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[Ge₂]^{4–} Dumbbells with Very Short Ge-Ge Distances in the Zintl Phase Li₃NaGe₂ – a Solid State Equivalent to Molecular O₂

Lavinia M. Scherf, Antti J. Karttunen, Oliver Pecher, Pieter C. M. M. Magusin, Clare P. Grey, Thomas F. Fässler*

Abstract: The novel ternary Zintl phase Li₃NaGe₂ comprises alkali metal cations and $[Ge_2]^{4-}$ dumbbells. The diatomic $[Ge_2]^{4-}$ unit is characterized by the shortest Ge-Ge distance of 2.390(1) Å observed in a Zintl phase and thus represents the first double bond as also suggested by the (8–*N*) rule. Raman measurements support these findings. The multiple bond character is confirmed by electronic structure calculations, and an upfield ⁶Li NMR shift of –10.0 ppm assigned to those Li cations surrounded by the π systems of three Ge dumbbells further underlines this interpretation. For the unperturbed, ligand-free dumbbell in Li₃NaGe₂ the π -bonding p_y and p_z orbitals are degenerate as in molecular oxygen having singly occupied orbitals. The partially filled π -type bands of the neat solid Li₃NaGe₂ cross the Fermi level, resulting in metallicity. Li₃NaGe₂ was synthesized from the elements as well as from binary reactants and subsequently characterized crystallographically.

Multiple bonds of the heavier homologues of boron, carbon and nitrogen have been intensively investigated during the last decades.^[1-3] Among those, the heavier tetrel homologues *E* all adopt a *trans*-bent structure in contrast to planar ethylene. The pyramidalization at the *E* atom in $R_2E=ER_2$ drastically influences the possibility of forming a classical π -bond between the two metal atoms for *E* = Ge, Sn, and Pb.

The CGMT model proposed by Trinquier and Malrieu^[4] traces the deviation from planarity of $R_2E=ER_2$ molecules back to the relative values of singlet and triplet energies of the monomeric units ER_2 (E_S and E_T, respectively) and the energy of the double bonded system: for $E_{\sigma^+\pi} > 2\Delta E_{S-T}$ a classical planar structure is obtained, whereas $E_{\sigma^+\pi} < 2\Delta E_{S-T}$ favors the *trans*-bent structure. $E_{\sigma^+\pi} < \Delta E_{S-T}$ results in monomeric ER_2 fragments without E E bonding.^[1,5] The interatomic *E*-*E* distance is another strong indicator for the bond order. Scheme 1 gives an overview of several homonuclear, double bonded Si and Ge species. For the second row elements, distances of double bonded atoms are considerably shorter than the sum of the covalent radii, whereas for the heavier homologues

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deviations occur due to the strict requirements of the often bulky ligands used to stabilize the reactive bonds or due to electrostatic repulsion in case of charged systems. For example, $R_2Ge=GeR_2$ bond lengths are in the range of 2.21 to 2.51 Å,^[2] most of them being shorter than the Ge-Ge single bond length of 2.44 Å, but in $[RGe=GeR]^{2^-}$ longer distances are observed.^[6]

In general, *trans*-bent structures are energetically favored over planar structures. Ligand-free and thus symmetrically unperturbed germanide dumbbells have been observed in Zintl phases.^[7] Assuming a complete valence electron transfer in BaMg₂Ge₂ results in a 14 valence electron [Ge–Ge]^{6–} dumbbell. The bond length of 2.58 Å^[8] is however significantly longer than the Ge single bond length and sketchily explained by Coulombic repulsion of the negatively charged atoms. Recently, the strong interaction of alkali earth metal (*Ae*) *d* states with anions in Zintl phases was experimentally established, hinting at a less effective electron transfer in case of *Ae* counter ions.^[9]

Ge dumbbells occur also in binary lithium germanides Li_xGe_{1-x} (0.692 $\leq x \leq 0.778$).^[10-13] However, according to the formal electron transfer of the Zintl Klemm concept they never contain a truly double bonded dumbbell [Ge=Ge]⁴⁻. Either a more highly charged species such as (Li⁺)₁₃(Ge⁴⁻)₂[Ge₂]⁵⁻ for Li₁₃Ge₄ and (Li⁺)₉([Ge₂]^{4.5-})₂ for Li₉Ge₄ or [Ge₂]⁴⁻ units with one or 0.5 excess electrons per dimer, respectively, have been considered. Yet, with decreasing Li content *π*-bonding contributions are expected to increase, and appropriately the Ge–Ge bond lengths decrease from 2.62 Å (Li₁₃Ge₄)^[12,13] to 2.44 Å (Li₉Ge₄)^[10] but in all cases remain longer than the Ge–Ge single bond.

	Ë= <u>Ĕ</u>		$\begin{bmatrix} R \\ \vdots \\ \vdots \\ R \end{bmatrix}^{2}$	[∶ ε ==ε;̀]⁴-
E = Si E = Ge	2.23 Å ^[14] -	2.14-2.29 Å ^{୲2} 2.21-2.51 Å ^{៲2}	- 2.39-2.46 Å ^ø	(≥ 2.34 Å ^[16]) (≥ 2.44 Å ^[10])
				(,

Scheme 1. Types of double bonded heavier tetrel systems and their respective E=E bond lengths. Values in parentheses correspond to tetrel dumbbells in Li₁₄Si₆ and Li₉Ge₄ which are closest to a double bond system.

For silicides, stannides and plumbides^[12] as well as pnicogenides^[16] the situation is similar – quite a few compounds containing dumbbells are known but their electron counts never suggest a true double bond and bond lengths are at least in the range of the elemental single bond.^[17]

Here, we report on the new Zintl phase Li₃NaGe₂ which contains anionic dumbbells [Ge₂]⁴⁻ that appear for the first time in a neat solid with a short Ge-Ge distance that can be characterized as truly double bonded. Furthermore, we show the first Raman spectrum of a solid state Ge=Ge double bond. Since it does not possess covalently bonded ligands it constitutes an analog to molecular O₂. After the recently discovered Li₁₈Na₂Ge₁₇,^[18] Li₃NaGe₂ is only the second compound in the ternary Li-Na-Ge system.

Red, lustrous single crystals of Li₃NaGe₂ were initially found as a product of a flux synthesis in the Li-Na-Ge system with excess Na. Subsequently, the air- and moisture-sensitive compound was synthesized directly from the pure elements which were heated up to 750 °C and characterized crystallographically. Nearly phase-pure samples of Li₃NaGe₂ were obtained from thoroughly mixed Li₁₇Ge₄, NaGe and Ge precursors by dwelling at 400°C.

a)



Figure 1. a) Projections of the unit cell of Li₃NaGe₂ onto the ac- (left) and abplanes (right); coordination spheres of b) a Ge-Ge dumbbell and c) of Li1 and Li3; d) Li1 surrounded by *n*-bonding Ge *p* orbitals (thermal ellipsoids at 90 % probability at 123 K).[19]

Li₃NaGe₂ crystallizes in a new structure type in the orthorhombic space group *Pnma* (No. 62) with *a* = 12.706(3) Å, *b* = 4.4571(9) Å, c = 7.7715(16) Å (Figure 1a).^[20] The structure contains Ge–Ge dumbbells which are coordinated by 15 alkali metal cations (Figure 1b). Three Li1 and three Li3 atoms form a six-membered ring in envelope configuration surrounding the dimer. Four Li2 and five Na1 atoms coordinate the Ge atoms in a terminal fashion. Selected distances for the Ge-Ge coordination sphere are given in [19]. Remarkably, the Ge-Ge bond of 2.390(1) Å is significantly shorter than the Ge single bond (2.44 Å) and shorter than Ge-Ge bonds in known intermetallic compounds. For molecular digermenes the Ge=Ge bond length reduction varies significantly due to the competing influences of steric and electrostatic repulsion. However, in dianions [RGe=GeR]²⁻ they fall in a similar range (2.39–2.45 Å).^[6]

According to the Zintl Klemm concept,^[21] Li₃NaGe₂ can also be described as (Li⁺)₃(Na⁺)[Ge₂]⁴⁻, suggesting a true double bond for the Ge dumbbell. This description and the very short bond length demand a large p- $p \pi$ bonding contribution to the Ge–Ge bond. Typically, for Zintl phases a small band gap at the Fermi level is expected. In this case, however, the Li₃NaGe₂ structure formally contains a double bonded dimer (formal charge of -2 for each Ge atom) which should behave similarly to molecular O2 according to the (8–N) rule. Unlike other double bonded π systems such as ethylene, O_2 does not have a preferred orientation for π -bonding because two sets of perpendicular p orbitals are available for π bonding like in triple bonded systems. This results in single occupation of degenerate antibonding MOs and paramagnetism of O₂. The coordination sphere of the Ge dumbbell in Li₃NaGe₂ does not show any preferred orientation and hypothetical [Ge₂]⁴⁻ in vacuum is also paramagnetic (Figure 2c). Thus, Li₃NaGe₂ may be viewed as a solid state equivalent to O₂.

In contrast, partially filled bands in the solid state result in metallic properties. A significant density of states at the Fermi level, partially occupied π -antibonding Ge(p) orbitals and thus metallic behavior are confirmed by a thorough electronic structure analysis using TB-LMTO-ASA^[22] (for details on the calculations see SI). The total and partial DOS are depicted in Figure 2a. Indeed, Li₃NaGe₂ does not have a band gap at the Fermi level E_F. The Ge(s) states are largely situated below -6 eV and thus do not contribute significantly to the bonding in Li₃NaGe₂. In the range of -5 to 4 eV, the total DOS is dominated by Ge(p) states involved in Ge-Ge bonding. Li(s) and Na(s) states are also partially occupied in this region and therefore play a non-negligible role in the overall bonding and stability of Li₃NaGe₂.

In order to elucidate the Ge p-p π bonding in Li₃NaGe₂ more closely, the Crystal Orbital Hamilton Population (COHP) of the Ge-Ge bond was calculated and a detailed band structure analysis performed. The COHP (Figure 2b) indicates that, as expected, some π -antibonding Ge(p) states are occupied at the Fermi level. The -iCOHP at Fermi level yields 3.27 eV bond⁻¹, affirming the large *p*-*p* π bonding contribution. For comparison, Ge dumbbells in Ae_7Ge_6 (Ae = Ca, Sr, Ba) which have been proven to feature significant π bonding exhibit –iCOHPs of 2.48 to 2.92 eV [23]



Figure 2. a) Total and partial DOS curves in the range of -12 eV to 4 eV for Li₃NaGe₂ (total: solid black, Ge1(s), Ge2(s): dotted black, Ge1(p), Ge2(p): dashed black, Li1(s), Li2(s), Li3(s): dotted grey, Na1(s): dashed grey); b) COHP (solid line) and integrated COHP (dotted line) of the Ge1–Ge2 bond in Li₃NaGe₂, iCOHP at $E_F = 3.27$ eV bond⁻¹; c) molecular orbital pictures of free [Ge₂]^{4⁻} and schematic MO diagrams of free [Ge₂]^{4⁻} and O₂ calculated at the DFT-PBE0/def2-TZVP level of theory (s. SI, the p_x orbital is chosen to be parallel to the bond axis as in Li₃NaGe₂); d) total DOS curve as well as band structures including Ge(p) fatbands in the range of -4.5 eV to 4 eV for Li₃NaGe₂.

A fatband analysis of the Ge(p) states serves to clarify the π bonding situation of [Ge₂]⁴⁻ (Figure 2d). Although the Ge dumbbell is not aligned exactly with any cell parameter, a separate view of the p_x , p_y , and p_z orbital contributions clearly shows that the Ge anion may justifiably be related to molecular O₂: The p_x orbitals are oriented almost parallel to the Ge–Ge bond and the corresponding bands are all located below E_F with respective antibonding bands well above E_F , representing the Ge–Ge σ bond. The p_y and p_z orbitals are oriented nearly perpendicular to the bond axis and are therefore available for π bonding. The corresponding bands are located around E_F , demonstrating a significant π bonding contribution. As also seen in the COHP analysis, the π -antibonding p_y and p_z bands are partially occupied, similar to molecular O₂.

According to magnetic measurements (see SI), Li_3NaGe_2 is diamagnetic with a molar magnetic susceptibility of -4.0 x 10^{-5} emu/mol at 5000 Oe, suggesting that the diamagnetic

contributions of Li⁺, Na⁺, and the Ge core and 3*d* electrons superimpose Pauli-paramagnetic contributions from the conduction electrons of the metal.^[24]

The Raman spectrum of Li₃NaGe₂ is shown in Figure 3. We assigned the Raman bands by utilizing density functional calculations (DFT-PBE/TZVP level of theory, see SI for full computational details). It was not possible to calculate Raman intensities because Li₃NaGe₂ shows metallic conductivity, but calculating the wavenumbers of the Raman active modes and inspecting the corresponding normal modes still enabled the complete interpretation of the spectrum (Table S1). The broad band centered at about 350 cm⁻¹ arises from vibrational motion of the Li atoms. The 230 cm⁻¹ band is due to Ge=Ge stretching vibrations of the Ge dumbbell. The calculated wavenumbers are slightly overestimated in comparison to experiment, showing one symmetric Ge=Ge stretching modes at 230 and 224 cm⁻¹.^[25] This

stretching frequency is in the range known from Zintl polyanions such as $[Ge_9]^{4-}$ (220–222 cm⁻¹) and $[Ge_4]^{4-}$ (274 cm⁻¹).^[26] Finally, the band centered at 90 cm⁻¹ is due to tilting and low-energy motion of the Ge dumbbell and falls in the range of libration modes of $[Ge_9]^{4-}$ clusters.^[27]



Figure 3. Raman spectrum of a Li₃NaGe₂ single crystal.

⁶Li and ²³Na magic angle spinning (MAS) NMR spectra were recorded for ⁶Li-enriched Li₃NaGe₂ (Figure 4). The spectra can be consistently interpreted with the aid of 2D ⁶Li spin-exchange, ⁷Li 1D and ²³Na MQMAS NMR (Figures S8 to S10) in combination with DFT-PBE calculation of chemical shifts and quadrupolar NMR parameters for Na₇LiGe₈ and Li₁₄Ge₆ (cf. SI for details on the experiments and calculations Tables S2, S3,).

The ⁶Li NMR spectrum contains two major signals at 94.5 and -10.0 ppm with integral ratio 1:2, which belong to Li₃NaGe₂. The nearness of the corresponding Li atom positions in the unit cell is proven by ⁶Li spin-exchange NMR (Figure S9). Negative shifts are known from Li⁺ located in π electron clouds such as in aromatic [Ge₅]⁶⁻ in Li₁₂Ge₇ (-17 ppm)^[28] or in the Li cyclopentadienide (Cp) monomer (-7.6 ppm)^[29]. Li1 and Li3 are surrounded by π -electrons of three Ge dumbbells (Figure 1). Therefore, we assign the signal at -10.0 ppm to these two atom positions (Figure 4a). The two Li sites are not resolved due to their similar coordination spheres. The Knight shifted ⁶Li signal at 94.5 ppm can be attributed to Li2 which coordinates the Ge dumbbells terminally (Figure 1b). Minor signals at 1.9 and 28.1 ppm belong to Na₇LiGe₈^[30] and Li₁₄Ge₆^[31] impurities, respectively (cf. SI Figure S11 and Table S2).

Apart from minor spinning sidebands, the ²³Na NMR spectrum shows two major and one minor center-band signals (Figure 4b), of which the highest at 315 ppm belongs to Na1 in Li₃NaGe₂. Like Li2, this Na environment has a Knight shift well outside the diamagnetic shift range, which is explained by the fact that it possesses a similar atomic environment and experiences the same local electronic band structure (cf. SI). The signals at -8.0 and 38.0 ppm are assigned to the impurity Na₇LiGe₈ also observed with ⁶Li NMR. The interpretation of 1D NMR spectra are corroborated by ²³Na MQMAS NMR and DFT-PBE calculations (Figure S10, Tables S2, S3).



Figure 4. a) ⁶Li and b) ²³Na MAS NMR spectra of ⁶Li₃NaGe₂ (15 kHz MAS spinning speed, 11.7 T, ambient temperature). Signal assignments for the main phase and the respective crystallographic positions in black as well as for the impurity phases in grey. Rotational sidebands are marked by asterisks.

In conclusion, we have discovered the novel phase Li₃NaGe₂ which contains $[Ge_2]^{4-}$ dumbbells. Ge–Ge distance and electronic structure calculations clearly indicate a true Ge=Ge double bond with a strong influence of its π electron system on ⁶Li NMR shifts. Li₃NaGe₂ can be viewed as a model system for heavier element multiple bonds in molecular compounds without preferential orientation of the π bonds, rendering it an example of a solid-state equivalent to molecular O₂. The resulting degeneracy of π -bonding *p* orbitals which is well known for paramagnetic O₂ leads to metallic behavior in the title compound.

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Keywords: germanium dumbbell • double bond • lithium • sodium • Zintl phases

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 Ge2–Na: 3.385(2), Li1–Li2: 2.80(2), Li1–Li3: 2.600(10)–2.65(2), Li2–
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Entry for the Table of Contents (Please choose one layout)

Layout 1:

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A true double bond in neat solids between two Ge atoms is observed in Li₃NaGe₂ and theoretically established by electronic structure calculations. The p bond character of the [Ge₂]⁴⁻ dumbbells is experimentally confirmed by the upfield ⁶Li NMR shift of the coordinating Li cations. As in molecular O₂ the π -bonding orbitals have no preferential orientation, are degenerate and partially filled, but are expanded into bands, resulting in



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