Growth, spectroscopy, and tunable laser operation of the disordered crystal LiGd(MoO$_4$)$_2$ doped with ytterbium

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The disordered crystal LiGd$_{0.936}$Yb$_{0.064}$(MoO$_4$)$_2$ was grown with high optical quality by the flux method using Li$_2$MoO$_4$ as a solvent. The crystal possesses tetragonal symmetry belonging to the space group $I\bar{4}$, with two nonequivalent sites occupied by the dopant. Spectroscopic studies at 5 and 300 K provided information on the Stark energy-level splitting and the absorption and emission cross sections of the Yb^{3+} ion. Laser operation of Yb^{3+} was obtained for the first time in such a Li-containing double tungstate or molybdate. The tuning range with a Ti:sapphire laser pumping extended over $\sim$32 nm for the σ polarization and 23 nm for the π polarization. Without a tuning element the laser performance was similar for the two polarizations. By using a 10% transmission output coupler, a maximum output power of $\approx$470 mW was obtained with a slope efficiency η = 64.5% and the absorbed pump power at threshold was 520 mW. Laser operation was also achieved by pumping with a tapered diode laser and a fiber-coupled diode laser module, with a Yb laser output power of 0.66 W in the latter case. © 2006 Optical Society of America

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1. INTRODUCTION

Several sodium double tungstate (DT) and double molybdate (DM) single crystals with tetragonal structure and formula NaT(XO$_4$)$_2$ (T: trivalent ion, e.g., lanthanide, and X = W or Mo) were studied recently as laser hosts for tunable Yb^{3+} emission in the 1 μm spectral range. In particular, laser operation was demonstrated for the first time to our knowledge in Yb:NaGd(WO$_4$)$_2$,[1–3] Yb:NaLa(WO$_4$)$_2$,[4,5] and Yb:NaLa(MoO$_4$)$_2$.[4,6,7] These lasers could be tuned in a relatively broad spectral region, e.g., up to 48 nm around 1040 nm for Yb:NaLa(MoO$_4$)$_2$.[7] Such tunability is associated with the strong coupling of the Yb^{3+} electronic transition to the lattice vibrations and with the increased spectral linewidths of trivalent lanthanides induced by the crystallographic disorder in the Na-T cationic sublattice. Further interest arises from the possibility of using this broadband emission to support ultrashort laser-pulse durations near 1 μm in the sub-100 fs regime. In fact, the first demonstration of a passively mode-locked Yb laser based on a disordered sodium DT crystal has already been reported.[8]

Only a limited number of DT and DM compounds can be obtained at room temperature as tetragonal crystals with structural disorder. With few exceptions, most of the K, Rb, and Cs DT and DM crystals exhibit ordered phases with lower than tetragonal crystalline symmetry at room temperature. For instance, most of the KT(WO$_4$)$_2$ compounds have different crystallographic phases at high (close to melting or decomposition) and low (room) temperatures.[9] In cases when the disordered tetragonal structure exists at high temperature, polymorphic transformation to the low-temperature structure hampers or even prevents the preparation of a stable tetragonal phase with optical quality at room temperature.

Laser operation in general has been demonstrated in the past in only two lithium DT or DM hosts, LiLa(MoO$_4$)$_2$ (Refs. 10–12) and LiGd(MoO$_4$)$_2$ (Refs. 11, 13, and 14), but in all cases the dopant was Nd^{3+}, which operates as a four-level system. With the title compound LiGd(MoO$_4$)$_2$, or LiGdMo for short, the first emission of Nd^{3+} at 1059.9 nm for the $^4F_{3/2} \rightarrow {}^4I_{3/2}$ transition was obtained at 300 K with flash-lamp pumping[13]; soon afterwards a similar setup operated on the $^4F_{3/2} \rightarrow {}^4I_{3/2}$ transition of Nd^{3+} at 1340 nm,[11] and finally cw lasing at 1059.9 nm was obtained in Ref. 14, again with flash-lamp pumping.

The tetragonal scheelitelike structure was identified for the whole series of Li DM compounds with T = La -Lu, Y.[15,16] Li-based DM compounds melt with decomposition; single crystals of all the DM compounds were obtained using Li_2MoO_4 fluxes.[15] The crystals obtained at high temperature do not exhibit polymorphic transformations upon cooling to room temperature.[15] On the basis of single-crystal x-ray structural analysis, the centrosymmetric space group $I4_1/a$ was assigned to $LiCe(MoO_4)_2$,[17] and also to $LiNd(MoO_4)_2$ and $LiY(MoO_4)_2$.[18] In the latter work it was suggested that the same symmetry $I4_1/a$ is characteristic for the whole series T = La-Lu, Y. The only Li-based DM for which a nontetragonal low-temperature phase was identified is $LiLa(MoO_4)_2$, for which the largest ionic radius difference between the monovalent and trivalent cations occurs.[19] However, in x-ray diffraction (XRD) analysis the tetragonal space groups $I4_1/a$ (No. 88), $I4_1$ (No. 80), and $I\bar{4}$ (No. 82) are distinguished only on the basis of weak reflections. Therefore a detailed crystallographic study of the Yb-doped LiGdMo crystal, as performed in the present work, was necessary to ascertain the true symmetry.

Recently, an extensive crystal growth and characterization study has been conducted on single crystals of $LiYb(MoO_4)_2$ (LiYbMo) grown using Li_2MoO_4 and $Li_2Mo_2O_7$ solvents.[20] The high Yb density in this stoichiometric material $(6.87 \times 10^{21}\ cm^{-3})$ results in a large peak optical absorption coefficient $(\alpha > 125\ cm^{-1})$ for both polarizations.[20] Hence, very thin (of the order of 0.2 mm and less) and difficult to handle active elements will be necessary to control the reabsorption effects in the three-level operational scheme of the Yb^{3+} ion. In the present work we investigate the optically inert host LiGdMo and demonstrate for the first time to our knowledge laser operation with such a type of disordered Yb-doped Li DM crystal.

2. CRYSTAL GROWTH AND STRUCTURAL CHARACTERIZATION

$LiGd_{1-x}Yb_xMo$ crystals were grown in air using a platinum crucible and a resistance vertical tubular furnace by the top-seeded solution growth technique. The starting chemical products used were antireflection-grade Li_2CO_3, MoO_3, Yb_2O_3, and Gd_2O_3 and the solvent was Li_2MoO_4. The growth mixture consisted of 50 mol.% Li_2MoO_4 and 50 mol.% $LiGd_{0.9}Yb_{0.1}Mo$. The crucible filled with the charge was heated in the furnace to 1030°C, i.e., 50°C above the observed charge melting. The liquid was kept at this constant temperature for four days to achieve a homogeneous melt. Some melt evaporation was evident from the nucleation of needle-shaped microcrystals around the pulling wire. The resolidified phase was identified as $Li_{0.04}MoO_3$.[21] The saturation temperature of the solution was determined by repeated seeding. During the growth process, the crystal was slowly cooled at a rate of 1°C/day and rotated at a rate of 4.5 rpm. Finally, the crystal was pulled out from the melt and cooled down to room temperature at a rate of 15°C/h. Figure 1 shows a polished crystal plate used in the present laser study.
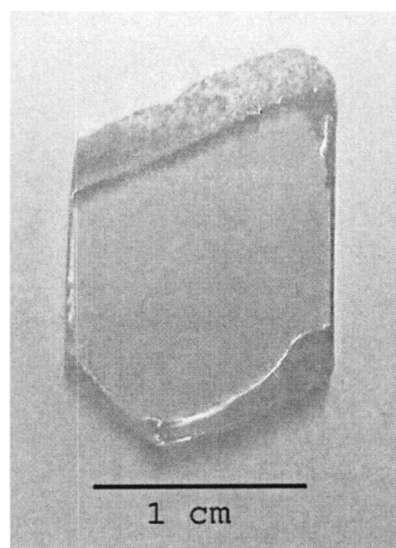

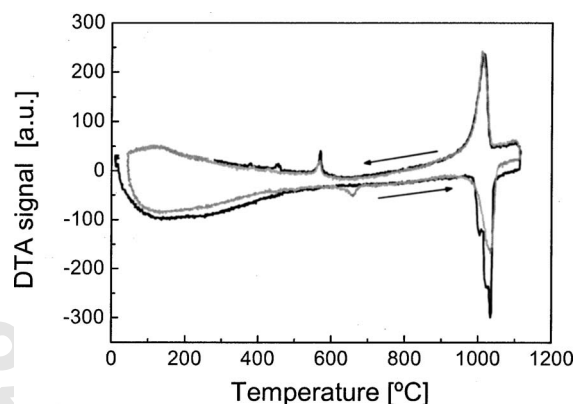Fig. 1.   Polished $LiGd_{0.936}Yb_{0.064}(MoO_4)_2$ plate used in the laser experiments.

Fig. 2.   Differential thermal analysis (DTA) of the $LiGd_{0.936}Yb_{0.064}(MoO_4)_2$ crystal. The first cycle is shown by the black curve, and the second cycle is shown by the gray curve. The decomposition melting character is evident from the complex nature of the endothermic peak at 1000°C, its change between the first and second cycles, and by the appearance of new peaks after the first cycle melting.

Simultaneous thermogravimetric and differential thermal analysis (DTA) were performed in air using Al_2O_3 as a reference. Heating and cooling ramps were at a 10°C/min rate. To assess the melting character, the samples were subjected to two consecutive melting and resolidification cycles. The DTA results revealed that the doped crystal melts with decomposition. This is evident from Fig. 2 by the presence of several overlapping peaks constituting the endothermic feature starting at ~1000°C. The comparison with the stoichiometric LiYbMo crystal[20] shows that the increase of the Yb concentration reduces the melting and decomposition temperature, in agreement with previous observations for the $LiT(MoO_4)_2$ series in the region T = Gd-Lu.[15]

The Yb concentration in the grown Yb:LiGdMo boule was estimated by inductively coupled plasma emission using an axial plasma spectrometer. It was 6.4 mol.% leading to an actual crystal composition of $LiGd_{0.936}Yb_{0.064}Mo$. This value along with the cell volume

obtained from the lattice parameters gives a Yb density of $4.06 \times 10^{20} \text{ cm}^{-3}$ in the grown boule.

XRD analysis of Yb:LiGdMo was performed with a selected prismatic-cut single crystal from the grown material, using the equipment and methods previously described.[20] Details for the data collection and refinement of the structure are summarized in Table 1.

Initially the structure of the Yb-doped LiGdMo crystal was refined for coordination of the atoms in the space group $I4_1/a$ that gave a discrepancy factor $R_1 = 0.0708$. The (Gd+Yb) and Li atoms in this unit cell randomly occupy the same $4b$ site with an occupancy factor of 0.5. However, several weak Bragg reflections $(0\ 0\ l)$, l is even but $l \neq 4n$ and $(h-h\ 0)$, $h \neq 2n$, which are forbidden for the space group $I4_1/a$, were present in the collected data. Seven of them, with an intensity higher than $3\sigma(F)$, constitute violations of the a plane but they are allowed for both $I4_1$ and $I\bar{4}$ space groups. However, the other two reflections, also with $I > 3\sigma(F)$, are forbidden for the 4_1 axis, which leads to the space group $I\bar{4}$. Moreover, the discrep-

Table 1. Crystal Data at 296 (2) K and Structure Refinement for Yb:LiGd(MoO$_4$)$_2$

Wavelength (Å)	0.71073
Crystal system, space group	Tetragonal, $I\bar{4}$
Unit cell dimensions (Å)	$a = b = 5.1814(10)$
	$c = 11.285(3)$
Volume (Å^3)	302.98(11)
Z, calculated density (Mg/m^3)	2, 5.306
Absorption coefficient (mm^{-1})	14.894
$F(000)$	430
Crystal size (mm)	$0.20 \times 0.20 \times 0.10$
θ range for data collection (deg)	4.33–28.85
Limiting indices	$-6 \leqslant h \leqslant 2$, $-6 \leqslant k \leqslant 6$, $-12 \leqslant l \leqslant 13$
Reflections collected/ independent (unique)	686/296 [$R(\text{int}) = 0.0263$]
Absorption correction	SADABS
Refinement method	Full matrix least squares on F^2
Data, restraints, parameters	296, 2, 34
Goodness of fit on F^2	1.119
Final R indices [$I > 2\sigma(I)$]	$R_1 = 0.0583$, $wR_2 = 0.1998$
R indices (all data)	$R_1 = 0.0605$, $wR_2 = 0.2060$
Extinction coefficient	0.1009(9)
	Site, atomic coordinates $(\times 10^4)$, $U_{(\text{eq})}{}^a$:
Gd(1)/Yb(1)/Li(1)	$2d$ 5000 0 2500, 9(2)
Gd(2)/Yb(2)/Li(2)	$2b$ 5000 5000 0, 7(1)
Mo(1)	$2a$ 0 0 0, 9(1)
Mo(2)	$2c$ 0 5000 2500, 9(1)
O(1)	$8f$ 2425(16) 1540(20) 857(10), 12(2)
O(2)	$8f$ 2401(18) 3490(20) 1657(14), 25(3)

^{a}Equivalent isotropic displacement parameters $U_{(\text{eq})}$(Å$^2 \times 10^3$) defined as one third of the trace of the orthogonalized U_{ij} tensor. The crystallographic refinement also included all positive anisotropic displacement, parameters, not given in the table for the sake of simplicity.

ancy factor when the crystal was refined in the space group $I\bar{4}$ decreased to $R_1 = 0.0583$, with satisfactory positive anisotropic displacement parameters for all atoms of the current Yb-doped LiGdMo crystal. Finally, we note that the noncentrosymmetric space group $I\bar{4}$ coincides with the one determined for LiYbMo (Ref. 20) and that the lattice parameters of LiGd$_{0.936}$Yb$_{0.064}$Mo lie between those of LiGdMo and LiYbMo.[15,20,22]

The above structural determination is important for the understanding of the optical properties of such disordered crystals, since the space group $I\bar{4}$ has two nonequivalent $2b$ and $2d$ sites with S_4 point symmetry occupied by Li$^+$, Gd^{3+}, or Yb^{3+} (see Table 1) with specific occupancy factors. From the macroscopic point of view provided by the XRD analysis, these sites are characterized by different cation–oxygen distances, namely, for the site $2d$ Gd(1)/Li(1)/Yb(1)-O(1) = 2.419(11)Å and Gd(1)/Li(1)/Yb(1)-O(2) = 2.448(13)Å, and for the site $2b$ Gd(2)/Li(2)/Yb(2)-O(1) = 2.437(11)Å, and Gd(2)/Li(2)/Yb(2)-O(2) = 2.433(13)Å. Moreover, the occupancy factors obtained without imposing charge restraints between both sites are (Gd+Yb)/Li [0.50(4)/0.49(1) in $2b$] and (Gd+Yb)/Li [0.51(4)/0.49(1) in $2d$]. Since these occupancy factors are very close to the statistical distribution over both sites, a high degree of disorder is assigned to the current crystal. This will give rise to a further contribution to the inhomogeneous broadening of the Yb^{3+} optical properties in comparison to the space group $I4_1/a$ assumption where a single $4b$ site for the Li$^+$, Gd^{3+}, or Yb^{3+} cations exists.

3. ABSORPTION AND EMISSION CROSS SECTIONS

Single-crystal LiGd$_{0.936}$Yb$_{0.064}$Mo samples oriented by the Laue technique were used in the optical studies. Polarized spectra are labeled as π ($\boldsymbol{E}|/\boldsymbol{c}$) and σ ($\boldsymbol{H}|/\boldsymbol{c}$). This crystal is transparent in the UV down to $\sim$350 nm at 300 K. At this temperature, the difference in the UV absorption edge for σ- and π-polarized light was very small, $<$0.5 nm. At 5 K the UV absorption edge moved to $\sim$340 nm ($\sim$1 nm shorter for π-polarized than for σ-polarized light).

To evaluate the crystal field splitting of the ground, $^2F_{7/2}$, and excited, $^2F_{5/2}$, multiplets of Yb^{3+}, we measured the absorption and emission spectra at a low temperature (5 K). Figure 3(a) shows the polarized absorption and Fig. 3(b) the unpolarized and π-polarized photoluminescence. The common absorption and photoluminescence band at $10,263 \text{ cm}^{-1}$ indicates the energy position (E_{zl}) of the $^2F_{7/2}(0) \leftrightarrow {}^2F_{5/2}(0')$ zero-phonon transition. Within our experimental uncertainty (0.2 nm) the position of this band in the 5 K photoluminescence was independent of the excitation wavelength, which means that the contribution of the coexisting Yb^{3+} centers cannot be resolved.

The other well-resolved bands in the 5 K absorption spectrum [Fig. 3(a)] are related to the energy position of the $^2F_{5/2}(n')$ levels. It is tempting to ascribe the spectral structure observed at 10,647 and $10,708 \text{ cm}^{-1}$ to separate contributions of Yb^{3+} ions in the aforementioned $2b$ and $2d$ crystal sites. This possibility was checked by applying
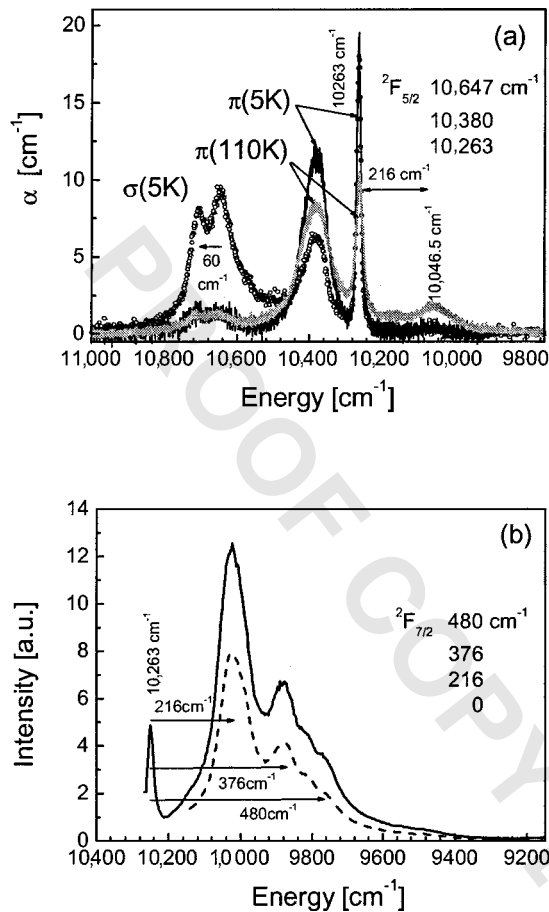
Fig. 3.   Low-temperature spectroscopy of Yb^{3+} in LiGdMo. (a) Polarized absorption spectra recorded at 5 and 110 K. (b) 5 K photoluminescence after excitation at 973 nm $(10,278\ cm^{-1})$ to the $^2F_{5/2}(0')$ level (dashed curve π-polarized record) and after excitation at 960 nm $(10,417\ cm^{-1})$ to the $^2F_{5/2}(1')$ level (solid curve, unpolarized record).

the semiempirical simple overlap model (SOM)[23] that allows us to estimate the crystal field parameters (CFPs) from the crystallographic positions of the Gd(or Yb)O_8 coordination polyhedra. Separate sets of CFPs for the Yb ions in the 2b and 2d sites were obtained from the atomic coordinates given in Table 1 and the corresponding Yb-O distances. In both cases the obtained values of the six CFPs corresponding to the S_4 symmetry were introduced in the simulation of the $^2F_{7/2}(n)$ and $^2F_{5/2}(n')$ Stark energy-level energies for the $4f$ (Ref. 13) configuration, along with free-ion (FI) parameters previously determined for the isostructural NaBi(WO$_4$)$_2$ host.[24] The simulation was performed using a previously developed code.[25] The used FI parameters, the derived SOM CFPs, and the corresponding energy-level schemes obtained for Yb^{3+} in each site are summarized in Table 2. The energy difference between both sites for the $^2F_{7/2}(0)\rightarrow{}^2F_{5/2}(2')$ transition obtained from this simulation was ~3 cm^{-1}. This is much less than the energy separation of the two bands at 10,647 and 10,708 cm^{-1} in Fig. 3(a) and it can be concluded that they are not due to Yb^{3+} in different crystal sites. In view of the agreement between the observed energy position of the band at 10,263 cm^{-1} in Fig. 3(a) with the energy-level calculations for the $^2F_{7/2}(0)\rightarrow{}^2F_{5/2}(0')$

transition that confirms the reliability of the method used, we ascribe the band at 10,647 cm^{-1} to the $^2F_{7/2}(0)$ $\rightarrow{}^2F_{5/2}(2')$ transition involving unresolved contributions from both sites and the secondary band observed at 10,708 cm^{-1} likely to coupling with a phonon of energy roughly equal to 60 cm^{-1}.

The energy position of the first excited $^2F_{7/2}(1)$ Stark level was determined from the absorption spectrum recorded at 110 K [see Fig. 3(a)]. The 5 K photoluminescence spectrum shown in Fig. 3(b) also exhibits well-resolved bands with an associated structure related to the simultaneous deexcitation of the $^2F_{5/2}(0')$ and $^2F_{5/2}(1')$ levels. The weak band observed at ~9810 cm^{-1} is also likely due to a 60 cm^{-1} phonon coupling of the sidebands. The analysis of these spectra allowed the assignment of the energy levels indicated in Figs. 3(a) and 3(b). These energy levels must be understood as the average energy positions of the Yb^{3+} centers induced by the nonequivalent 2b and 2d lattice sites and their multiple environments due to the random distribution of Li$^+$, Gd^{3+}, and Yb^{3+} cations over the same 2b and 2d lattice sites.

The room-temperature Yb^{3+} absorption cross sections are shown in Fig. 4. The Yb^{3+} emission cross sections can be calculated by the reciprocity method[26] using the relationship $\sigma_e(h\nu)=(Z_l/Z_u)\sigma_a(h\nu)\times\exp[(E_{zl}-h\nu)/k_BT]$ where Z_l and Z_u are the partition functions of the lower and upper multiplets and $Z=\Sigma_i d_i \exp(-E_i/k_BT)$ with d_i denoting the level degeneracy. With the determined energy positions (E_i) of the Yb^{3+} levels in the ground and excited multiplets, we arrived at $Z_l/Z_u=0.937$. Figure 4 shows the calculated emission cross sections and their comparison with the 300 K photoluminescence. It is obvious that reabsorption affects the measured photoluminescence signal at short wavelengths.

The other (alternative) method for calculation of the emission cross sections, the Fürchtbauer–Ladenburg equation, is not applicable in the case of the three-level

Table 2. FI (E^0 and ζ) Parameters and Calculated SOM CFPs (B_0^2, B_0^4, B_4^4, B_0^6, B_4^6, and S_4^6) Used to Determine the $^2F_{7/2}(n)$ and $^2F_{5/2}(n')$ Energy Levels of Yb^{3+} in the 2b and 2d Sites of LiGdMo^a

Site	2b	2d
E^0	4608.7	
ζ	2902.2	
B_0^2	486	516
B_0^4	−612	−600
B_4^4	±652	±630
B_0^6	−51	−97
B_4^6	±531	±522
S_4^6	±123	±105
$^2F_{5/2}(2')$	10,623	10,628
$^2F_{5/2}(1')$	10,361	10,353
$^2F_{5/2}(0')$	10,263	10,266
$^2F_{7/2}(3)$	475	483
$^2F_{7/2}(2)$	328	322
$^2F_{7/2}(1)$	211	206
$^2F_{7/2}(0)$	0	3

^{a}Parameters and energies are given in cm^{-1}. Overlap between Yb-ligand (oxygen) orbital wave functions: $\rho=0.067$; effective charge for oxygen, −0.8.
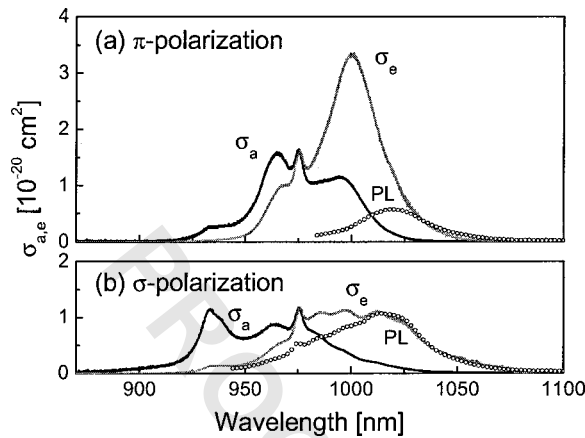
Fig. 4. Room-temperature absorption cross sections (σ_a, black curves) and comparison between the calculated emission cross sections (σ_e, gray curves) and the 300 K photoluminescence (PL, circles).

Yb system because of the above-mentioned reabsorption. However, by requiring that the results obtained by the two methods are identical, one can estimate the radiative lifetime. This is equivalent to calculating it from $(1/\tau_{rad})$ $=8\pi n^2 \int [\langle \sigma_e(\nu)\rangle /\lambda^2]d\nu$ where the averaging is over the polarization.[27] We obtained $\tau_{rad}=330\,\mu s$. The fluorescence lifetime was measured using powder obtained from single crystals of Yb:LiGdMo, diluted in ethylene glycol. The result was $\tau_f=250\,\mu s$. Both values can be compared with those similarly obtained for LiYbMo ($\tau_{rad}=235\,\mu s$ and τ_f $=183\,\mu s$) using slightly updated results in comparison with Ref. 20.

The maximum absorption cross sections near 975 nm (suitable for diode pumping) amount to 1.64 and 1.17 $\times 10^{-20}\,cm^2$ for the π and σ polarizations, respectively. The absorption peak near 933 nm, however, is also very high for the σ polarization ($1.15\times 10^{-20}\,cm^2$). The maximum emission cross section is obtained for the π polarization near 1000 nm: $3.3\times 10^{-20}\,cm^2$.

4. LASER EXPERIMENTS

The initial laser experiments with Ti:sapphire and tapered diode laser pumping were conducted at room temperature using an uncoated 2.6 mm thick plate sliced from the $LiGd_{0.936}Yb_{0.064}Mo$ boule. The relatively large sample (see Fig. 1) was in contact with the copper holder only at two edges of one of the surfaces and was practically uncooled. It had a *c* axis approximately 45° out of the surface plane and was inserted under Brewster angle in the three-mirror astigmatically compensated laser cavity, shown in Fig. 5(a). The 74 cm long cavity consisted of an end mirror M_1 with a radius of curvature of –5 cm; a folding mirror M_2 with a radius of curvature of –10 cm, through which the pump beam was focused by a f $=6.28$ cm lens L; and a plane output coupler M_3 of transmission T_{OC}. Pumping was realized only in a single pass since mirror M_1 was also highly transmitting at the pump wavelength. The pump beam was polarized always in the plane of Fig. 5(a) and the Brewster geometry forced the Yb laser to oscillate also in the same plane. This plane also contained the normal to the polished surfaces and

the *c* axis. Under these conditions the two nonequivalent possibilities for positioning the Yb:LiGdMo sample corresponded to orientation of the electric field vector of the light propagating in it at roughly 20° and 70° to the *c* axis, in which case we designate also as π and σ polarizations.

Finally, a 50 W unpolarized fiber-coupled diode laser module was used to study the power limits with diode pumping with the hemispherical cavity shown in Fig. 5(b). In this case an uncoated sample with a thickness of 1.9 mm and an aperture of 3 mm × 3 mm was cut from the same plate and mounted in a copper holder with water cooling to maintain the room temperature. The *c* axis was in the surface plane, which allowed natural selection of the polarization. M_1 was a plane high reflector for the laser wavelength on a 3 mm thick quartz substrate that was highly transmitting at the pump wavelength and the concave output coupler had a radius of curvature of –5 cm. The sample was placed as close as possible (≈ 0.2 mm) to M_1.

A. Results under Ti:sapphire laser pumping

The Ti:sapphire laser pump beam was focused to a spot of $\sim 22\,\mu m$ (Gaussian waist) and the pump power incident onto the Yb:LiGdMo crystal was limited in this case to 1.8 W. The pump wavelength was 975 nm, corresponding to the maximum of the absorption spectra (Fig. 4). The transmission of the output mirrors used was $T_{OC}=1.2\%$, 2.7%, 5.5%, and 10%.

Figure 6 shows the change of the crystal absorption measured with increasing incident pump power. Strong absorption bleaching can be seen in the absence of lasing, and laser operation had a recycling effect resulting in recovery of the absorption. This leads to a rather weak dependence of the actual crystal absorption on the incident pump power. The dependence was similar for the two polarizations studied and the absorption was only slightly higher for the π polarization. Even for the highest outcoupling used, the intracavity intensity was high enough to substantially compensate the bleaching effect.
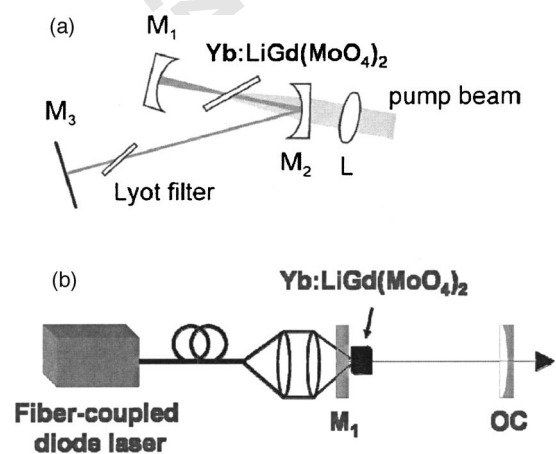

Fig. 5. (a) Schematic of the Yb:LiGdMo laser cavity with the Lyot filter for tuning used with Ti:sapphire and tapered diode lasers as pump sources. (b) Plane–concave cavity without polarization-selecting elements for pumping with the fiber-coupled diode module. OC, output coupler.
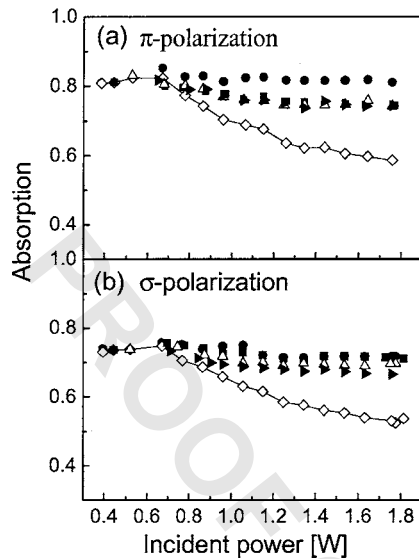
Fig. 6. Single-pass crystal absorption versus incident pump power without lasing (diamonds) and under laser operation for T_{OC}=1.2%, 2.7%, 5.5%, and 10%. (right triangles), (a) π polarization and (b) σ polarization.
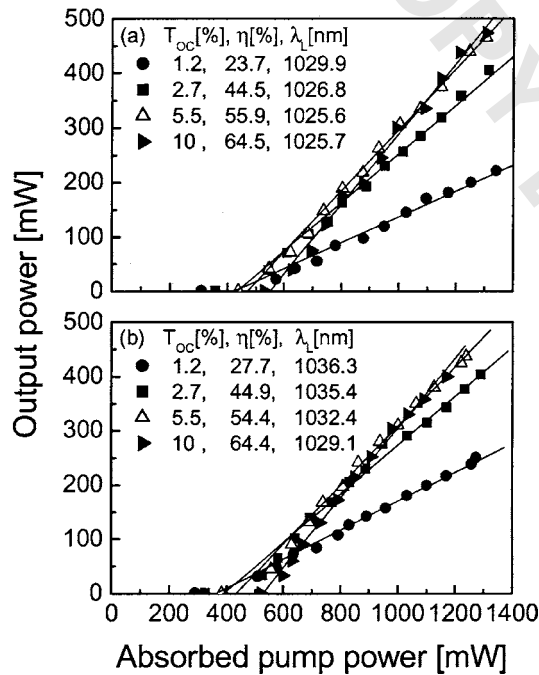

Fig. 7. Output power versus absorbed pump power at 975 nm for the cw Yb:LiGdMo laser (symbols) for (a) π polarization and (b) σ-polarization. The linear fits shown give the slope efficiencies η obtained for the indicated output coupler transmissions T_{OC}.

Figure 7 shows the laser performance (output versus absorbed power) for the two polarization configurations. For π polarization [Fig. 7(a)], a maximum output power of 473 mW was obtained with T_{OC}=10% at a laser wavelength λ_L=1025.7 nm. In this case the maximum slope efficiency (η=64.5%) was also obtained and the absorbed pump power for reaching the laser threshold was ~520 mW. The overall laser performance was quite similar for the two polarizations. The minimum thresholds measured with T_{OC}=1.2% were of the order of 300 mW.

The only substantial difference that could be observed as a trend was that the laser wavelength was always longer for σ polarization.

The laser tunability was studied by inserting under Brewster angle a two-plate Lyot filter in the M_2-M_3 cavity arm close to the output coupler (Fig. 5). Under optimum alignment the output power reduction with the filter inside the cavity did not exceed 5%. Figure 8 shows the tunability achieved with two different output couplers, T_{OC} =1.2% and 5.5%, for an incident pump power of 1.8 W. In both cases these tunability curves were limited on the shortwave side by the spectral characteristics of the cavity mirrors and in particular by the output couplers used. The total tuning range obtained extends from 1016 to 1049 nm. In addition to the longer lasing wavelengths, the tuning curves for σ polarization are broader than those for π polarization.

The average inversion $\langle\beta\rangle=(1/d)\int_0^d\beta(z)\mathrm{d}z$ necessary to achieve the threshold for a sample thickness d and Yb-ion density N can be estimated from $\langle\beta\rangle=[2\sigma_a Nd-ln(1-T_{OC})]/[2Nd(\sigma_a+\sigma_e)]$. The computed results for the two polarizations and two different output couplers are plotted in Fig. 9. The comparison with the experimentally observed wavelength dependence is justified because we established that the wavelength changed only slightly from threshold to maximum power (increasing by less that 1 nm for the π and less than 3 nm for the σ polarization for all four output couplers used in Fig. 7). It can be seen from Fig. 9 that for a given output coupling the threshold average inversion has its minimum at longer wavelengths for the σ polarization in accordance with the experimental observations in Figs. 7 and 8. Moreover, it can be clearly seen from Fig. 9 that for both polarizations the oscillation wavelength gets shorter with increasing output coupling, a trend experimentally observed not only without a tuning element (Fig. 7) but also evident from the tuning behavior in Fig. 8.

B. Results under Diode Laser Pumping
The tapered diode laser (TDL) used for demonstration of diode pumping consisted of a 1 mm long ridge waveguide
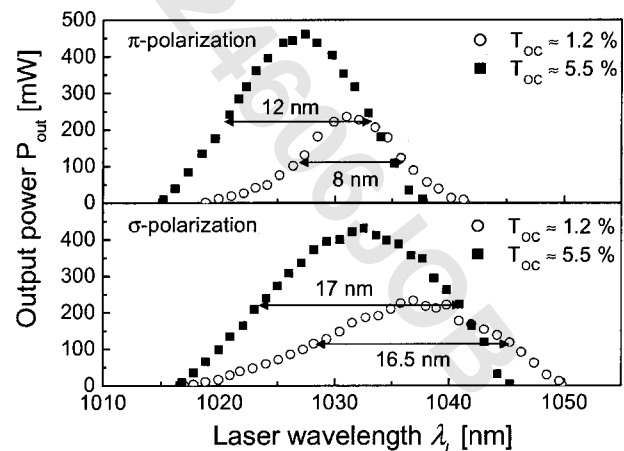

Fig. 8. Wavelength tunability of the Yb:LiGdMo laser under Ti:sapphire laser pumping for an incident pump power 1.8 W and the two polarizations. Two different output couplers were used: T_{OC}=1.2% and 5.5% and in both cases the tunability was limited on the shortwave side by the transmission characteristics of the output couplers.
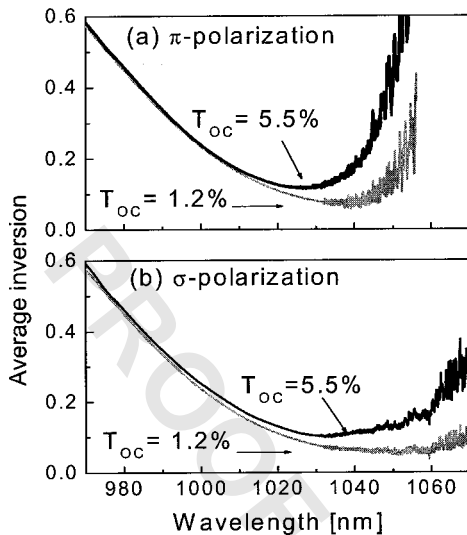
Fig. 9.   Calculated average inversion $\langle \beta \rangle$ necessary to achieve threshold versus wavelength for the two polarizations and two different output couplers (T_{OC}).
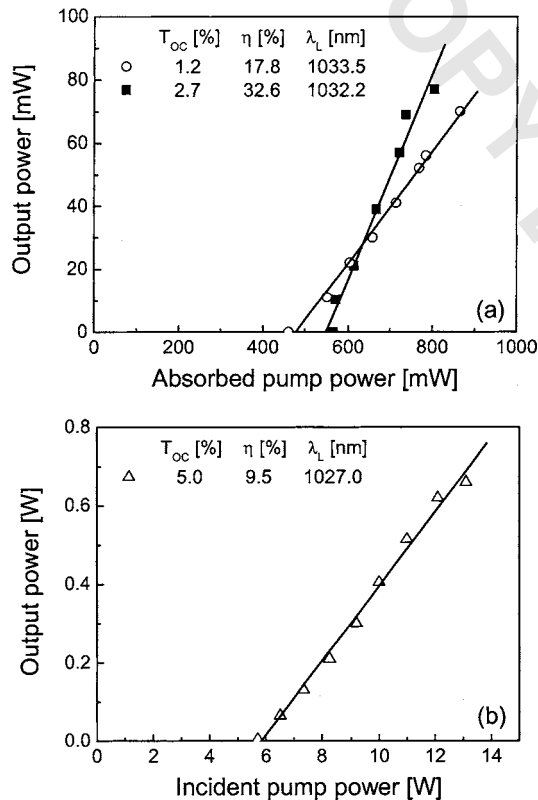

Fig. 10.   Output power of the π-polarized cw Yb:LiGdMo laser pumped (a) by a TDL at 975 nm and (b) by an unpolarized fiber-coupled diode laser module at 980 nm. The lines are the fits used to calculate the slope efficiencies η corresponding to the used output couplers (T_{OC}).

and a 3 mm long tapered amplifier section. The highly reflecting ($>90\%$) facet was $\sim 3\ \mu m$ wide and the output aperture had a width of $300\ \mu m$. The TDL had an output power of up to 1.25 W at $M^2 < 3$ for the slow axis emission. A collimated pump beam could be formed by relatively simple optics. The spectral linewidth near 975 nm was ~ 1 nm. Two output couplers, $T_{OC} = 1.2\%$ and 2.7%, were

used with the same setup depicted in Fig. 5(a). We studied only the π polarization.

The absorption bleaching effect was relatively small for the available pump power: The absorption dropped in the absence of lasing from $\sim 85\%$ to 75% with increasing pump level. With the output coupler aligned, the recycling effect resulted in some recovery to $\sim 81.5\%$. An output power of 77 mW at 1032.2 nm was obtained for $T_{OC} = 2.7\%$ [Fig. 10(a)] and the maximum slope efficiency with this output coupler was $\eta = 32.6\%$. For $T_{OC} = 1.2\%$, the absorbed pump power at the laser threshold was 460 mW.

The 50 W fiber-coupled diode laser module operated near 980 nm and had a linewidth of 4 nm. The fiber core diameter was $200\ \mu m$ with a N.A. = 0.22. The unpolarized pump beam was focused to a spot size of $\sim 100\ \mu m$ using 1:1 imaging optics [Fig. 5(b)]. The best results were obtained with $T_{OC} = 5\%$ [Fig. 10(b)]. Note that they were affected also by the imperfect parallelism of the uncoated sample. Since the pump beam divergence prevented a precise estimation of the absorbed power, the slope was calculated with respect to the incident pump power and is correspondingly lower. The deviation of the fixed pump wavelength from the absorption maximum (the absorption estimated without lasing was only $\sim 37\%$) and the relatively large pump spot size resulted in an increased threshold (5.7 W) and low slope efficiency. Nevertheless the maximum output power achieved was 0.66 W. Without any polarization-selective elements, the laser oscillated in the π polarization.

5. CONCLUSION

We demonstrated, for the first time to our knowledge, laser operation of Yb^{3+} in a Li-based DM host belonging to the general class of disordered $LiT(XO_4)_2$ compounds. Efficient and tunable laser operation of $Yb:LiGd(MoO_4)_2$ was obtained by pumping it near 975 nm with a Ti:sapphire laser. In accordance with the gain characteristics that depend on both the emission and the absorption cross sections, longer wavelengths and broader tunability were observed for σ polarization. Diode pumping with $\sim 50\%$ higher threshold was achieved with a TDL and an output power of 0.66 W was obtained with a fiber-coupled diode laser module. It has been determined that this crystal belongs to the $I\bar{4}$ space group; therefore the presence of two different sites each with multiple environments contributes to the inhomogeneous broadening of the electron transitions that ensures relatively broad laser tunability.

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