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1	Declarations of interest: none
2	
3 4	The effect of melt composition and oxygen fugacity on manganese partitioning between apatite and silicate melt
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11	Abstract:
12	Oxygen fugacity and melt composition are both known to have a strong influence on the partitioning
13	of trace elements between coexisting minerals and melt. Previous work has suggested that Mn
14	partitioning between apatite and silicate melt may be strongly affected by oxygen fugacity and could,
15	therefore, act as an oxybarometer. Here, we present a new study on the partitioning of Mn between
16	apatite and melt at high temperature (1400-1250 °C) and 1 GPa pressure, for various melt
17	compositions and oxygen fugacities (NNO +4.7 to NNO -10). We find that there is no demonstrable
18	variation in the partition coefficient for Mn between apatite and silicate melt ($D_{Mn}{}^{Ap\cdot m}$) across the
19	range of fO_2 conditions studied here. Instead, we find that D_{Mn}^{Ap-m} varies significantly with melt
20	composition and that in particular, the proportion of non-bridging oxygens strongly influences
21	partitioning of Mn between apatite and melt. We propose that variations in the Mn content of natural
22	apatite, previously thought to reflect variations in fO_2 , are instead related to the degree of melt
23	polymerization. These findings are consistent with the results of Mn K-edge XANES spectroscopy,
24	which demonstrate that Mn in coexisting apatite and silicate glass is present predominantly as Mn ²⁺
25	regardless of fO_{2} . Furthermore, XANES spectra from a series of silicate glasses synthesised at various
26	oxygen fugacities demonstrate that Mn^{2+} is the predominant species, and that the average Mn
27	oxidation state does not vary over a wide range of fO_2 -T conditions.

Keywords: Apatite, Manganese, element partitioning, oxygen fugacity, experimental petrology, melt
 composition

30 Introduction:

31 Apatite [nominally Ca₅(PO₄)₃(F,Cl,OH)] is an accessory mineral found in many igneous, metamorphic, 32 and sedimentary rocks. The three primary apatite end-members (fluor-, chlor- and hydroxyapatite) 33 relate to the three anion end members of apatite (F, Cl and OH respectively). The incorporation of 34 these volatiles as major constituents in the apatite crystal structure make it a critical mineral for 35 understanding melt volatile contents in terrestrial (Douce and Roden 2006; Scott et al. 2015) and 36 extra-terrestrial systems (Gross et al. 2013; McCubbin et al. 2016). Apatite is also an important tool 37 for exploring the trace element budgets of magmas (Nagasawa 1970; Sha and Chappell 1999), as its 38 crystallographic structure allows for the incorporation of more than half the stable naturally-occurring 39 elements (Hughes 2015). Recent work (Miles et al. 2014; Konecke et al. 2017) has also suggested that 40 the substitution of redox sensitive elements (e.g. Mn, S, Ce, Eu) into apatite could be used to constrain 41 the oxygen fugacity (fO_2) of the melt from which it has crystallized, providing a much needed new 42 oxybarometer.

43 Oxygen fugacity is an important parameter within igneous systems because, among other things, it 44 controls mineral stabilities (Hensen 1986; Toplis and Carroll 1995), dictates gas species released from 45 volcanoes (Wallace and Carmichael 1992; Moussallam et al. 2016), and has implications on the metallogenesis of primary igneous ore bodies (Fleet et al. 1991). Furthermore, estimates of fO_2 are 46 47 used in modelling geophysical processes in the mantle (e.g. creep, electrical conductivity; Ryerson et 48 al. 1989; Wood and Nell 1991; Dai and Karato 2014) and understanding the timing of planetary scale 49 processes, such as the oxidation of the mantle and atmosphere (Scaillet and Gaillard 2011). Oxygen 50 fugacity is also an important variable when constraining the composition of the earth's core (Li and 51 Agee 2001) and early continental crust (Yang et al. 2014), as well as for the timing of the onset of plate 52 tectonics (Brounce et al. 2015). However, despite the importance of fO_2 it remains one of the most

53 difficult geological parameters to accurately constrain. As such, any dependence of chemical 54 substitutions in apatite on fO_2 would be invaluable in constraining planetary processes.

55 Manganese in apatite is a minor element that can substitute for more than 1 atom per formula unit 56 (a.p.f.u.) onto the Ca2 and P sites (Hughes et al. 2004; Deer et al. 2013). Several studies have indicated 57 that while partitioning of Mn between apatite and silicate melt is sensitive to oxygen fugacity, it is also 58 dependent on parameters such as melt SiO₂ and total alkali content, as well as melt structure (Sha and 59 Chappell 1999; Belousova et al. 2001; Chu et al. 2009; Miles et al. 2014). This work aims to constrain 60 the sensitivity of Mn partitioning between apatite and melt to fO_2 for a variety of silicate melt 61 compositions and to assess its viability as an oxybarometer.

- 62 1.1 Experimental rational
- 63

64 We primarily set out to test the conclusion of Miles et al. (2014) that the Mn content of apatite, from 65 a range of calc-alkaline intermediate to silicic rocks, can be directly related to fO₂ via the below 66 empirical relationship:

67
$$\log fO_2 = -0.0022 (\pm 0.0003) Mn (ppm) - 9.75 (\pm 0.46)$$
 [1]

This dependence implies that variations in fO2 influence Mn apatite-melt partitioning because of a 68 69 change in the relative proportions of Mn oxidation states in the melt (Mn can be found in the 2+, 3+, 70 4+, and 5+ oxidation states in nature). However, under most geological conditions, Mn²⁺ is by far the 71 dominant species in silicate melts (Watson 1977, Schmidt et al. 2006). In apatite, Mn²⁺ has an ionic 72 radius of 0.90 Å in 7-fold (Shannon 1976) and 1.0 Å in 9-fold coordination (Miles et al. 2014). The corresponding values for Ca²⁺ are 1.06 Å and 1.18 Å (Shannon 1976). Mn³⁺ has a smaller ionic radius 73 (0.62-0.67 Å; Miles et al. 2014); hence Mn²⁺ should preferentially partition into apatite from the melt 74 75 compared to Mn³⁺. However, Marks et al (2016) questioned the general applicability of a Mn-in-76 apatite oxybarometer, suggesting that additional controls such as temperature, melt composition and 77 the co-precipitation of other Mn-bearing phases may also influence the Mn content of natural apatites. We have conducted an experimental calibration of Mn oxidation state in silicate melts as a function of fO_2 , in order to investigate its impact on apatite-melt partitioning. Experiments were designed to examine how Mn oxidation state, in both apatite and melt, vary as a function of fO_2 for a range of melt compositions. Apatite in three end-member melt compositions, ranging from basaltic andesite to rhyolitic, were chosen so that any effects of crystal chemistry and melt composition on Mn partitioning could also be determined.

84 1.2 Starting Materials

Starting materials consisted of mixtures of high-purity analytical oxide (SiO₂, Al₂O₃, TiO₂, MgO, Fe₂O₃, 85 86 MnO_2) and carbonate (CaCO₃, Na₂CO₃, and K₂CO₃) powders. Before use, powders were fired using the 87 following conditions: SiO₂ 8 hours 1250 °C, TiO₂ 16 hours 1000 °C, Al₂O₃ 18 hours 1250 °C, Fe₂O₃ 1 hour 88 800 °C, MgO 18 hours 1250 °C, CaCO₃ 4 hours 400 °C, Na₂CO₃ 12 hours 400 °C and K₂CO₃ 8 hours 500 °C. 89 Following firing, powders were stored in a 110 °C oven. The starting compositions were ground under 90 acetone, using an agate mortar and pestle, for thirty minutes, to ensure homogeneity. Powders were 91 then loaded in a box furnace and decarbonated using a controlled heating program, which ramped 92 the temperature from 600- to 1000 ° C over a 6 hour period. Following decarbonation, an apatite 93 component consisting of various amounts of tri-calcium phosphate $[TCP-Ca_3(PO_4)_2]$ and either CaF₂ or 94 CaCl₂ was added to the starting material, and the mixture was reground for a further thirty minutes in 95 acetone. At this stage, water (1.5 wt. % H₂O) was also added to the SH3 and BMT2 starting 96 compositions in the form of gibbsite $(Al(OH)_3)$ or brucite $(Mg(OH)_2)$. The different bulk starting 97 compositions used in this study can be found in Table 1. Note PM1 and HAP10 were two iterations of 98 the same starting compositions.

99 1.3 Experimental Procedures

Experiments were carried out in an end-loaded piston cylinder, using ½" talc-pyrex assemblies with an
 internal graphite resistance furnace, and inner crushable alumina spacers. Runs were first pressurised
 to ~0.3 GPa before heating, then pressurised simultaneously with temperature, which was raised at a

103 rate of 100 °C/min. After reaching super liquidus conditions runs were slightly over pressurised to 104 compensate for pressure loss during the first hour (hot piston out technique). Temperature was 105 continuously monitored using a $Pt_{100-}Pt_{87}Rh_{13}$ thermocouple placed ≈ 10 °C from the hot spot of the 106 assembly. All experiments were run at 1 GPa.

107 Runs were initially taken to super liquidus conditions (Table 2) to ensure the homogeneity of the 108 starting composition and promote growth of larger apatite crystals. Subsequently, the temperature 109 was cooled isobarically to the final T at different rates (see Table 2). The final T and dwell time for each 110 run (Table 2) was chosen to ensure apatite was the only phase present within the melt, allow 111 equilibration between crystals and melt, and to also ensure apatite grains and areas of glass were a 112 suitable size for analysis. On completion of the run samples were isobarically quenched by shutting 113 off power to the heating circuit; T dropped to <50 °C in less than 30 seconds.

Oxygen fugacity was variably controlled in experiments by the sample environment. Initial runs were carried out using a single Pt capsule, which is assumed to have an oxygen fugacity in the vicinity of the Ni – NiO (NNO) solid buffer. Although fO_2 cannot be determined directly from the run products, it can be estimated by comparison with other experiments run using the same experimental procedure. In a separate series of experiments, run using the same bulk composition but doped with Eu, with the same sample preparation procedure and sample assembly, Eu XANES spectra indicate an Eu²⁺/Eu³⁺ ratio in the melt phase close to that of the NNO buffer, as expected (Burnham et al. 2015).

A second suite of experiments were carried out using an inner graphite bucket inside a Pt capsule. As described by Médard et al. (2008), the presence of a graphite liner in the Pt capsule results in buffering of O by reaction with C, and more reducing conditions close to C - CO (CCO-1; Médard et al. 2008). Again, we cannot verify fO_2 conditions directly in these experiments, and it is possible that use of a Fefree starting material here, for example, could result in slightly more reduced conditions than those described by Médard et al. (2008). However, it is unlikely that fO_2 conditions in graphite-Pt capsules will be more oxidising than CCO (Médard et al. 2008), providing a more reducing conditions than

128 samples run in just Pt or with an oxidising buffer. Oxygen fugacity in subsequent experiments was 129 controlled using a double capsule technique (e.g., Eugster 1957; Eugster and Wones 1962; Jakobsson 130 2012). In these runs, fO_2 was controlled using a series of metal-metal oxide oxygen buffers (i.e. Ni-NiO, Cr-Cr₂O₃, Fe₃O₄-Fe₂O₃, Fe₃O₄-FeO) within an outer Pt capsule that enveloped an internal Pt 131 132 capsule containing the starting material. Four µL of deionized (DI) water was micro-syringed into the 133 bottom of all outer capsules regardless of starting composition prior to filing with the buffer and 134 welding. In the CaCl₂ starting composition runs, 4μ L of DI water was added into the inner capsule prior 135 to filing and welding. Calculated oxygen fugacity values are given in Table 2 and were calculated based 136 on the relevant redox buffer equations (Holzheid and O'Neill 1995, Jacobsson 1985, Médard et al. 137 2008, O'Neill 1988, O'Neill and Pownceby 1993). The fO₂ values for experiments run with a solid state 138 buffer could deviate from the calculated buffer values if the activity of water inside the capsule is not 139 in unity, and in water under-saturated samples, fO₂ values will be lower than the coexisting hydrated 140 buffer (Matjuskin et al. 2015), and indicated in Table 2. The purpose of high-pressure experiments 141 here, however, due to complexities in experimental design, is to synthesise apatite in equilibrium with 142 silicate melt under a range of fO_2 conditions, from oxidising to reducing. Gas-mixing synthesis 143 experiments are then used to accurately constrain the influence of fO_2 on Mn oxidation state.

144 Additional experiments were carried out in a vertical tube gas mixing furnace to synthesise Mn bearing 145 glasses at controlled fO_2 conditions, at atmospheric pressure. A Mn-Fe doped granitic starting 146 composition and a Mn doped granitic starting composition was mixed with poly-vinyl acetate (PVA) to 147 create beads, each bead was hung on the end of a Pt loop. Multiple beads were organised on a 148 chandelier and placed in the hot-spot of the furnace on the end of an alumina rod. Redox conditions 149 inside the furnace were determined using CO₂-H₂ mixtures, with in line Bronkhorst mass flow 150 controllers to regulate gas proportions as calculated using the tables of Deines (1970). All experiments 151 were held at 1300 °C for 24 hours to equilibrate before being rapidly quenched by quickly raising the 152 alumina rod out of the hot-spot.

153 1.4 Analytical Techniques: EPMA

Recovered capsules were mounted in crystal bond, ground to the desired level, and polished using diamond paste. Crystal bond was later dissolved using acetone. After cleaning, samples were mounted in 1" indium mounts and carbon coated. Glass beads from the gas mixing furnace experiments were embedded in epoxy, ground, and polished prior to analysis. Electron probe microanalysis data was collected using the JEOL JXA8530F Hyperprobe field emission gun electron microprobe analyser (FEG-EPMA) at the University of Bristol.

Apatite was analysed using a 10 μm beam for all conditions; the first protocol measured Na, Ca, P, F, and Cl with an accelerating voltage of 15keV and 5nA current. The second protocol used a 15 keV accelerating voltage and 20nA current to measure Si, Mg, K, S, Fe, Mn, Ni, Cr, and Ti on the same spot as the first protocol. Standards used were albite (Na, Si), Saint John's Island olivine (Mg), sanidine (K), barite (S), NaCl (Cl), fayalite (Fe), Mn metal (Mn), Ni metal (Ni), Cr₂O₃ (Cr), TiO₂ (Ti), and Durango apatite (Ca, P, F).

The glass was analysed with the following conditions: (1) 15keV accelerating voltage, 10nA current for
Ca, Ti, Si, Al, Na, Mg, F, K, Cl, Fe and Ni (2) 15keV accelerating voltage, 40nA current for Mn, Cr, and P.
All analyses were collected with a 10 μm beam and both protocols analysed the same spot. Standards
were as follows: wollastonite (Ca), Saint John's Island olivine (Mg), TiO₂ (Ti), albite (Si, Na), MgF₂ (F),
sanidine (Al, K), NaCl (Cl), fayalite (Fe, Mn), Cr₂O₃ (Cr), Durango apatite (P), and Ni metal (Ni).

171 1.5 Analytical Techniques: SIMS

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Volatile measurements in the apatite and coexisting glasses were collected by secondary ion mass
spectrometry (SIMS) on a Cameca ims-4f at the NERC ion probe facility at the University of Edinburgh.
Prior to SIMS analysis samples were cleaned with ethanol, and then coated with ~35 nm gold to
provide electrical conductivity.

177 Analyses were done using a 160⁻ primary beam with nominal current of 2.2nA, and 10.8KeV 178 accelerating voltage was targeted on a sample with a 4.5keV voltage to produce a ~15keV net impact 179 energy. Apatite and glass measurements of the secondary ions 1H, 11B, 12C, 16O, 19F, 26Mg, 30Si, 180 31P, 35Cl, 39K, 42Ca, 55Mn, 88Sr, 138Ba, 139La, 140Ce and 151Eu were counted on an electron 181 multiplier for 79 seconds respectively, for each 6 cycles of the magnet. A 3-minute pre-sputter period 182 at a raster size of 10µm was used to pre-clean the analysis area. Secondary ions were acquired with a 25µm diameter image field, and the area analysed was restricted to 8-10µm spot size using a circular 183 184 field aperture. An energy offset of 50eV was applied (40eV window). Higher resolution CO2 185 measurements of glass and apatite were collected on additional spots (size permitting). These were 186 acquired after a 4-minute pre sputter, with a 20µm raster, to pre clean the analysis area. Data was 187 collected with a 60µm diameter image field, 50eV offset (40eV window), and a field aperture was used 188 to restrict ions to a spot size of \sim 15µm in diameter. The mass resolution employed was 900-1000, 189 which is sufficient to separate $25Mg^{2+}$ from $12C^+$. The majority of CO₂ measurements reported for 190 apatite are based on the low-resolution window measurements, using higher resolution measurement to identify the need of an Mg correction. An Mg^{2+} correction based on 25 Mg^{2+} was applied. 191

Only SIMS measurements for H₂O and CO₂ are given here, quantified by the 1H and 12C isotopes respectively. Other elements were measured to confirm that the analyses were not contaminated, and Eu/Ce measurements are for use in future investigation. Apatite standards (Bristol in-house standards – Riker et. al 2018) were used to quantify ion yields for 1H, and CO₂ analyses based on H/Ca and C/Ca ratios. Basaltic glass standards (ST-2 and ST-6) were used to quantify ion yields for 1H, and CO₂ analyses based on H/Si and C/Si. Absolute error for SIMS data is assumed to be better than 10%.

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1.6 Analytical Techniques: XANES

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200 Mn and Fe K-edge XANES spectra were collected on experimental capsules and glasses in 1" epoxy 201 mounts at beamline I18 of the Diamond Light Source (Harwell, UK) in one three-day session. These 202 samples were then removed from epoxy for subsequent analysis techniques (EPMA, SIMS). Spectra

203 were acquired in fluorescence mode, using a Ge detector. Beam energy was finely tuned using a Si(111) 204 crystal monochromator. The beam size was selected by using a variety of slits and was focussed on an 205 area of 5 μ m × 5 μ m. Mn K-edge scans were measured over an energy range of 6400-6750 eV with a 206 1s count time per point. The pre-edge region was measured in 5 eV steps from 6400-6520 eV and from 207 6520-6532 eV with 1 eV steps. The edge region from 6532-6562 eV was measured with 0.2 eV steps 208 and 6562-6580 eV with 1 eV steps. The post-edge region was collected with 3 eV steps from 6600-209 6750 eV. Mn K-edge energy was calibrated by defining the first inflection point in the derivative of a 210 Mn foil standard to 6539 eV.

Fe K-edge spectra were collected from 7000-7320 eV with a 1s count time per point. The pre-edge region was measured from 7000-7090 eV with 6.25 eV steps, and from 7090-7107 with 1 eV steps. The edge region was scanned with 0.25 eV steps from 7107-7120, 0.3eV steps from 7120-7140 eV and 1 eV steps over 7140-7160 eV. The post edge region from 7160-7320 was measured with 3 eV steps. Fe spectra energy was calibrated to the first inflection point in the derivative of a Fe foil standard to 7112 eV.

Multiple Fe and Mn XANES scans on a same sample were taken to check for a change in oxidation with beam exposure. Additional repeat scans were taken on the same sample but at additional spots at the start and end of the beamline session to check the reproducibility of the runs. No obvious shift in the pre-edge energy position was noted for spectra collected at the start and end of the session which is in fitting with the findings of Mosselmans et al. (2009), that the pre-edge position of Ti for scans taken 36 hours apart on beamline I18 shifted by ~0.03eV, which is less than the resolution of the beamline.

Raw data was imported into the software Athena (Ravel and Newville 2005), where multiple scan data was merged. Scans were also deglitched, corrected for self-absorption based on the FLUO algorithm (Haskel 1999) and then the data was pre-edge/post-edge normalised.

Pre-edge peak fitting was carried out on the normalised data in Fityk (Wojdyr 2010). Pre-edge peaks
were obtained from the data by fitting a convex hull spline baseline across the pre-edge region.

Gaussian peaks of fixed half-width at half-maximum of 0.7 were fitted to the data. All data was fit assuming two Gaussian contributions. The Gaussian contributions were used to calculate the average integrated intensity for each fit, and was calculated from the sum of the integrated intensities of each contributing peak. The 'average' centroid position was calculated from the area weighted centroid positions of each Gaussian peak.

233 Experimental Results

234 All run products contained apatite and quenched melt (glass). In run PM1 H an additional unidentified 235 mineral phase was found. Apatite throughout the experiments varies in size from 2 to 500µm, in the 236 longest dimension, with the largest apatite found in the least silicic melts. Modal proportions of 237 apatite crystals are generally less than 30% and the habits of most crystals are euhedral. Apatite 238 commonly display elongate melt inclusions trapped in their cores, running parallel to the longest axis 239 of the crystals. Major, trace, and volatile measurements for glasses from each group of starting 240 compositions are listed in Table 2. Quenched glasses for all runs are homogenous in appearance and 241 have low standard deviation in major and minor element analyses. Apatites reveal no evidence of 242 chemical zonation. Furthermore, standard deviations for major and minor elements, calculated for 243 multiple analyses, are low (Table 3), and comparable to errors on individual analyses calculated by 244 JEOL software from counting statistics. This suggests that all our experiments are in equilibrium.

245 Glass compositions for runs with the PM1/Hap10 starting mix resulted in a range of compositions from 246 trachyte to phonotephroite/basaltic trachyandesite (Fig. 1). This range is due to varying degrees of 247 apatite saturation, and the additional effect of crystallising another phase in PM1 H. This also resulted 248 in different degrees of melt polymerisation as quantified by the ratio of non-bridging oxygens to 249 tetrahedrally coordinated cations (NBO/T). PM1 H contained the most polymerised melt upon 250 quenching (NBO/T = 0.12) and PM1 Pt the least (NBO/T = 0.43; Table 2), while the average NBO/T for 251 the PM1/Hap10 quenched glasses \approx 0.26. MnO contents of the glasses vary from a minimum of 4.31 252 wt% in PM1 H to 9.21 wt% in PM1 Pt. Apatite from the PM1/Hap10 experiments has a range of Mn

contents from 4.94 wt% MnO (0.74 a.p.f.u. PM1 Pt) to 7.89 wt% MnO (1.16 a.p.f.u. - PM1 NNO), with an average of 6.51 wt% MnO. Molar proportion of Cl-H₂O for these apatite displays a range from more Cl rich (X^{Ap}_{Cl} = 0.89, PM1 Pt) in the PM1 samples to closer to pure hydroxyapatite compositions (X^{Ap}_{H2O} = 0.90, Cr 1200 Hap10) in HAP10 runs.

Sample Cr 1200 Hap5 was run in an additional starting composition (Table 1). Hap 5 is similar in composition to Hap10 but contained half the amount of Mn (5 wt.%). This experiment resulted in a trachyitic glass composition. Apatite in this run contained 6.58 wt% MnO (0.96 a.p.f.u.) and the anion content is close in composition to the hydroxyl end member (X^{Ap}_{H2O} = 0.85).

261 The SH3 composition resulted in glasses that are basaltic in composition, ranging from 4.00-4.32 wt% 262 total alkalis (Na₂O + K₂O) and 45.42-46.47 wt% SiO₂ (Fig. 1). SH3 MH shows elevated FeO levels (1.75 263 wt% FeO) compared to other samples in this composition, presumably due to contamination from the 264 buffer, occurring because of diffusion of Fe through the Pt into the sample. The MnO contents in these 265 samples ranges from a minimum of 0.73 wt% MnO in SH3 Cr to a maximum of 0.90 wt% MnO in SH3 266 Pt. Apatite from this composition contains 0.90 wt% MgO (0.22 a.p.f.u.), and a maximum of 0.19 wt% 267 MnO (0.03 a.p.f.u.). All apatite are close to the F end member composition with X^{Ap}F ranging from 0.86-0.96 (Table 3). 268

269 Experiments run in the BMT2 system, with the exception of BMT2 MH, contain glasses that range from 270 tephriphonolite to phonolite in composition (Fig. 1). Sample BMT2 MH was subject to Fe 271 contamination from the buffer, with the addition of 12.1wt% FeO^T that changed the glass composition 272 to a Fe-rich foidite (Fig. 1). The MnO content varied between 0.07-0.80 wt% in these quenched glasses, 273 thought to be a result of differences in the amount of apatite crystallised. Apatite from this starting 274 composition contained variable amounts of Mn compared to SH3, with 0.02, 0.04 and 0.07 a.p.f.u. of 275 Mn for samples BMT2 Cr, MH and Pt respectively. BMT2 Cr is closest in composition to end member 276 FAp (3.53 wt. %), whilst BMT2 Pt has the least amount of F (3.26 wt. %).

The compositions of 1 atm. gas mixing furnace glasses are given in Table 4. Both set of glasses are trachyitic in composition, with the exception of Fe-Mn granitic glasses $Log fO_2$ -5.1 and -6.3, which are tephriphonolite and trachyandesite, respectively. The Fe-Mn granitic glass $Log fO_2$ -6.3 is deficient in Na₂O and K₂O compared to the other samples. All samples were synthesised at the same temperature and left to dwell for the same time. Therefore, the difference in composition is likely to reflect a larger surface area to bead volume for this sample.

283 284

1.7 X-ray absorption near edge structure spectroscopy

285 X-ray absorption near edge structure (XANES) spectra were collected for the PM1 composition 286 (samples NNO, Pt, PtC) to determine if any change in the oxidation state of Mn was detectable in 287 samples synthesised under different fO_2 conditions. Sample PtC should be much more reduced (NNO-288 2.3, where NNO \pm x refers to fO_2 relative to the Ni-NiO buffer in log 10 units), than both the Pt and 289 NNO samples, which are both expected to be more oxidised, around NNO+0, even accounting for any 290 uncertainties in the exact fO_2 at which samples were buffered. Spectra were obtained on both the 291 quenched glass and apatite crystals to determine any change in Mn oxidation state. The XANES spectra 292 from the quenched glass (Fig. 2) are remarkably similar in their shape, edge energy position and pre-293 edge peak energy position between samples PM1 NNO, Pt and PtC suggesting there is very little/no 294 change in the oxidation state of Mn between these melts. When compared to tephroite (Mn₂SiO₄-295 Mn^{2+}) and $Mn_2O_3(Mn^{3+})$ standards, the edge position is very close to the tephroite standard suggesting 296 Mn is in the 2+ oxidation state. There is no variation in the pre-edge peak position or intensity position 297 (Table 5) for these glasses. This suggests no major change in oxidation state or Mn coordination 298 between the quenched melts, with Mn incorporated exclusively, or almost exclusively, as Mn²⁺.

299

XANES spectra for the PM1 apatite grains are also displayed in Figure 2. There are slight differences in
 these 3 spectra with the position of the edge crest varying from a minimum of 6551.33eV for PM1 Pt,

302 6551.75eV for PM1 PtC and 6553.947 eV for PM1 NNO. This shift is thought to represent a change in 303 the local structure around Mn or an effect of crystal orientation (Dyar et al. 2002) rather than any 304 systematic changes in Mn mean oxidation state. A change in oxidation state is unlikely, as no 305 significant energy shift is observed in the pre-edge region (Table 5.), which is less sensitive to the 306 environment of the absorbing atom (Chalmin et al. 2009). Apatite spectra also suggest no major 307 change in Mn oxidation state between samples, with Mn dominantly present as Mn²⁺.

308

309 Additional XANES spectra were collected on Mn and Fe-Mn doped glasses from the HAP10 310 composition, which were synthesised under carefully controlled oxygen fugacites (log fO₂ ranging 311 from -9.4 to -0.7 at 1300 °C) in a gas-mixing furnace. Mn pre-edge peak spectra for the Mn K-edge of 312 the glasses are shown in Figure 3A (Fe-Mn bearing glasses) and 3C (Mn glasses). Once again, there is 313 no observable shift in the Mn pre-edge peak energy as a function of fO_2 , suggesting that in all samples 314 Mn is in a similar valence and environment. The position of the intensity averaged centroid and the location of the main edge suggest Mn²⁺ dominates in these silicate glasses. As there is no difference 315 between the Fe-free and Fe-bearing Mn spectra, this suggests that there is no reduction or oxidation 316 317 of the Mn³⁺ driven by redox changes in Fe during quench in the Fe-bearing samples. In contrast, the 318 Fe K-edge pre-edge peaks obtained from the same glass samples (Fig. 3B), show a clear shift in pre-319 edge peak centroid to higher energy with more oxidising conditions. The oxidation state of Fe in the 320 Fe bearing glasses has been determined using the variogram diagram from Fiege et al. (2017), which 321 is based on the work of Wilke et al. (2001). Spectra collected on our glasses range from nearly 100% Fe²⁺ (samples 66 and 76) down to a minimum of 22.8% Fe²⁺ (sample air/100; Fig. 4), and are consistent 322 323 with glasses in 5-fold, or a mix of 4- and 5-fold coordination.

Over the large range of fO_2 -T space explored for the high pressure and 1 atm. glasses we have been unable to detect any measurable change in the oxidation state of Mn. XANES spectra instead imply that Mn²⁺ dominates in all samples. The average oxidation state of Mn in apatite appears not to change

as a function of fO_2 -T conditions over the range of conditions used in this study. As such, we find no evidence to support the assertion that variation of fO_2 in magmas should result in any significant change in Mn valence.

330 Discussion

331 Oxygen fugacity and D_{Mn}^{Ap-m}

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Our results demonstrate that the majority of Mn occurs as Mn^{2+} in melts across a range of fO_2 333 334 conditions. Here we test the empirical relationship defined by Miles et al. (2014; Fig. 5) by plotting 335 D_{Mn}^{Ap-m} (Nernst partition coefficient, $D_x = [x]_{mineral}/[x]_{melt}$) as a function of fO_2 . All samples in the SH3 haplo-basaltic composition show no significant variation in D_{Mn}^{Ap-m} as a function of fO_2 , even though 336 337 there is a variation of approximately 17 log units in fO_2 between samples. Similarly, there is no obvious relationship between fO_2 and D_{Mn}^{Ap-m} for PM1 and Hap10 melt compositions. This suggests that 338 339 oxygen fugacity is not the main control on Mn apatite-melt partitioning in these compositions, although significant variation in Mn partitioning is noted for experiments involving different melt 340 compositions. Samples in the BMT2 composition show an increase in D_{Mn}^{Ap-m} with decreasing fO_2 . This 341 is unlikely to be due to changes in the availability of Mn^{2+} in the melt (i.e. due to imposed fO_2) as the 342 XANES data indicates that the Mn^{2+}/Mn^{3+} ratio does not vary over a similar range in fO_2 to these 343 experiments. Instead, the apparent fO_2 - D_{Mn}^{Ap-m} trend for BMT2 coincidentally corresponds to 344 345 changes in melt composition.

346 Melt polymerisation and D_{Mn}^{Ap-m}

347

It has long been recognised that trace element partition coefficients tend to be higher for minerals in equilibrium with more silicic melts (Blundy and Wood 2003). Although this may partly reflect disparity in the temperature and pressures of these systems it is thought the extent of melt polymerization is a key control on element partitioning (Mahood and Stimac 1990; Schmidt et al. 2006). Polymerisation is a measure of the extent to which silica tetrahedra within the melt structure are linked together. As 353 tetrahedral sites are linked together by shared oxygens (bridging oxygens), an increase in the extent 354 of polymerisation typically means a reduction in the ability of the melt to incorporate other elements. 355 Polymerisation of a melt can be compared in a variety of ways. In more felsic systems the aluminium 356 saturation index (ASI: molar ratio Al₂O₃/ (Na₂O+K₂O+CaO)) provides an indication of melt 357 polymerisation (Prowatke and Klemme 2005). However, the ratio of non-bridging oxygens (NBO) to 358 tetrahedrally coordinated cations (T) is becoming the standard indicator of melt polymerisation 359 (Cottrell et al. 2009). NBO/T = 0 represents a fully polymerized melt, whereas 4 represents a fully 360 depolymerised melt of isolated tetrahedral (Kohn and Schofield 1994). Here NBO/T was calculated 361 using the procedure originally documented by Mysen et al. (1982, 1984). All Fe has been assigned as network modifiers, due to lack of information on Fe³⁺/Fe^T in these samples. 362

Figure 6 displays the relationships between D_{Mn}^{Ap-m} with NBO/T (Fig. 6A) and ASI (Fig. 6B) for samples 363 364 reported here and equivalent literature studies. ASI data from McCubbin et al. (2015) have not been 365 included given the absence of K₂O in their compositions. Both ASI and NBO/T parameters are 366 intrinsically linked, but there appears to be a stronger relationship between NBO/T and D_{Mn} than ASI, 367 indicating that melt polymerisation is more important than aluminosity in controlling Mn apatite-melt 368 partitioning. The partitioning of Mn into apatite in more polymerised melts appears to be considerably 369 greater than in more depolymerised melts, implying that melt structure has a strong influence on the 370 partitioning of Mn between apatite and melt. We would expect NBO/T to play a key role in dictating D_{Mn}^{Ap-m} over a range of melt compositions which are hosts of terrestrial apatite. Other parameters 371 which could influence D_{Mn}^{Ap-m} are temperature and pressure. However, a similar relationship with 372 D_{Mn}/D_{Zn} (olivine-melt) and NBO/T was documented by Kohn and Schofield (1994), who also 373 374 demonstrated that melt composition has a stronger effect on mineral-melt partitioning than 375 temperature.

The relationship of NBO/T with D_{Mn}^{Ap-m} may arise if Mn ions are coordinated with non-bridging oxygens in silicate melts (Watson 1977). In more polymerised melts, the availability of non-bridging

378 oxygens is lower, and therefore, the extent to which melt networks incorporate metal cations such 379 as Mn^{2+} will be reduced. Alternatively, the observed relationship can be explained based on Mn^{2+} 380 interaction with melt networks, where Mn-O bonding in more polymerised melts is less favourable 381 due to the destabilising effect it has on the silicate framework (Kohn and Schofield 1994). It is unlikely 382 that the high D_{Mn}^{Ap-m} are due to the octahedral site preference energy of the Mn^{2+} ion, as Mn^{2+} has an 383 octahedral site preference energy of zero because of the ion's d⁵ electron configuration (Kohn and 384 Schofield 1994).

385 386

Mn content of apatite at Criffel pluton

387 Miles et al. (2014) highlighted that variations in the Mn content of apatite from different zones within 388 the Criffell granitic pluton (southern Scotland) appeared to correlate with changes in melt fO2, and 389 were relatively independent of other parameters such as temperature and the concentration of Mn in the melt. As the partitioning data presented here indicates there is no relationship between D_{Mn}^{Ap-} 390 391 ^m and fO_2 we used the range of compositions from our study to test other hypotheses. If we apply the NBO/T to D_{Mn}^{Ap-m} power relationship given in Figure 6 to the data for the Criffel pluton, we can 392 393 determine apparent D_{Mn}^{Ap-m} for the relevant melt compositions. NBO/T was calculated from whole 394 rock compositions (Stephens and Halliday 1980; Miles et al. 2013) for the 4 petrologically distinct 395 zones of the pluton for which apatite Mn concentrations were available (Miles et al. 2014). These 396 compositions have been chosen to obtain the largest change in NBO/T. Whole rock MnO contents were combined with the modelled D_{Mn}^{Ap-m} based on NBO/T to predict apatite MnO, and then data was 397 398 converted to ppm for comparison. This method predicts a reasonable fit of the Mn apatite content 399 (ppm) based on NBO/T with measured apatite concentrations from EPMA and SIMS data (Miles et al. 400 2014; Fig. 7). The error in Figure 7 is large indicating the sensitivity of apatite Mn content to the power 401 law fit. The predicted Mn apatite content is also very sensitive to the modelled bulk rock composition 402 and bulk rock MnO content. However, to a first approximation, it appears that the variation seen in 403 apatite Mn concentration at Criffel can be explained solely by variations in NBO/T rather than fO_2 . This

404 is, perhaps, unsurprising as XANES work indicates that over the range of fO_2 conditions inferred for 405 this pluton there is no change in the oxidation state of Mn²⁺ in silicate melt.

406

407 Volatile partitioning as a function of melt structure and oxygen fugacity

408

409 Manganese is unlikely to be the only element for which apatite-melt partitioning is affected by melt 410 composition. Aside from the effect which melt polymerisation should have on partitioning of other 411 cations readily incorporated onto octahedral sites in silicate melt structure, partitioning of anions 412 between apatite and melt may also be affected by NBO/T. Figure 8 shows that D_{Cl}^{Ap-m} is also linked to melt structure for dominantly hydroxyapatite compositions (X^{Ap}_{H2O} > 0.5 a.p.f.u.). In Figure 8, data 413 414 from this study as well as Doherty et al. (2014) and Prowatke and Klemme (2006), all show an increase 415 in D_{Cl}^{Ap-m} with increasing polymerisation of the melt. This is similar to the trend noted for D_{Mn}^{Ap-m}, with 416 Cl becoming more incompatible in the melt with higher degrees of melt polymerisation. This may be 417 because the most polymerised melts also have lower wt% of Ca, Fe, Na and Mg, which are species 418 known to enhance the solubility of Cl in silicate melts (Mathez and Webster 2009).

It has been argued that Nernst partition coefficients are not suitable when examining the distribution of halogens and water between apatite and melt (Boyce et al., 2014; McCubbin et al., 2015) as the combined concentration of Cl, F and OH in the apatite structure is fixed based on stoichiometry (McCubbin et al. 2015). A more suitable measure of volatile partitioning in apatite is the exchange coefficient, K_D (Eq. 2). Equation 2 describes the exchange of OH-F between apatite and melt:

424
$$K_D^{OH-F} = \frac{\left(\frac{OH_{ap}}{OH_m}\right)}{\left(\frac{F_{ap}}{F_m}\right)} = \frac{\left(\frac{OH_{ap}}{F_{ap}}\right)}{\left(\frac{OH_m}{F_m}\right)}$$
 [2]

The exchange coefficient allows partitioning behaviour of two major elements competing for the same
crystallographic site to be determined. The exchange coefficient treats F, Cl and H, as major elements,
and models them in a similar fashion to the behaviour of Mg and Fe between olivine and melt (Roeder

and Emslie 1970). Assuming ideal behaviour, K_D can be calculated using the mole fraction or wt. % of
each species.

430 As such, instead of examining the relationship of D_{Cl}^{Ap-m} with NBO/T, a more suitable parameter may 431 be K_D^{Cl-H2O} . However, K_D^{Cl-H2O} data from published literature for apatite of this composition is much 432 more restricted, making it difficult to determine if there is a relationship with NBO/T.

In contrast, D_F^{Ap-m} appears to be unaffected by melt composition for runs carried out in the fluorapatite
end member composition, likely reflecting that F behaves slightly different in the melt structure to Cl,
or that partitioning of F is more dependent on apatite crystal chemistry, and therefore less affected
by melt composition.

437 Throughout the entire dataset there is no obvious influence of fO_2 on partitioning/exchange of the 438 anions. Therefore, it is likely that mineral-melt partitioning of these elements is controlled by 439 additional variables, such as, element availability in the melt, temperature, and structural controls 440 from apatite and melt.

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443 Conclusions

Apartite-silicate melt partitioning experiments described here demonstrate that fO_2 has no discernible 444 445 influence on Mn or volatile partitioning. This is consistent with the results of XANES spectroscopy 446 which demonstrate that over a wide range of fO₂ conditions Mn is incorporated into both apatite and a range of silicate melts as Mn²⁺. Instead, results of this study, combined with literature data show 447 448 that melt structure, and in particular the extent of melt polymerization as defined by the ratio NBO/T, 449 controls partitioning of Mn between apatite and a range of silicate melts. The varying Mn content of 450 apatite from the Criffel pluton can be explained by the varying degrees of melt polymerisation for 451 samples from different zones of the pluton, rather than changes in the oxygen fugacity of the parental

452 magma, as previously suggested. More polymerised melts generally have a much higher mineral-melt 453 partition coefficient for trace elements (e.g. Mn apatite-melt), than corresponding, less polymerised 454 melts. Therefore, empirical relationships between variables such as oxygen fugacity and the Mn 455 content in apatite are not valid across systems where melt composition changes, and further terms in 456 such equations are needed to constrain the influence of melt composition.

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640 Figure captions

Figure 1: Total alkali-silica (TAS) plot for quenched melt compositions. Sample names refer to oxygen
 fugacity buffer and starting bulk composition used in each run, see text and table 2 for details.

Figure 2: (A) Stacked XANES spectra for PM1- Pt (solid green line), PtC (solid red line), and NNO glasses
(solid purple line) (B) focus on the pre-edge region of A. (C) Stacked XANES spectra of PM1- Pt (solid
green line), PtC (solid red line) and NNO (solid purple line) apatite (D) focus on pre-edge peaks of C.
Also plotted on A) and C) is an additional XANES spectrum of PM1 Pt with the spectra for the standards

- 647 tephroite ($Mn_2SiO_4 Mn^{2+}$) (solid black line) and $Mn_2O_3(Mn^{3+})$ (dashed black line).
- Figure 3: (A) Mn K-edge, (B) Fe K-edge spectra acquired on the same set of Fe-Mn doped granitic glass, and (C) Mn K-edge spectra from Mn doped granitic glass. For reference $\log fO_2$ of NNO = -6.5 at 1300 °C.

Figure 4: Variogram (Wilke et al. 2001) that shows the average centroid position and integrated preedge intensity for Fe reference compounds in single valence/single coordination (grey ellipses and black dots). The black dotted lines indicate binary mixes between the end member reference compounds. Also plotted are the centroid position and intensity for the Fe-Mn doped granitic glasses from this study (crosses). For reference log fO_2 of NNO = -6.5 at 1300 °C.

- **Figure 5:** D_{Mn}^{Ap-m} plotted as a function of log fO_2 for apatite-melt partitioning experiments from this study.
- Figure 6: D_{Mn}^{Ap-m} plotted as a function of ASI (Aluminium saturation index) and NBO/T (number of nonbridging oxygen per tetrahedral cation) for samples in this study and those from Sha and Chappell (1999), Belousova et al. (2001), and McCubbin et al. (2015). The power fit is fitted using data from this study, McCubbin et al. (2015) and Sha and Chappell (1999). Belousova et al. (2001) data is not used for the power law fit because although the data shows a similar trend to other studies, there is an unexplained offset to lower NBO/T.

Figure 7: Boxplot of Mn apatite concentration (ppm) for the 4 petrologically distinct zones of the Criffel
 Pluton (Miles et al. 2014). The blue box highlights the interquartile range for the dataset, with the blue
 line indicating the median value. The black whiskers represent the range of the dataset. Overlaying
 this data is the apparent Mn apatite concentration (red crosses) determined from the power law
 relationship between NBO/T and D_{Mn}^{Ap-m} shown in Fig. 6, and associated error (red lines) based on the
 90% confidence intervals.

Figure 8: D_{Cl}^{Ap-m} against melt NBO/T for our study and literature data (Mathez and Webster 2005;
 Webster et al. 2009; Doherty et al. 2014).

1 Table 1: Target melt compositions, wt% oxide, without additional apatite component. Added to each bulk composition was a 7:3 of starting material to

- 2 apatite component, consisting of 1.5 mol Ca₃(PO₄)₂ to either 1 mol. CaCl₂ (PM1/HAP10/HAP5) or 1 mol. CaF₂ (BMT2/SH3). H₂O added in the form of brucite
- 3 or gibbsite
- 4
- 5

	PM1	HAP10	HAP5	BMT2	SH3
SiO2	64.1	63.7	72.2	61.3	52.2
TiO ₂	-	-	-	0.4	-
Al ₂ O ₃	17.1	17.1	13.4	18.4	17.3
FeO	-	-	-	3.1	-
MnO	7.4	7.3	5.3	1.0	1.0
MgO	-	-	-	0.2	11.5
CaO	-	-	-	0.9	12.1
Na₂O	4.7	4.7	5.4	6.1	2.7
K ₂ O	6.8	7.2	3.7	7.0	1.8
H₂O*	-	-	-	1.5	1.5

7 **Table 2:** Experimental parameters, and average major and trace element composition of melt for experimental runs. Errors represent 1σ. Duration relates

8 to hours at final T, while ramp rate is degrees cooled per hour. Note: Pt = single Pt capsule, H= single Pt capsule with added H₂O, NNO = Ni-NiO buffer, MH =

9 $Fe_3O_4-Fe_2O_3$ buffer, MW = Fe_3O_4-FeO buffer, Cr = Cr-Cr_2O_3 buffer, PtC = graphite capsule inside Pt. H₂O measurements were made by SIMS. NBO/T = number

10 of non-bridging oxygen per tetrahedrally coordinated cation (see text for details).

	DM1 D+				MH 1200	MW 1200	Cr 1200	NNO 1200	Cr 1200
				FIVITFIC	Hap10	Hap10	Hap10	Hap10	Hap5
Initial T	1450	1450	1450	1450	1400	1400	1400	1400	1400
Final T	1250	1100	1250	1250	1200	1200	1200	1200	1200
Ramp Rate	60	60	60	60	60	60	60	60	60
Est. fO ₂ ¹	-7.0	-8.8	-7.0	-9.9	-2.9	-9.3	-17.6	-7.6	-17.6
Duration	23	18	20	21	19	20	24	21	21
Glass									
SiO2	53.39 ± 0.64	61.86 ± 0.78	52.38 ± 0.38	56.13 ± 0.68	50.07 ± 0.58	56.89 ± 1.11	53.48 ± 0.41	52.13 ± 0.51	63.00 ± 0.85
TiO ₂									
Al ₂ O ₃	14.40 ± 0.18	17.85 ± 0.27	15.51 ± 0.20	16.08 ± 0.18	14.60 ± 0.16	16.11 ± 0.17	15.18 ± 0.15	15.41 ± 0.41	12.93 ± 0.13
FeO [⊤]					1.66 ± 0.12	1.57 ± 0.09	0.03 ± 0.01		
MgO					0.10 ± 0.01	0.25 ± 0.02	0.07 ± 0.01	0.08 ± 0.01	0.08 ± 0.01
MnO	9.21 ± 0.52	4.31 ± 0.34	9.06 ± 0.32	6.69 ± 0.27	7.57 ± 0.31	7.16 ± 0.53	7.18 ± 0.22	7.65 ± 0.33	3.88 ± 0.03
CaO	6.10 ± 0.27	1.71 ± 0.08	4.56 ± 0.11	4.43 ± 0.13	4.05 ± 0.19	3.27 ± 0.29	3.32 ± 0.09	4.61 ± 0.23	2.58 ± 0.08
K ₂ O	5.14 ± 0.13	6.67 ± 0.10	4.45 ± 0.07	4.85 ± 0.1	5.20 ± 0.10	6.22 ± 0.16	5.54 ± 0.12	5.47 ± 0.13	3.48 ± 0.08
Na₂O	5.21 ± 0.14	5.45 ± 0.07	2.51 ± 0.09	3.38 ± 0.13	3.32 ± 0.19	4.03 ± 0.12	3.68 ± 0.26	3.68 ± 0.25	4.57 ± 0.13
P ₂ O ₅	3.46 ± 0.23	0.82 ± 0.07	2.85 ± 0.12	1.76 ± 0.11	3.62 ± 0.40	2.12 ± 0.08	2.70 ± 0.20	3.41 ± 0.32	2.20 ± 0.02
Cl	2.45 ± 0.13	0.47 ± 0.03	0.29 ± 0.03	2.08 ± 0.11	0.46 ± 0.01	0.22 ± 0.01	0.15 ± 0.00	0.56 ± 0.05	0.08 ± 0.01
F									
Total	99.35 ± 0.29	99.14 ± 0.99	91.6 ± 0.48	95.39 ± 0.52	90.71 ± 0.89	97.91 ± 0.45	91.39 ± 0.39	93.27 ± 0.87	92.93 ± 0.89
H₂O	0.92		7.79	2.78	7.97		7.63	5.60	6.44
CO ₂	0.16		0.11	0.03				0.20	
NBO/T	0.43	0.12	0.27	0.21	0.29	0.24	0.25	0.30	0.15

12 Table 2 continued

	SH3 Cr	SH3 PtC	SH3 Pt	SH3 MH	BMT2 Cr	BMT2 Pt	BMT2 MH
Initial T	1400	1400	1400	1400	1500	1500	1500
Final T	1250	1250	1250	1250	1350	1350	1350
Ramp Rate	60	60	60	60	10	10	10
Est. fO ₂ ¹	-16.76	-9.91	-7.00	-2.32	-15.20	-6.00	-1.28
Duration	24	24	24	24	9	9	9
Glass							
SiO ₂	46.47 ± 0.43	46.36 ± 0.45	45.78 ± 0.43	45.42 ± 0.50	60.56 ± 0.45	55.80 ± 0.63	45.38 ± 0.49
TiO₂					0.27 ± 0.04	0.30 ± 0.02	0.25 ± 0.01
Al ₂ O ₃	15.64 ± 0.18	15.59 ± 0.18	15.47 ± 0.01	15.33 ± 0.11	18.45 ± 0.09	17.49 ± 0.09	13.34 ± 0.21
FeO [™]		0.04 ± 0.01		1.75 ± 0.05		1.29 ± 0.05	12.08 ± 0.20
MgO	10.00 ± 0.07	10.18 ± 0.06	10.00 ± 0.05	8.55 ± 0.07	0.32 ± 0.01	0.30 ± 0.01	0.28 ± 0.01
MnO	0.73 ± 0.02	0.79 ± 0.02	0.90 ± 0.01	0.86 ± 0.01	0.07 ± 0.01	0.76 ± 0.01	0.58 ± 0.01
CaO	16.47 ± 0.20	16.46 ± 0.06	16.16 ± 0.30	15.56 ± 0.05	3.60 ± 0.03	4.31 ± 0.12	8.11 ± 0.13
K ₂ O	1.58 ± 0.05	1.58 ± 0.06	1.49 ± 0.05	1.66 ± 0.02	6.97 ± 0.09	6.27 ± 0.08	5.09 ± 0.08
Na₂O	2.62 ± 0.13	2.42 ± 0.14	2.51 ± 0.17	2.66 ± 0.14	6.34 ± 0.17	5.73 ± 0.26	4.76 ± 0.26
P ₂ O ₅	4.15 ± 0.12	3.99 ± 0.03	4.19 ± 0.01	4.03 ± 0.02	1.35 ± 0.05	2.48 ± 0.08	5.74 ± 0.18
Cl							
F	1.35 ± 0.05	1.36 ± 0.05	1.37± 0.03	1.28 ± 0.08	1.00 ± 0.06	1.07 ± 0.08	1.22 ± 0.07
Total	99.07 ± 0.48	98.78 ± 0.45	97.90 ± 0.20	97.11 ± 0.65	99.09 ± 0.25	95.78 ± 0.37	96.88 ± 0.51
H₂O	1.60	0.89	1.89	1.89	0.48	2.07	1.90
CO2	0.69	0.29					
NBO/T	0.86	0.86	0.86	0.79	0.11	0.15	0.36

 ${}^{1}fO_{2}$ estimated based on $a_{H2O}=1$; for all double capsule experiments it is possible that fO_{2} is lower due to reduced a_{H2O} .

15 Table 3: Major and trace element compositions of apatite, and associated partition coefficients for

experimental runs. Errors represent 1σ. Note: Pt = single Pt capsule, H= single Pt capsule with added

 H_2O , NNO =Ni-NiO buffer, MH = Fe₃O₄-Fe₂O₃ buffer, MW = Fe₃O₄-FeO buffer, Cr = Cr-Cr₂O₃ buffer, PtC

18 = graphite capsule inside Pt. H_2O measurements were made by SIMS. $X^{Ap}F = F^{Ap}/3.767$, $X^{Ap}CI =$

 $Cl^{Ap}/6.809$ and $X^{Ap}H2O = H_2O^{Ap}/1.81$. Structural formulas are calculated based on 26 anions.

	PM1 Pt	PM1 H	PM1 NNO	PM1 C	MH 1200 Hap10
# Analyses	17	20	28	19	9
SiO ₂	0.37 ± 0.06	0.48 ± 0.10	0.27 ± 0.05	0.45 ± 0.05	0.18 ± 0.01
Al ₂ O ₃	0.04 ± 0.01	0.06 ± 0.02		0.06 ± 0.03	
FeO [⊤]					0.14 ± 0.04
MgO					
MnO	4.94 ± 0.15	6.6 ± 0.25	7.89 ± 0.28	5.93 ± 0.24	5.67 ± 0.50
CaO	48.28 ± 0.24	46.93 ± 0.37	47.51 ± 0.35	47.34 ± 0.49	49.23 ± 0.46
K ₂ O	0.14 ± 0.03	0.19 ± 0.05	0.07 ± 0.03	0.13 ± 0.02	0.02 ± 0.00
Na ₂ O	0.59 ± 0.04	0.60 ± 0.05	0.06 ± 0.03	0.33 ± 0.04	0.13 ± 0.03
P ₂ O ₅	39.89 ± 0.46	39.81 ± 0.27	40.5 ± 0.47	39.7 ± 0.63	40.83 ± 0.49
Cl	6.03 ± 0.38	5.20 ± 0.60	0.94 ± 0.05	5.79 ± 0.91	1.15 ± 0.02
F					
Total	100.27 ± 0.64	99.87 ± 0.62	97.28 ± 0.51	99.72 ± 0.65	97.11 ± 0.39
H₂O	0.09		1.40	0.17	1.43
CO ₂	0.08		0.07	0.04	0.06
Al	0.01	0.01		0.01	
Fe					0.02
Mg					
Mn	0.74	0.99	1.16	0.89	0.83
Ca	9.12	8.86	8.84	8.99	9.12
К	0.06	0.09	0.03	0.06	0.01
Na	0.20	0.21	0.02	0.11	0.04
Sum Ca Site	10.13	10.15	10.06	10.06	10.03
Р	5.95	5.94	5.96	5.95	5.98
Si	0.05	0.07	0.04	0.07	0.03
Sum P site	6.00	6.01	6.00	6.02	6.01
Cl	1.80	1.55	0.28	1.74	0.34
F					
H ₂ O	0.11		1.62	0.20	1.65
Sum X site	1.91		1.90	1.94	1.99
X ^{Ap} Cl	0.89		0.14	0.85	0.17
X ^{Ap} F					
X ^{Ap} H ₂ O	0.05		0.77	0.09	0.79
X ^r total	0.94	4.50	0.91	0.94	0.96
$D_{Mn}^{ap/m}$	0.54	1.53	0.87	0.89	0.75
	0.10		0.18	0.06	0.18
	21.18		0.22	5.30	0.21
K _{DF} ^{r-n20}					

24 Table 3 continued

	MW 1200	Cr 1200	NNO 1200	Cr 1200
	Hap10	Hap10	Hap10	Hap5
# Analyses	4	8	7	6
SiO2	0.55 ± 0.23	0.23 ± 0.02	0.27 ± 0.02	0.27 ± 0.02
Al ₂ O ₃				
FeO ^T	0.67 ± 0.12			
MgO	0.15 ± 0.01	0.03 ± 0.00		0.06 ± 0.01
MnO	7.14 ± 0.49	7.30 ± 0.61	6.61 ± 0.06	6.58 ± 0.08
CaO	46.57 ± 0.66	47.76 ± 0.57	48.11 ± 0.21	48.24 ± 0.22
K ₂ O	0.11 ± 0.04	0.06 ± 0.02	0.05 ± 0.01	0.04 ± 0.01
Na₂O	0.28 ± 0.03	0.08 ± 0.02	0.19 ± 0.03	0.18 ± 0.07
P ₂ O ₅	40.24 ± 0.59	40.92 ± 0.56	40.51 ± 0.50	41.21 ± 0.29
Cl	1.89 ± 0.10	0.63 ± 0.04	1.93 ± 0.03	0.74 ± 0.05
F	0.24 ± 0.30			
Total	97.86 ± 0.60	96.73 ± 0.65	97.41 ± 0.64	97.15 ± 0.27
H₂O		1.63	1.35	1.54
CO ₂		0.08	0.11	0.08
Al				
Fe	0.10			
Mg	0.04	0.01		0.02
Mn	1.06	1.07	0.97	0.96
Са	8.71	8.85	8.98	8.90
К	0.05	0.03	0.02	0.02
Na	0.09	0.03	0.06	0.06
Sum Ca Site	10.05	9.98	10.04	9.95
Р	5.95	5.99	5.97	6.01
Si	0.08	0.03	0.04	0.04
Sum P site	6.03	6.02	6.01	6.04
Cl	0.56	0.18	0.57	0.22
F	0.13			
H₂O		1.88	1.57	1.77
Sum X site		2.07	2.14	1.98
X ^{Ap} Cl		0.09	0.28	0.11
X ^{Ap} F				
X ^{Ap} H₂O		0.90	0.75	0.85
X ^{Ap} total		0.99	1.03	0.96
D _{Mn} ^{ap/m}	1.00	1.02	0.86	1.70
D _{H20} ^{ap/m}		0.21	0.24	0.24
K _D ^{CI-H2O}		0.28	0.40	0.90
K _{DF} ^{F-H2O}				

31 Table 3 continued

	SH3 Cr	SH3 PtC	SH3 Pt	SH3 MH
# Analyses	5	6	4	8
SiO ₂	0.39 ± 0.05	0.42 ± 0.06	0.39 ± 0.10	0.39 ± 0.10
Al ₂ O ₃				
FeO ^T				0.05 ± 0.04
MgO	0.88 ± 0.01	0.86 ± 0.02	0.90 ± 0.01	0.87 ± 0.02
MnO	0.16 ± 0.01	0.19 ± 0.01	0.18 ± 0.01	0.19 ± 0.02
CaO	54.3 ± 0.18	53.96 ± 0.23	54.51 ± 0.10	54.36 ± 0.45
K ₂ O		0.02 ± 0.01		0.02 ± 0.01
Na₂O				
P ₂ O ₅	41.28 ± 0.67	41.79 ± 0.48	41.39 ± 0.22	41.71 ± 0.39
Cl	0.02 ± 0.02	0.04 ± 0.01	0.03 ± 0.01	0.02 ± 0.02
F	3.24 ± 0.19	3.34 ± 0.12	3.61 ± 0.31	3.25 ± 0.18
Total	100.33 ± 0.58	100.71 ± 0.31	101.06 ± 0.19	100.92 ± 0.44
H₂O	0.37	0.26	0.20	0.40
CO2	0.28	0.21	0.43	0.24
Al				
Fe				0.01
Mg	0.22	0.22	0.23	0.22
Mn	0.02	0.03	0.03	0.03
Са	9.85	9.74	9.86	9.79
К		0.01		0.01
Na				
Sum Ca Site	10.10	9.99	10.11	10.05
Р	5.92	5.96	5.91	5.94
Si	0.05	0.06	0.05	0.05
Sum P site	5.97	6.02	5.97	5.99
Cl	0.01	0.01	0.01	0.01
F	1.74	1.78	1.93	1.73
H ₂ O	0.42	0.29	0.23	0.45
Sum X site	2.16	2.08	2.16	2.18
X ^{Ap} Cl		0.01		
X ^{Ap} F	0.86	0.89	0.96	0.86
X ^{Ap} H₂O	0.20	0.14	0.11	0.22
X ^{Ap} total	1.07	1.04	1.07	1.09
D _{Mn} ^{ap/m}	0.22	0.24	0.20	0.22
D _{H20} ^{ap/m}	0.23	0.30	0.11	0.21
K _D ^{CI-H2O}				
K _{DF} ^{F-H2O}	4.38	3.21	9.90	4.43

	BMT2 Cr BMT2		BMT2 MH
# Analyses	4	7	5
SiO ₂	0.32 ± 0.11	0.17 ± 0.01	0.17 ± 0.02
Al ₂ O ₃			
FeO ^T		0.14 ± 0.02	0.61 ± 0.03
MgO	0.12 ± 0.01	0.10 ± 0.01	0.07 ± 0.00
MnO	0.15 ± 0.04	0.49 ± 0.06	0.27 ± 0.01
CaO	54.64 ± 0.45	54.44 ± 0.15	54.40 ± 0.23
K ₂ O	0.09 ± 0.04	0.06 ± 0.02	0.05 ± 0.01
Na₂O	0.11 ± 0.03	0.08 ± 0.02	0.09 ± 0.05
P ₂ O ₅	41.47 ± 0.36	41.52 ± 0.52	41.82 ± 0.42
Cl			0.02 ± 0.02
F	3.53 ± 0.21	3.26 ± 0.13	3.40 ± 0.12
Total	100.5 ± 0.47	100.28 ± 0.50	100.94 ± 0.34
H₂O	0.16	0.26	0.36
CO ₂	0.15	0.21	0.09
Al			
Fe		0.02	0.09
Mg	0.03	0.03	0.02
Mn	0.02	0.07	0.04
Са	9.93	9.90	9.84
К	0.04	0.03	0.02
Na	0.04	0.03	0.03
Sum Ca Site	10.06	10.07	10.03
Р	5.96	5.96	5.98
Si	0.05	0.02	0.02
Sum P site	6.00	5.99	6.00
Cl			0.01
F	1.89	1.75	1.82
H₂O	0.18	0.29	0.41
Sum X site	2.07	2.04	2.23
X ^{Ap} Cl			
X ^{Ap} F	0.94	0.87	0.90
X ^{Ap} H ₂ O	0.09	0.14	0.20
	1.03	1.01	1.10
$D_{Mn}^{ap/m}$	2.14	0.65	0.47
D _{H20} ^{ap/m}	0.34	0.16	0.19
K _D CI-H2O			
K _{DF} ^{F-H2O}	6.50	14.39	7.94

A - Mn granitic glasses							
Log fO ₂	-0.7	<-4	-5.1	-6.3	-7.8	-8.9	-9.4
SiO2	61.20 ± 1.22	62.40 ± 1.04	61.66 ± 0.54	62.01 ± 0.73	60.76 ± 1.27	62.85 ± 1.70	63.48 ± 0.55
Al ₂ O ₃	19.06 ± 0.39	17.64 ± 0.52	17.89 ± 0.66	18.07 ± 0.29	18.61 ± 0.70	17.62 ± 0.91	17.93 ± 0.10
FeO [⊤]							
MnO	8.64 ± 0.75	8.70 ± 0.58	9.26 ± 0.35	9.25 ± 0.34	9.23 ± 0.42	9.29 ± 0.93	6.55 ± 0.80
K ₂ O	6.68 ± 0.18	6.80 ± 0.18	6.81 ± 0.07	6.82 ± 0.14	6.59 ± 0.15	6.39 ± 0.12	7.10 ± 1.25
Na₂O	4.83 ± 0.08	4.73 ± 0.17	4.75 ± 0.10	4.64 ± 0.07	4.58 ± 0.09	4.14 ± 0.12	4.38 ± 0.74
Total	100.54 ± 0.69	100.37 ± 0.45	100.48 ± 1.12	100.90 ± 0.32	99.88 ± 0.61	100.41 ± 0.53	99.56 ± 0.80

Table 4: Major element compositions of Mn granitic and Fe-Mn granitic glasses synthesised in a 1 atm. Gas mixing furnace under reducing conditions. Errors represent 1σ.

B - Mn-Fe granitic glasses

			2				
Log fO₂	-0.7	<-4	-5.1	-6.3	-7.8	-8.9	-9.4
SiO ₂	59.67 ± 1.37	60.62 ± 1.19	57.00 ± 0.75	61.28 ± 1.34	60.41 ± 2.26	61.33 ± 0.86	61.10 ± 2.03
Al ₂ O ₃	19.36 ± 0.90	18.08 ± 0.97	18.43 ± 0.13	18.56 ± 0.66	17.78 ± 0.60	16.95 ± 0.26	17.75 ± 0.52
FeO [⊤]	4.81 ± 0.17	5.40 ± 0.11	7.61 ± 0.17	8.43 ± 0.66	6.65 ± 0.97	7.05 ± 0.25	6.06 ± 0.89
MnO	3.91 ± 0.36	3.58 ± 0.27	4.38 ± 0.11	4.13 ± 0.35	3.75 ± 0.58	3.64 ± 0.12	3.48 ± 0.53
K₂O	6.96 ± 0.19	7.02 ± 0.08	6.78 ± 0.07	3.88 ± 0.22	6.71 ± 0.23	6.76 ± 0.11	7.01 ± 0.21
Na₂O	5.02 ± 0.11	4.88 ± 0.13	4.83 ± 0.20	3.16 ± 0.12	4.57 ± 0.10	4.38 ± 0.16	4.39 ± 0.12
Total	99.76 ± 0.41	99.60 ± 0.52	99.07 ± 0.51	99.48 ± 0.24	99.90 ± 0.50	100.14 ± 0.55	99.80 ± 0.41

Table 5: Pre edge peak fitting for PM1 Pt, PM1 PtC and PM1 NNO. Apt = apatite.

Sample	Centroid (eV)	Intensity
PM1 Pt Glass	6540.46	0.078
PM1 PtC Glass	6540.43	0.080
PM1 NNO Glass	6540.46	0.071
PM1 Pt Apt.	6540.44	0.064
PM1 PtC Apt.	6540.44	0.063
PM1 NNO Apt.	6540.49	0.054

	Zone 1 (09-17) ¹	Zone 2 (09-18) ¹	Zone 3 (09-21) ¹	Zone 4 (274) ²
SiO ₂	63.92	66.04	69.52	71.35
TiO₂	0.58	0.53	0.36	0.21
Al ₂ O ₃	15.75	15.33	15.10	15.39
FeO ^T	3.24	2.74	1.89	1.33
MnO	0.06	0.05	0.04	0.04
MgO	2.51	1.97	0.93	0.97
CaO	3.82	2.67	2.02	0.90
K₂O	3.35	3.86	4.20	4.70
Na₂O	4.12	3.99	3.84	3.84
P ₂ O ₅	0.29	0.26	0.17	0.10
NBO/T	0.18	0.13	0.06	0.03

Table 6: Whole rock geochemistry for thought to be representative of the four petrologic zones at theCriffel Pluton. ¹Values from (Miles et al., 2013. ²Values from (Stephens & Halliday, 1980).











L Pt		Hap5 Cr
LΗ	Δ	SH3 Cr
L NNO	Δ	SH3 PtC
L PtC	Δ	SH3 Pt
10 MH	Δ	SH3 MH
10 MW		BMT2 Cr
10 Cr		BMT2 Pt
10 NNO		BMT2 MH
	. Pt . H . NNO . PtC .10 MH .10 MW .10 Cr .10 NNO	. Pt ● . H △ . NNO △ . PtC △ .10 MH △ .10 MW ▶ .10 Cr ▶ .10 NNO ▶



	Sha and Chappell 1999	McCubbin et al. 2015
×	This study	Belousova et al. 2001





