



The influence of highly dispersed Cu₂O-anchored MoS₂ hybrids on reducing smoke toxicity and fire hazards for rigid polyurethane foam



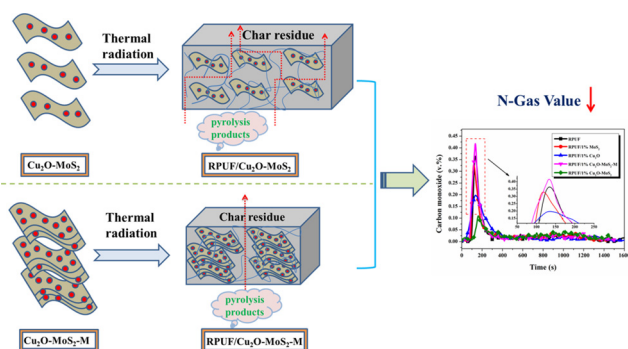
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GRAPHICAL ABSTRACT



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ABSTRACT

The extensive utilization of rigid polyurethane foam (RPUF) as construction insulation material has brought two main troubles to our society: fire risks and toxic hazards. To reduce the fire hazards of RPUF, a layered MoS₂ decorated with Cu₂O nanoparticles was creatively obtained by hydrothermal technology and facile wet chemical treatment for reducing the toxic product formations of polyurethane nanocomposites during combustion. Due to the low weight ratio of Cu₂O attached onto MoS₂, the resulting Cu₂O-MoS₂ hybrid effectively prevented the MoS₂ nanosheets from restacking. However, the Cu₂O-MoS₂-M hybrid was produced by increasing content of Cu₂O, which has the characteristic stacked layer structure of MoS₂. Reduced harmful organic volatiles and the toxic gases (e.g. a respective decrease of ca. 28% and 53% for CO and NO_x products) were obtained because of synergistic effect between the physical adsorption of MoS₂ and catalysis action of Cu₂O. Notably, the addition of Cu₂O-MoS₂ hybrids led to high char formation of the RPUF nanocomposite, indicating the effectively catalytic carbonization property. In addition, the N-Gas model for predicting fire smoke toxicity was developed and demonstrated. Furthermore, the research offers direct proofs of the negative influence of the stacked MoS₂ on reducing the smoke toxicity for RPUF nanocomposites.

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1. Introduction

Polyurethane (PU) is a significant member of polymers that is widely applied in many fields such as building insulation, household items, and automotive parts. Among PU materials, rigid polyurethane foam (RPUF), one of the most important commercial products, has been applied in household equipment and building material by reason of its gorgeous mechanical behavior and heating insulation property (Santos et al., 2017; Kacperski and Spsychaj, 1999; Pauzi et al., 2014). Nevertheless, once combustion, RPUF will produce extreme amounts of heat because of its the cellular structure (Chattopadhyay and Webster, 2009; Zhang et al., 2014), generating toxic smoke and poisonous gases (Zhang et al., 2016). These deleterious products can bring fire disaster and increased the death toll (Wang et al., 2014a; Levchik and Weil, 2005; Carty and White, 1994; Ling et al., 2015). Consequently, generous investigations and efforts have been made to reduce the generation of smoke toxicity suppression which has great influence on the human society.

With the increasing demand for reduction of toxic hazards, intensive attention should be drawn to the development of catalysts for reducing poisonous gases (Yu et al., 2016). Generally, the deleterious gaseous agents composes by carbon monoxide (CO), nitrogen oxide (NO_x), hydrogen cyanide (HCN) and harmful smoke (Ling et al., 2015), in the process of combustion and degradation for RPUF. Normally, metal oxide catalysts have been applied to reduce the exhaust gases of internal combustion engines (Sun et al., 2010; White et al., 2006; Kröcher and Elsener, 2009; Moreno et al., 2014; Shi et al., 2017). Transition metal oxide nanoparticles have received much attention from researchers which play an important role in carbonization and detoxification of the deleterious products. Among various transition metal oxides catalysts, Cu₂O and its derivatives with high activity and stability have been investigated as highly-efficient catalysts for the treatments of CO, NO_x, HCN and organic compounds (Zhao et al., 2006; Sun et al., 2014), which render Cu₂O a promising suppressant for toxic gaseous products. Tu and Wang (1996) found that smoke emission of polyvinyl chloride (PVC) can be controlled by Cu₂O resulting from its charring effect and reductive-coupling in the condensed phase. White et al. (2006) confirmed that Cu₂O exhibits predominant activity toward CO oxidation in CO/N₂/O₂ mixtures, and more than 99.5% transformation of CO to CO₂ has been realized.

Recently, MoS₂ has received considerable attention, which possesses a layered structure by van der Waals forces similar to graphene, because of its excellent performances such as thermal and chemical stability, and catalytic potential in the field of environmental and industrial catalysis research (Dinter et al., 2009; Breyse et al., 2008; Yu et al., 2015). More recently, MoS₂ nanosheets have been applied into polymer composites, which enormously improved their physical and chemical properties (Wang et al., 2017). Deng et al. found that water-dispersed noble metal nanocrystal modified MoS₂ nanosheets exhibit improved electrochemical catalytic activity compared to the individual property (Lee et al., 2010). Nevertheless, exfoliated MoS₂ nanosheets actually have a trend toward agglomeration and even restacking due to the strong van der Waals force that reduces their compatibility in polymers. It is reported that only the uniform dispersion of MoS₂ nanosheets can achieve the improvement of fire safety (Yuwen et al., 2014). Notably, Feng et al. found that the restacking of the two-dimensional MoS₂ nanosheets could be efficiently prevented by the densely anchored nanoparticles, that was beneficial for the homogeneous dispersion of MoS₂ nanosheets with polymers (Feng et al., 2014). Nonetheless, MoS₂ nanosheets on boosting the activity of Cu₂O on the MoS₂ surface for catalytic oxidation of CO production of RPUF nanocomposites are scarcely reported. Thus, it is reasonable to expect that the Cu₂O loaded on MoS₂ is the critical point lies in achieving high dispersion of MoS₂ nanosheets and thus restraining the fire hazards of RPUF.

To provide and guide the primary and immediate toxicity studies in a fire, the N-Gas model for predicting toxic hazard has been proposed

by NIST (the USA National Institute of Standards and Technology) for many years (Levin, 1996). This method makes the calculation of the toxic potency of fire smoke components available on the basis of the interaction of acute toxicity among the main components, oxygen depletion (Δ O₂), carbon monoxide (CO), carbon dioxide (CO₂), hydrogen cyanide (HCN) and nitrogen oxide (NO_x), hydrogen chloride (HCl), hydrogen bromide (HBr) (Levin, 1996). Essentially, the method of calculating toxic potency is based on the ratios of gaseous concentrations. On account of the experience formerly, the following 7-gas N-Gas model was presented:

$$\begin{aligned} \text{N-Gas Value} = & \frac{m[\text{CO}]}{[\text{CO}_2] - b} + \frac{21 - [\text{O}_2]}{21 - \text{LC}_{50}(\text{O}_2)} \\ & + \left(\frac{[\text{HCN}]}{\text{LC}_{50}(\text{HCN})} \times \frac{0.4[\text{NO}_2]}{\text{LC}_{50}(\text{NO}_2)} \right) + 0.4 \left(\frac{[\text{NO}_2]}{\text{LC}_{50}(\text{NO}_2)} \right) \\ & + \frac{[\text{HCl}]}{\text{LC}_{50}(\text{HCl})} + \frac{[\text{HBr}]}{\text{LC}_{50}(\text{HBr})} \end{aligned} \quad (1)$$

Where the LC50 (ppm) values are reference concentrations (RFC) of main gas components, which lead to death when inhaled for a specified time, typically 30 min. The terms “m” and “b” in equation connect with the elevated ventilation rate ascribed to the CO₂ concentration. For researches when the CO₂ concentration surpasses 5%, ‘m’ and ‘b’ equal 23 and -38600, respectively, whereas the ‘m’ and ‘b’ are -18 and 122,000 if the CO₂ concentration is 5% or less.

Herein, we reported a facile wet chemical route for the preparation of Cu₂O-MoS₂ hybrids with different loadings of tiny Cu₂O nanoparticles onto the surface of MoS₂ nanosheets by a facile wet chemical route. The optimal proportion of Cu₂O and MoS₂ was 1:1 which contributed to preventing restacking of MoS₂ nanosheets. However, a large numbers of restacked MoS₂ nanosheets were identified obviously with high weight ratio of Cu₂O to MoS₂, which was labeled as Cu₂O-MoS₂-M. In particular, Cu₂O-MoS₂ hybrid boosting the activity of toxicity suppression in RPUF has been clarified by comparison to Cu₂O-MoS₂-M hybrid. More importantly, the Cu₂O-MoS₂ hybrid provided the incremental quantity of catalytic active-sites afforded by the absence of stacked MoS₂ layers (Yang et al., 2015). Our research offers tangible proofs of the passive effects of the stacked MoS₂ nanosheets of Cu₂O-MoS₂-M hybrids on improving the ability of toxicity suppression and highlighting the importance of more rational use of Cu₂O loaded on MoS₂ for reducing the fire hazards of RPUF by means of N-Gas values which were generated and calculated by N-Gas model using the concentrations of fire effluents.

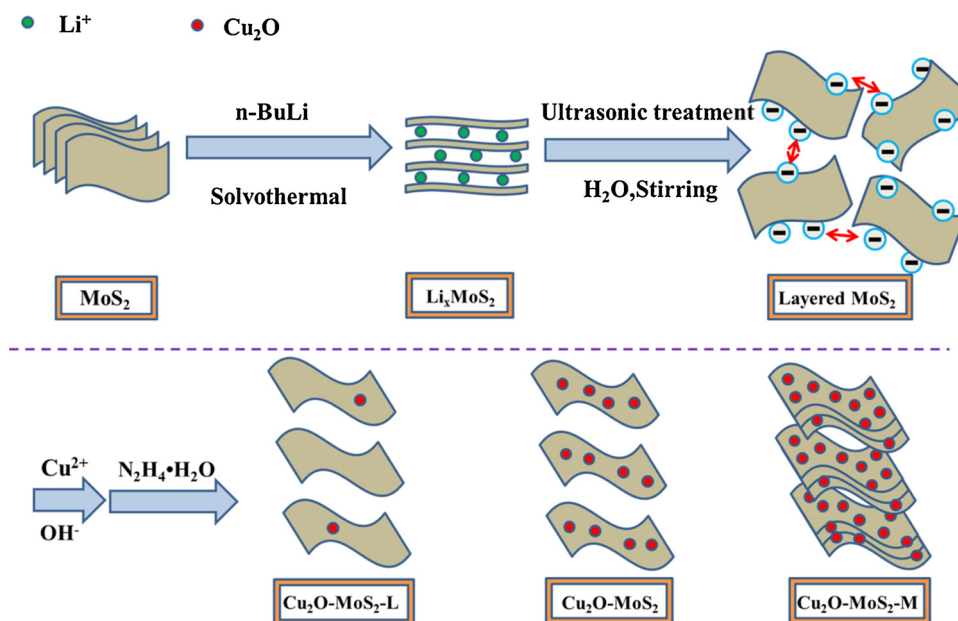
2. Experimental section

2.1. Raw materials

Molybdenum disulfide (MoS₂, AP), n-hexane (AP), copper acetate monohydrate (Cu(CH₃COO)₂·H₂O, AP), poly(N-vinylpyrrolidone) (PVP), hydrazine hydrate (85% aq.), absolute ethanol (C₂H₅OH) and triethanolamine (TEOA) were all obtained from Sinopharm Chemical Reagent Co., Ltd. N-Butyl lithium was supplied by Aladdin Industrial Corporation. Polyol LY-4110, polyaryl polymethylene isocyanate, dibutyltin dilaurate (LC), triethylenediamine (A33), and silicone surfactant were graciously provided by Jiangsu Luyuan New Materials Co., Ltd, China.

2.2. Preparation of LixMoS₂

LixMoS₂ was obtained using the hydrothermal route by adding 36 mL of a 0.5 M solution in hexane of n-butyl lithium to 1.0 g of raw MoS₂ in a teflon autoclave. Subsequently, the autoclave was heated at 100 °C for 4 h, followed by cooling naturally. The resulting products were rinsed with anhydrous hexane by filtration, followed by drying at 55 °C for 6 h in a vacuum chamber.



Scheme 1. Schematic presentation of the formation mechanism of the $\text{Cu}_2\text{O-MoS}_2$ hybrids.

Table 1

The formulae of pristine RPUF, RPUF/ MoS_2 , RPUF/ $\text{Cu}_2\text{O-MoS}_2$ and RPUF/ $\text{Cu}_2\text{O-MoS}_2\text{-M}$ composites.

Samples	RPUF-1	RPUF-2	RPUF-3	RPUF-4	RPUF-5
LY4110 (g)	100	100	100	100	100
A33 (g)	1	1	1	1	1
LC (g)	0.5	0.5	0.5	0.5	0.5
Water (g)	1	1	1	1	1
Si-Oil (g)	2	2	2	2	2
TEA (g)	3	3	3	3	3
MoS_2	0	2.45	0	0	0
$\text{Cu}_2\text{O-MoS}_2$	0	0	0	2.45	0
$\text{Cu}_2\text{O-MoS}_2\text{-M}$	0	0	0	0	2.45
PM-200 (g)	135	135	135	135	135

2.3. Preparation of $\text{Cu}_2\text{O-MoS}_2$ hybrids

The acquisition of ultrathin MoS_2 was based on a one-step ultrasonic and hydrolysis route of Li_xMoS_2 into deionized water at ambient conditions and $\text{Cu}_2\text{O-MoS}_2$ hybrids obtained by the very facile wet chemical treatment is shown in Scheme 1. In the first step, 0.2 g of Li_xMoS_2 was added into 500 mL of deionized water and ultrasonically agitated at ambient temperature for 5 h to acquire a colloidal suspension. After that, $\text{Cu}(\text{CH}_3\text{COO})_2\cdot\text{H}_2\text{O}$ (0.2 g) was mixed with 0.2 g of PVP aqueous solution (10 mL) and then added in the above suspension with 24 h

continuous agitation at room temperature. Subsequently, 0.03 g of hydrazine hydrate ($\text{N}_2\text{H}_4\cdot\text{H}_2\text{O}$) was dropped with constant agitation and ultrasonication. 20 min later, the black deposit was centrifuged and washed several times, dried in a chamber and stored under nitrogen. Other samples with different mass (i.e. 0.1 g and 0.4 g) from $\text{Cu}(\text{CH}_3\text{COO})_2\cdot\text{H}_2\text{O}$ were prepared using the similar procedure, which are labeled as $\text{Cu}_2\text{O-MoS}_2$, $\text{Cu}_2\text{O-MoS}_2\text{-L}$ and $\text{Cu}_2\text{O-MoS}_2\text{-M}$.

2.4. RPUF/ $\text{Cu}_2\text{O-MoS}_2$ composites fabrication

In a typical procedure, 0.78 g of $\text{Cu}_2\text{O-MoS}_2$ hybrid was added into 45 g of PAPI with agitation and ultrasonication to obtain a uniform suspension at ambient condition. Afterwards, the RPUF composite was prepared by a one-pot and free-rise treatment according to our previous work (Yuan et al., 2016). Briefly, all the basic material except above suspension was agitated vigorously in a beaker. Subsequently, the two mixtures were adequately intermixed for 12 s and then quickly transferred into a mould. In addition, Table 1 lists sample descriptions. Finally, the RPUF/ $\text{Cu}_2\text{O-MoS}_2$ composite was allowed to cure for 28 h. In addition, the same procedure was followed to prepare pristine RPUF, RPUF/ $\text{Cu}_2\text{O-MoS}_2$, RPUF/ $\text{Cu}_2\text{O-MoS}_2$ and RPUF/ $\text{Cu}_2\text{O-MoS}_2\text{-M}$ (-L) composites, while the nanofiller loading in all the samples was maintained at 1 wt%.

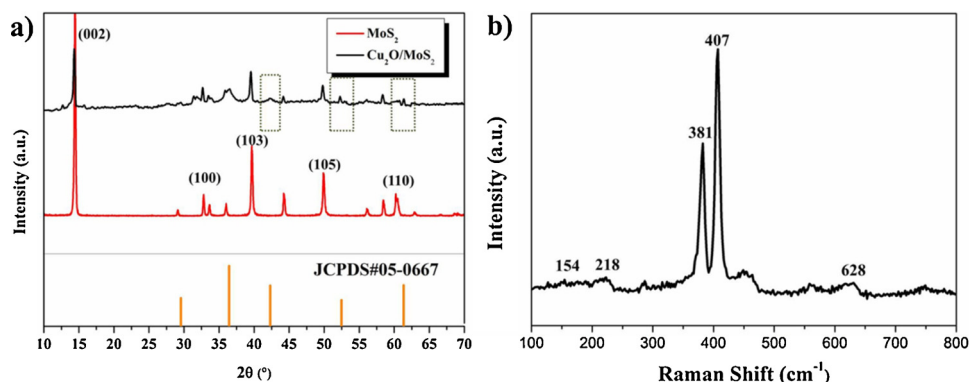


Fig. 1. XRD patterns of the MoS_2 and $\text{Cu}_2\text{O-MoS}_2$ hybrid (a) and Raman spectrum of $\text{Cu}_2\text{O-MoS}_2$ hybrid (b).

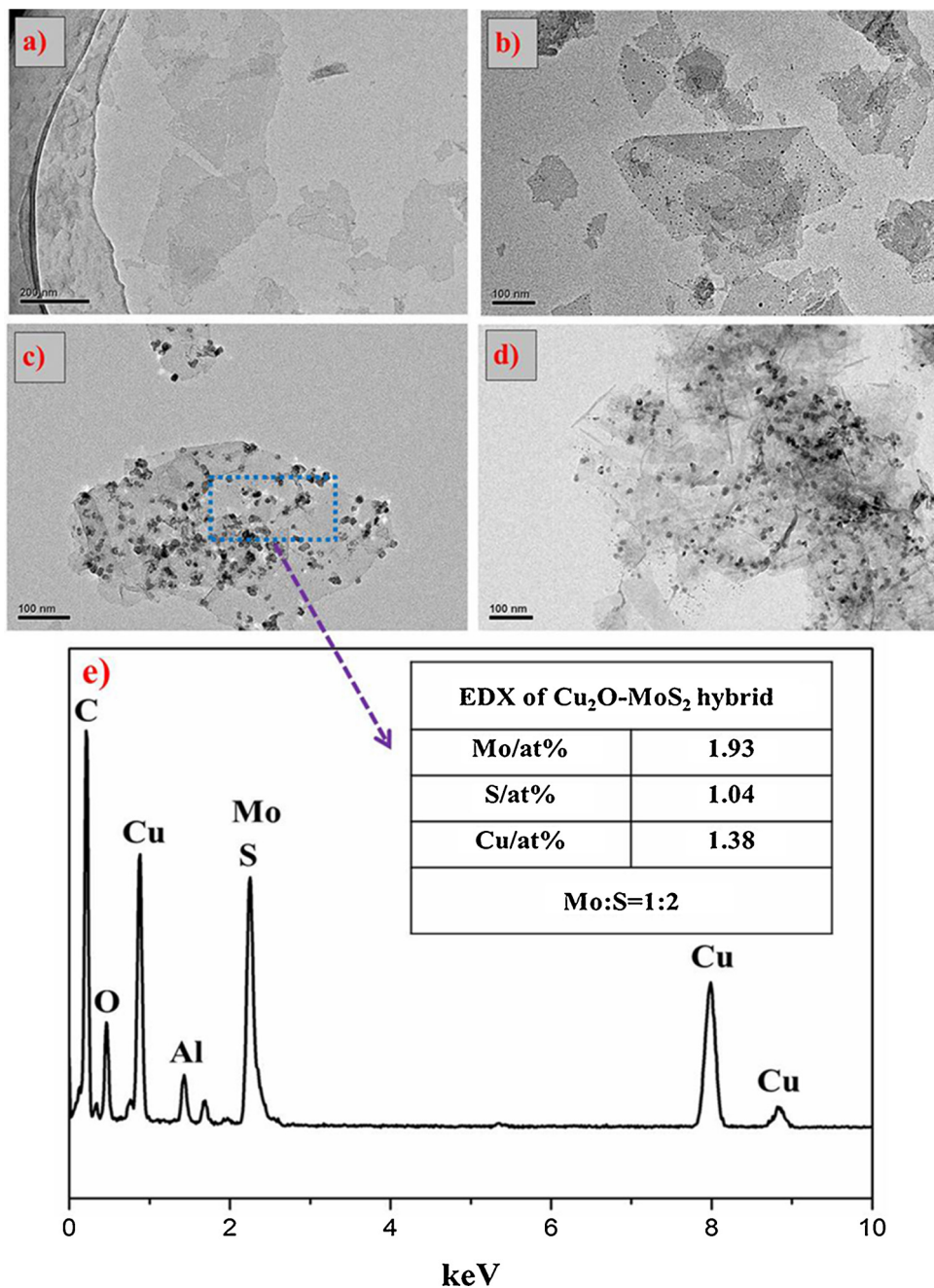


Fig. 2. TEM of the MoS₂ and Cu₂O-MoS₂ hybrids images: MoS₂ nanosheets (a), Cu₂O-MoS₂-L hybrid (b), Cu₂O-MoS₂ hybrid (c), Cu₂O-MoS₂-M hybrid (d) and EDX spectrum of Cu₂O-MoS₂ hybrid (e).

2.5. Characterization

Crystal-phase properties of the samples were studied by a powder X-ray diffractometer (Japan Rigaku D Max-Ra) using a rotating anode X-ray diffractometer accompanied with a Ni filtered Cu-K α tube ($\lambda = 1.54178 \text{ \AA}$) in the 2θ range from 10° to 70° with a sweep speed of 4 min^{-1} .

Element composition and morphology of Cu₂O-MoS₂ hybrid were observed by transmission electron microscopy (JEM-2100 F, Japan) with an acceleration voltage of 200 kV and energy dispersive X-ray spectroscopy.

Laser Raman spectra were measured from 100 to 2000 cm^{-1} on a SPEX-1403 laser Raman spectrometer (SPEX Co., USA) using a 514.5 nm argon laser.

The zeta potentials of the series of Cu₂O-MoS₂ composites were

recorded by a nanoparticle analyzer (Nano-ZS90, Malvern) at ambient temperature (Piao et al., 2014).

The morphologies of Cu₂O-MoS₂ hybrids in the polymer matrix were analyzed with transmission electron microscopy. The specimens were embedded in epoxy resin, and cut into thin films with a thickness of ca. 70 – 90 nm by ultramicrotomy at 25 °C.

The density of samples was tested according to ASTM D1622. The size of the specimen for the measurement was $30 \times 30 \times 30 \text{ mm}^3$, and the average density was received from at least five specimens. In addition, compressive properties of RPUF nanocomposites were characterized with a universal testing machine (Instron 1185) at room temperature. Values reported herein were the average of 5 tests.

Combustion properties of RPUF and its composites were performed on a microscale combustion calorimeter (MCC, GOVMARK) and thermal stability of the foams was performed by thermogravimetric

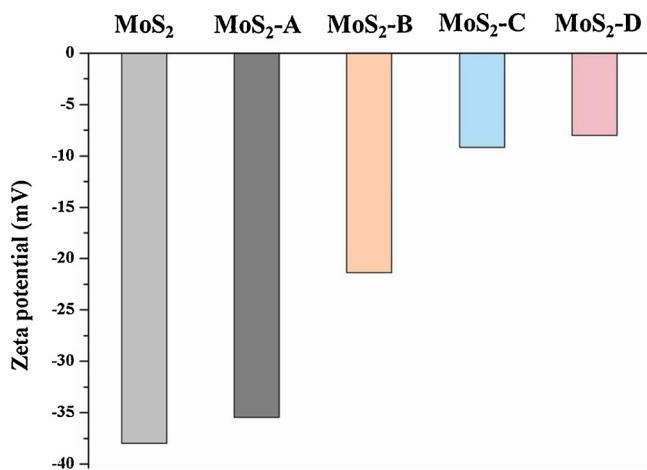


Fig. 3. The zeta potential of MoS₂, MoS₂-A, MoS₂-B, MoS₂-C and MoS₂-D hybrids when the weight ratio of MoS₂ and Cu(CH₃COO)₂·H₂O was 1:0, 1:1, 1:1.5, 1:2, 1:3.

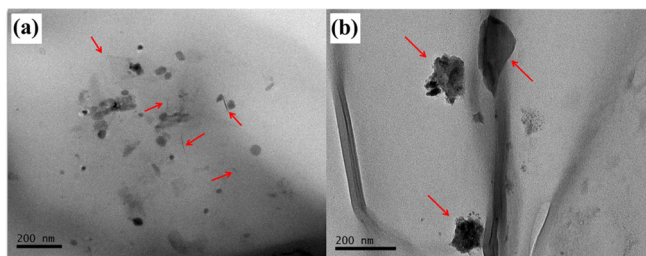


Fig. 4. TEM ultrathin observations of the RPUF/Cu₂O-MoS₂ composite (a) and RPUF/Cu₂O-MoS₂-M composite (b).

analysis (TGA) using a Q5000IR thermo-analyzer instrument at a linear heating rate of 20 °C min⁻¹ from 30 °C to 800 °C under nitrogen or air condition.

Cone calorimeter (Stanton Redcroft, UK) tests were performed according to ISO 5660 standard. Each specimen was wrapped with aluminum foils and exposed to an external heat flux of 35 kW/m² with the size of 100 × 100 × 3 mm³.

Scanning electron microscopy (Gemini500, ZEISS Co., German) was conducted to study the microstructures of the char residues of RPUF composites after cone calorimeter. The sample was coated with gold/palladium alloy.

Thermogravimetric analysis/infrared spectrometry (TG-IR) of RPUF and RPUF composites was carried out through a TGA Q5000 thermogravimetric analyzer, which used a stainless steel transfer pipe to combine with a Nicolet 6700 spectrophotometer. Thermal analyzer was conducted in the range from 25 to 500 °C with a heating rate of 20 °C/min under nitrogen condition.

Smoke toxicity was analyzed using a steady state tube furnace tests (SSTF) which was assessed on the basis of the ISO TS 19700. 20 g of samples in the form of granules were distributed evenly along the combustion quartz boat in the tube furnace with 825 °C at a constant rate. The air flow rate of 3.5 L min⁻¹ went through the tube furnace to promote combustion.

The quantitative measurement of the evolved gases like CO, CO₂, NO_x and HCN was conducted by the tubular furnace method. 0.5 g of samples loaded into the crucible are fed into the furnace with 800 °C for at least 18 min. The combustion gases were continuously extracted using a pump with 2 L min⁻¹. A series of tests have been performed. Subsequently, specific analyzers containing nondispersive infrared detectors for CO₂, CO, NO_x and HCN were located at the end of the line.

3. Results and discussion

3.1. Structural characterization and morphology

Scheme 1 displays the typical synthesis and possible mechanism of Cu₂O-MoS₂ hybrids. At first, the bulk MoS₂ is inserted with metal ions (Li⁺) by the solvothermal treatment, followed by exfoliation to form a single or few-layer nanosheet in deionized water through hydrolysis

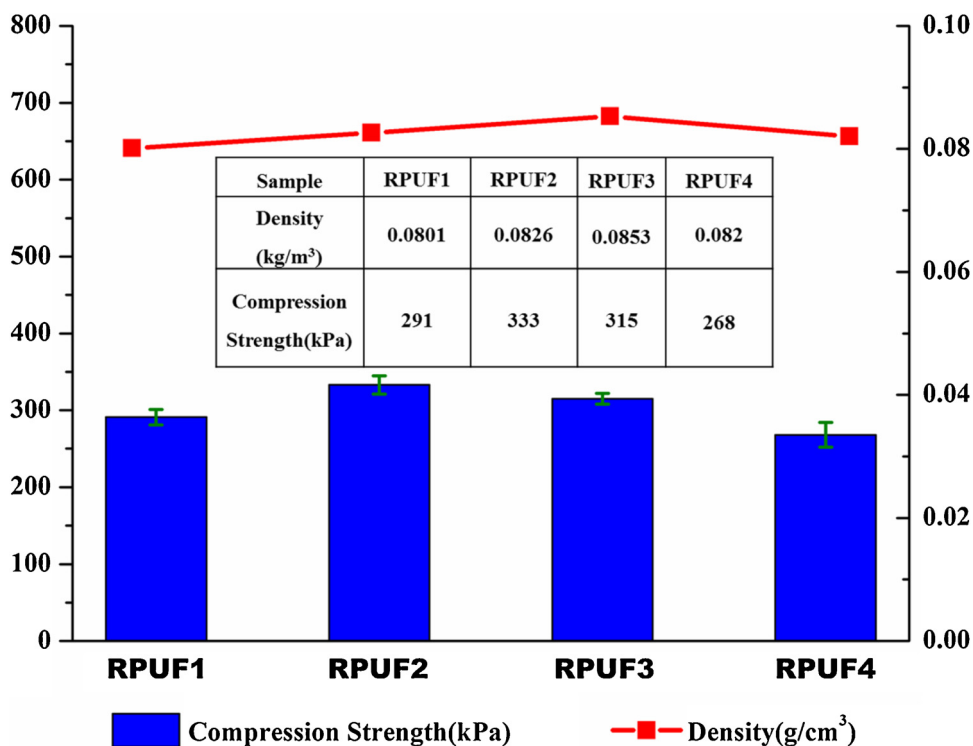


Fig. 5. Compressive strength and density of pristine RPUF, RPUF/MoS₂, RPUF/Cu₂O-MoS₂-L, RPUF/Cu₂O-MoS₂ and RPUF/Cu₂O-MoS₂-M composites.

Table 2
Compressive strengths of RPUF formulations and reported literatures.

Formulation	Δ Compressive strength (%)	Ref.
RPUF/HGM10-EG0-20 wt%	-70.68	<i>J. Appl. Polym. Sci.</i> 109 (2008) 1935-1943 (Bian et al., 2008)
RPUF/pEG-PMMA-10 wt%	-10.39	<i>Polym. Degrad. Stab.</i> 94 (2009) 971-979 (Ye et al., 2009)
RPUF/EG-10 wt%	-16%	<i>J. Appl. Polym. Sci.</i> 114 (2009) 853-863 (Meng et al., 2009)
RPUF/APP-10 wt%	-32%	<i>J. Appl. Polym. Sci.</i> 114 (2009) 853-863 (Meng et al., 2009)
PU/PFAPP/PL20 wt%	8.15%	<i>J. Polym. Res.</i> 20 (2013) 234 (Xing et al., 2013)
RPUF/CNTs-0.3 wt%	16%	<i>Polym. Int.</i> 61 (2012) 1107-1114 (Yan et al., 2012)
RPUF/Cu ₂ O-2 wt%	11.8%	<i>Composites Part A</i> 112 (2018) 142-154 (Yuan et al., 2018b)
RPUF/MoO ₃ -2 wt%	11.7%	<i>Composites Part A</i> 112 (2018) 142-154 (Yuan et al., 2018b)
RPUF/Cu ₂ O-MoS ₂ -1 wt%	8.3%	<i>This work</i>

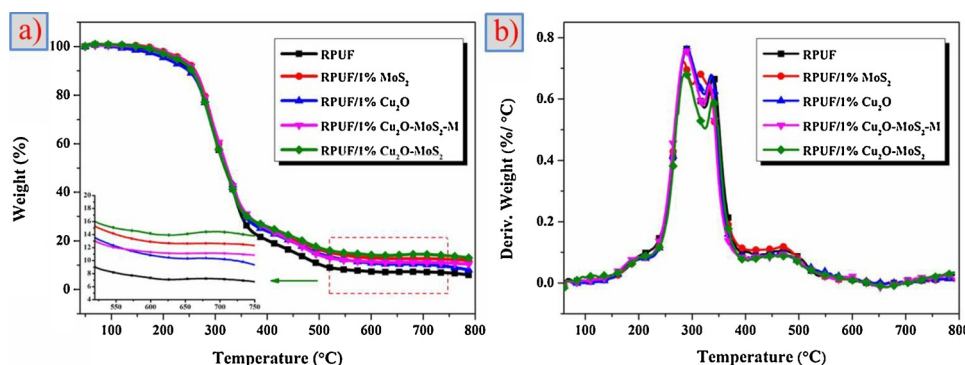


Fig. 6. TG/DTG profiles of pristine RPUF, RPUF/MoS₂, RPUF/Cu₂O, RPUF/Cu₂O-MoS₂-M and RPUF/Cu₂O-MoS₂ composites under nitrogen atmosphere.

Table 3
TGA results of pristine RPUF, RPUF/MoS₂, RPUF/Cu₂O-MoS₂ and RPUF/Cu₂O-MoS₂-M composites under nitrogen atmosphere.

Samples	T _{-5%} (°C)	T _{-50%} (°C)	Residue (wt%)
RPUF	218	318	5.85
RPUF/MoS ₂	230	322	10.32
RPUF/Cu ₂ O-MoS ₂ -M	205	320	7.85
RPUF/Cu ₂ O-MoS ₂	236	323	11.82

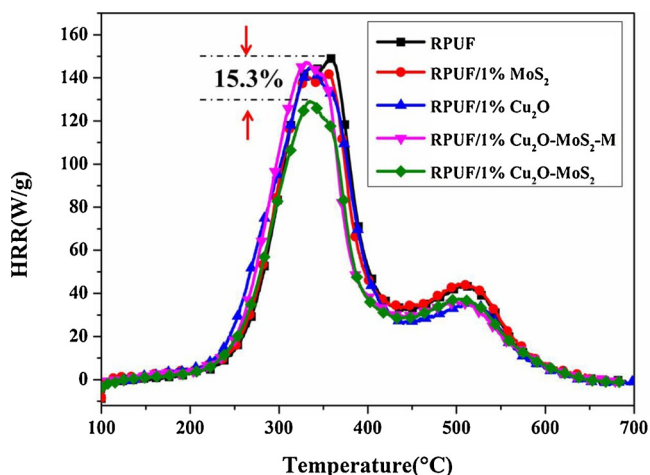


Fig. 7. The heat release rate (HRR) curves of pristine RPUF, RPUF/MoS₂, RPUF/Cu₂O, RPUF/Cu₂O-MoS₂ and RPUF/Cu₂O-MoS₂-M composites.

reaction. Then, a series of Cu₂O-MoS₂ hybrids with strong interfacial interaction are well obtained by a very facile wet chemical method. Specifically, the Cu²⁺ ions are attached to the surface of MoS₂ nanosheets by electrostatic attraction and the Cu(OH)₂ nanoparticulates will be transformed with the introduction of N₂H₄·H₂O. Thus the self-assembly of Cu₂O-MoS₂ hybrids are acquired.

Fig. 1 displays the X-ray diffraction (XRD) patterns of the Cu₂O-MoS₂ hybrids and the re-stacked MoS₂. In the XRD pattern of Cu₂O, five narrow peaks appeared at 2 θ of 29.5, 36.3, 42.3, 61.3 and 73.5° with lattice distance $a = 4.269 \text{ \AA}$ (JCPDS file no. #05-0667) corresponding to the (110), (111), (200), (220) and (311) reflections, respectively. Compared with pristine MoS₂, the peaks indexed at 2 $\theta = 36.3^\circ$, 42.3°, and 61.3° are observed for Cu₂O-MoS₂ hybrids, which were attributed to typical diffraction peaks of Cu₂O. For as-prepared Cu₂O-MoS₂ hybrids, the characteristic peaks at 14.1, 32.5, 40.0, and 59.0° are consistent with (002), (100), (103), and (110) crystal planes of the MoS₂ with high crystallinity, indicating the formation of Cu₂O crystalline phase on the surface of MoS₂. Moreover, the intensity of the diffraction peak for (002) (MoS₂) in the Cu₂O-MoS₂ hybrid decreases, indicating that the face-to-face stacking is broken due to the deposition of Cu₂O on both sides of MoS₂ nanosheets.

In order to further illustrate the phase of the Cu₂O-MoS₂ hybrid, Raman spectrum of Cu₂O-MoS₂ hybrid is also analyzed in detail, as shown in Fig. 1b. The characteristic peak at 381 cm⁻¹ is associated with the in-plane E_{2g} vibration modes of two sulfur atoms with the molybdenum atom, while the peak at 407 cm⁻¹ is attributed to the out-of-plane A_{1g} vibration modes of sulfur atoms (Lee et al., 2010; Fan et al., 2015; Coleman et al., 2011). Notably, the Raman shifts of Cu₂O at 218, 154 and 628 cm⁻¹ appear in accordance with a second-order overtone and two infrared-allowed modes respectively after the loading of Cu₂O on MoS₂ nanosheets (Powell et al., 1975; Li et al., 2017). Meanwhile, the vibrational fingerprint at 218 cm⁻¹ of Cu₂O-MoS₂ sample should be assigned to T_{2u} and Raman scattering features from Cu₂O (Fan et al., 2016; Hardcastle, 2011; Prabhakaran and Murugan, 2014; Ohsaka et al., 1978).

TEM was applied to reveal the morphology and structure information of MoS₂ and Cu₂O-MoS₂. As depicted in Fig. 2a, two-dimensional MoS₂ presents a typical platelet shape with several hundred nanometers large, which certifies that the MoS₂ is fully exfoliated into thin layers. As for the Cu₂O-MoS₂-L (less loading of Cu₂O) hybrid, only a tiny fraction of Cu₂O nanoparticles are anchored on the surface of MoS₂ nanosheets (Fig. 2b). It is noteworthy that the Cu₂O-MoS₂-L hybrid displays an exfoliated layer with a fairly low loading of Cu₂O. From

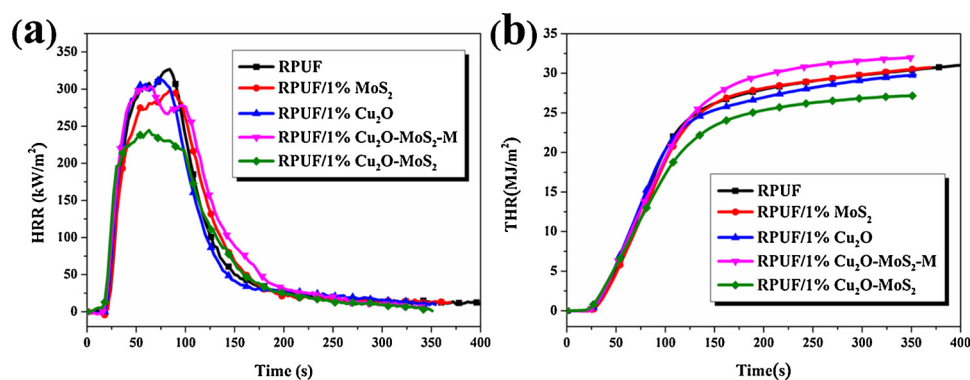


Fig. 8. Heat release rate (a) and total heat release (b) versus time curves of pristine RPUF and its composites.

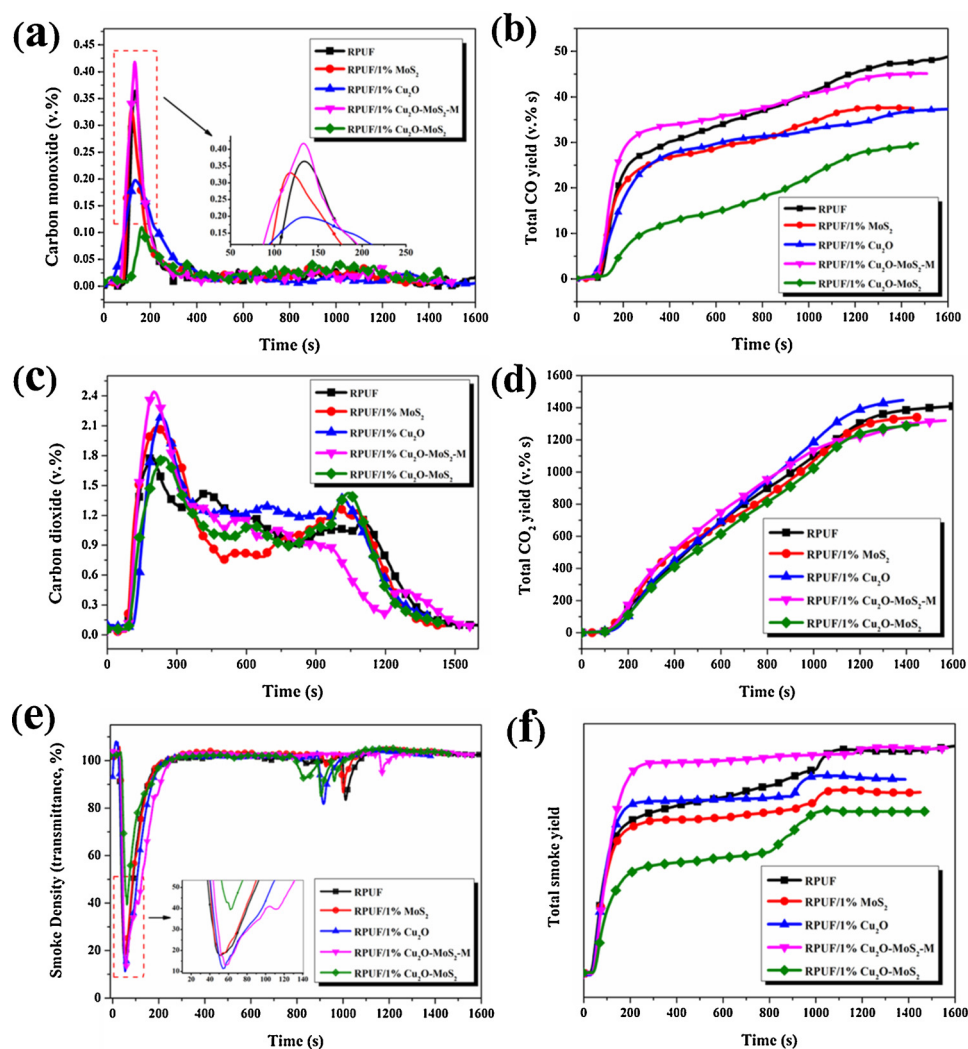


Fig. 9. CO curves (a, b), CO₂ curves (c, d), and total smoke density (e, f) of pristine RPUF and its composites obtained from SSTF tests.

Fig. 2c, the Cu₂O nanoparticles are uniformly and firmly decorated on the surface of the MoS₂ nanosheets. Compact lamellar structures and no aggregations of the MoS₂ nanolayers can be seen for the Cu₂O-MoS₂ hybrids, demonstrating that the Cu₂O nanoparticles are anchored stiffly on the surface of MoS₂ nanosheets. As seen in Fig. 2d, the morphology of the Cu₂O-MoS₂-M hybrid displays abundant layers tangle. The re-stacking of MoS₂ nanosheets is probably attributed to the presence of vast numbers of positive charges by Cu²⁺ ions precipitating on the surface of MoS₂ nanosheets during preparation. For further

confirmation of the Cu₂O-MoS₂ hybrids, Fig. 2e displays the elementary composition of the selected fields by energy dispersive X-ray spectroscopy (EDX). It is noticeable that the Mo, S, Cu and O elements existed in Cu₂O-MoS₂ hybrid. Notably, the C and Al peaks are derived from the aluminum grids (Cherstiouk et al., 2016). Hence, the emergence of O and Cu is due to the Cu₂O nanoparticles attached to the surface of the MoS₂ nanosheets. The atomic ratios of Mo, S and Cu are inserted in Fig. 2e and the EDX spectrum further indicates that the Cu₂O-MoS₂ hybrid was successfully prepared.

Table 4

Quantitative analysis of gaseous products of degradation of pristine RPUF, RPUF/MoS₂, RPUF/Cu₂O-MoS₂ and RPUF/Cu₂O-MoS₂-M composites by tubular furnace method at 825 °C.

Samples	Products					N-Gas value Eqn.(1)
	HCN (ppm)	NO _x (ppm)	CO (ppm)	CO ₂ (ppm)	O ₂ (ppm)	
RPUF	325	30	5000	115000	18.3	1.079
RPUF/MoS ₂	335	24	4900	100000	18.6	1.095
RPUF/Cu ₂ O-MoS ₂ -M	320	34	5200	105000	18.7	1.154
RPUF/Cu ₂ O-MoS ₂	275	14	3600	120000	17.9	0.787

To ascertain the TEM results, the zeta potential of a series of Cu₂O-MoS₂ hybrids with the same concentration was measured. Generally, the zeta potential is an important parameter for characterizing the stability of colloidal dispersions and provides a measure of the magnitude and sign of the effective surface charge associated with the double layer around the colloid particle (Gupta et al., 2015). Generally, particles with zeta potentials more positive than +30 mV or more negative than -30 mV are considered to form stable dispersions due to interparticle electrostatic repulsion. Fig. 3a exhibits the pristine MoS₂ nanosheets achieve a fairly high negative charge (-37.5 ± 2.5 mV) on the surface. Nevertheless, the surface charge begins to go up and reaches -21.3 mV with the loading of appropriate Cu²⁺ ions. More importantly, the addition of Cu²⁺ ions with a high loading can severely decrease the negative potential from 21.3 to 8.9 mV. In addition, significant face-to-face stacking is broken due to the reduced van der Waals forces by the introduction of Cu₂O nanoparticles. Consequently, the self-assembly of the Cu₂O-MoS₂ hybrid is unfavorable due to the less electrostatic repulsion of interlamination of MoS₂ nanosheets (Gupta et al., 2015).

3.2. Mechanical properties of RPUF composites

3.2.1. Dispersion state of Cu₂O-MoS₂ hybrids in RPUF composites

The interfacial interaction and dispersion of nanofillers in polymers

reveal a key factor to influence the performance of polymeric nanocomposites (Bao et al., 2011; Yuan et al., 2018a). To assess the dispersion states of different Cu₂O-MoS₂ hybrids in the RPUF nanocomposites, the morphologies of fractured surfaces were analyzed using TEM, as portrayed in Fig. 4. The Cu₂O-MoS₂ hybrids are finely dispersed and embedded in the RPUF without aggregation under higher magnification (Fig. 4a). By comparison, after the incorporation of Cu₂O-MoS₂-M hybrids, the hybrids agglomerate severely in the RPUF nanocomposites, which are probably attributed to a forceful van der Waals and intralayer covalent bonding with the presence of vast numbers of positive charges promoting the restacking of MoS₂ nanolayers in the RPUF nanocomposites during preparation.

3.2.2. Foam density and compressive strength

Basically, the compressive strength of the RPUF composites will be enhanced if the nanoadditives are finely distributed in the RPUF substrate because of the superior interaction between the nanoadditives and polymers. As portrayed in Fig. 5, a distinct improvement in compressive strength is obtained with the introduction of the Cu₂O-MoS₂ hybrid. Compared with the data from Table 2, the compressive strength increases effectively, which is possibly due to the highly dispersed Cu₂O-anchored MoS₂ have an enormous number of hydroxyl groups to form the crosslinking system with favorable compatibility between the nanoparticles and RPUF matrix (Huang et al., 2013; Yuan et al., 2018b), which promote its excellent dispersion in the matrix, improving the mechanical property between the Cu₂O-MoS₂ hybrid and RPUF matrix. The sample with 1% concentration of Cu₂O-MoS₂-M hybrid shows a decrease of 7.9% in the compressive strength, which should be attributed to the poor compatibility between the Cu₂O-MoS₂-M hybrid and RPUF matrix (Wang et al., 2016). Incorporating Cu₂O-MoS₂ hybrid into the RPUF matrix leads to an increase of 8.3% in the compressive strength, achieving the purpose of reinforcement for the RPUF matrix (Thirumal et al., 2007). Compared with the data from Table 2, the compressive strength increases effectively, which is possibly due to the highly dispersed Cu₂O-anchored MoS₂ hybrids have hydroxyl group can react with the R-NCO groups of isocyanurate to form the crosslinking system with favorable compatibility between the particles and the matrix

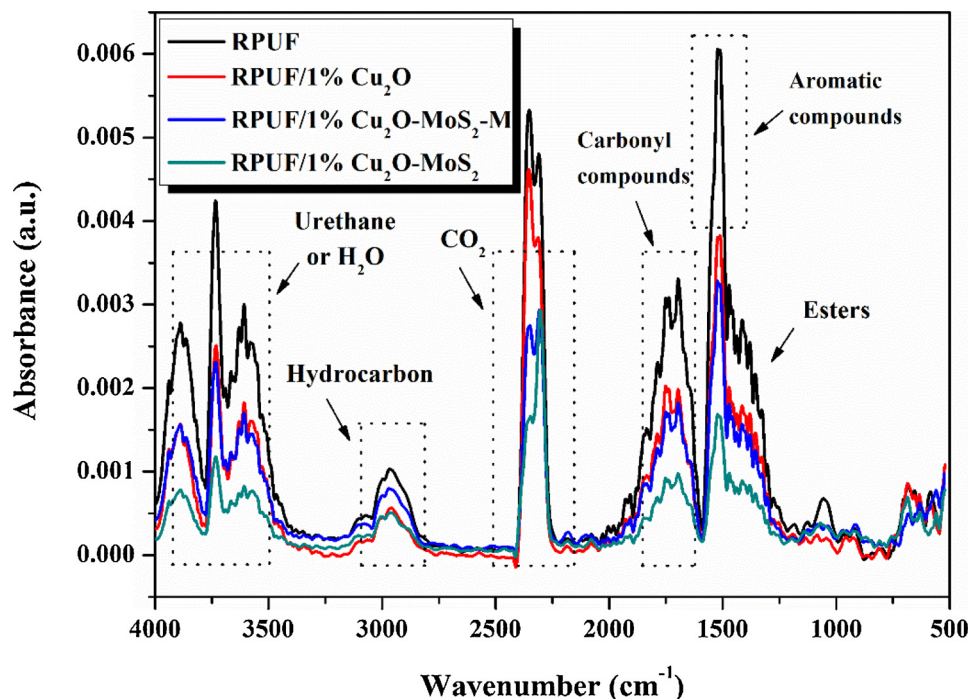


Fig. 10. TG-IR spectra of pyrolysis products of pristine RPUF, RPUF/Cu₂O, RPUF/Cu₂O-MoS₂ and RPUF/Cu₂O-MoS₂-M composites at maximum decomposition rate.

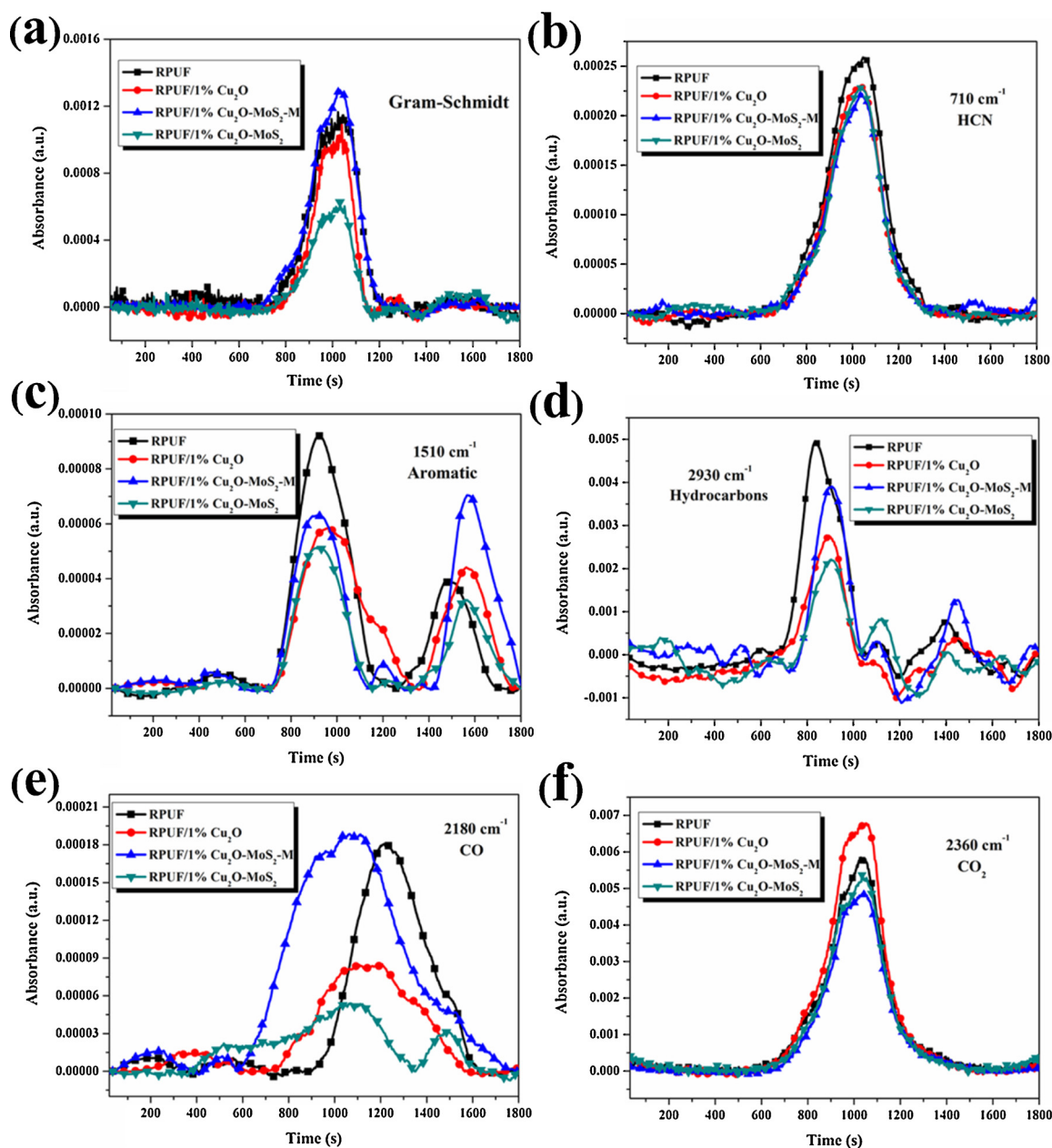


Fig. 11. Gram-Schmidt curves (a) and Intensity of characteristic peaks for pyrolysis products of pristine RPUF, RPUF/Cu₂O, RPUF/Cu₂O-MoS₂ and RPUF/Cu₂O-MoS₂-M composites at maximum decomposition rate (b)-(f).

3.3. Thermal property

The influence of Cu₂O-MoS₂ hybrid on the thermal stability property of RPUF was analyzed by TGA. The TGA curves of RPUF/MoS₂, RPUF/Cu₂O, RPUF/Cu₂O-MoS₂ and RPUF/Cu₂O-MoS₂-M composites in anaerobe conditions are described in Fig. 6. In addition, the typical thermal parameters were presented in Table 3. It is noticeable that the pyrolysis process under anaerobe atmosphere includes two-stage steps referred to the DTG profile (Camino et al., 1984). As can be observed from Fig. 6a, the first step occurs in the ranges of 223–422 °C, which derives from the degradation of hard-segment (Chattopadhyay and Webster, 2009). The weak C–NH bond brings about the generation of isocyanate primary or secondary amine and alcohol (Yuan et al., 2016). The subsequent step is ascribed to the decomposition of the soft-segment which contains a stable urea structure (Modesti et al., 2008). Interestingly, a 18 °C increment and a 13 °C decrement in T_{5%} are observed for RPUF/Cu₂O-MoS₂ and RPUF/Cu₂O-MoS₂-M respectively,

rivalled by pristine RPUF. It could be caused by the restacking of MoS₂ for the Cu₂O-MoS₂-M hybrid (Bissessur et al., 1993). The TG results present that the residues of RPUF/MoS₂, RPUF/Cu₂O-MoS₂ and RPUF/Cu₂O-MoS₂-M composites are much more than that of pristine RPUF (5.85 wt%), which are 10.32 wt%, 7.85 wt% and 11.82 wt%, respectively, revealing that the Cu₂O-MoS₂ possess impressive catalytic charring behavior for RPUF. Additionally, as seen from DTG curves, the maximum mass loss rate of the RPUF/Cu₂O-MoS₂ is the lowest, indicating that the barrier effect and catalytic charring of Cu₂O-MoS₂ hybrid plays an important role in enhancing thermal stability significantly (Cao et al., 2010).

3.4. Fire hazards

3.4.1. Flammability of RPUF composites

MCC is a valid laboratory test to evaluate RPUF and its composites for their fire-retardant property. The heat release rate curves of pristine

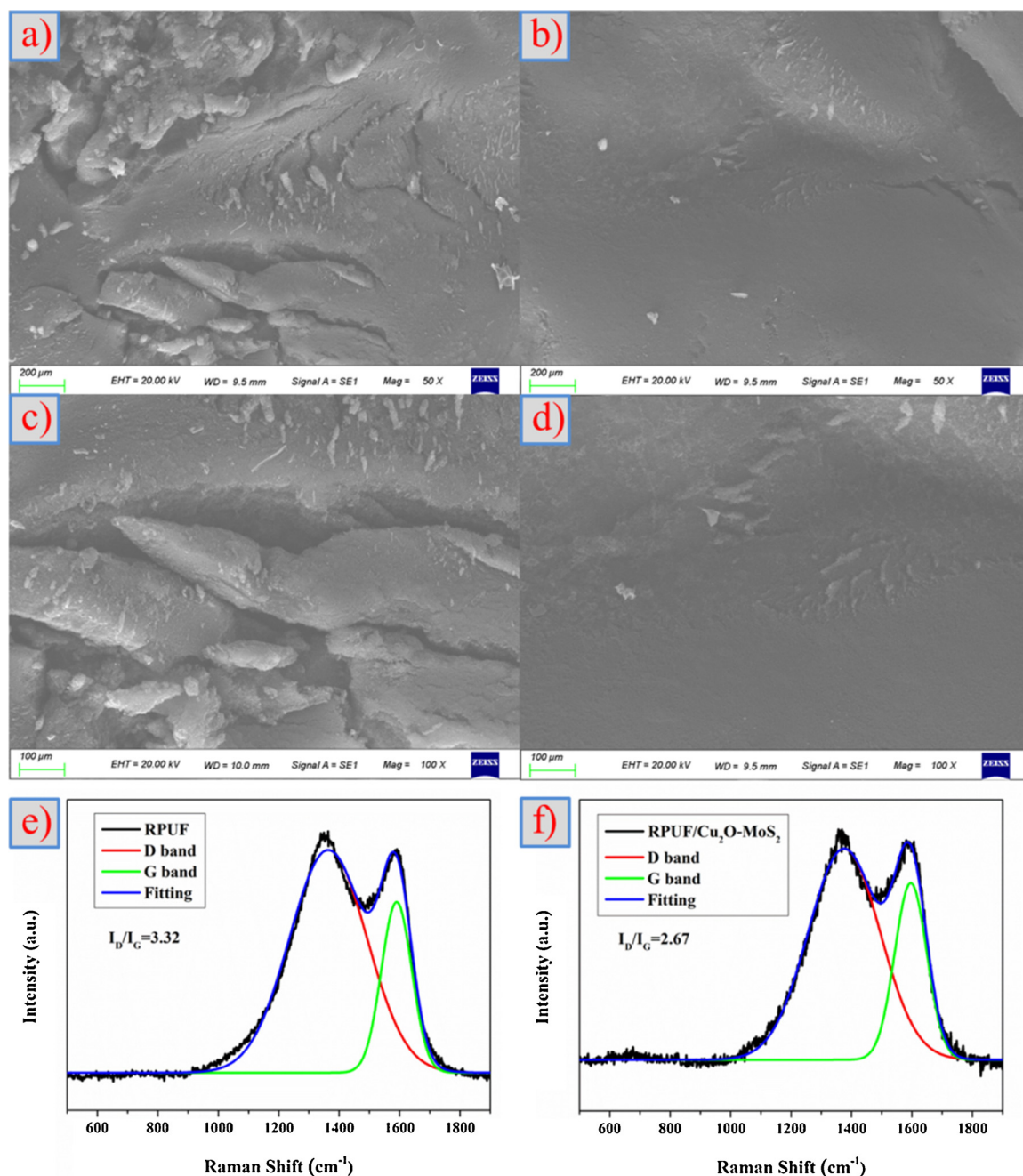


Fig. 12. SEM micrographs (a–d) and Raman spectra (e–f) of the surface residues from pristine RPUF (a, c) and RPUF/Cu₂O-MoS₂ (b, d) after cone calorimeter tests.

RPUF, RPUF/MoS₂, RPUF/Cu₂O, RPUF/Cu₂O-MoS₂ and RPUF/Cu₂O-MoS₂-M composites are portrayed in Fig. 7. A decrease of 15.3% in the peak heat release rate (pHRR) for RPUF/Cu₂O-MoS₂ composite is obtained, compared to pristine RPUF, which is ascribed to the layered barrier effect and catalytic carbonization of the Cu₂O-MoS₂ hybrids. Unfortunately, only slight decrease in pHRR is observed when the Cu₂O-MoS₂-M hybrids are incorporated into RPUF. These results certify that the thin layer structure of MoS₂ nanosheets have significant effect on decreasing the pHRR.

To evaluate the influence of incorporation of Cu₂O-anchored MoS₂ hybrids on the combustion behaviors of RPUF composites, HRR and THR curves of pristine RPUF and its composites are plotted in Fig. 8. It is noticeable that the addition of pristine MoS₂ and Cu₂O (1 wt%) gives rise to a 9.8% and 3.6% reduction in pHRR. Furthermore, incorporating 1 wt% Cu₂O-MoS₂ into RPUF results in the lowest peak HRR value

(26.3% reduction). Also, the addition of 1 wt% Cu₂O-MoS₂ gives rise to the lowest total heat release (THR) and highest char yield. The mechanism of Cu₂O-MoS₂ in reducing the flammability of RPUF is probably attributed to the barrier effect and charring effect which could obtain a protective char and slow down the heat and mass transfer.

3.4.2. Smoke toxicity analysis by SSTF

To better realize the roles of the MoS₂, Cu₂O, Cu₂O-MoS₂ and Cu₂O-MoS₂-M on decreasing the smoke toxicity, the steady state tube furnace test were performed on pristine RPUF and its nanocomposites. It is well known that rigid polyurethane foam is easily combustible and produces a great deal of smoke and toxic gases. Accordingly, the reduction of the amount of toxic gases and smoke has significant influence on saving lives during combustion process. Fig. 9 exhibits the curves of CO, CO₂ and smoke yield as a function of time for the RPUF nanocomposites

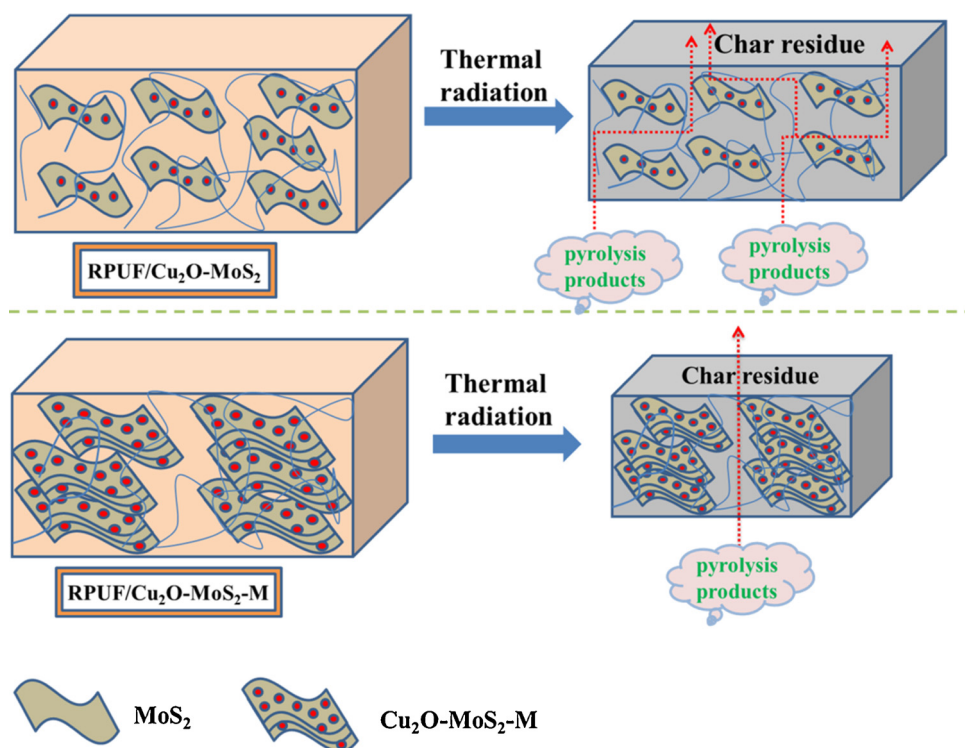


Fig. 13. Schematic illustration for the mechanism of catalytic oxidation of the $\text{Cu}_2\text{O-MoS}_2$ hybrids in the thermal degradation process of the RPUF composite.

derived from SSTF results. As seen in Fig. 9, RPUF/ MoS_2 depicts mildly decreased CO yields and smoke density compared with pristine RPUF. It is attributed to the physical adsorption effect of the thin layer structure of MoS_2 nanosheets with huge specific surface area (Wang et al., 2014b). CO concentration for RPUF/ Cu_2O is reduced, while the CO_2 yields have been increased for the nanocomposites as compared to that of pristine RPUF, indicating that the catalytic oxidation of Cu_2O is certified under air atmosphere. Specifically, the CO yields and smoke during combustion for the sample with $\text{Cu}_2\text{O-MoS}_2$ hybrid are dramatically decreased compared to pristine RPUF. The reduction should be derived from the synergistic effect between the adsorption of MoS_2 and the restricted mobility in the polymer chains by the restriction of Cu_2O (Laachachi et al., 2007, 2005). MoS_2 is apt to absorb small gaseous molecules, thus retarding the escape of the organic volatiles to form smoke, owing to the physical adsorption effect. Meanwhile, the Cu_2O nanoparticles catalyze the oxidation of organic volatiles, thereby decreasing the fire hazards.

3.4.3. Volatile gases analysis by tubular furnace test

To investigate the adsorption and catalytic oxidation behavior of $\text{Cu}_2\text{O-MoS}_2$ hybrid to reduce the fire hazards of RPUF nanocomposites, the amounts of toxic gases were gathered to measure the composition from the tubular furnace experiments. It can be seen from Table 4, with the addition of $\text{Cu}_2\text{O-MoS}_2$ hybrids into RPUF, the acquisition of toxic gases during thermal degradation in open porcelain pans set in a tube furnace is decreased substantially. The introduction of $\text{Cu}_2\text{O-MoS}_2$ hybrids retards the evolution of principal toxicants (CO and NO_x) significantly compared to other samples which is considered as an important finding. The lower yield of principal toxicants and the higher yield of CO_2 indicate that $\text{Cu}_2\text{O-MoS}_2$ is an efficient catalyst which can accelerate the reaction of CO and NO_x to produce desired harmless gases during decomposition of RPUF (Shi et al., 2015).

The toxicants were predicted by NIST according to the N-Gas model in Eqn. (1). This model expresses the ratio of the concentration of each toxicant to its lethal concentration, and then multiplies the sum of this ratio by the hyperventilation factor (Stec and Hull, 2011). Generally,

animal deaths will start to occur when the N-Gas value is above 0.8 and 100% of the animals will die when the value is above 1.3 (Liu et al., 2016). The results of this experiment according to Eqn. (1) are presented in Table 4, which indicate that the incorporation of $\text{Cu}_2\text{O-MoS}_2$ hybrids for reducing the fire hazards is significantly enhanced.

3.4.4. Evolved gases analysis by TG-IR

TG-IR was employed to analyze the evolved volatiles of RPUF and RPUF nanocomposites during thermal decomposition process and explore the mechanism of smoke toxicity suppression. The FTIR spectra of pyrolysis products at maximum degradation rate during the pyrolysis of RPUF, RPUF/ Cu_2O , RPUF/ $\text{Cu}_2\text{O-MoS}_2$ and RPUF/ $\text{Cu}_2\text{O-MoS}_2\text{-M}$ nanocomposites are presented in Fig. 10. Peaks in the range of $3850\text{--}3550\text{ cm}^{-1}$ are assigned to the vibrations of O-H stretch bond of water or N-H stretch bond in urethane. It is obvious that the emergence of vibrations at 1238, 1505, 2185, 2358 and 2920 cm^{-1} derive from esters, aromatic compounds, CO, CO_2 and hydrocarbons.

To further understand the changes of the pyrolysis products, the relative intensity of the pyrolysis gases of pristine RPUF and its nanocomposites are depicted in Fig. 11. From Gram-Schmidt curve, it is clearly observed that the $\text{Cu}_2\text{O-MoS}_2$ hybrids into RPUF can change the thermal degradation process and bring about a depressed emission of pyrolysis products (Cao et al., 2010). Meanwhile, the dramatical reduction of fire hazards and toxic gases for RPUF/ $\text{Cu}_2\text{O-MoS}_2$ nanocomposite lead to a decrement in the fire hazard, which will reduce the smoke toxicity of RPUF and be beneficial for fire rescue.

3.5. Char residue analysis

Fig. 12 presents the SEM images of the char residues for pristine RPUF and RPUF/ $\text{Cu}_2\text{O-MoS}_2$ after cone calorimeter tests. It can be depicted from Fig. 12(a-d), the char residue of pristine RPUF exhibits relatively loose and fragile with many cracks from Fig. 12(a-d), which cannot act as a shielding effect to protect the polymer matrix. In contrast, RPUF/ $\text{Cu}_2\text{O-MoS}_2$ nanocomposites obtain compact char residues during combustion, which can hinder the flammable gases and afford

better shielding action. To investigate the condensed-phased products of RPUF nanocomposites after combustion, Fig. 12 shows the Raman spectra of the surface residues of RPUF and RPUF/Cu₂O-MoS₂ nanocomposites. The spectra display similar shape overlapping peaks at approximately 1595 and 1360 cm⁻¹, which belongs to G band and D band (Fang et al., 2010). The intensity ratio of the G to D band (I_G/I_D) of RPUF/Cu₂O-MoS₂ is higher than that of pristine RPUF, suggesting that the introduction of Cu₂O-MoS₂ hybrids could form compact graphitic char residues, which act as a useful barrier and protect underlying material.

On the basis of the above-mentioned results, the mechanism for the improvement of thermal stability and the reduction of smoke toxicity of RPUF/Cu₂O-MoS₂ nanocomposite is illustrated in Fig. 13 and proposed as follows: (i) the physical adsorption effect of MoS₂ and the restricted mobility of the polymer chains by the restriction effect of Cu₂O is responsible for enhancing the thermal stability; (ii) the synergistic effect between the physical adsorption of MoS₂ and the catalysis action of Cu₂O play a synergistic role in decreasing the fire hazards. Additionally, the synergistic effect of RPUF/Cu₂O-MoS₂-M nanocomposite is inferior to those of RPUF/Cu₂O-MoS₂ nanocomposite, which provides direct proofs of the negative influence of the stacked MoS₂ on reducing the smoke toxicity for RPUF nanocomposites.

4. Conclusions

In the present study, layered MoS₂ decorated with Cu₂O nanoparticles has been prepared to decrease smoke toxicity for RPUF nanocomposites. The results indicate that the activity of the layered MoS₂ for reducing the fire hazards is markedly improved with the addition of the Cu₂O cocatalyst. The structural and morphology characterization illustrate that the surface of layer-structured material was decorated with Cu₂O nanoparticle successfully. Moreover, the results suggest that the Cu₂O-MoS₂ hybrid effectively prevented the MoS₂ nanosheets from restacking but Cu₂O-MoS₂-M hybrid displayed a characteristic stacked layer structure. TGA investigation exhibited that the thermal stability of RPUF is markedly improved by adding Cu₂O-MoS₂ hybrid at a low loading (1 wt%) and the physical adsorption and catalytic carbonization of Cu₂O-MoS₂ hybrids are believed to be important factors. Furthermore, catalytic oxidation of Cu₂O-MoS₂ hybrids on thermal decomposition and generation of toxic gases of RPUF was studied by TG-IR and calculated by N-Gas model. It was found that the amounts of the toxic organic volatiles were decreased owing to the synergistic effect between the physical adsorption of MoS₂ and the catalysis action of Cu₂O. The work depicts that the development of noble-metal-free layer-structured composites, such as Cu₂O-MoS₂ hybrid, containing an inexpensive and environmentally benign Cu₂O cocatalyst, is feasible and has great potential for reducing smoke toxicity of fire safety RPUF composites.

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