

Synthesis, characterization, and performance evaluation of multilayered photoanodes by introducing mesoporous carbon and TiO₂ for humic acid adsorption

Soraya Hosseini¹
Hossein Jahangirian²
Thomas J Webster²
Salman Masoudi Soltani³
Mohamed Kheireddine
Aroua¹

¹Department of Chemical Engineering, University of Malaya, Kuala Lumpur, Malaysia; ²Department of Chemical Engineering, Northeastern University, Boston, MA, USA; ³Department of Chemical Engineering, Imperial College London, South Kensington Campus, London, UK

Abstract: Nanostructured photoanodes were prepared via a novel combination of titanium dioxide (TiO₂) nanoparticles and mesoporous carbon (C). Four different photoanodes were synthesized by sol-gel spin coating onto a glassy substrate of fluorine-doped tin oxide. The photocatalytic activities of TiO₂, TiO₂/C/TiO₂, TiO₂/C/C/TiO₂, and TiO₂/C/TiO₂/C/TiO₂ photoanodes were evaluated by exposing the synthesized photoanodes to UV-visible light. The photocurrent density observed in these photoanodes confirmed that an additional layer of mesoporous carbon could successfully increase the photocurrent density. The highest photocurrent density of ~1.022 mA cm⁻² at 1 V/saturated calomel electrode was achieved with TiO₂/C/C/TiO₂ under an illumination intensity of 100 mW cm⁻² from a solar simulator. The highest value of surface roughness was measured for a TiO₂/C/C/TiO₂ combination owing to the presence of two continuous layers of mesoporous carbon. The resulting films had a thickness ranging from 1.605 μm to 5.165 μm after the calcination process. The presence of double-layer mesoporous carbon resulted in a 20% increase in the photocurrent density compared with the TiO₂/C/TiO₂ combination when only a single mesoporous carbon layer was employed. The improved performance of these photoanodes can be attributed to the enhanced porosity and increased void space due to the presence of mesoporous carbon. For the first time, it has been demonstrated here that the photoelectrochemical performance of TiO₂ can be improved by integrating several layers of mesoporous carbon. Comparison of the rate of removal of humic acid by the prepared photoanodes showed that the highest performance from TiO₂/C/C/TiO₂ was due to the highest photocurrent density generated. Therefore, this study showed that optimizing the sequence of mesoporous carbon layers can be a viable and inexpensive method for enhanced humic acid removal.

Keywords: renewable energy, photocatalysis, mesoporous carbon, TiO₂ nanoparticle, multilayer photoelectrode, humic acid

Correspondence: Soraya Hosseini
Department of Chemical Engineering,
University of Malaya, Wilayah Persekutuan,
50603 Kuala Lumpur, Malaysia
Email soraya20h@gmail.com

Hossein Jahangirian
Department of Chemical Engineering,
313 Snell Engineering Center,
Northeastern University, 360 Huntington
Avenue, Boston, MA 02115, USA
Email kamran.jahangirian@gmail.com

Introduction

Recently, the development of highly active photocatalysts has attracted considerable attention in the field of solar energy.¹ Several metal oxides and their modifications are currently being widely used as candidates for improving the photocatalytic processes. Titanium dioxide (TiO₂) is one of the most efficient photocatalysts due to its chemical stability, nontoxicity, low cost, safety to humans and the environment, strong oxidizing abilities, super hydrophilicity, long durability, and transparency to visible light.²⁻⁴ Due to the two distinct types of photochemical reactions on a TiO₂ surface, many studies on its photocatalytic activity have been conducted on various types of surface-modified

TiO₂ catalysts. The photocatalytic activity of TiO₂ has been extensively studied because of its potential use in the environmental and energy areas, hydrogen evolution, and air and water purification systems, including self-cleaning surfaces, sterilization, photoelectrochemical conversion as well as sanitation and remediation applications.⁵⁻⁸

Various morphologies of titanium oxide with different properties have been recently fabricated.⁹ These include spheres, nanorods, fibers, tubes, sheets, and interconnected architectures. Parameters such as particle size, specific surface area, exposed surface area, pore volume, porosity, and crystalline phase can have significant effects on photocatalytic performance. Therefore, tuning the synthesis parameters is crucial for obtaining a final product with desirable properties. So far, various modifications including surface enhancements, coupling with metal oxides, and metal depositing have been performed on TiO₂ nanoparticles in order to improve the photocatalytic activity.^{10,11} The majority of TiO₂ photocatalysts possess low specific surface areas and adsorption efficiencies, which, in turn, lead to low photocatalytic activities. The addition of different layers by loading WO₃ and Fe₂O₃ on TiO₂ thin films has been reported to increase photocatalytic performance. This has been associated with the structural disorder present on adjacent layers. This disorder in structure leads to enhanced scattering of free electrons, consequently lowering electron mobility.^{12,13} Carbonaceous materials spreading out in a broad spectrum, including mesoporous carbon, graphene, fullerene, and nanotubes, are considered to be attractive materials in solar applications and in photocatalytic processes due to their high surface areas.^{14,15}

In this study, TiO₂ was selected as a widely used heterogeneous photocatalyst.¹⁶ The effect of mesoporous carbon as a porous layer in different combinations with TiO₂ was investigated for enhancing the photocatalytic activity of the TiO₂ catalyst under UV-visible (UV-Vis) illumination. TiO₂ nanoparticles were used as the primary layer with mesoporous carbon as the additional surface. Here, several combinations of TiO₂ nanoparticles/mesoporous carbon were tested, and their corresponding photocatalytic activities were measured and subsequently compared with that of the TiO₂ thin film. Thin TiO₂ photocatalytic films were fabricated via a sol-gel method. In addition, mesoporous carbon was synthesized via a self-assembly soft template technique with the F127 surfactant acting as the pore-forming agent. A comparative study of photoanodes was performed over the film's surface using TiO₂, TiO₂/C/TiO₂, TiO₂/C/C/TiO₂, and TiO₂/C/TiO₂/C/TiO₂ on fluorine-doped tin oxide (FTO) substrates. All synthesized photoanodes were tested under UV-Vis light irradiation.

Materials and methods

Materials and chemicals

Triblock poly(ethylene oxide)-*b*-poly(propylene oxide)-*b*-poly(ethylene oxide) copolymer Pluronic F127 (MW = 12,600, PEO₁₀₆-PPO₇₀-PEO₁₀₆), furfuryl alcohol (99%), humic acid (HA), and nitric acid (65%) were purchased from Sigma-Aldrich Co., St Louis, MO, USA. A mixture of 80% anatase and 20% rutile phases with an average particle size of 35 nm was also supplied by Sigma-Aldrich. All the reagents used were of analytical grade and were used as received. FTO glass (sheet resistance ruthenium <15 U cm⁻¹) was provided by Pilkington Glass Co (Northwood, OH, USA).

Preparation of mesoporous carbon

Mesoporous carbon was synthesized by the polymerization of furfuryl alcohol (99%), surfactant (F127), ethanol (99%), and nitric acid (65%) through a surfactant-templating method. In this approach, furfuryl alcohol was used as the carbon source, triblock copolymer Pluronic F127 as a noncarbonizing template polymer, and nitric acid as the catalyst for the polymerization of furfuryl alcohol. First, a prescribed amount of Pluronic F127 was completely dissolved in 5 mL of ethanol and was then continuously stirred for 30 minutes on an automatic shaker at a constant temperature of 25°C. After the complete dissolution of the surfactant in ethanol, 3 g of furfuryl alcohol was added to the surfactant solution followed by the addition of 0.2 mL of nitric acid. Next, the solution was placed and kept inside a cold bath for 2 hours under continual stirring. Finally, a homogeneous solution (light brown) was formed. The polymer solution was coated on the TiO₂ surface as an intermediate layer. The solidification of carbon was followed by a thermal treatment in an oven at 100°C for 24 hours. The dried photoanodes were then carbonized in a tubular furnace under inert (nitrogen) atmosphere with a flow rate of 70 cm³/min. The carbonization was run at 550°C with a heating rate of 3°C/min for a period of 4 hours.

Preparation of the photoanode using TiO₂/mesoporous carbon

The TiO₂ nanostructured photoanode was prepared via a sol-gel spin coating method. In a typical preparation, 5 g of titanium oxide was mixed with 50 mL of ethanol. The mixture was sonicated for 1 hour in a cooled water bath to maintain the temperature at 30°C. This step was vital for a homogeneous dispersion of the nanoparticles. The films of TiO₂/mesoporous carbon were prepared on glass substrates by applying a spin coating technique. The glass substrate (FTO)

was cleaned with acetone in an ultrasonic bath for 30 minutes to remove impurities and was then dried on a hot plate. In order to fabricate the photoanode, the films were spin coated using a spin coater (model WS-400-6NPP; Laurell Technologies Corporation, North Wales, PA, USA) at two fixed rotation speeds: for 5 seconds at 500 rpm and then for 25 seconds at 5,000 rpm. This allowed the deposition of a few drops of the TiO_2 solution onto the FTO substrate. Initially, 50 μL of the suspension was dropped over the glass substrate mounted on a spin coater before spinning. The final rpm of the spin coater was 5,000 rpm with varying accelerations depending on the nanoparticle diameter. After the spin coating process, a drying process was carried out on a hot plate at 250°C for 20 minutes in order to remove the residues in the film. The spin coating and drying processes were repeated to produce a clean thin film. Next, the TiO_2 nanoparticles were deposited on the FTO glass and were calcined under air at 450°C with a heating and cooling rate of $5^\circ\text{C}/\text{min}$ for a period of 2 hours. This was to improve the interconnectivity of the particles. After calcination, the polymer solution was applied as the second layer via a spin coating method. The mesoporous carbon solution was spin coated for 3 minutes at 1,000 rpm. The substrate was dried at 100°C for 6 hours, and the spin coating was repeated to form a thin and porous layer. Finally, the substrate was dried at 100°C for 24 hours in order to remove the residual solvent and moisture. The carbon layer was carbonized under nitrogen gas flow at 550°C for 4 hours with a heating rate of $3^\circ\text{C}/\text{min}$. The carbon solution was coated onto the TiO_2 photoanode as the second and/or third layer.

Adsorption experiments

A stock solution of HA with a concentration of 1,000 ppm was prepared. The HA solutions were further prepared by diluting the stock solution using distilled water to the desired concentrations. The HA adsorption into the synthesized photoanode was conducted in a concentration range of 200 ppm at an optimal pH value. A UV-Vis spectrophotometer (Shimadzu UV) was used for the determination of the HA concentration at 255 nm wavelength. Photoanodes were immersed into the HA solution for various contact times at room temperature and were then rinsed in distilled water to remove the nonadsorbed HA molecules.

Characterization of the synthesized photoanodes

X-ray diffraction (XRD) patterns of the $\text{TiO}_2/\text{mesoporous carbon}$ nanocomposites were produced using a Bruker AXS D8 advance diffraction spectrometer ($\text{CuK}\alpha$ $\lambda = 1.54056 \text{ \AA}$,

operated at 40 kV and 40 mA) at a scan rate of $0.02\theta/1.5 \text{ s}$. The specific surface area of mesoporous carbon was evaluated by isothermal nitrogen adsorption/desorption measurements at 77 K and then calculated using the Brunauer–Emmett–Teller equation. The samples were degassed at 250°C for 15 hours prior to the nitrogen adsorption experiments. The pore size distribution was calculated by employing the Barrett–Joyner–Halenda equation. Surface roughness measurements were conducted using an atomic force microscope (AFM, SPI3800N; Seiko Inc., Tokyo, Japan). All scratch tests were scanned using tapping mode by a sensitive silicon nitride tip, with a curvature radius of 10 nm and a nominal spring constant of 45 N/m. The surface morphologies of the $\text{TiO}_2/\text{mesoporous carbon}$ films were studied using a HITHACHI-4300 scanning electron microscope (SEM).

Photoelectrochemical cell

The photoelectrochemical cell was prepared by setting up two electrodes and an electrolyte in a container equipped with a quartz optical window. The container was filled with an aqueous electrolyte with the light being able to pass through the optical window as shown in Figure 1. The electrodes were

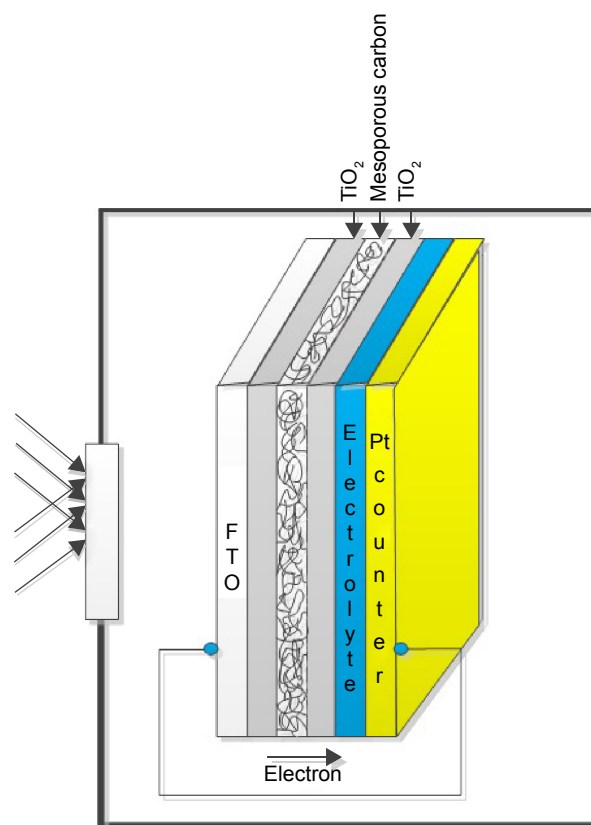


Figure 1 The structure of the photoelectrochemical cell ($\text{TiO}_2/\text{C}/\text{TiO}_2$).

Abbreviations: TiO_2 , titanium dioxide; FTO, fluorine-doped tin oxide; Pt, platinum.

wrapped up in an aqueous electrolyte. Each electrode could be photoactive. The electron–hole pairs were created due to the absorption of photons on the photoelectrode with sufficient energy and the subsequent transfer of the electron from the valence to the conduction band of the semiconductor.¹⁷ Water was oxidized through the holes on the semiconductor surface with the gaseous oxygen evolving at the photoanode surface. Hydrogen ions were converted into gaseous hydrogen due to a reduction reaction as a result of the ion migration to the cathode within the aqueous electrolyte. In the three-electrode cell, a reference electrode was typically used. All experiments were carried out in triplicate.

Results and discussion

Characterization of photoanodes

The XRD patterns of pure TiO₂ and TiO₂/C with various configurations are presented in Figure 2. In the XRD patterns of TiO₂ nanoparticles, the anatase phase was the predominant structure compared with the rutile phase.¹⁸ Anatase has a more open crystalline structure, whereas rutile has a compact, rod-like, and crystalline structure as depicted in Figure 3. Studies have also suggested that the anatase phase, due to its open structure, results in a more active and selective catalyst compared with the rutile phase. It is widely believed that anatase is the active phase in photocatalytic reactions.¹⁸ All peaks are well indexed as the anatase phase (JCPDS card no 89-4921) in the pure TiO₂ sample. The average crystallite size was understood to be 8.29 nm based on the Debye–Scherrer equation:¹⁹

$$t = \frac{K\lambda}{\beta \cos \theta_B} \quad (1)$$

The mean crystallite size can be calculated based on the value corresponding to the maximum peak's half value. Where λ is the wavelength of the X-ray radiation ($\lambda=0.154056$ nm), K is the Scherrer constant ($K=0.89$), θ is the diffraction angle, and β is the line width at half-maximum height of the most intense peak. A major peak corresponding to (1 0 1) reflections of the anatase phase of TiO₂ was apparent at an angle of 25.3°, whereas the minor peaks appearing at 34.50°, 48.5°, 53.5°, and 55.1° represented the indices of (0 0 4), (2 0 0), (1 0 5), and (2 1 1) planes of anatase and TiO₂, respectively.

The average crystallite size of anatase TiO₂ was calculated from the Debye–Scherrer equation using the (1 0 1) diffraction peak of anatase TiO₂ at 25.3°. Upon adding mesoporous carbon content, the peak intensities of mesoporous carbon grew and the anatase phase of TiO₂ became stronger, indicating that the crystallinity of the TiO₂ anatase was preserved during the addition of the mesoporous carbon

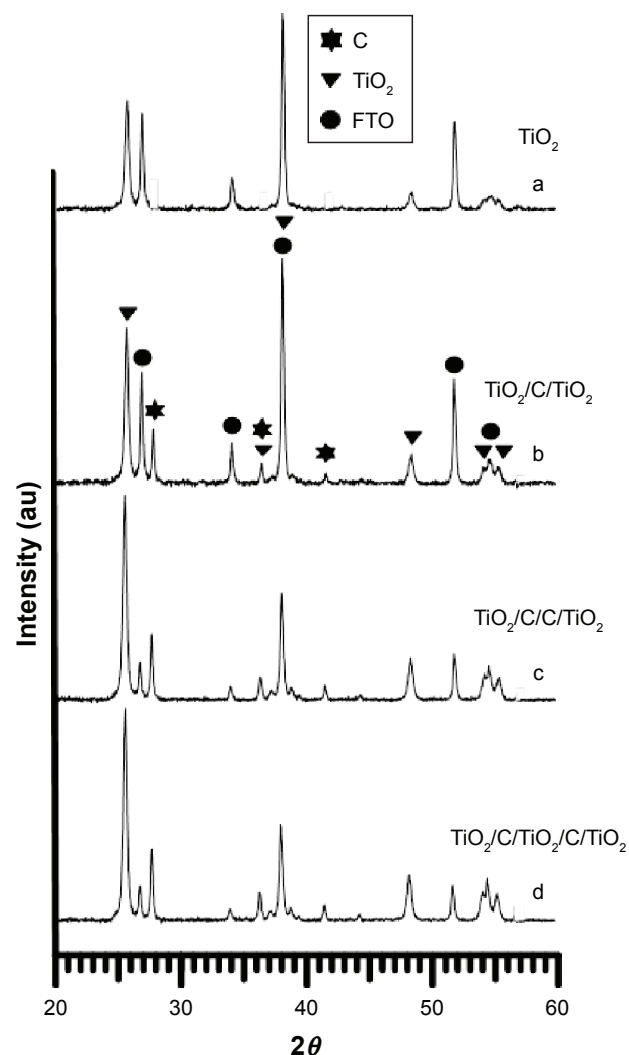


Figure 2 XRD results for (a) TiO₂, (b) TiO₂/C/TiO₂, (c) TiO₂/C/C/TiO₂, and (d) TiO₂/C/TiO₂/C/TiO₂ samples. **Abbreviations:** XRD, X-ray diffraction; TiO₂, titanium dioxide; FTO, fluorine-doped tin oxide.

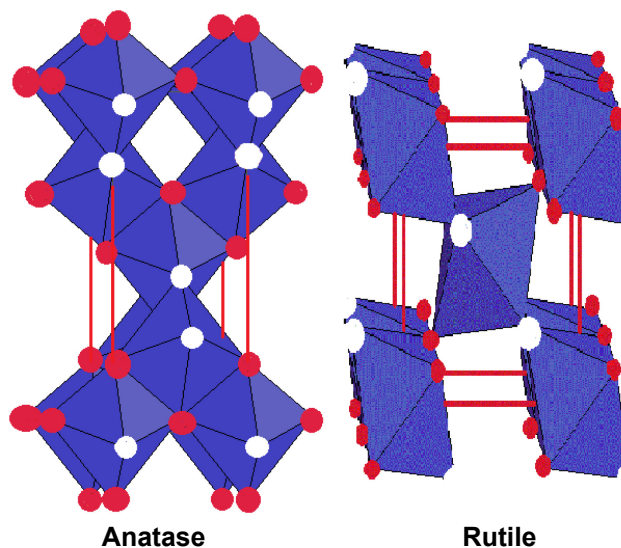


Figure 3 A schematic of anatase and rutile phases of TiO₂ structure. **Abbreviation:** TiO₂, titanium dioxide.

layers. The TiO_2/C nanocomposites did not show any peak shift, indicating that the TiO_2 matrix was well maintained as the anatase phase. By adding mesoporous carbon as an additional layer, some of the diffraction peaks became sharp and narrow while some others became broader. There are distinct peaks in the TiO_2/C pattern: one at 25.3° that belongs to the anatase phase of TiO_2 and the others occurring at 38° and 43° that belong to graphite.

The SEM images of TiO_2 , $\text{TiO}_2/\text{C}/\text{TiO}_2$, $\text{TiO}_2/\text{C}/\text{C}/\text{TiO}_2$, and $\text{TiO}_2/\text{C}/\text{TiO}_2/\text{C}/\text{TiO}_2$ are shown in Figure 4. Photocatalysts with unique styles of a coating layer (one to five layers) on the FTO substrate were fabricated. The film thickness was measured by SEM, which ranged from $1.605\ \mu\text{m}$ to $5.165\ \mu\text{m}$. Uncoated FTO thin oxide conductive layers had a thickness of $\sim 820\ \text{nm}$. The coarseness of the surface was different for each synthesized photocatalyst as a result of the incorporation of mesoporous carbon as the intermediate layer. It is clear that the coarseness of the surface increased when the mesoporous carbon was selected as the intermediate layer. The $\text{TiO}_2/\text{C}/\text{C}/\text{TiO}_2$ sample showed the highest degree of coarseness. This could be the consequence of the presence of two continuous layers of mesoporous carbon providing numerous void space, which can also lead to the enhancement in the absorption of light beams. By the introduction of a mesoporous carbon layer, the coarseness also increased in all three samples compared with pure TiO_2 . However, with TiO_2 as the middle layer ($\text{TiO}_2/\text{C}/\text{TiO}_2/\text{C}/\text{TiO}_2$), a reduction in the coarseness of the surface and, consequently, a decrease

in photocurrent density was observed. In this case, TiO_2 may play a role as a barrier to electron transfer. Obviously, at least three coating layers are crucial for the fabrication of TiO_2 photocatalyst. This intermediate layer is required in order to increase the coarse pore structure of the layer and large surface irregularities or defects leading to the dispersion of UV–Vis light. When more than three coatings were further added to the structure, the photocurrent density started to decrease ($\text{TiO}_2/\text{C}/\text{TiO}_2/\text{C}/\text{TiO}_2$). It was observed that increasing the layers is not effective for the photocatalytic process, whereas increasing the porous layer (mesoporous carbon) had a strong impact on the photocurrent.

N_2 adsorption/desorption isotherms (Figure 5) illuminated that mesoporous carbon and TiO_2 nanoparticles exhibited a typical type IV isotherm according to the IUPAC classification,²⁰ indicating mesopore-rich textural properties for both samples. In mesoporous carbon, a steep and sharp H1-type hysteresis loop at a high relative pressure ($P/P_0 > 0.6$) was observed. However, capillary condensation at a low relative pressure ($P/P_0 \approx 0.4$) in the TiO_2 nanoparticle mesopores was detected. The Brunauer–Emmett–Teller surface area and pore volume of mesoporous carbon and TiO_2 nanoparticles are shown in Table 1. In Figure 5, mesoporous carbon possess a high surface area, ie, three times of that of TiO_2 nanoparticles. By introducing mesoporous carbon as a sublayer between the TiO_2 layers, the photocurrent density could improve due to enhanced surface area and roughness. This could similarly lead to an enhanced efficiency in light absorption.

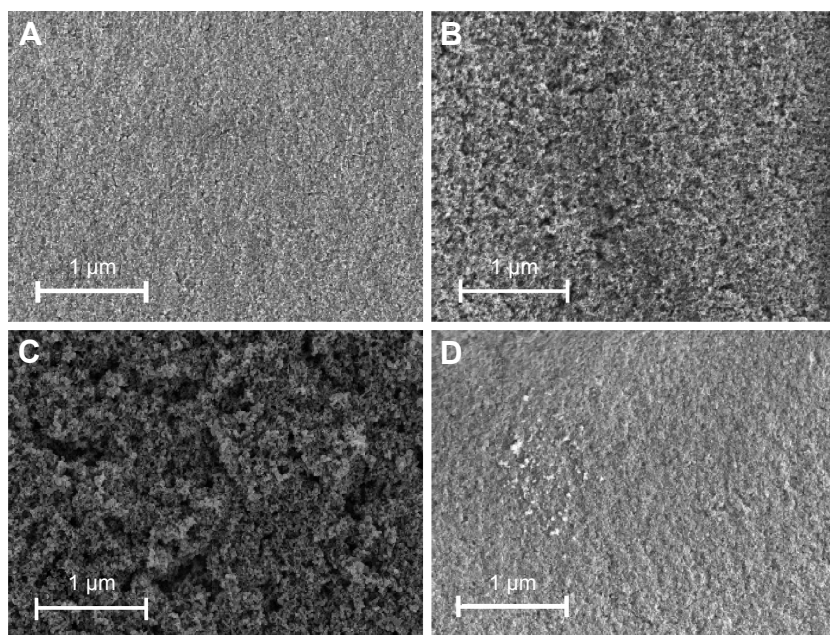


Figure 4 SEM images of the film surface deposited by a spin coating solution on the FTO substrate with a thickness of about 820 nm.

Notes: (A) TiO_2 , (B) $\text{TiO}_2/\text{C}/\text{TiO}_2$, (C) $\text{TiO}_2/\text{C}/\text{C}/\text{TiO}_2$, and (D) $\text{TiO}_2/\text{C}/\text{TiO}_2/\text{C}/\text{TiO}_2$.

Abbreviations: SEM, scanning electron microscope; FTO, fluorine-doped tin oxide; TiO_2 , titanium dioxide.

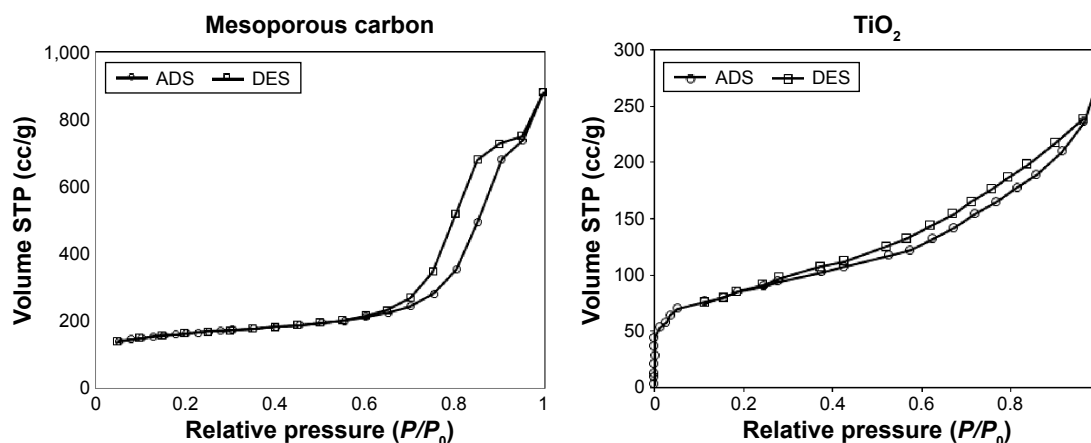


Figure 5 Nitrogen adsorption/desorption of mesoporous carbon and TiO₂ nanoparticles.
Abbreviations: TiO₂, titanium dioxide; ADS, adsorption; DES, desorption; STP, standard temperature and pressure.

Figure 6 shows the AFM images (3D mode) of the surfaces of TiO₂ and TiO₂/C films. The surface plots of all samples demonstrate that the films are quite rough at the nanometer scale with a granular structure consisting of interconnected grain particles fused together. The particles create indentations and protrusions as seen in Figure 6. The surface roughness increases in the samples with mesoporous carbon as an additional layer. Figure 6 clearly reveals the rough surface of TiO₂/C/C/TiO₂ in comparison with the other samples. The results are in agreement with the SEM images presented in Figure 4. The low surface roughness for TiO₂ implies that the films consist of fine grains and amorphous structures, whereas the large grains were found in the sample with mesoporous carbon. The samples with different layer combinations and styles took on quite different surface morphologies. The surface of TiO₂ shows a small degree of roughness and is more homogeneous compared with the samples with mesoporous carbon. AFM results successfully demonstrated the effect of mesoporous carbon on surface roughness.

Photocurrent density

Carbons are strong light-absorbing materials and can be employed in the production of photoanodes. Activated carbon–titania composites as photocatalysts have shown high efficiencies for the photodegradation of pollutants.²¹ Velasco et al investigated the photochemical behavior

of activated carbons under UV–Vis light irradiation. They suggested that UV–Vis light could penetrate inside the pores of carbon.¹⁴ The carbon/UV–Vis light interactions occur at the carbon surface promoting the photogeneration of charge carriers (electron–hole pairs) that could migrate through the graphitic sheets in the carbon and then transfer to the other layers. Moreover, oxygen functionalities on mesoporous carbon could act as charge injectors upon UV light excitation. Unfortunately, the relationship between the nature of carbon and their photocatalytic activity under UV–Vis light is not well clear. However, the mechanism (electron–hole pairs) in mesoporous carbon may be associated with the porosity of carbon or can be attributed to carbon surface functions. This mechanism is explained by the penetration of UV light into carbon porosity. Moreover, carbon–photon interactions can occur at the external carbon surfaces subsequently propagated through the mesoporous carbon.^{14,22,23}

When carbon is illuminated under UV light, a fraction of the incident photon flux would provoke the generation of charge carriers, which will diffuse randomly through the mesoporous carbon. The role of the graphitic sheets of carbons would be of paramount importance for the migration of the photogenerated electrons, minimizing recombination, and favoring the electron transfer to the other sheets.²⁴ The characteristics of photocurrent density versus photovoltage for the cells are plotted in Figure 7. The corresponding parameters are also presented in Table 2. The surface roughness was enhanced in a sequential fashion, ie, TiO₂/C/C/TiO₂ > TiO₂/C/TiO₂ > TiO₂/C/TiO₂/C/TiO₂ > TiO₂. This order shows that TiO₂/C/C/TiO₂ has the highest degree of surface roughness, and therefore, the highest value of photocurrent density was linked to this photoanode. In this case, the porous surface of photoanodes provided

Table I BET analysis of mesoporous carbon and TiO₂ nanoparticles

	Surface area (m ² /g)	Pore volume (cm ³ /g)	Average size (nm)
Mesoporous carbon	980	1.05	8.75
TiO ₂ nanoparticle	285	0.65	8.12

Abbreviations: BET, Brunauer–Emmett–Teller; TiO₂, titanium dioxide.

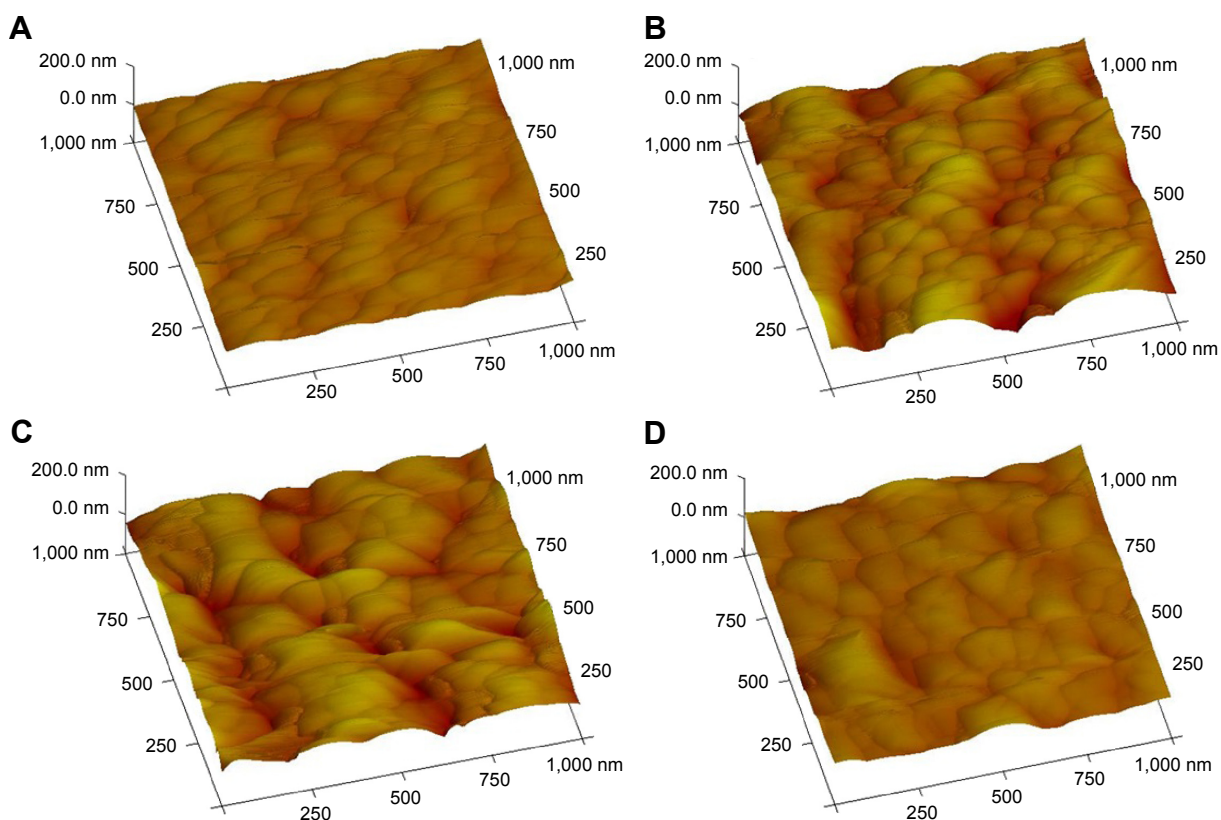


Figure 6 AFM images of the samples.

Notes: (A) TiO_2 , (B) $\text{TiO}_2/\text{C}/\text{TiO}_2$, (C) $\text{TiO}_2/\text{C}/\text{C}/\text{TiO}_2$, and (D) $\text{TiO}_2/\text{C}/\text{TiO}_2/\text{C}/\text{TiO}_2$.

Abbreviations: AFM, atomic force microscope; TiO_2 , titanium dioxide.

high surface area (to absorb maximum light), and a double layer of mesoporous carbon facilitated the rapid transport of charges. Figure 7 confirms that adding a layer of mesoporous carbon could increase the photocurrent density compared with the photoanode fabricated by the TiO_2 nanoparticles alone.

The efficiency of TiO_2 in solar energy applications is restricted due to the fast photogenerated electron–hole recombination in the UV–Vis light absorption. In the present

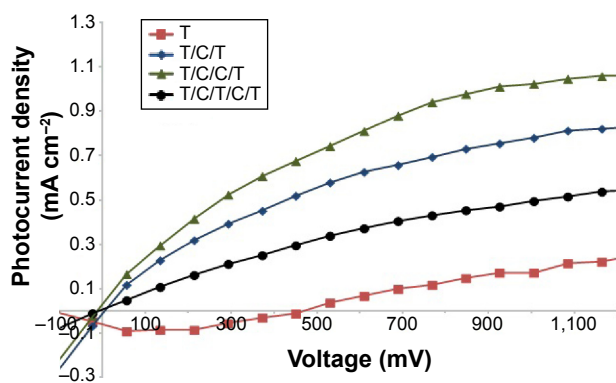


Figure 7 Photocurrent values of different photoanodes: TiO_2 (T), $\text{TiO}_2/\text{C}/\text{TiO}_2$ (T/C/T), $\text{TiO}_2/\text{C}/\text{C}/\text{TiO}_2$ (T/C/C/T), and $\text{TiO}_2/\text{C}/\text{TiO}_2/\text{C}/\text{TiO}_2$ (T/C/T/C/T).

Abbreviation: TiO_2 , titanium dioxide.

study, the design of layers is based on thin TiO_2 /carbon films where the carbon is of mesoporous scale. Mesoporous carbon represents a type of carbon, which is neither perfectly crystalline nor completely amorphous, and is composed of sp^2 microstructures lacking a long-range crystalline order. The synthesized carbon is of a porous and amorphous structure with a porosity ranging from <2 nm (micropores) and 2–50 nm (mesoporous) to >50 nm (macropores) with a high surface area. The exact structural mechanism of carbon/ TiO_2 is not clear, and the addition of carbon in the TiO_2 lattice causes a significant bandgap narrowing.²⁵ Zhao et al have studied TiO_2 loaded onto activated carbon fibers demonstrating enhanced photocatalytic activity and a red-shift in excitation wavelength ($E_g = 1.68$ eV or 2.10 eV)²⁶ resulting in red shift in the absorption spectrum. This difference in red shift is associated with the carbon location in the lattice, which is dependent on the preparation methods employed. The nature of the carbon dopant state in the TiO_2 lattice can be generated in different bonds such as Ti–C or O–Ti–C. It is also demonstrated that the concentration of doped carbon in TiO_2 is an important factor for observing the visible light activity of TiO_2 . Some studies indicated that the dopant states of carbon in TiO_2 are highly undesirable because

Table 2 Photocurrent density and various parameters of photoanodes using TiO₂ and mesoporous carbon with different arrangements

Sample	Grain size (nm)	Surface roughness, R _z (nm)	Film thickness (μm)	Open circuit potential (V)	Photocurrent density (mA cm ⁻²)	
					At 0 V/SCE	At 1 V/SCE
TiO ₂	5	20±1	1.605	-0.119	-0.024	0.203
TiO ₂ /C/TiO ₂	11	47±1	3.493	-0.008	-0.018	0.804
TiO ₂ /C/C/TiO ₂	16	63±1	4.525	-0.004	0.032	1.022
TiO ₂ /C/TiO ₂ /C/TiO ₂	8	29±1	5.165	-0.011	-0.006	0.503

Abbreviations: SCE, saturated calomel electrode; TiO₂, titanium dioxide.

they promote a charge carrier recombination.²⁷ Some studies have pointed out that the enhanced activity was attributed to improved charge carrier separation at the interface between the carbon and the TiO₂ films.²⁸

Mesoporous carbon could promote photocatalytic activity due to the higher surface area of the composites (C/TiO₂) and consequently absorption of more light beams. Mesoporous carbon can provide good electronic properties in the C/TiO₂ composites. Therefore, carbon with more homogeneity in structure has a large electron-storage capacity (one electron for each 32 carbon atoms) and a fast transmission rate of photogenerated electrons. Moreover, the photogenerated electrons are transferred from the TiO₂ conduction band toward the carbon surface due to a lower Fermi level. Some of the proposed effects contributing to better photocatalytic performance (TiO₂/C) were suggested by Sellappan et al,²⁸ which are illustrated in Figure 8.

Furthermore, in the presence of carbon, the photogenerated electrons on TiO₂ can continuously move to the carbon surface, spatially separating the electron and the hole, thus, effectively decreasing recombination.^{29,30} Adding mesoporous carbon may affect charge carrier dynamics and, in turn, affect photocatalytic activity.

Experiments on pure TiO₂ show a decrease in photocurrent density at low voltages. Current density is the sum of the capacitive and ionic components. For TiO₂, the capacitive behavior appeared at low voltages (negative current), whereas the capacitive current is zero for photoanodes containing TiO₂ and mesoporous carbon. A negative current value or downward deflection of a current trace is typically referred to as an inward current. A negative current value (ie, inward current) can reflect either the movement of positive ions (cations) or negative ions (anions). O²⁻ anions (from the water content) are diffusing inward into the metallic surface (TiO₂) and, thus, a growth of an anodic layer can take place leading to the ejection of a great number of electrons, ie, generating the highest electronic current response. So the current decays during the first minutes of reaction. Mesoporous carbon as an intermediate layer can act as a current collector to prevent capacitive current.

Adsorption study

HA is a major type of dissolved organic matter that contains a variety of functional groups such as aromatic units, amide, amine, and various oxygen-containing groups. HA adsorption was studied and compared by using two photoanodes

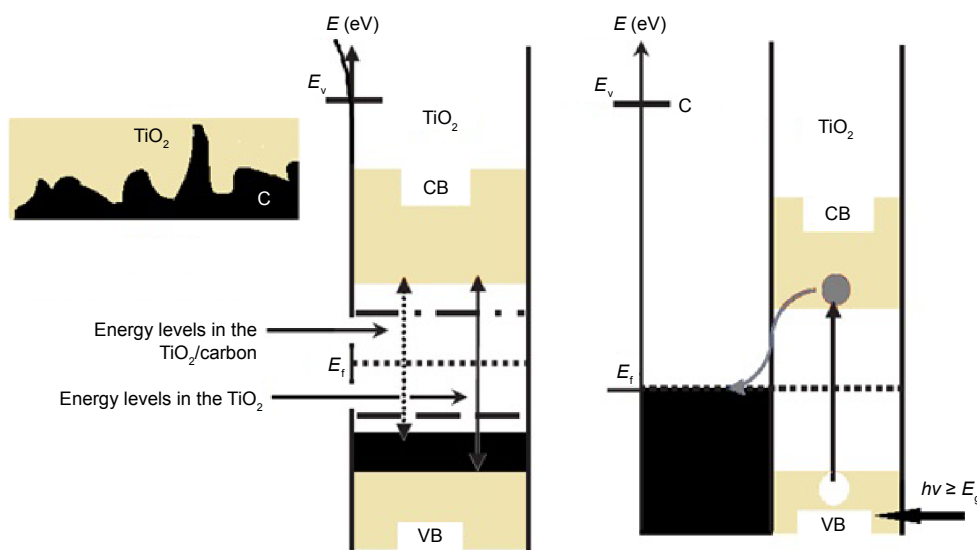


Figure 8 Energy level diagram of charge carrier separation at the interface between carbon and TiO₂.

Abbreviations: C, carbon; CB, conduction band; E_v, vacuum energy level position; E_g, optical bandgap; E_f, Fermi energy level position; hν, energy of photon; TiO₂, titanium dioxide; VB, valence band.

(with C and without C). The amount of HA obtained from adsorption can be calculated by measuring the HA concentration at each time interval. As soon as the system reached equilibrium, the HA concentration was measured using a UV–Vis spectroscope. The effect of pH on HA removal was studied within a pH range of 4–10 by using $\text{TiO}_2/\text{C}/\text{C}/\text{TiO}_2$ as photoanode. The results are presented in Figure 9. The adsorption attains an optimum at pH 5 for the adsorption of HA on photoanodes. Two photoanodes (TiO_2 and $\text{TiO}_2/\text{C}/\text{C}/\text{TiO}_2$) were employed in order to adsorb HA. This was due to the fact that these two photoanodes corresponded to producing the minimum and maximum current densities, respectively. The HA removal performances by both $\text{TiO}_2/\text{C}/\text{C}/\text{TiO}_2$ and TiO_2 are shown in Figure 10. The adsorbed HA indicated that $\text{TiO}_2/\text{C}/\text{C}/\text{TiO}_2$ had a higher efficiency compared with TiO_2 . This shows that $\text{TiO}_2/\text{C}/\text{C}/\text{TiO}_2$, having both the highest degree of surface roughness and the porous surface area, provides the maximum light absorption efficiency.

The decay rate of the HA concentration was found to follow a pseudo first-order reaction. The rate could be determined by the following integrated first-order decay expression:

$$\ln \frac{C_0}{C_t} = -kt \quad (2)$$

The adsorption capacity at equilibrium (q_e , mg/g) was determined by using:

$$q_e = \frac{(C_0 - C_e)V}{m} \quad (3)$$

where C_0 and C_e (mg/L) are the initial and equilibrium concentrations of HA, respectively, m (g) is the mass of adsorbent, and V (L) is the volume of adsorbate solution.

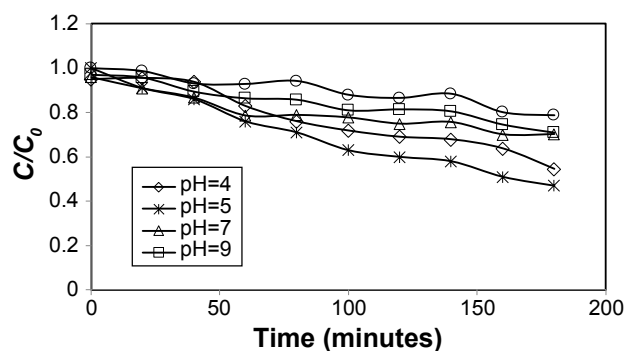


Figure 9 Effect of pH on HA removal by using $\text{TiO}_2/\text{C}/\text{C}/\text{TiO}_2$ (initial HA concentration: 200 ppm).

Abbreviations: HA, humic acid; TiO_2 , titanium dioxide.

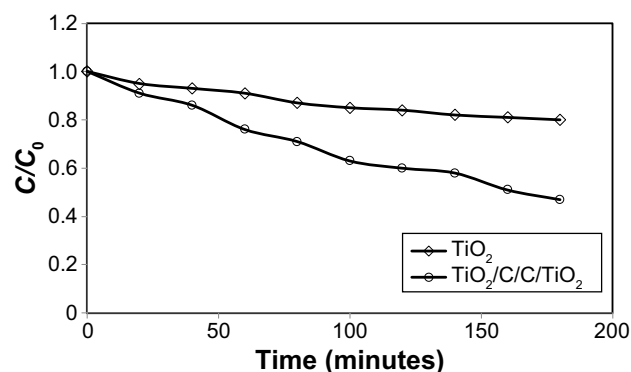


Figure 10 HA concentration during the photocatalytic activity under visible light irradiation (initial HA concentration: 200 ppm, pH=5).

Abbreviations: HA, humic acid; TiO_2 , titanium dioxide.

In the present study, $\text{TiO}_2/\text{C}/\text{C}/\text{TiO}_2$ had a photocurrent higher than pure TiO_2 , partly due to its larger specific surface area as well as larger porous surface area. A comparative study of the photocatalytic activity of the mesoporous sample with the bulk sample (TiO_2) indicated the importance of mesoporosity on the recombination of excited charge carriers and photocatalysis. These results demonstrated that the migration of the photoexcited electrons and the holes in the mesoporous sample and bulk sample should be quite different due to significant differences in surface structure.

Conclusion

Mesoporous carbon with high surface area and porosity was employed in the synthesis of multilayered photoanodes. Two different types of photoanodes were tested in order to investigate the effect of mesoporosity of carbon on photocurrent intensity and surface roughness. A higher photocurrent density was observed with double layers of mesoporous carbon between TiO_2 when compared with only one layer of mesoporous carbon. A double layer of mesoporous carbon between TiO_2 also gave a higher degree of surface roughness in comparison with one layer of mesoporous carbon. A significant enhancement in photocurrent was measured by adding additional layers as seen for the two photoanodes: $\text{TiO}_2/\text{C}/\text{TiO}_2$ and $\text{TiO}_2/\text{C}/\text{TiO}_2/\text{C}/\text{TiO}_2$. The addition of two carbon layers increased the graphite sheets between the TiO_2 layers. The graphite sheets may have facilitated rapid transport of charges or contributed to the generation of charged carriers due to the functional groups of mesoporous carbon. Adsorption coupled with a photocatalytic process was used here as an economic approach for waste water treatment, such as the removal of HA. A comparative study of the photocatalytic activity of two photoanodes ($\text{TiO}_2/\text{C}/\text{C}/\text{TiO}_2$ and TiO_2) demonstrated that the mesoporosity of carbon and the recombination of excited charge carriers could increase HA removal.

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Disclosure

The authors report no conflicts of interest in this work.

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