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Quantitative WD-XRF Calibration for Small Ceramic Samples and Their Source Material

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A wavelength-dispersive X-ray fluorescence (WD-XRF) calibration is developed for small powdered samples (300 mg) with the purpose of analyzing ceramic artifacts that might be available only in limited quantity. This is compared to a conventional calibration using a larger sample mass (2 g). The comparison of elemental intensities obtained in both calibrations shows that the decrease in analyzed sample mass results in a linear decrease in measured intensity for the analyzed elements. This indicates that the small- and large-sample calibrations are comparable. Moreover, the elemental contents of four ceramic sherds and two potential clay sources fall well within the range of the certified reference materials that are the basis of the calibration curves. The advantage with the analytical method presented here is that it is rapid and requires only a small amount of sample that can easily be re-used for further analyses. This method has great potential in ceramic provenance studies. © 2011 Wiley Periodicals, Inc.

INTRODUCTION

In archaeometry a wide range of chemical techniques has been used to assess the geochemical content of various archaeological objects (e.g., Adan-Bayewitz, Asaro, & Giaque, 1999; Szökefalvi-Nagi et al., 2004; Asaro & Adan-Bayewitz, 2007; Cultrone et al., 2010; Mazar et al., 2010; Renson et al., 2011). Among these objects are ceramic sherds, which, if accurately quantified for their elemental content and analyzed in a multivariate statistical approach, can provide information on the source and trading of ceramics in ancient times. The main problem of analyzing archaeological objects is that, for conservation reasons, the availability of sample is often limited. In the case of ceramics, the sample often consists of small fragments found on archaeological sites or retrieved from collections. Archaeometrists are therefore in

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constant search for analytical techniques that allow elemental quantification on a minimum amount of material.

X-ray fluorescence (XRF) is an analytical technique that has been widely used in many industrial, environmental, geological, and archaeological investigations (see West et al., 2010, for a recent review of XRF applications). Among the applications, XRF has been widely used in the earth sciences (e.g., Potts, 1987, and references therein; Gardner, 1990; Cheburkin, Frei, & Shotyk, 1997; Zambello & Enzweiller, 2002; West et al., 2010, and references therein) and has been developed to analyze vegetation samples (e.g., Cheburkin & Shotyk, 1996; Margui, Hidalgo, & Queralt, 2005) and liquid matrices (e.g., Cumming & McDonald, 1985; Varga et al., 1995; Oprea et al., 2009). In archaeological studies, XRF has a wide range of applications, and samples can be potentially re-used for further analysis depending on the sample preparation (e.g., Adan-Bayewitz, Asaro, & Giaque, 1999; Hein et al., 2004; Craig et al., 2007; Tripathi & Rajamani, 2007; Cultrone et al., 2010; West et al., 2010, and references therein). The common drawbacks of XRF are the so-called matrix effects, which are principally caused by the composition, grain size, and mineralogy of the sample. A way to minimize these effects is to finely powder the samples and to apply subsequent pretreatments such as lithium borate fusion or pelletization using a wax binder (Potts, 1987). However, these processes alter the properties of the samples, which then cannot be re-used for further analyses. Classical XRF on powder, pellets, or fused beads also requires several grams of sample to achieve an accurate elemental quantification (e.g., Zambello & Enzweiller, 2002; Cultrone et al., 2010). This limits its application to archaeological artifacts that are available only in sufficiently large quantity. For small samples, direct analysis of powdered material would therefore be generally preferred. Such an application needs to be carefully developed to provide accurate elemental quantification (Adan-Bayewitz, Asaro, & Giaque, 1999).

New generations of wavelength-dispersive XRF (WD-XRF) offer the potential to decrease the quantity of material by carefully developing calibrations on samples as small as a few hundred milligrams. A common WD-XRF setup allows the determination of major and selected trace elements without specific spectrometric design. In this short note we provide the details of a calibration that was purposely developed for quantifying selected elements in small ceramic fragments and their potential raw materials. We present some of the technical aspects of the selection and preparation of the certified reference materials (CRMs) as well as a specific calibration developed to optimize WD-XRF for ceramic matrices and their source materials. This specific calibration for small sample mass (300 mg) is validated by comparing the intensities obtained with a conventional calibration (i.e., using 2-g samples).

METHODOLOGY

Selection of Certified Reference Materials

Thirty-five commercially available CRMs were selected from our laboratory stock. All our CRMs are finely powdered and stored in desiccators. They consist of sedimentary and igneous rocks (NIST1d, NIST70a, NIST278, NIST2780, NIST688), river sediments (NIST8704, STSD3), lake sediments (LKSD1, LKSD4, BIL1, JLK1), estuarine sediments (NIST1646a), and marine sediments (NIST1944, NIST2702, NIST2703, MAG1, PACS2, MESS3) as well as sands (NIST81a, NIST1413), clays (NIST97b, NIST98b, NIST679), and soils (SO2, SO3, SO4, NIST2711, NIST2586, NCSDC73325, NCSDC73326, JSO1) from sedimentary outcrops, and coal fly ash (JCFA1, NIST1633a, NIST1633b, ZUK1), the latter type being added to extend the range of various trace elements. Rather than exclusively choosing fine sediments, we have included several sediment and rock types to represent mixed matrices (i.e., fine material and temper) that may be found in ceramics (Rapp, 2009).

Sample Preparation

All the analyses were carried out directly on CRM powders. Two calibrations were developed: one based on 2 g of each CRM, as commonly done in XRF analyses, and a second based on 300 mg of powder that was specifically developed here. For the first calibration, 2 g of dried and homogenized loose powder was transferred to 40-mm-opening plastic cups with a 2.5- μ m transparent mylar film at its base. These samples were placed in 36-mm-opening metallic cup holders for analysis. The challenging part of this study was the development of the second calibration, using 300 mg of powder, which was transferred to a plastic cup with a 10-mm opening, also with a 2.5- μ m mylar film at the base. In these small cups the powder was slightly compressed manually to reduce porosity effects. These plastic cups were subsequently loaded in a metallic cup holder with an opening (i.e., analytical) diameter of 8 mm.

XRF Optimization

All the analyses were conducted using a Bruker S8-Tiger WD-XRF analyzer equipped with a Rh anticathode X-ray tube, four analyzing crystals (XS-55, PET, LIF200, LIF220), a flow proportional counter (for light elements), and a scintillation counter (for heavy elements). Analyses were performed under helium atmosphere to avoid powder loss in the analytical chamber, which may occur under vacuum conditions. The main parameters, used for each calibration and each element, are reported in Table I. Such a WD-XRF setup allows the determination of a wide range of elements, generally from C to U. In this study, the following elements were selected: Na, Mg, Al, Si, P, S, K, Ca, Ti, V, Mn, Fe, Ni, Cu, Zn, Br, Rb, Sr, Ba, and Pb. This selection is based on the certified values available in each CRM, which are matched with the expected detectable concentrations of our WD-XRF spectrometer. The WD-XRF resolution and sensitivity have been optimized by combining the proper collimators, crystals, and counters for each element (Table I). The peak acquisition time was optimized using the built-in SpectraPlus® software interface, which calculates the time necessary to achieve an average selected detection limit. The acquisition time varies, therefore, among elements as well as among concentrations of a given element (i.e., the lower the concentration, the greater the time). To avoid any excessively long acquisition times, a maximum time was set for each element, which varies from 30 to 100 s (Table I). The background measurement times have been fixed manually, generally at 10 s, except for a few trace elements

		Number		Concentration	ration			Collimator			Acqı Tin	Acquisition Time (s)	Overlav			Acc	Renr
	Unit	of CRM	Min.	Max.	Line	kV	ШA	(°)	Crystal	Detector	Max.	Bckgd.	Correction	R^{2}	ILLD	(%)	(%)
arge	cups (4(Large cups (40 mm diameter, analytical surface 36 mm)	eter, ana	dytical s	surface	36 mm)	_										
Na	%	13	0.03	1.90	$K\alpha$	30	135	0.46	XS-55	FC	100	10		0.99	0.003	က	66
Mg	%	26	0.04	4.98	$K\alpha$	30	135	0.23	XS-55	FC	45	10		0.99	0.004	6	66
К	%	21	0.08	9.79	$K\alpha$	50	81	0.46	LIF200	FC	30	10		0.99	0.001	က	66
Ca	%	30	0.02	37.5	$K\alpha$	50	10	0.23	LIF200	FC	100	10		0.99	0.002	7	66
AI	%	21	0.28	15.1	$K\alpha$	30	20	0.23	PET	FC	40	10		0.98	0.010	9	66
Si	%	27	1.92	40.0	$K\alpha$	30	20	0.23	PET	FC	30	10		0.96	0.020	ы С	66
ï	$\mu g. g^{-1}$	24	100	20100	$K\alpha$	50	81	0.46	LIF200	FC	30	10	$\operatorname{Ba} \operatorname{L}_{\alpha}$	0.99	2	5	66
Fe	%	21	0.05	9.05	$K\alpha$	60	29	0.23	LIF220	\mathbf{SC}	100	10		0.99	0.001	က	66
Mn	$\mu g. g^{-1}$	26	47	3098	$K\alpha$	60	67	0.46	LIF200	\mathbf{SC}	60	20	Fe K α	0.99	e G	4	66
Ba	$\mu g. g^{-1}$	14	210	2250	$K\alpha$	60	67	0.23	LIF220	\mathbf{SC}	50	30	Ti K α	0.99	21	7	93
Rb	$\mu g. g^{-1}$	17	2	190	$K\alpha$	60	67	0.46	LIF200	\mathbf{SC}	100	09		0.99	1	4	66
\mathbf{Sr}	$\mu g. g^{-1}$	23	26	3200	$K\alpha$	60	67	0.23	LIF220	\mathbf{SC}	30	20		0.99	1	ы С	66
S	%	10	0.01	0.35	$K\alpha$	30	135	0.46	PET	FC	30	10		0.98	0.001	25	89
Ь	$\mu g. g^{-1}$	19	129	3000	$K\alpha$	30	135	0.46	PET	FC	30	10	$\operatorname{Ca} \operatorname{K} \alpha$	0.95	5	20	97
Br	$\mu g. g^{-1}$	10	33	86	$K\alpha$	60	67	0.23	LIF200	\mathbf{SC}	30	10		0.98	1	27	8
Cu	$\mu g. g^{-1}$	17	9	380	$K\alpha$	60	67	0.46	LIF200	\mathbf{SC}	30	20	Fe K α	0.99	1	8	$\overline{96}$
Zn	$\mu g. g^{-1}$	14	18	656	$K\alpha$	60	67	0.46	LIF220	\mathbf{SC}	30	10		0.99	2	က	66
\mathbf{Pb}	$\mu g. g^{-1}$	17	33	330	$\Gamma\beta$	60	67	0.23	LIF220	\mathbf{SC}	100	10		0.99	1	14	94
ïZ	$\mu g. g^{-1}$	19	4	150	$K\alpha$	60	67	0.46	LIF200	\mathbf{SC}	100	09	Y K α	0.99	1	7	66
Λ	$\mu g. g^{-1}$	23	36	297	$K\alpha$	50	81	0.23	LIF220	FC	100	30		0.98	1	ũ	66

			Mundour		Concentration	bration			Collimator			Tin	Time (s)	Orrowlerr			400 V	Done
Name if the paralytical surface 8 mm) Na % 10 001 1.58 Ka 30 135 0.46 XS-55 FC 100 30 209 0.009 K % 10 0.01 1.58 Ka 30 135 0.23 LIF200 FC 30 20 0.99 0.009 0.		Unit	of CRM	Min.	Max.	Line	kV	ШĄ		Crystal	Detector	Max.	Bckgd.	Correction	R^2	ILLD	(%)	(%)
Na % 10 001 1.59 Ka 30 135 0.46 XS-55 FC 100 30 20 0.99 0.003 K % 19 0.03 3.45 Ka 30 135 0.23 LIP200 FC 30 10 0.09 0.090 0.003 K % 19 0.03 3.45 Ka 50 10 20 0.90 0.003 Ni % 24 192 4.00 Ka 50 20 0.23 EFT FC 30 10 0.99 0.003 Ni μgg^{-1} 25 100 1300 Ka 50 20 0.23 LIP200 FC 30 10 0.99 0.030 Ni μgg^{-1} 15 Ka 50 51 117200 FC 30 10 109 10 Kit 60 67 0.35 LIP200 FC	Smal	ll cups (1	0 mm diam€	iter, ana	lytical s		8 mm)											
	Na	%	10	0.01	1.59	$K\alpha$	30	135	0.46	XS-55	FC	100	30		0.99	0.007	Ð	92
K % 19 0.08 3.45 Ka 50 81 0.23 LIP200 FC 30 10 99 0.003 Ca % 26 0.02 150 Ka 50 10 0.23 LIP200 FC 100 10 99 0.003 Xi % 21 0.02 150 Ka 30 20 0.23 PET FC 100 10 99 0.003 Xi 27 100 130 Ka 50 30 20 0.23 LIP200 FC 100 10 99 0.003 Mn μgg^{-1} 26 47 3008 Ka 50 11 20 50 70 10 100	Mg	%	21	0.04	4.98	$K\alpha$	30	135	0.23	XS-55	FC	30	20		0.99	0.009	7	93
Qa % 26 0.02 15.0 Ka 50 10 0.23 LHP200 FC 100 10 0.39 0.003 Xi % 10 0.28 6.42 Ka 30 20 0.23 PET FC 180 10 0.39 0.003 Yi μgg^{-1} 25 100 1430 Ka 50 51 0.46 LHP200 FC 100 10 0.39 0.010 Min μgg^{-1} 25 100 1430 Ka 50 51 0.46 LHP200 SC 100 114 0.39 0.010 Min μgg^{-1} 18 24 Ka 50 52 147 117 93 117 94 117 94 114 114 114 114 114 114 114 114 114 114 114 114 114 114 114 114 114 114 <	K	%	19	0.08	3.45	$K\alpha$	50	81	0.23	LIF200	FC	30	10		0.99	0.002	က	06
M1 % 10 0.28 6.42 Ka 30 20 0.23 PET FC 180 10 0.98 0.096 0.036 Ti μgg^{-1} 25 100 14300 Ka 30 20 0.23 PET FC 30 10 0.96 0.096 0.036 Ti μgg^{-1} 25 100 14300 Ka 50 51 0.46 LIF200 FC 10 0.96 0.096 0.01 Mn μgg^{-1} 25 47 3098 Ka 60 67 0.46 LIF200 SC 100 10 7.97 0.99 0.13 Mn μgg^{-1} 10 180 1490 Ka 60 67 0.23 LIF220 SC 100 60 7.1 10 19 14 14 14 14 14 14 14 14 14 14 14 14 14 <	Ca	%	26	0.02	15.0	$K\alpha$	50	10	0.23	LIF200	FC	100	10		0.99	0.005	7	06
Ni 24 1.92 400 Ka 30 20 0.23 PET FC 30 10 0.96 0.96 0.93 0.96 0.90 0.93 0.93 0.96 0.90 </td <td>AI</td> <td>%</td> <td>10</td> <td>0.28</td> <td>6.42</td> <td>$K\alpha$</td> <td>30</td> <td>20</td> <td>0.23</td> <td>\mathbf{PET}</td> <td>FC</td> <td>180</td> <td>10</td> <td></td> <td>0.98</td> <td>0.010</td> <td>7</td> <td>91</td>	AI	%	10	0.28	6.42	$K\alpha$	30	20	0.23	\mathbf{PET}	FC	180	10		0.98	0.010	7	91
$ \begin{array}{ ccccccccccccccccccccccccccccccccccc$	Si	%	24	1.92	40.0	$K\alpha$	30	20	0.23	\mathbf{PET}	FC	30	10		0.96	0.030	4	06
Fe % 21 0.05 9.05 Ka 50 5 0.23 LIF220 SC 100 10 0.99 0.99 0.919 Mn μgg^{-1} 25 47 3098 Ka 60 67 0.46 LIF220 SC 100 60 Fe Ka 0.99 0.114 Ba μgg^{-1} 18 2 190 Ka 60 67 0.23 LIF220 SC 100 60 T1 Ka 0.99 0.114 Kb μgg^{-1} 18 2 1041 Ka 60 67 0.23 LIF220 SC 100 60 T1 Ka 0.99 0.10 Kb μgg^{-1} 28 Ka 60 67 0.23 LIF220 SC 100 60 Fe Ka 0.99 0.10 Kb μgg^{-1} 38 Ka 60 67 0.23 LIF220 SC 100 60 Fe Ka	ï	$\mu g.g^{-1}$	25	100	14300	$K\alpha$	50	81	0.46	LIF200	FC	100	10	$\operatorname{Ba} \operatorname{L}_{lpha}$	0.99	13	5 C	92
	\mathbf{Fe}	%	21	0.05	9.05	$K\alpha$	50	Ð	0.23	LIF220	$^{\rm SC}$	100	10		0.99	0.010	4	92
Ba μgg^{-1} 10 180 1490 Ka 60 67 0.23 LlF220 SC 100 60 Ti Ka 0.99 114 Rb μgg^{-1} 18 2 190 Ka 60 67 0.46 LlF220 SC 100 60 70 99 11 Sr μgg^{-1} 21 26 1041 Ka 60 67 0.23 LlF220 SC 100 60 70 99 12 Br μgg^{-1} 22 129 1279 Ka 60 67 0.23 LlF200 SC 100 60 772 202 Br μgg^{-1} 16 6 310 Ka 60 67 0.23 LlF200 SC 100 60 772 202 202 Cu μgg^{-1} 16 6 61 67 0.23 LlF200 SC 100 60 772	Мn	$\mu g.g^{-1}$	25	47	3098	$K\alpha$	60	67	0.46	LIF200	\mathbf{SC}	100	09	Fe K α	0.99	4	ŋ	92
Rb $\mu g.g^{-1}$ 18 2 190 Ka 60 67 0.46 LlF200 SC 100 60 97 1 Sr $\mu g.g^{-1}$ 21 26 1041 Ka 60 67 0.23 LlF220 SC 100 60 0.99 2 S $\mu g.g^{-1}$ 22 129 1279 Ka 60 67 0.23 LlF200 SC 100 60 0.72 202 Dr $\mu g.g^{-1}$ 8 3 86 Ka 60 67 0.23 LlF200 SC 100 60 72 202 Zu $\mu g.g^{-1}$ 16 6 71 0.23 LlF200 SC 100 60 72 202 Zu $\mu g.g^{-1}$ 16 6 61 67 0.46 LlF200 SC 100 60 72 202 Zu $\mu g.g^{-1}$ 17 8 144	\mathbf{Ba}	$\mu g.g^{-1}$	10	180	1490	$K\alpha$	60	67	0.23	LIF220	\mathbf{SC}	100	09	Ti K α	0.99	114	6	91
Sr $\mu g g^{-1}$ 21 26 1041 $K \alpha$ 60 67 0.23 LlF220 SC 100 60 0.99 2 P $\mu g g^{-1}$ 22 127 $K \alpha$ 30 135 0.46 PET FC 30 10 0.80 0.02 0.30 0.02 0.30 0.02 0.20 <	Rb	$\mu g.g^{-1}$	18	2	190	$K\alpha$	60	67	0.46	LIF200	\mathbf{SC}	100	09		0.97	1	11	93
S % 19 0.01 1.57 Ka 30 135 0.46 PET FC 30 10 0.80 0.00 P $\mu g g^{-1}$ 22 129 1279 Ka 60 135 0.46 PET FC 30 10 Cak 0.72 202 Br $\mu g g^{-1}$ 8 3 86 Ka 60 67 0.23 LIF200 SC 100 60 72 202 Zn $\mu g g^{-1}$ 16 6 310 Ka 60 67 0.46 LIF200 SC 100 60 72 202 Zn $\mu g g^{-1}$ 17 8 414 Ka 60 67 0.46 LIF200 SC 100 60 72 203 Ni< $\mu g g^{-1}$ 17 8 150 Ka 60 67 0.46 LIF200 SC 100 60 73 12	\mathbf{Sr}	$\mu g.g^{-1}$	21	26	1041	$K\alpha$	60	67	0.23	LIF220	\mathbf{SC}	100	09		0.99	2	6	93
$ \begin{array}{l c c c c c c c c c c c c c c c c c c c$	s	%	19	0.01	1.57	$K\alpha$	30	135	0.46	\mathbf{PET}	FC	30	10		0.80	0.002	36	66
Br $\mu g.g^{-1}$ 8 3 86 Ka 60 67 0.23 LIF200 SC 100 60 69 7 0.98 1 Cu $\mu g.g^{-1}$ 16 6 310 K α 60 67 0.46 LIF200 SC 100 60 Fe K\alpha 0.99 2 Zn $\mu g.g^{-1}$ 9 12 330 L β 60 67 0.46 LIF200 SC 100 60 67 3 Ni $\mu g.g^{-1}$ 17 8 150 K α 60 67 0.46 LIF200 SC 100 60 7 3 V $\mu g.g^{-1}$ 17 8 150 K α 60 67 0.46 LIF200 SC 100 60 7 9 1 V $\mu g.g^{-1}$ 17 8 150 K α 50 81 0.23 LIF220 SC 100 6	Ь	$\mu g.g^{-1}$	22	129	1279	$K\alpha$	60	135	0.46	\mathbf{PET}	FC	30	10	$\operatorname{Ca} \operatorname{K} \alpha$	0.72	202	26	88
$ \begin{array}{c ccccccccccccccccccccccccccccccccccc$	$_{\mathrm{Br}}$	$\mu g.g^{-1}$	8	33	86	$K\alpha$	60	67	0.23	LIF200	\mathbf{SC}	100	09		0.98	1	27	47
$ \begin{array}{c ccccccccccccccccccccccccccccccccccc$	Cu	$\mu g.g^{-1}$	16	9	310	$K\alpha$	60	67	0.46	LIF200	\mathbf{SC}	100	09	Fe K α	0.99	2	4	78
$ \begin{array}{c ccccccccccccccccccccccccccccccccccc$	\mathbf{Zn}	$\mu g.g^{-1}$	26	18	414	$K\alpha$	60	67	0.46	LIF200	\mathbf{SC}	100	09		0.97	33	10	91
Ni $\mu g.g^{-1}$ 17 8 150 K α 6.0 6.7 0.46 LIF200 SC 300 6.0 Y K α 0.99 1 V $\mu g.g^{-1}$ 19 36 297 $K\alpha$ 50 81 0.23 LIF220 FC 150 60 7 4 Kv: kilovolts, mA: milli-amperes; XS-55: W/Si multilayer crystal; LIF200 and 220: lithium fluoride crystals; PET: pentaerythrite crystal; LLD: lower limit of detection. TF	\mathbf{Pb}	$\mu g.g^{-1}$	6	12	330	$\Gamma\beta$	60	67	0.23	LIF220	\mathbf{SC}	100	09		0.99	5	4	88
$\frac{V}{Kv} \frac{\mu g g^{-1}}{\mu lovolts} \frac{19}{mA}: \frac{36}{milli-amperes}, XS-55; W/Si multilayer crystal; LIF200 and 220: lithium fluoride crystals; PET: pentaerythrite crystal; LLD: lower limit of detection. TF (200) and 220: lithium fluoride crystals; PET: pentaerythrite crystal; LLD: lower limit of detection. TF (200) and 220: lithium fluoride crystals; PET: pentaerythrite crystal; LLD: lower limit of detection. TF (200) and 220: lithium fluoride crystals; PET: pentaerythrite crystal; LLD: lower limit of detection. TF (200) and 220: lithium fluoride crystals; PET: pentaerythrite crystal; LLD: lower limit of detection. TF (200) and 220: lithium fluoride crystals; PET: pentaerythrite crystal; LLD: lower limit of detection. TF (200) and 220: lithium fluoride crystals; PET: pentaerythrite crystal; LLD: lower limit of detection. TF (200) and 220: lithium fluoride crystals; PET: pentaerythrite crystal; LLD: lower limit of detection. TF (200) and 220: lithium fluoride crystals; PET: pentaerythrite crystal; LLD: lower limit of detection. TF (200) and 220: lithium fluoride crystals; PET: pentaerythrite crystal; LLD: lower limit of detection. TF (200) and 220: lithium fluoride crystals; PET: pentaerythrite crystal; LLD: lower limit of detection. TF (200) and 220: lithium fluoride crystals; PET: pentaerythrite crystal; LLD: lower limit of detection. TF (200) and 220: lithium fluoride crystals; PET: pentaerythrite crystal; LLD: lower limit of detection. TF (200) and 220: lithium fluoride crystals; PET: pentaerythrite crystal; LLD: lower limit of detection. TF (200) and 220: lithium fluoride crystal; PET: pentaerythrite crystal; LLD: lower limit of detection. TF (200) and 220: lithium fluoride crystal; PET: pentaerythrite crystal; LLD: lower limit of detection. TF (200) and 220: lithium fluoride crystal; PET: pentaerythrite crystal; LLD: lower limit of detection. TF (200) and 220: lithium fluoride crystal; PET: pentaerythrite crystal; LLD: lower limit of detection. TF (200) and 220: lithium fluoride crystal; PET: pentaerythri$	Ni	$\mu g. g^{-1}$	17	8	150	$K\alpha$	60	67	0.46	LIF200	\mathbf{SC}	300	09	Y K α	0.99	1	8	82
Kv: kilovolts; mA: milli-amperes; XS-55: W/Si multilayer crystal; LJF200 and 220: lithium fluoride crystals; PET: pentaerythrite crystal; LLD: lower limit of detection. T	Λ	$\mu g.g^{-1}$	19	36	297	$K\alpha$	50	81	0.23	LIF220	FC	150	09		0.99	4	c,	90
is the median value obtained by communities (Removed to the corrected VBF concentrations for each alement in each standard. The removing the feat of the removal of the	Kv: ki istha	ilovolts; m median va	A: milli-ampel	res; XS-5E	5: W/Si m	ultilayer (BMs and	crystal; I	LIF200 a	und 220: lithiur	n fluoride cr	ystals; PET: p	entaeryth	urite crystal;	LLD: lower lim	uit of dete	sction. The	accurac	cy (Acc.

Table I. (Continued)

where intensities were expected to be low (see Table I). The overall measurement time is about 40 min per sample (i.e., 36 samples/24 h).

The calibration procedure is achieved in two steps: (1) all the CRMs are scanned at standard conditions to allow for optimization of sensitivity, resolution, and detection limits (see preceding), and then (2) a full analysis of the CRMs selected to give the best fit (i.e., to minimize standard error) of the ultimate calibration line is conducted. This necessary selection depends on (a) the available certified values; (b) the behavior of each CRM matrix in both calibrations (2 g loose powder vs. 300 mg pressed powder); and (c) the targeted standard error, which can be decreased by carefully removing values slightly off the calibration line without changing its slope. This selection process explains why, for a given element, the number of CRMs may vary between both calibrations (Table I). A posteriori corrections are also applied. They are of two types: (1) a software-built matrix correction that takes into account the inter-elemental interferences in the sample (i.e., a fluorescence X-ray originating from an element that causes electronic transitions in elements with lower energies), and (2) overlay corrections that are applied individually for peak overlaps commonly encountered in XRF spectra (see Table I). Finally, a quantitative calibration curve has been developed for each element by using a linear regression between the corrected intensities and the CRM concentrations through the built-in software.

In addition, an in-house standard (STG) is measured daily to check the analytical performance of the XRF. Three additional internal standards (SQ1, SQ3, Ausmon) are used to monitor the long-term (yearly) drift of the device and to correct intensities if necessary. Lower limits of detections (LLD) are calculated as follows:

$$LLD = \frac{3}{S_i} \sqrt{\frac{I_b}{T_b}}$$

where S_i is the analytical sensitivity (Cps/concentration), I_b is the background intensity (in kilocounts per second, Cps), and T_b is the background measurement time (s). It is also possible to recalculate a corrected concentration for each CRM by plotting its corrected intensity on the calibration line. This value was used to monitor the total analytical accuracy for each CRM and each element as follows:

$$Accuracy = \frac{\sqrt{(C_{CRM} - C_{COR})^2}}{C_{CRM}} \times 100$$

where C_{CRM} is the certified concentration and C_{cor} is the corrected (i.e., recalculated) concentration. Finally, reproducibility has been assessed for both calibrations using an external CRM (which is not included in the calibration process) with a similar matrix (NCS-ZC73002, soil). Multiple (n = 10) runs of this CRM provide a reproducibility expressed as follows:

$$reproducibility (\%) = \frac{Average C_{n=10} - Stdeva C_{n=10}}{Average C_{n=10}} \times 100$$

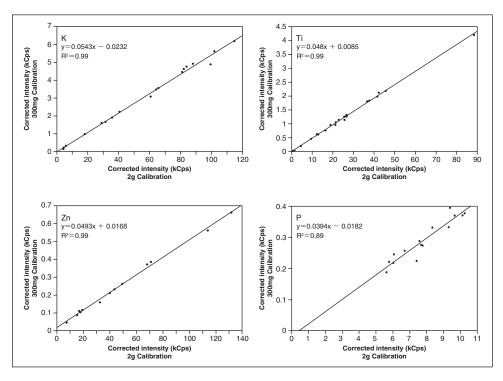


Figure 1. Biplot comparison of intensities (kilocounts per second—kCps) obtained using the large and the small calibrations for K, Ti, Zn, and P. Black dots represent the CRMs.

where *Average* and *Stdeva* represent the mean (n = 10) CRM concentration and standard deviation, respectively. Results, as well as detection limits, accuracies, and reproducibilities are reported in Table I.

RESULTS AND DISCUSSION

Potassium (K), titanium (Ti), zinc (Zn), and phosphorus (P) were chosen to illustrate the comparison of the two calibrations and the reliability of the small-sample calibration. Potassium (major constituent) and Ti (minor constituent) were selected because they represent two important lithogenic elements commonly used in provenance studies. Zinc illustrates the behavior of heavy metals and phosphorus a calibration that displays greater uncertainty. Figure 1 presents biplots of the corrected intensities for both large-cup and small-cup calibrations. Because the number of CRMs for a given element differs between the large-cup and small-cup calibrations, only the intensities of the CRM common to each calibration are plotted together. For most of the studied elements, the CRM intensities plot nicely along a line ($R^2 \ge$ 0.89 for elements in Figure 1), which indicates that decreasing the amount of material results in a linear decrease in the peak intensities. The intensities generally decrease by a factor 15 to 25 from the 2-g to the 300-mg samples. However, both elemental peak and background intensities decrease, allowing for a good peakbackground resolution. This is also supported by the low and comparable LLD found in both calibrations: a few tens of $\mu g g^{-1}$ and a few $\mu g g^{-1}$ for major and trace elements, respectively (Table I). These values are generally below or equal to previously reported data (Cheburkin, Frei, & Shotyk, 1997; Boyle, 2000; Hein et al., 2002; Zambello & Enzweiller, 2002; Cultrone et al., 2010). Similarities between the two calibrations can also be observed for the accuracies, which remain within $\leq 10\%$ for most elements except for S, P, and Br. These accuracy values are slightly below values encountered in the literature (Cheburkin, Frei, & Shotyk, 1997; Boyle, 2000; Zambello & Enzweiller, 2002; Cultrone et al., 2010). Improvements in both LLD and accuracy can be explained by the continuous analytical development of commercially available XRF and the careful selection of appropriate CRMs and WD-XRF optimization. Lower detection limits and accuracies have been reported by Adan-Bayewitz, Asaro, & Giaque (1999), who have used a specific Ag-cathode XRF setup. While such a configuration results in lower LLD and accuracies, it does not allow the determination of all major elements, which is crucial in provenance studies. Moreover, such a configuration is of limited availability compared to largely commercially available Rh-cathode WD-XRF. Compared to the large-cup calibration, reproducibility values are generally 5–10% lower for the small-cup calibration (Table I). This can be explained by the smaller analytical surface (8-mm vs. 36-mm opening), which results in a lower signal integration. However, these reproducibilities are fully satisfactory, with values mostly above 90%, except for a very few elements (e.g., Cu: 78% and Ni: 82%). One notable exception is Br, which shows rather poor reproducibility (47%). This may be explained by (1) the low concentration encountered in the CRM analyzed for this test (2.8 μ g g^{-1} , (2) the relatively low amount of CRM with certified values for Br (n = 8; see Table I), and (3) the low intensities (0-0.15 kilocounts per second) recovered for this element. Bromine concentrations should therefore not be taken as a fully quantified element using the small-cup calibration, but rather as indicative of large variations, which may be taken into account when using multivariate statistics, where covariance is considered rather than absolute values.

The linear relationship between the two calibration intensities also implies that results are independent of the calibration. In other words, the results of samples quantified with one calibration can be compared directly with the results obtained using the other. In the specific context of a provenance study, it is therefore possible to compare source materials where a large amount of material is available, such as soils, rocks, and sediments, with small ceramic sherds (where the amount of material may be limited) using the two different calibrations based on the same CRM, but differing in the amount of material (2 g vs. 300 mg) and analytical surface (36 mm vs. 8 mm). Both major and trace elements display linear behaviors similar to K, Ti, and Zn, with low intercept offsets. Two elements can be accurately quantified using the large-cup calibration. The comparison between the two calibrations is poor because P and Ba are inaccurately quantified using the small-cup calibration. This can be explained by a narrow CRM concentration range for P and bad

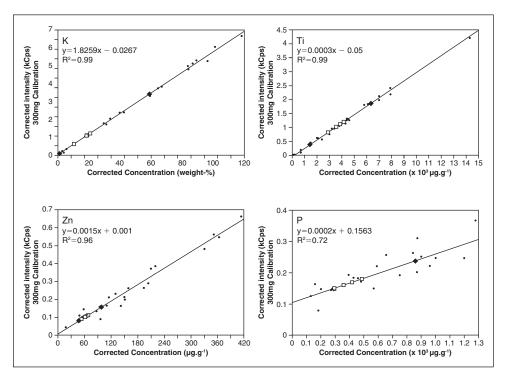


Figure 2. Calibration lines (corrected intensities in kCps vs. CRM concentrations) obtained using the small cups for K, Ti, Zn, and P. Black dots represent the CRM. Open squares represent the pottery sherds. Black diamonds represent two potential raw materials from Cyprus.

peak–background ratios for P and Ba. Caution is therefore required when using these elements in ceramic provenance studies where their concentrations have been measured with the technique described here. However, in provenance studies where multivariate statistical approaches are often used to identify pottery groups and provenances, the loss of two minor elements out of a total suite of 20 elements will not influence significantly the results of multivariate statistical calculations.

The small-cup calibration curves for K, Ti, Zn, and P are reported in Figure 2. To briefly illustrate the applicability of the small-cup calibration to ceramic artifacts, the elemental concentrations of four Cypriot Late Bronze age ceramic sherds and two potential raw materials have been measured; complete analyses of those samples will be presented in a separate publication. The concentration ranges found in the sherds and sediments are well enclosed within the CRM concentration ranges (Figure 2). As shown by the Zn biplots, trace metal concentrations are found to be relatively low in these sherds and sediments. This is also the case for the Pb concentrations ($<15\mu g$ g⁻¹, not shown here), which may potentially limit the applicability of XRF to trace metals in ceramic provenance studies. These considerations regarding elemental chemistry need, however, to be confirmed by the analysis of a wider range of ceramic

artifacts in order to better explore the concentration range of each element using WD-XRF.

One last but major advantage of WD-XRF, besides providing quantitative data on a large amount of elements (here 18 to 20 elements), is that these same powdered samples can be re-used for further analyses such as mineralogy determination by X-ray diffraction, or isotopic analyses. The calibration developed here renders this re-use possible even on very small samples. It should finally be emphasized that while 300 mg of sample powder is sufficient, one should make sure that such a small amount is sufficiently representative of the whole object.

CONCLUSIONS

This study presents a rapid, non-destructive (i.e., the powdered samples can be re-used) WD-XRF protocol developed for small sample masses, down to 300 mg of dry powder. This protocol allows the accurate determination of major (Na, Mg, Al, Si, K, Ca, Fe), minor (Ti, P, S, Mn), and trace (V, Ni, Cu, Zn, Br, Rb, Sr, Ba, Pb) elements. With the exception of three elements (P, Ba, Br), this specific calibration is fully comparable to conventional calibration using larger amounts of sample (2 g). Lower limits of detections and accuracies are comparable to, if not below, those found in the literature using similar XRF designs. Such a small sample calibration developed for pottery sherds may be widely applied in the field of ceramic provenance studies.

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