Valence change of praseodymium in Pr_{0.5}Ca_{0.5}CoO₃ investigated by x-ray absorption spectroscopy

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X-ray absorption spectroscopy measurements in $Pr_{0.5}Ca_{0.5}CoO_3$ were performed at the Pr $M_{4.5}$, Pr L_3 , and Ca $L_{2.3}$ absorption edges as a function of temperature below 300 K. Ca spectra show no changes down to 10 K while a noticeable thermally dependent evolution takes place at the Pr edges across the metal-insulator transition. Spectral changes are analyzed by different methods, including multiple scattering simulations, which provide quantitative details on an electron loss at Pr 4f orbitals. We conclude that in the insulating phase a fraction $[15(\pm 5)\%]$ of Pr^{3+} undergoes a further oxidation to adopt a hybridized configuration composed of an admixture of atomiclike $4f^1$ states (Pr^{4+}) and f-symmetry states on the O 2p valence band $(Pr^{3+}\underline{L})$ states indicative of a strong 4f-2p interaction.

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I. INTRODUCTION

Metal-insulator transitions (MIT) in perovskite cobaltites are a fundamental research topic due to the relevance of the spin state of Co for electron mobility in these strongly correlated oxides. ¹⁻⁸ In particular, $(Pr,Ln)_{1-x}Ca_xCoO_3$ cobaltites (Ln: lanthanide or rare earth) with perovskite structure are attracting great interest due to their unusual physical behavior. ⁹⁻¹⁶ As in other heavily doped isostructural cobaltites (such as $Ln_{0.5}Sr_{0.5}CoO_3$), the metallic state in $Pr_{1-x}Ca_xCoO_3$ is based on the Co-O $e_g(\sigma^*)$ band, and the electronic configuration of cobalt ions can be described as $t_{2g}^5(\sigma^*)^{1-x}$. Thus, the carriers introduced by doping in $Pr_{0.5}Ca_{0.5}CoO_3$ can be viewed as low-spin (LS) Co^{4+} (S = 1/2, t_{2g}^5) species moving through the matrix of intermediate spin (IS) Co^{3+} (S = 1, $t_{2g}^5e_g^1$) centers.

A sudden spin-state transition from IS $\mathrm{Co^{3+}}$ to diamagnetic LS $\mathrm{Co^{3+}}$ (S = 0, t_{2g}^6) was proposed as the origin of the MIT in $Pr_{0.5}Ca_{0.5}CoO_3$ at $T_{MI} \sim 75$ K.⁹ In contrast with the ferromagnetic order in compositions that do not show a MIT, the abrupt decrease in the conductivity at $T < T_{\rm MI}$ in $Pr_{0.5}Ca_{0.5}CoO_3$ avoids the onset of long-range order ferromagnetism. Initially, this sharp change was solely attributed to the onset of a new electronic configuration of Co (spin-state change of Co³⁺ ions). Understanding of the detailed mechanism for electron localization in this system, however, requires further efforts. Decreasing the *Ln* size and the Co-O-Co bending angle diminishes the $e_{\varrho}(\sigma^*)$ bandwidth and the carrier mobility. With Ln = Nd or smaller lanthanides in $Ln_{1-x}Ca_xCoO_3$ ($x \le 0.5$) the conductivity decreases, but there are no signs of abrupt spin-state changes. 17-20 suggesting only smooth thermally assisted variations of the Co³⁺ spin state. In addition, a first-order MIT is only found as a function of temperature or applied pressure in $(Pr_{1-y}Ln_y)_{1-x}Ca_xCoO_3$ compositions having Pr atoms. 13,15,17-19

Recent findings point to an active participation of Pr atoms in the transition. There is a remarkable volume reduction (\sim 2%) of the unit cell across $T_{\rm MI}$. Neutron

diffraction revealed that this contraction in the insulator phase of $Pr_{0.5}Ca_{0.5}CoO_3$ is the result of an A-O bond shortening (both Pr and Ca occupy the A site of the perovskite) and it is not due to changes in the size of CoO_6 octahedra. In agreement with the theoretical predictions by Knížek *et al.*, 3 a Schottky anomaly in the specific heat of $(Pr_{0.85}Y_{0.15})_{0.7}Ca_{0.3}CoO_3$ reported by Hetjmánek *et al.* indicated the presence of some Kramers Pr^{4+} ions in this compound at low temperature. In Finally, x-ray absorption experiments at the Pr L_3 and Co $L_{2,3}$ edges of $Pr_{0.5}Ca_{0.5}CoO_3$ allowed García-Muñoz *et al.* to confirm the existence of an exceptional mechanism based on a charge migration from Pr to Co at the MIT.

Some works have already pointed to a decisive role of Pr-O bonds in this family and a rather unique mechanism for the MIT in heavily Ca-doped Co perovskites that also contain Pr. Nevertheless, a proper understanding of the remarkable electronic and magnetic properties of the reference Pr_{0.5}Ca_{0.5}CoO₃ system requires additional investigations. For instance, they should help us to understand the observed photoinduced phase transition in Pr_{0.5}Ca_{0.5}CoO₃ under illumination by laser radiation, ¹⁶ where metallic domains can be stimulated, and which seems to be related to induced spin-state changes. It is also important to ascertain the actual occupation of atomic 4*f* states versus other *f*-symmetry extended states not localized on Pr ions, and to clarify the degree of covalency associated to distinct individual Pr-O*i* bonds in the insulating regime.

To obtain more detailed insight on the physical implications associated with the reported (Pr,Ca)-O bond contraction at $T_{\rm MI}$, we report and analyze here the changes occurring at Pr $M_{4,5}$ in the x-ray absorption spectra of Pr_{0.5}Ca_{0.5}CoO₃ when undergoing the MIT. Also, we have performed a thorough analysis of the Pr L_3 edge results presented in Ref. 15, including multiple scattering simulations. We clearly demonstrate that the spectral evolution is related to a further oxidation of Pr³⁺ ions in the insulating phase and make an estimation of the Pr³⁺:Pr⁴⁺ ratio at 10 K, although the ionic description is too crude in this system where Pr-O covalency effects are crucial. At this temperature, our findings support that \sim 15%

of Pr atoms have lost the trivalent $4f^2$ state, which has been substituted by a more oxidized mixed configuration composed of mainly tetravalent $4f^1$ states but also of a tetravalent Pr with electronic occupation of f-symmetry-projected extended states on O atoms. In addition, our study of the spectra recorded with hard x rays rules out intermediate valence models for praseodymium ($Pr^{3+\delta}$) in $Pr_{0.5}Ca_{0.5}CoO_3$.

II. EXPERIMENTAL

 $Pr_{0.5}Ca_{0.5}CoO_3$ (PCCO) samples were prepared in polycrystalline form by solid-state reaction using high-purity $Pr_6O_{11},\,Co_3O_4,\,$ and $CaCO_3$ precursors, following the method described in Ref. 11. The final annealing temperature was 1160 °C under O_2 atmosphere. To reach the optimal oxygen content, the samples were also treated under high oxygen pressure (at 900 °C with $P_{O2}=200$ bars for 14 h, and at 475 °C with $P_{O2}=150$ bars for 6 h). Samples were single phased, and fully stoichiometric according to Rietveld analysis of neutron diffraction data. Resistivity and magnetization curves displayed an abrupt transition signaling that electron localization occurs at $T_{\rm MI}\sim75$ K. 11

X-ray absorption spectra (XAS) measurements were made at the synchrotron radiation source of the Helmholtz Zentrum in Berlin.²² Spectra were collected in beamlines MAGS (Pr L_3 edge) and PM3 (Pr $M_{4,5}$ and Ca $L_{2,3}$ edges) by means of bulk-sensitive fluorescence yield (FY) and surface-sensitive total electron yield (TEY). MAGS beamline provides a focused beam in the energy range 4-30 keV, monochromatized by a Si (111) double-crystal monochromator, with a photon flux of the order of 10^{12} photons/s at 10 keV. The energy resolution is of the order of 2 eV at 6 keV. The PM3 dipole beamline delivers radiation within the 20-1900 eV energy range. Circular incident light polarization was employed. The measured maximum flux is $\sim 10^{10}$ photons/s at 900 eV, and the horizontal and vertical divergences of the beam are 1.5 and 1.0 mrad, respectively. The energy resolution is better than 0.3 eV at 900 eV. The Pr_{0.5}Ca_{0.5}CoO₃ polycrystalline sample was placed either in a displex-type (MAGS) or a He flow (PM3) cryostat that allowed temperature-dependent measurements. Spectra were collected on heating between 10 and 300 K. Fluorescence photons were detected either by a scintillation counter (MAGS) or a photodiode (PM3) placed at an angle of $\sim 90^{\circ}$ with respect to the incoming beam and vertically oriented with respect to the sample. Data normalization was performed following standardized procedures in all cases. Background removal was performed by subtraction of a straight line. At the Pr L_3 edge, absorption was normalized to an arbitrary value in the extended energy region, \sim 50 eV beyond the peak (i.e., the white line) position. In Pr $M_{4.5}$ spectra, we normalized to unity at the maximum of the M_5 edge.

III. RESULTS AND DISCUSSION

A. Soft x-ray spectra: Pr $M_{4,5}$ and Ca $L_{2,3}$ edges across the MIT

The top panel of Fig. 1 shows the experimental TEY-detected absorption spectra (Pr $M_{4,5}$ edge) of PCCO at different temperatures between 10 and 300 K. A clear evolution is observed. When crossing the MIT (from high to low temperature), the spectra appear slightly displaced towards

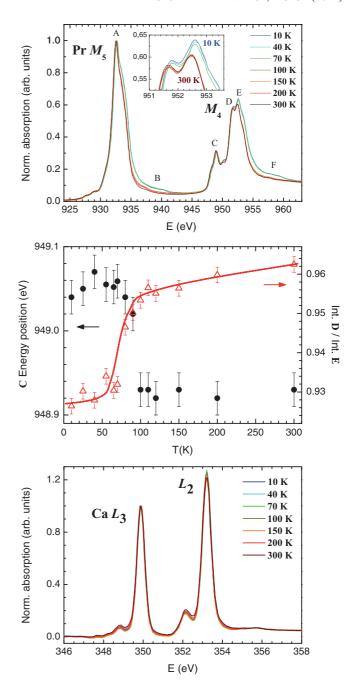


FIG. 1. (Color online) Experimental soft x-ray absorption in $Pr_{0.5}Ca_{0.5}CoO_3$. Top: spectra at the Pr $M_{4,5}$ edges as a function of T. A clear shift in the energy position and intensities of B, E, and F features occurs at the MIT as shown in the inset. Middle: fingerprints of the Pr valence change at the MIT such as the energy position of C (circles), and D/E relation of intensities (triangles); line is a guide to the eye. Bottom: spectra at the Ca $L_{2,3}$ edges as a function of T.

higher energy values by up to ~ 0.15 eV (Fig. 1, top; middle: circles, left scale). In addition, the spectral shape evolves (Fig. 1, top). B and F features appearing at $T < T_{\rm MI}$ are characteristic of compounds containing \Pr^{4+} . As explained within the single-impurity Anderson model, they originate due to the presence of a ligand hole (\underline{L}) in the 2p states of nearest-neighboring oxygen atoms. $^{23-25}$ Indeed, an atomiclike $4f^1$ state seems to be energetically unfavorable.

Covalency effects are large, and the spectroscopic features of nominally Pr⁴⁺ reference compounds like PrO₂ are only understood by considering a significant charge transfer from oxygen atoms. It has been reported that the combination of pure Pr^{4+} (4 f^{1}) and a nominally less oxidized Pr cation associated to a ligand hole $(4f^2\underline{L})$ leads to a qualitatively correct reproduction of experimental spectra. Actually, the latter component is often nearly as important as the atomiclike one. 25-27 The partial conversion of Pr³⁺ into Pr⁴⁺ at the MIT produces an evolution in the relative intensities of D and Epeaks, too, as seen in the middle panel of Fig. 1 (triangles, right scale). These changes seem to make sense after inspection of the absorption spectra of Pr₂O₃ (Pr³⁺) and BaPrO₃ (Pr⁴⁺) reference compounds reproduced from Hu et al.²⁸ in Fig. 2, which show remarkable differences. In the latter, the maximum at the Pr M_5 is found at 934 eV, ~ 1.5 eV beyond the Pr M_5 peak in PCCO at 300 K, and it shows a very large broad maximum at the Pr M_4 edge, also shifted to higher energies with respect to the multiple-featured edge in PCCO. The substitution of Pr^{3+} by Pr^{4+} therefore produces a decay in the D/E amplitude ratio in the insulating phase.

The bottom panel of Fig. 1 shows the absorption spectra of PCCO at the Ca $L_{2,3}$ edges as a function of temperature.

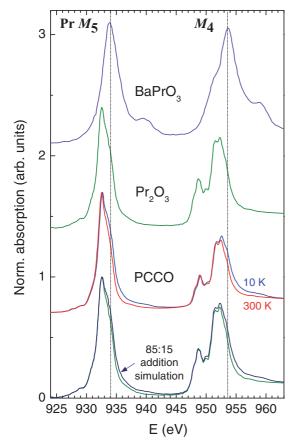


FIG. 2. (Color online) From top to bottom: experimental Pr $M_{4,5}$ absorption spectra of Pr tetravalent BaPrO₃ (violet) and trivalent Pr₂O₃ (green) [both from Ref. 28], Pr_{0.5}Ca_{0.5}CoO₃ measured at 10 K (blue) and 300 K (red), and a comparison of Pr₂O₃ to the 85:15 weighted addition (dark blue) of Pr₂O₃ and BaPrO₃ spectra, i.e., ideally the spectrum of a compound containing in average Pr^{+3.15}. Some spectra have been vertically shifted for clarity reasons.

It is worth reminding that Pr and Ca atoms randomly occupy the A position of the perovskite. We expect changes in Pr-O bonds not to be accompanied by a variation in the electronic state of Ca atoms. Indeed, we do not observe any significant change in the spectral shape between 10 and 300 K. Thus, we can safely ascribe the changes reported in the structural environment around the A site of the PCCO perovskite to Pr atoms only.¹¹

In the following, we attempt to estimate the actual charge donation by Pr^{3+} cations across the MIT. We start by assuming that no Pr^{4+} is present in our sample at room temperature. In Fig. 2, we can thus align the spectrum of PCCO at 300 K with that of Pr_2O_3 from Ref. 28 in the energy axis (at the Pr M_5 peak). The small misalignment detected in the energy position at the Pr M_4 edge is most likely due to slight differences in the spin-orbit coupling strength in both compounds. Indeed, PCCO at 300 K and the Pr_2O_3 spectra are remarkably alike. It is worth noticing that, contrary to hard x-ray absorption where the electronic band structure and the local geometry are the fundamental parameters modeling the spectral shape, the core-hole related effects (charge transfer, spin orbit, and exchange and multiplet interactions) dominate in the soft x-ray regime and push crystal structure effects into the background.²⁹

The weighted addition of Pr₂O₃ and BaPrO₃ spectra can thus be used to simulate the effect of an oxidation state of Pr ions between +3 and +4 in PCCO at $T < T_{MI}$. In the bottom of Fig. 2 we plot the result for the best weighted addition of these two references (85:15). The sum spectrum shows an increase of amplitude in B and F features comparable to that observed between the experimental spectra in PCCO at 300 and 10 K. We also see a decrease in the D/E amplitude ratio, although it is less noticeable than in PCCO. In general, the sum spectrum becomes a good simulation of the 10 K spectrum of PCCO. Minor discrepancies may be related to the approximation of BaPrO₃ as a Pr⁴⁺ reference. It is known that Pr 4f O 2porbital hybridization effects can strongly modulate Pr M_{4.5} absorption spectra and the amplitude of B and F features is one of its fingerprints.^{27,29} Nevertheless, they have been found to be similar in compounds with a well differentiated structure at room temperature, such as PrO₂ (Fm-3m) and BaPrO₃ (Imma)²⁸ Therefore we can expect a comparable spectral contribution of covalency effects in PCCO and BaPrO₃, both with a perovskite structure. Consequently, taking into account the uncertainties inherent in the method employed to analyze Pr $M_{4,5}$ absorption data, we can conservatively estimate that ${\sim}15(\pm5)\%$ of Pr³⁺ ions undergo a further oxidation process by crossing the MIT. This value is not far from, but somewhat lower than the estimation in Ref. 15 for the same compound, and below the calculated amount of oxidized Pr⁴⁺ in (Pr_{0.85}Y_{0.15})_{0.7}Ca_{0.3}CoO₃ from specific heat measurements.14

B. Hard x-ray spectra: Pr L_3 edge results and analysis

Despite the presented absorption study at the Pr $M_{4,5}$ ($3d \rightarrow 4f$) edge, which seems to be a more direct way to look into the actual oxidation state of Pr atoms, the eventual presence of Pr⁴⁺ is also reflected at the Pr L_3 ($2p_{3/2} \rightarrow 5d$) edge. $^{15,30-34}$

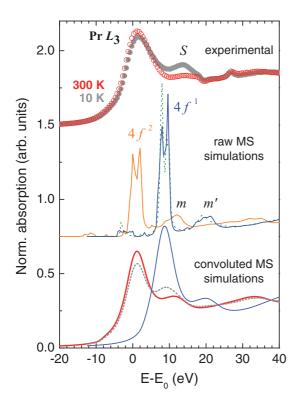


FIG. 3. (Color online) Top: experimental absorption spectra at the Pr L_3 edge of PCCO at 10 K (gray filled circles) and 300 K (red open circles). Middle: FDMNES calculations at the Pr L_3 edge before convolution by the core-hole energy broadening by using a cluster with radius 6.1 Å around the central Pr atom and a $4f^2$ (orange solid line) and $4f^1$ (blue solid line) electronic configurations, and for a 4.0 Å cluster (27 atoms) with a $4f^1$ configuration (green dotted line). Bottom: convoluted FDMNES calculated spectra for the larger cluster and a $4f^2$ (red thick solid line) and $4f^1$ (blue solid line) electronic filling. The dotted gray line is the result of an 85:15 weighted addition of the latter, respectively. Some spectra have been vertically shifted for clarity.

The top of Fig. 3 shows the experimental spectra at 10 and 300 K. The Pr⁴⁺ fingerprint feature is labeled as S. In Ref. 15 we attributed the enhancement of S in PCCO absorption spectra at $T < T_{\text{MI}}$ to the appearance of Pr⁴⁺ ions. However, to further confirm this interpretation it is necessary to analyze in detail the origin of this feature since multiple scattering (MS) of the photoelectron with the locally surrounding atoms around the central absorbing Pr may be significant at this energy value. Moreover, the amplitude of this likely contribution might vary with temperature. Regarding these concerns, on the one hand we must take into account that S is observed with different intensities in the metallic regime of PCCO (at high temperature, where praseodymium is present as Pr³⁺) and in other Pr³⁺ reference compounds. 15,30,35 Besides, the atomic displacements observed at $T_{\rm MI}$ induce geometrical variations that could influence the MS.

To get a deeper insight on the Pr L_3 edge changes, we have performed x-ray absorption simulations based on the MS formalism by means of the FDMNES code.³⁶ Calculations are relativistic and make use of the muffin-tin approximation for the calculus of the Hedin-Lundqvist potentials. The energy axis is relative to E_0 , the calculated absorption edge energy for

the $4f^2$ electronic configuration. The experimental spectrum has been shifted to align the Pr L_3 white line with that of calculations to facilitate comparisons. In the central area of Fig. 3 we show the result of simulations for a cluster of radius 6.1 Å around Pr atoms prior to convolution by the natural core-hole and experimental energy widths. Simulations have been performed for a PrCoO₃ cluster where we have used the crystallographic structure of PCCO at $T > T_{\rm MI}$ for a Pr $4f^2$ electronic configuration, and that of the low-temperature insulator phase for an excited Pr $4f^1$ configuration. The inclusion of Ca atoms in the cluster at the same crystallographic positions of Pr was discarded due to computational limitations. Nevertheless, for smaller clusters we checked that the Ca presence barely influences the position and amplitude of the main spectral features at the Pr L_3 edge. The MS simulation for the excited state $(4f^1, Pr^{4+})$ shows a larger white line shifted by \sim 8 eV to a higher energy with respect to the Pr $4f^2$ state simulation. The feature arising from MS is present and shows a similar magnitude (m, m') in the central area of Fig. 3) in both simulations. Moreover, we have obtained that the simulation of MS for the $4f^1$ state shows the same amplitude as for a smaller cluster containing only 27 atoms. Hence, we must conclude that the enhancement of S below the MIT in the experimental spectra cannot be attributed to changes in the MS signal associated with the reported structural variation at $T \approx 75$ K, but it has to be the consequence of the occupation of excited $4f^1$ states, i.e., to the appearance of Pr^{4+} .

Actually, the so-called S feature is formed by two components: the MS and the Pr⁴⁺ contributions, close in the energy scale. In the previous section we estimated the presence of \sim 15% of Pr⁴⁺ at 10 K in PCCO from Pr $M_{4,5}$ experimental absorption data. Next we will check the consistency of that result against the Pr L_3 edge data, considering it as an input for the FDMNES simulations. So, in the bottom part of Fig. 3 we show the calculated Pr L_3 absorption spectra after energy convolution. The gray (filled circles) curve depicts the result of the weighted addition of $4f^2$ and $4f^1$ states calculations according to the ratio 0.85:0.15. The comparison to experimental spectra in the top of Fig. 3 shows a good qualitative agreement. The low-temperature spectrum with average Pr^{3.15+} content reproduces (i) the enhancement of S in a nearly quantitative manner $(S_{10K}/S_{300K} = 1.21)$ and 1.17 for the experimental and simulated cases, respectively), (ii) the slight displacement towards lower energy, and (iii) the decrease in the amplitude of the main white line and the small shift of the main edge towards a higher energy by \sim 0.4 eV. There is a small discrepancy in the overall energy position of features in the calculated spectra with respect to experimental ones but it does not affect the basic grounds of our analysis. It is important to notice that the transfer of spectral amplitude from the white line to S at $T < T_{MI}$ would be not easily explainable in a scenario where the low-temperature phase was described in terms of an intermediate $Pr^{3+\delta}$ valence

More important is to note that the white line decrease at 10 K is overestimated in the calculations performed (Fig. 3). We consider that the explanation likely resides in the ligand-hole contribution. This can be due to covalency effects and f electron delocalization due to strong interaction between Pr 4f and O 2p states. In many formally tetravalent rare-earth

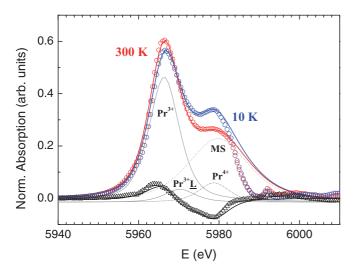


FIG. 4. (Color online) Experimental (circles) and pseudo-Voigt-composed best fitting curves (solid lines) of Pr L_3 absorption spectra. Individual pseudo-Voigt curves contributing to the spectrum at 10 K are also shown (gray solid lines). Black triangles and black line are the result of the subtraction of 300–10 K experimental spectra and fitting curves, respectively.

(Pr⁴⁺, Ce⁴⁺, Tb⁴⁺) based compounds, covalency promotes the presence of ligand holes in the neighboring cations and therefore the coexistence of $4f^n$ and $4f^{n+1}L$ states.^{31,32} Figure 3 suggests that, beyond the atomic orbital description, one should also consider Pr 4f states that present large hybridization with O 2p states and produce occupation of $4 f^2 L$ states. We have analyzed Pr L_3 spectra performing a fit of the absorption spectra by means of pseudo-Voigt (PV) curves after removal of the spectroscopic contribution of high-energy continuum states by subtraction of an arctangent curve. ^{26,32} A key point in this analysis is to reduce at maximum the number of free parameters. We have employed one PV curve per Pr final-state electronic configuration. Thus, at 300 K, assuming Pr valence is purely 3, we used one PV curve to account for the white line centered at 5966.5 eV and a second, broader one to simulate the spectral weight of the secondary multiple scattering feature centered at 5979.7 eV. By fixing the amplitude of the latter PV, we diminished the spectral weight of the first curve to simulate the loss of Pr³⁺ in the 10 K spectrum. This is reintroduced into the spectrum as a gain in Pr^{4+} and $Pr^{3+}\underline{L}$, which are simulated by the addition of two new PV curves (with the same full width at half maximum as that corresponding to Pr³⁺) to the spectrum centered at 5978.7 eV and 5970.2 eV, respectively.²⁶ The best fitting curves to experimental spectra at 300 and 10 K are shown in Fig. 4, where 0.15e- per Pr atom have been transferred to produce a mixed state with 0.09(2)e- of pure Pr^{4+} and 0.06(2)e- of $Pr^{3+}\underline{L}$. Namely, $15(\pm 5)\%$ of Pratoms are further oxidized at low temperature into a final state $|\Psi_{\rm oxid}\rangle = \alpha |4f^1\rangle + \beta |4f^2\underline{L}\rangle$ with $|\alpha|^2 \sim 0.6$ and $|\beta|^2 \sim 0.4$. The large disagreement between experimental and simulated spectra at energies beyond 5982 eV is due to the arctangent truncation function employed. In the bottom part of Fig. 4 we can see the experimental and calculated difference spectra between high and low temperature, where the ligand-hole spectral component accounts for the small kink present at \sim 5970 eV.

IV. CONCLUSIONS

We have presented a comprehensive spectroscopic study of the thermal evolution of Pr electronic properties in the structurally simple perovskite $Pr_{0.5}Ca_{0.5}CoO_3$. X-ray absorption spectra at both the Pr $M_{4,5}$ and L_3 edges undergo clear changes across the metal-insulator transition. These can be correlated to the anomalous shortening of the mean (Pr,Ca)-O distance in the insulating state as observed by neutron diffraction. We have also shown that Ca $L_{2,3}$ spectra do not display perceptible changes across the transition and hence we conclude that only Pr atoms account for this bond-length reduction. The magnitude of the Pr-O1 (-2.4%) and Pr-O2 (-2.1%) distances contraction 15 suggests a large hybridization between Pr 4f states wave functions and oxygen orbitals and/or electron migration involving changes in the valence of Pr ions.

Analysis of experimental spectra, performed following several methods, demonstrates that the spectral evolution is associated to a partial oxidation of Pr³⁺ ions into Pr⁴⁺. The low-temperature spectrum of PCCO at the Pr $M_{4.5}$ edges could be well reproduced by a semiempirical simulation making use of nominally Pr³⁺ and Pr⁴⁺ reference compounds in a 85:15 ratio. Pr L₃ edge experimental spectra at both sides of the transition were analyzed taking into account the multiple scattering resonance and the influence of the structural modifications at the transition. Within these considerations, the Pr L_3 spectrum at 10 K phase supports the coexistence of roughly an 85:15 admixture of (i) localized $4f^2$ states (Pr³⁺), and (ii) further-oxidized Pr⁴⁺ atoms. As usually found in formal Pr⁴⁺ compounds, a high degree of covalency is expected in the second type of Pr atoms, which can be modeled through a ligand hole in oxygen atoms. Our simulations indicate that nominally Pr4+ atoms can be described by an admixture configuration of atomiclike $4f^1$ states (\sim 60%) and $4f^1$ states with a large degree of covalent mixing with O 2p orbitals that induce the occupation of f-symmetry states localized on the valence band of O atoms $(Pr^{3+}L)$ states, \sim 40%), respectively. Indeed, this nonionic contribution needs to be considered in order to correctly account for some features in the experimental spectra.

Our analysis supports the charge transfer interpretation of the changes detected at the metal-insulator transition, and are consistent with density functional theory (DFT)-type calculations that indicate a transfer of holes from cobalt to praseodymium sites, 13 which would promote a Co spin-state transition to the LS t_{2g}^{6} configuration. As a result of the charge transfer, the insulating state seems due to the substitution of a broad itinerant Co-O band by more narrow and localized Pr-O states at the Fermi level.

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