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# Improved $J_c$ of MgB<sub>2</sub> superconductor by ball milling using different media

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## Abstract

In this paper, the effects of ball milling B powders using different media, such as acetone, ethanol, and toluene on the superconducting properties of MgB<sub>2</sub> have been studied. It was observed that toluene medium was the most effective of them all for enhancing  $J_c$ .  $J_c$  was estimated to be  $5 \times 10^3$  A/cm<sup>2</sup> at 8 T and 5 K. This value is much higher than that of pure MgB<sub>2</sub> that was not ball milled, by a factor of 20. It was considered that ball-milled B using toluene leads to smaller MgB<sub>2</sub> grains, resulting in enhanced  $J_c$  at low operating temperature and high field.

(Some figures in this article are in color only in the electronic version)

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## 1. Introduction

The discovery of the superconductivity of MgB<sub>2</sub> with a critical temperature of 39 K has offered the promise of important large-scale and electronic device applications at around 20 K [1]. A significant enhancement in the  $J_c$  of MgB<sub>2</sub> has been achieved through chemical doping with carbon (C) containing compounds, such as SiC, C, B<sub>4</sub>C, and CNT. However, doping effects have been limited by the agglomeration of nanosized dopants and poor reactivity between boron (B) and C [2-7]. Moreover, the self-field and low-field  $J_c$  were depressed due to the decrease in the superconducting volume. To improve these properties, various methods have been reported. They include, for example, a ball-milling method [8,9], a thermo-mechanical processing method [10], and the use of MgH<sub>2</sub> powder instead of Mg [11]. Among them, the ball-milling method is particularly interesting, as size control of the starting material may induce more effective pinning without any dopants. Specifically, the properties of the B powder, such as purity and size, may play an important role in determining the final properties of the MgB<sub>2</sub> [12]. This is because melting Mg can be diffused into B powder to form MgB<sub>2</sub> phases during sintering.

Fischer et al. fabricated MgB<sub>2</sub> wires and bulks by a mechanical alloying method [10]. They obtained one of the highest  $J_c$  values without any other elements added. The very fine-grained nanocrystalline microstructure of the superconducting phase seems to be responsible for the excellent  $J_c$  values ( $1 \times 10^5$  A/cm<sup>2</sup> at 2.1 T and 20 K). Fang et al. fabricated MgB<sub>2</sub>/Fe tapes via the powder-in-tube method using an ultra-fine Mg and B precursor mixture prepared by high energy ball milling. The  $J_c$  was estimated to be  $2.0 \times 10^5$  A/cm<sup>2</sup> at 20 K and 0.6 T, and  $1.1 \times 10^5$  A/cm<sup>2</sup> at 1.5 T [8]. Kondo et al. reported that the magnetic  $J_c$  of MgB<sub>2</sub> depended on the different insert gases, Ar and H<sub>2</sub>, that were used during ball milling [9]. However, there was no systematic study of the effects of different ball-milling media, such as acetone (C<sub>3</sub>H<sub>6</sub>O), ethanol (C<sub>2</sub>H<sub>6</sub>O), and toluene (C<sub>7</sub>H<sub>8</sub>). It is these liquid media that help to make mixing homogeneous. On the other hand, these media consist of carbon (C), oxygen (O), and hydrogen (H), so that the O can easily react with Mg and B to form MgO and B<sub>2</sub>O<sub>3</sub> during the drying process after ball milling. In our

study, therefore, we evaluated the effects of ball-milled B powders that were produced using different media, such as acetone, ethanol, and toluene, on the superconducting properties of MgB<sub>2</sub>. The lattice parameters,  $J_c$ ,  $T_c$ , and microstructure are presented in comparison with reference pure MgB<sub>2</sub>.

## 2. Experiment details

MgB<sub>2</sub> pellets were prepared by an *in-situ* reaction process. B powders (99%) were prepared in different ball-milling media, such as acetone, ethanol, and toluene (with the samples named after the corresponding medium). The ball-milling process was carried out for 4 hr with a rotating speed of 160 rpm under air. These three kinds of B powders were mixed, ground, and pressed with Mg (99%) powder. All samples were sintered at 650°C for 30 min under high purity argon gas. The heating rate was 5°C/min. A pure reference MgB<sub>2</sub> sample (referred to as pure) was also fabricated for comparison by applying the same process, except for the ball milling. All samples were characterized by X-ray diffraction (XRD) and scanning electron microscopy (SEM). The crystal structure was refined with the aid of the program FullProf.  $T_c$  was defined as the onset temperature at which diamagnetic properties were observed. The magnetization was measured at 5 and 20 K using a Physical Property Measurement System (PPMS, Quantum Design) in a time-varying magnetic field with sweep rate 50 Oe/s and amplitude 8.5 T. Since there is a large sample size effect on the magnetic  $J_c$  for MgB<sub>2</sub>, all the samples for measurement were made to the same size (1 x 2 x 3 mm<sup>3</sup>) for comparison. The magnetic  $J_c$  was derived from the width of the magnetization loop using Bean's model [13].  $J_c$  versus magnetic field was measured up to 8.5 T.

## 3. Results and discussion

Table 1 shows the data measured for the pure reference MgB<sub>2</sub> and the ball milled MgB<sub>2</sub> samples using different media. The  $a$  and  $c$  lattice constants for all samples did not change within the limit of calculation error, suggesting that C could not substitute into B sites during ball milling. Even

though the hydrocarbon media used contained C, there were no apparent substitution effects. However, it is to be noted that  $T_c$  for all ball-milled samples was a little decreased. In addition, the  $a$  lattice parameter for the toluene sample is shorter than that of the other ball-milled samples. This is probably because the toluene sample has a low  $T_c$  value. This low  $T_c$  value might be one of the possible reasons for the low crystallinity of ball milling sample. These observations were further explained by the FWHM values and refined grain sizes, to be discussed later.

The XRD patterns for all samples are shown in Figure 1. The XRD measurements were performed on the ground  $\text{MgB}_2$  pellets. It was observed that all samples seemed to be well developed  $\text{MgB}_2$  with small amounts of  $\text{MgO}$ . There were no visible differences in this respect between the samples. From the XRD results, the FWHMs of the (100), (002), and (110) peaks were evaluated as listed in Table 1. It is to be noted that the FWHMs of the 3 main peak positions for the toluene sample were much larger. This significant broadening of the (002) and (110) peaks can be explained by depression of crystallinity due to ball milling. According to the Scherrer equation, an increased FWHM value also indicates that the grain size has decreased as follows:

$$D_p = 0.94\lambda / \beta_{1/2} \cos \theta$$

where  $D_p$  is crystal size,  $\lambda$  is wavelength,  $\beta_{1/2}$  is FWHM, and  $\theta$  is peak position. From this equation, crystal size is easily calculated as a function of peak width (FWHM). This approach does, however, neglect the effect that strain can have on crystallite size [14]. As a result, broadening FWHM values may be due to ball milling with different media. Yamada *et al.* reported that a small  $\text{MgB}_2$  grain size was effective in enhancing flux pinning because the grain boundaries of  $\text{MgB}_2$  represented effective pinning centers, as in the case of A15 metallic superconductors [15].

Figure 2 shows SEM images for all samples sintered at  $650^\circ\text{C}$  for 30 min. It was observed that the specimens consisted of quite well consolidated material. The grain size of the material is quite small, around 100-300 nm. Specifically, the grain sizes of the ethanol and acetone samples are

larger than for the pure reference sample (~200 nm), while the grain size of the toluene sample was slightly smaller (~100 nm). It is to be noted that the reference sample has better grain homogeneity, but the toluene sample seems to have enhanced clumping of fine particles into coral-like agglomerates due to ball-milling effects. The different behaviours of  $J_c(B)$  and  $T_c$  can be attributed to this microstructural difference. Since melting Mg is diffused into B powder to form  $MgB_2$  phases during sintering,  $J_c(B)$  properties can be related to the B powder properties.

Figure 3 shows the magnetic field dependence of  $J_c$  for all samples at 5 and 20 K. We observe that the  $J_c$  value of toluene sample was estimated to be  $5 \times 10^3 A/cm^2$  at 8T and 5K. This value is comparable to those of chemically doped samples under high field. The  $J_c$  value is much higher than that of the pure  $MgB_2$  made without any ball-milling process, by a factor of 20. Using ball milled B with toluene as the ball-milling medium was a highly effective method to enhance the  $J_c(B)$  performance under high field. This is because toluene can prevent the oxidation of B powder during ball milling, and the small grain size is effective for enhancing flux pinning at the grain boundaries, which represent effective pinning centers. However, at 20K, the  $J_c(B)$  performance of the toluene sample value is slightly lower than that of the reference  $MgB_2$  sample. This is probably because the ball-milled sample had poor grain connectivity. This observation is further supported by the FWHM and SEM results (Table 1 and Figure 2).

The critical temperature ( $T_c$ ) for all samples is presented in Figure 4.  $T_c$  of the reference  $MgB_2$  was higher than that of the ball-milled samples. This is related to the improvement of crystallinity of the  $MgB_2$  superconductor. It is to be noted that the  $T_c$  variations of the ball-milled samples are significantly different for the different media. In addition,  $T_c$  of the toluene sample was slightly lower than for the acetone and ethanol samples. For example, the  $T_c$  value of the toluene sample was 35.7 K, while the corresponding values for the acetone and ethanol samples were 37.2 K and 36.8 K, respectively. This depressed result for the toluene sample is considered to be due to its poor crystallinity, resulting in a significant amount of strain and crystal defects in the sample after ball milling.

#### **4. Summary**

We evaluated the effects of ball-milled B powders using different media, such as acetone, ethanol, and toluene, on the superconducting properties of MgB<sub>2</sub>. It was observed that the *a* and *c* lattice parameters did not change for any of the samples, suggesting that C could not substitute into B sites during ball milling. Specifically, ball-milled B using toluene leads to smaller MgB<sub>2</sub> grains, resulting in enhanced *J<sub>c</sub>* under high field. The estimated *J<sub>c</sub>* was  $5 \times 10^3$  A/cm<sup>2</sup> at 8 T and 5 K. This value is much higher than that of the pure reference MgB<sub>2</sub>, by a factor of 20. At 20K, however, the *J<sub>c</sub>(B)* performance of the toluene sample is slightly worse than for the pure sample. This depressed value is probably related to the poor crystallinity. Further studies on the ball-milling process promise to allow adjustment of the desired phase form, such as for MgB<sub>2</sub> made with low grade boron powder via solid state reaction for future applications.

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**Fig. 2.** SEM images for the MgB<sub>2</sub> samples.

**Fig. 3.** The magnetic critical current density  $J_c$  as a function of field for the MgB<sub>2</sub> samples.

**Fig. 4.** The  $T_c$  curves for the MgB<sub>2</sub> samples.



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Table 1. X. XU et al.

Samples	FWHM (deg.)			Lattice constant		
	(100)	(110)	(002)	a(Å)	c(Å)	c/a
pure	0.4300	0.4860	0.4940	3.0848	3.5298	1.1442
Toluene	0.4640	0.6800	0.6800	3.0812	3.5244	1.1438
Ethanol	0.3620	0.4700	0.4700	3.0843	3.5292	1.1442
Acetone	0.3440	0.4560	0.4960	3.0830	3.5258	1.1436

Fig. 1. X. XU et al.

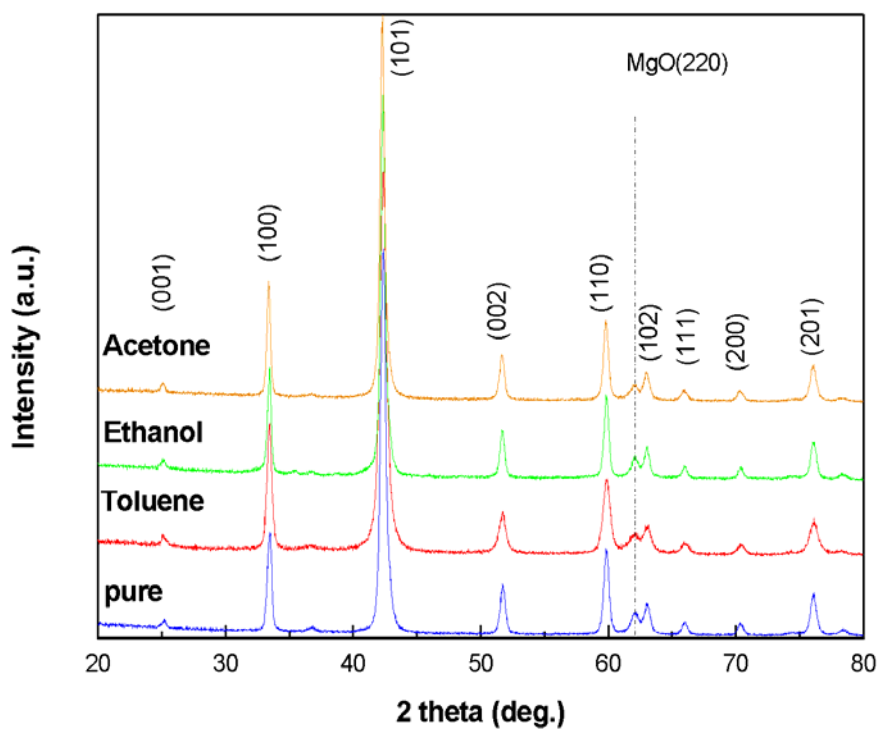


Fig 2. X. XU et al.

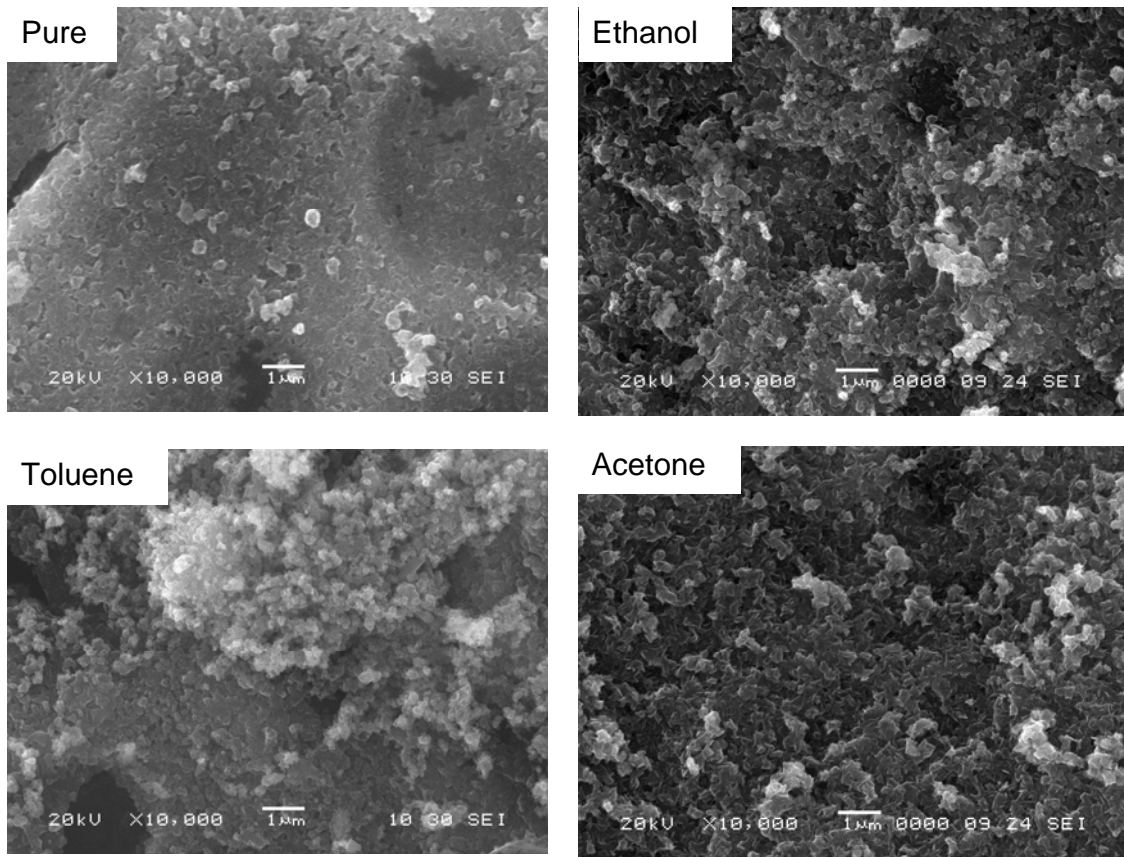


Fig 3. X. XU et al.

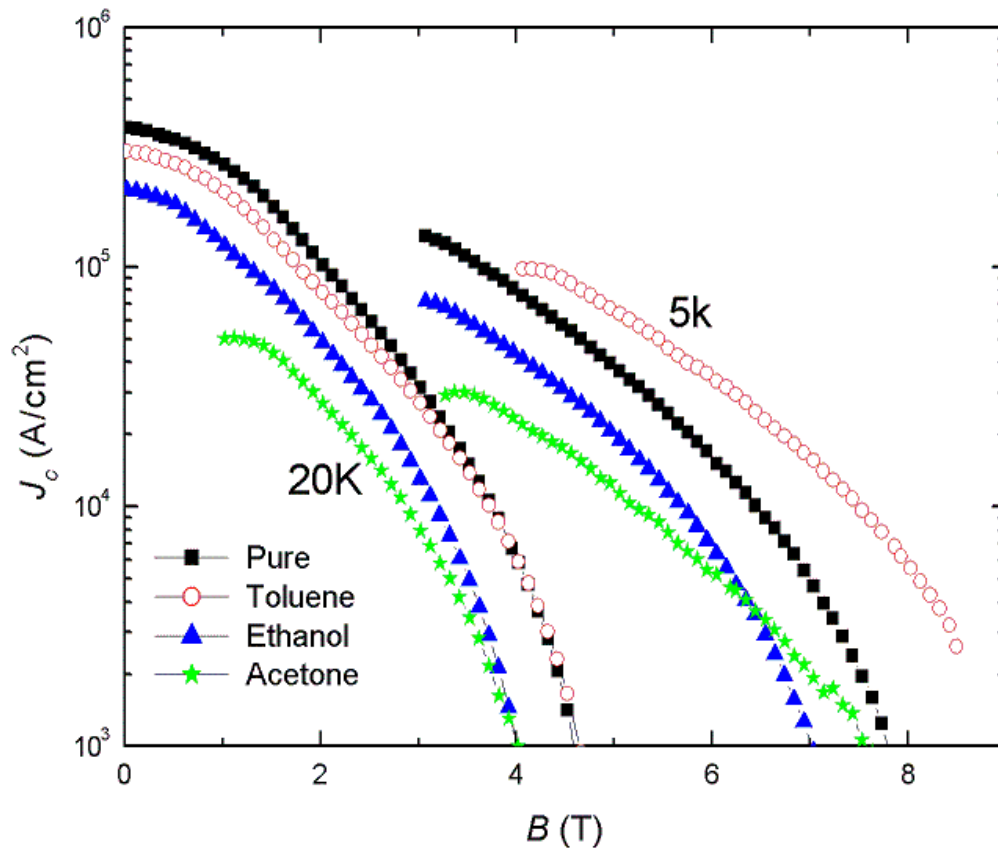


Fig 4. X. XU et al.

