

Hydrogen Retention in Tungsten Materials Studied by Laser Induced Desorption

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Introduction

Tritium retention in plasma facing components (PFCs) due to plasma wall interactions is one of the most critical safety issues for ITER and future fusion devices. Tungsten is foreseen as PFC material in the divertor of ITER and the most promising candidate of PFCs in future reactors. Its fuel retention behaviour is subject of present R&D.

In this work the retention of fuel (deuterium) in bulk tungsten has been studied applying laser induced desorption (LID). This method allows the local measurement of hydrogen isotopes and is also proposed for in situ diagnostic at ITER to monitor tritium retention.

Trapping of hydrogen in tungsten is strongly dependent on material properties and temperature during plasma exposure. Both effects can influence the results obtained by laser induced desorption spectroscopy.

Diagnostic Method

Laboratory:

Laser Induced Desorption with Quadrupole Mass Spectrometer

LID: Nd:YAG laser: $\lambda = 1064 \text{ nm}$, $E \leq 40 \text{ J}$, $t_{\text{pulse}} \leq 3 \text{ ms}$,

absorbed intensity: $P/A \leq 2 \text{ GW/m}^2$, $f = 6 \text{ cm}$ focussing lens

QMS: scan 1-50 amu in 1 s, pressure calibration in H_2 , D_2 , CH_4 , CD_4 , ...

TEXTOR: Laser Induced Desorption Spectroscopy

LID: Nd:YAG laser: $\lambda = 1064 \text{ nm}$, $E \leq 40 \text{ J}$, $t_{\text{pulse}} \leq 3 \text{ ms}$,

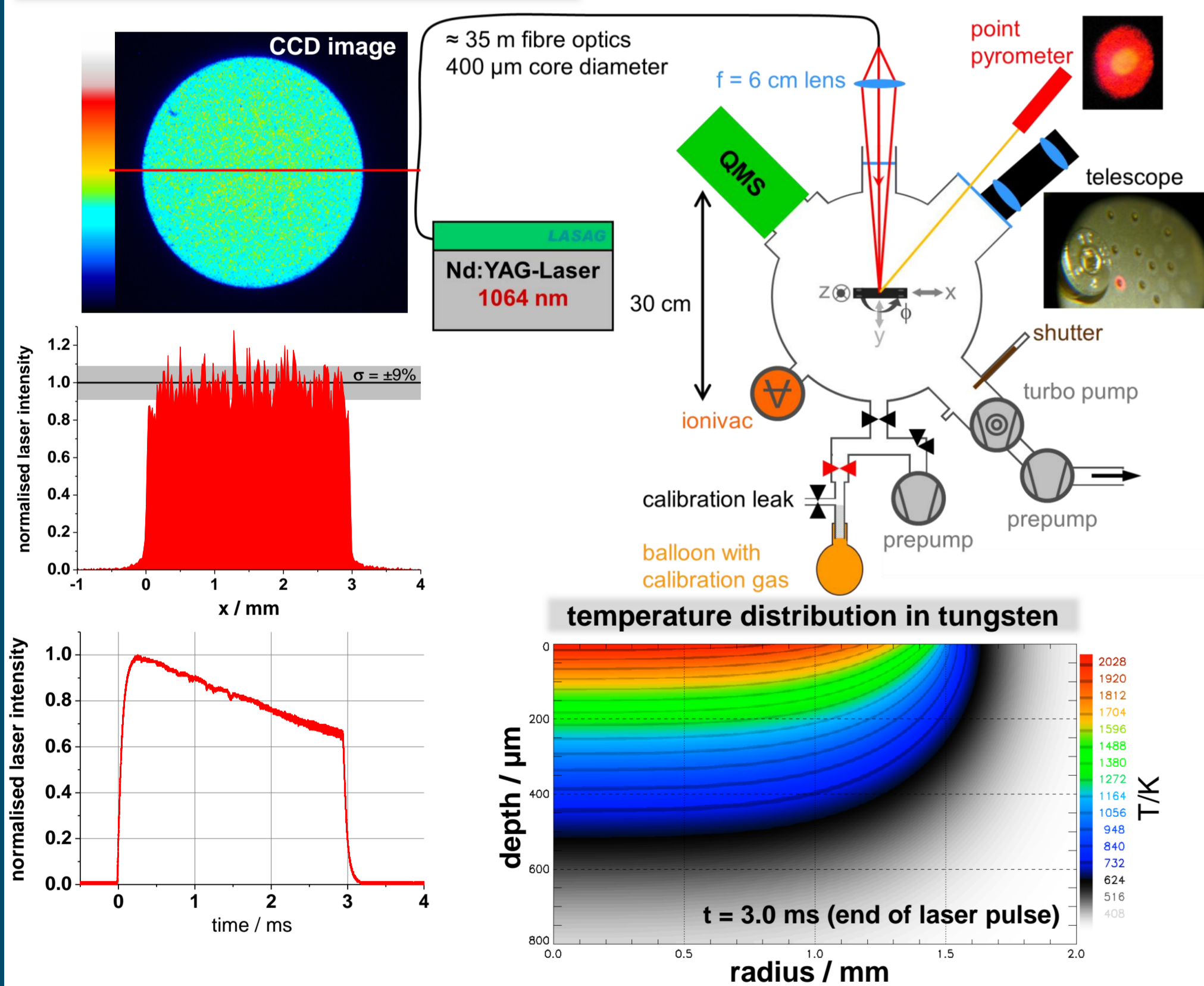
absorbed intensity: $P/A \leq 500 \text{ MW/m}^2$, $f = 30 \text{ cm}$ focussing lens

LIDS: optical spectroscopy of Balmer Alpha lines (H_α)

spatial resolution: 3 mm

spatial distribution and temporal development of standard laser pulse at the W target surface

LID-QMS experimental set-up in laboratory



Glow Discharge Exposure

unpolished Goodfellow tungsten, annealed at 1273 K for >1 hour

low flux: $2.8 \times 10^{19} \text{ D/m}^2\text{s}$, $T_{\text{expo}} = 533 \text{ K}$

fluence: $2 \times 10^{23} \text{ D/m}^2$ with 2 h exposure time

twin sample: $7.35 \times 10^{19} \text{ D/m}^2$ (TDS) after D_2^+ exposure in RF glow discharge

laser parameter optimization on homogeneously loaded W by glow discharge

pulse length variation, but keeping the heat flux factor constant by energy adaptation

$$I\sqrt{t} = \frac{P}{A}\sqrt{t} = \frac{E}{A\sqrt{t}}$$

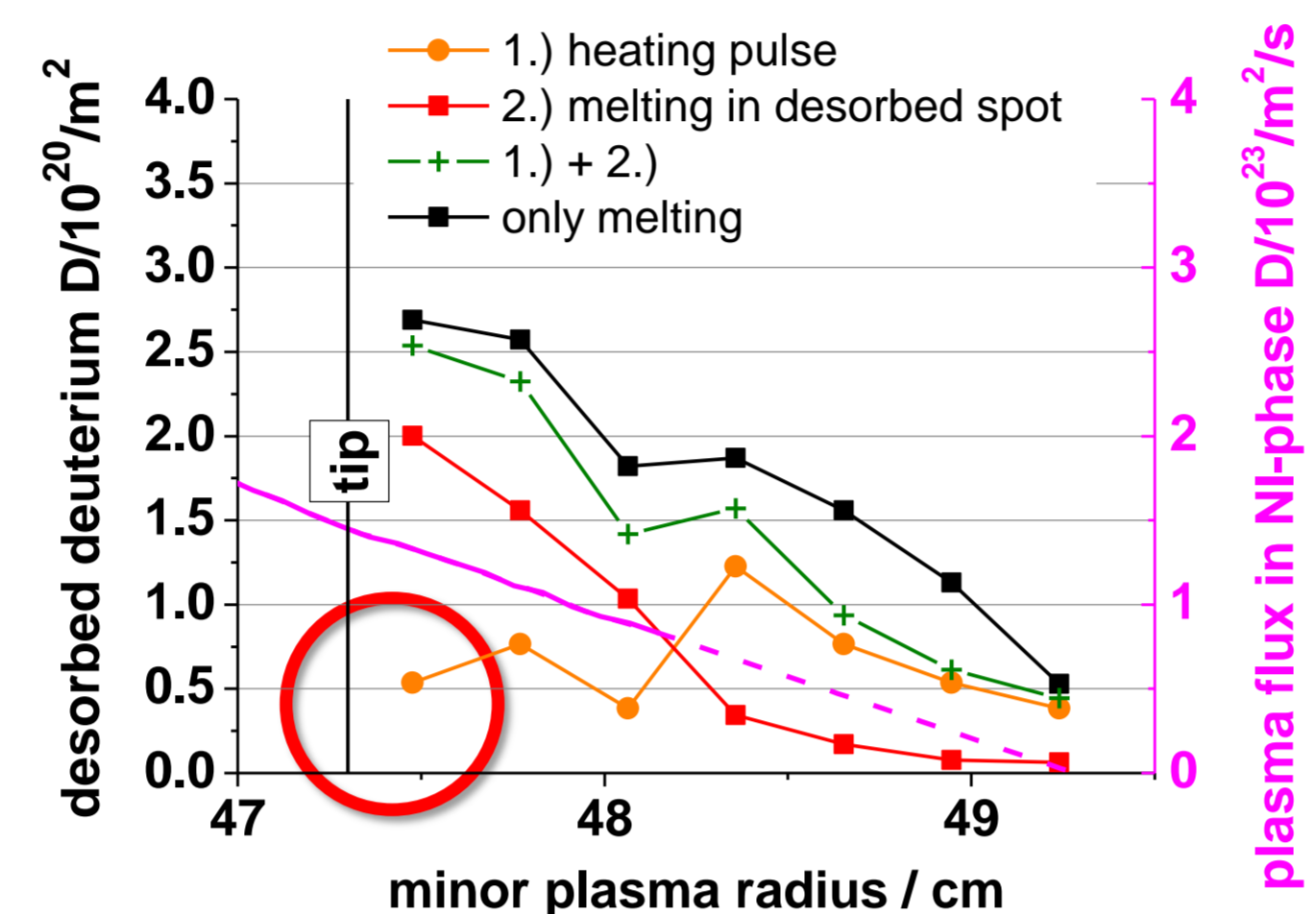
pulse length =	0.5	1.5	3	6	12	3 ms
E _{absorbed} =	3.2	5.6	7.9	11	16	J
I =	1.6	0.9	0.65	0.5	0.33	GW/m ²
E/A =	0.8	1.4	2	2.8	4	MJ/m ²
heat flux factor =	36	36	36	36	36	MW/m ² s ^{1/2}

65% desorbed

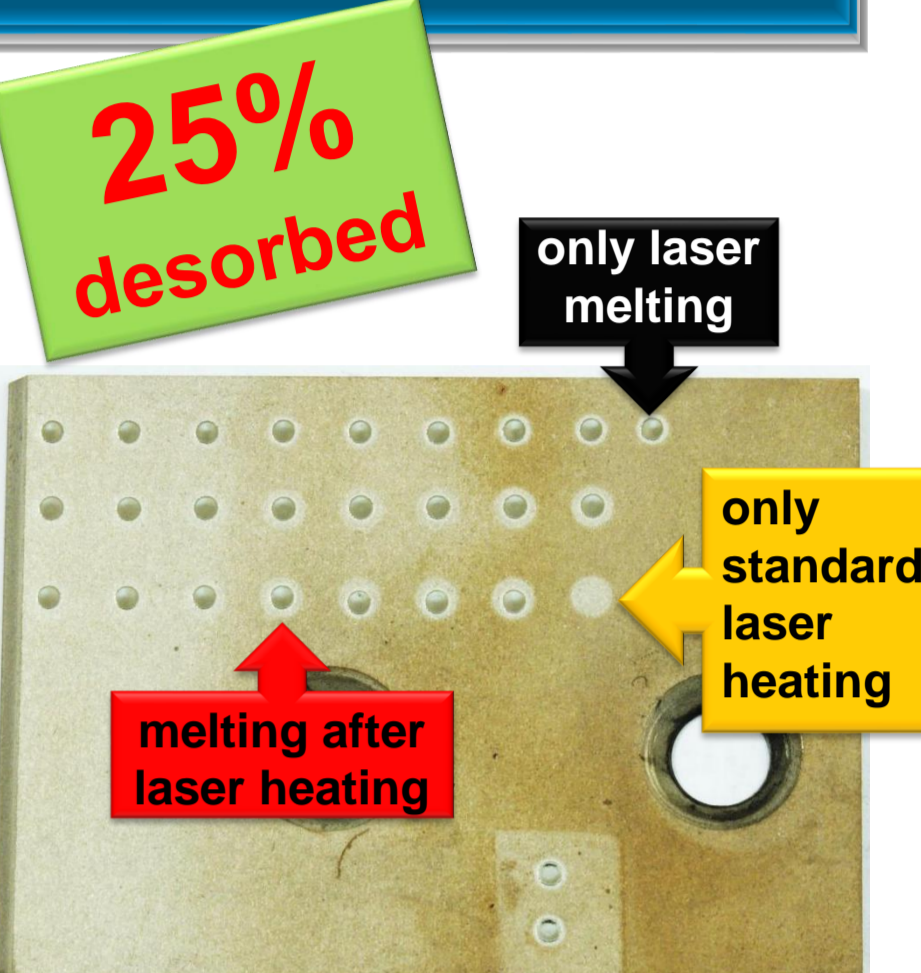
- short laser pulses \rightarrow time for hydrogen diffusion short \rightarrow low desorption
- too long pulses \rightarrow lateral heat diffusion
- a) reduced maximum temperature \rightarrow low desorption
- b) fuzzy temperature profile \rightarrow fuzzy desorption volume
- partial overlapping of neighbouring desorption volumes
- scatter

Results

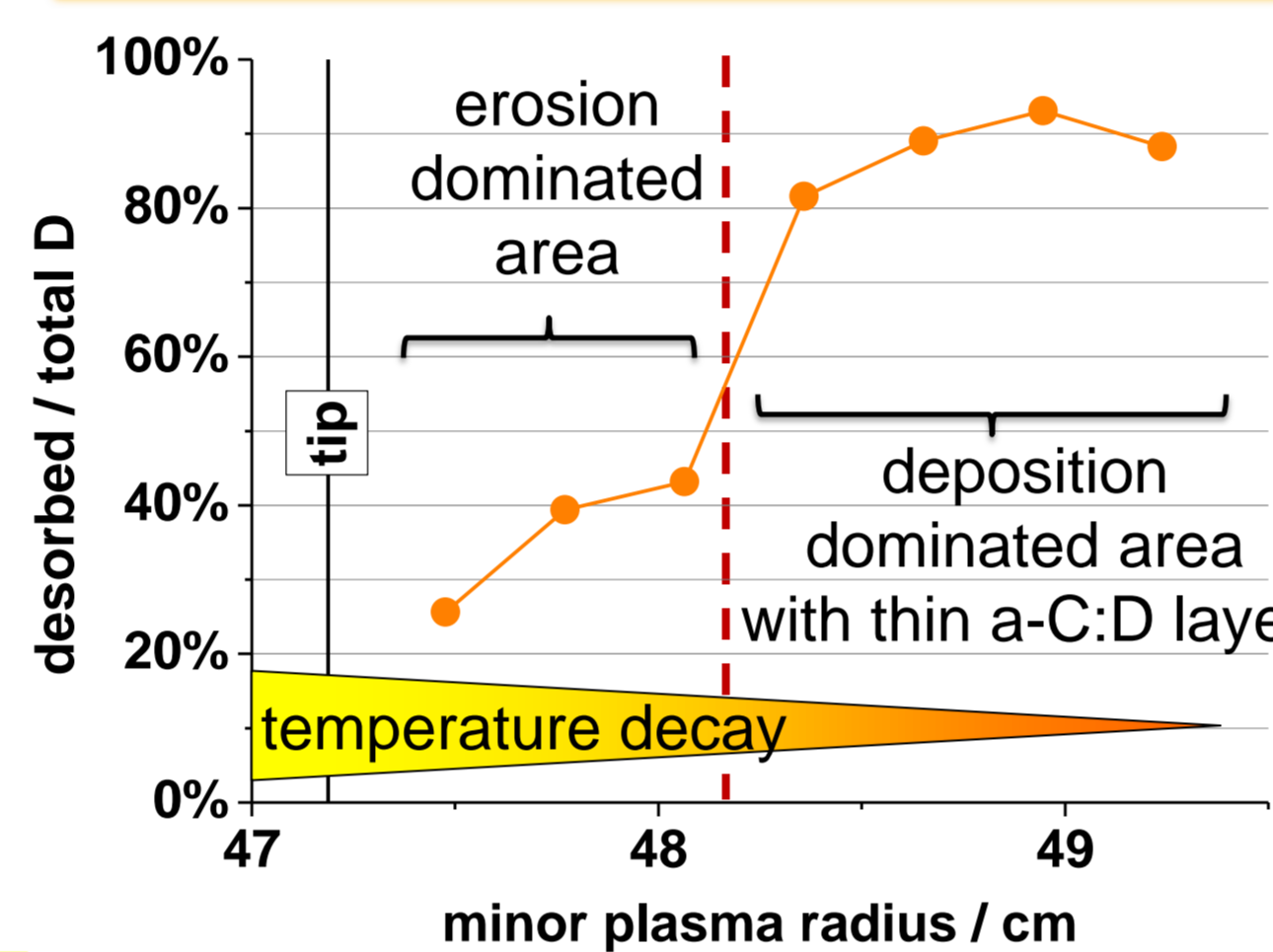
TEXTOR Tokamak Plasma Exposure



unpolished Goodfellow tungsten, annealed at 1273 K for >1 hour
medium flux: $1.5 \times 10^{23} \text{ D/m}^2\text{s}$
fluence: $1.7 \times 10^{25} \text{ D/m}^2$ with 110 s exposure time
with flux profile decay in SOL
surface temperature: 400 - 650 K, temperature excursions 100-150 eV D and minority impurity fluxes of C (2-4 %), O and He
bulk W plate on roof limiter, sample tip at 47.3 cm (LCFS: 46 cm)
exposed at: $n_e \leq 5 \times 10^{18} / \text{m}^3$, $T_e \leq 30 \text{ eV}$
post mortem analysis by LID:
1.) laser heating pulse: 500 MW/m², ϕ 3 mm, 1800 K
2.) melting by laser: 1.6 GW/m², ϕ 2 mm, >4000 K



Melting pulse after heating pulse proves incomplete desorption.



Thin amorphous hydrocarbon layers on the surface prevent D diffusion into the W bulk because D implantation zone is now in the layer.

Combination of heating and melting pulses identifies areas with different retention mechanisms.

High Flux Exposure (Pilot-PSI)

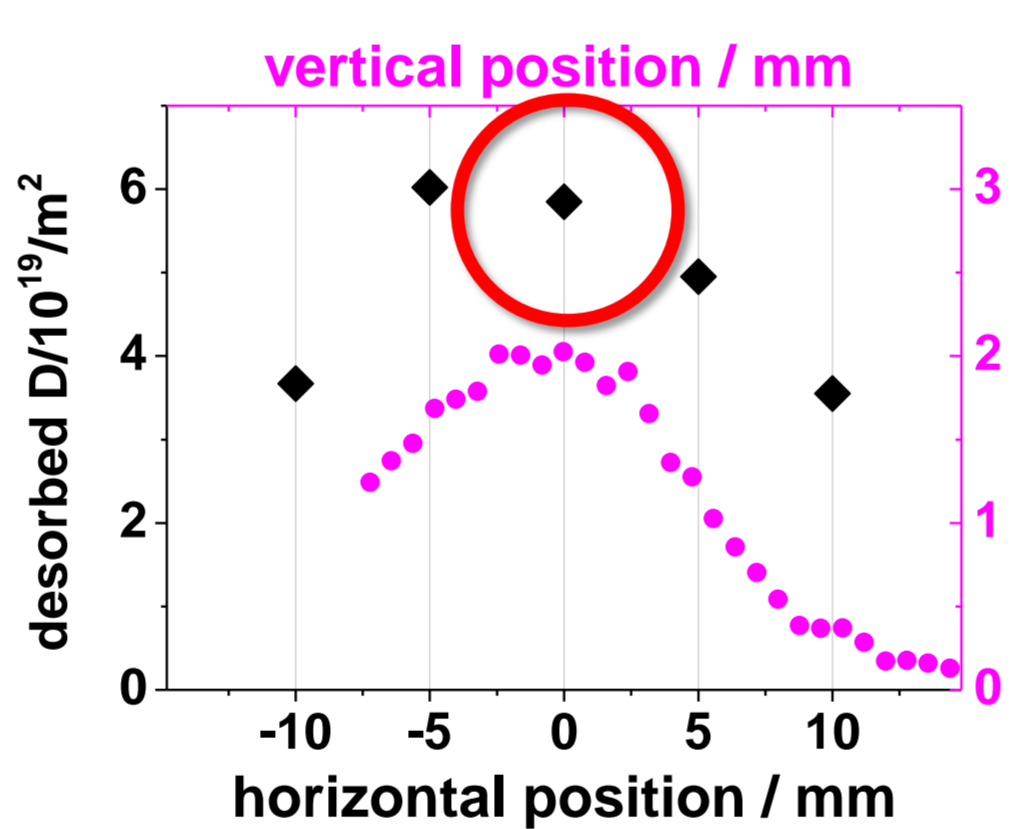
polished Goodfellow tungsten, annealed at 1273 K for >1 hour

high flux: $2 \times 10^{24} \text{ D/m}^2\text{s}$

fluence: $2 \times 10^{25} \text{ D/m}^2$ with 10 s exposure time

$T_{\text{max}} = 500\text{-}580 \text{ K}$, temperature profile according to beam profile, direct water cooling

1.2 eV D, $3.7 \times 10^{20} / \text{m}^2$ with additional -55 V biasing



Similar fluence as in TEXTOR, yields similar amount of desorbed D by standard heating pulse.

Horizontal plasma maximum shift to the left induced by biasing is visible in D inventory.

ECR Plasma Exposure

unpolished Plansee-tungsten, annealed at 1200K for 1 hour

low flux: $10^{20} \text{ D/m}^2\text{s}$ at 370 K

deuterium inventory: TDS: $7.5 \times 10^{20} \text{ D/m}^2$

fluence: $6 \times 10^{24} \text{ D/m}^2$ with 18 h exposure time

ECR heated plasma with 38 eV/D

91% desorbed

Best match of LID with other methods:

TDS: $7.5 \times 10^{20} \text{ D/m}^2$

LID: $6.8 \times 10^{20} \text{ D/m}^2$

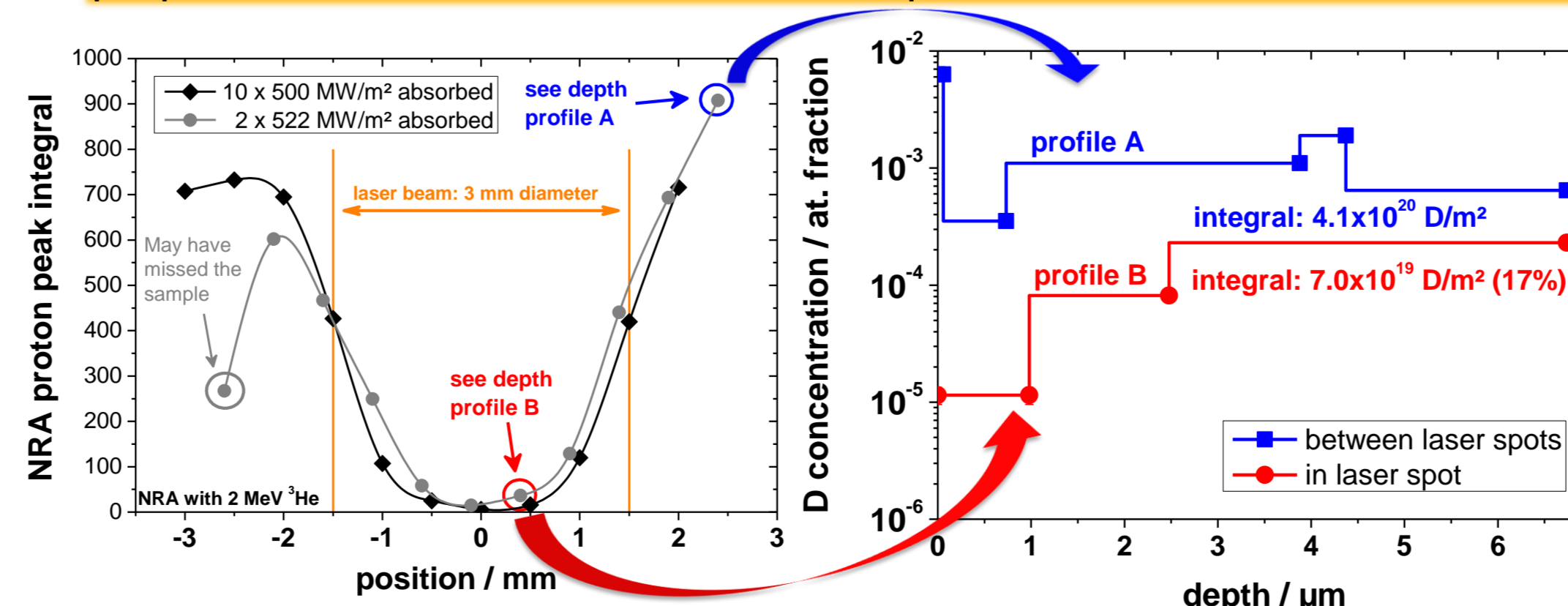
NRA up to 12 μm (with 6 MeV ^3He):

$5.5 \times 10^{20} \text{ D/m}^2$

NRA up to 7 μm (with 2 MeV ^3He):

$4.1 \times 10^{20} \text{ D/m}^2$

general observation for all samples: Repetitive standard heating pulses on one position yield <10% of first pulse signal due to lateral and perpendicular D diffusion into the desorption volume.



NRA scan over laser spots qualitatively shows the lateral D depletion in the first 7 μm

NRA depth profiles quantitatively prove >83% desorption in first 7 μm and decrease of surface concentration by \sim factor 1000. Compare with diffusion modeling!

Summary

Aim: Measurement of hydrogen isotopes in tungsten by spot laser heating and plasma spectroscopy and/or mass spectroscopic detection (QMS)
 \rightarrow in situ hydrogen monitoring during plasma operation

This work: Laser induced desorption and QMS detection, comparison with NRA and slow ramp thermal desorption

Loading conditions:

$370 \text{ K} \leq T_w \leq 1500 \text{ K}$

fluence: $2 \times 10^{23} \text{ D/m}^2 \leq \Gamma \leq 2 \times 10^{25} \text{ D/m}^2$

Laser desorption (single heating pulse)

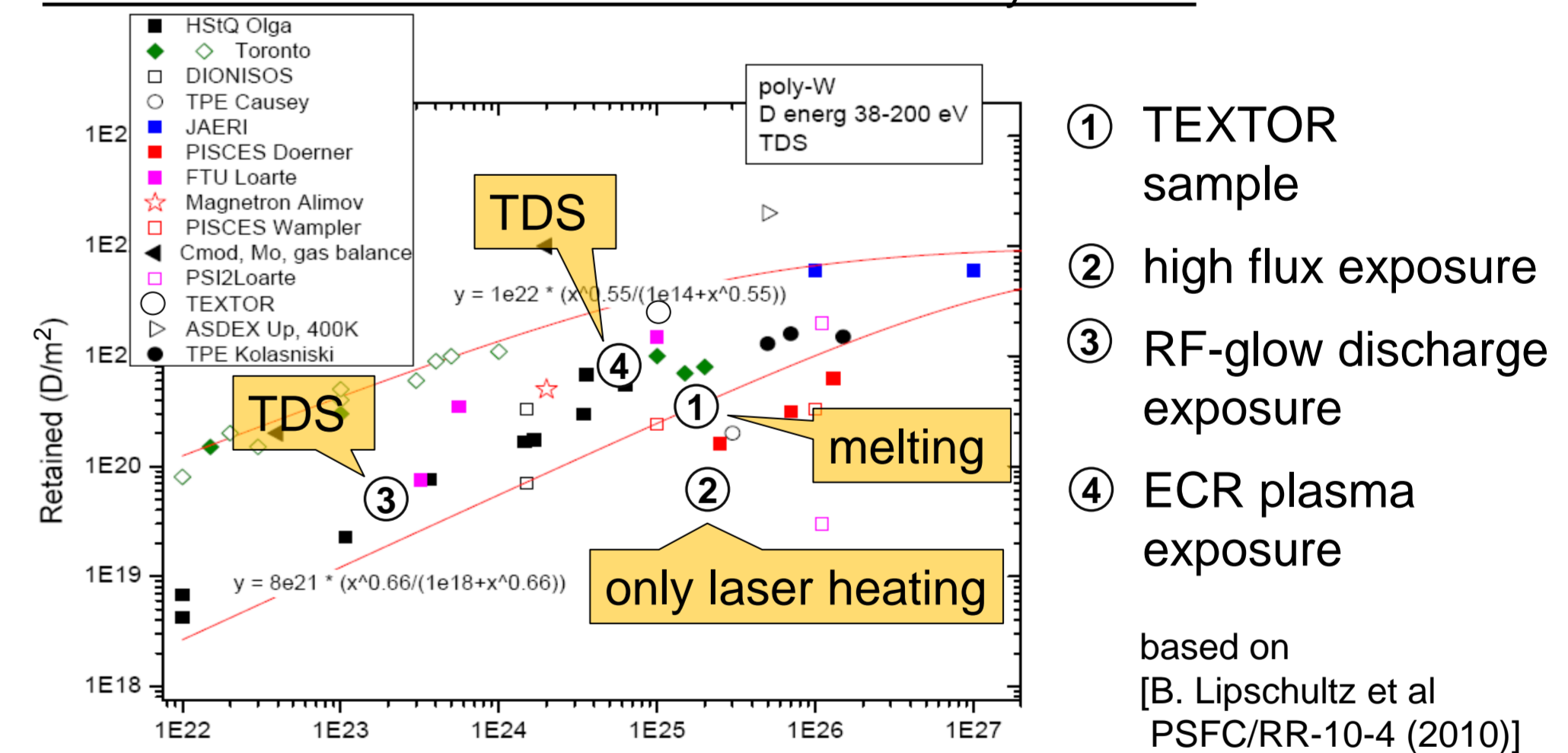
- > 90 % of fuel from a-C:H layers on W, C, CFC and from bulk C
- max 83 % from bulk W, for low T ($\approx 450 \text{ K}$, Plansee material)
- decrease of released hydrogen with increasing loading temperature down to $\sim 20\%$

Reasons:

- deep diffusion of part of H in the material for high T_w or long t_{exposure}
- different material properties with increased hydrogen storage on traps with higher binding energy

local spot melting releases all stored fuel within depth of $\sim 100 \mu\text{m}$
Strategy: frequent desorption of the same monitoring spots on first wall before the fuel is out of laser desorption range

Results in context to literature and community values:



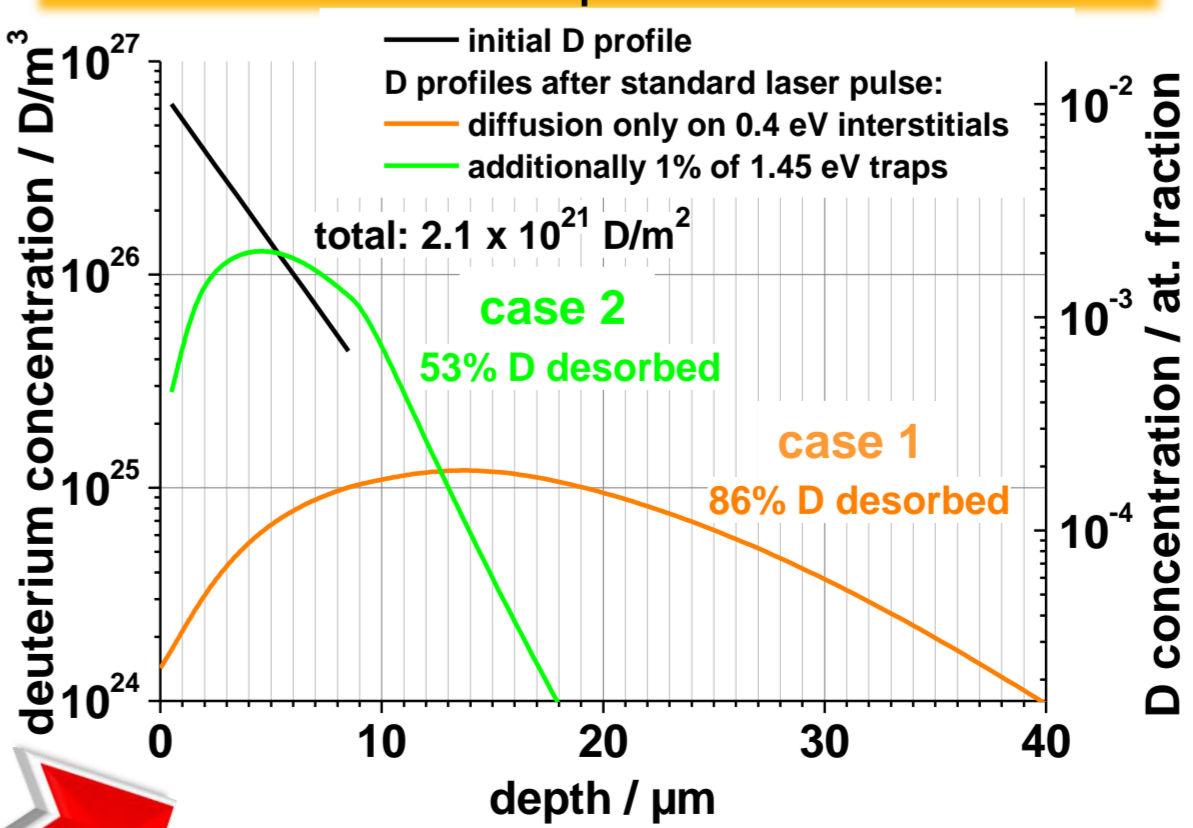
Laser Heating Simulation

1D-code for diffusion of heat and particles: TMAP7 [G. R. Longhurst et al, INEEL/EXT-04-01657 (2008)]

assumed initial D profile up to 9 μm with 3 μm decay length performing standard heating pulse (500 MW/m², 3 ms)

modeling shows: Stronger trapping and deeper deuterium start profile reduce laser desorption.

case 1: only interstitial sites as "traps" diffusion on interstitial traps: $D = 4.1 \times 10^{-7} e^{-0.4eV/kT} \text{ m}^2/\text{s}$ (experimentally determined by Frauenfelder, 1968)



case 2: additionally 1% of lattice density are 1.45 eV traps diffusion on these traps only: $D = 1.5 \cdot 10^{-7} e^{-1.45eV/kT} \text{ m}^2/\text{s}$

The inward diffusing fraction of the inventory can not be desorbed (except it reaches a grain boundary). The profile maximum shifts deep into the bulk.