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# Measurement of $T_c$ suppression in tungsten using magnetic impurities

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We have measured the effects of dilute magnetic-atom doping on the superconducting transition temperature of tungsten thin films. Our “ $T_c$  tuning” technique is accurate, precise, and simple. Experiments were performed using dc-magnetron-sputtered tungsten films with undoped values of  $T_c$  in the range of 70–150 mK. The magnetic-atom doping was achieved using ion implantation. Specific  $T_c$  suppressions of between 5% and 65% were targeted and observed in this study. The transition width of each undoped sample was  $\approx 1$  mK and the transition widths remained sharp after implantation with  $^{56}\text{Fe}^+$  ions. Our data are in good agreement with predictions of a linear dependence of  $T_c$  suppression with increasing magnetic-atom concentration, in the small concentration limit. At higher concentrations, antiferromagnetic coupling between the magnetic dopant atoms becomes important and the  $T_c$ -suppression effect is diminished. We use our  $T_c$  data to calculate the Abrikosov–Gor’kov (AG) and Ruderman–Kittel–Kasuya–Yosida (RKKY) spin-flip relaxation parameters  $\tau_{\text{AG}}$  and  $\tau_{\text{RKKY}}$ . We conclude with a brief discussion of applications of the  $T_c$ -tuning technique, and present our plans for future studies in this area. © 1999 American Institute of Physics. [S0021-8979(99)03624-5]

## MOTIVATION AND BACKGROUND

Thin films of superconducting materials have been used as phonon sensors for many years. The exact design of phonon sensors varies according to the intended applications.<sup>1</sup> For example, some of the cryogenic detectors developed by the Cold Dark Matter Search (CDMS) collaboration to search for weakly interacting massive particle (WIMP) dark matter rely on collecting thermal phonons using overlapping thin films of superconducting aluminum and tungsten.<sup>2</sup> These phonon sensors are called “W/Al quasi-particle-trap-assisted electrothermal-feedback transition-edge sensors (QETs).”<sup>3</sup> The energy sensitivity of a QET phonon sensor depends critically on the  $T_c$  of the superconducting W film, and also on the width of the superconducting transition.

Our tungsten and aluminum films are deposited using a Balzers 450 dc-magnetron sputtering system and a 99.999% pure, vacuum-pressed tungsten target purchased from Johnson Matthey. This metalization system provides high-quality films with sufficiently narrow superconducting transitions (width  $\leq 1$  mK), however, the “as-deposited” W films often have values of  $T_c$  above our target value of 65 mK. The exact value of the tungsten film  $T_c$  obtained in a given metalization run depends significantly on the details of deposition parameters and specific vacuum chamber conditions. The variability in  $T_c$  from one batch of devices to the

next is likely due to differing stress characteristics in the deposited films, as well as to minute quantities of various residual impurities present in our vacuum chamber during deposition. Because film deposition parameters cannot be absolutely controlled, and even slight deviations from our prescribed deposition recipe can give rise to nonoptimal QET  $T_c$  values, we recently focused on developing a technique to adjust the  $T_c$  of our sputtered tungsten films, by postprocessing. We were specifically interested in developing a technique which would suppress  $T_c$  by a prescribed amount, yet no broaden the superconducting transition nor degrade device performance.

## DEMONSTRATION OF $T_c$ -TUNING TECHNIQUE

Our idea was to use the precision and accuracy of conventional ion-implantation techniques to tune the  $T_c$  of individual W films. To study this possibility, we used a collection of 350-Å-thick, sputtered W films with as-deposited  $T_c$ 's of  $\approx 70$ –150 mK. The films were deposited in a series of separate depositions; the substrates used were identical, and consisted of a 210-Å-thick layer of amorphous silicon on top of a (100) silicon wafer. Immediately adjacent samples, each of area  $\approx 0.25$  cm<sup>2</sup>, were taken from our as-deposited W films and separately ion-implanted with  $^{56}\text{Fe}^+$  ions at 50 keV. At this energy, a negligible fraction of the incident  $^{56}\text{Fe}$  ions fully penetrate a 350-Å-thick tungsten film. Doses ranging from  $1.00 \times 10^{12}$  to  $1.00 \times 10^{13}$  ions/cm<sup>2</sup> ( $\pm 0.01\%$ ) were used in the study presented here. A plot of the depth profile

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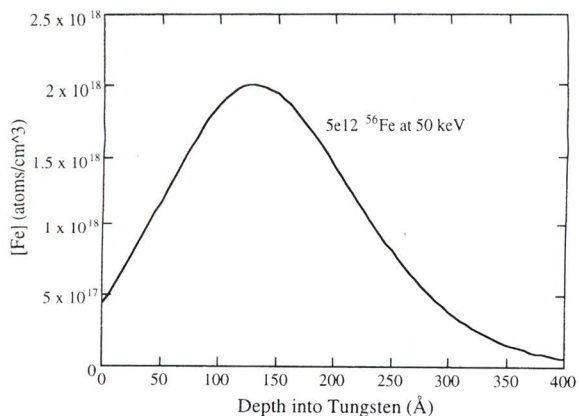


FIG. 1. Calculated Fe concentration vs depth into the W film for one Fe-ion implantation dose used in our study.

for one of our implants ( $5.00 \times 10^{12} \text{ cm}^{-2}$  at 50 keV), showing the concentration of implanted  $^{56}\text{Fe}^+$  ions versus depth into the tungsten film, is presented in Fig. 1. The plot in Fig. 1 was calculated using PROFILE CODE™, a software simulation package purchased from Implant Sciences, Inc. The sputtering target used to deposit all of our tungsten films contains a residual impurity concentration of 0.9 ppm Fe. In our study, this Fe concentration would correspond to an equivalent 50 keV Fe-atom dose of  $\approx 1.3 \times 10^{11} \text{ cm}^{-2}$ .

In Fig. 2, we plot the measured film  $T_c$  versus ion dose for 50 keV  $^{56}\text{Fe}^+$  implanted into a 350-Å-thick tungsten film with unimplanted superconducting  $T_c$  of  $(135 \pm 2)$  mK. In Fig. 3, we plot the  $T_c$  data in an alternate form; we plot  $T_c$  (control sample)  $- T_c$  (implanted sample) versus peak  $^{56}\text{Fe}^+$  concentration in atomic ppm, calculated for each of the measured implant doses, and for the one control sample. Our control sample was not implanted; it contained only the 0.9 ppm Fe contributed from the high-purity W sputtering target. A quadratic fit to the combined data provides an empirical relationship between [Fe], the peak concentration of Fe ions in our W samples, and  $\Delta T_c$ , the resulting shift in the transition temperature:

$$\Delta T_c = -3.16 + 1.66[\text{Fe}] - 0.00457[\text{Fe}]^2,$$

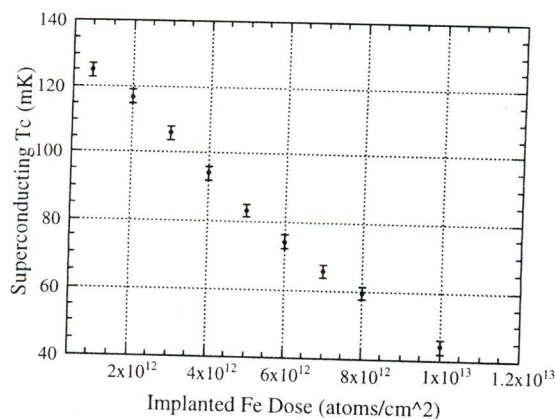


FIG. 2. Measured superconducting transition temperature for 350-Å-thick tungsten films implanted with  $^{56}\text{Fe}^+$  at 50 keV kinetic energy.

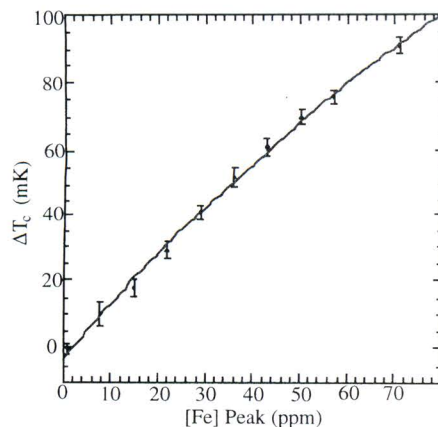


FIG. 3. Measured  $T_c$  suppression for tungsten samples as a function of peak Fe concentration in our tungsten films. The control sample with 0.9 ppm [Fe] had a  $T_c$  of  $135 \pm 2$  mK. The plotted points include the [Fe]=0.9 ppm from the sputtering target. The fitted curve is a quadratic.

where  $\Delta T_c = T_c$  (control sample)  $- T_c$  (implanted sample) is measured in mK and [Fe] is in ppm, atomic fraction.

We have demonstrated the  $T_c$ -tuning process on several additional W films with unimplanted  $T_c$ 's of 70–150 mK. In the simplest approach, we used the data shown in Fig. 2 to determine the dose of  $^{56}\text{Fe}^+$  needed to shift the  $T_c$  of a given film the desired amount. We assumed a straight-line fit to the low-dose data plotted in Fig. 2, then shifted the  $y$  intercept of that line up or down to match the  $T_c$  of the new unimplanted control sample. The slope ( $= 1.5 \text{ mK/ppm}$ ) was not adjusted because the  $^{56}\text{Fe}^+$ -implanted energy for all samples was held at 50 keV. The ion dose necessary to shift  $T_c$  by a prescribed amount was then read from the  $x$  axis of the intercept-adjusted graph. Using this simple approach, we have successfully tuned several films to within a few mK of a specified target  $T_c$ . Most importantly, for all of our samples, the narrow superconducting transition widths of  $\approx 1$  mK were maintained even after the  $T_c$ -tuning procedure was completed.

In Fig. 4, we show the 135 mK film data, along with a fit to the theoretical prediction of Abrikosov and Gor'kov (AG) for the general dependence of  $T_c$  on magnetic-impurity concentration. Based on the AG model,<sup>4</sup> we assume an antisymmetric exchange coupling between the magnetic impurity ions (Fe) and the lattice host atoms in our W films. We assume a one-electron Hamiltonian is sufficient to describe the W system. We also assume the Fe concentration is sufficiently low that a dilute-limit approximation is valid, and that the presence of impurity atoms results only in the scattering of W conduction electrons. That is, we assume no coupling exists between one impurity atom and another. Finally, we assume the pair potential for Cooper pairs in the W film is position independent. For reference, the coherence length in our superconducting W films is  $\xi_0 \approx 0.3 \mu\text{m}$ , which is much greater than the film thickness  $\approx 350 \text{ \AA}$ . In the AG model, the interaction between a W-atom electron and an iron impurity can be represented by a perturbing Hamiltonian term of the form:

$$H_{\text{perturbation}} = K(\mathbf{r}_e - \mathbf{r}_i) \mathbf{S}_e \cdot \mathbf{S}_i,$$

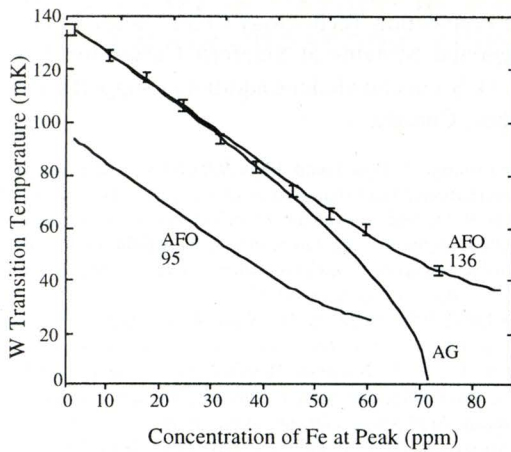


FIG. 4. Superconducting transition temperature of tungsten as a function of iron dopant concentration (at peak of ion-implant distribution). Curve “AG” corresponds to the Abrikosov and Gor’kov model. Curves “AFO” correspond to the AG model modified to include antiferromagnetic ordering between Fe atoms. Fit “AFO 136” corresponds to  $T_{co} = 136$  mK. For comparison, we include fit “AFO 95,” the predicted curve for films with  $T_{co} = 95$  mK. Our Fe in W data are best fit for the AFO model with  $\alpha = -(0.06 \pm 0.01)$ .

where  $\mathbf{r}_e$  and  $\mathbf{r}_i$  are the position vectors of the W-atom electron and the magnetic impurity atom, respectively. Similarly,  $\mathbf{S}_e$  and  $\mathbf{S}_i$  are the spin vectors of the W-atom electron and the magnetic-ion impurity.  $K(\mathbf{r}_e - \mathbf{r}_i)$  is the exchange coupling constant, which can be positive or negative because it operates differently (asymmetrically) on the two electron ( $\uparrow\downarrow$ ) in a Cooper pair. The AG model for superconductors with magnetic impurities yields the following relationship between the reduced transition temperature  $T_c/T_{co}$  and the reduced magnetic-impurity concentration  $x/x_c$  of a sample:<sup>5</sup>

$$\ln(T_c/T_{co}) = \psi\{1/2\} - \psi\{(1/2) + (1/4)(e^{-\gamma})(x/x_c)\} \times (T_{co}/T_c),$$

where  $\psi$  is the digamma function, i.e.,  $\psi$  is the derivative of the logarithm of the  $\Gamma$  function:

$$\psi = -\gamma - (1/z) - \sum_{k=1}^{\infty} \{1/(z+k) - (1/k)\},$$

and  $\gamma$  is the Euler–Mascheroni constant ( $\gamma = 0.57721566\dots$ ). At low-impurity concentrations the AG model predicts a linear decrease in  $T_c$  as a function of  $x/x_c$ . At substantially higher concentrations, i.e., when  $T_c \ll T_{co}$ , the model predicts an extremely rapid decrease of  $T_c$  such that the superconductivity vanishes altogether at a certain critical concentration  $x_c$ . The critical concentration can be evaluated from the slope of the  $T_c$  vs  $x$  curve in the low concentration limit:

$$\left. \frac{dT_c}{dx} \right|_{x=0} \cong \frac{-\pi^2 T_{co}}{8e^{\gamma} x_c}.$$

For our W data, and referenced to the peak of each Fe-implant distribution, this calculation yields a critical concentration  $x_c = 69$  ppm.

The AG curve shown in Fig. 4 fits our data well at low-impurity concentrations, where  $T_c$  decreases linearly with increasing impurity concentration. For higher concentrations, however, i.e., for doping levels exceeding  $\approx 30$ – $40$  ppm Fe, our data deviate substantially from the AG curve. In particular, our data show that the  $T_c$  suppression due to antisymmetric electron–ion exchange coupling in the film becomes less significant at the higher Fe concentrations studied. At these higher concentrations it is probable that the weak coupling approximation for the W–Fe system is not valid and Fe–Fe interactions can be no longer ignored. Indeed, we believe that the observed deviation from the AG model at high Fe concentrations (i.e.,  $T_c \ll T_{co}$ ) is due to antiferromagnetic interactions between the Fe impurities in our samples. The antiferromagnetic ordering increases the superconducting order parameter, which allows the system to superconduct when it would otherwise be normal ( $x \geq x_c$ ). In contrast, ferromagnetic (rather than antiferromagnetic) coupling between impurities would inhibit the persistence of the superconducting state. Including the effect of nearest-neighbor magnetic interactions between the dopant atoms, we get

$$\ln(T_c/T_{co}) = \psi\{1/2\} - \psi\{(1/2) + (1/4)(e^{-\gamma})(x/x_c)\} \times (T_{co}/T_c)[1 + \alpha(x/x_c)(T_{co}/T_c)],$$

where  $\alpha$  is a magnetic coupling parameter<sup>5</sup>

$$\alpha = \tau_{AG}(x_c)/\tau_{RKKY}(x_c, T_{co}),$$

$\alpha > 0$  for ferromagnetic ordering (parallel spins) and  $\alpha < 0$  for antiferromagnetic ordering (antiparallel spins). The best fit to our data shown in Fig. 4 has  $\alpha = -0.06 \pm 0.01$ . The parameters  $\alpha_{AG}$  and  $\tau_{RKKY}$  (Ruderman–Kittel–Kasuya–Yosida) give the relative strengths of the impurity–host atom (Fe–W) interactions and the impurity–impurity (Fe–Fe) interactions, respectively. Using our data, we can readily calculate  $\tau_{AG}$  and  $\tau_{RKKY}$  for the Fe–W system. In the dilute dopant limit:

$$\left. \frac{dT_c}{dx} \right|_{x=0} = \frac{\pi}{4\tau_{AG}x}.$$

Thus

$$\tau_{AG}(x_c) = -\frac{\pi}{4(-1.5 \text{ mK/ppm})(69 \text{ ppm})} = 7.6 \text{ K}^{-1},$$

and

$$\tau_{RKKY}(x_c, T_{co}) = (7.6 \text{ K}^{-1})/(-0.06) = -130 \text{ K}^{-1}.$$

Magnetic ordering in superconductors is discussed in depth by Roshen and Ruvalds,<sup>5</sup> Maple,<sup>6</sup> Fulde and Keller,<sup>7</sup> and Dupont, Ziemniak, and Usadel.<sup>8</sup>

## CONCLUSION AND FUTURE DIRECTIONS

The  $T_c$ -tuning technique we have developed is reliable and reproducible, and offers flexibility for fabricating cryogenic devices with specific superconducting transition temperatures. The fact that the measured transition widths of our  $T_c$ -suppressed films are not affected by the doping process is encouraging, since we expect that a broadening of the tran-

sition would accompany a decrease in the critical current of our superconducting films. For our primary applications to cryogenic particle and photon detectors, such a decrease in critical current would adversely affect the overall performance of our devices. Soon, we will measure directly the effects of dilute Fe-atom doping on the critical current and heat capacity of the W films.

We are in the process of extending our  $T_c$ -suppression work to W-Co and W-Ni systems, to further study the effects on  $T_c$  of magnetic ordering in superconducting films. Those results will be presented in a future publication.

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