1 Synergy achieved in silver-TiO₂ nanocomposites for the inhibition of

2 biopatina on limestone

J. Becerra^{1,*}, A.P. Zaderenko¹, M.J. Sayagués², R. Ortiz¹, P. Ortiz¹

¹ Departamento de Sistemas Físicos, Químicos y Naturales, Universidad Pablo de Olavide, ES-41013
Seville, Spain.

6 ² Instituto de Ciencia de Materiales de Sevilla (CSIC-US), ES-41092-Seville, Spain.

7 *Corresponding author: J. Becerra: jbeclun@upo.es

8 Abstract

9 Biodeterioration of stone monuments is estimated to be as high as 20-30% of the total degradation 10 suffered by Cultural Heritage constructions. With regard to this problem, bactericidal treatments are 11 mainly based on cleaning. These processes, while effective in the short term, require frequent 12 reapplications increasing potential damages to the monument. Silver nanoparticles offer many 13 advantages over traditionally employed products, such as their prolonged biocide efficacy and their 14 low toxicity to humans and environment. The aim of this study was to evaluate the applicability and 15 effectiveness of seven nanocomposite treatments based on titanium dioxide and/or silver nanoparticles to prevent biodeterioration of limestone monuments. These nanocomposites were characterized by 16 17 UV-Visible spectrophotometry, Dynamic Light Scattering and Electron Microscopy. To assess their 18 bactericidal activity, accelerated weathering tests were performed on limestones from the quarry of 19 Utrera, a source widely employed in such iconic monuments as the Cathedral of Seville (Spain). 20 Furthermore, the samples of biopatina employed in our assays stemmed from the façades of historical buildings from Seville. Our results show that silver and titanium dioxide nanocomposites stabilized by 21 22 citrate achieve a high biocide effect while maintaining color alterations at a low level.

23 Keywords

24 Biopatina, biocide, limestone, conservation and restoration, silver-titanium dioxide nanocomposite,25 stone biodeterioration

26 1. Introduction

27 Biodeterioration of stone monuments and archaeological remains is an undesirable change caused by 28 biological agents, whose impact is estimated at around 20-30% of the total degradation suffered by 29 Cultural Heritage constructions [1]. Biological agents involved in biodeterioration include a wide 30 variety of organisms and microorganisms. Bacteria and algae are the most common life forms 31 dwelling on stone buildings, favoring the appearance of biopatina over the stone surface. These 32 biopatinas act through physical and chemical processes. Physical processes are related to the adhesion 33 systems of the microorganisms on the stone substrate, generating chromatic changes, while chemical 34 processes imply reactions that transform or break down the stone substrate causing disintegration and 35 crumbling. In the case of microorganisms, chemical processes are associated with their metabolism as 36 they can disaggregate carbonate stones due to the generation of organic acids. Bacteria are involved in 37 significant chemical processes that vary according to their feeding. Autotrophic bacteria are 38 characterized by inducing solubilisation processes that include acidification of the medium and eating 39 away the stone structure, while the adverse effects of heterotrophic bacteria are mainly due to the 40 production of chelating substances and organic and inorganic acids and alkalis. Nevertheless, in both 41 types of bacteria the main alterations observed are brown and black patinas and exfoliation [2].

42 Algae and cyanobacteria are often the first microorganisms to colonize the stone surface [3]. 43 For that, specific environmental conditions are necessary such as intense light, high humidity or water, 44 high temperature and alkaline pH. Patinas and crusts of different colors are the principal alterations loosely attributed to weathering [4] which depend on factors like amount of light, available nutrients, 45 46 temperature, colony aging, etc...[1]. Furthermore, these patinas favor the build-up of dust, spores, 47 other organic compounds, etc, that constitute a special dwelling ground for other biological colonizers 48 (lichens, fungi and vascular plants) [5], and they maintain high moisture levels due to mucilaginous 49 pods.

50 The absolute cost caused by biodeterioration is difficult to quantify [2], and the cost of 51 cleaning, restoration and maintenance is increasing owing to the increase of detrimental external 52 factors, such as pollution. Some limestones widely employed in the construction of historical buildings 53 [6], are colonized by microorganisms owing to their porous nature. This microbiological colonization, which takes place especially in dark and damp areas of buildings built with this material, leads to the development of biopatina and biocrust that constitute a fertile substrate for higher order plants. Besides favoring colonization by plants, microorganisms are a major cause of deterioration by themselves, as biopatina and biocrust not only produce undesirable alterations of the appearance of the material, but they can also produce fractures and loss of material, as mentioned above. Figure 1 shows representative images of biopatina and biocrust growth on limestone surfaces from monuments in southern Spain.



61

Fig. 1. General (A) and expanded (B) views of biocrust in the vaults of the Cathedral of Seville. New onset of*biopatina in the main façade of the City Hall of Seville (C) one year after its restoration.*

Current bactericidal treatments are mainly based on cleaning. These cleaning processes, while 64 effective in the short term, require frequent reapplications which also increase potentially the damage 65 66 to the monument. Different treatments are currently employed to prevent the appearance and growth 67 of these alteration agents and their effects on buildings. Most of them are chemical methods like tributyltin, phenolic compounds, quaternary ammonium compounds, etc. These methods have 68 69 important disadvantages, however, like their high toxicity to humans and environment, their short-70 lasting effects and possible incompatibility with the substrate stone. For example, limestones are 71 highly sensitive to acid action involved in most treatments, or the color changes caused by tributyltin 72 on travertine stones [7]. Bastian et al. [8] highlighted that these treatments can cause irreversible 73 changes in the original materials, and have even favored the appearance of new microorganisms. New 74 treatments based on silver nanoparticles (AgNPs) with biocide effects bear the promise to avoid these disadvantages [9,10] which is endorsed by the good results obtained in other disciplines such as 75

medicine, cosmetics, textile industry or environmental remediation [11]. There are several explanations for the interaction of AgNPs with microorganisms, although the precise mechanism is not yet well understood [12]. It is a multifactorial process in which the cellular wall and the plasma membrane of the bacteria are damaged, and so is the inhibition of the protein synthesis and DNA replication [13–15]. These effects are due to the slow release of Ag^+ ions over time in the case of some nanoparticles that favors their long-term biocide effects at lower concentrations than silver salts.

AgNPs biocide properties have been confirmed in different microorganisms, among them, Gram-negative bacteria (*Escherichia coli, Pseudomonas aeruginosa* and *Acinetobacter baumannii*) and Gram-positive bacteria (*Staphylococcus aureus, Enterococcus faecalis, Candida albicans* and *Listeria monocytogenes*) [12–15]. Notably, biocide activity was related to nanoparticle size [14], shape [16] or reducing agent employed in their synthesis [17]. The main advantages offered over traditionally employed products are their prolonged biocide efficacy against a wide variety of microorganisms and their low toxicity to humans and environment [10,18].

89 On the other hand, titanium dioxide nanoparticles (TiO₂NPs) are one of the most versatile 90 nanoparticles used in construction and building materials due to their stability over time, their 91 compatibility with traditional materials and their high photocatalytic activity which makes them 92 suitable for self-cleaning surfaces [19–21]. This last property, together with hydrophilic properties in 93 presence of UV radiation [20,22], allows to generate a protecting film against deposits from 94 environmental pollution. TiO₂NPs decompose both organic and inorganic substances that cause 95 staining and degradation of stone walls [20,21]. The durability of treatments and their inalterability 96 have been studied too [19,23,24], as well as the mechanism for their biocide properties. As biocide, 97 TiO₂NPs cause oxidation in the microorganism cell walls and hinder or at least interfere with their respiration [25,26], especially under UV radiation. 98

99 The use of combined silver and titanium dioxide into a single nanocomposite (Ag/TiO₂) seeks 100 to complement and improve the specific properties on construction materials that these nanoparticles 101 exhibit separately. The presence of silver on the surface of TiO₂NPs increases its chemical reactivity 102 under visible light, so its catalytic activity is not dependent on the availability of UV radiation [27,28]. 103 The biocide effects of Ag/TiO₂ nanocomposites have been checked for other applications against

104 several species of bacteria (Pseudomonas stutzeri, Stenotrophomonas maltophilia, Micrococcus luteus, 105 Bacillus subtilis, Staphylococcus aureus, Escherichia coli, Klebsiella pneumonia and Pseudomonas 106 aeruginosa [27,29,30]) and fungi (Candida albicans [29,30]). The structure of this kind of 107 nanocomposite is a silver shell around a TiO₂ core; although the inverse structure, TiO₂ shell around a 108 silver core, has been investigated too, and both configurations were shown to be efficient as biocides 109 [31-33]. La Russa et al. [27] have made a first approach confirming the biocide effectiveness of 110 Ag/TiO₂ nanocomposites in a diffusion test on agar plates that extrapolates to stone protection. 111 Nevertheless, inhibition of biopatina formation on stone has not been demonstrated until now. In a study by Bellissima et al. [9], although they applied AgNPs functionalized with tetraethylorthosilicate 112 113 (TEOS) on stone, they do not measure the biopatina inhibition faculty of these nanoparticles, only 114 their effectiveness as an antibacterial treatment.

115 In this paper, the applicability of Ag/TiO_2 nanocomposites to inhibit biopatinas on stone 116 façades and walls has been studied, checking their biocide properties to determine their effectiveness 117 and applicability under real-world conditions. Remarkably, the samples of biopatina employed in our 118 assays stemmed from the façades of historical buildings.

119 2. Materials and methods

120 2.1. Stone sampling

Due to the abundance of quarries of carbonate stone in western Andalusia, this type of stone has been widely used as building material in this region since ancient times. Our study focuses specifically on a representative limestone from one of the main suppliers, the quarry of Utrera (Seville, Spain). Remarkably, this limestone has been employed in the construction of some of the most iconic historic buildings in southern Spain, such as the Cathedral or the Town Hall in Seville [34,35], so it is considered one of the most important quarries in southern Spain.

Limestone from the quarry of Utrera is a carbonated bioclastic limestone with a high content
of quartz and fossils (2-5%) [34] and a high open porosity (10 %) [36] with pore sizes of 0.1-1 μm
[34]. Additionally, the high content and widespread distribution of fossil material generates an intra-

130 porosity that is not connected. The samples used in this study have a size of $1.5 \times 1.5 \times 0.5$ cm.

131 2.2. Synthesis of silver nanoparticles and silver/titanium dioxide nanocomposites

132 Silver nitrate (AgNO₃) and trisodium citrate were purchased from Panreac, titanium dioxide *133* (P25) from DEGUSSA and sodium borohydride (NaBH₄) from Sigma-Aldrich. All other chemicals *134* were reagent grade. Water was purified using a Milli-Q reagent grade water system from Millipore.

135 Silver nanoparticles stabilized by citrate (Ag(acit)) were synthesized according to the method 136 described by Flores et al [37]. Briefly, an aqueous solution of AgNO₃ (1 mL, 0.005 mmol) was added 137 to an aqueous solution of trisodium citrate (16 mL, 0.169 mmol) under magnetic stirring in an ice bath. 138 Thereafter, 100 µL of an aqueous solution of NaBH₄ (0.01 mmol) were added dropwise and the 139 reaction mixture was stirred for 1 h and 45 min. The resulting pale yellow aqueous-colloidal 140 suspension of Ag@cit nanoparticles was kept in darkness until its use in biocide assays. This protocol 141 was adapted to obtain two silver/TiO₂ nanocomposites, Ag@cit/TiO₂ and 2xAg@cit/TiO₂, with molar ratios of Ag:TiO₂ of 0.04 and 0.08, respectively, by adding proper amounts of TiO₂ to the initial 142 143 AgNO₃ aqueous solution before the reduction reaction.

144 In addition to the citrate-stabilized nanocomposites, three nanocomposites based on naked 145 silver (i.e., without adding citrate as stabilizing agent during the synthesis of the silver nanoparticles) 146 were prepared. Briefly, an aqueous solution of NaBH₄ (4 mL, 0.04 mmol) was added drop by drop to 147 20 mL of an aqueous solution of AgNO₃ (0.02, 0.01 or 0.005 mmol) and TiO₂ (0.12 mmol) under 148 magnetic stirring in an ice bath. The reaction mixture was stirred for 10 min. Depending on the 149 amount of silver used in the synthesis three different nanocomposites were obtained: Ag/TiO₂, 2xAg/TiO₂ and 4xAg/TiO₂, with molar ratios of Ag:TiO₂ of 0.04, 0.08 and 0.16, respectively. The 150 151 resulting pale yellow aqueous-colloidal suspensions of naked-silver/TiO₂ nanocomposites were kept in 152 darkness until being used in biocide assays.

All the obtained nanoparticles and nanocomposites have been characterized by UV-Visible
(UV-Vis) spectrophotometry and Dynamic Light Scattering (DLS). UV-Vis spectra were recorded on
an Ocean optics spectrometer equipped with a HR4000 detector (Dunedin, FL. USA). Hydrodynamic
diameter (HD) and zeta potential (ζ) were determined by DLS on a Zetatrac Analyzer (Microtrac,

USA). Measurements were carried out at 25 °C in water, by placing proper amounts of the 157 158 nanocomposite suspensions in the sample holders. The morphology and structure of the $Ag(a)cit/TiO_2$ 159 nanocomposite at the microscopic level were analyzed using Scanning Electron Microscopy (SEM) 160 and Transmission Electron Microscopy (TEM). SEM images were recorded in a GeminiSEM 300 161 microscope. For TEM experiments we employed a high resolution TEM with field mission gun (FEG-162 HRTEM) from FEI Company (TECNAI G2 F30 S-twin), with a Fischione high angle annular dark-163 field detector (HAADF) (0.16 nm point resolution) to work in Scanning Transmission Electron 164 Microscopy (STEM) mode, and one INBCA ZX-max 80 silicon drift detector for Energy-dispersive X-ray spectroscopy (EDXS) analysis. The experiments were performed at 300 kV with a resolution of 165 0.2 nm. The analysis of HR micrographs and the Fast Fourier transform (FFT) for phase interpretation 166 167 were performed with Digital Micrograph software (Gatan Inc.) and the Java version of JEM Software. 168 The analysis of the HAADF-STEM images and the EDS spectra profile were done with ES Vision 169 software (FEI Company). The samples for the microcharacterization of both composites were prepared 170 by drying a drop of the suspensions on a carbon coated copper grid.

171 2.3. Biopatina sampling and characterization

172 The vulnerability of historical buildings to different agents in Seville was previously studied by some of the authors of this work in the façades of thirty monuments from this city [38]. In the 173 174 present study we have evaluated the frequency and degree of biodegradation of these façades and the 175 necessity of a preventive conservation treatment based on biocides. The biodegradation levels were 176 classified according to the Vulnerability Index described by Ortiz and Ortiz [39] as follows: low 177 frequency, which indicates that the presence of the biodegradation is incipient and still difficult to 178 detect; medium frequency, which indicates that weathering forms are easily identified, and high 179 frequency, which indicates a high rate of occurrence.

180 To design the accelerated weathering test we used real-life algae grown on monuments 181 affected by biodeterioration. Thirteen samples of biocrust from the façades were cut out with a scalpel 182 and studied by scanning electron microscopy (SEM-EDX JEOL JSM-5400), X-ray diffraction (XRD 183 Bruker D8C) and Zarbeco handheld digital microscope. Afterwards, the samples of biocrust were 184 cultured in a phosphate-rich medium (NaH₂PO₄x2H₂O,20 mg/L) and the main algal species identified

185 by means of optical microscopy (Leica DM2500 optical microscope).

186 2.4. Accelerated weathering test

The previously prepared algae cultures were employed to perform an accelerated weathering 187 188 test on limestones by immersing stone slabs (1.5 x 1.5 x 0.5 cm) inside a culture tank at room 189 temperature for 28 days with permanent lighting (tungsten-halogen filament lamp whose emission of 190 UV radiation is about 2% of total emission [40], 100W). At the end of this period the stone slabs were 191 allowed to dry for 5 days at room temperature. Additionally, accelerated weathering tests on treated 192 slabs were performed to establish the capability of the nanocomposites to inhibit biopatina growth on 193 limestones by depositing doses up to 200 μ L of a water suspension of the nanocomposite on the stone 194 slab surface, and allowing the spot to dry at room temperature before immersing the slide in the tank. 195 The concentration of the nanoparticle suspension was adjusted to obtain 0.03 mg/mL referred to silver. 196 The surfaces of the samples were analyzed before and after the accelerated weathering test by means 197 of a Zarbeco handheld digital microscope and a Leica DM4500M optical microscope to determine the 198 appearance of biofilms, and a colorimeter X-Rite SP60 to evaluate chromatic changes induced by the 199 nanocomposites and/or the generation of biopatina. Untreated stones, subjected to the accelerated 200 weathering test, when required, were used as control for the chromatic changes determinations.

201 Chromatic changes (ΔE^*) were calculated according to the parameters defined by the 202 CIELAB color-system according to Eq. (1).

203
$$\Delta E^* = \sqrt{\Delta L^{*2} + \Delta a^{*2} + \Delta b^{*2}}$$
 Eq. (1)

204 Where ΔL^* , Δa^* and Δb^* characterize variations of the color values as referred to the control in 205 the black-white (brightness), red-green and yellow-blue axes, respectively.

- 206
- 207

209 **3. Results and discussion**

210 3.1. Treatments

211 Silver nanoparticles, TiO₂ nanoparticles and silver/TiO₂-nanocomposites have been selected to evaluate their effectiveness in preventing biopatina growth on limestone. Silver/TiO₂-nanocomposites 212 213 were obtained by reducing silver nitrate with a strong reducing agent, sodium borohydride, in an 214 aqueous medium containing TiO₂ nanoparticles. Different nanocomposites were prepared by varying 215 the molar ratio Ag:TiO₂ (Table 1) and, additionally, a set of nanocomposites was prepared by 216 introducing a second additive, citrate, during the reduction process. We have chosen citrate because it 217 is a stabilizing agent widely employed in the synthesis of silver nanoparticles, in order to diminish the 218 hydrodynamic diameter of nanocomposites, as we will discuss later on. Regarding TiO₂, our source 219 was P-25 from Degussa. We have chosen P-25 because it is a standard material widely employed in 220 photocatalytic applications [41], including the photodegradation of organic matter [42]. P-25 contains 221 anatase and rutile phases in a 3:1 ratio, with average sizes of 85 and 25 nm, respectively, as 222 determined by transmission electron microscopy [43].

Table 1. Molar Ag:TiO₂ ratio, hydrodynamic diameter (HD) and zeta potential (ζ) of the different types of nanocomposites compared to TiO₂ and Ag@cit nanoparticles.

Treatment	Ag:TiO ₂	HD (nm)	ζ (mV)		
TiO ₂	-	184±49	-17±2		
Ag@cit	-	36±8	-63±3		
Ag@cit/TiO ₂	0.04	72±18	-24.8±0.3		
2xAg@cit/TiO2	0.08	52±11	-27±1		
Ag/TiO ₂	0.04	234±32	- 16±1		
2xAg/TiO ₂	0.08	170±26	-17±1		
4xAg/TiO ₂	0.16	94±33	-17±3		

In summary, we have developed two types of treatments based on silver/TiO₂nanocomposites, one based on silver nanoparticles stabilized by citrate and one based on naked silver nanoparticles. In addition to nanocomposites we have also assayed silver and TiO_2 nanoparticles to unpick the effect in isolation versus in combination.

230 As can be seen in Table 1, the least favorable HDs are obtained for TiO₂ nanoparticles and 231 naked-silver nanocomposites. Their high HD and low zeta potential are indicative of low colloidal 232 stability and aggregation [44]. Far more favorable HDs are achieved by citrate-stabilized silver 233 nanocomposites owing to the stabilizing effect of trisodium citrate. In fact, Ag@cit nanoparticles show 234 the lowest HD and extreme zeta potential (-63 mV). Moreover, Ag@cit nanoparticles have a narrow 235 surface plasmon peaking at 400 nm (Fig. S.1), which indicates that they are small and have a narrow 236 size distribution. Nanocomposites obtained from citrate-stabilized silver nanoparticles exhibit higher 237 stability in aqueous medium than those obtained from naked silver nanoparticles. As in the case of 238 Ag@cit nanoparticles, their TiO₂-derived nanocomposites show a silver plasmon peaking at 400 nm, 239 whereas naked-silver nanocomposites plasmon peaking appears at higher wavelength, 440 nm, 240 indicating that the size of the latter is larger (Fig. S.1). These results are in good agreement with the 241 values of HD (Table 1). It is worth noting that both for Ag@cit and naked-silver nanocomposites, 242 decreasing the molar ratio Ag:TiO₂ (i.e. the amount of silver in the formulation) leads to 243 destabilization of the resulting nanocomposites as evidenced by the increase in HD and zeta potentials 244 closer to zero [30].

245 The TEM analysis of both composites indicates that the samples are formed by agglomerated TiO₂ and Ag particles. An example of each type of nanocomposite is shown in Figures 2.a and 3a. 246 247 Rounded silver nanocrystals (darker contrast) are deposited on top of the TiO₂ nanoplatelets; therefore, 248 it was difficult to measure the Ag interplanar spacing. However in the Ag/TiO₂ composite it was 249 possible to found a small nanocrystal of silver (marked square) with cubic structure (space group 225=Fm-3m; cell parameter a=4.09 Å) as can be seen in the HRTEM image presented in Fig. 2.b; such 250 Ag crystal is oriented along the [1 0 1] zone axis as indicate the inset FFT. In the case of Ag@cit/TiO2 251 252 composite only (2 0 0) silver planes were measured, marked in the micrograph presented in Fig. 3.b. 253 TiO₂ nanocrystals present faceted shape and correspond with the lighter contrast in the TEM images. 254 In Figures 2.d, 3.c and 3.d, three HRTEM micrographs of some of these nanoplatelets are presented. 255 Which are oriented along the $\begin{bmatrix} 1 & 1 \\ 1 \end{bmatrix}$ zone axis for Ag/TiO₂ and along $\begin{bmatrix} 1 & 0 \\ 0 \end{bmatrix}$ and $\begin{bmatrix} 1 & 0 \\ 1 \end{bmatrix}$ for Ag(a)cit/TiO₂, and belong to the tetragonal structure of Anatase (space group 141 = I41 / amd; 256 parameters of cell a = 3.78 and c = 9.51Å), the corresponding FFT are inserted. 257



258

Fig. 2. (a) TEM image of the Ag/TiO₂ composite; (b) HRTEM image showing a silver crystallite (square marked) oriented along [1 0 1], the corresponding FFT is inset; (c) HRTEM micrograph of TiO₂ nanocrystal with the (1 0 1) planes marked and (d) TiO₂ crystallite oriented along [1 1 1] zone axes and FFT inset.



Fig. 3. (a) TEM image of the Ag@cit/TiO₂ composite; (b) HRTEM micrograph showing a silver nanocrystal
and (c and d) HRTEM images of TiO₂ nanocrystals oriented along [1 0 0] and [1 1 0] zone axes respectively,
the corresponding FFT are inset.

The chemical microanalysis was carried out using an EDS spectroscopic technique in STEM mode (HAAD detector) and the most representative results are shown in Figures 4 and 5. At this point it is important to note that the image contrast obtained in TEM and STEM is inverse, In TEM images, the darker contrast corresponds to heavier elements while in STEM images corresponds to lighter elements, due to the different detectors used to get the images.



271

Fig. 4. (a) STEM image of the Ag/TiO₂ composite and (b and c) EDX spectra corresponding to points one and

two respectively, marked in the STEM image.



Fig. 5. (a) STEM image of the Ag@cit/TiO₂ composite (the red line corresponds to the EDX profile; five point
analysis) and EDX spectra of point 1 (top inset) and point 3 (bottom inset). (b) Ti/Ag atomic percentage ratio
along the profile

An EDS analysis of the Ag/TiO₂ composite was done in two different positions, one in a bright contrast particle and another in a dark contrast particle, marked in the STEM image presented in Fig. 4.a (1 and 2 respectively). The obtained spectra for both positions are depicted in Fig. 4.b and 4.c, the brighter position (1) clearly corresponds to a silver nanocrystal on the top of TiO₂ particles, and 282 position two belongs to a TiO₂ nanoplatelet. The C-K and Cu-K picks are coming from the grid. A 283 composition profile was made for the Ag@cit/TiO₂ composite along the red line marked in the STEM 284 micrograph (Fig. 5.a). Figure 5.b shows the Ti/Ag atomic percentage along such profile (five points 285 were analyzed). The obtained EDS spectra recorded in point one and three are inset in the top and bottom of the Fig. 5.a respectively. In the same way that the previous composite, the darker contras 286 (point 1) belongs to a TiO₂ nanoplatelet and the lighter contrast (point 3) corresponds to a silver 287 288 crystal, which is located on the top of TiO₂ particles. In addition to Ti, O and Ag elements, carbon was 289 also found in the analyzed profile; C element must come from the citrate molecules as the C from the 290 grid is discarded since the analyzed area was supported on a hole. Si and S elements must be 291 impurities in both composites.

3.2. Biopatina characterization and use in accelerated weathering

293 On site diagnosis of the thirty churches studied in Seville show that biopatina and vegetation appear in 90% of these buildings (Fig.6), that means that 27 of the 30 churches studied need a 294 295 preventive conservation treatment against biocolonization. The frequency of the biocrust is low and medium, and the frequency of vegetation is mainly low and medium, although it reaches high levels if 296 297 preventive conservation of roofs is not carried out. Moreover, the frequency of these weathering forms 298 change on the walls of a same building, as the distribution of these pathologies depends on the 299 orientation; they appear commonly on Northern and Southern walls, and on the dampness due to 300 drainpipes, water sprouts or capillarity.



302 Fig. 6. Vulnerability to biocrust (blue bars) and vegetation (green bars) in thirty churches studied in Seville.

In order to achieve a representative green biopatina culture for the biocrust that develops in the climatic conditions of Seville, samples taken from the façades of different churches in Seville have been collected, characterized and grown to obtain a culture. These façades were built with limestones, sandstones and calcarenites similar to slabs from Utrera (Spain)[38]. The samples were collected in different conditions and times (autumn and spring according to the rain periods in Seville).

308 The SEM study of the samples showed that these biopatinas are spread on high porosity 309 substrates generating biological deposits with thicknesses between $10-400\mu m$ (Fig. S.2.a). 310 Furthermore, highly porous stones favor the penetration of microorganisms through intergranular areas 311 causing fissures and powdering (Fig. S.2.b), two of the most important physical damages associated to 312 this kind of alteration agent. The results of the EDX analysis show only silicon (Si) and Calcium (Ca) 313 in the composition of the samples (Fig. S.2.c), corroborating that the damages of these biopatinas and 314 biocrusts are only physical and aesthetic because no sub-products of alteration have been identified..

315 Green biocrust samples were grown, thoroughly mixed and cultured. The rich mixture 316 contained viable micro-organisms that had proven to be especially adapted to grow on limestone, 317 covering a broad range of environmental and surface conditions, and are therefore deemed to be 318 suitable for a quick colonization of fresh limestone slabs. This formed the basis for our accelerated 319 weathering test.

Five types of green algae (Chlorophyta) were identified in the culture, corresponding to species of the genera Chlorella, Botryococcus, Cyclotella, Monoraphidium and phytoflagellate and two blue-green algae (Cyanobacteria) corresponding to the species of the genera Gloeocapsa and Nostoc.

324 3.3. Aesthetical impact of the treatments to inhibit biopatina

The chromatic changes in stone slabs were measured after the application of nanocomposites both before (ante weathering, aw) and after (post weathering, pw) weathering tests. Measures taken ante weathering test have the purpose of evaluating the initial aesthetic impact of the different treatments on the stone, while measures taken post weathering test show the effectiveness of treatments as inhibitors of the formation of biopatina. Table 2 shows the chromatic coordinate

- 330 increments ΔL^* , Δa^* and Δb^* obtained for the stone slabs ante and post-weathering tests, as well as the
- 331 corresponding ΔE^* , calculated according to Eq.1, and the total increments.
- 332 Table 2. Increases of color caused by the application of nanocomposites and the generation of
- 333 biopatina

	ante-Weathering test [aw]				post-Weathering test [pw]				Final increments			
Product	ΔL^*	∆a*	$\varDelta b^*$	ΔE^*	ΔL^*	∆a*	$\varDelta b^*$	ΔE^*	ΔL^*	∆a*	$\varDelta b^*$	ΔE^*
TiO ₂	0.31	-0.10	-0.77	0.84	-8.09	6.7	14.52	17.92	-7.78	6.60	13.75	17.12
Ag@cit	-5.92	0.72	3.05	6.69	-7.25	6.89	15.21	18.20	-13.17	7.61	18.26	23.76
Ag@cit/TiO2	-8.40	2.37	5.13	10.12	-3.64	2.35	1.75	4.67	-12.04	4.72	6.88	14.65
2xAg@cit/TiO2	-8.07	1.50	2.70	8.63	-1.32	2.2	4.42	5.11	-9.39	3.70	7.12	12.34
Ag/TiO ₂	-19.12	8.40	12.30	24.24	-5.51	-3.78	-1.47	6.84	-24.63	4.62	10.38	27.30
2xAg/TiO2	-9.38	4.61	9.15	13.89	-8.26	0.85	10.01	13.01	-17.64	5.46	19.16	26.61
4xAg/TiO2	-4.97	0.99	0.88	5.14	-14.47	7.84	17.93	24.34	-19.44	8.83	18.81	28.46
Untreated sample	-	-	-	-	-18.66	5.47	17.88	26.42	-18.66	5.47	17.88	26.42

334

335 The visual impact of treatment is shown in Figure 7, in which stone slabs are shown after the 336 deposition of the different treatments. Ag/TiO₂ and 2xAg/TiO₂ nanocomposites exhibit a noticeable 337 change in color when compared to control. This is in keeping with data quantified using $\Delta E^*[aw]$ tool.



338

Fig. 7. Digital photographs of limestone slabs after the application of treatments (200 μ L of an aqueous suspension of nanocomposites). All of the silver-bearing treatments have the same silver dose.

341 Remarkably, the chromatic changes (a* and b*) produced by the application of the treatments 342 (Fig. 8) remain within the range of variability of the control untreated limestone from Utrera, except 343 for Ag/TiO₂ and $2xAg/TiO_2$. These nanocomposites are significantly shifted to higher values (i.e. red 344 and yellow). Other treatments remain close to the chromatic values of the untreated limestone.



345

Fig. 8. Chromatic changes a*/b* and brightness L* (inset) of the limestone slabs following treatment (anteweathering test). Red lines highlight the variability of the untreated stone samples.

348 The effect of initial treatment on the ΔL^* variable is generally negative causing darkening, 349 excepted for Ag@cit and TiO₂. For other samples, the darkening remains within the variability range 350 of the untreated stone (Figure 8, inset).

351 The high color change originated by Ag/TiO₂ and 2xAg/TiO₂ nanocomposites, when compared to 4xAg/TiO₂, can be explained based on their colloidal stability. 4xAg/TiO₂ nanocomposite 352 353 has a smaller hydrodynamic diameter and similar zeta potential than its analogues Ag/TiO₂ and 2xAg/TiO₂, and is therefore expected to be more stable. Moreover, the concentration of the aqueous 354 355 suspension of all nanocomposites was adjusted to obtain 0.03 mg/mL (referred to silver) in the assays. The TiO₂ amount was 0.53 mg/mL in the sample without silver, in Ag/TiO₂ and in Ag@cit/TiO₂, 356 357 whereas in suspensions with higher relative silver amounts (doubled and quadrupled) than that of their 358 analogues, the TiO_2 amount is therefore lowered in the same proportion. As a consequence, both 359 Ag/TiO_2 and $2xAg/TiO_2$ aggregate and excess nanocomposite precipitates, they saturate the pores on 360 the stone surface and bring about a change in color (Fig. 9).





362 Fig. 9. Magnified (140x) stone surfaces after treatment, analyzed by Zarbeco handheld digital microscope

363 In the case of citrate-capped silver/TiO₂ nanocomposites the situation is similar; however, as 364 these nanocomposites are much smaller and more stable than the related nanocomposites obtained 365 from naked-silver nanoparticles, the effect is not so remarkable. The general trend of nanocomposites 366 to aggregate after deposition is also observed by SEM (Fig. S.3).

367 3.4. Effectiveness of the treatments in accelerated weathering test

368 As mentioned above, the suitability of the nanocomposites to inhibit the generation of 369 biopatina was assayed by means of spectrophotometric color measurements, a method that is emerging 370 as an improved alternative to previous methods such as chlorophyll α determination or fluorescein 371 diacetate hydrolysis [45–48]. Among the advantages of the spectrophotometric color measurement 372 method, it should be noted that it is a non-destructive technique that can be performed on-site, not 373 requiring expensive and time-consuming sample preparation [46–48].

374 Three ΔE^* values have been calculated, the first one before the weathering test $\Delta E^*[aw]$, the second one after the weathering test $\Delta E^{*}[pw]$ and the third one $\Delta E^{*}[t]$. $\Delta E^{*}[aw]$ characterizes the 375 376 initial aesthetical impact of treatment, as it measures the chromatic change from untreated stone slabs 377 to freshly treated slabs, $\Delta E^{*}[pw]$ from freshly treated slabs to weathered slabs, and $\Delta E^{*}[t]$ describes the overall change from untreated slabs to treated and weathered ones. Fig. 10 summarizes the 378 379 chromatic changes for the treatments studied. Although there is no standardized rule for the 380 interpretation of color changes in Cultural Heritage materials as a result of the application of biocide 381 products, three ranges of chromatic changes have been proposed for dyes in Cultural Heritage porous materials: $\Delta E^{*} < 5$, chromatic changes cannot be detected by the human eye; $5 < \Delta E^{*} < 10$, chromatic changes can be detected by the human eye but still acceptable; and $\Delta E^{*} > 10$, chromatic changes are clearly visible [49]. This global color variation, $\Delta E^{*} < 10$, has also been established as threshold by other authors [50]. In our case, the initial aesthetic impact of the different treatments on the stone, i.e. *ante*-weathering test, have a global color variation ΔE^{*} [aw]<10 except for Ag/TiO₂ and 2xAg/TiO₂ nanocomposites TiO₂ nanoparticles and the 4xAg/TiO₂ nanocomposite achieve ΔE^{*} [aw]<5.



388

389 Fig. 10. $\Delta E^*[aw]$, $\Delta E^*[pw]$ and $\Delta E^*[t]$ for the different treatments.

As can be seen in Figure 10, both TiO₂ and Ag@cit nanoparticles inhibit the biopatina growth. Nevertheless, the extent of this inhibition is modest, and $\Delta E^*[pw]$ is reduced by only 25-30%, as referred to the untreated sample. On the other hand, the combination of these active ingredients into a nanocomposite, Ag@cit/TiO₂ nanocomposites, greatly increases the effectiveness of treatment when compared to nanoparticles themselves so that ΔE^* is reduced by over 80%. Due to this synergy, increasing the Ag:TiO₂ ratio does not slow down biopatina growth, as can be seen in Figure 10. This synergy could be due to a lowering by silver of the band gap in TiO₂ [27].

397 On the other hand, the overall chromatic changes, $\Delta E^*[t]$, were smaller in the treated slabs 398 when compared to the untreated ones, except for the naked silver/TiO₂ nanocomposites. Remarkably, 399 above 80% reduction in biopatina formation was achieved by Ag@cit/TiO₂ and 2xAg@cit/TiO₂ 400 nanocomposites, with $\Delta E[t]^*$ values close to 10.

401 4. Conclusions

402 A variety of treatments that we designed to prevent biopatina formation on limestone were 403 investigated. The initial impact was assessed, and then the stone slabs, which stem from a quarry in Utrera, were subjected to accelerated weathering. The accelerated weathering test used real-world 404 algae grown on historical monuments affected by biodeterioration. Whereas both silver nanoparticles 405 406 and titanium dioxide P25 nanoparticles achieved some degree of inhibition on their own, it was the 407 combination of both in a small-size, stable, citrate promoted nanocomposite that achieved a very 408 significant reduction. These nanocomposites minimize the aesthetical impact on the surfaces on which 409 they were applied. They can be prepared in an aqueous medium, avoiding the need for organic 410 solvents. They could be applied both by spray or by brush, or be mixed with other formulations such 411 as mortars to confer them biocide properties as preventive conservation measures.

412 Acknowledgements

413 This study has been partially supported by the projects: Art-Risk, RETOS project of Ministerio de 414 Economía y Competitividad and Fondo Europeo de Desarrollo Regional (FEDER), (code: BIA2015-415 64878-R (MINECO/FEDER, UE)), CTQ2013-48396-P of Fondo Europeo de Desarrollo Regional 416 (FEDER-Unión Europea) and Ministerio Economía y Competitividad and the research teams P10-417 FQM-6615, TEP-199 and FQM-319 from Junta Andalucía. J. Becerra is grateful to the Ministerio de 418 Educación, Cultura y Deporte for his pre-doctoral fellowship.

419

420 Bibliography

- 421 [1] M.F. Macedo, A.Z. Miller, A. Dionisio, C. Saiz-Jimenez, Biodiversity of cyanobacteria and green algae
 422 on monuments in the Mediterranean Basin: An overview, Microbiology. 155 (2009) 3476–3490.
 423 doi:10.1099/mic.0.032508-0.
- 424 [2] C. Gaylarde, M. Ribas Silva, T. Warscheid, Microbial impact on building materials: an overview, Mater.
 425 Struct. 36 (2003) 342–352. doi:10.1007/BF02480875.
- 426 [3] Y. Nuhoglu, E. Oguz, H. Uslu, A. Ozbek, B. Ipekoglu, I. Ocak, I. Hasenekoglu, The accelerating effects of

- the microorganisms on biodeterioration of stone monuments under air pollution and continental-cold
 climatic conditions in Erzurum, Turkey, Sci. Total Environ. 364 (2006) 272–283.
 doi:10.1016/j.scitotenv.2005.06.034.
- *K. Sterflinger, G. Piñar, Microbial deterioration of cultural heritage and works of art Tilting at windmills?, Appl. Microbiol. Biotechnol.* 97 (2013) 9637–9646. doi:10.1007/s00253-013-5283-1.
- 432 [5] C.A. Crispim, C.C. Gaylarde, Cyanobacteria and biodeterioration of cultural heritage: A review,
 433 Microb. Ecol. 49 (2005) 1–9. doi:10.1007/s00248-003-1052-5.
- 434 [6] J.P. Calvo, M. Regueiro, Carbonate rocks in the Mediterranean region from classical to innovative
 435 uses of building stone, Geol. Soc. London, Spec. Publ. 331 (2010) 27–35. doi:10.1144/sp331.3.
- 436 [7] M.P. Nugari, O. Salvadori, Biocides and Treatment of Stone: Limitations and Future Prospects, in: R.J.
- 437 Koestler, V.H. Koestler, A.E. Charola, F. Nieto-Fernandez (Eds.), Art, Biol. Conserv. Biodeterior. Work.
 438 Art, The Metropolitan Museum of Art, New York, 2003: pp. 518–535.
- 439 [8] F. Bastian, V. Jurado, A. Nováková, C. Alabouvette, C. Saiz-Jimenez, The microbiology of Lascaux
 440 Cave, Microbiology. 156 (2010) 644–652. doi:10.1099/mic.0.036160-0.
- 441 [9] F. Bellissima, M. Bonini, R. Giorgi, P. Baglioni, G. Barresi, G. Mastromei, B. Perito, Antibacterial
 442 activity of silver nanoparticles grafted on stone surface, Environ. Sci. Pollut. Res. 21 (2014) 13278–
 443 13286. doi:10.1007/s11356-013-2215-7.
- 444 [10] A.M.M. Essa, M.K. Khallaf, Biological nanosilver particles for the protection of archaeological stones
 445 against microbial colonization, Int. Biodeterior. Biodegrad. 94 (2014) 31–37.
 446 doi:10.1016/j.ibiod.2014.06.015.
- 447 [11] B. Gutarowska, J. Skora, K. Zduniak, D. Rembisz, Analysis of the sensitivity of microorganisms
 448 contaminating museums and archives to silver nanoparticles, Int. Biodeterior. Biodegrad. 68 (2012) 7–
 449 17. doi:10.1016/j.ibiod.2011.12.002.
- 450 [12] J.S. Kim, E. Kuk, K.N. Yu, J.H. Kim, S.J. Park, H.J. Lee, S.H. Kim, Y.K. Park, Y.H. Park, C.Y. Hwang,
 451 Y.K. Kim, Y.S. Lee, D.H. Jeong, M.H. Cho, Antimicrobial effects of silver nanoparticles, Nanomedicine
 452 Nanotechnology, Biol. Med. 3 (2007) 95–101. doi:10.1016/j.nano.2006.12.001.
- 453 [13] E. Cavaliere, S. De Cesari, G. Landini, E. Riccobono, L. Pallecchi, G.M. Rossolini, L. Gavioli, Highly

- 454 bactericidal Ag nanoparticle films obtained by cluster beam deposition, Nanomedicine Nanotechnology,
 455 Biol. Med. 11 (2015) 1417–1423. doi:10.1016/j.nano.2015.02.023.
- 456 [14] C.-N. Lok, C.-M. Ho, R. Chen, Q.-Y. He, W.-Y. Yu, H. Sun, P.K.-H. Tam, J.-F. Chiu, C.-M. Che, Silver
 457 nanoparticles: partial oxidation and antibacterial activities, JBIC J. Biol. Inorg. Chem. 12 (2007) 527–
 458 534. doi:10.1007/s00775-007-0208-z.
- 459 [15] I. Sondi, B. Salopek-Sondi, Silver nanoparticles as antimicrobial agent: A case study on E. coli as a
 460 model for Gram-negative bacteria, J. Colloid Interface Sci. 275 (2004) 177–182.
 461 doi:10.1016/j.jcis.2004.02.012.
- 462 [16] S. Pal, Y.K. Tak, J.M. Song, Does the Antibacterial Activity of Silver Nanoparticles Depend on the Shape
 463 of the Nanoparticle? A Study of the Gram-Negative Bacterium Escherichia coli, Appl. Environ.
 464 Microbiol. 73 (2007) 1712–1720. doi:10.1128/AEM.02218-06.
- 465 [17] P. Van Dong, C. Ha, L. Binh, J. Kasbohm, Chemical synthesis and antibacterial activity of novel-shaped
 466 silver nanoparticles, Int. Nano Lett. 2 (2012) 9. doi:10.1186/2228-5326-2-9.
- 467 [18] M. Rai, A. Yadav, A. Gade, Silver nanoparticles as a new generation of antimicrobials, Biotechnol. Adv.
 468 27 (2009) 76–83. doi:10.1016/j.biotechadv.2008.09.002.
- 469 [19] E. Quagliarini, F. Bondioli, G.B. Goffredo, A. Licciulli, P. Munafò, Smart surfaces for architectural
 470 heritage: Preliminary results about the application of TiO2-based coatings on travertine, J. Cult. Herit.
 471 13 (2012) 204–209. doi:10.1016/j.culher.2011.10.002.
- 472 [20] E. Quagliarini, F. Bondioli, G.B. Goffredo, C. Cordoni, P. Munafò, Self-cleaning and de-polluting stone
 473 surfaces: TiO2 nanoparticles for limestone, Constr. Build. Mater. 37 (2012) 51–57.
 474 doi:10.1016/j.conbuildmat.2012.07.006.
- 475 [21] P. Munafò, G.B. Goffredo, E. Quagliarini, TiO2-based nanocoatings for preserving architectural stone
 476 surfaces: An overview, Constr. Build. Mater. 84 (2015) 201–218.
 477 doi:10.1016/j.conbuildmat.2015.02.083.
- 478 [22] F. Pacheco-Torgal, S. Jalali, Nanotechnology: Advantages and drawbacks in the field of construction
 479 and building materials, Constr. Build. Mater. 25 (2011) 582–590.
 480 doi:10.1016/j.conbuildmat.2010.07.009.

- 481 [23] L. Graziani, E. Quagliarini, F. Bondioli, M. D'Orazio, Durability of self-cleaning TiO2 coatings on fired
 482 clay brick façades: Effects of UV exposure and wet & dry cycles, Build. Environ. 71 (2014) 193–203.
 483 doi:10.1016/j.buildenv.2013.10.005.
- 484 P. Munafò, E. Quagliarini, G.B. Goffredo, F. Bondioli, A. Licciulli, Durability of nano-engineered TiO2 [24] 485 self-cleaning treatments on limestone, Constr. Build. Mater. 65 (2014)218–231. 486 doi:10.1016/j.conbuildmat.2014.04.112.
- 487 [25] M.F. La Russa, S.A. Ruffolo, N. Rovella, C.M. Belfiore, A.M. Palermo, M.T. Guzzi, G.M. Crisci,
 488 Multifunctional TiO2 coatings for Cultural Heritage, Prog. Org. Coatings. 74 (2012) 186–191.
 489 doi:10.1016/j.porgcoat.2011.12.008.
- 490 [26] H.A. Foster, I.B. Ditta, S. Varghese, A. Steele, Photocatalytic disinfection using titanium dioxide:
 491 Spectrum and mechanism of antimicrobial activity, Appl. Microbiol. Biotechnol. 90 (2011) 1847–1868.
 492 doi:10.1007/s00253-011-3213-7.
- 493 [27] M.F. La Russa, A. Macchia, S.A. Ruffolo, F. De Leo, M. Barberio, P. Barone, G.M. Crisci, C. Urz??,
 494 Testing the antibacterial activity of doped TiO2 for preventing biodeterioration of cultural heritage
 495 building materials, Int. Biodeterior. Biodegrad. 96 (2014) 87–96. doi:10.1016/j.ibiod.2014.10.002.
- 496 [28] Y. Zhao, B. Yang, J. Xu, Z. Fu, M. Wu, F. Li, Facile synthesis of Ag nanoparticles supported on TiO 2
 497 inverse opal with enhanced visible-light photocatalytic activity, Thin Solid Films. 520 (2012) 3515–
 498 3522. doi:10.1016/j.tsf.2011.12.076.
- 499 [29] I. Yaşa, N. Lkhagvajav, M. Koizhaiganova, E. Çelik, Ö. Sari, Assessment of antimicrobial activity of
 500 nanosized Ag doped TiO2 colloids, World J. Microbiol. Biotechnol. 28 (2012) 2531–2539.
 501 doi:10.1007/s11274-012-1061-y.
- 502 [30] M. Lungu, Ş. Gavriliu, E. Enescu, I. Ion, A. Brătulescu, G. Mihăescu, L. Măruţescu, M.C. Chifiriuc,
 503 Silver-titanium dioxide nanocomposites as effective antimicrobial and antibiofilm agents, J.
 504 Nanoparticle Res. 16 (2014). doi:10.1007/s11051-013-2203-3.
- 505 [31] Y. Lin, W. Qiqiang, Z. Xiaoming, W. Zhouping, X. Wenshui, D. Yuming, Synthesis of Ag/TiO 2
 506 Core/Shell Nanoparticles with Antibacterial Properties, Bull. Korean Chem. Soc. 32 (2011) 2607–2610.
 507 doi:10.5012/bkcs.2011.32.8.2607.
- 508 [32] D. Nithyadevi, P. Suresh Kumar, D. Mangalaraj, N. Ponpandian, C. Viswanathan, P. Meena, Improved 22

- 509 microbial growth inhibition activity of bio-surfactant induced Ag-TiO2 core shell nanoparticles, Appl.
 510 Surf. Sci. 327 (2015) 504–516. doi:10.1016/j.apsusc.2014.11.169.
- 511 [33] X.H. Yang, H.T. Fu, X.C. Wang, J.L. Yang, X.C. Jiang, A.B. Yu, Synthesis of silver-titanium dioxide
 512 nanocomposites for antimicrobial applications, J. Nanoparticle Res. 16 (2014). doi:10.1007/s11051513 014-2526-8.
- 514 [34] M.A. Guerrero Montes, Diagnóstico del estado de alteración de la piedra del palacio consistorial de
 515 Sevilla. Causas y mecanismos., University of Seville, 1990.
- 516 [35] M.Á. Bello López, Caracterización y estado de alteración química de los materiales empleados en la
 517 construción de la Catedral de Sevilla, Univerisdad de Sevilla, 1988.
- 518 [36] P. Ortiz, M.A. Guerrero, M.A. Vázquez, R. Ortiz, J.M. Martín, M.C. Peña, Accelerated weathering test
 519 as environmental behaviour trials on calcareous stone, in: Proceeding 11th Int. Congr. Deterior.
 520 Conserv. Stone., 2008: pp. 223–231.
- 521 [37] C.Y. Flores, C. Diaz, A. Rubert, G.A. Benítez, M.S. Moreno, M.A. Fernández Lorenzo de Mele, R.C.
 522 Salvarezza, P.L. Schilardi, C. Vericat, Spontaneous adsorption of silver nanoparticles on Ti/TiO2
 523 surfaces. Antibacterial effect on Pseudomonas aeruginosa, J. Colloid Interface Sci. 350 (2010) 402–408.
 524 doi:10.1016/j.jcis.2010.06.052.
- 525 [38] F. Colao, R. Fantoni, P. Ortiz, M.A. Vazquez, J.M. Martin, R. Ortiz, N. Idris, Quarry identification of
 526 historical building materials by means of laser induced breakdown spectroscopy, X-ray fluorescence and
 527 chemometric analysis, in: Spectrochim. Acta Part B At. Spectrosc., 2010: pp. 688–694.
 528 doi:10.1016/j.sab.2010.05.005.
- 529 [39] R. Ortiz, P. Ortiz, Vulnerability Index: a new approach for preventive conservation of monuments., Int.
 530 J. Archit. Herit. (2016). doi:10.1080/15583058.2016.1186758.
- 531 [40] T.T. Schaeffer, Effects of Light on Materials in Collections : Data on Photoflash and Related Sources by
 532 Effects, 2001. http://ccl.worldcat.org/oclc/46785588.
- 533 [41] A. Ibhadon, P. Fitzpatrick, Heterogeneous Photocatalysis: Recent Advances and Applications, Catalysts.
 534 3 (2013) 189–218. doi:10.3390/catal3010189.
- 535 [42] A. Mills, S. Le Hunte, An overview of semiconductor photocatalysis, J. Photochem. Photobiol. A Chem.

- *536 108 (1997) 1–35. doi:10.1016/S1010-6030(97)00118-4.*
- 537 [43] T. Ohno, K. Sarukawa, K. Tokieda, M. Matsumura, Morphology of a TiO2 Photocatalyst (Degussa, P538 25) Consisting of Anatase and Rutile Crystalline Phases, J. Catal. 203 (2001) 82–86.
 539 doi:10.1006/jcat.2001.3316.
- 540 [44] K. Suttiponparnit, J. Jiang, M. Sahu, S. Suvachittanont, T. Charinpanitkul, P. Biswas, Role of Surface
 541 Area, Primary Particle Size, and Crystal Phase on Titanium Dioxide Nanoparticle Dispersion
 542 Properties, Nanoscale Res. Lett. 6 (2011) 1–8. doi:10.1007/s11671-010-9772-1.
- 543 [45] B. Prieto, B. Silva, O. Lantes, Biofilm quantification on stone surfaces: Comparison of various methods,
 544 Sci. Total Environ. 333 (2004) 1–7. doi:10.1016/j.scitotenv.2004.05.003.
- 545 [46] W. De Muynck, A.M. Ramirez, N. De Belie, W. Verstraete, Evaluation of strategies to prevent algal
 546 fouling on white architectural and cellular concrete, Int. Biodeterior. Biodegrad. 63 (2009) 679–689.
 547 doi:10.1016/j.ibiod.2009.04.007.
- 548 [47] P. Sanmartín, D. Vázquez-Nion, B. Silva, B. Prieto, Spectrophotometric color measurement for early
 549 detection and monitoring of greening on granite buildings., Biofouling. 28 (2012) 329–38.
 550 doi:10.1080/08927014.2012.673220.
- [48] P. Sanmartín, F. Villa, B. Silva, F. Cappitelli, B. Prieto, Color measurements as a reliable method for
 estimating chlorophyll degradation to phaeopigments, Biodegradation. 22 (2011) 763–771.
 doi:10.1007/s10532-010-9402-8.
- 554 [49] O. García, K. Malaga, Definition of the procedure to determine the suitability and durability of an anti555 graffiti product for application on cultural heritage porous materials, J. Cult. Herit. 13 (2012) 77–82.
 556 doi:10.1016/j.culher.2011.07.004.
- 557 [50] P. Ortiz, V. Antúnez, R. Ortiz, J.M. Martín, M.A. Gómez, A.R. Hortal, B. Martínez-Haya, Comparative
 558 study of pulsed laser cleaning applied to weathered marble surfaces, Appl. Surf. Sci. 283 (2013) 193–
 559 201. doi:10.1016/J.APSUSC.2013.06.081.