

## Article

**Assessing the Performance of Dispersionless and Dispersion-accounting Methods: Helium Interaction with Cluster Models of the TiO<sub>2</sub>(110) Surface**

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5                   **Dispersion-accounting Methods: Helium Interaction with Cluster**  
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7                   **Models of the TiO<sub>2</sub>(110) Surface**  
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## Abstract

As a prototypical dispersion-dominated physisorption problem, we analyze here the performance of dispersionless and dispersion-accounting methodologies on the helium interaction with cluster models of the TiO<sub>2</sub>(110) surface. A special focus has been given to the dispersionless density functional dIDF and the dIDF+D<sub>as</sub> construction for the total interaction energy (K. Pernal, R. Podeswa, K. Patkowski, and K. Szalewicz, *Phys. Rev. Lett.* **109** (2009) 263201), where D<sub>as</sub> is an effective inter-atomic pairwise functional form for the dispersion. Likewise, the performance of Symmetry-Adapted Perturbation Theory (SAPT) method is evaluated, where the interacting monomers are described by density functional theory (DFT) with the dIDF, PBE, and PBE0 functionals. Our benchmarks include CCSD(T)-F12b calculations and comparative analysis on the nuclear bound states supported by the He-cluster potentials. Moreover, intra- and inter-monomer correlation contributions to the physisorption interaction are analyzed through the method of increments (H. Stoll, *J. Chem. Phys.* **97** (1992) 8449) at CCSD(T) level of theory. This method is further applied in conjunction with a partitioning of the Hartree-Fock interaction energy to estimate individual interaction energy components, comparing them with those obtained using the different SAPT(DFT) approaches. The cluster size evolution of dispersionless and dispersion-accounting energy components is then discussed, revealing the reduced role of the dispersionless interaction and intra-monomer correlation when the extended nature of the surface is better accounted for. On the contrary, both post-Hartree-Fock and SAPT(DFT) results clearly demonstrate the high transferability character of the effective pairwise dispersion interaction whatever the cluster model is. Our contribution also illustrates how the method of increments can be used as a valuable tool not only to achieve the accuracy of CCSD(T) calculations using large cluster models, but also to evaluate the performance of SAPT(DFT) methods for the physically well-defined contributions to the total interaction energy. Overall, our work indicates the excellent performance of a dIDF+D<sub>as</sub> approach in which the parameters of the dispersion function are optimized using the smallest cluster model of the target surface. It also paves the way for further assessments of the dIDF+D<sub>as</sub> approach including periodic boundary conditions as a cost-efficient and accurate method to treat van der Waals adsorbate-surface interactions.

**Keywords:** Symmetry-Adapted Perturbation-Theory; Post-Hartree-Fock Method of Increments; Helium-Surface Potential; Dispersionless Density-Functional-Theory; van der Waals Adsorbate-Surface Interactions; Titanium Dioxide

## I. INTRODUCTION

Nowadays, the first-principle determination of He-surface interaction potentials is a subject of growing interest. This is partly due to the development of low energy helium atom diffraction<sup>1</sup> as an experimental technique to probe surface structure with high resolution, without being perturbed from surface charging and damaging problems. The interest on the He-surface interaction problem has been renewed after the appearance of the helium droplet mediated deposition technique as an experimental tool for the cold attachment of embedded molecules and clusters on surfaces.<sup>2</sup> The microscopic understanding of this process requires quantum dynamical simulations,<sup>3</sup> using realistic He-surface interaction potentials. Last but not least, the physisorption of a helium atom on a surface represents an ultimate case of dispersion-dominated van der Waals (vdW) adsorbate-surface interaction, making it a good test to address more general problems in surface science where vdW interactions play a significant role. Both as a technological relevant surface<sup>4-6</sup> and a prototype transition metal-oxide substrate with well-characterized properties,<sup>4</sup> we have chosen the perfect rutile  $\text{TiO}_2(110)-(1\times 1)$  surface.<sup>3,7</sup>

Naturally, cost-efficient calculations of He-surface interaction potentials should use standard periodic electronic structure codes (*e.g.*, CRYSTAL<sup>8</sup> and VASP<sup>9</sup>), in which methods based on density functional theory (DFT), and Hartree-Fock (HF) or hybrid HF/DFT approaches are implemented using either localized atom-centered Gaussian-type orbitals or plane-wave-based one-electron basis sets. Since these methods are not appropriate to describe the long-range correlation effects in weakly bound systems, the inclusion of long-range dispersive corrections (referred to as +D corrections) has become one of the current standards in DFT-based calculations when physisorption problems are addressed (for a recent review, see Ref. 10). Early attempts to correct the asymptotic behavior of DFT-based approaches when applied to weakly bound complexes date back to the mid-nineties.<sup>11,12</sup> Thus, DFT+D schemes that include semiempirical functions of inter-atomic distances and higher order terms for the dispersion energy, were first proposed by Gianturco and collaborators,<sup>11</sup> and more recently, by Wu and Yang,<sup>13</sup> and by Grimme and collaborators (*e.g.*, the DFT-D2<sup>14</sup> and DFT-D3<sup>15</sup> approaches). Another DFT+D strategy consists in adding the density dependent dispersion correction by Steinmann and Corminboeuf,<sup>16</sup> using the non-local correlation (vdW-DF) functional of Langreth and co-workers,<sup>17</sup> or intermolecular perturbation theory

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3 in a localized orbital basis.<sup>18</sup> Also, a corrected DFT method has been recently developed  
4 to include vdW  $C_6/R^6$  terms using maximally localized Wannier functions and applied to  
5 physisorption problems in metal surfaces (see, *e.g.*, Ref. 19). The choice of the exchange-  
6 correlation functional in DFT+D schemes can be crucial to avoid overbinding effects. This  
7 way, the application of the localized molecular orbital decomposition analysis (LMO-EDA)<sup>20</sup>  
8 by Jordan and collaborators<sup>21</sup> has illustrated how the overestimation of attractive induction  
9 energy components with standard DFT-based methods (typically attributed to deficiencies  
10 in the exchange functional) can compensate the shortcomings in the dispersion contribution,  
11 giving rise to overbinding effects in DFT+D constructions. Early studies of rare gas cluster  
12 ions<sup>22,23</sup> demonstrated the major influence of the chosen exchange functional in DFT-based  
13 treatments on the ability to get interaction energies in reasonable agreement with higher  
14 level *ab initio* methods, enabling at the same time the calculation of multi-dimensional po-  
15 tential energy surfaces for quantum dynamics simulations.<sup>12</sup> In particular, the necessity of  
16 including partial or full “exact” single determinant exchange to avoid overbinding effects  
17 was clearly emphasized.<sup>22</sup> Aimed to avoid over-corrections of the inter-monomer interac-  
18 tions in DFT+D schemes, the key idea put forward very recently by Pernal et al.<sup>24</sup> (see  
19 also Refs. 25–27) consisted in designing a functional (the so-called dlDF functional) which  
20 accounts for the dispersionless interaction energy only so that the dispersion corrections can  
21 be safely added later. The dlDF functional is a hybrid meta exchange-correlation functional  
22 similar to M05-2X by Zhao, Schultz, and Truhlar<sup>28</sup> from which it differs in the number  
23 and values of the DFT parameters. These parameters are optimized using a training set of  
24 weakly bound dimers to reproduce benchmark dispersionless interaction energies. To the  
25 best of our knowledge, the performance of the dlDF approach in a prototypical He-surface  
26 interaction problem is evaluated here for the first time.

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46 Another possibility to describe weakly interacting complexes consists in calculat-  
47 ing inter-monomer interaction energies through Symmetry-Adapted Perturbation Theory  
48 (SAPT).<sup>29,30</sup> It has the advantage of providing a decomposition of the total interaction  
49 energy into physical meaningful electrostatic, exchange-repulsion, induction and dispersion  
50 contributions. When combined with a DFT description of the monomer properties, these  
51 SAPT(DFT) approaches become particularly efficient.<sup>31–33</sup> The SAPT(DFT) method was  
52 simultaneously developed by Misquitta and collaborators<sup>32</sup> (see also Ref. 33), computed in  
53 the SAPT program, and by Hesselmann and Jansen,<sup>31</sup> who implemented the method into  
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3 the MOLPRO package.<sup>34</sup> The application of both implementations to weakly bound systems,  
4 for which dispersion dominates the attractive interaction energy component, provided inte-  
5 raction energies of similar accuracy to those obtained with the CCSD(T) method but with  
6 much better scaling properties with the basis set size (*e.g.*, see, Ref. 35), in particular when  
7 the density-fitting versions of the method are applied.<sup>36,37</sup> Very recently, the SAPT(DFT)  
8 method has been applied to adsorbate-surface interaction problems.<sup>7,38</sup> In this work, we  
9 evaluate the performance of both dlDF and the SAPT(DFT) method using different DFT  
10 approaches in dealing with the physisorption interaction of helium with clusters modeling  
11 the TiO<sub>2</sub>(110) surface.  
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19 For molecular systems, the reliability of dispersion-accounting methods such as the  
20 dlDF+D approach is ideally evaluated through the benchmarking with coupled-cluster sin-  
21 gles, doubles and noniterative triples [CCSD(T)] calculations using large basis sets or ex-  
22 trapolations to the complete basis set limit. The convergence of the CCSD(T) correlation  
23 energy contribution with respect to the basis set size can be accelerated when functions  
24 depending explicitly on the inter-electronic coordinates are introduced, for example through  
25 the frozen-geminal (F12) CCSD(T)-F12x ( $x = a, b$ ) approaches.<sup>39,40</sup> During the last years,  
26 the implementation of the density-fitting approach, has made possible to perform explicitly  
27 correlated CCSD(T)-F12 calculations for small molecular systems. Naturally, the appli-  
28 cation of these methods is still not feasible using fully periodic models of the He-surface  
29 extended system. Concerning the physisorption on non-conducting surfaces, reference inte-  
30 raction energies with periodic models can be obtained at local second-order Möller-Plesset  
31 (LMP2) level of theory, as implemented in the CRYSCOR code by Pisani et al.,<sup>41</sup> using the  
32 reference crystalline HF orbitals that are provided by a periodic Gaussian-type orbitals  
33 calculation with the CRYSTAL code.<sup>8</sup> This method has been recently applied to determine  
34 the He-MgO(100) potential interaction.<sup>42</sup> Using the projector-augmented-wave method and  
35 plane-wave representations, implementations of non-local *canonical* MP2 and coupled-cluster  
36 approaches were introduced,<sup>43,44</sup> but no He-surface potentials have been reported yet. Al-  
37 ternatively, a finite cluster *ansatz* can be adopted to include post-HF effects using coupled-  
38 cluster approaches on top of the periodic HF calculations, as applied with the method of  
39 increments. This method was originally developed by Stoll<sup>45</sup> and successfully applied to ac-  
40 count for correlation effects in many extended systems such as diamond,<sup>45</sup> bulk rutile,<sup>46</sup> and  
41 carbon-based surfaces.<sup>47</sup> The method of increments has been very recently applied by Paulus  
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3 and collaborators<sup>48,49</sup> and Staemmler<sup>50</sup> to adsorption<sup>49</sup> and physisorption problems<sup>50,51</sup> using  
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5 both hydrogen-terminated clusters<sup>51</sup> and point-charge-based cluster embedding.<sup>49,50</sup>  
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7 Aimed to include post-HF effects using embedded cluster models, the “cluster-in-solid”  
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9 technique<sup>52</sup> was developed and implemented as an interface<sup>53</sup> between the CRYSTAL code and  
10 the MOLPRO package of *ab initio* programs.<sup>34</sup> Very recently, this method has been successfully  
11 applied to get the correlated band structure of bulk rutile TiO<sub>2</sub>.<sup>54</sup> Later on, it was applied  
12 to add correlation corrections at CCSD(T) level of theory into the interaction between a  
13 helium atom and an embedded cluster modeling the TiO<sub>2</sub>(110) surface.<sup>7</sup> Linear-dependence  
14 problems in the periodic HF calculations with Gaussian-type orbitals prevented from using  
15 the specially diffuse basis set that is necessary to characterize the He-surface interaction.  
16 Nonetheless, the similarity between the correlation energies obtained using embedded cluster  
17 models with smaller basis sets and the hydrogen-terminated cluster counterparts, indicated  
18 the suitability of using hydrogen-saturated clusters to get insights into the physisorption  
19 interaction at higher levels of *ab initio* theory.  
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28 In this study, hydrogen-saturated cluster models of the TiO<sub>2</sub>(110) surface are used for  
29 the benchmarking of dispersionless and dispersion-accounting approaches. Previous periodic  
30 calculations using the Perdew-Burke-Ernzerhof (PBE) functional<sup>55</sup> and an electronic  
31 basis set tailored to minimize the basis set superposition error (BSSE), yield a short and  
32 medium-range interaction He-surface potential in very reasonable agreement with LMP2  
33 results obtained with a large cluster of stoichiometry (TiO<sub>2</sub>)<sub>9</sub>(H<sub>2</sub>O)<sub>16</sub>. In contrast, the  
34 PBE+D2 scheme produced clear overbinding effects. The first part of this paper is aimed  
35 to understand the fundamental reasons of this behavior using the LMO-EDA scheme of  
36 Su and Li.<sup>20</sup> More importantly, using a sequence of clusters, this work is aimed to evaluate  
37 the performance of both the dlDF+D approach proposed by Pernal et al.<sup>24</sup> and the  
38 SAPT(DFT) method<sup>31,32</sup> using different DFT approaches to account for the dispersionless  
39 and dispersion components of the interaction. For this purpose, reference interaction energies  
40 are obtained through CCSD(T)-F12b, CCSD(T), and LCCSD(T) calculations depending on  
41 the cluster size. Our benchmarking is not only performed on interaction energies but also  
42 on the nuclear bound states and corresponding vibrational frequencies. Further physical  
43 insights into the physisorption interaction and additional benchmarking are achieved by  
44 applying the methods of increments at CCSD(T), LCCSD(T), LMP2 and MP2 levels of  
45 theory. As will be shown, this method allows to separate the correlation contribution to  
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3 the total interaction energy into inter- and intra-monomer parts. When combined with  
4 a LMO-EDA-based decomposition of the HF interaction energy, estimates for the electro-  
5 static, exchange-repulsion, induction and dispersion energy components can be obtained  
6 and compared with the SAPT(DFT) counterparts, using different density functionals for  
7 the monomers. Next, the dispersion energies for a given cluster size are fitted to the dis-  
8 persion functional form proposed by Podeszwa and Szalewicz<sup>26</sup> (denoted as  $D_{as}$ ), and the  
9 dispersion energies directly calculated with the SAPT(DFT) method (hereafter referred to  
10 as  $D$ ) are compared with those extracted with the  $D_{as}$  functional using clusters of increasing  
11 size. The method of increments is also applied to prove the transferability of the effective  
12 pairwise dispersion interactions in expanding the cluster model. Moreover, the combined  
13 application of SAPT(DFT) and supermolecular DFT and post-HF methods with LMO-EDA  
14 and incremental-based decompositions has allowed us to explain *how and why* the physisorp-  
15 tion interaction is modified by the modeling of the rutile surface with clusters of increasing  
16 size, with the emphasis on the relative weights of dispersion and dispersionless contributions  
17 as well as intra- and inter-monomer correlation effects.

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19 This paper is structured as follows. Section II contains the computational details and  
20 a brief outline of the applied methods. In the first part of Section III, the results for the  
21 He-(TiO<sub>2</sub>)(H<sub>2</sub>O)<sub>3</sub> complex are presented and discussed, with the main purpose of under-  
22 standing the characteristics of the interaction and assessing the performance of the applied  
23 methodologies. The second part of this section extends the application of SAPT(DFT) and  
24 dlDF+ $D$  approaches as well as the method of increments to larger clusters modeling the  
25 TiO<sub>2</sub>(110) surface. Section IV closes with the conclusions.

## 26 27 28 29 30 31 32 33 34 35 36 37 38 39 40 41 42 43 44 45 **II. APPLIED METHODS AND COMPUTATIONAL DETAILS**

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47 All the electronic structure calculations have been performed with the MOLPRO suite of  
48 programs.<sup>34</sup> For this purpose, we have implemented the LMO-EDA scheme of Su and Li<sup>20</sup>  
49 in the MOLPRO package using the MATROP facility. To implement the dlDF functional,<sup>24</sup> we  
50 modified the parameters of the M05-2X functional.<sup>28</sup> On the other hand, the nuclear bound-  
51 state calculations were accomplished using the Truhlar-Numerov procedure,<sup>56</sup> as described  
52 below.



## A. Cluster models

The hydrogen-saturated clusters chosen to characterize the He–TiO<sub>2</sub>(110) interaction are represented in Fig. 1. They are very similar to those used by Rittner et al.<sup>57</sup> to model the N<sub>2</sub>–TiO<sub>2</sub>(110) system. We consider the He-cluster interaction as a function of the vertical height  $z$  above the most stable adsorption site (the five-coordinated Ti(5f) atom), with a  $z$  grid in the range [2.8–10.0] Å. As already mentioned in Ref. 7, the number of hydrogen atoms was chosen to make the clusters electrically neutral, adding them along the O–Ti direction in the extended system. As in Ref. 7, we define the main interaction region as that formed by the Ti(5f) adsorption site and their nearest in plane surface and bridging oxygen atoms (denoted as O<sub>p</sub> and O<sub>b</sub> in Fig. 1). The geometry of the clusters has been frozen to the experimental geometry of the TiO<sub>2</sub>(110)–(1×1) surface.<sup>58</sup> This approach is justified by the fact that only minor surface ion displacements upon helium physisorption were observed (0.0013 Å as much).<sup>3</sup> The cluster of stoichiometry (TiO<sub>2</sub>)(H<sub>2</sub>O)<sub>3</sub> (denoted as  $\mathcal{C}_1$ , see Fig. 1) has served to characterize the interaction of the helium atom with the Ti(5f) adsorption site and its nearest (in-plane and sub-surface) oxygen neighbors from the TiO<sub>2</sub>(110) surface. The relatively small size of this cluster has allowed us to perform all-electron explicitly correlated CCSD(T)-F12b calculations for benchmarking purposes. The next cluster model (the  $\mathcal{C}_2$  cluster with stoichiometry (TiO<sub>2</sub>)<sub>3</sub>(H<sub>2</sub>O)<sub>5</sub>, see Fig. 1) is obtained by extending the  $\mathcal{C}_1$  cluster along the Ti(5f) row (i.e., the crystallographic [001] direction). This way, our analysis is extended to the physisorption interaction of helium with the two next-nearest Ti(5f) atoms and their respective coordination shells of oxygen atoms. The largest cluster of stoichiometry (TiO<sub>2</sub>)<sub>9</sub>(H<sub>2</sub>O)<sub>7</sub> (the  $\mathcal{C}_3$  cluster, see Fig. 1) has enabled to characterize the role of the bridging oxygen atoms sticking out of the surface<sup>7</sup>. It involves the expansion of the  $\mathcal{C}_2$  cluster along the crystallographic [ $\bar{1}00$ ] direction. As shown in Fig. 1, the  $\mathcal{C}_1$  and  $\mathcal{C}_2$  clusters model the top molecular layer only while the largest  $\mathcal{C}_3$  cluster partially accounts for the interaction with atoms from the first under-surface molecular layer.

## B. Supermolecular calculations

Using the supermolecular approach, the interaction energy between a single helium atom and the cluster modeling the  $\text{TiO}_2(110)$  surface  $\mathcal{C}(\text{TiO}_2)$  is expressed as,

$$E_{\text{int}}^{\text{tot}} = E_{\text{He}-\mathcal{C}(\text{TiO}_2)} - E_{\text{He}} - E_{\mathcal{C}(\text{TiO}_2)}, \quad (1)$$

where  $E_{\text{He}-\mathcal{C}(\text{TiO}_2)}$  is the total energy of the  $\text{He}-\mathcal{C}(\text{TiO}_2)$  complex, and  $E_{\text{He}}$  and  $E_{\mathcal{C}(\text{TiO}_2)}$  are the energies of the monomers. We used the counterpoise procedure proposed by Boys and Bernardi<sup>59</sup> to account for the intra-molecular basis set superposition error (BSSE). To estimate the dispersionless contribution to the total interaction energy, we have used the dI DF functional developed by Szalewicz and collaborators.<sup>24</sup> The parameters of the M05-2X functional<sup>28</sup> as present in MOLPRO were modified to implement the dI DF functional. This allowed us to calculate dI DF interaction energies and to perform SAPT(dI DF) calculations. The implementation was tested by reproducing the same dI DF interaction energies presented as a training set data in Ref. 24.

Using wave-function-based post-Hartree-Fock methods,  $E_{\text{int}}^{\text{tot}}$  is partitioned as,

$$E_{\text{int}}^{\text{tot}} = E_{\text{int}}^{\text{HF}} + E_{\text{int}}^{\text{corr}} \quad (2)$$

where  $E_{\text{int}}^{\text{HF}}$  and  $E_{\text{int}}^{\text{corr}}$  represent the Hartree-Fock and correlation contributions to the total interaction energies. Along with standard correlation methods, the term  $E_{\text{int}}^{\text{corr}}$  has been calculated with the method of local increments.<sup>45</sup> Detailed descriptions of this method can be found in Refs. 47,60,61 and, as applied to adsorption energies, in Refs. 50,51. In this method, the correlation energy  $E^{\text{corr}}$  is first expressed as a cumulant expansion of contributions (increments) from groups of occupied localized orbitals on different atoms (or group of atoms),<sup>46</sup>

$$E^{\text{corr}} = \sum_A \epsilon(A) + \sum_{A<B} \Delta\epsilon(A-B) + \sum_{A<B<C} \Delta\epsilon(A-B-C), \quad (3)$$

where  $A, B, C, \dots$  denote localized orbital groups on atoms (or atomic groups) indexed by  $A, B, C, \dots$ , respectively. The one-body term  $\epsilon_A$  represents the correlation energy obtained when the electrons from the localized orbital group  $A$  are correlated, freezing the rest of localized orbitals. The two-body increment is defined as the non-additive part of the correlation energy associated to the simultaneous correlation of the electrons from the

localized orbital groups  $A$  and  $B$ :

$$\Delta\epsilon(A - B) = \epsilon(A - B) - \epsilon(A) - \epsilon(B). \quad (4)$$

Similarly, the three-body increments  $\Delta\epsilon(A - B - C)$  is calculated by subtracting the one- and two-body energy increments involving the orbital groups  $A$ ,  $B$  and  $C$  from the energy obtained upon the simultaneous correlation of the electrons occupying the same localized orbitals (see, *e.g.*, Ref. 46).

To obtain the correlation contribution to the interaction energy  $E_{\text{int}}^{\text{corr}}$ , it is necessary to calculate the increment modifications due to the interaction between the monomers, including all the orbital groups from the cluster  $\mathcal{C}(\text{TiO}_2)$  modeling the  $\text{TiO}_2(110)$  surface that are interacting with the helium atom. In this particular case, the most important orbital groups of the surface would be the occupied localized orbitals generated from the  $2s$  and  $2p$  atomic orbitals centered at the oxygen atoms in the vicinity of the adsorbed species. Using a similar notation to that of Ref. 51, the one-body increment contribution to  $E_{\text{int}}^{\text{corr}}$  from the group of localized orbitals centered on the  $i$ th oxygen atom of the  $\mathcal{C}(\text{TiO}_2)$  cluster can be written as,

$$\eta(\text{O}^{(i)}) = \epsilon^{\text{He}-\mathcal{C}(\text{TiO}_2)}(\text{O}^{(i)}) - \epsilon^{\mathcal{C}(\text{TiO}_2)}(\text{O}^{(i)}). \quad (5)$$

Similarly, the two-body increment contribution to  $E_{\text{int}}^{\text{corr}}$  arising from the groups centered on the  $i$ th and  $j$ th oxygen atoms is calculated as,

$$\eta(\text{O}^{(i)} - \text{O}^{(j)}) = \Delta\epsilon^{\text{He}-\mathcal{C}(\text{TiO}_2)}(\text{O}^{(i)} - \text{O}^{(j)}) - \Delta\epsilon^{\mathcal{C}(\text{TiO}_2)}(\text{O}^{(i)} - \text{O}^{(j)}). \quad (6)$$

When the increments involve groups of both the surface and the adsorbate,  $\eta(A - B)$  and  $\Delta\epsilon(A - B)$  share the same definition. Therefore,  $E_{\text{int}}^{\text{corr}}$  can be expanded as,

$$E_{\text{int}}^{\text{corr}} = \eta(\text{He}) + \sum_i \eta(\text{O}^{(i)}) + \sum_i \eta(\text{He}-\text{O}^{(i)}) + \sum_{i < j} \eta(\text{O}^{(i)} - \text{O}^{(j)}) + \sum_{i < j} \eta(\text{He}-\text{O}^{(i)} - \text{O}^{(j)}) + \dots$$

It should be noticed that all the increments are calculated using the full basis set so that  $E_{\text{int}}^{\text{corr}}$  becomes counterpoise corrected. Along with the incremental scheme, benchmark calculations have been performed using the CCSD(T)-F12b method for the  $\text{He}-\text{C}_1$  complex, as discussed below. In both cases, the semi-core  $3sp$  orbital from the Ti atom were kept frozen and not correlated.

### C. Calculations using Symmetry-Adapted Perturbation Theory

In this work, the SAPT method has been used with a DFT description of the helium atom and the cluster modeling the rutile surface, as implemented in the DFT-SAPT code.<sup>31,36</sup> We will refer this method as SAPT(DFT), replacing DFT by the functional name when a particular functional is considered. An independent implementation of this method by Misquitta *et al.* has been also reported.<sup>33</sup> This method enables to calculate the interaction energy as a sum of individual first- and second-order interaction terms, namely electrostatic  $E_{elst}^{(1)}$ , exchange  $E_{exch}^{(1)}$ , induction  $E_{ind}^{(2)}$ , and dispersion  $E_{disp}^{(2)}$  terms, along with their respective exchange corrections ( $E_{exch-ind}^{(2)}$  and  $E_{exch-disp}^{(2)}$ ). The  $\delta$ (HF) estimate<sup>36,62</sup> of the higher-order induction plus exchange-induction contributions has been added to the DFT-SAPT interaction energy. The density-fitting (DF) technique has been used to calculate SAPT(DFT) interaction energies of the He- $C_2$  and He- $C_3$  complexes (see Fig. 1), as implemented within MOLPRO by Hesselmann, Jansen and Schütz.<sup>36</sup> No DF was employed for the SAPT(DFT) calculations of the He- $C_1$  complex. Molecular orbitals for SAPT(DFT) were obtained from Kohn-Sham (KS) calculations using the PBE,<sup>55</sup> PBE0,<sup>63</sup> and dIDF<sup>24</sup> functionals.

In all calculations, the exchange-correlation ( $xc$ ) potentials were asymptotically corrected following the gradient-regulated scheme of Grüning *et al.*<sup>64</sup>. As described in Ref. 64, this correction to the exchange-correlation ( $xc$ ) potential was devised to enforce the right long-range behaviour (i.e., decaying as  $-1/r$ ), shifting at the same time the highest occupied molecular orbital (HOMO) energy  $\epsilon_{HOMO}$  to the ionization potential (IP) value  $I_p$  (i.e., satisfying the condition  $\epsilon_{HOMO} = -I_p$ ). To achieve this goal<sup>64</sup>, the potential is shifted by  $I_p + \epsilon_{HOMO}$ . The eigenvalues of the KS orbitals (in particular the unoccupied orbitals) supported by exchange-correlation ( $xc$ ) potentials with wrong asymptotic decay are usually too high, as occurs with the HOMO eigenvalue. The inaccuracies in the KS eigenvalue spectra (e.g., the differences between the occupied and unoccupied energies) are naturally reflected in the response functions employed to determine the induction and dispersion energies. The HOMO eigenvalues of the monomers have also a pronounced influence on the exchange-repulsion component (for details see Ref. 65). Likewise, asymptotically corrected  $xc$  potential are necessary to get accurate tails of the monomer densities, and then of the charge overlaps entering into the electrostatic energy determination<sup>65</sup>. This way, it has been shown<sup>33,65</sup> that the inclusion of the asymptotic correction in SAPT(DFT) calculations

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3 improve much the interaction energies, making them less dependent on the density functional  
4 used. We employed ionization potential (IP) values of 0.9036 and 0.3491 a.u. for atomic He  
5 and the fragment modeling the rutile surface, respectively, and the HOMO energies (resulting  
6 from the corresponding standard DFT calculations on the monomers) to calculate the shift  
7 parameter  $I_p + \epsilon_{\text{HOMO}}$ . The IP values are those reported in the NIST Chemistry Web Book<sup>66</sup>  
8 for atomic helium and (gas-phase)  $\text{TiO}_2$ . Another alternative would have been to use either  
9 the  $\text{TiO}_2(110)$  surface work-function (0.2021 a.u. from Ref. 67) or the Koopmans' theorem  
10 IP values from HF computations at each cluster size as in Ref. 38. However, a very similar IP  
11 value (0.3409 a.u.) to the  $\text{TiO}_2$  gas-phase value was obtained for the  $\mathcal{C}_1$  cluster at CCSD(T)  
12 level while the HF counterpart was about 0.08 a.u. too high. On the other hand, the  
13 Hartree-Fock IP value varied very little in going from the  $\mathcal{C}_1$  to the  $\mathcal{C}_3$  cluster (to within 7%)  
14 and a similar trend can be expected for the correlated IP counterpart. We have also tested  
15 that the employment of the  $\text{TiO}_2(110)$  work-function as the IP value for the SAPT(DFT)  
16 calculations on the He- $\mathcal{C}_3$  complex make the interaction energies about 15% less attractive  
17 and worsen the agreement with the best benchmarking.

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The adiabatic local density approximation (ALDA)<sup>68</sup> has been used in all SAPT(PBE)  
calculations. Since the PBE0 and dI DF exchange potentials contain 25% and 61.44% of  
single-determinant exchange, a hybrid-ALDA kernel with the same percentage of Fock ex-  
change has been employed to get both the static and frequency-dependent response functions  
in SAPT(PBE0) and SAPT(dI DF) calculations for the He- $\mathcal{C}_1$  complex. However, the current  
density-fitting implementation of the SAPT(DFT) method within MOLPRO can only deal with  
pure ALDA kernels in computing the response functions. Therefore, this approximation is  
also used in SAPT(DFT) calculations with hybrid functionals to calculate the dispersion  
energy component for the He- $\mathcal{C}_2$  and He- $\mathcal{C}_3$  complexes. To stress the use of a pure ALDA  
kernel in conjunction with the hybrid PBE0 and dI DF functionals in the dispersion energies,  
the SAPT calculations will be referred to as SAPT(ALDA/PBE0) and SAPT(ALDA/dI DF),  
respectively.

#### D. Localized molecular orbital energy decomposition analysis

The localized molecular orbital energy decomposition analysis (LMO-EDA) of Su and Li<sup>20</sup>  
has been used to separate the total interaction energy obtained from supermolecular HF and

PBE calculations into electrostatic, exchange-repulsion (or exchange), induction, and inter-monomer correlation contributions, comparing them to those obtained from SAPT-based calculations. This scheme has been implemented in the MOLPRO package through the MATROP module. The electrostatic and exchange contributions are calculated by freezing the occupied monomer molecular orbitals. The former is defined as the Coulomb interaction between the electrons and nuclei of the two monomers and the explicit expression is equivalent to that implemented for the SAPT  $E_{elst}^{(1)}$  component (Eq. (2) of Ref. 36). Using the LMO-EDA partitioning in DFT, the exchange contribution is defined as,

$$E_{exch} = E_x[\rho_{\text{He}} + \rho_{\mathcal{C}(\text{TiO}_2)}] - \{E_x[\rho_{\text{He}}] + E_x[\rho_{\mathcal{C}(\text{TiO}_2)}]\}, \quad (7)$$

where  $\rho_{\text{He}}$  and  $\rho_{\mathcal{C}(\text{TiO}_2)}$  denote KS densities of the helium atom and the cluster modeling the rutile surface, respectively. On the other hand, the calculation of the repulsion contribution requires the mutual orthogonalization of the occupied monomer molecular orbitals. This contribution is defined as the difference between the KS energy (excluding the correlation term) obtained with the first-order reduced density matrices calculated with the orthogonalized molecular orbitals, and the counterpart using the original molecular orbitals. For example, the exchange contribution to the repulsion term ( $E_{rep}^{exch}$ ) is given as,

$$E_{rep}^{exch} = E_x[\rho_{\text{He}}^{ortho} + \rho_{\mathcal{C}(\text{TiO}_2)}^{ortho}] - E_x[\rho_{\text{He}} + \rho_{\mathcal{C}(\text{TiO}_2)}], \quad (8)$$

where the subscript ‘‘ortho’’ indicates that the densities have been obtained after orthogonalizing the monomer molecular orbitals. Besides the changes in the electrostatic terms, the repulsion contribution also accounts for the modification in the one-particle kinetic energies due to the orthogonalization procedure.

Using the LMO-EDA scheme, the KS orbital relaxations due to the inter-monomer interaction itself are accounted for within the induction and inter-monomer correlation contributions, with the latter expressed as,

$$E_{corr}^{inter} = E_c[\rho_{\text{He}+\mathcal{C}(\text{TiO}_2)}] - \{E_c[\rho_{\text{He}}] + E_c[\rho_{\mathcal{C}(\text{TiO}_2)}]\}, \quad (9)$$

where  $\rho_{\text{He}}$  and  $\rho_{\mathcal{C}(\text{TiO}_2)}$  denote KS densities of the helium atom and the cluster modeling the rutile surface, respectively, and  $\rho_{\text{He}+\mathcal{C}(\text{TiO}_2)}$  is the electron density minimizing the He- $\mathcal{C}(\text{TiO}_2)$  KS energy. This correlation contribution has been identified with the short-range contribution to the dispersion energy,<sup>20,21</sup> adopting the same interpretation here. In comparing the

LMO-EDA and SAPT(DFT) results, the dispersion and exchange-dispersion SAPT(DFT) contributions were combined as were the induction, exchange-induction and  $\delta$ (HF) contributions. Similarly, the exchange and repulsion LMO-EDA terms were added to calculate the net exchange-repulsion energy contribution.

## E. Electronic Basis Sets and Orbital Localization

### 1. Basis Sets in DFT, LCCSD(T), and SAPT(DFT) Calculations

Unless otherwise specified, supermolecular DFT/LCCSD(T) and SAPT(DFT) calculations were performed using an augmented triple- $\zeta$  valence all-electron basis set<sup>69</sup> for O and Ti atoms (the so-termed ext-TVAE\*\* basis<sup>7</sup>). This is a dual basis that includes additional diffuse functions for the O and Ti atoms within the main interaction region (see Ref. 7 for the details). Both helium and hydrogen atoms were described with the standard aug-cc-pVTZ basis set of Dunning and collaborators.<sup>70,71</sup> The density fitting technique was used in both dIDF and SAPT-based calculations on the He-C<sub>2</sub> and He-C<sub>3</sub> complexes, as implemented within MOLPRO.<sup>36,72</sup> The DF of Coulomb and exchange J/K integrals has used the auxiliary basis set developed for the cc-pVTZ basis by Weigend,<sup>73</sup> while the aug-cc-VTZ/MP2Fit basis<sup>74</sup> was employed to fit the integrals containing virtual orbitals. The suitability of both the ext-TVAE\*\* basis and the density fitting approximation was tested for either DFT or SAPT(DFT) calculations as follows:

- Using the additional diffuse functions of the ext-TVAE\*\* basis for all the atoms composing the C<sub>2</sub> cluster, the He-C<sub>2</sub> interaction energies deviated from those obtained with the dual basis by less than 2%. Similarly, dispersionless and dispersion SAPT(DFT) interaction energies agreed with those calculated with the dual basis to within 1% and 4%, respectively. On the other hand, the omission of these functions for the atoms within the main interaction region resulted in interaction energy differences amounting to about 6% and 15% in dIDF and SAPT(dIDF) calculations, respectively.
- The error in dIDF interaction energies due to the density-fitting was below 0.1 cm<sup>-1</sup> for the He-C<sub>2</sub> complex at the vdW minimum, with the relative error being below 1% at the relevant range of inter-monomers He-Ti(5f) distances. Similar errors were found in the SAPT(DFT) energy components upon the density-fitting approximation with the

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3 exception of the dispersion terms when hybrid DFT functionals are used (see Section  
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5 III).  
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## 9 2. Basis Sets in Explicitly Correlated Coupled-Cluster Calculations

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12 Following an accurate study of transition-metal containing molecules, explicitly correlated  
13 CCSD(T) calculations were performed using the CCSD(T)-F12b method<sup>40</sup>, with the cc-  
14 pVnZ-F12 ( $n=T, Q$ ) series of orbital basis sets for H, He and O<sup>75</sup> and the aug-cc-pVnZ  
15 ( $n=T, Q$ ) basis of Dunning and collaborators<sup>70,71</sup> for Ti. These orbital basis sets will be  
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17 termed AVTZ and AVQZ for  $n =T$  and  $Q$ , respectively. Density-fitting within the correlation  
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19 treatment used the aug-cc-pVnZ/MP2Fit<sup>74</sup> ( $n=T, Q$ ) auxiliary basis sets for H, He and O  
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21 atoms, respectively. The Fock matrix fitting was carried out with the def2-QZVPP/JKFit<sup>76</sup>  
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23 auxiliary basis sets for Ti and He atoms, and the cc-pVnZ/JKFit ( $n=T, Q$ ) sets for the  
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25 other elements. The complementary auxiliary basis approach<sup>77</sup> for the resolution of the  
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27 identity employed the OptRI auxiliary basis sets matching to the corresponding orbital basis  
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29 sets.<sup>78-80</sup> The DF/MP2Fit fitting basis was used along with the complementary auxiliary  
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31 basis optimized for the aug-cc-pVnZ ( $n=T, Q$ ) sets centered on the Ti atom (see Ref. 81).  
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33 As in previous studies of transition-metal containing systems,<sup>80,82</sup> the value of the geminal  
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35 Slater exponent was fixed to  $1.4 a_0^{-1}$ .  
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37 We extrapolated the CCSD(T)-F12b correlation energies to the complete basis set limit  
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39 (CBS) by applying the scheme proposed by Helgaker and co-workers<sup>83</sup> for standard CCSD(T)  
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41 calculations,

$$42 \quad E_n^{corr} = E_{CBS}^{corr} + A \ell_{max}^{-3} \quad (10)$$

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44 where the angular quantum number  $\ell_{max}$  is simply settled to the  $n$  cardinal number in the  
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46 AVnZ ( $n =T, Q$ ) basis set. The HF interaction energies were fixed to those obtained with  
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48 the largest basis. When applied to very weak interacting dimers with explicitly correlated  
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50 coupled-cluster methods, an extensive analysis by Patkowski<sup>84</sup> has shown that the  $1/n^3$   
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52 scheme provides reliable estimations of the CBS limit. Different extrapolation schemes were  
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54 tested (see, *e.g.*, Ref. 85), providing very similar values. It should be noticed that basis  
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56 set extrapolation schemes specially designed for F12 methods have been developed<sup>86</sup> but  
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58 not yet reported for transition-metal containing systems. In any case, the CCSD(T)-F12b  
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60 interaction energies with the AVQZ basis were found to be very close to the estimated CBS



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limit (see Section III).

### 3. Basis sets and Localization in Calculations with the Method of Increments

The method of increments has been applied to the  $\text{He}-(\text{TiO}_2)(\text{H}_2\text{O})_3$  complex by using the polarized correlation-consistent valence-triple- $\zeta$  basis (cc-pVTZ) of Dunning and collaborators<sup>70,71</sup> for Ti and H, and the aug-cc-pVTZ basis for He and O (to be denoted as (A)VTZ). As in the CCSD(T)-F12b/AVQZ calculations, only the valence electrons were correlated. For the incremental expansion, the occupied valence orbital space was partitioned into six orbital groups that can be attributed to the helium atom and individual (Ti-)O<sub>p</sub>H or (Ti-)O<sub>sub</sub>H<sub>2</sub> groups from the  $\mathcal{C}_1$  cluster model (see Fig. 1). The Foster-Boys localization procedure<sup>87</sup> was employed, as implemented in MOLPRO.<sup>34</sup> Each (Ti-)O<sub>b</sub>H [(Ti-)O<sub>sub</sub>H<sub>2</sub>] atomic group consists of four double occupied orbitals. They are mainly composed of the 2s and 2p atomic orbitals centered at the surface in-plane oxygen anions O<sub>p</sub> (the sub-surface oxygen atom O<sub>sub</sub>). Clearly, titanium atoms also participate in the orbital groups, which is manifested through the hybridization (mainly with Ti(3d) atomic orbitals) during the Foster-Boys localization. Obviously, the adsorbate species subunit consists of a double occupied localized orbital, as generated from the atomic orbital 1s centered at the helium atom. The incremental expansion was truncated after the most important three-body terms. The suitability of this setup was evaluated by comparing the correlation contributions from the incremental expansion ( $-73.128/-47.598$  cm<sup>-1</sup> at  $z = 3.5/3.875$  Å, respectively) to those obtained from full CCSD(T) calculations ( $-72.774/-47.316$  cm<sup>-1</sup>), agreeing to within 0.5%.

The incremental scheme has been also applied to the  $\text{He}-(\text{TiO}_2)_9(\text{H}_2\text{O})_7$  complex by calculating the most important correlation contributions to the interaction energy. In this case, the increments were calculated at MP2, LMP2, LCCSD(T), and CCSD(T) levels of theory. As for the  $\text{He}-(\text{TiO}_2)(\text{H}_2\text{O})_3$  dimer, only the valence electrons were correlated and the counterpoise correction was included. We used the cc-pVDZ basis for H and Ti, and the aug-cc-pVDZ basis for O and He. For the atoms outside the main interaction region<sup>7</sup>, the polarization functions were omitted. For the  $\text{He}-(\text{TiO}_2)(\text{H}_2\text{O})_3$  dimer, the omission of these functions resulted in differences for the main correlation contributions to within 1–4%. Hereafter, this basis will be referred to as (A)VDZ.

## F. Calculation of Nuclear Bound States

To calculate the bound states supported by the He-(TiO<sub>2</sub>)(H<sub>2</sub>O)<sub>3</sub> potential energy curve (PEC), the interaction energies were first fitted to a Morse functional form with a damped long-range  $-f(z; z_0, a) \left(\frac{C_3}{z^3}\right)$  term, where,

$$f(z; z_0, a) = \frac{1}{2} \left\{ 1 + \tanh \left[ \frac{z - z_0}{a} \right] \right\}. \quad (11)$$

Next, the energies and nuclear wave-functions were calculated by numerically solving the one-dimensional Schrödinger equation through the Truhlar-Numerov procedure,<sup>56</sup> using a numerical grid of 100000 points in the range [1.0–1000.0] a.u. The atomic masses were fixed to the following values (in amu):  $m_{\text{Ti}}=46.8671$ ,  $m_{\text{O}}=15.9994$ ,  $m_{\text{H}}=1.00794$ , and  $m_{4\text{He}}=4.002602$ .

## III. RESULTS AND DISCUSSION

The results are organized as follows: The different clusters modeling the rutile surface are shown in Fig. 1. Using the smallest cluster model of stoichiometry (TiO<sub>2</sub>)(H<sub>2</sub>O)<sub>3</sub> (referred to as  $\mathcal{C}_1$ ), Figures 2 and 4 provide the He-cluster interaction energies as a function of the inter-monomer distance. For the sake of clarity, Table I collects the magnitudes characterizing these potential energy curves. Next, the individual components of the interaction at the vdW minimum are schematically represented in Fig. 3. The interaction is further analyzed using correlation increment contributions, which are collected in Table II. Additional benchmarking is obtained by calculating the nuclear bound states supported by the He- $\mathcal{C}_1$  potential energy curves, with the leading vibrational energies and density distributions shown in Table III and Fig. 5, respectively.

The second part of this section extends our study to the larger cluster models of stoichiometries (TiO<sub>2</sub>)<sub>3</sub>(H<sub>2</sub>O)<sub>5</sub> and (TiO<sub>2</sub>)<sub>9</sub>(H<sub>2</sub>O)<sub>7</sub> (denoted as  $\mathcal{C}_2$  and  $\mathcal{C}_3$ , respectively). The potential energies curves of the He- $\mathcal{C}_2$  and He- $\mathcal{C}_3$  complexes are represented in Figs. 6 and 9, while the cluster size evolution of the individual interaction energy components, as calculated with the SAPT(DFT) method, can be followed from Fig. 7. The method of increments has been also applied to the He- $\mathcal{C}_3$  system with the main correlation contributions summarized in Table IV. Finally, the cluster size evolution of dispersionless and dispersion interaction energies depending on the chosen DFT method (within the SAPT(DFT) framework) is graphically represented in Figure 8.

### A. The He-(TiO<sub>2</sub>)(H<sub>2</sub>O)<sub>3</sub> complex: performance of DFT+D approaches

As already mentioned, Figure 2 shows He-(TiO<sub>2</sub>)(H<sub>2</sub>O)<sub>3</sub> interaction energies as a function of the distance between the helium atom and the Ti(5f) cation, using the different methods considered in this work (see also Table I). Focusing first on the PECs obtained with the PBE and PBE+D2 approaches, it can be immediately noticed that the PBE functional provides a very reasonable short- and medium-range potential energy curve, in contrast with the PBE+D2 scheme. Thus, the PBE well-depth and minimum energy positions are very close to those obtained in the conventional CCSD(T)/(A)VTZ calculations. Naturally, due to the inability of the semi-local PBE functional to account for long-range dispersion, the attractive tail of the potential is not described correctly, which is also reflected in the discrepancy between PBE and CCSD(T)/(A)VTZ zero-point energies. The inclusion of the empirical dispersion correction with the Grimme's D2 scheme<sup>14</sup> improves the asymptotic region (at distances larger than  $\sim 5.0$  Å). However, it also causes a very pronounced overbinding effect, with the well-depth being a factor of  $\sim 2.4$  larger than the CCSD(T)-F12b/AVQZ value. Many previous studies have shown that the PBE+D2 scheme works reasonably well in extended systems where the vdW interaction plays a key role (*e.g.*, soft layered materials<sup>88</sup> or water physisorption<sup>51</sup> to name just two). In order to understand the fundamental reasons for the good performance of the PBE approach at the vdW minimum (and the consequent poor behavior of the PBE+D2 construction), we have decomposed the net PBE interaction energy using the LMO-EDA scheme developed by Su and Li<sup>20</sup> (see Section II). The results are shown in Fig. 3 along with those obtained through SAPT(DFT) calculations and the LMO-EDA-based decomposition of the HF interaction energy, with and without adding the correlation contributions extracted from supermolecular CCSD(T)/(A)VTZ calculations with the method of increments, as described below.

As clearly apparent from Fig. 3 (notice the small magnitudes of the electrostatic and induction contributions), the weak physisorption interaction results from the balance of the short-range exchange-repulsion and the attractive long-range dispersion energy term, with the latter dominating at the vdW minimum and at larger inter-monomer distances. Of course, the dispersion energy arises from a pure electron correlation effect so that the HF interaction is repulsive. Adding the exchange-repulsion and dispersion-like contributions to  $E_{\text{int}}^{\text{tot}}$  from CCSD(T) calculations with the method of increments, we get an attractive

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3 interaction of similar magnitude to that obtained with the SAPT(PBE0) approach. The  
4 individual  $E_{exch-rep}$  and  $E_{disp}$  contributions also agree very well to each other, which is  
5 remarkable. From Figure 3, it can be also noticed that the exchange-repulsion is enhanced  
6 when the correlation obtained with the CCSD(T) method is accounted for. Contrarily, when  
7 the correlation is estimated through the PBE approach, the exchange-repulsion component  
8 has a very similar magnitude to the uncorrelated HF counterpart. As apparent from Figure 3,  
9 the good performance of the PBE approach can be attributed to both the recovering of some  
10 part of the short-range dispersion through the inter-monomer correlation contribution, as  
11 identified within the LMO-EDA framework<sup>21</sup> (see Eq. 9), and to a cancellation of errors:  
12 this functional underestimates the exchange-repulsion LMO-EDA energy, and overestimates  
13 the attractive induction contribution, partially compensating for the underestimation of the  
14 dispersion component. It follows that the inclusion of an additional attractive dispersion  
15 term through the DFT-D2 scheme overcorrects the interaction.  
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26 The tendency of DFT methods to overestimate the induction contribution in weakly  
27 bound systems such as water clusters has been noticed previously<sup>21</sup> and attributed to the  
28 self-interaction error in DFT.<sup>89</sup> In contrast to the LMO-EDA of water clusters, however, the  
29 overestimation of the induction He-(TiO<sub>2</sub>)(H<sub>2</sub>O)<sub>3</sub> interaction with the PBE approach is not  
30 accompanied by that of the exchange-repulsion. This could explain one of the reasons why  
31 dispersion-corrected PBE+D approaches work better in water clusters. The larger value  
32 of the electrostatic term with the PBE approach (see Fig. 3) reflects the strongest overlap  
33 between the monomer densities. Hence, a larger magnitude for the exchange-repulsion would  
34 be expected, as opposed to the results shown in Fig. 3. This feature can be attributed to  
35 the exchange functional. This way, the replacement of the PBE exchange functional by the  
36 revPBE variant<sup>90</sup> (which differs by a single parameter to mimic the short-range Hartree-  
37 Fock behavior) modifies the exchange energy to such extent that the net revPBE exchange-  
38 repulsion becomes the largest between those shown in Fig. 3. The same holds true when  
39 the PBE exchange is replaced by the single-determinant counterpart (i.e., from the KS Fock  
40 operator).  
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53 As mentioned in the introduction, the dIDF functional of Pernal et al.<sup>24</sup> was designed to  
54 account for the dispersionless interaction energy only, avoiding over-corrections in DFT+D  
55 constructions. Consequently, as apparent from Fig. 2, the dIDF functional provides a purely  
56 repulsive PEC for the He-(TiO<sub>2</sub>)(H<sub>2</sub>O)<sub>3</sub> interaction and, once the dispersion is accounted for  
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(leading to the dIDF+ $D$  approach, see Fig. 2), we obtain a PEC in very reasonable agreement with the benchmark obtained using the CCSD(T)-F12b method (see also Table I). Since the parameters entering the dIDF functional were not optimized with a training set containing transition-metal atoms, the good performance of the dIDF approach is remarkable.

### B. The He-(TiO<sub>2</sub>)(H<sub>2</sub>O)<sub>3</sub> complex: analysis of the interaction with the method of increments

Let us now analyze the He-cluster interaction from a local viewpoint. This is possible by applying the method of increments, which can distinguish the correlation contributions coming from different cluster regions. Moreover, this method allows to separate between intra- and inter-monomer correlation effects. Selected correlation-energy contributions to the total interaction energy are collected in Table II. Focusing first on the increment contributions at  $z = 3.5$  Å, by far the largest contribution comes from the two-body interaction of the helium atom with the adjacent O<sub>p</sub> atoms located at the surface plane (see Fig. 1). The increments involving the subsurface oxygen atoms that beneath the Ti(5f) adsorption site are very small, only amounting to 6% of the total correlation energy contribution. The total attractive contribution is overshooted by about 30%. The most important reduction comes from the one-body increments of the adjacent O<sub>p</sub> atoms (or rather their modification with respect to those calculated for the bare (TiO<sub>2</sub>)(H<sub>2</sub>O)<sub>3</sub> cluster). The contributions from the three-body interaction between the helium atom and two surface O<sub>p</sub> atoms (and the corresponding modifications of these surface O<sub>p</sub> two-body increments with respect to the bare (TiO<sub>2</sub>)(H<sub>2</sub>O)<sub>3</sub> cluster) equal to 6% of the two-body counterparts.

As discussed in Refs. 50,51, the two-body and three-body inter-monomer increments can be interpreted as two-body and three-body dispersion-like terms. This interpretation is further supported by the excellent agreement of the net dispersion energies with those calculated through the SAPT(DFT) method (see Fig. 3 and the bottom panel of Fig. 4). Indeed, the correlation part coming from the interaction of He and surface oxygen atoms (O<sub>p</sub>) fits to better than 0.15 cm<sup>-1</sup> (3%) to a  $-C_6/R^6 - C_8/R^8$  functional form over the considered range of He-O<sub>p</sub> distances (from 3.3 to 5.4 Å). The root-mean-square (RMS) of residuals (0.21 cm<sup>-1</sup>) increased by a factor of 7 if the  $R^{-8}$  term is omitted in the fitting, with the C<sub>6</sub> coefficient raising from 8.737 to 15.158 a.u. These values are around the one calculated

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3 with the Slater-Kirkwood formula for the interaction of a Helium atom  $O^{2-}$  anion (11.470  
4 a.u from Ref. 3). As already mentioned, however, it should be taken into account that the  
5 valence  $Ti(3d)$  orbitals also participate in the localized orbitals groups centered on the  $O_p$   
6 atoms. To estimate  $C_6$  coefficients for the  $He-Ti^{4+}$  interaction, we would need to correlate  
7 the semicore  $Ti(3sp)$  orbitals. Due to the small  $Ti^{4+}$  static polarizability (more than an  
8 order of magnitude smaller than for the  $O^{2-}$  anion,<sup>91</sup>) the associated two-body increments  
9 are expected to be much smaller. Considering also the different weight factors for Ti and  
10  $O_p$  atoms, it can be inferred that the correlation of the  $Ti(3sp)$  orbitals contributes very  
11 little to the total dispersion energy. The fitting of the increments accounting for the helium  
12 interaction with the subsurface oxygen atom required an extra  $R^{-10}$  term. These increments  
13 contribute to less than 4% of the net inter-monomer correlation at the vdW minimum. As  
14 can be seen from Table II, the (short-range) three-body  $He-O_p-O_p$  increments contribute  
15 very little (6% of the two-body counterpart), showing the effective pairwise behavior of the  
16 dispersion interactions.  
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19 Focusing now on the intra-monomer correlation terms (see Table II), we have found an  
20 exponential decay of the one- and two-body repulsive increment contributions as a func-  
21 tion of the  $He-Ti(5f)$  distance. As shown in Fig. 3, we have identified these terms with  
22 the correlation contribution to the exchange-repulsion energy component. The enhanced  
23 exchange-repulsion when the correlation is accounted for has been interpreted in terms of  
24 the effect called “truncation of the correlation space” by Staemmler.<sup>50,92</sup> Indeed, a number  
25 of virtual excitations experienced by the electrons of one monomer to avoid each other be-  
26 comes blocked when some orbitals are occupied by the electrons of the other monomer. It  
27 is thus strongly influenced by the energy location and shape of the “blocking” monomer  
28 occupied orbital (*e.g.*, the  $He(1s)$  orbital) and also the energy gap to the virtual orbitals.  
29 This repulsive contribution is added to the repulsion LMO-EDA energy component from  
30 the Hartree-Fock wave-function, reflecting the enhanced inter-electronic Coulomb repulsion  
31 within each monomer upon the orthogonalization of the HF occupied orbitals to those of the  
32 other monomer. As a short-range correlation contribution, the magnitude of the exchange-  
33 repulsion-like term increases steeply as the inter-monomer distance decreases. For example,  
34 it amounts to 21% and 38% of the total correlation energy at  $z = 3.875$  and  $3.5 \text{ \AA}$ , respec-  
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58 At this point, we would like to stress two aspects. First, it is not possible to rigorously  
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map SAPT-based energy components to the correlation contributions from supermolecular post-HF calculations. Therefore, the performance of SAPT(DFT) approaches is usually tested through the comparison with total CCSD(T) interaction energies. Aimed to fill this gap, the SAPT(CCSD) method has been recently developed by Korona (see, *e.g.*, Ref. 93 and references cited therein). In this work, the incremental partitioning of the CCSD(T) correlation has been applied along with the LMO-EDA of the HF interaction energy to perform the mapping between SAPT(DFT) energy components and CCSD(T) correlation contributions. For very weakly interacting systems composed by neutral closed-shell monomers, intra- and inter-monomer correlation contributions can be well identified with the correlation contribution to the exchange-repulsion/dispersion energy components. The excellent agreement between CCSD(T) and SAPT(PBE0) individual components confirms the reliability of this mapping. Second, the LMO-EDA scheme of Su and Li<sup>20</sup> identifies the whole CCSD(T) correlation contribution with the dispersion energy component instead of being separated as in our case. Similarly, the DFT correlation energy ( $E_c$ ) modification when going from the separated monomers to the complex is entirely accounted for within the inter-monomer LMO-EDA correlation energy component  $E_{corr}^{inter}$  (see Eq. 9), being identified with the short-range dispersion. Our results suggest that the changes in  $E_c$  could be also taken into account within the exchange-repulsion energy term, in an analogous way to the DFT exchange energy contribution to the repulsion term (see Eq. 8).

### C. Assessing the accuracy of the dlDF and SAPT(DFT) methods: He-(TiO<sub>2</sub>)(H<sub>2</sub>O)<sub>3</sub> interaction energies and nuclear bound states

To assess the accuracy of the dlDF and SAPT(DFT) methods, we have performed CCSD(T)-F12b calculations using all-electron augmented triple- and quadruple- $\zeta$  orbital basis sets (AVTZ and AVQZ, see Section II), followed by the extrapolation to the CBS limit. For benchmarking purposes, we have also calculated the nuclear bound states supported by the He-(TiO<sub>2</sub>)(H<sub>2</sub>O)<sub>3</sub> potential (see Table III and Figure 5). By comparing the CCSD(T)-F12b/AVTZ and CCSD(T)-F12b/CBS results (see Fig. 2 and Table III), it can be noticed that the interaction energies with the AVTZ basis are not yet converged. The well-depth minimum is  $\sim 8 \text{ cm}^{-1}$  (16.5 %) above the CCSD(T)-F12b/CBS vdW minimum, and the transition energies  $\nu_{1\leftarrow 0}$  and  $\nu_{2\leftarrow 1}$  between vibrational energy levels are red-shifted

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3 by about  $3 \text{ cm}^{-1}$  (14–33%) with respect to the benchmark values (see Table III). On a  
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5 contrary, the CCSD(T)-F12b/AVQZ interaction energies are very close to the estimated  
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7 CBS limit and the transition energies  $\nu_{1\leftarrow 0}$  and  $\nu_{2\leftarrow 1}$  are red-shifted by  $1.5 \text{ cm}^{-1}$  only. The  
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9 helium nuclear wave-functions are also very close, with the only distinguishable difference  
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11 being the delocalization degree of the highest-energy bound-state (see Fig. 5). The improve-  
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13 ment of coupled-cluster results upon the inclusion of functions explicitly depending on the  
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15 inter-electronic distance is already a well-known feature (see, *e.g.*, Ref. 84 for an exten-  
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17 sive analysis on weakly interacting dimers). In our case (see Fig. 2), this improvement is  
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19 particularly noticeable at the short-range region while the attractive tail is already well de-  
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21 scribed at CCSD(T)/(A)VTZ level. This can be expected from the better description of the  
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23 inter-electronic cusp with CCSD(T)-F12 methods, and thus short-range correlation effects  
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25 as those associated with the exchange-repulsion. An automatic implementation of the incre-  
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27 mental CCSD(T)(F12) scheme has been recently reported by Friedrich and collaborators<sup>94</sup>  
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29 but no applications to transition-metal oxides have been reported yet.

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31 Comparing the SAPT(DFT) and CCSD(T)-F12b potential energy curves (see the upper  
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33 panel of Figure 4), it can be noticed that the SAPT(DFT) interaction energies using the  
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35 PBE0 and dlDF functionals closely follow those from the CCSD(T)-F12b method. The  
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37 SAPT(PBE) approach also provides a very reasonable potential, but clearly too repulsive  
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39 at short-range, indicating the overestimation of the exchange-repulsion contribution. On a  
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41 contrary, as shown in the bottom panel of Figure 4, the dispersion energies depend weakly  
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43 on the density functional used. Still, they are slightly overestimated with the PBE func-  
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45 tional so that the potential at the medium-range region is a little bit more attractive than  
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47 the benchmark. Interestingly, the overestimation of the repulsive (exchange) component  
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49 is somewhat compensated by that of the attractive (dispersion) contribution so that the  
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51 nuclear bound-state energies agree to within 2% with those obtained through CCSD(T)-  
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53 F12b/AVTZ calculations (see Table III). Using the PBE0 functional, both the PEC (see  
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55 Fig. 4) and the nuclear bound-state energies lay in between the CCSD(T)-F12b counter-  
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57 parts using the AVTZ and AVQZ basis sets. The SAPT(dlDF) approach clearly offers an  
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59 even better agreement with the benchmark: the interaction energies at the vdW minimum  
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and beyond differ from those obtained through CCSD(T)-F12b/AVQZ calculations by less  
than 2% while, in the repulsive potential region, they become relatively closer to the es-  
timated CBS limit. This is the reason why the highest-energy nuclear wave-function (see



Fig. 5) is slightly less extended than the CCSD(T)-F12b/AVQZ counterpart. On the other hand, the transition energies  $\nu_{1\leftarrow 0}$  and  $\nu_{2\leftarrow 1}$  agree to within 0.1% with those calculated at CCSD(T)-F12b/AVQZ level.

Besides testing the performance of SAPT(DFT) approaches through the comparison with CCSD(T) total interaction energies, it is convenient to extend the analysis to the individual energy components. For this purpose, the estimations of the dispersion energies at CCSD(T)/(A)VTZ level, as extracted using the method of increments, are also shown at the bottom panel of Fig. 4. Taking into account the different theoretical grounds of the applied methods, it is worth stressing that the SAPT(DFT) dispersion contributions agree remarkably well with the CCSD(T)/(A)VTZ incremental counterparts (to within 3% for both PBE0 and dI DF functionals). Moreover, the incremental scheme at CCSD(T) level and the SAPT(PBE0) approach provide values for the correlation contribution to the exchange-repulsion in quantitative agreement within 4% (26.92 *vs.* 27.52 cm<sup>-1</sup> at the vdW minimum). Once again, the SAPT(PBE) approach overestimates the magnitude of this component, with the differences from the corresponding SAPT(PBE0) of about 35%. The exchange-repulsion becomes reduced in SAPT(dI DF) calculations by almost the same percentage. This can be explained considering that the KS orbitals of the monomers are more localized when the dI DF functional is used. As for the intra-monomer correlation contribution to the exchange-repulsion, it affects mostly the potential wall, making it even more repulsive. Our results indicate the overestimation of these contributions when calculated at CCSD(T)/(A)VTZ level. This way, it can be understood why the SAPT(dI DF) potential is closer to the CCSD(T)-F12b/AVQZ potential than the SAPT(PBE0) potential.

Having assessed the accuracy of SAPT(DFT) interaction energies (both total and dispersion and dispersionless energy components), we now evaluate the performance of the dI DF and dI DF+*D* approaches, with the dispersion contribution (denoted as *D*) extracted from the SAPT(dI DF) calculations. By subtracting from the benchmark interaction energies, the dispersionless dI DF values, it is clear that we should recover the dispersion component of the interaction as a function of the inter-monomer distance. As clearly apparent from the bottom panel of Fig. 4, this is indeed the case. The dI DF+*D* potential is in fact very close to the benchmark (see Fig. 2), with the only noticeable difference in the region between the minimum and the long-range tail (the dI DF+*D* potential is slightly more attractive). The accuracy of the dI DF+*D* potential can be further assessed by comparing directly the

bound-state energies and wave-functions (see Table III and Fig. 5). The average relative error of the vibrational energies is 3.3% only, with the dlDF+ $D$  values being larger. This way, the vibrational states with  $v = 1$  and 2 are more localized than the benchmarks (see Fig. 5). Taking into account the approximations used for our benchmark calculations (*e.g.*, the neglect of higher order correlation in the CCSD(T) method and the correlation of the valence electrons only), we can consider that the performance dlDF+ $D$  approach on the He- $\mathcal{C}_1$  interaction is very satisfactory.

#### D. Fitting of the dispersion interaction energies to the $D_{as}$ functional form

To analyze the transferability properties of the dispersion upon augmenting the  $\mathcal{C}(\text{TiO}_2)$  cluster size, we have fitted the dispersion He- $\mathcal{C}_1$  interaction energies to the effective interatomic functional form proposed in Ref. 24 and improved in Ref. 26 (referred to as  $D_{as}$ ),

$$D_{as} = \sum_{x \in \mathcal{C}_1} -\frac{\sqrt{C_6^{\text{He}} C_6^x}}{R_{\text{He}-x}^6} f_6 \left( \sqrt{\beta_{\text{He}} \beta_x} R_{\text{He}-x} \right) - \frac{\sqrt{C_8^{\text{He}} C_8^x}}{R_{\text{He}-x}^8} f_8 \left( \sqrt{\beta_{\text{He}} \beta_x} R_{\text{He}-x} \right) \quad (12)$$

where  $x$  numbers the atoms within the cluster  $\mathcal{C}_1$  and  $f_{6,8}$  are the damping functions of Tang and Toennies<sup>95</sup>. It should be mentioned that the parameters of the  $D_{as}$  function were not previously fitted on molecular systems containing the Ti atom. Therefore, we optimized the  $C_6^{\text{Ti}}$ ,  $C_8^{\text{Ti}}$  and damping  $\beta^{\text{Ti}}$  coefficients, freezing the parameters of the other atoms to the original values in Ref. 26. A relative RMS of 1.8% was thus obtained. By performing a global fitting of all the parameters, the relative RMS was slightly lower (1.5%) with the optimized values departing little from those reported in Ref. 26. This indicates an effective atom-in-molecule-like behaviour. We notice, however, that the  $C_6^x$  parameters entering into the  $D_{as}$  function should not be confused with the  $C_6$  coefficients extracted with the method of increments for the helium interaction with localized orbital groups centered on the O atoms. As already mentioned, the Ti(3d) atomic orbitals also contribute to these groups due to the partial covalent bonding. In any case, the  $C_6$  coefficients associated to He-O and He-Ti atomic pairs within the  $D_{as}$  function are not comparable to those obtained for the ionic He-O<sup>2-</sup> and He-Ti<sup>4+</sup> counterparts. Thus, from the  $\sqrt{C_6^{\text{He}} C_6^{\text{O}}}$  term in Eq. 12, we obtain a value of 4.07 a.u., which is about three times smaller than that calculated using the Slater-Kirkwood formula (11.47 a.u.).

### E. The He-(TiO<sub>2</sub>)<sub>3</sub>(H<sub>2</sub>O)<sub>5</sub> complex

Focusing on the next cluster size (the  $\mathcal{C}_2$  cluster, see Fig. 1), dispersionless and dispersion-accounting interaction energies of the He- $\mathcal{C}_2$  complex as a function of the He-Ti(5f) distance are plotted in Fig. 6 while the decomposition of the net interaction at the vdW minimum is shown in Fig. 7 (along with the He- $\mathcal{C}_1$  and He- $\mathcal{C}_2$  counterparts). The reference CCSD(T) interaction energies were calculated using a dual basis set: (1) the He atom and the O and Ti atoms in the main interaction region<sup>7</sup> were described with the same basis set as used for the incremental calculations (denoted as (A)VTZ, see above) but supplemented with a hydrogenic aug-cc-pVTZ set of mid-bond functions; (2) the cc-pVDZ basis set of Dunning was employed for the atoms outside the main interaction region.

The first thing to note from Fig. 6 is that both dIDF and SAPT(DFT) approaches provide very similar values for the dispersionless contribution over the whole range of inter-monomer distances. This energy component is rather similar to that obtained for the He- $\mathcal{C}_1$  complex. Still, as can be seen from Fig. 7, the exchange-repulsion is somewhat smaller. As compared with the  $\mathcal{C}_1$  counterpart, the  $\mathcal{C}_2$  cluster partially accounts for the extended nature of the surface along the Ti(5f) atom row. The additional oxygen atoms are rather far away from the helium atoms so that the enhancement of the net dispersion term is rather modest (about 15% at most). The decreased exchange-repulsion and the intensified dispersion-attraction come together to make the potential well about 20 cm<sup>-1</sup> deeper (see Fig. 6).

It is also worth stressing that the differences on the dispersionless SAPT(DFT) energies depending on the functional adopted are smoothed out when the cluster is enlarged. This is even more clear in Fig. 8, where the dispersionless contribution to the total interaction energy has been plotted for each cluster size. By comparing SAPT(ALDA/dIDF) and SAPT(PBE) interaction energies, we can notice that the dispersion component of the interaction is apparently influenced by the chosen functional. However, as discussed in Ref. 36, this is a consequence of using a pure ALDA kernel in calculations with non-local  $xc$  potentials when the density-fitting approximation is used. The results obtained for the He- $\mathcal{C}_1$  complex show that the dispersion energy is underestimated by about 7.5% (17%) when the ALDA/PBE0 (ALDA/dIDF) kernel is used instead of an hybrid  $xc$  kernel with 25% (61.44%) of single-determinant exchange. Accordingly, SAPT(ALDA/dIDF) and SAPT(ALDA/PBE0) interaction energies become very close to those calculated with the SAPT(PBE) approach when

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3 their respective dispersion components are augmented by 17% and 7.5%, respectively.  
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5 As can be seen in Fig. 6, the SAPT(ALDA/PBE0) potential is almost indistinguishable  
6 from that obtained through CCSD(T) calculations. The small underestimation of the dis-  
7 persion energy with the SAPT(ALDA/PBE0) model is thus compensated by the basis set  
8 incompleteness in our benchmark CCSD(T) calculations for the He- $\mathcal{C}_2$  complex. We have  
9 tested that the CCSD(T) well-depth with the same basis differs by about 12% from the CBS  
10 estimate for the He- $\mathcal{C}_1$  system. The SAPT(DFT) interaction energies are better converged  
11 with respect to the orbital basis set size so that additional diffuse functions causes a shift  
12 of the dispersion component by  $\sim 4\%$  only (see Section II). An overall very good agreement  
13 between SAPT(PBE0) and CCSD(T) interaction energies of weakly bound systems has been  
14 pointed out by Jansen and collaborators (see, for example, Ref. 96). Considering the basis  
15 set incompleteness in our CCSD(T) calculations, it seems that the SAPT(PBE) approach  
16 performs also rather well for the He- $\mathcal{C}_2$  complex.  
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26 Considering the interaction energies calculated with the dlDF+ $D$  and dlDF+ $D_{as}$  approa-  
27 ches (see Fig. 6), it can be noticed that SAPT(PBE) and dlDF+ $D_{as}$  PECs are almost  
28 identical to each other, differing also very little from the dlDF+ $D$  curve. Hence, it turns  
29 out that the dlDF+ $D_{as}$  approach is also an excellent performer for the He- $\mathcal{C}_2$  system. More-  
30 over, the similarities between the dispersion energies fitted from the He- $\mathcal{C}_1$  complex, and  
31 calculated directly for the He- $\mathcal{C}_2$  complex (and also the He- $\mathcal{C}_3$  system, as discussed below),  
32 indicate that this energy component is effectively pairwise additive, and highly transferable  
33 from one cluster size modeling the extended system to another. We also mention that the  
34  $D_{as}$  dispersion energies with the parameters obtained through the global fitting differed very  
35 little from those calculated with the constrained fitting (i.e., optimizing the Ti parameters  
36 only).  
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#### 48 **F. The He-(TiO<sub>2</sub>)<sub>9</sub>(H<sub>2</sub>O)<sub>7</sub> complex: correlation effects**

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51 As can be seen in Fig. 1, the largest cluster model of the rutile surface (the  $\mathcal{C}_3$  cluster  
52 with stoichiometry (TiO<sub>2</sub>)<sub>9</sub>(H<sub>2</sub>O)<sub>7</sub>) already includes the bridging oxygen atom O<sub>b</sub> rows. The  
53 key role of the O<sub>b</sub> rows has been highlighted in previous physisorption studies on the same  
54 surface.<sup>7,57</sup> As will be shown, the subdivision of the net interaction energy provides further  
55 insights into the reasons of this behavior. Figure 7 illustrates how the individual energy  
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components of the interaction are being modified upon expanding the cluster size from  $\mathcal{C}_1$  to  $\mathcal{C}_3$ , as evaluated with the SAPT(PBE0) approach. They are calculated at an intermediate He–Ti(5f) distance between the vdW minima of the He– $\mathcal{C}_{1,2}$  and He– $\mathcal{C}_3$  dimers ( $z = 3.25 \text{ \AA}$ ). As can be seen in Fig. 7, the He-cluster interaction becomes a factor of 4 more attractive upon expanding the cluster model from  $\mathcal{C}_1$  to  $\mathcal{C}_3$ . The most obvious reason is the inclusion of the dispersion He– $O_b$  interaction with the  $\mathcal{C}_3$  cluster. In the vdW minimum region, the distance separating the helium atom and the two nearest bridging oxygen atoms  $O_b$  differs very little from the He– $O_p$  distance (*e.g.*, by  $0.15 \text{ \AA}$  at  $z = 3.25 \text{ \AA}$ ). As mentioned above (see Table II), the two-body He– $O_p$  increments contribute to more than 90% of the net dispersion He– $\mathcal{C}_1$  interaction energy. Considering the effectively pairwise additive character of the dispersion He– $\mathcal{C}_2$  interaction, it can be expected that the two adjacent  $O_b$  atoms in  $\mathcal{C}_3$  enhance the net dispersion contribution by a factor of  $\sim 1.5$ . As can be observed from Fig. 7, this is certainly the case: the dispersion energy increases from  $-150 \text{ cm}^{-1}$  in  $\mathcal{C}_1$  to about  $-220 \text{ cm}^{-1}$  in  $\mathcal{C}_3$ . From Fig. 7, it can be also noticed that the induction interaction is more attractive (by about  $22 \text{ cm}^{-1}$ ) when atoms from internal molecular layers are included with the  $\mathcal{C}_3$  cluster model of the rutile surface. On a contrary, the induction component changes very little when the  $\mathcal{C}_1$  cluster is extended along the parallel direction to the surface with the  $\mathcal{C}_2$  cluster model. Nonetheless it is very important to notice that these are not the only factors in making the whole interaction more attractive: the He– $\mathcal{C}_1$  exchange-repulsion is reduced by about 30% ( $\sim 50 \text{ cm}^{-1}$ ) for the He– $\mathcal{C}_3$  complex. This is somewhat surprising, assuming that effective pairwise He– $O_b$  Pauli repulsive interactions are added to those of the He– $O_p$  pairs. The electronic density around the  $O_p$  atoms, however, differs considerably in  $\mathcal{C}_1$  and  $\mathcal{C}_2$  clusters. As pointed out by Rittner et al.,<sup>57</sup> the  $O_p$  atoms become squeezed out when bounded to Ti atoms within the  $\mathcal{C}_3$  cluster (*i.e.*, instead of the hydrogen atoms saturating the  $\mathcal{C}_1$  counterpart). This compression of the electronic density is reflected in the reduction of the electrostatic attraction by a factor of 1.6 (see Fig. 7). This explains why the uncorrelated HF exchange-repulsion decreases by  $20 \text{ cm}^{-1}$  when going from  $\mathcal{C}_1$  to  $\mathcal{C}_3$ . Since the exchange-repulsion energy is further overshooted, it is obvious that there is also a correlation effect at play in reducing this contribution. As clearly apparent from Fig. 9, both the decreased exchange-repulsion and the increased induction concur in providing dispersionless He– $\mathcal{C}_3$  interaction energies rather below the He– $\mathcal{C}_1$  counterparts, over the whole range of intermonomer distances.

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3 In order to get insights into the main correlation effect responsible for the reduction in the  
4 exchange-repulsion, the method of increments has been applied to the He- $\mathcal{C}_3$  complex with  
5 the MP2, LMP2, LCCSD(T) and CCSD(T) methods using the (A)VDZ basis (see Section II  
6 for the computational details). The most important incremental contributions are collected  
7 along with the He- $\mathcal{C}_1$  counterparts in Table IV. The values obtained for the He- $\mathcal{C}_1$  complex  
8 with the (A)VTZ basis are also tabulated. From the comparison to the results with the  
9 (A)VTZ basis, we can notice that the dispersion CCSD(T) He- $O_p$  term is underestimated  
10 by about 13% using the (A)VDZ basis, with the dispersionless contributions lying to within  
11 16% the (A)VTZ counterpart. Focusing on the incremental contributions obtained with the  
12 CCSD(T) approach (shown in boldface), it can be noticed that the correlation contribution  
13 to the He- $\mathcal{C}_3$  interaction energy  $E_{\text{int}}^{\text{corr}}$  is a factor of 1.7 larger than the  $E_{\text{int}}^{\text{corr}}$  value obtained  
14 for He- $\mathcal{C}_1$  dimer and the same basis. In agreement with the analysis of SAPT(DFT) energy  
15 components, only part of this increased correlation can be attributed to the additional  
16 attractive He- $O_b$  interactions in the He- $\mathcal{C}_3$  system: it turns out that the dominant (repulsive)  
17 one-body incremental contribution is more than doubled in the  $\mathcal{C}_1$  dimer. In both cases,  
18 this correlation term is dominated by the one-body  $\eta(O_p)$  increment modification when  
19 going from the bare  $\mathcal{C}_x$  ( $x = 1, 2$ ) cluster to the He- $\mathcal{C}_x$  dimer. Once again, this is the  
20 main correlation contribution to the exchange-repulsion. Albeit its magnitude raises as the  
21 inter-monomer He- $\mathcal{C}_3$  distance decreases, it stays well below the He- $\mathcal{C}_1$  counterpart. As  
22 mentioned above, this exchange-repulsion-like contribution can be rationalized in terms of  
23 the correlation space truncation of the electrons from the  $O_p$  atoms by the He(1s) orbital. It  
24 is influenced by both the energy and shape of the He(1s) orbital with respect to the valence  
25  $O_p(2s, 2p)$  orbitals and the energy gap to the low-lying virtual orbitals. At HF level, the  
26 energy of the border HOMO orbital (mostly formed by the  $O_p(2p)$  atomic orbital) varies  
27 very little from the  $\mathcal{C}_1$  to the  $\mathcal{C}_3$  cluster. On a contrary, the HOMO-LUMO gap in  $\mathcal{C}_3$   
28 is almost half of the value in  $\mathcal{C}_1$ . As in Ref. 46, the appearance of lower-lying excitations  
29 can be explained by the better description of the  $O_p$  diffuse tail when the electrons are  
30 allowed to flow to the neighbor titanium atoms in the  $\mathcal{C}_3$  cluster. On the other hand,  
31 the energy of the orbital oriented in the  $O_p$ -H direction in  $\mathcal{C}_1$  is very close to that of the  
32 He(1s) orbital, thus enhancing the correlation space truncation effect. The energy and space  
33 location of the  $O_b(2p)$  orbitals make them less affected, despite the very similar He- $O_p$  and  
34 He- $O_b$  distances. As a result, the  $\eta(O_b)$  increment contribution is a factor of 10 smaller  
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3 than the  $\eta(O_p)$  counterpart (see Table IV). For ionic materials such as the MgO(100)  
4 surface, the electronic charges are more confined and therefore the exchange-repulsion term  
5 is less influenced by the cluster model (see, for example, Ref. 38). As a partially covalent  
6 material, the realistic *ab initio* estimation of the intra-monomer correlation contribution to  
7 the interaction energy, necessitates the inclusion of the main interacting atoms from the  
8 rutile surface along with their first coordination shells within the cluster model. This is  
9 in contrast to the inter-monomer correlation contribution (i.e., the dispersion): as clearly  
10 shown in Table IV, the two-body He- $O_p$  dispersion interaction changes very little (5% at  
11 worst) when going from  $C_1$  to the  $C_3$  cluster for all the methods considered. This finding  
12 is further confirmed by the comparison between the d1DF+ $D$  and d1DF+ $D_{as}$  He- $C_3$  PECs  
13 shown in Fig. 9: the dispersion energies calculated directly for the He- $C_3$  complex match  
14 almost perfectly those fitted from the He- $C_1$  interaction.  
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25 The CCSD(T) results reported in Table IV can be also compared with those obtained  
26 with the MP2, LMP2 and LCCSD(T) approaches. The first thing to observe is that all  
27 these approaches significantly underestimate the correlation contribution to the interaction  
28 energy. At MP2 level of theory, both the overestimated repulsive intra-monomer correlation  
29 terms and the underestimated attractive He-O inter-monomer increments are responsible for  
30 such behavior. On the other hand, from the comparison of the most important LCCSD(T)  
31 and CCSD(T) incremental contributions as well as the LMP2 and MP2 counterparts, the  
32 tendency of the local correlation treatment to further reduce the dispersion terms is clearly  
33 apparent. It should be stressed, however, that MOLPRO default parameters were employed  
34 within the LMP2 and LCCSD(T) frameworks so that their optimization should provide  
35 results closer to the MP2 and CCSD(T) counterparts.  
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#### 46 **G. The He-(TiO<sub>2</sub>)<sub>9</sub>(H<sub>2</sub>O)<sub>7</sub> complex: potential energy curves**

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49 Let us now compare the PECs shown in Fig. 9, as obtained from supermolecular and  
50 SAPT-based calculations. We have calculated a reference value of the He- $C_3$  interaction  
51 energy at LCCSD(T) level of theory, using the ext-TVAE\*\* basis set and choosing a geometry  
52 at the minimum region ( $z = 3.75 \text{ \AA}$ , see Fig. 9). The interaction becomes more attractive  
53 by about 22% at LCCSD(T) level ( $-108.2 \text{ cm}^{-1}$ ) as compared with the LMP2 counterpart  
54 ( $-84.7 \text{ cm}^{-1}$ ). A similar enhancement (about 25%) is found for the main correlation part  
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of the interaction energy when it is treated at CCSD(T) level of theory instead of MP2 (see Table IV). The inclusion of the Grimme's spin correction in MP2 makes the interaction even less attractive (by about 20%). Previous studies of the He–MgO(100) extended system with the CRYSCOR code have already pointed out the underestimation of the dispersion contribution at LMP2 level of theory<sup>97</sup>.

Similar to the He– $\mathcal{C}_1$  system, it is evident from Fig. 8 that the SAPT(PBE) approach provides a too repulsive dispersionless interaction for the He– $\mathcal{C}_3$  complex. As occurred when the  $\mathcal{C}_1$  cluster was expanded along the Ti(5f) rows to build the  $\mathcal{C}_2$  model (i.e., the crystallographic [001] direction), it can be expected that a further extension of the  $\mathcal{C}_3$  cluster along the crystallographic  $[\bar{1}00]$  direction weakens down the differences between the SAPT(DFT) estimations for the net dispersionless term, and then the whole interaction energy. Interestingly, the overestimation of the dispersionless He– $\mathcal{C}_3$  interaction energy with the SAPT(PBE) approach is of very similar magnitude to the underestimation of the dispersion term by the SAPT(ALDA/PBE0) and SAPT(ALDA/dlDF) variants. Thus, the differences between the SAPT(PBE) and SAPT(ALDA/PBE0) PECs are almost indistinguishable (see Fig. 9). To avoid the dispersion energy underestimation in the SAPT(ALDA/PBE0) approach, Jansen and collaborators (see, *e.g.*, Ref. 36) have proposed a modified version that uses the Localized-Hartree-Fock method. Based on the very weak influence of the chosen functional for the dispersion He– $\mathcal{C}_1$  energies (see the bottom panel of Fig. 4), we have opted for a simpler solution: the dispersion energies calculated at SAPT(PBE) level (denoted as  $D$ ) have been added to the dispersionless energies evaluated with the SAPT(PBE0) approach. The PEC resulting from this scheme (referred to as SAPT(PBE0)+ $D$ ) is also plotted in Fig. 9. The interaction energy at  $z = 3.75 \text{ \AA}$  ( $-109.8 \text{ cm}^{-1}$ ) agrees well with the value obtained at LCCSD(T)/ext-TVAE\*\* level ( $-108.2 \text{ cm}^{-1}$ ), considering the approximations involved. The  $E_{int}$  value at  $z = 3.25 \text{ \AA}$  ( $-171.8 \text{ cm}^{-1}$ ) is also consistent with the estimated interaction energy at CCSD(T)/(A)VDZ level ( $-147.5 \text{ cm}^{-1}$ ), taking into account the different basis sets. Thus, when the He– $\text{O}_p$  increment using the (A)VDZ basis is upscaled by a factor of  $\sim 1.2$ , accounting for the difference with that calculated for the He– $\mathcal{C}_1$  complex using the (A)VTZ basis (see Table IV), the estimated  $E_{int}$  value becomes lowered to  $-170.4 \text{ cm}^{-1}$ . Finally, the excellent agreement between dlDF+ $D_{as}$  and SAPT(PBE0)+ $D$  potential energy curves is worth mentioning. Similarly to the He– $\mathcal{C}_1$  and He– $\mathcal{C}_2$  complexes, the dlDF+ $D_{as}$  approach has remarkably provided interaction He– $\mathcal{C}_3$  energies lying closer to the expected



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3 “exact” values than those calculated at higher levels of *ab initio* theory.  
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#### 7 IV. CONCLUDING REMARKS AND FUTURE PROSPECTS 8 9

10 The benchmark results presented in this contribution clearly show that the  
11 SAPT(DFT)<sup>31,32</sup> and d1DF+ $D^{24}$  methods perform very well in the ultimate dispersion-  
12 dominated physisorption problem, involving the helium interaction with clusters modeling  
13 a prototype transition-metal-oxide surface.  
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17 Using the smallest cluster and the localized molecular orbital energy decomposition  
18 (LMO-EDA),<sup>20</sup> we have analyzed the reasons why the most common DFT approach to  
19 determine He-surface potentials (the PBE functional) works reasonably well without disper-  
20 sion corrections and, consequently, poorly in DFT+D constructions. The application of the  
21 method of increments<sup>45</sup> to extract the intra- and inter-monomer correlation contributions  
22 along with the LMO-EDA partitioning of the HF interaction energy, has allowed us to esti-  
23 mate the individual energy terms characterizing the He-cluster interaction at CCSD(T) level  
24 of theory, comparing them with SAPT(DFT)-based components using different DFT func-  
25 tionals (PBE, d1DF, and PBE0). To refine our benchmarks, we have calculated the nuclear  
26 bound-state energies and wave-functions supported the He-cluster interaction potentials at  
27 CCSD(T)-F12b level of theory, including an estimation of the CBS limit. Although the  
28 three considered SAPT(DFT) versions provide results (i.e., bound-state energies) in quanti-  
29 tative agreement with the CCSD(T)-F12b counterparts, the individual dispersionless energy  
30 components calculated with the SAPT(DFT) method using the hybrid PBE0 functional are  
31 closer to those estimated at CCSD(T) level of theory by means of the method of increments.  
32 A similar conclusion has been reached in a systematic analysis using the SAPT(CCSD)  
33 method.<sup>93</sup> The hybrid meta functional d1DF, however, provides total interaction energies  
34 in better agreement with those obtained at CCSD(T)-F12b level using the largest basis  
35 set. The PBE functional within SAPT(DFT) overestimates the exchange-repulsion energy  
36 component, giving rise to the least attractive interaction. On a contrary, the dispersion  
37 energies calculated with the three SAPT(DFT) approaches agree remarkably well to each  
38 other and to those extracted with the incremental scheme at CCSD(T) level. Interestingly,  
39 the supermolecular d1DF+D approach has the best agreement with our best benchmark.  
40 For instance, the transition vibrational energies differ by less than 0.2% from those obtained  
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3 at CCSD(T)-F12b level.  
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5 Naturally, whatever the cluster modeling the rutile surface is, the dispersion domi-  
6 nates the attractive interaction. Both SAPT(DFT) and wave-function post-HF (CCSD(T),  
7 LCCSD(T), MP2, and LMP2) methods applied within the incremental scheme, demonstrate  
8 the high transferability character of the effective pairwise dispersion interaction in extending  
9 the cluster model along the different crystallographic directions. This has been made clear by  
10 the similarity between the dispersion energies estimated from the fitting of the SAPT(DFT)  
11 results for the smallest cluster (referred to as  $D_{as}$ ), and directly calculated at largest cluster  
12 sizes (denoted as  $D$ ). It has been further demonstrated through the explicit calculation of  
13 the dominant inter-monomer increments, characterizing the dispersion interaction, for both  
14 the smallest and the largest cluster: they differ by 5% at worst. On a contrary, the short-  
15 range intra-monomer correlation (associated to the exchange-repulsion) is not transferable  
16 between the two cluster models. For the smallest cluster, it is by far the most difficult  
17 correlation effect to deal with and its accurate estimation necessitates explicitly correlated  
18 methods. Fortunately, this contribution is much less important when the extended surface  
19 is better modeled. When going from the smallest to the largest cluster, the reduction in the  
20 exchange-repulsion is as important as the increase of the dispersion interaction in making  
21 the potential well a factor of four deeper. The He-cluster induction interaction becomes sig-  
22 nificantly more attractive when adding atoms from the internal molecular layers of the rutile  
23 surface with the largest cluster model. Contrarily, the extension of the cluster model only  
24 along a direction parallel to the surface leaves the inductive component almost unchanged,  
25 as the net dispersionless interaction. Therefore, further improvement of the He-TiO<sub>2</sub>(110)  
26 interaction energies at CCSD(T) level with the incremental scheme would require periodic  
27 HF calculations using slab models, enabling a more precise estimation of the long-range  
28 inductive term. Similarly, the more accurate determination of the He-TiO<sub>2</sub>(110) potential  
29 energy surface with the dlDF+ $D_{as}$  scheme requires further calculations including periodic  
30 boundary conditions. One possible route is to perform periodic dlDF calculations directly  
31 and the other possibility is to estimate the intramonomer correlation contribution from dlDF  
32 calculations using cluster sizes large enough to ensure its convergence.  
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55 The results presented in this contribution have also clearly demonstrated the capability  
56 of the method of increments to achieve the accuracy of the CCSD(T) calculations using large  
57 cluster models of the TiO<sub>2</sub>(110) surface. Moreover, we have illustrated how the combination  
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3 of this method with an energy partitioning of the HF interaction energy (as the LMO-  
4 EDA scheme) enables to estimate interaction energy components, which can be used as  
5 reference values in the design of ad-hoc DFT-based treatments. In this regard, the excellent  
6 performance of the dlDF approach (optimized to give the dispersionless component only) is  
7 particularly reassuring. Work along these lines is currently in progress.  
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11 In concluding, through the application of SAPT(DFT), the method of increments and  
12 explicitly correlated CCSD(T) on the helium interaction with cluster models of the rutile  
13 surface, we have unambiguously demonstrated both the accuracy of the dlDF+ $D$  approach  
14 and the high transferability of the effective pairwise dispersion interactions between the dif-  
15 ferent clusters. As a result, an even simpler dlDF+ $D_{as}$  scheme, in which the  $D_{as}$  dispersion  
16 function parameters are fitted using the smallest cluster model, is very reliable. Both the  
17 simplicity and accuracy of the dlDF+ $D_{as}$  scheme hold promises for the practical determi-  
18 nations of He-surface potential energy surfaces. Extensions to periodic boundary conditions  
19 are underway. Likewise, work is currently ongoing to extend the strategy described in this  
20 contribution to assess the performance of the dlDF+ $D_{as}$  approach on other substrates in-  
21 cluding graphite and metals, as well as different adsorbates for which the dispersion might  
22 be relevant but not necessarily dominant. In this regard, it is worth recalling that Pernal  
23 et al.<sup>24</sup> have demonstrated that the dlDF functional works also very well on molecular sys-  
24 tems with attractive interactions well beyond the van der Waals regime. As in the present  
25 contribution, we expect that these efforts serve to illustrate how the combined DFT and  
26 post-HF perspectives may be effective to tailor simple and reliable *ab initio*-driven schemes  
27 in surface science applications.  
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#### 44 **Acknowledgments**

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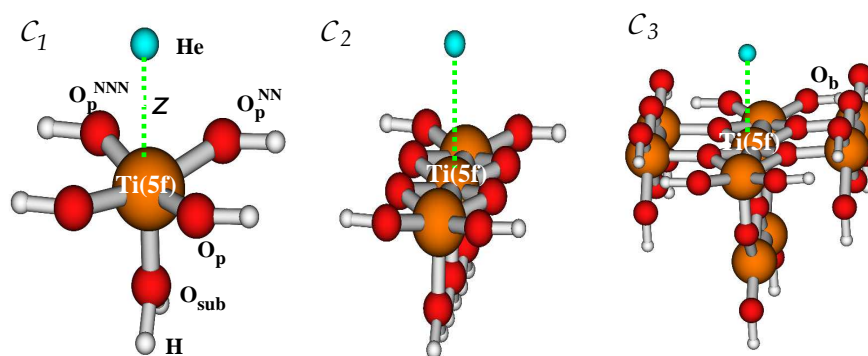


FIG. 1: (color online) Hydrogen-terminated clusters chosen to model the  $\text{TiO}_2(110)$  surface of stoichiometries  $\mathcal{C}_1$ :  $(\text{TiO}_2)(\text{H}_2\text{O})_3$ ,  $\mathcal{C}_2$ :  $(\text{TiO}_2)_3(\text{H}_2\text{O})_5$ , and  $\mathcal{C}_3$ :  $(\text{TiO}_2)_9(\text{H}_2\text{O})_7$ . A reference oxygen (in-plane) surface atom ( $O_p$ ), the nearest-neighbor ( $O_p^{\text{NN}}$ ) and the next-nearest-neighbor ( $O_p^{\text{NNN}}$ ) are highlighted along with a sub-surface oxygen atom ( $O_{\text{sub}}$ ), the adsorption site (the Ti(5f) cation), and a bridging oxygen atom at the  $\mathcal{C}_3$  cluster. The vertical height of a helium atom above the Ti(5f) site (referred to as  $z$ ) is also indicated.

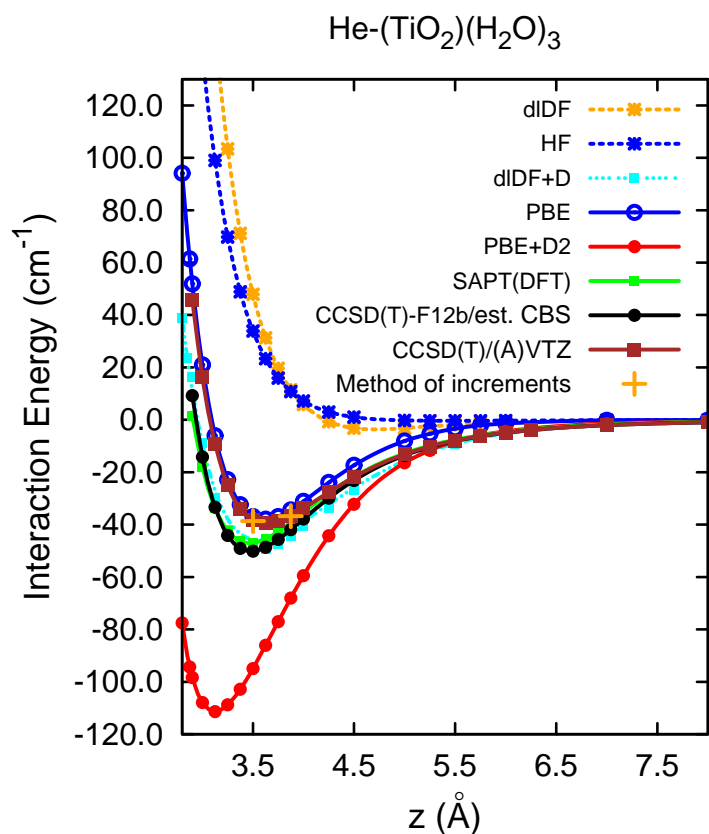


FIG. 2: (color online) He- $C_1$  interaction energies as a function of the He-Ti(5f) distance ( $z$ ) using different methodologies (see Fig. 1). The dIDF functional has been used for both the total SAPT(DFT) interaction energies and the dispersion energy component within the dIDF+ $D$  construction.

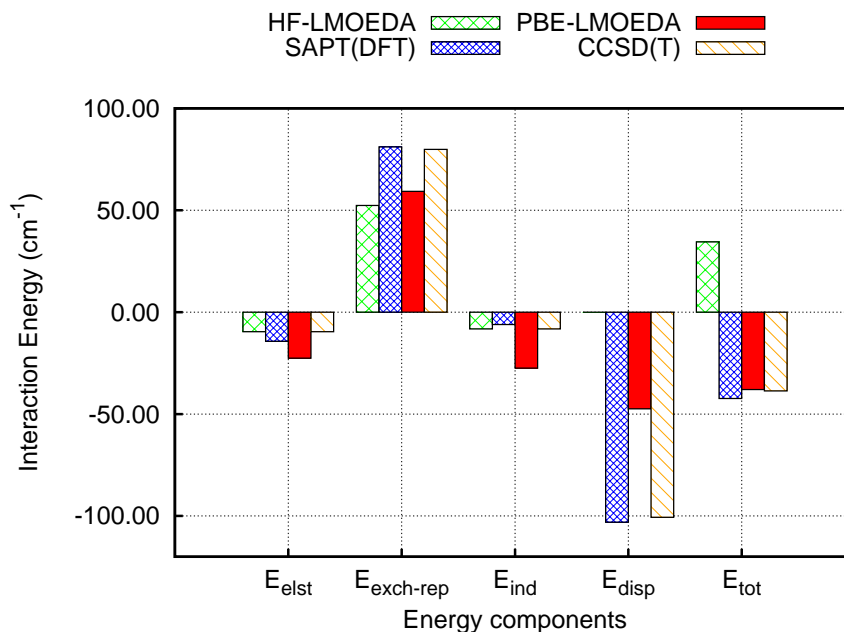


FIG. 3: (color online) Electrostatic, exchange-repulsion, induction, and dispersion contribution to the total interaction energies of the He-(TiO<sub>2</sub>)(H<sub>2</sub>O)<sub>3</sub> complex at the He-Ti distance of  $R = 3.5$  Å. The PBE0 functional has been used within the SAPT(DFT) framework. The partitioning of the PBE and HF interaction energies uses the LMO-EDA approach. The CCSD(T) energy components have been obtained by adding the exchange-repulsion and dispersion-like contributions extracted with the method of increments to the HF counterparts (see text).

TABLE I: Minimum energy positions ( $z_{\text{min}}$ ), well-depths ( $D_e$ ) and zero point energies (ZPE) of the He- $\mathcal{C}_1$  potential energy curves shown in Figs. 2 and 4.

	$z_{\text{min}}(\text{Å})$	$D_e(\text{cm}^{-1})$	ZPE( $\text{cm}^{-1}$ )
CCSD(T)-F12b: AVTZ/AVQZ/CBS	3.54/3.51/3.48	41.87/46.58/50.14	-26.22/-29.94/-32.80
SAPT(DFT): PBE/PBE0/dlDF	3.67/3.58/3.46	41.07/42.80/46.95	-25.73/-27.25/-30.54
dlDF+ $D$	3.53	50.07	-33.88
PBE	3.61	37.90	-19.68
CCSD(T): (A)VTZ	3.62	39.44	-24.63
PBE+D2	3.13	111.36	-80.68

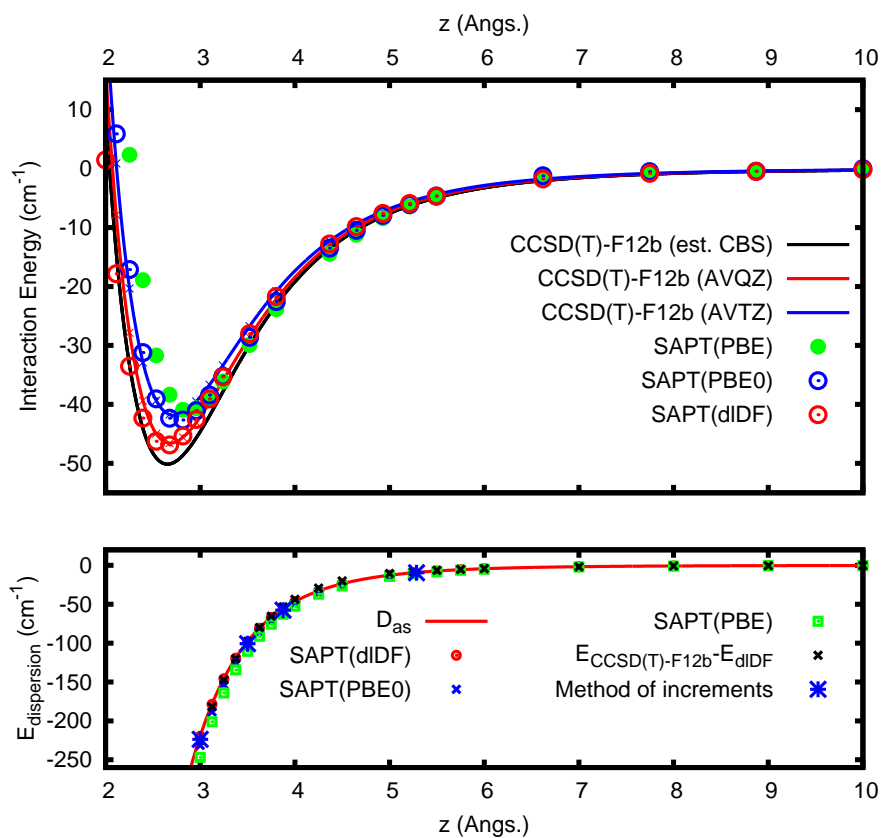


FIG. 4: (color online) Upper panel: Potential energy curves for the He-C<sub>1</sub> complex as obtained with the CCSD(T)-F12b and SAPT(DFT) methods. Bottom panel: Dispersion energies corresponding to the He-C<sub>1</sub> interaction. The dispersion energies obtained with the method of increments include three-body contributions (see text).



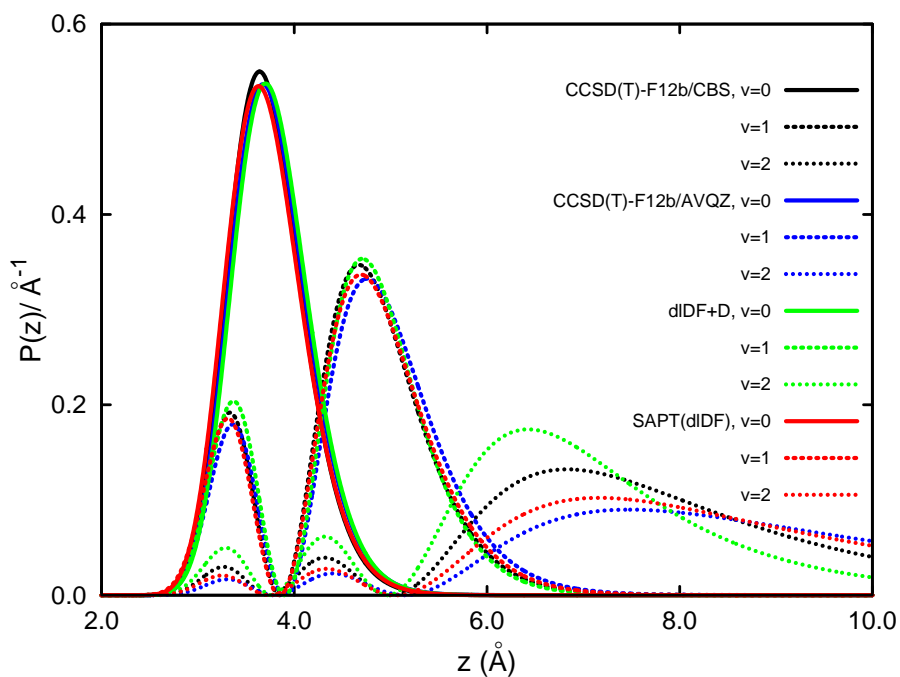


FIG. 5: (color online) Helium probability density distributions of the He- $C_1$  complex as a function of the He-Ti(5f) distance (see Table III). The distributions are normalized to unity.

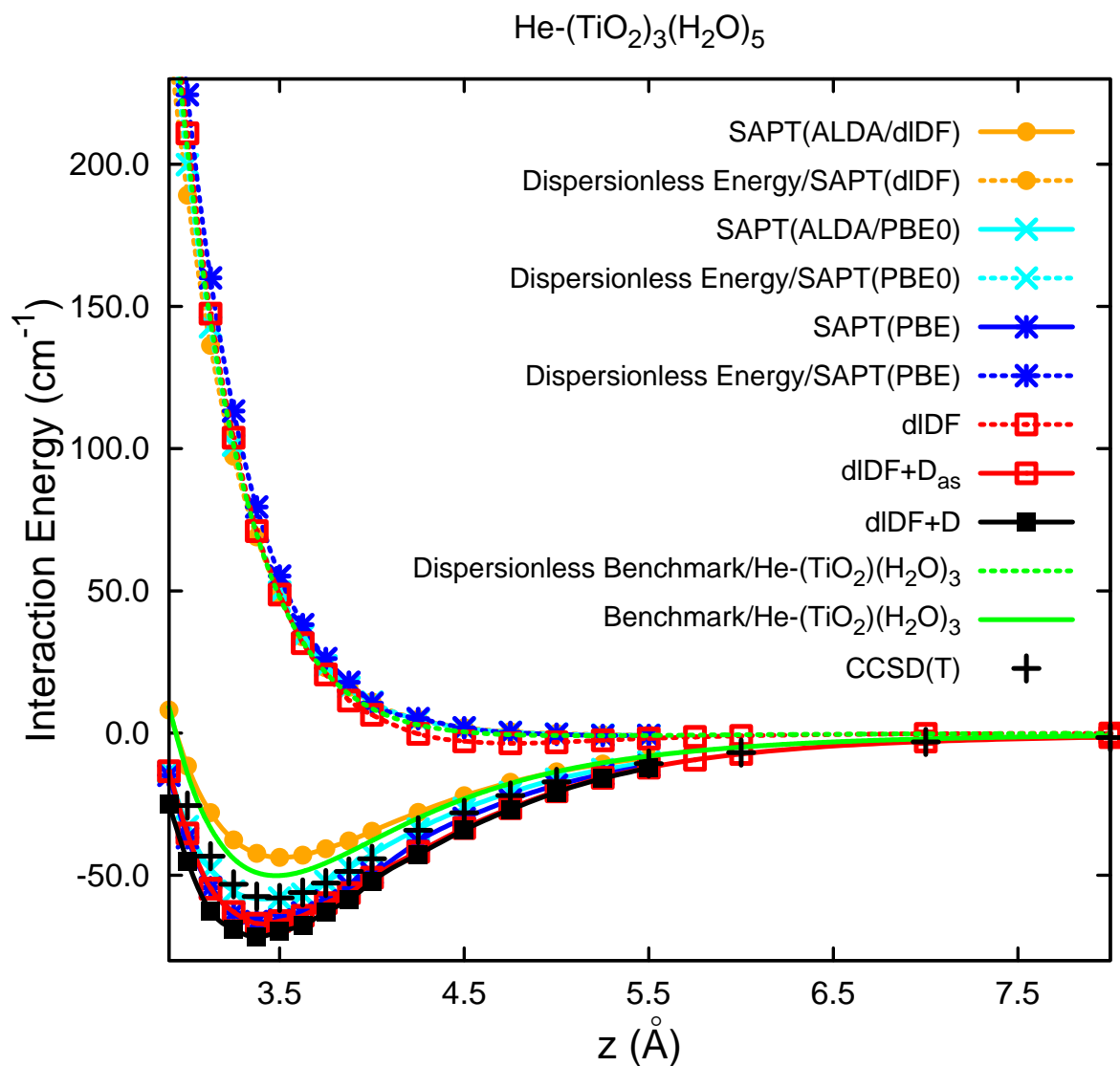


FIG. 6: (color online) Interaction energies of the He- $\mathcal{C}_2$  complex (see Fig. 1) as a function of the He-Ti(5f) distance using different methods. The benchmark for the He-(TiO<sub>2</sub>)(H<sub>2</sub>O)<sub>3</sub> complex corresponds to the CCSD(T)-F12b/CBS total interaction energies. The dispersionless benchmark refers to the difference between CCSD(T)-F12b/CBS total interaction energies and the dispersion energies calculated with the method of increments (see also the bottom panel of Fig. 4). The notation SAPT(ALDA/PBE0) and SAPT(ALDA/dIDF) indicates the use of a pure ALDA kernel in calculating dispersion energies with the hybrid PBE0 and dIDF functionals (see text).

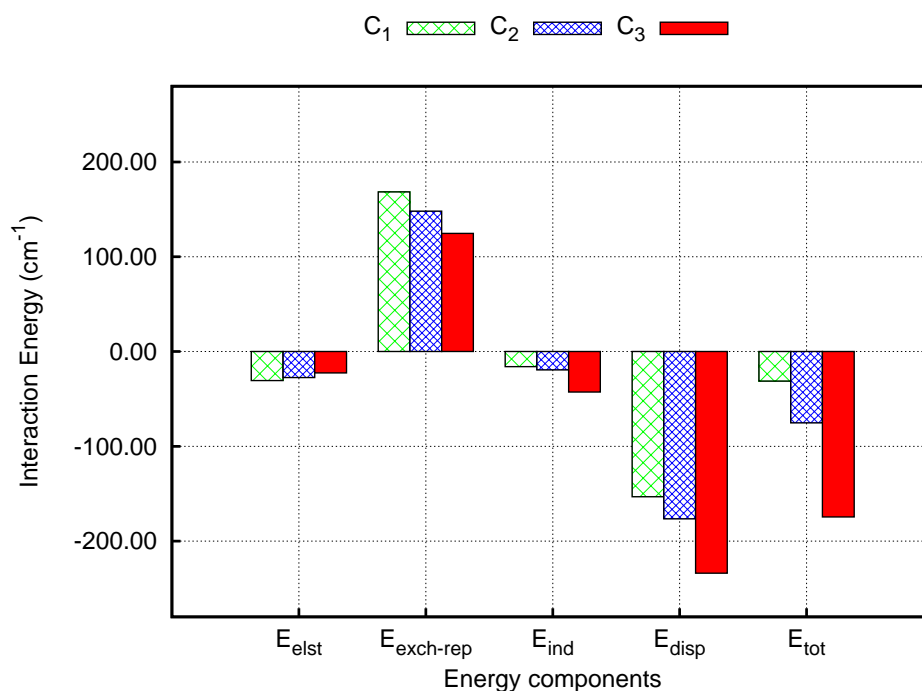


FIG. 7: (color online) Electrostatic, exchange-repulsion, induction, and dispersion contribution to the total interaction energies of the He- $C_1$ , He- $C_2$ , and He- $C_3$  complexes (see Fig. 1) at the He-Ti(5f) distance of  $R = 3.25 \text{ \AA}$ . The interaction energy components have been calculated with the SAPT(PBE0) method with the exception of the  $E_{\text{disp}}$  term in  $C_2$  and  $C_3$ , which has been calculated using the PBE functional.

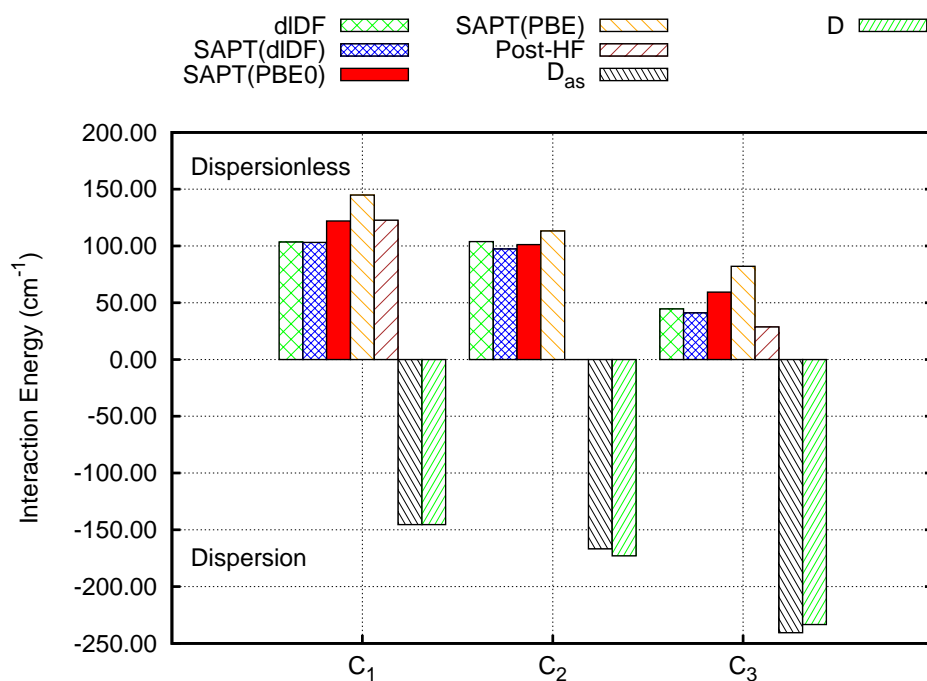


FIG. 8: (color online) Dispersionless contribution to the total interaction energies of the He-C<sub>1</sub>, He-C<sub>2</sub>, and He-C<sub>3</sub> complexes (see Fig. 1) at the He-Ti(5f) distance of  $R = 3.25 \text{ \AA}$ , as calculated with the dIDF and SAPT(DFT) approaches. The post-HF values correspond to the sum of the HF interaction energy and the intra-monomer correlation contribution, as extracted with the method of increments at CCSD(T) level of theory using the (A)VTZ and (A)VDZ basis sets for the He-C<sub>1</sub> and He-C<sub>3</sub> complexes, respectively. In the He-C<sub>3</sub> case, only the main incremental contributions to the intra-monomer correlation have been accounted for. For comparison purposes, the dispersion energies fitted from the calculations on the He-C<sub>1</sub> complex (referred to as  $D_{as}$ ), and directly calculated for the He-C<sub>2</sub> and He-C<sub>3</sub> complexes (denoted as  $D$ ) are also represented (see text).

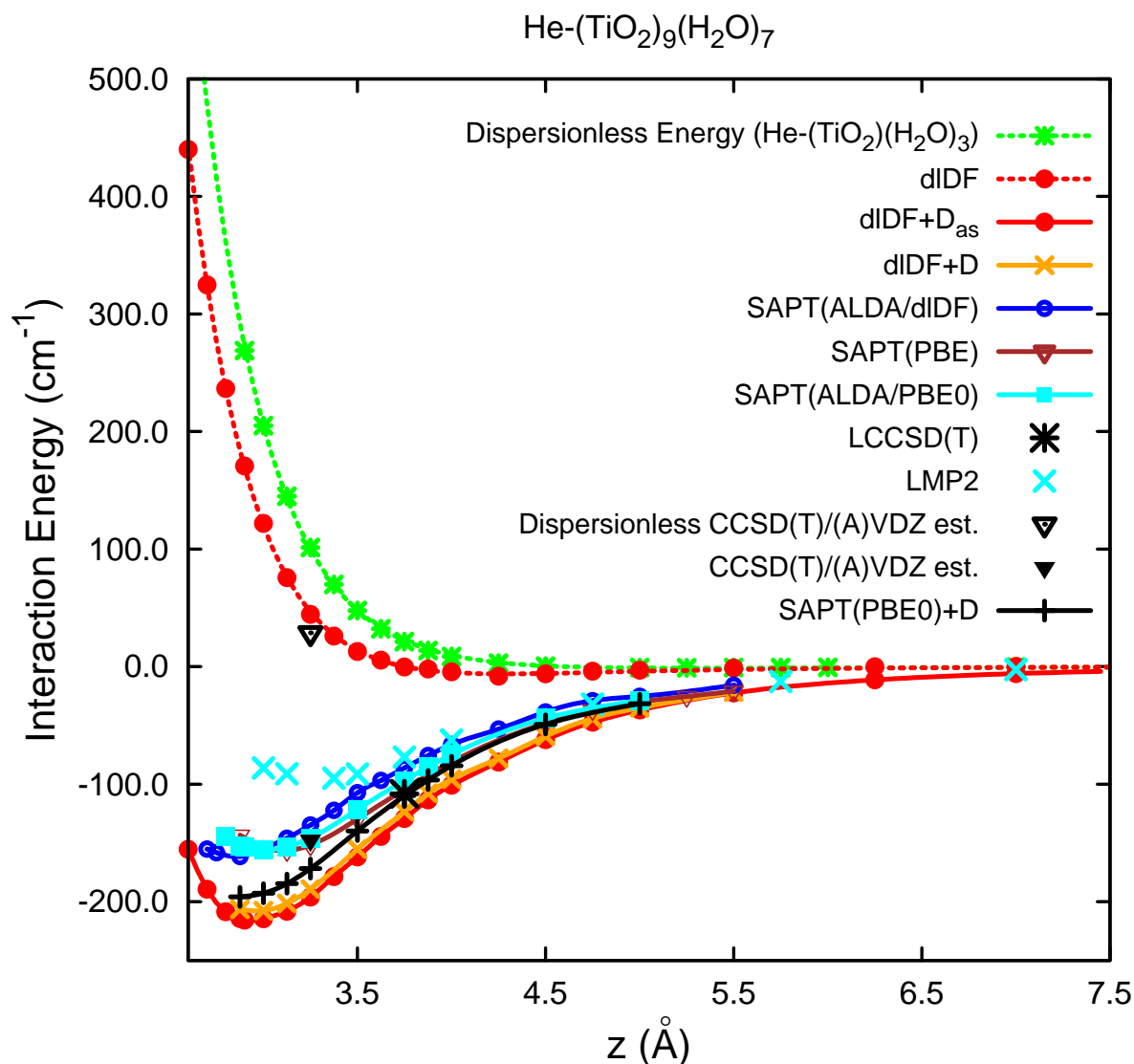


FIG. 9: (color online) Interaction energies of the He- $C_3$  complex as a function of the He-Ti(5f) distance using different methods (see Fig. 1). The LMP2 interaction energies are those reported in Ref. 7. The dispersionless energy of the He-(TiO<sub>2</sub>)(H<sub>2</sub>O)<sub>3</sub> complex has been calculated as the difference between CCSD(T)-F12b/CBS interaction energies and the dispersion energy components extracted with the method of increments. The SAPT(PBE0)+*D* notation refers to the sum of dispersionless SAPT(PBE0) interaction energies and dispersion energies calculated with the SAPT(PBE) approach. The CCSD(T) estimations correspond to the sum of the HF interaction energy and the main correlation contributions, as calculated with the method of increments at CCSD(T)/(A)VDZ level of theory (see text).

TABLE II: Selected correlation-energy increment contributions (in  $\text{cm}^{-1}$ ) to the total interaction energy from different orbital groups of the He- $\text{C}_1$  complex at distances of 3.5 and 3.875 Å (see Fig. 1). <sup>b</sup> Correlation contribution to the total interaction energy ( $E_{\text{int}}^{\text{corr}}$ ) from full CCSD(T)/(A)VTZ calculations (see text).

Orbital group	Increment contribution	
	$z = 3.5 \text{ \AA}$	$z = 3.875 \text{ \AA}$
One-body contributions		
$\eta(\text{He})$	1.7123	0.5779
$\sum_i \eta(\text{O}_p^{(i)})$	17.212	5.9600
$\eta(\text{O}_{sub})$	-0.0458	-0.0214
Two-body contributions		
$\sum_i \eta(\text{He-O}_p^{(i)})$	-102.806	-58.111
$\eta(\text{He-O}_{sub})$	-3.9367	-2.3257
$\sum_{i<j} \eta(\text{O}_p^{(i)}-\text{O}_p^{\text{NN}(j)})$	5.2022	1.9073
$\sum_{i<j} \eta(\text{O}_p^{(i)}-\text{O}_p^{\text{NNN}(j)})$	3.3180	1.5165
$\sum_i \eta(\text{O}_p^{(i)}-\text{O}_{sub})$	0.1216	0.0052
Three-body contributions		
$\sum_{i<j} \eta(\text{He-O}_p^{(i)}-\text{O}_p^{\text{NN}(j)})$	6.6833	3.5825
$\sum_{i<j} \eta(\text{He-O}_p^{(i)}-\text{O}_p^{\text{NNN}(j)})$	-0.5889	-0.6893
$E_{\text{int}}^{\text{corr}}$	-73.128	-47.598
	-72.774 <sup>b</sup>	-47.316 <sup>b</sup>
$E_{\text{int}}^{\text{HF}}$	34.482	10.832
$E_{\text{int}}^{\text{tot}}$	-38.646	-36.766

TABLE III: Energies of the bound states supported by the He- $\mathcal{C}_1$  potential energy curves shown in Fig. 2 and 4.

	$E_{v=0}(\text{cm}^{-1})$	$E_{v=1}(\text{cm}^{-1})$	$E_{v=2}(\text{cm}^{-1})$
CCSD(T)-F12b: AVTZ/AVQZ/CBS	-26.22/-29.94/-32.80	-6.34/-8.15/-9.60	-0.51/-0.62/-0.86
SAPT(DFT): PBE/PBE0/dlDF	-25.73/-27.25/-30.54	-6.33/-7.11/-8.65	-0.50/-0.50/-0.65
dlDF+ $D$	-33.88	-11.13	-0.91
CCSD(T)/(A)VTZ	-24.63	-5.94	-0.51

TABLE IV: Selected correlation-energy increment contributions (in  $\text{cm}^{-1}$ ) to the total interaction from different orbital groups of the He- $\mathcal{C}_3$  and He- $\mathcal{C}_1$  complexes at the He-Ti(5f) distance of 3.25 Å (see Fig. 1). For the sake of comparison, some increment contributions calculated at CCSD(T)/(A)VTZ level are also tabulated.

Orbital group	$z = 3.25 \text{ \AA}$								
	$\mathcal{C}_3/(A)VDZ$				$\mathcal{C}_1/(A)VDZ$				
	MP2	LMP2	LCCSD(T)	CCSD(T)	MP2	LMP2	LCCSD(T)	CCSD(T)	CCSD(T)
One-body									
$\eta(\text{He})$	0.51	7.72	5.70	0.73	(2.26)	(4.56)	(5.23)	(3.47)	(3.30)
$\sum_i \eta(\text{O}_p^{(i)})$	23.54	22.25	17.77	<b>17.62</b>	(47.12)	(45.96)	(37.83)	<b>(38.99)</b>	(33.84)
$\sum_i \eta(\text{O}_b^{(i)})$	3.80	3.0	1.74	2.19					
Two-body									
$\sum_i \eta(\text{He-O}_p^{(i)})$	-108.84	-99.35	-96.91	<b>-136.51</b>	(-103.59)	(-99.10)	(-95.23)	<b>(-133.12)</b>	(-153.59)
$\sum_i \eta(\text{He-O}_b^{(i)})$	-35.0	-32.37	-31.57	-39.75					
$E_{\text{int}}^{\text{corr}}$	-115.99	-98.75	-103.27	<b>-155.72</b>	(-54.22)	(-48.58)	(-52.17)	<b>(-90.66)</b>	(-116.45)

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