects, doping and substitution
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Abstract – We report a new strategy to induce superconductivity in iron-based oxyarsenide. Instead of F^- substitution for O^{2-}, we employed Th^{4+} doping in GdFeAsO with the consideration of "lattice match" between Gd_2O_2 layers and Fe_2As_2 ones. As a result, superconductivity with T_c^{onset} as high as 56 K was realized in a $Gd_{0.8}Th_{0.2}FeAsO$ polycrystalline sample. This T_c value is among the highest ever discovered in the iron-based oxypnictides.

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Introduction. – Following the discovery of superconductivity in an iron-based arsenide $LaO_{1-x}F_xFeAs$ with a superconducting transition temperature (T_c) of 26 K [1], T_c was pushed up surprisingly to above 40 K by either applying pressure [2] or replacing La with Sm [3], Ce [4], Nd [5] and Pr [6]. The maximum T_c^{onset} has climbed to 55 K, observed in $SmO_{1-x}F_xFeAs$ [7,8] and $SmFeAsO_{1-x}$ [9]. The value of T_c was found to increase with decreasing lattice parameters in $LnFeAsO_{1-x}F_x$ (Ln stands for the lanthanide elements) at an apparently optimal doping level. However, the F^- doping in GdFeAsO (whose lattice constants are the smallest) produced T_c only below 40 K [10,11].

The LnFeAsO family [12] crystallizes in a tetragonal ZrCuSiAs-type [13] structure with space group *P4/nmm*. From the crystal chemistry point of view, the crystal structure can be described as an alternate stacking of Ln_2O_2 fluorite-type block layers and Fe_2As_2 antifluorite-type block layers along the *c*-axis. The two block layers are connected by CsCl-type layers (see fig. 1). Therefore, the chemical stability of LnFeAsO depends, to some extent, on the lattice match between the two block layers. A rough estimate based on the effective ionic radii [14] indicates that the Ln_2O_2 planar lattice is substantially smaller than the Fe_2As_2 lattice. The layer mismatch becomes more significant for the LnFeAsO member with a smaller Ln^{3+} ion. This may explain why TbFeAsO, DyFeAsO and other heavy-lanthanide–containing LnFeAsO members were hard to synthesize previously [12].

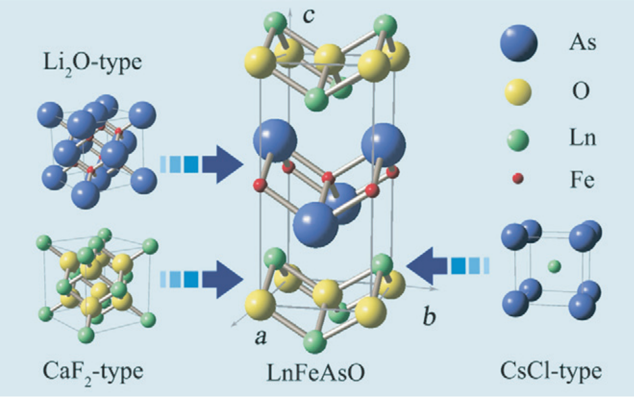

Fig. 1: Crystal chemistry understanding of the structure of LnFeAsO (Ln = lanthanides). The stacking of fluorite (CaF_2) layers, CsCl-type layers and antifluorite (Li_2O) layers along the *c*-axis forms the LnFeAsO structure. The lattice constant along the stacking direction can be expressed by the formula $c \simeq \frac{1}{2}a_{CaF_2} + \frac{1}{2}a_{CsCl} + \frac{1}{2}a_{Li_2O}$, which basically satisfies the experimental results. Note that the lattice match between the Ln_2O_2 layers and the Fe_2As_2 layers affects the chemical stability of LnFeAsO.

As the family member with the relatively small Ln^{3+} ion, GdFeAsO is a promising parent compound to have a

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higher T_c through carrier doping. Similar to cuprate superconductors in which superconductivity emerges when charge carriers are induced into CuO_2 planes by chemical doping at "charge reservoir layers" [15], superconductivity in $LnFeAsO_{1-x}F_x$ is realized by partial substitution of O^{2-} with F^-. The F^--for-O^{2-} substitution introduces extra positive charges in the insulating Ln_2O_2 layers and negative charges (electron doping) in the Fe_2As_2 layers. An earlier preliminary experiment showed signs of superconductivity below 10 K in $GdFeAsO_{1-x}F_x$ [10]. Later, T_c was pushed up to 36 K in a polycrystalline sample with a nominal composition of $GdO_{0.83}F_{0.17}FeAs$ [11]. Very recently the T_c value was increased to 53.5 K in oxygen-deficient $GdFeAsO_{1-x}$ by using high-pressure synthesis [16]. It is of great interest whether the T_c can be elevated further in electron-doped GdFeAsO systems.

Up to now, electron doping in the iron-based oxyarsenides was realized through the chemical substitution only at the oxygen site in Ln_2O_2 layers (it is here noted that hole doping in $La_{1-x}Sr_xFeAsO$ system, which was reported to show superconductivity at 25 K, was on the La-site [17]). Because the ionic radius of F^- $(1.31\,\text{Å}, \text{CN} = 4)$ is distinctly smaller than that of O^{2-} $(1.38\,\text{Å}, \text{CN} = 4)$ [14], F^- substitution for O^{2-} in GdFeAsO leads to more severe lattice mismatch as mentioned above. In other words, doping F^- (or oxygen vacancy) in GdFeAsO is particularly difficult, which is probably the main obstacle to elevate T_c. Substitution of Ln^{3+} by relatively large tetravalence ions is an alternative route to introduce electrons. A successful example was the electron doping in $Ln_{2-x}Ce_xCuO_4$ (Ln = Pr, Nd or Sm), which has led to the discovery of *n*-type cuprate superconductors [18]. Th^{4+} is a very stable tetravalence ions and is as large as Gd^{3+} [14], therefore, we pursued the Th^{4+} substitution for Gd^{3+} in the GdFeAsO system.

Experimental. – Polycrystalline samples of $Gd_{1-x}Th_xFeAsO$ were synthesized by a solid state reaction in an evacuated quartz tube. All the starting materials (Gd, Gd_2O_3, ThO_2, Fe and As) are with high purity ($\geqslant 99.95\%$). First, GdAs was presynthesized by reacting Gd tapes with As pieces in vacuum at 773 K for 10 hours and then at 1173 K for 12 hours. Similarly, FeAs was prepared by reacting Fe powders with As shots at 773 K for 6 hours and then at 1030 K for 12 hours. Then, powders of GdAs, Gd_2O_3, ThO_2, Fe and FeAs were weighed according to the stoichiometric ratio of $Gd_{1-x}Th_xFeAsO$. The weighed powders were mixed thoroughly by grinding, and pressed into pellets under a pressure of $4000\,\text{kg/cm}^2$ in an argon-filled glove box. The pressed pellets were wrapped with Ta foils, and sealed in an evacuated quartz ampoule. The sealed ampoule was slowly heated to 1423 K, holding for 48 hours. Finally the samples were rapidly cooled to room temperature.

Powder X-ray diffraction was performed at room temperature using a D/Max-rA diffractometer with Cu K_α radiation and a graphite monochromator. The XRD diffractometer system was calibrated using standard Si powder. Lattice parameters were refined by a least-squares fit using at least 20 XRD peaks. Energy-dispersive X-ray (EDX) spectra were obtained by using the Phoenix EDAX equipment attached to a field-emission scanning electron microscope (SIRION FEI). The ground sample powder was placed directly on the copper sample holder for making the SEM specimen.

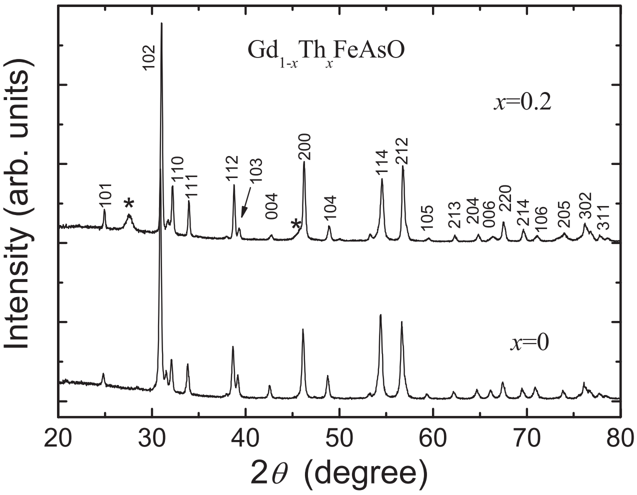

Fig. 2: Powder XRD of $Gd_{1-x}Th_xFeAsO$ polycrystalline samples. The asterisked peaks come from unreacted ThO_2.

The electrical resistivity was measured with a standard four-terminal method. Samples were cut into a thin bar with typical size of $4\,\text{mm} \times 2\,\text{mm} \times 0.5\,\text{mm}$. Gold wires were attached onto the samples' newly abraded surface with silver paint. The size of the contact pads leads to a total uncertainty in the absolute values of resistivity of 10%. The electrical resistance was measured using a steady current of 5 mA, after checking the linear *I*-*V* characteristic.

Temperature dependence of magnetization was measured on a Quantum Design Magnetic Property Measurement System (MPMS-5). For the measurement of the undoped compound, the applied field was 1000 Oe. For the measurement of the Th-doped superconducting samples, both the zero-field-cooling and field-cooling protocols were employed under the field of 10 Oe.

Results and discussion. – Figure 2 shows the X-ray diffraction (XRD) patterns of the $Gd_{1-x}Th_xFeAsO$ samples. The XRD peaks of the undoped compound can be well indexed based on the tetragonal ZrCuSiAs-type structure, indicating single phase of GdFeAsO. As for the Th-doped samples, small amount of unreacted ThO_2 can be seen. The refined lattice parameters are $a = 3.9154(2)\,\text{Å}$ and $c = 8.4472(4)\,\text{Å}$ for the parent compound, basically consistent with the previously reported values [12]. For the Th-doped sample of $x = 0.2$, the fitted lattice parameters are $a = 3.9161(2)\,\text{Å}$ and $c = 8.4386(3)\,\text{Å}$. Therefore,

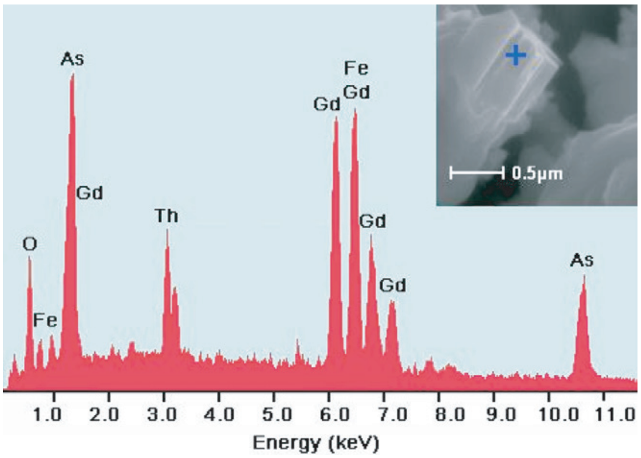

Fig. 3: A representative EDX spectrum of a crystalline grain of Gd$_{0.8}$Th$_{0.2}$FeAsO sample, indicating the incorporation of thorium into the lattice. The inset shows the SEM image of the same sample. The marked spot is the position where the EDX spectrum was collected.

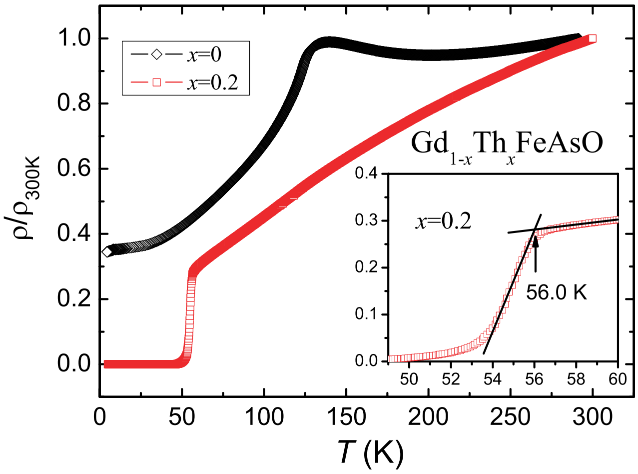

Fig. 4: Temperature dependence of electrical resistivity (ρ) in Gd$_{1-x}$Th$_x$FeAsO. The data are normalized to $\rho_{300\,\mathrm{K}}$ as the resistivity measured on polycrystalline samples is often higher than the intrinsic value due to the grain boundary and surface effect. The inset is an expanded plot to show the superconducting transition at 56 K for the sample of $x = 0.2$.

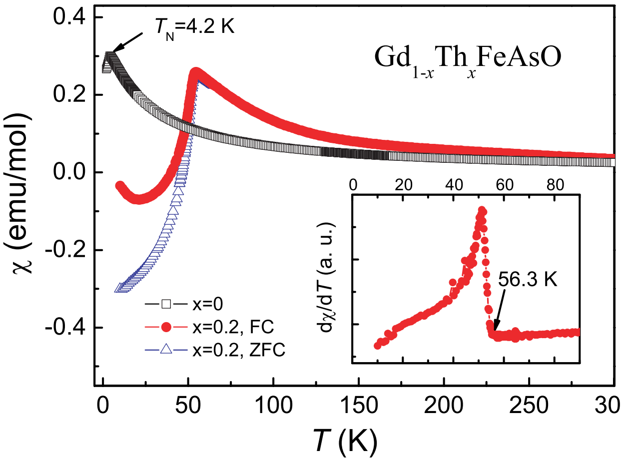

Fig. 5: Temperature dependence of magnetic susceptibility (χ) of Gd$_{1-x}$Th$_x$FeAsO. The lower inset shows the differential χ_{FC} curve of Gd$_{0.8}$Th$_{0.2}$FeAsO powder sample, which locates the T_c^{onset} at 56.3 K. ZFC, zero-field cooling; FC, field cooling.

though the a-axis almost remains unchanged, the c-axis is shorten significantly by the Th-doping, indicating that thorium is indeed incorporated into the lattice. The shrinkage of the c-axis is attributed to the strengthening of the interlayer Coulomb attraction as a consequence of Th^{4+} doping.

More direct evidence of Th incorporation into the lattice comes from the chemical composition measurement by energy-dispersive X-ray (EDX) microanalysis. Figure 3 shows that the microcrystal in the SEM image contains remarkably Th in addition to Gd, O, Fe and As. Quantitative analysis gives the Gd : Th : Fe : As ratios as 0.83 : 0.19 : 0.96 : 1.00 for the $x = 0.2$ sample (here we omit the oxygen content because the amount of oxygen cannot be measured so precisely by EDX technique). This result demonstrates that most of the Th was successfully doped for the sample of $x = 0.2$.

Figure 4 shows the $\rho(T)$ curve for Gd$_{1-x}$Th$_x$FeAsO samples. For the undoped GdFeAsO, the $\rho(T)$ curve exhibits an obvious anomaly at 135 K, characterized by a resistivity drop with decreasing temperature. In LaFeAsO a similar resistivity anomaly was found at 150 K [1,19] which has been recently suggested to be associated with a structural phase transition and/or an antiferromagnetic spin-density-wave transition [19–21]. We speculate that the present resistivity anomaly in GdFeAsO has a similar physical origin with that in LaFeAsO. For $x = 0.2$ such resistivity anomaly disappears, instead, resistivity drops abruptly to zero below 56 K, indicating a superconducting transition. In addition, the linear temperature-dependence of normal-state resistivity near T_c suggests possible non-Fermi liquid behavior in the present system, similar to that in SmFeAsO$_{1-x}$F$_x$ [8].

The magnetic measurement on Gd$_{1-x}$Th$_x$FeAsO samples was shown in fig. 5. The high-temperature magnetic susceptibility (χ) of the parent compound follows the Curie-Weiss law. The fitted effective magnetic moment was 7.96 μ_B per formula unit, in good agreement with the magnetic moment of a free Gd^{3+} ion. Below 4.2 K, χ drops sharply, indicating an antiferromagnetic ordering of Gd^{3+} magnetic moments. Note that similar behavior was observed in SmFeAsO, where the Neel temperature was 4.6 K [22].

For the Th-doped samples, the normal-state susceptibility roughly obeys the Curie-Weiss law. Below 55 K, χ drops steeply to negative values, confirming the superconductivity observed above. After subtracting the paramagnetic susceptibility of Gd^{3+} ions, the volume fraction of magnetic shielding at 10 K was estimated to be over 50%, indicating bulk superconductivity. The differential χ_{FC} curve in the inset of fig. 5 shows that the T_c^{onset} is over 56 K, consistent with the resistance measurement. Therefore, Th-doping in GdFeAsO elevates the T_c by 20 K and 2.5 K, respectively, compared with $GdO_{0.83}F_{0.17}FeAs$ [11] and high-pressure synthesized $GdFeAsO_{1-x}$ [16]. To our knowledge, this value of T_c^{onset} is among the highest ever discovered in iron-based oxypnictides.

Our observation of superconductivity in $Gd_{1-x}Th_xFeAsO$ indicates that the Ln-site substitution in LnFeAsO is feasible to realize electron doping, hence the high-temperature superconductivity. Moreover, the electron-doping is more easily induced by the Th^{4+}-substitution compared with the F^--substitution in GdFeAsO. It is thus expected that the thorium-doping strategy can be applied to other iron-based oxypnictides.

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