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Ab initio lattice dynamics simulations and inelastic neutron scattering spectra for studying phonons in BaFe₂As₂: Effect of structural phase transition, structural relaxation, and magnetic ordering

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We have performed extensive ab initio calculations to investigate phonon dynamics and their possible role in superconductivity in BaFe2As2 and related systems. The calculations are compared to inelastic neutron scattering data that offer improved resolution over published data [Mittal et al., Phys. Rev. B 78, 104514 (2008), in particular at low frequencies. Effects of structural phase transition and full and/or partial structural relaxations, with and without magnetic ordering, on the calculated vibrational density of states are reported. Phonons are best reproduced using either the relaxed magnetic structures or the experimental cell. Several phonon branches are affected by the subtle structural changes associated with the transition from the tetragonal to the orthorhombic phase. Effects of phonon-induced distortions on the electronic and spin structure have been investigated. It is found that for some vibrational modes, there is a significant change in the electronic distribution and spin populations around the Fermi level. A peak at 20 meV in the experimental data falls into the pseudogap region of the calculation. This was also the case reported in our recent work combined with an empirical parametric calculation [Mittal et al., Phys. Rev. B 78, 104514 (2008)]. The combined evidence for the coupling of electronic and spin degrees of freedom with phonons is relevant to the current interest in superconductivity in BaFe₂As₂ and related systems.

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I. INTRODUCTION

Every time a new class of superconducting material is discovered, another door is opened to an improved understanding of the phenomenon of high-temperature superconductivity. This holds, in particular, when transition temperatures beyond some 20 K are reached. Examples are the research activities triggered by the discovery of superconductivity in doped fullerenes, boron carbides, and MgB₂. Despite the undeniable new insights procured by these studies, the question of high-temperature superconductivity continues to elude description by physical models. Oxypnictides, a family of superconductors with rather high transition temperatures that share structural and electronic two dimensonality with copper oxides, offer a new class of materials to study high-temperature superconductivity.

Superconductivity has been evidenced in iron oxypnictide LaOFeP (Ref. 1) with $T_c \sim 4$ K, increasing to ~ 7 K with F doping.² A similar transition temperature has been measured for nickel oxypnictide LaONiP (Ref. 3) ($T_c \sim 4$ K). An unexpectedly high T_c ($T_c \sim 26$ K) was recently discovered in the F-doped iron oxypnictide La $[O_{1-x}F_x]$ FeAs.^{4,5} The parent compound LaFeAsO offers a unique picture of the chemical bonding; dominantly ionic in the LaO layers and covalent in the FeAs layers. Thus charge transfer can be assumed to occur according to (LaO)+(FeAs)-. It is established that the superconductivity in LaFeAsO originates mainly from structural and electronic properties of the FeAs layers. Similar structural and electronic features have been observed in the ternary oxygen-free iron arsenide BaFe2As2. The only difference between the two compounds is that the FeAs layers, which are in principle electronically identical in both cases, are separated in BaFe2As2 by Ba atoms instead of LaO layers. In this case the charge transfer occurs according to Ba²⁺_{0.5}(FeAs)⁻. BaFe₂As₂ becomes superconducting under pressure at 29 K (Ref. 6) and exhibits a phase transition around 140 K involving a structural transition (distortion) from tetragonal (high temperature) to orthorhombic (low temperature). 15 BaFe₂As₂, featuring the simplest possible buffer in between the FeAs layers, is therefore ideal for studying the coupling of nuclear and electronic degrees of freedom.

Phonons and electron-phonon coupling have recently been studied computationally in the parent compound LaOFeAs.^{7,8} It is generally concluded that conventional electron-phonon coupling is insufficient to explain superconductivity in the family of Fe-As-based superconductors, while some reports in literature⁹ indicate the possibility of nonconventional electron-phonon coupling. The example of MgB₂ showed that phonon coupling in conjunction with particular Fermi-surface nesting effects is a viable way to describe transition temperatures of up to 40 K.¹⁰ If phonons do not directly mediate the electron pairing interaction, other excitations are required and spin fluctuations have been considered and measured in BaFe₂As₂.¹¹

Experimentally, phonons in Fe-As superconductors have been investigated by inelastic neutron scattering (INS) in fluorine doped LaOFeAs (Ref. 12) and BaFe₂As₂. ¹³ Thermal neutrons were used, allowing spectra to be measured in neutron downscattering from 4 K up to room temperature, but this spectrometer configuration tends to limit energy resolution for low-frequency modes, compared to using cold neutrons in an upscattering configuration. In LaOFeAs, experimental data was compared with published ab initio phonon calculations. 14 The agreement between experiment and calculation is reasonable, corresponding band frequencies being within a few meV, but this level of inaccuracy leads to strong peaks at 30 and 40 meV in the experimental spectra being compared to a single peak at 35 meV in the calculation. In the case of BaFe₂As₂ the experimental resolution was not sufficient to see evidence of the structural phase transition at 140 K in the phonon spectrum. In order to better evaluate coupling between electronic and nuclear structure and phonons, higher-resolution INS data is required that can be rigorously compared with ab initio calculations.

In this context, we present a combined theoretical and experimental investigation of the lattice dynamics of BaFe₂As₂, including coupling to electronic and spin degrees of freedom. We report higher resolution INS spectra, above and below the phase-transition temperature, which complement our published data. The generalized vibrational density of states (DOS) is calculated for both phases using *ab initio* methods and allows the atomic contributions to the phonons to be evaluated. Phonon calculations have been repeated on experimentally determined crystal structures and on relaxed geometries, including spin polarization, in order to identify critical structural degrees of freedom. Finally the effect of phonons on the spin-resolved electronic density of states is investigated.

II. EXPERIMENTAL DETAILS

The neutron spectroscopy measurements were performed on the cold time-of-flight spectrometer IN6 (Institut Laue Langevin, Grenoble, France), in upscattering mode using an incident wavelength of 5.1 Å and time focusing at 4 meV, giving very good resolution below 20 meV. Compared to our previous work, ¹³ where the thermal-neutron time-of-flight spectrometer IN4 was used in downscattering, we have better resolution but cannot access the full spectrum below a temperature of 140 K. The measurements from the two instruments are in this respect fully complementary and both structural phases can just be measured on IN6. Two grams of sample of polycrystalline BaFe2As2 were prepared and characterized as reported elsewhere. 15 The sample was held in a thin Al-foil container and thermalized within a nitrogen cryoloop for measurements at 140 and 300 K. The container scattering was subtracted and the detectors calibrated with a vanadium standard. After correction for energy-dependent detector efficiency, the one-phonon generalized phonon density of states (GDOS) was obtained from the angle-integrated data (17 to 114°) using the incoherent approximation. ¹⁶ Multiphonon terms to the scattering were determined selfconsistently in an iterative procedure. Figure 1 (bottom

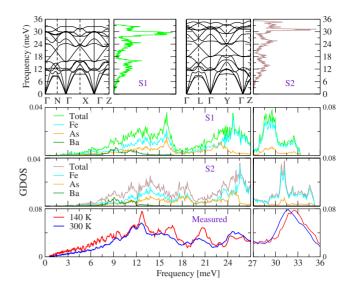


FIG. 1. (Color online) Top panels: dispersion relations and vibrational DOS for tetragonal (S1) and orthorhombic (S2) phases. For the tetragonal phase the high-symmetry points are N = (1/2,0,0), X = (0,0,1/2), and Z = (1/2,1/2,-1/2). For the orthorhombic phase they are L = (1/2,0,0), Y = (1/2,0,1/2), and Z = (1/2,1/2,0). Middle panels: Weighted partial vibrational DOS and the generalized vibrational DOS for the two phases. Experimental generalized vibrational DOS, measured at two temperatures.

panel) shows the measured spectra. The temperature of the structural transition is reported to be 140 K (Ref. 15) or 142 K, ¹⁷ with peak intensities starting to change significantly at temperatures as high as 150 K. The 140 K spectrum is therefore measured very close to the transition temperature but it shows some notable differences with the 300 K spectrum, such as the hardening of the highest frequency vibrations and the band at 15 meV. These trends tend to be reproduced by the calculations (see below). Bigger differences between the spectra of the two structural phases are not expected since the data measured over a wide temperature range, at lower resolution, on IN4 did not reveal any marked changes. ¹³

III. COMPUTATIONAL DETAILS

Electronic and ionic first principle calculations were performed using the projector-augmented wave (PAW) formalism¹⁸ of the Kohn-Sham density-functional theory (DFT) (Refs. 19 and 20) at the generalized gradient approximation (GGA) level, implemented in the Vienna *ab initio* simulation package (VASP).^{21,22} The GGA was formulated by the Perdew-Burke-Ernzerhof (PBE) (Refs. 23 and 24) density functional. The Gaussian broadening technique was adopted and all results are well converged with respect to *k* mesh and energy cutoff for the plane-wave expansion.

Experimentally refined crystallographic data in both the low-temperature (orthorhombic) and high-temperature (tetragonal) phases were taken from reference. These structures were used to calculate the GDOS and dispersion relations for both the tetragonal and orthorhombic phases. Hereafter, the corresponding simulations are labeled S1 and S2, respectively. The orthorhombic phase was then taken for

TABLE I. Labels for the different $ab\ initio$ simulations presently reported.

Label	Structural Data	Phase	
S1	Experimental	Tetragonal	
S2	Experimental	Orthorhombic	
S31	Fully relaxed	Orthorhombic	
	Partially optimized		
S32	[only z(As) is optimized]	Orthorhombic	
S41	Fully relaxed+magnetic ordering	Orthorhombic	
	Partially optimized		
	[only $z(As)$ is optimized]		
S42	+magnetic ordering	Orthorhombic	

a case study for which full (S31) and partial (S32) structural relaxations were performed. By partial relaxation we mean only the As z position is optimized. The effect of the observed magnetic ordering 17 in the orthorhombic phase was taken into account in further full (S41) and partial (S42) optimizations. Table I provides a summary of these simulations. In the magnetic structure the Fe moments are parallel to the longer a axis. Spins are aligned ferromagnetically along the shorter b axis and antiferromagnetically along the a and c axes. 17

In the lattice dynamics calculations, in order to determine all interatomic force constants, the supercell approach has been adopted. Therefore, the single cells were used to construct $(3^*a, 3^*b, c)$ and $(2^*a, 2^*b, c)$ supercells for the tetragonal and orthorhombic cases, respectively, a and b being the shorter cell axes. The former contains 18 formula units (90 atoms) and the latter contains 16 formula units (80 atoms). Total energies and interatomic forces were calculated for the 12 and 18 structures (in the tetragonal and orthorhombic supercells, respectively) resulting from individual displacements of the three symmetry-inequivalent atoms, along the three inequivalent Cartesian directions $(\pm x, \pm y, \text{ and } \pm z)$. The 15 phonons branches corresponding to the five atoms in the primitive cell, were extracted in subsequent calculations using the PHONON software. $\frac{26}{a}$

IV. RESULTS

A. Generalized phonon density of states and dispersion relations

Figure 1 shows the calculated dispersion curves and the GDOS for both the tetragonal (high-temperature: S1) and orthorhombic (low-temperature: S2) phases using the experimental structures. In order to compare with experimental data, the calculated GDOS was determined as the sum of the partial DOS weighted by the atomic scattering cross sections and masses: $GDOS = \sum_{i} \frac{\sigma_i}{M_i} pDOS_i \left[\frac{\sigma_i}{M_i} = 0.21 (Fe), 0.07 (As), 0.02 (Ba); i=\{Fe,As,Ba\}]$. Both the full (GDOS) and partial generalized phonon density of states (pDOS) are included in Fig. 1. The comparison of the calculated and measured spectra is generally good. The acoustic region extends in both phases to about 9 meV. The lowest-lying optic modes give

rise to intensity starting at about 12 meV. As can be seen from the phonon relations, the corresponding dispersion sheets are not flat. While at the Γ point these low-frequency optic modes correspond to rigid displacements of the FeAs layers with respect to the Ba spacers, distortions within the FeAs layers are present in the zone-boundary modes. Γ -point modes above 18 meV distort the FeAs layers and the Ramanactive ones are illustrated in Fig. 2(c). The calculated vibrational spectrum is clearly separated into three bands via two pseudogaps, which arise from the strongly dispersive sheets in the regions around 20 and 28 meV.

The calculated phonon dispersion is sensitive to structural details, particularly for the optic bands in the intermediate frequency range from 10–20 meV. Flat bands lead to a clear two-peak structure in the GDOS at 13 and 18 meV in the orthorhombic phase. These peaks are found at 13 and 16 meV in the tetragonal phase in good agreement with peaks in the corresponding experimental data. In particular, the experimentally observed peak at 16 meV in the orthorhombic phase softens to 15 meV in the tetragonal phase. Otherwise at the highest frequencies, bands at 24 and 29 meV in the tetragonal phase harden to 25 and 31 meV in the orthorhombic phase, following the trends observed experimentally. It is the higher resolution INS data that allows the validity of the phonon calculations, with respect to the structural phase transition, to be established.

However there is a peak observed at 20–21 meV which is absent in the calculated spectra for both phases. This constitutes the only significant discrepancy between experiment and calculation. As can be seen in the partial DOS, the dispersion branches in the region under question represent motions of Fe and As.

B. Effect of structural relaxation

A generally good description of the phonon spectra has been obtained by using the experimental structures for the ab initio calculations. However, experimental structures are not exactly the equilibrium structures that are typically used for phonon calculations. Phonon calculations have therefore been repeated in the orthorhombic phase for which full (S31atomic z coordinate of As atom plus lattice constants) and partial (S32-atomic coordinate only) structural relaxation were performed. In Fig. 2(a) the corresponding spectra are compared with the calculation performed on the experimental cell (S2) and the experimental spectrum. The S31 and S32 spectra are almost identical, meaning that the lattice constants alone have only a rather minor influence on the phonon spectra. The lattice constant which changes most on full relaxation is the c parameter, which determines the spacing between Ba and Fe-As layers (see Table II). The highest frequency modes, above 24 meV, react most significantly to these structural changes. For example, the band at 25 meV splits in both S31 and S32, whereas a much weaker splitting is observed in the experimental data. The agreement with the experimental data is worse than for the calculation performed on the experimental structure due to the hardening of the highest frequency vibrations. The only internal structural parameter is the z coordinate of the As atom, which deviates by

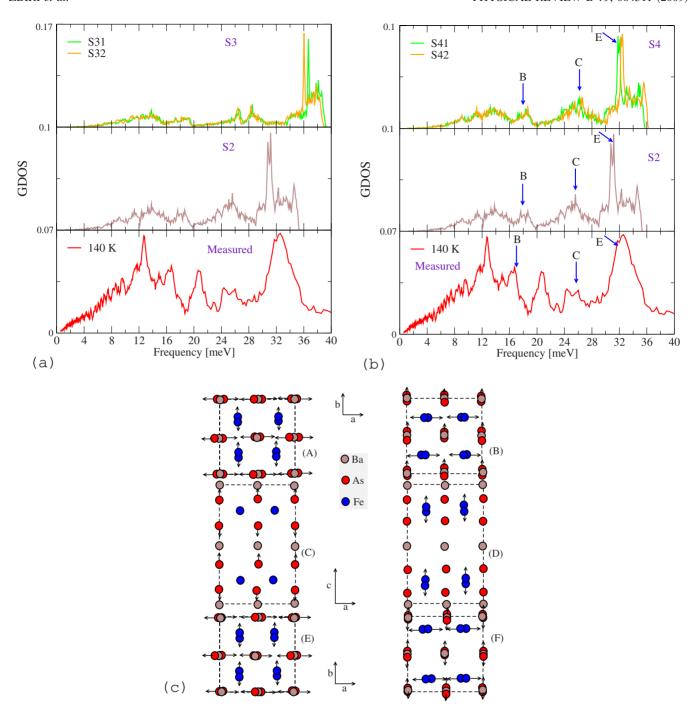


FIG. 2. (Color online) Measured and simulated GDOS in the orthorhombic phase (a) for relaxed nonmagnetic structures (S31 and S32) and (b) for relaxed structures including the known magnetic ordering (S41 and S42). In both cases, the calculation based on the experimental orthorhombic structure S2 is shown for comparison. Labels B, C, and E in (b) indicate specific phonon modes which will be considered to discuss their effect on the electronic structure of BaFe₂As₂ (cf. Fig. 3). The full displacement patterns of the six Raman-active modes (A: 18 meV, B: 18 meV, C: 26 meV, D: 29 meV, E: 33 meV, and F: 34 meV), in the orthorhombic phase, are illustrated in (c).

0.1 Å from the experimental value upon relaxation. This is the cause of the disagreement at high frequency.

C. Effect of magnetic ordering

Further full (S41) and partial (S42) structural relaxation in the orthorhombic phase were performed taking into account the observed magnetic ordering¹⁷ [see Fig. 2(b)]. This leads

to a description of the GDOS which is significantly better than that shown in Fig. 2(a) for the optimized nonmagnetic structures and is comparable to that obtained with the experimental structure. This is appropriate from a theoretical point of view as the phonon calculations should start from a fully relaxed structure. The optimized c axis is within 0.5% of the experimental value and the As z coordinate is close to the experimental value (see Table II). In addition, the calculated

TABLE II. Experimental and optimized lattice parameters and fractional As z coordinates for both structural phases. NM, F, and $M_{\rm obs}$ label nonmagnetic, ferromagnetic, and observed magnetic ordering. Values between brackets indicate optimized As z position obtained from partial structural relaxations.

	Orthorhombic phase			Tetragonal phase		
	Expt.a	M_{obs}	NM/F	Expt. ^a	M_{obs}	NM/F
a	5.6146	5.6611	5.6026	3.9625	3.9746	3.9553
b	5.5742	5.5940	5.5998	3.9625	3.9746	3.9553
c	12.9453	12.8793	12.6302	13.0168	12.7154	12.6849
z(As)	0.3538	0.3512 (0.3510)	0.3456 (0.3446)	0.3545	0.3469 (0.3467)	0.3455 (0.3442)

^aReference 15.

orthorhombic distortion is strongest for the observed magnetic structure and closely matches the experimental value. Magnetic ordering of spin states in Fe cations is therefore important for calculating FeAs bonding and giving the best calculated vibrational density of states. The data in Table II also reveals that, from a structural point of view, the ferromagnetic case is similar to the nonmagnetic case. Also, including the correct magnetic ordering $M_{\rm obs}$ (Ref. 17) results in a volume expansion, which is of interest in light of the apparent pressure-induced superconductivity in the parent compounds of the FeAs-based superconductors. 27,28

D. Effect of phonons on the electron density of states

To gain insight into the effect of phonons on electronic structure, we have calculated the electronic density of states (eDOS) for frozen phonon patterns using the recently observed magnetic structure in the orthorhombic phase. ¹⁷ The

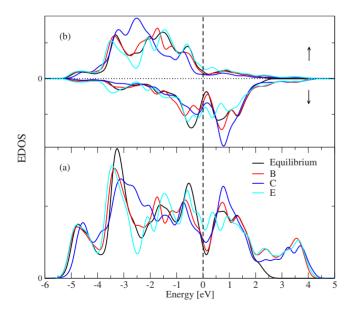


FIG. 3. (Color online) Bottom panel (a): total eDOS of the unperturbed (equilibrium), and the structurally perturbed orthorhombic phase of $BaFe_2As_2$. Total eDOS of spin-down channel is identical to the up-channel. Top panel (b): the corresponding spin-resolved eDOS of the d electrons per Fe cation. Labels B, C, and E indicate phonon-perturbed equilibrium structures (cf. Figs. 2(b) and 2(c); B: 18 meV, C: 26 meV, and E: 32 meV).

mean amplitude was chosen as ~ 0.05 Å, which corresponds approximately to a single occupation of the mode. The results are shown in Fig. 3(a) for the three Gamma point modes that most strongly perturb the equilibrium electronic structure. These modes are Raman active and are labeled B, C, and E in Fig. 2(c) and have frequencies 18, 26, and 32 meV. The main features of the eDOS for the unperturbed structure are in good agreement with other electronic structure calculations of the tetragonal phase.^{29–31} The bands close to the Fermi level and in particular the eDOS at the Fermi level depend strongly on the phonon distortion via the d bands of Fe. As the Fermi level of the equilibrium structure is found at the minimum of a rather steep-walled valley of the eDOS, phonon modes that displace the Fermi level only slightly create a large fluctuation of the electron density at the Fermi level.

In view of the importance of magnetic ordering on the equilibrium structure of BaFe₂As₂, we have also investigated the spin-resolved eDOS of the d electrons of the Fe cation [see Fig. 3(b)]. Complex changes in the eDOS are captured in the magnetic moment of the Fe cation. The calculated magnetic moment for the equilibrium structure is $1.94\mu_R$. This value is much larger than the observed magnetic moment (μ_B) as reported by other authors³² and explained in terms of frustrating magnetic interactions reducing the observed moment (see Ref. 17 and references therein). On the computational side, standard settings, notably for band occupation at the Fermi level, tend to result in more localized and therefore bigger moments. Settings could be tuned to give a reduced magnetic moment but here we simply want to use the equilibrium magnetic moment as a reference value to quantify the effect of structural distortions. Mode B increases the moment slightly to $2.07\mu_B$, mode C increases it to $2.55\mu_B$, and mode E decreases it to $1.58\mu_B$. From Fig. 3(b) it is seen that modes B and C cause a decrease in the spindown population and an increase in the spin-up population and therefore an increase in the magnetic moment. On the other hand, mode E causes a rocking of the spin population from the majority up-channel to the minority down-channel, decreasing the magnetic moment. Mode E tends to reduce the magnetic ordering. For all three modes, the Fermi level changes by less 0.5%.

V. CONCLUSION AND PERSPECTIVES

Small variations in the GDOS between the tetragonal and orthorhombic phases are observed in the high-resolution ex-

perimental data and these are generally reproduced by the calculations. The variations in the distances between Fe atoms and between Fe and As atoms are less than 0.01 Å, 15 revealing the sensitivity of phonons to subtle structural changes. Phonon calculations performed on the experimental structure and on the fully relaxed structure, including the observed magnetic ordering, best match the experimental spectra. Nonmagnetic and ferromagnetic relaxed structures show bigger discrepancies with the measured spectra. The key structural parameter is the z coordinate of the As atom, which governs the distortion of the Fe-As layer. Phonon modes that modulate the Fe-As layer cause the most significant changes in the eDOS at the Fermi level, which indicate the extent of electron-phonon coupling. In addition the magnetic moment and spin-resolved eDOS of the Fe cation vary significantly with the phonon displacements, demonstrating spin-phonon coupling, which is relevant to the search for a composite mechanism for high-temperature superconductivity that goes beyond simple electron-phonon coupling.

The main shortcoming of the calculation is the inability to reproduce the observed experimental peak at 20 meV. The calculation predicts in this region highly dispersed branches leading to a pseudogap in the density of states. Experimentally, the *Q* dependence of this peak is similar to that of other phonon bands. There is, therefore, no experimental indication that we are dealing with a magnetic excitation, especially at the temperatures of these measurements. In recent work by Reznik and co-workers³³ combining *ab initio* linear-response phonon calculations and inelastic x-ray scattering, the peak close at 20 meV is observed experimentally but not reproduced computationally.

We have also tried to generate spectral intensity at 20 meV within the framework of our phonon calculations. This is achieved by trying to fit interatomic force constants to change the frequencies of selected phonons at particular wave vectors. The only phonon that could be significantly modified in this way was a mode at 24 meV at the zone boundary (0,0,0.5), but the change in spectral intensity is small. This effect is however consistent with the sensitivity that we have demonstrated of the phonons to the z coordinate of the As atom. The importance of the c-axis phonons in FeAs-based superconductors has also been highlighted recently.³³

However, if we believe that modern DFT codes correctly reproduce the main spectral features in, structurally, rather simple systems such as BaFe₂As₂, then we have no clear explanation for this peak within the phonon model implemented here. Therefore, any statement on the origin of the 20 meV peak without further experiments has to remain speculative. Strong nonconventional electron-phonon coupling has been proposed in the literature⁹ for LaOFeAs. The possibility of a singularity close to 20 meV is evoked on the basis of gap openings at the Fermi surface. The same physical arguments should hold for BaFe₂As₂ as only the FeAs layers are concerned.

Finally we plan to extend our measurements and calculations to other systems in this family of compounds, to determine whether the anomaly at 20 meV is a general feature or whether it is specific to the Ba system. In this context, we are motivated by initial work on the strontium system for which we obtain excellent agreement between measured and calculated Raman frequencies.

¹Y. Kamihara, H. Hiramatsu, M. Hirano, R. Kawamura, H. Yanagi, T. Kamiya, and H. Hosono, J. Am. Chem. Soc. **128**, 10012 (2006).

²C. Y. Liang, R. C. Che, H. X. Yang, H. F. Tian, R. J. Xiao, J. B. Lu, R. Li, and J. Q. Li, Supercond. Sci. Technol. **20**, 687 (2007).

³T. Watanabe, H. Yanagi, T. Kamiya, Y. Kamihara, H. Hiramatsu, M. Hirano, and H. Hosono, Inorg. Chem. **46**, 7719 (2007).

⁴O. Kamihara, T. Watanabe, M. Hirano, and H. Hosono, J. Am. Chem. Soc. **130**, 3296 (2008).

⁵F. Hunte, J. Jaroszynski, A. Gurevich, D. C. Larbalestier, R. Jin, A. S. Sefat, M. A. McGuire, B. C. Sales, D. K. Christen, and D. Mandrus, Nature (London) 453, 903 (2008).

⁶P. L. Alireza, J. Gillett, Y. T. C. Ko, S. E. Sebastian, and G. G. Lonzarich, J. Phys.: Condens. Matter **21** (1), 012208 (2008).

⁷D. J. Singh and M.-H. Du, Phys. Rev. Lett. **100**, 237003 (2008).

⁸L. Boeri, O. V. Dolgov, and A. A. Golubov, Phys. Rev. Lett. 101, 026403 (2008).

⁹H. Eschrig, arXiv:0804.0186 (unpublished).

¹⁰W. Pickett, Nature (London) **418**, 733 (2002).

¹¹R. A. Ewings, T. G. Perring, R. I. Bewley, T. Guidi, M. J. Pitcher, D. R. Parker, S. J. Clarke, and A. T. Boothroyd, Phys. Rev. B 78, 220501(R) (2008).

¹² A. D. Christianson, M. D. Lumsden, O. Delaire, M. B. Stone, D. L. Abernathy, M. A. McGuire, A. S. Sefat, R. Jin, B. C. Sales, D. Mandrus, E. D. Mun, P. C. Canfield, J. Y. Y. Lin, M. Lucas,

M. M. Kresch, J. B. Keith, B. Fultz, E. A. Goremychkin, and R. J. McQueeney, Phys. Rev. Lett. **101**, 157004 (2008).

¹³R. Mittal, Y. Su, S. Rols, T. Chatterji, S. L. Chaplot, H. Schober, M. Rotter, D. Johrendt, and T. Brueckel, Phys. Rev. B 78, 104514 (2008).

¹⁴D. J. Singh and M. H. Du, Phys. Rev. Lett. **100**, 237003 (2008).

¹⁵M. Rotter, M. Tegel, D. Johrendt, I. Schellenberg, W. Hermes, and R. Pöttgen, Phys. Rev. B 78, 020503(R) (2008).

¹⁶H. Schober, A. Tölle, B. Renker, R. Heid, and F. Gompf, Phys. Rev. B **56**, 5937 (1997).

¹⁷Q. Huang, Y. Qiu, W. Bao, M. A. Green, J. W. Lynn, Y. C. Gasparovic, T. Wu, G. Wu, and X. H. Chen, Phys. Rev. Lett. **101**, 257003 (2008).

¹⁸P. E. Blöchl, Phys. Rev. B **50**, 17953 (1994).

¹⁹P. Hohenberg and W. Kohn, Phys. Rev. **136**, B864 (1964).

²⁰W. Kohn and L. J. Sham, Phys. Rev. **140**, A1133 (1965).

²¹G. Kresse and J. Furthmüller, Comput. Mater. Sci. 6, 15 (1996).

²²G. Kresse and D. Joubert, Phys. Rev. B **59**, 1758 (1999).

²³J. P. Perdew, K. Burke, and M. Ernzerhof, Phys. Rev. Lett. 77,

<sup>3865 (1996).
&</sup>lt;sup>24</sup> J. P. Perdew, K. Burke, and M. Ernzerhof, Phys. Rev. Lett. **78**, 1396 (1997).

²⁵ K. Parlinski, Z. Q. Li, and Y. Kawazoe, Phys. Rev. Lett. 78, 4063 (1997).

²⁶ K. Parlinksi, PHONON, 2003.

- ²⁷ A. Kreyssig, M. A. Green, Y. Lee, G. D. Samolyuk, P. Zajdel, J. W. Lynn, S. L. Bud'ko, M. S. Torikachvili, N. Ni, and S. Nandi, arXiv:0807.3032 (unpublished).
- ²⁸ W. Yu, A. A. Aczel, T. J. Williams, S. L. Bud'ko, N. Ni, P. C. Canfield, and G. M. Luke, arXiv:0811.2554, Phys. Rev. B (to be published).
- ²⁹I. R. Shein and A. L. Ivanovskii, arXiv:0806.0750 (unpublished).
- ³⁰F. Ma, Z.-Y. Lu, and T. Xiang, arXiv:0806.3526 (unpublished).
- ³¹D. J. Singh, Phys. Rev. B **78**, 094511 (2008).
- ³²S. Lebègue, Z. P. Yin, and W. E. Pickett, arXiv:0810.0376, New J. Phys. (to be published).
- ³³D. Reznik, K. Lokshin, D. C. Mitchell, D. Parshall, W. Dmowski, D. Lamago, R. Heid, K.-P. Bohnen, A. S. Sefat, M. A. McGuire, B. C. Sales, D. G. Mandrus, A. Asubedi, D. J. Singh, A. Alatas, M. H. Upton, A. H. Said, Yu. Shvyd'ko, and T. Egami, arXiv:0810.4941 (unpublished).