

Open Archive Toulouse Archive Ouverte (OATAO)

OATAO is an open access repository that collects the work of Toulouse researchers and makes it freely available over the web where possible.

This is an author-deposited version published in: <u>http://oatao.univ-toulouse.fr/</u> Eprints ID: 11743

> Identification number: DOI: 10.1016/j.jpowsour.2013.08.049 Official URL: http://dx.doi.org/10.1016/j.jpowsour.2013.08.049

To cite this version:

El Hannach, Mohamed and Soboleva, Tatyana and Malek, Kourosh and Franco, Alejandro A. and Prat, Marc and Pauchet, Joel and Holdcroft, Steven *Characterization of pore network structure in catalyst layers of polymer electrolyte fuel cells.* (2014) Journal of Power Sources, vol. 247 (n° 1). pp. 322-326. ISSN 0378-7753

Any correspondence concerning this service should be sent to the repository administrator: staff-oatao@inp-toulouse.fr

Characterization of pore network structure in catalyst layers of polymer electrolyte fuel cells

Mohamed El Hannach^{a,*}, Tatyana Soboleva^b, Kourosh Malek^{b,c}, Alejandro A. Franco^{a,1,2}, Marc Prat^{d,e}, Joël Pauchet^a, Steven Holdcroft^{b,c}

^a CEA, LITEN, LCPEM, 17 rue des Martyrs, F-38054 Grenoble, France

^b Department of Chemistry, Simon Fraser University, Burnaby, BC, Canada V5A 1S6

^c National Research Council of Canada, Institute for Fuel Cell Innovation, 4250 Wesbrook Mall, BC, V6T 1W5 Vancouver, Canada

^d University of Toulouse, INPT, UPS, IMFT, Avenue Camille Soula, 31400 Toulouse, France

^e CNRS, IMFT, 31400 Toulouse, France

HIGHLIGHTS

Keywords:

Catalyst laver

Pore network modeling

Polymer electrolyte fuel cell

Nitrogen adsorption isotherm Pore size distribution

- We build a comprehensive model of nitrogen adsorption isotherm.
- The results confirm the existence of different sorption regimes.
- We validate the representation of a catalyst layer porous media using a pore network.
- The effect of ionomer content on pore size distribution was successfully modeled.

G R A P H I C A L A B S T R A C T



ABSTRACT

We model and validate the effect of ionomer content and Pt nanoparticles on nanoporous structure of catalyst layers in polymer electrolyte fuel cells. By employing Pore network modeling technique and analytical solutions, we analyze and reproduce experimental N₂-adsorption isotherms of carbon, Pt/ carbon and catalyst layers with various ionomer contents. The porous catalyst layer structures comprise of Ketjen Black carbon, Pt and Nafion ionomer. The experimental pore size distributions obtained by N₂-adsorption are used as an input to generate porous media using the pore network approach. Subsequently, the simulated porous structures are used to produce simulated N₂-adsorption isotherms, which are then compared to the experimentally measured isotherms. The results show a good agreement in the prediction of the effect of the ionomer content on the microstructure of catalyst layers. Moreover, the analysis of the isotherms confirms the hypothesis of ionomer distribution on the surface of agglomerates as well as the existence of different sorption regimes in primary and secondary pores of fuel cell catalyst layers.

1. Introduction

* Corresponding author. Simon Fraser University, Mechatronic Systems Engineering, School of Engineering Science, 250-13450 102 Avenue, Surrey, BC, Canada V3T 0A3.

E-mail address: melhannach@gmail.com (M. El Hannach). ¹ Laboratoire de Réactivité et de Chimie des Solides (L.R.C.S.), Université de

Picardie Jules Verne, CNRS UMR 7314-33 rue Saint Leu, Amiens, France.

 2 Réseau sur le Stockage Electrochimique de l'Energie (RS2E), FR CNRS 3459, France.

In current and past physical models of Catalyst Layer (CL) in Polymer Electrolyte Fuel Cells (PEFCs), there has been a lack of clear CL nanostructure picture, in particular, structure and function of ionomer, nature and distribution of active interfaces, structure and composition of agglomerates, porosity, mixed wettability, and water distribution [1]. Simulations of microstructure formation, including self-organization of components in CL, predict the interplay between fabrication parameters, structure and transport properties [2,3]. Moreover, understanding CL structure is imperative for development of novel materials and improvement of CL physicochemical properties. Nevertheless, in order to evaluate the impacts on "pseudo" steady state performance and durability, modeling efforts are needed to couple basic structural parameters obtained from modeling the CL microstructure formation to the fuel cell performance modeling [4,5].

In order to complete basic structural aspects, the remaining issues include establishing a relationship between composition and microstructure in terms of ionomer and Pt content, pore size distribution, and pore network characteristics. The Pore Network Modeling (PNM) approach is widely used to study the transport phenomena inside porous materials and its application has increased recently in the field of Proton Exchange Membrane Fuel Cell (PEMFC) materials. The main input of the PNM is the pore size distribution (PSD). Nitrogen physisorption is a common tool used to evaluate microstructures of porous materials and obtain pore size distribution curves. The PSD can be estimated either from adsorption or desorption isotherms using modified forms of Kelvin equation for capillary condensation.

In this work, we used the PSDs extracted from the nitrogen adsorption isotherms for various catalyst layer compositions as an input parameter for the PNM. The experimentally-derived isotherms were reported in Ref. [6] and explained in the Experimental section. Then the adsorption isotherms are simulated using the model described below to reproduce the experimental isotherms. The flexibility of the PNM allows for simulation of the nitrogen adsorption process with the consideration of pores blockage due to nitrogen condensation. A comparison between the experimental and simulated isotherms validates representation of the CL using the pore network model.

2. Experimental

In a previous study by Soboleva et al. [6], Ketjen Black carbon and a catalyst powder with 46 wt% Pt on Ketjen Black from Tanaka Kikinzoku Kogyo (TKK) [1,6] were used for N₂-adsorption studies and for fabrication of the catalyst coated membranes (CCMs). All CCMs were fabricated by spray-deposition of catalyst inks on Nafion 211 membrane, used as received. CCMs with 5, 10, 30 and 50 wt% of ionomer in the CL corresponding to 0.047 mg cm⁻² (0.1 l/C), 0.094 mg cm⁻² (0.2 l/C), 0.375 mg cm⁻² (0.8 l/C), and 0.875 mg cm⁻² (1.8 l/C) ionomer loadings were prepared. Carbon and Pt loadings were held constant at 0.475 mg cm⁻² and 0.4 mg cm⁻² respectively. The thickness of the catalyst layers was $12 \pm 2.0 \ \mu m$.

N₂-physisorption experiments, previously published in Ref. [6], were performed at 77 K on a 3100 Surface Area Analyzer (Beckman CoulterTM). N₂ gas of ultra-high purity 99.999% was used. Carbon and Pt/carbon samples of ~100 mg were degassed at 160 °C for 12 h prior to the adsorption experiment to remove any residual water from the samples. CCMs of total 50 cm² were cut into thin strips, placed in the sample tube and degassed for 12 h at 105 °C. Adsorption experiments were performed immediately after degassing. A blank adsorption experiment on Nafion 211 membrane was performed and it was found that contribution of the membrane to N₂ adsorption of CCMs is negligible. Adsorption isotherms of Pt/carbon powders and CCMs were all re-plotted after normalizing by carbon content in each sample. BET surface areas were found from BET equation [7], by plotting left-hand side term of the equation against p/p_0 in the range from 0.05 to 0.2 p/p_0 . Monolayer volume, n_m , was found from the slope and the intercept of such a BET plot, Eq. (1):

$$n_m = \frac{1}{\text{slope} + \text{intercept}} \tag{1}$$

Surface area is then calculated from Eq. (2):

$$SA_{BET} = \frac{n_m \cdot N \cdot s}{V} \tag{2}$$

where N – Avogadro's number, s – cross-sectional area of the N₂ molecule (0.162 nm²); V – molar volume of adsorbent gas.

Total pore volume was found at a partial pressure of $0.98 \ p/p_0$ (STP) by converting to liquid nitrogen volume using the conversion factor of 0.001547 [8].

Pore size distributions (PSDs) were assessed using the Barret– Joyce–Halenda (BJH) method, which is based on the Kelvin equation and accounts for the formation of a condensed layer adsorbed on the pore wall prior to capillary condensation, which is accounted for using Harkins–Jura equation:

$$t_{\rm N2} = \left(\frac{13.99}{0.034 - \log(p/p_0)}\right)^{1/2} (\rm{\mathring{A}})$$
(3)

where t_{N2} – thickness of the adsorbed layer on the pore wall, Å; p – partial pressure of N₂; p_0 – saturation pressure of N₂ at 77 K [9].

3. Model development

The PNM is based on using a network of pores and throats (links between adjacent pores) (Fig. 1) to represent the structure of a porous medium [10,11]. The pore network used in this work is constructed from spherical shape pores and cylindrical shape throats. The distance between two adjacent pores, i.e., network spacing, is constant. Thus, the pore network is regular. We use only 3D networks where every pore is connected to 6 adjacent pores (Fig. 1). The CL consists of agglomerated carbon/platinum particles. Two types of pores are considered: primary pores (d < 5 nm) representing pores inside agglomerates, and secondary pores (d > 5 nm) representing pores between agglomerates.

Secondary pores are represented by a 3D regular pore network with a network spacing equal to 150 nm. The value is chosen to be higher than the maximum pore diameter to avoid any overlap between two adjacent pores. The size of the network is measured by a number of pores in x, y, and z directions. The size of the main pore network is 15 \times 15 \times 15, which is equivalent to 3375 pores and 10,800 throats. The size of the simulated domain is equivalent to



Fig. 1. Pore network representing the primary pores (sub-network in the left) and the secondary pores (main network in the right).

The composition of investigated samples and the corresponding sizes of sub-networks.

Table 1

Sample composition	Sub-network size
Ketjen-Black (KB)	$5 \times 5 \times 5$
Pt/KB	4 imes 4 imes 4
Pt/KB + 5% ionomer	$3 \times 3 \times 3$
Pt/KB + 10% ionomer	$2 \times 2 \times 2$
Pt/KB + 30% ionomer	$1 \times 1 \times 1$
Pt/KB + 50% ionomer	0

 $2.4\times2.4\times2.4~\mu m$ which is large enough to capture the range of pore sizes indicated by the PSD.

The primary pores are represented with 2744 sub-networks. Each sub-network is geometrically located in the cubic space between 8 pores and 12 throats of the main network (Fig. 1) and represents the location of an agglomerate [11]. The pores located at the edges of sub-networks are connected to the adjacent throats of the main network. The size of the sub-network is chosen to represent the porous structure of an agglomerate with a diameter of ~ 100 nm. However, the pores inside the agglomerates can be blocked by the ionomer when its loading increases, assuming that the ionomer is mainly distributed on the surface of the agglomerates [10]. To represent this blockage of primary pores by ionomer, the number of pores in the sub-network was varied from $5 \times 5 \times 5$ to $1 \times 1 \times 1$ so that only the accessible primary pores are represented (Table 1). Experimentally obtained PSDs for different samples are used to generate the porous structures in the pore network model, Fig. 2.

To simulate the nitrogen adsorption isotherms using constructed pore networks, the relative pressure of nitrogen p/p_0 is increased gradually from 0 to 1 and then decreased following the same steps. The volume of the adsorbed nitrogen is calculated during adsorption (increase of p/p_0) and desorption (decrease of p/ p_0). At each step, the thickness of the adsorbed nitrogen is calculated as function of the relative pressure using Eq. (3).

The increase of t_{N2} reduces the nitrogen-free diameter d_{free} of the pores and the throats. This diameter is expressed as follows:

$$d_{\rm free} = d - 2t_{\rm N2} \tag{4}$$

where *d* is the geometrical diameter of the pore or the throat (at $t_{N2} = 0$ Å). As the relative pressure increases during adsorption process, d_{free} decreases.



Fig. 2. Pore size distributions of different catalyst layer structures obtained by Soboleva et al. [6].

The capillary condensation of nitrogen results in the complete filling of a pore with liquid nitrogen, $d_{\text{free}} = 0$. The process occurs when the radius of the meniscus r_{m} formed by the liquid/gas interface is lower than a critical radius r_{c} given by Kelvin equation:

$$r_{\rm c} = -\frac{2\gamma_{N2}V_m}{RT\ln(p/p_0)} 10^7 (\rm nm)$$
(5)

where γ_{N2} (N m⁻¹) is the surface tension and $V_{\rm m}$ (m³ mol⁻¹) is the molar volume of liquid nitrogen. The radius of the meniscus depends on the diameter of the pores and on the shape of the meniscus itself. We suppose that the shape of the meniscus curvature changes from the adsorption process to the desorption process (see Fig. 3). The expressions of $r_{\rm m}$ in the adsorption process and the desorption process are given in the Eqs. (6a) and (6b), respectively:

$$\frac{2}{r_{\rm m}} = \frac{1}{r} + \frac{1}{\infty} \Rightarrow r_{\rm m} = 2r = d_{\rm free}$$
(6a)

$$\frac{2}{r_{\rm m}} = \frac{1}{r} + \frac{1}{r} \Rightarrow r_{\rm m} = r = d_{\rm free}/2 \tag{6b}$$

From Eqs. (6a) and (6b), the capillary condensation occurs at higher relative pressure during the adsorption process ($d_{\rm free} < r_{\rm c}$) than during the desorption process ($d_{\rm free} < 2r_{\rm c}$). As a consequence a hysteresis is obtained.

The model is based on Hoshen–Kopelman algorithm [12]. The throats located at the boundaries of the main network are considered to be opened to the surroundings, where nitrogen is introduced. These throats are considered as starting points for the algorithm to investigate the networks and fill the pores and the throats with the correct amount of absorbed and liquid nitrogen using recursive functions.

For example, starting from a pore, the algorithm investigates one of the adjacent throats. If the capillary condensation occurs, this throat is considered as blocked ($d_{\text{free}} = 0$) and the second throat is investigated. If the capillary condensation does not occur, the algorithm investigates the next pore and continues until it finds a pore or a throat that is blocked by condensed liquid N₂. Then the algorithm goes back and investigates the last pore or throat with $d_{\text{free}} \neq 0$. The simulation continues until all possible pathways for nitrogen starting from the throats located on the boundaries of the main network are investigated. This process is repeated during all nitrogen adsorption and desorption steps. Additionally, the pore trapping phenomenon is considered in the simulation. This process occurs during desorption, when all pathways leading to pores inside the structure are blocked with condensed liquid nitrogen resulting in inability of these pores to desorb nitrogen at a partial pressure prescribed by Kelvin equation.

Ideally, all pores larger than the molecular diameter of nitrogen (3.5 Å) can be detected by N_2 adsorption technique. The following



Fig. 3. Illustration of the meniscus shape (a) during the adsorption process, (b) during the desorption process.

adsorption regimes are distinguished during nitrogen adsorption. The physical adsorption in micropores [13,14] (<2 nm) that occurs at very low gas partial pressures ($<0.01 p/p_0$) is described by the theory of micropore filling [15-17]. Adsorption in these pores occurs through volume filling, with the adsorption potential strongly enhanced by the overlap of adsorption forces from the opposite walls of the pore; i.e., neither mono-/multilayer adsorption or capillary condensation takes place. Concurrent with the filling of micropores, adsorption of monolayer of nitrogen molecules occurs on the pore walls of meso- and macropores (>2 nm) and on the external surface of a particle. Although it is difficult to distinguish between these concurrent processes, the consensus is that the steeper the rise in the adsorption isotherm at these very low partial pressures, the larger is the total volume of micropores [18,19]. As the partial pressure is increased in the experiment, monolayer formation evolves into multilayer formation until the pore radius related to the capillary condensation at a given partial pressure is reached, instantly filling the pore. Thus, at medium and high partial pressures, two processes may be observed to occur simultaneously: multilayer formation and capillary condensation. At a partial pressure close to unity, either all pores are filled with nitrogen, and the isotherm reaches a plateau because the saturation point is reached, or multilayer formation continues in macropores (>100 nm) and no saturation point in the isotherm is observed.

By comparing the simulated and experimental adsorption isotherms, the latter extracted from Soboleva et al. [6], we can assess the validity of representing the CL structure using a pore network model.

4. Results and discussion

The results obtained with the PNM are presented in Fig. 4. In general, it is possible to distinguish three different regions in the adsorption isotherms. The effect of the PSD and in particular the amount of primary pores is evident from the low partial pressure section of adsorption isotherms ($<0.05 p/p_0$) where the pore partial filling process is occurring (no capillary condensation). At the values of relative pressure >0.05, the amount of adsorbed nitrogen increases almost linearly, which is consistent with the hypothesis of monolayer adsorption.

The second part of the isotherm $(0.35 < p/p_0 < 0.7)$ corresponds to the filling of the primary pores by capillary condensation. These

3.0E+08 KB Pt/KB Volume of adsorbed nitrogen (nm³) 2.5E+08 Pt/KB+5% lonomer Pt/KB+10% lonomer Pt/KB+30% lonomer 2.0E+08 Pt/KB+50% lonomer 1.5E+08 1.0E+08 5.0E+07 0.0E+00 0 0.2 0.4 0.6 0.8 1

Fig. 4. Simulated N₂ adsorption isotherms obtained using the pore network model.

p/po

pores have smaller diameters (~5 nm) than the secondary pores and therefore they are filled first. The third part of the isotherm $(0.7 < p/p_0)$ corresponds mainly to the filling of the secondary pores by capillary condensation. The amount of accumulated nitrogen increases faster in this part of the isotherm because the secondary pores are larger and therefore have a larger contribution to the porosity.

As it can be observed from Fig. 5, good agreement between the experimental and the simulation results is found. The model predicts well the position of the hysteresis. However, there are discrepancies in the shape of the isotherm between experimental and simulation results, especially in the case of Pt/KB 50% ionomer CL. In the simulation, the accumulated volume of the adsorbed nitrogen is much higher than experimental results. Thus, the model overestimates the total volume occupied by the pores of a given diameter. This overestimation is believed to be due to the assumption of the regular pore network, i.e., constant spacing between pores. In a microstructure model based on almost spherical particles, we can assume that the length of the throats (cylindrical shape pores) is nearly equal to their diameters. Hence, with the assumption of constant spacing (throat length) of 150 nm, the length of the small throats (e.g., 10 nm) may be overestimated, together with their volume. If the number of the small pores is relatively higher than the number of the large pores (with a diameter close to the network spacing) then overestimation of the volume occupied by the small pores is not negligible. This phenomenon is observed in the case of flat PSDs, such as for Pt/KB 50% ionomer catalyst layer, where the volume occupied by each pore size class is similar. The case of Pt/KB 50% ionomer illustrates the limit of the regular pore network assumption, where the simulation results are drastically different from the experimental. For all other samples, the PSD is characterized by a distinctive peak at \sim 50 nm; in this case the overestimation of the volume occupied by the small pores is negligible compared to the total volume occupied by the large pores around the peak. In this case, representing the porous structure with a regular pore network is acceptable and validated by experimental results.

We observe in Fig. 5 that the total measured and calculated pore volume decreases when ionomer content in the catalyst layer is increased. In terms of PSD, the presence of the ionomer reduces the volume occupied by each pore size class, especially the primary pores with diameters around 5 nm (see Fig. 2). In the simulation, the primary pores, i.e., intra-agglomerate pores, are assumed to be blocked by the ionomer distributed on the surface of the agglomerates. This assumption is modeled by reducing the number of the



Fig. 5. A comparison between simulated and experimental N₂ adsorption isotherms.

accessible pores in the sub-networks (Table 1). The assumption leads to simulation results that are in good agreement with the experimental ones. Thus, we conclude that the simulation results confirm that the ionomer is distributed on the surface of agglomerates rendering the intra-agglomerate pore space inaccessible to gaseous nitrogen.

From these findings, we conclude that the regular pore network comprising primary and secondary pore networks is acceptable to represent the microstructure of fuel cell catalyst layers. The assumption of a regular pore network is valid for PSDs that possess a distinct peak and is not applicable for flat PSDs where pore volumes of all pores are similar. Since realistic catalyst layers are represented by PSDs with a peak at a certain pore size, the assumption is then valid for most of conventional catalyst layer structures. Moreover, the described results validate the hypothesis that ionomer is distributed on the surface of the agglomerates.

Acknowledgment

The authors gratefully acknowledge partial funding from CEA, Grenoble, France, as well as the NRC-CEA collaboration program and Simon Fraser University. The CEA funding was received from the European Union's Seventh Framework Programme (FP7/2007–2013) for the Fuel Cells and Hydrogen Joint Technology Initiative under grant agreement n° [256798] (PEMICAN).

References

- M.H. Eikerling, K. Malek, Electrochemical Materials for PEM Fuel Cells: Insights from Physical Theory and Simulation in Modern Aspects of Electrochemistry, Springer-Verlag, New York, 2009, pp. 169–247.
- [2] K. Malek, M. Eikerling, Q. Wang, T. Navessin, Z. Liu, J. Phys. Chem. C 111 (2007) 13627.
- [3] K. Malek, T. Mashio, M. Eikerling, Electrocatalysis 2 (2011) 141-157.
- [4] A.A. Franco, in: C. Hartnig, C. Roth (Eds.), Polymer Electrolyte Membrane and Direct Methanol Fuel Cell Technology (PEMFCs and DMFCs), Fundamentals and Performance, vol 1, Woodhead, Cambridge, UK, 2012.
- [5] K. Malek, A.A. Franco, J. Phys. Chem. B 115 (2011) 8088.
- [6] T. Soboleva, X. Zhao, K. Malek, Z. Xie, T. Navessin, S. Holdcroft, ACS Appl. Mater. Interfaces 2 (2010) 375–384.
 [7] S. Brunauer, P.H. Emmett, E. Teller, J. Am. Chem. Soc. 60 (1938) 309–319.
- [7] S. Buhadet, F.H. Elmett, E. Feller, J. All, Chem. Soc. 66 (1936) 505–515.
 [8] M. Uchida, Y. Fukuoka, Y. Sugawara, N. Eda, A. Ohta, J. Electrochem. Soc. 143 (1996) 2245–2252.
- [9] S.J. Gregg, K.S.W. Sing, Adsorption, Surface Area and Porosity, Acedemic Press, New York, 1982.
- [10] M. El Hannach, J. Pauchet, M. Prat, Electrochim. Acta 56 (2011) 10796–10808.
 [11] M. El Hannach, M. Prat, J. Pauchet, Int. J. Hydrogen Energy 24 (2012) 18996–
- 19006.
- [12] A. Al-Futaisi, T.W. Patzek, Phys. A 321 (2003) 665-678.
- [13] K.S.W. Sing, D.H. Everett, R.A.W. Haul, L. Moscou, R.A. Pierotti, J. Rouquerol, T. Siemieniewska, Pure Appl. Chem. 57 (1985) 603–619.
 [14] J. Rouquerol, D. Avnir, C.W. Fairbridge, D.H. Everett, J.H. Haynes, N. Pernicone,
- [14] J. Rouquerol, D. Avnir, C.W. Fairbridge, D.H. Everett, J.H. Haynes, N. Pernicone, J.D.F. Ramsay, K.S.W. Sing, K.K. Unger, Pure Appl. Chem. 8 (1994) 1739–1758.
 [15] M. Kruk, M. Jaroniec, Chem. Mater. 13 (2001) 3169–3183.
- [16] M.M. Dubinin, Carbon 25 (1987) 593–598.
- [17] J. Choma, M. Jaroniec, J. Piotrowska, Carbon 26 (1988) 1-6.
- [18] J.M. Thomas, Introduction to the Principles of Heterogeneous Catalysis, Academic Press, New York, 1967.
- [19] N.A. Seaton, J.P.R.B. Walton, N. Quirke, Carbon 27 (1989) 853-861.