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# **Optical and Electrical Studies on Crystalline Tin Sulphide**

by

M. Merdan, BSc

Thesis submitted to the  
University of Nottingham  
for the degree of  
Doctor of Philosophy

August 1977



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## ABSTRACT

A Fourier transform far infrared spectrometer has been constructed to perform reflectivity and transmission measurements on small semiconductor samples in the temperature range 10-300 °K. Far infrared and Raman spectra selection rules have been obtained by the Correlation method from the factor group analysis of the SnS subset. Crystals of n- and p- type SnS have been prepared, and the preparation methods are critically reviewed with reference to the T-p-x diagram. The electrical properties (Hall effect, carrier concentration, Hall mobility and barrier height) are measured for these specimens in the temperature range 40-300°K. Methods of obtaining electrical contacts to SnS are exhaustively studied. Broadband photoconductivity and photovoltaic measurements are reported on n- and p-type specimens in the spectral range 0.5-2.1 eV at temperatures from 300°K to 10°K. Band structure and impurity energies are suggested to account for the observed results. The band edge shift with temperature is also measured. Far infrared reflectivity and transmission measurements are made between 300°K and 10°K in order to determine the lattice vibration frequencies. Kramers-Krönig analysis and computer fitting routines are used to obtain this information from the reflectivity data. The interlayer forces are compared with those obtained in the isomorphous materials GeS and GeSe. Similar spectroscopic and analytic techniques have also been used to examine the room temperature optical properties of ZnS+Fe (Marmatite).

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# CHAPTER ONE

## INTRODUCTION

### 1.1 Extent of this thesis

The thesis describes investigations into the electrical and optical properties of crystalline SnS from room temperature down to 10 °K. In an appendix we also present the room temperature infrared transmission and reflection data of marmatite.

We begin with a general review of the far infrared properties of layered semiconductors, giving (for the first time) a group theoretical analysis of SnS subset by means of the correlation method [Fataley et al., 1972], thereby deducing the lattice vibrational modes.

The methods of preparing single crystals of SnS are discussed in detail, with reference to the T - p - x diagram in Chapter Two. We also discuss the preparation of n- and p-type crystalline SnS in a form suitable for electrical and optical investigation.

In Chapter Three, the electrical properties (Hall effect, carrier concentration, Hall mobility and barrier height measurements) are outlined together with a discussion of various contact procedures which have been used to form ohmic and non-ohmic contacts for the electrical and optoelectrical measurements.

In Chapter Four, we report for the first time broad-band (0.5-2.1 eV) photoconductivity and photovoltaic measurements for various temperatures from 300 °K down to 10 °K and

we suggest a band edge structure to SnS which provides an explanation of our photoconductivity data. This explanation is given in the absence of any band edge structure literature for SnS. Appendix A to this chapter describes the apparatus used to perform photoconductivity and photovoltaic effects in the temperature range 10 - 300 °K. The band edge shift with temperature is also measured.

Chapter Five covers the description of apparatus used for far infrared spectroscopy together with the modifications made on the electronic units. The sample cell design is discussed and the optical, electronic and mechanical problems encountered during the far infrared reflection and transmission experiments are reviewed.

Far infrared reflectivity and transmission data on n- and p-type crystalline SnS at various temperatures from 300 °K down to 10 °K are reported in Chapter Six. Lattice vibrational modes of SnS were obtained by means of Kramers-Krönig analysis and least squares-fit computer programs applied to our far infrared reflectivity spectrum. Interlayer forces were discussed in comparison with those in the isomorphic materials GeS and GeSe for which data has been obtained by Wiley et al. 1976, and Chandrasekhar and Zwick 1976, respectively. Appendix B of this chapter is given in order to explain the physical meaning of the least squares-fit computer program. In Appendix C the room temperature optical properties of ZnS+Fe (marmatite) are investigated. The optical constants were obtained by means of similar Kramers-Krönig analysis and least squares-fit computer program.

## 1.2 Layered Compounds

These compounds form a class of molecular crystals in which the molecular unit (the individual layer) is an infinitely extended sheet in two dimensions. These compounds are characterised by highly anisotropic bonding forces and structurally identical layers; the forces between the layers are known to be weak by comparison with the forces within the layer. These materials display a wide range of behaviour varying from almost two-dimensional to almost three-dimensional with a similar variation in the isotropy of their properties.

Recently, there has been an intensive interest in both the electrical and optical properties of layered crystals in order to determine their fundamental physical properties. [Wilson and Yoffe, 1969 and Van der Valk and Haas, 1977, and references in these.] These compounds show a broad variation of electrical properties, ranging from electrical insulators such as  $\text{HfS}_2$  through semiconductors such as  $\text{SnS}$  and  $\text{MoS}_2$  subsets, to metals such as  $\text{NbSe}_2$ , which is a superconductor below 7 °K. [Wieting and Verble, 1971.]

### 1.2.1 General properties of $\text{MX}_2$ type layered crystals.

In order to understand the properties of the MX type layered materials, we first present a short review of  $\text{MX}_2$  systems. The  $\text{MX}_2$  layer materials form a structurally and chemically well defined family. Electrically, however, they cover a wide spectrum of properties as mentioned earlier, from an insulator like  $\text{HfS}_2$ , through semiconductors like  $\text{MoS}_2$ ,  $\text{SnS}_2$ ,  $\text{SnSe}_2$  and semi-metals like  $\text{NbS}_2$  and  $\text{VSe}_2$ . [Wilson and Yoffe, 1969.]

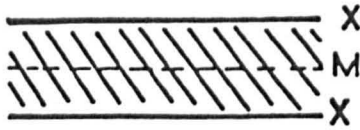
The  $\text{MX}_2$  layer crystal structure consists of layers of the M atoms sandwiched between two layers of the X atoms, as seen, for example, in Figure 1.1. The bonding within the layer is generally thought to be mainly covalent. However, the bonding between neighbouring double layers is most likely to be of the van der Waalstype, although Smith et al., 1977 and Lucovsky et al., 1976, suggest that long range coulomb forces are also found to some extent in these materials, such as  $\text{SnS}_2$  and  $\text{SnSe}_2$ . They also remark that the coulomb interaction is more important in the sulphide than the selenide, due to the larger electronegativity difference between Sn and S. There is no proof in the literature on the bonding force in these materials, that it is indeed of a van der Waals type.

The cleavage properties of the stoichiometric materials are quite remarkable. The crystals are not friable and may be pulled apart on sellotape. Obviously the extremely anisotropic character of the layer compounds, built in at the atomic level, dominates all the mechanical, electrical and optical properties of such materials.

The frequencies of the transverse optical mode parallel and perpendicular to the layers differ by a factor of nearly two. [Clasen et al. 1975, Grisel et al. 1976, Lucovsky et al. 1976a, 1976b.] Recently, Van der Valk and Haas (1977) have published a paper in which they review the anisotropic behaviour of lattice vibrations of layered compounds.

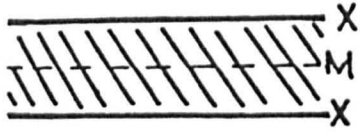
The expansion coefficient is about a factor of 10 in a direction perpendicular to the layers rather than parallel

Fig:1-1: Basic form of layer compounds



a) General Form

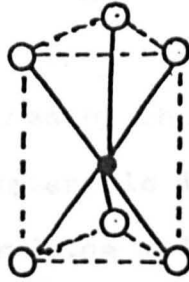
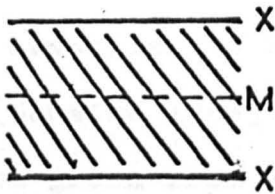
van der Waals gap



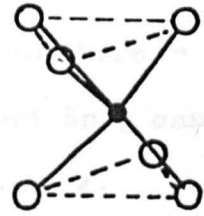
Sandwich of

'Anion' and 'Cation' sheets.

b) Coordination units for  $MX_2$  layer structures.

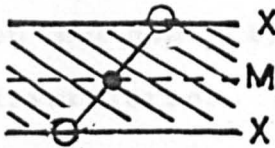


AbA  
trigonal prism

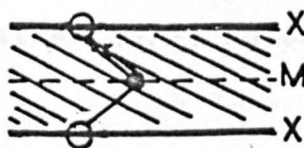


AbC  
octahedron

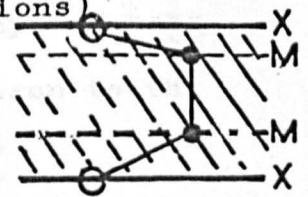
c) Further types of sandwich ( $11\bar{2}0$  sections)



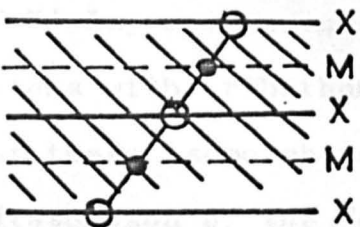
octa.  
 $CdI_2, CdCl_2$   
types.



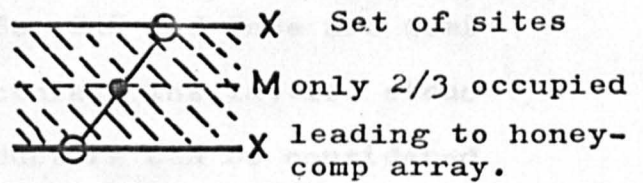
trig. pr.  
 $MoS_2, NbS_2$   
types.



Gas, Gase.

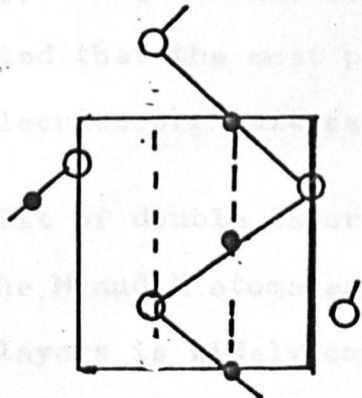


$Bi_2Te_3$ .

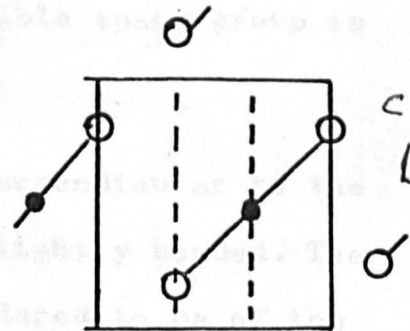


$BiI_3, CrCl_3$ .

$NiAs$   
(BB)



$CdI_2$   
(c6)



Relation of the  $MX_2$  layer structure to MX structures.

to them [Brixner, 1963]. Conversely, the velocity of sound in this latter direction is twice that parallel to  $\vec{c}$ .

[Guseinov and Rasulov, 1966.] The thermal conductivity is also up by almost a factor of 10 parallel to the layers.

The electrical conductivity perpendicular to the planes is down by a factor of at least 100 on that in the planes for  $\text{MoS}_2$  etc.  $\text{MoS}_2$  and  $\text{CdI}_2$  show an excitonic absorption.

Only  $\text{MoS}_2$  occurs naturally in appreciable quantities. The other crystals are best obtained by chemical vapour transport [Schafer 1964]. Layer materials like  $\text{CdI}_2$  and  $\text{SnS}_2$  can also be grown from solution and the melt respectively.

A recurrent theme in all the studies of these materials has been the effects of the much weaker interlayer forces [Zallen and Slade 1974, Zallen 1974] in comparison to the intralayer forces.

### 1.2.2 MX type layered crystals: General Properties

We principally consider the IV-VI group of MX crystals.

The IV-VI layer compounds  $\text{GeS}$ ,  $\text{GeSe}$ ,  $\text{SnS}$  and  $\text{SnSe}$  are semiconductors with orthorhombic structure. The lattice structure of these isomorphous semiconductors can be considered as a distortion of the  $\text{NaCl}$  structure. For  $\text{GeS}$ ,  $\text{GeSe}$  [Dutto and Jeffrey, 1965],  $\text{SnS}$  [Hofmann, 1935] and  $\text{SnSe}$  [Okazaki, 1956] it is reported that the most probable space group is  $D_{2h}^{16}$ , with four molecules per unit cell.

The crystals consist of double layers perpendicular to the  $c$  axis in which the  $M$  and  $X$  atoms are tightly bonded. The bonding of these layers is widely considered to be of the

van der Waals type. However, it may be possible that long range forces of a Coulomb type are also involved. This has been suggested for MX systems such as GaSe by Fischer and Brebner (1962) and for MX<sub>2</sub> crystals such as SnSe<sub>2</sub> and SnS<sub>2</sub> by Lucovsky et al., 1976 and Smith et al., 1977. The evidence for this suggestion is the large local effective charge deduced from infrared measurements.

Every M atom is surrounded by six X atoms and every X atom by six M atoms, so that the bond lengths can be grouped into a set of three almost orthogonal (shorter) M-X bonds and a set of three (longer) M-X bonds.

One can, therefore, expect that this type of bonding will result in lattice dynamical, electrical and optical anisotropy. It has been reported that crystals like SnS display anisotropic electrical and optical properties. For example, the Hall mobility parallel to the c axis is approximately five times less than that perpendicular to the c axis [Albers et al. 1960, Haas and Corbay 1961, and Albers et al. 1962] and the effective mass parallel to the c-axis

$m_{||}^* \approx m_0$ , whereas in a plane perpendicular to the c-axis  $m_{\perp}^* \approx 0.2m_0$  [Haas and Corbay 1961, and Albers et al. 1962].

Vlachos et al. (1976) reported that GeSe has an indirect gap of 1.22 eV  $\vec{E} \parallel \vec{a}$  and 1.18 eV of  $\vec{E} \parallel \vec{b}$  at room temperature.

Wiley et al. (1975) also remark that GeS has a highly anisotropic absorption edge, being directly allowed for  $\vec{E} \parallel \vec{a}$  and  $\vec{E} \parallel \vec{c}$  directions, and forbidden for  $\vec{E} \parallel \vec{b}$ .

Despite the recent investigations on the IV-VI orthorhombic GeS-like semiconductors, many of their physical



constants are still not yet clearly determined.

### 1.3 Far Infrared Properties of Semiconductors

#### 1.3.1 Introduction

The examination of semiconductors in the far infrared can be of interest for several reasons:

Firstly, the optical data can be easily related to the semiconductor transport properties because at these long wavelengths a simple non-quantum mechanical theory holds [Perkowitz and Thorland, 1974].

Secondly, over a wide range of carrier concentration in most semiconductors, the lower branch of the coupled plasmon-phonon mode falls almost wholly in the far infrared and its frequency can be observed at the reflectivity minimum.

Thirdly, the fundamental lattice vibrations - phonon frequencies - can be observed in the far infrared reflectivity and transmission experiments, thereby allowing us to obtain the lattice contribution to dielectric constants by means of Kramers-Krönig analysis and oscillator fits. In addition, the information about the 'interlayer-intralayer' forces in semiconductors can be obtained by performing both the far infrared (reflectivity) and Raman (back scattering) experiments.

The reflection coefficient for a semi-infinite medium in the case of normal incidence is given by

$$R = [(n-1)^2+k^2]/[(n+1)^2+k^2] \quad (1.1)$$

where  $n$  and  $k$  are the real and imaginary parts of the complex index of refraction and

$$n^2 = \epsilon(\omega)$$

where  $\epsilon$  is the complex dielectric function and  $\omega$  is the optical frequency. The dielectric function can be written as the sum of a lattice term and a free carrier term:

$$\epsilon(\omega) = \epsilon_{\text{lat}}(\omega) + \epsilon_{\text{fc}}(\omega).$$

The lattice contribution is given by Born and Huang (1954)

$$\epsilon_{\text{lat}}(\omega) = \epsilon_{\infty} + \frac{(\epsilon_0 - \epsilon_{\infty}) \omega_T^2}{(\omega_T^2 - \omega^2) + i\Gamma\omega}$$

where  $\epsilon_0$  and  $\epsilon_{\infty}$  are the low and high frequency dielectric constants. And we may write

$$\epsilon(\omega) = \epsilon_1(\omega) - i\epsilon_2(\omega)$$

so

$$\epsilon_1(\omega) = n^2 - k^2 = \epsilon_{\infty} + \frac{(\epsilon_0 - \epsilon_{\infty})^2 (\omega_T^2 - \omega^2) \omega_T^2}{(\omega_T^2 - \omega^2)^2 + \omega^2 \Gamma^2}$$

and

$$\epsilon_2(\omega) = 2nk = \frac{(\epsilon_0 - \epsilon_{\infty}) \omega_T^2 \omega \Gamma}{(\omega_T^2 - \omega^2)^2 + \omega^2 \Gamma^2}.$$

The lattice dielectric function exhibits a resonance at the transverse optical frequency  $\omega_T$  with a damping constant given by  $\Gamma$ .

The free-carrier term is given by Lyden, (1964)

$$\epsilon_{\text{fc}} = \omega_p^2 \epsilon_{\infty} \left[ - \left\langle \frac{\mathcal{J}_k^2}{1 + \omega^2 \mathcal{J}_k^2} \right\rangle + \frac{i}{\omega} \left\langle \frac{\mathcal{J}_k}{1 + \omega^2 \mathcal{J}_k^2} \right\rangle \right]$$

where the plasma frequency  $\omega_p$  is defined by

$$\omega_p^2 = \frac{4\pi N e^2}{m^* \epsilon_\infty}$$

where  $N$  is the free electron concentration,  $e$  is the electronic charge,  $m^*$  is the electronic effective mass and  $\tau_k$  is the energy-dependent electron scattering time. The angular brackets indicate averaging over the electron-distribution function.

In recent years a number of papers have been published on the vibrational spectra of SnS subset. For example, GeS [Wiley et al. 1975, 1976, Eymord and Otto 1976, Mihalovič et al. 1976, Gregore and Stetter 1975, Gregore et al. 1976]; SnS [Nikolič et al. 1974, Chamberlain et al. 1976, Nikolič et al. 1977]; GeSe [Siapkas et al. 1976, Chandrasekhar and Zwick 1976]. All these workers have deduced the optical constants of the respective IV-VI orthorhombic compounds by using Kramers-Krönig analysis and oscillator-fit computer programs.

### 1.3.2 Infrared and Raman Spectroscopy - Selection Rules:

With the recent increase in interest in infrared and Raman spectra of crystals, it has become important to determine which vibrational modes are optically active and to know the character of these modes. Hornig (1948), Winston and Halford (1949) and Bhagavantam and Venkatarayudu (1951) pioneered the development of methods for obtaining selection rules which determine this character.

In this section, it is intended to apply space-group selection rules to the analysis of the infrared lattice

spectrum of the SnS subset together with a brief review of 'correlation method' [Fataley et al., 1970] used. This technique has not previously been applied to this material and we present this treatment for the first time.

We concentrate on the use of the correlation tables in obtaining the vibrational selection rules for solids, but we do not review the detailed theory of the method. This has been discussed in several recent papers and books [Winston and Halford, 1949; Bhagavantam and Venkatarayudu, 1951; Slater 1963-67]. The procedure consists in writing down the character table and irreducible representations of the isomorphous point group corresponding to the space groups of the crystal.

The first step in the correlation method is to find out the crystal structure of the sample under investigation. Crystallographic information may be obtained from the international crystallography tables, for example, see Wyckoff 1963-4 and Henry and Lonsdale 1965.

The Bravais space cell is used in the correlation method to obtain the irreducible representation for the lattice vibrations. The crystallographic unit cell may be identical with the Bravais cell, or it may be larger by a simple multiple. This can be ascertained from the capital letter in the X-ray symbol for the crystal structure. For all crystal structures designated by a symbol containing P (for primitive) such as for orthorhombic SnS (Pnma), the crystallographic unit cell and Bravais unit cell are identical.

Crystal structures designated by other capital letters A, B,

C, F, I, R have crystallographic unit cells that contain 2, 2, 2, 4, 2, 3 or 1, [Fataley et al. 1972, p.4] respectively.

Now, we will proceed to use the correlation method for defining the selection rules of the optical modes of the SnS subset. For SnS, the space group is designated  $D_{2h}^{16}$  or Pnma or 62 SnS, with four molecules per Bravais unit cell [Hofmann, 1935]. There are therefore four equivalent Sn atoms and four equivalent S atoms in the Bravais unit cell. We now need to know the possible 'site symmetry' or point groups for the space group  $D_{2h}^{16}$ . The possible site symmetries for this space group  $D_{2h}^{16}$  are given  $2C_i(4)$ ;  $C_s(4)$ ;  $C_1(8)$  [Fataley et al. 1972, Appendix 1, p.173 or Fataley et al. 1971]. Here, the most useful information is the number contained in parenthesis, since it represents the number of equivalent atoms which have that particular site symmetry; for example,  $C_s(4)$  indicates that there are four equivalent atoms occupying sites of symmetry  $C_s$ .

Some of the site symmetries have numerical coefficients, such as  $2C_i(4)$ . The coefficient two shows the presence of two different and distinct kinds of  $C_i$  site in this unit cell. Each can accommodate four equivalent atoms.

Now, consider the case of SnS. First, four equivalent Sn atoms can be accommodated in the possible site symmetry. There are two possible site symmetries,  $C_i$  and  $C_s$ ; both of which can accommodate four equivalent atoms. One or the other will be correct, but additional information is needed to make the choice. For this we turn again to the crystallographic tables, which state [Wyckoff, 1964] that the Sn atoms

lie on the  $C_s$  site.

If we look [Wyckoff, 1964] for SnS

$$V_h^{16}(\text{Pbnm}) \quad (4c) \pm (uv\frac{1}{2}; \frac{1}{2}-u, v+\frac{1}{2}, \frac{1}{2})$$

is seen.

We can write the following alphabetical ordering from the above data.

Table 1

Site	Alphabetical order	Wyckoff notation or alphabetical ordering of site position	Atom
$2C_i(4)$	$C_i(4)$	a	
	$C_i(4)$	b	
$C_s(4)$	$C_s(4)$	c	Sn and S (7c)
$C_1(8)$	$C_1(8)$	d	

Now we have enough information to form the correlation between site symmetries and factor group analysis.

Table 2

The correlation for the lattice vibrations of the Sn atoms in SnS crystals  
between the Site Group  $C_s$  and the Factor Group  $D_{2h}$ .

$f^\gamma$	$t^\gamma$	$C_s$ site species, $\gamma$	Correlation $\sigma(zx)$	$D_{2h}$ factor group species $\alpha$	$C_\alpha$	$\frac{a_\gamma}{a_\alpha} = a_{A'} + a_{A''}$
8	2(Tx,y)	A'	→	A <sub>g</sub>	1	2
				B <sub>2g</sub>	1	2
				B <sub>1u</sub>	1	2
				B <sub>3u</sub>	1	2
4	1(Tz)	A''	→	B <sub>1g</sub>	1	1
				B <sub>3g</sub>	1	1
				A <sub>u</sub>	1	1
				B <sub>2u</sub>	1	1

Note: Table 2 was formed using the following information:

$t^\gamma$  is the number of translations in a site species  $\gamma$ ;

$f^\gamma$  - the degrees of vibrational freedom present in each site species  $\gamma$  for an equivalent set of atoms, ions or molecules. This can be calculated  $f^\gamma = t^\gamma \cdot n$ , where  $n$  is the number of atoms in an equivalent set.

$a_\gamma$  represents the degrees of freedom contributed by each site species  $\gamma$  to a factor group species  $\alpha$ . The value of  $a_\gamma$  can be calculated as follows:

$$f^\gamma = a_\gamma \sum_{\alpha} \gamma C_{\alpha},$$

where  $C_{\alpha}$  is the degeneracy of the species  $\alpha$  of the factor group. The usual values of  $C_{\alpha}$  are summarised in Table 3.

Table 3

<u>Species</u>	<u>Value of <math>C_{\alpha}</math></u>
A	1
B	1
E	2
F	3
G	4
H	5

The total irreducible representation of each equivalent set of atoms in the  $\Gamma_{eg}$  set is constructed in the following manner:

$$\Gamma_{eg} \text{ set} = \sum_{\alpha} a_{\alpha} \cdot \alpha$$

where  $a_{\alpha}$  is the number of lattice vibrations of the equivalent set of atoms in the species  $\alpha$  of the factor group.

The total irreducible representation of the crystal,  $\Gamma_{crys.}$ , can be constructed as follows:  $\Gamma_{crys} = \Gamma_{eg} \text{ set } 1 + \Gamma_{eg} \text{ set } 2 + \dots$



This irreducible representation of  $\Gamma_{\text{crys}}$  contains the acoustical vibrations, so:

$$\Gamma_{\text{cryst.vib}} = \Gamma_{\text{cryst}} - \Gamma_{\text{acoustic}}.$$

Now for Sn atoms in  $D_{2h}$ ,

$$\Gamma_{\text{eg.set 1}}^{(\text{Sn})} = 2.A_g + 2.B_{2g} + 2.B_{1u} + 2.B_{3u} + B_{1g} + B_{3g} + A_u + B_{2u}.$$

If we repeat this procedure for S atoms we will get the same results as for Sn atoms. Because S atoms are also accommodated in  $C_s$  site symmetry, and the number of S atoms is equal to Sn atoms in SnS, then

$$\Gamma_{\text{eg.set 2}}^{(\text{S})} = 2.A_g + 2.B_{2g} + 2.B_{1u} + 2.B_{3u} + B_{1g} + B_{3g} + A_u + B_{2u}.$$

Thus,

$$\begin{aligned} \Gamma_{\text{snS crystal}} &= \Gamma_{\text{eg.set 1}} + \text{set 2} = \\ &= 4.A_g + 4.B_{2g} + 4.B_{1u} + 4.B_{3u} + 2.B_{1g} + 2.B_{3g} + 2.A_u + 2.B_{2u} \end{aligned}$$

is obtained.

But

$$\Gamma_{\text{cryst.vib}} = \Gamma_{\text{crys}} - \Gamma_{\text{acoustic}}$$

and the acoustical modes are readily identifiable in factor groups, since they have the same character as the translation. Table 4 shows this identification. (See next page.)

Table 4 gives the following selection rules for the order of Raman and infrared active modes in SnS:

Raman spectrum: 12 fundamental lattice vibrations allowed, four  $A_g$ , four  $B_{2g}$ , two  $B_{1g}$  and three  $B_{3g}$ .

Table 4

$D_{2h}$  Factor Group Species, Translations, Acoustical Modes, Number of Lattice Vibrations  
and Infrared and Raman Activity of SnS Crystal

$D_{2h}$ factor group species	Translation species	Acoustical mode species	$\Gamma$ SnS crystal coefficients	$\Gamma$ SnS.vib coefficients	Infrared activity	Raman polarization tensor species	Raman activity
$A_g$			4	4		$\alpha_{xx}, \alpha_{yy}, \alpha_{zz}$	✓
$B_{2g}$			4	4		$\alpha_{xz}$	✓
$B_{1u}$	$T_z$	1	4	3	✓		
$B_{3u}$	$T_x$	1	4	3	✓		
$B_{1g}$			2	2		$\alpha_{xy}$	✓
$B_{3g}$			2	2		$\alpha_{yz}$	✓
$A_u$			2	2			
$B_{2u}$	$T_y$	1	2	1	✓		

Infrared spectrum: 7 fundamental lattice vibrations allowed; three  $B_{1u}$ , three  $B_{3u}$  and one  $B_{2u}$ .

Two vibrations,  $A_u$ , will be inactive in both the infrared and the Raman spectra.

If we look at the above result obtained for SnS subset, it is seen to be the same as that obtained by Zeyher et al. [unpublished, 1976] who used a different notation and a different method to obtain the mode assignments and selection rules.

Kress et al. (1976) reported that the optic active modes split into Raman and infrared active vibrations due to a centre of inversion between the layers. This splitting vanishes if the interlayer forces are neglected, since the diperiodic group [Wood, 1974] DG32 has no inversion centre. Recent measurements [Wiley et al. 1976, Chandrasekhar and Zwick 1976] have shown that many of the infrared active modes are nearly degenerate with the Raman active modes. This kind of splitting has also been observed by Verble and Wieting (1970), Wieting and Verble (1971), Zallen et al. (1971). These researchers have carried out infrared and Raman experiments for the layered compounds  $MoS_2$ ,  $As_2S_3$  and  $As_2Se_3$  and have reached the same conclusion as Kress et al. (1976).

The above 'correlation method' for selection rule analysis is seen to be very simple and unambiguous.

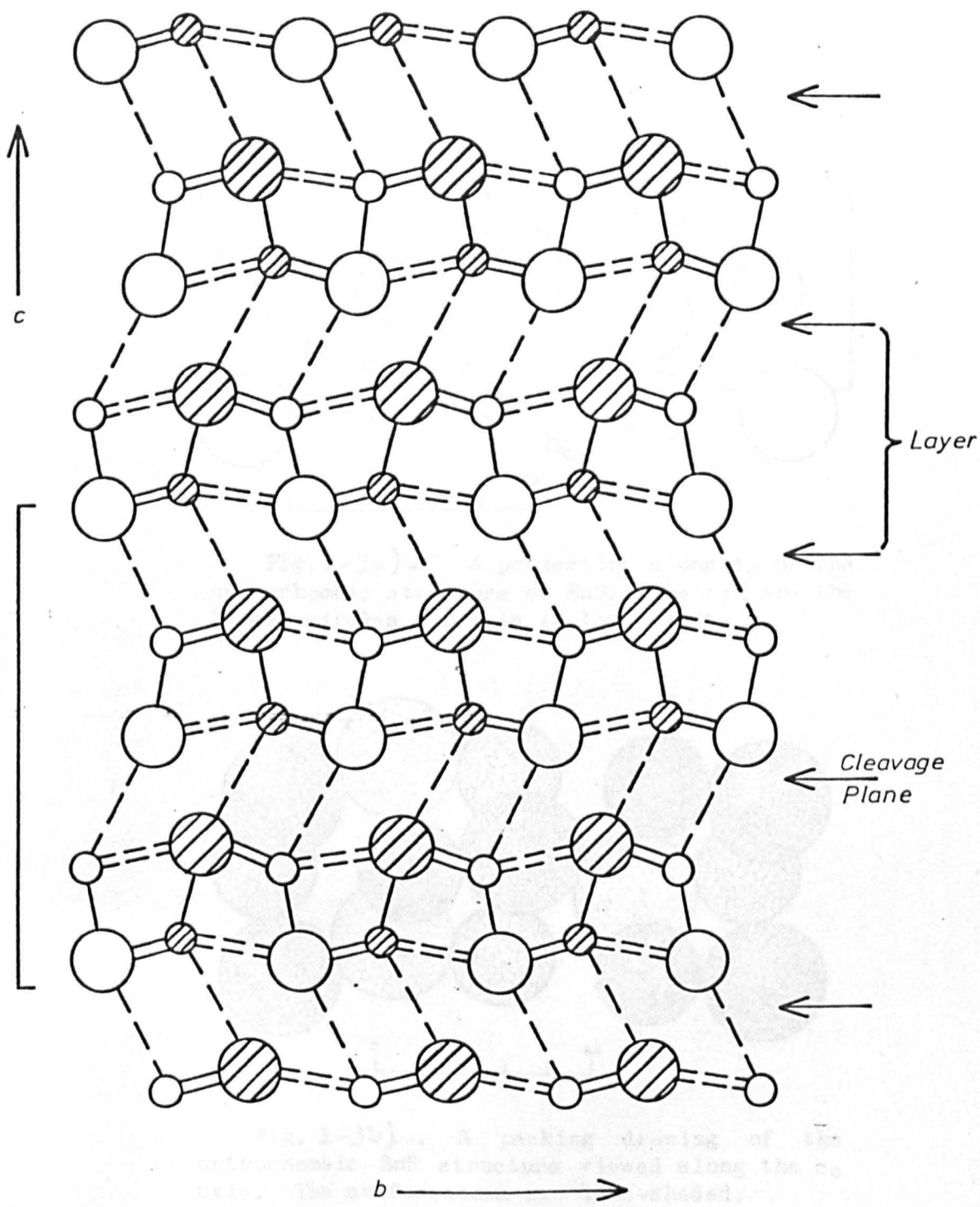
#### 1.4 Review of SnS

##### 1.4.1 Crystal structure and bond structure! SnS crystallises

in orthorhombic ( $D_{2h}^{16}$ ) structure [Hofmann, 1935] with lattice parameters  $a = 3.98 \text{ \AA}$ ,  $b = 4.33 \text{ \AA}$ ,  $c = 11.18 \text{ \AA}$ . There are four molecules in the unit cell. This structure, which is pseudo-tetragonal [Albers et al. 1960], may be thought of as a rather strongly distorted sodium chloride structure. Every Sn atom is surrounded by six S atoms, three at a short distance ( $2.68 \text{ \AA}$ ) with the interatomic directions almost perpendicular to each other ( $88^\circ 10'$ ,  $88^\circ 10'$  and  $95^\circ 8'$ ), and three atoms at a somewhat longer distance ( $3.88 \text{ \AA}$ ) with the interatomic directions at angles  $118^\circ$ ,  $118^\circ$  and  $75^\circ$  respectively. The interatomic distances of the second group form an angle of  $78^\circ$  with the interatomic directions of the first group of S atoms [Lambros et al. 1974].

Several papers [Albers et al. 1960, 1961; Albers and Schol 1961; Nikolić et al. 1974, 1976; Chamberlain et al. 1976] have reported the growing of small crystals of SnS from the melt. Albers and Schol (1961) have given the T-p-x phase diagram of the Sn-S system (this will be discussed in Chapter 2).

The crystal consists of double layers perpendicular to the c-axis in which the Sn and S atoms are tightly bound. The bonding between these layers is much weaker, having a (001) cleavage plane. This structure represents an intermediate state between a true layer structure and a three-dimensional crystal. The crystal structure of SnS is shown in Figs 12 and 13. Unfortunately there is no literature on the band structure of SnS. However, we tentatively suggest a band structure in Chapter 4 in order to explain our photo-







-  Sn in 1/4
-  Sn in 3/4
-  S in 1/4
-  S in 3/4

Fig:1-2) Crystal structure of SnS after Hofmann

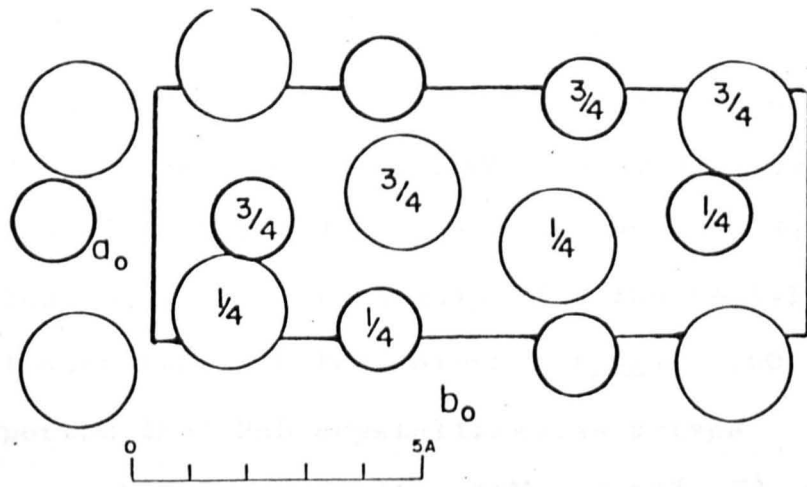


Fig. 1-3a). A projection along  $c_0$  of the orthorhombic structure of SnS. The tin are the larger circles. Origin in lower left.

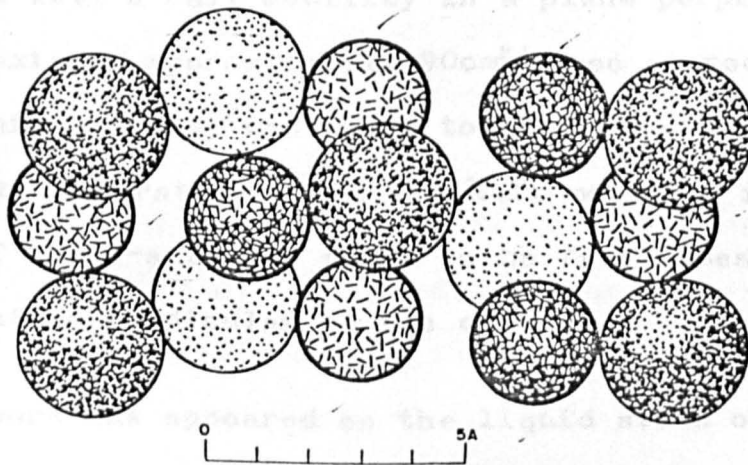


Fig. 1-3b). A packing drawing of the orthorhombic SnS structure viewed along the  $c_0$  axis. The sulfur atoms are line-shaded.

(After Wyckoff R.W.G. Vol.I-1960)

conductivity data.

#### 1.4.2 Electrical Properties

Anderson et al. (1943, 1945a, 1945b) have studied the electrical conduction and the thermoelectric power [Anderson et al. 1947] of SnS pellets. They found p-type conductivity and a band gap approximately 1.2 eV at room temperature. Braithwaite (1951) found photoconductivity in SnS powder with a long wavelength threshold of  $1.1\mu$  ( $\approx 1.13$  eV) again at room temperature. Later, Albers et al. (1960, 1961, 1962) reported that SnS crystallises as p-type material with a carrier concentration  $10^{17}$  and  $10^{18}$   $\text{cm}^{-3}$  with Hall mobility  $\mu_H$  of  $65\text{cm}^2/\text{v}\cdot\text{sec}$  at room temperature. They also performed Hall effect and mobility experiments on oriented samples with a Hall mobility in a plane perpendicular to the c-axis of approximately  $90\text{cm}^2/\text{v}\cdot\text{sec}$  at room temperature. This mobility was found to be proportional to  $T^{-2.2}$  at higher temperatures, and the Hall mobility in the direction of the c-axis was found to be five times smaller than that perpendicular to the c-axis.

Some published work has appeared on the liquid state of SnS by Boutin and Bourgon (1961) who concluded that SnS was a semiconductor in the liquid state, having a band gap of 1.9 eV. The conductivity was  $24\text{ohm}^{-1}\text{cm}^{-1}$  at  $895^\circ\text{C}$  and  $31.2\text{ohm}^{-1}\text{cm}^{-1}$  at  $930^\circ\text{C}$ , as obtained by a.c and d.c conductivity measurements. Recent work on the liquid state of SnS has been published by Handfield et al. (1966) who concluded that in the liquid state it forms a similar structure to that of the solid.

### 1.4.3 Optical Spectra

Gobrecht and Barschat (1957) studied the infrared transmission, photoconductivity and reflective spectra of layers of SnS, and estimated a band gap of 1.25 eV at room temperature. Albers et al. (1960) deduced the bandgap to be  $1.02 \pm 0.04$  eV at room temperature from the infrared transmission measurements on cleavage plates perpendicular to the c-axis and obtained an effective mass of the holes of  $m^* = 0.4 m_0$ . A later paper by Albers et al. (1961) reports that the indirect bandgap was 1.08 eV at 300°K and 1.115 eV at 77°K as obtained by absorption measurements. The effective charge of the atoms  $e^* = 0.7 e_0$  and the effective mass of the holes in the three principal crystal directions ( $m_a^* = m_b^* = 0.2 m_0$ ,  $m_c^* \approx m_0$ ) were determined from reflection measurements in the infrared. They remark that the number of equivalent maxima of the valence band is at least four. Haas and Corbay (1961) investigated infrared reflection spectra of SnS in the wavelength region 2 to 25  $\mu\text{m}$  ( $5000 - 400 \text{ cm}^{-1}$ ) and reported that the observed dispersion was a combined effect of lattice vibrations and free holes. They calculated that the index of refraction  $n_0 = 3.6 \pm 0.1$ , the dielectric constant  $\epsilon = 19.5 \pm 2$ , the effective charge on the atom  $e^* = 0.7 e_0$ , and the hole effective mass  $m_1^* = 0.2 m_0$ ,  $m_c^* \approx m_0$ . These values were in agreement with the previous work of Albers. Recently, Takahashi et al. (1972) have studied the optical properties of single crystals of SnSe and SnS at 300°K and 77°K by measuring the absorption spectra along the three principal axes in the energy range 0.1 to 1.3 eV. They found



the absorption bands were around 0.33 eV and 0.88 eV for  $\vec{E}_{11a}$ , around 0.83 eV for  $\vec{E}_{11b}$ , and around 0.33 eV and 0.88 eV for  $\vec{E}_{11c}$ . They attributed these absorption bands to free carrier absorption. More recently Lambros et al. (1974) performed optical absorption experiments on crystalline SnS in the wavelength range 2.2 - 0.8  $\mu\text{m}$  ( $\approx 0.56$ -1.55 eV) at temperatures between 100 - 300°K. They found indirect band-gaps of 1.142 and 1.095 eV for  $\vec{E}_{11a}$   $\vec{E}_{11b}$  at room temperature. They also mentioned that the phonon assisted transitions with energies 0.033 or 0.038 eV and 0.082 or 0.113 eV, with reference to the two main axes  $\bar{a}$  and  $\bar{b}$ . Nikolić et al. (1974) performed far infrared reflectivity experiments over the wavelength range 0.5 to 200  $\mu\text{m}$ . They derived the optical constant of SnS from a Kramers-Krönig analysis and calculated the transverse optical and longitudinal optical frequencies at room temperature. In this connection we reported (Chamberlain et al. 1976) far infrared reflectivity and transmission data using both polarised and partially polarised light in the three main directions, in the temperature range 10 - 300°K. The most recent work appearing on Raman scattering of single crystals of SnS at room temperature [Nikolić et al. 1977) obtains values for the Raman active modes in this material.

## CHAPTER TWO

### DESCRIPTION OF SPECIMEN PREPARATION

#### 2.1 Introduction

Single-crystalline samples of the majority of the IV-VI compounds can be prepared by melting suitable amounts of the components in evacuated quartz ampoules. In the preparation of SnS, GeS, PbS etc., where the reacting elements have a high vapour pressure, it is necessary to increase the furnace temperature very slowly, and to allow considerable time for reaction. This may take at least a day to bring the temperature to the melting point of the compound in order to avoid explosions, which are usually due to the vapour of free unreacted sulphur. Obviously, clean conditions and a dust free atmosphere are essential in this work. Another advisable precaution is to ensure that the reacting metal is cut into small pieces in order to make a good mixture. Soft metals such as In, Pb and Sn may be cut with a sharp knife, and it is advisable to etch the metals to be used after cutting. Suitable etchants for metals are to be found in various metallurgical handbooks (Smithells, 1955, Lyman, 1948).

Preparations of SnS specimen have been reported previously by a few workers (Albers et al, 1960; Haas and Corbay, 1961; Rau, 1966; Takahashi et al, 1972; Vujatović and Nikolić, 1976). The techniques of Albers' group involve a double distillation and introduction of compensating impurities, finally producing p-type specimens of  $\sim 5 \times 10^{16} \text{ cm}^{-3}$  (300 °K) for use in optical work. We have used techniques developed by Vujatović and Nikolić which result in higher mobility samples but with impurity concentration marginally greater than those achieved by Albers for his optical specimens, although the samples used for their electrical measurements are virtually identical with our own ( $10^{17}$ - $10^{19} \text{ cm}^{-3}$ ). Our sample growing technique will be discussed in next section. We now review for completeness the method of Albers et al.

The technique of the Albers' group was to prepare SnS in a quartz ampoule from components taken in the stoichiometric ratio. The ampoule was heated to 900 °C, while one part of the tube was kept at 400 °C in order to avoid explosions. The whole tube was heated up slowly to 900 °C. A molten zone was passed through the sample at a rate of about 1 cm per hour in order to obtain single crystals. The resultant SnS contains various impurities (Fe, Pb, Al, Sb, Ni, Mn, Cu) with a total concentration of the order of  $10^{18}$  to  $10^{19} \text{ cm}^{-3}$ . The impurity content is then chemically reduced (Albers et al, 1961) by means of a double distillation at 960 °C. Spectro-chemical analysis of this purified SnS showed that

the concentration of all detectable elements was below the limits of detectability with the exception of Pb, which was still present at a concentration of  $7.10^{16}$  at/cm<sup>3</sup>.

## 2.2 Growth Technique

The single crystal samples of SnS were produced from the pure elements by means of the Bridgman technique as described previously by Vujatović and Nikolić, (1976). The stoichiometric amount of the constituent elements (manufactured by Koch-Light, 99.9999 o/o grade) were weighed out. The total weight was always taken to be less than five gr in order to avoid explosions due to over pressure of sulphur at 900 °C (approximately 100 atmosphere, Albers et al. 1960). Sulphur was powder like and Sn was cut into small pieces with a clean sharp knife, and then the components were put together in a 12 mm diameter, thick wall (~ 1.5 - 3 mm) quartz ampoule as seen in Figure 2.1. The lower part of the ampoule had a conical shape and was about 35mm long. The ampoules were then evacuated by means of a rotary-diffusion pump system. When the pressure reached  $10^{-5}$  torr or better, the ampoule was sealed off.

Single crystals of SnS were grown in two stages as follows: in the first stage, the ampoule, containing the constituent elements, was placed in a cold furnace and heated up at a rate of about 25 °C/hr until the temperature reached 1000 °C. This is well above the melting points of both elements but it is essential to work at this

temperature in order to increase the efficiency of the reaction. Normally, this temperature was kept for 2-3 hours before cooling down gradually to room temperature. In the second stage, those ampoules with polycrystalline SnS were placed in a second quartz tube in order to prevent oxidation in case of possible breaking of the first ampoule during the solidification process. The second ampoule was evacuated and sealed off as before when the pressure was  $10^{-5}$  torr or better. The resultant double-walled quartz ampoules were placed in an initially cold vertical furnace whose final temperature profile is shown in Figure 2.2.

The furnace was heated up at a rate of  $25^{\circ}\text{C/hr.}$  until this profile was achieved, and the sample temperature was  $1000^{\circ}\text{C.}$  This temperature was kept constant for several hours. Then the ampoule was lowered down into the cooler zone at an initial rate of 2 cm/day until the melt solidifies (approximately  $870^{\circ}\text{C.}$ ); the rate was then increased to 24 cm/day until the sample was cooled down to room temperature. The process typically took one week.

### 2.3 Results and Conclusions

In order to grow single crystals under controlled conditions it is necessary to have a knowledge of the phase diagram (T-p-x) of the system involved.

The T-p-x-diagram of the Sn-S system was investigated by Albers and Schol (1961). The T-x relation of this diagram is given in Figure 2.3. There are two large regions of liquid immiscibility. One of these occurs on the Sn-rich side (10-48 at o/o S) and extends to

Fig.2-1) Quartz ampoule to  
grow SnS.

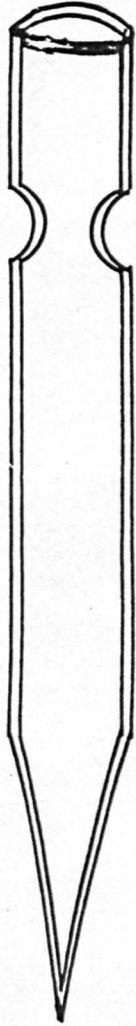
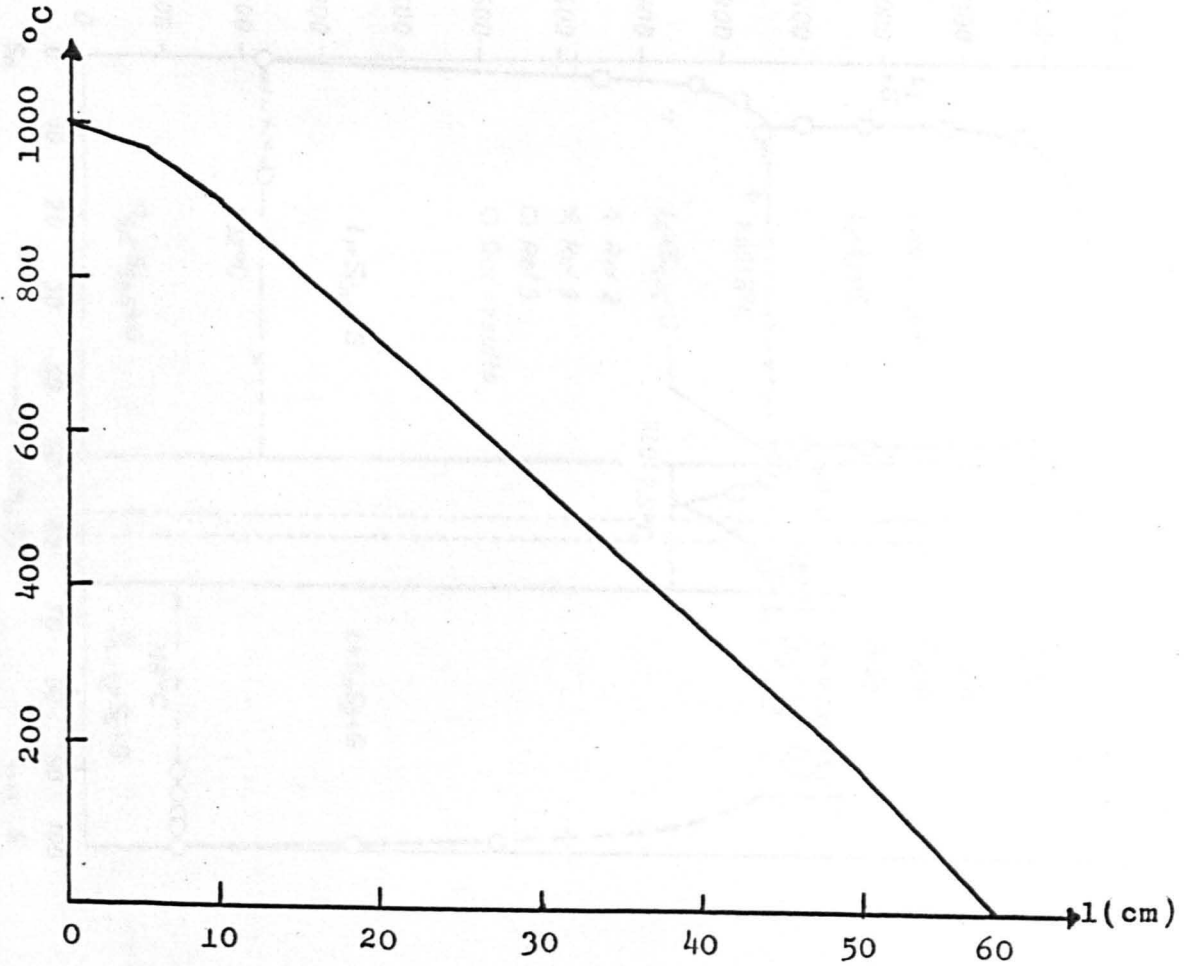


Fig.2-2) Temperature profile of the furnace used for  
growing the single crystals.



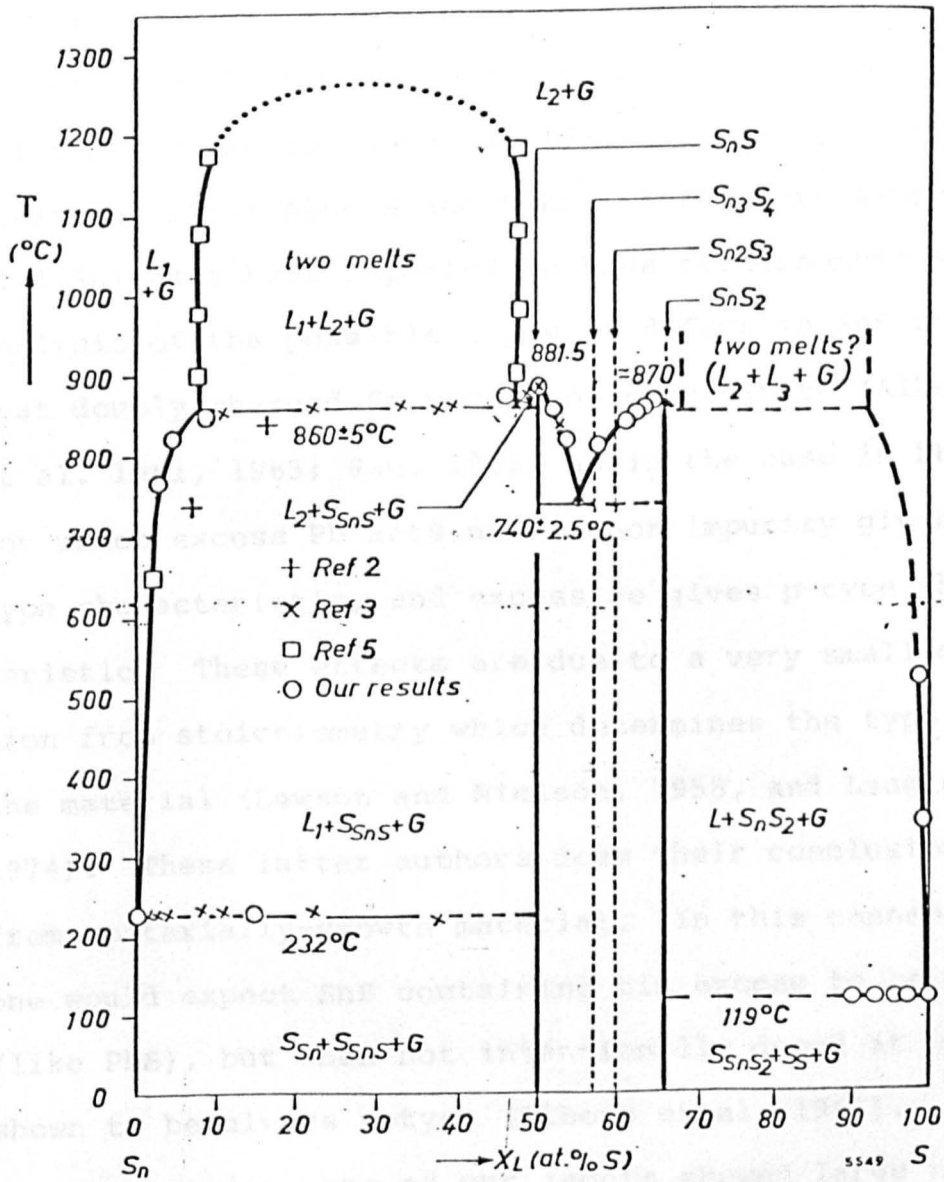


Fig.2-3)  $T-X_L$  Projection of the system Sn-S  
 (After Aibers and Schol, 1961)

An electron-diffraction investigation of SnS films (Badachhape and Goswami, 1964) deposited on substrates kept at 300-350 °C has shown that SnS crystallises in the cubic NaCl-type lattice with  $a=5.16 \text{ \AA}$ . However, solid SnS crystals (Hofmann, 1935) and  $\text{Sn}_2\text{S}_3$  (Mootz and Kunzmann, 1962; Albers and Schol, 1961; Karakhanova et al 1966) have been reported to have orthorhombic structure. Analysis of the possible types of defect in SnS shows that doubly charged Sn vacancies predominate (Albers et al. 1961, 1963; Rau, 1966) as in the case in PbSe, for which excess Pb acts as a donor impurity giving n-type characteristic, and excess Se gives p-type characteristic. These effects are due to a very small deviation from stoichiometry which determines the type of the material (Lawson and Nielson, 1958, and Laugier, 1974). These latter authors drew their conclusions from epitaxially-growth material. In this connection, one would expect SnS containing tin excess to be n-type (like PbS), but when not intentionally doped it has been shown to be always p-type (Albers et al. 1961). However, on one occasion, one of our ingots showed large n-type regions as measured by a thermopile method. Unfortunately, the process was unrepeatable; the n-type character was probably due to Sn excess in a similar fashion to PbSe as discussed earlier.

Our ampoules had a conical tip, so that initially only a small volume of the melt is supercooled. The object of using this design is to ensure that only a few nuclei (ideally only one) will be formed initially. Although we had no control of the orientation of the



the quadruple point at 860 °C, where solid SnS co-exists with a liquid phase having a composition close to SnS and an Sn excess. Another occurs on the S-rich side (68-95 at. o/o S), where solid SnS<sub>2</sub> co-exists with a liquid phase having a composition close to SnS<sub>2</sub> but with liquid sulphur excess.

SnS and SnS<sub>2</sub> are both found in the Sn-S system: their melting points are 881.5±2.5 °C and 870±2.5 °C, respectively. SnS and SnS<sub>2</sub> form a eutectic at 55 at. o/o S with a melting point of 740±2.5 °C. It is concluded that in the Sn-S system, Sn<sub>2</sub>S<sub>3</sub> and Sn<sub>3</sub>S<sub>4</sub> compounds may also exist. (Gerassimov et al. 1937; Rok and Poeyens, 1957; and Albers and Schol, 1961). Sn<sub>2</sub>S<sub>3</sub> and Sn<sub>3</sub>S<sub>4</sub> are formed by a peritectic reaction at 745 °C and 710 °C respectively. A subsequent re-investigation of Sn-S system by Karakhanova et al (1966) was made in the 40-62 at o/o S range in order to study in more detail the problem of the existence of the Sn<sub>2</sub>S<sub>3</sub> and Sn<sub>3</sub>S<sub>4</sub> phases. In this work it is reported that the melting point of SnS is 875±5 °C which is in agreement with earlier investigations within the limits of experimental error. SnS exhibits a polymorphic transition (Karakhanova, 1966) in the region of temperature 585-600 °C. The most recent investigations are by Abrikosov et al. (1969), who also report this effect in the heating and cooling curves of the alloys in the 40-56 at o/o S range. Measurements of the specific heat of SnS were made by Orr and Christensen (1958), who also reported this polymorphic transition at 602 °C.

first single crystal seed, the resultant ingot was always  $20-25^\circ$  off one of <sup>the</sup> main orientation directions ( $\vec{a}$  or  $\vec{b}$  axis). Because of the cracking of the ampoules during the solidification of the IV-VI group crystals of the SnS subset, we always used double-walled evacuated ampoules. Similar effects have been reported by Yahumoto (1958); Asanabe and Okazaki (1960); and Asanabe (1959) for GeS, GeSe and SnSe respectively. We noted that twinning (mostly simple twinning) occurs frequently as reported earlier by Hofmann (1935) and Albers et al (1960). However, it must be remarked that twinning can be minimised by using narrow ampoules, less than 10 mm inner diameter. Thus the crystal can be grown as one solid phase at one orientation. X-ray Laue diagrams of SnS crystals yielded pin point spots with no trace of spreading; these pin points also observed by Vujatović (1976). All crystals obtained show a perfect cleavage on the (040) face. Each time approximately 5 gr SnS crystals were obtained for the subsequent electrical and optical measurements which will be reported in later chapters. We must remark that it is relatively difficult to achieve a high degree, and a large area of homogeneity of SnS. Several etchants were used in order to improve sample surfaces including that which is given by Nikolić et al 1974. (100 ml ethanol + 12 ml HCl + 6 ml HNO<sub>3</sub> at 60 °C for 15 min.). Only limited success, however, was obtained with this etch.

## CHAPTER THREE

### ELECTRICAL MEASUREMENTS ON CRYSTALLINE TIN SULPHIDE

#### 3.1 Introduction

In this chapter techniques for the provision of electrical contacts to homogenous samples of n- and p- type tin sulphide are established. The characteristics of these contacts are examined, and measurements of barrier heights are reported for a number of contact materials. In addition, some Hall-effect and conductivity measurements which were undertaken to characterise the samples prior to optical studies are discussed.

A number of procedures for making electrical contacts to tin sulphide will now be discussed.

#### 3.2 Contact Technique

Soldered, silver paste (painted), alloyed and welded contacts onto crystalline n- and p- type SnS specimens were made by using ordinary (lead-tin) solder, silver paint, indium pellets, tin and lead pellets. All samples were freshly cleaved from the ingot unless otherwise stated. The various methods are now considered in turn.

### 3.2.1 Welded Contacts

38 S.W.G. platinum wires were used to form welded contacts onto SnS by using a condenser discharge. In many cases platinum wire diffused into the specimens or the material was physically damaged by the discharge. The contacts produced by this technique were weak mechanically, and it was concluded that welded contacts should be avoided if possible.

### 3.2.2 Soldered Contact

The second method involved attaching leads to SnS using ordinary commercial solder (lead-tin) and an acid flux. The contacts made by this method were not particularly reliable or reproducible and varied from sample to sample, in some cases showing blocking behaviour and in other cases non-blocking behaviour.

### 3.2.3 Silver Paint

Silver paint contacts were made in order to measure D.C. conductivity of SnS. Those contacts were mechanically quite strong between 77 °K and 700 °K. However, they were noisy and showed blocking behaviour. Because of this they could not be used for D.C. conductivity measurements.

### 3.2.4 Alloyed Contacts

With this method, metal-SnS contacts were made with a variety of metals and their voltage-current characteristics were examined (Nikolić et al., 1976).

Two similar methods were used to make the alloying contacts. The apparatus used in the first technique is shown in Fig. 3.1. The heater is wound out of Kanthal wire and has a resistance of 100 ohm. The heater is powered by a variac, and the sample is placed on a thin quartz plate over the heater. The sample temperature is monitored with a chromel-alumel thermocouple placed below the quartz plate. A silica tube arrangement is placed over the whole sample/heater assembly.

Prismatical samples with dimensions of about 15 x 4 x 0.3 mm were cut from the cleaved thin plate. Indium pellets with diameter of 0.8 mm were divided into two pieces. The In pieces were etched with diluted  $\text{HNO}_3$ , rinsed with acetone and then washed with plenty of distilled water. The hemispherical In pellets were placed on the SnS and pressed slightly against SnS with match sticks. The specimens were then placed on the quartz plate. Before starting to heat, pure Argon gas was flushed through the jar in order to remove away remaining air in the container. The heater was then turned on, and the temperature increased at a rate of about 50 °C per minute. When the temperature reached 150 °C, HCl vapour was introduced by bubbling pure Argon gas

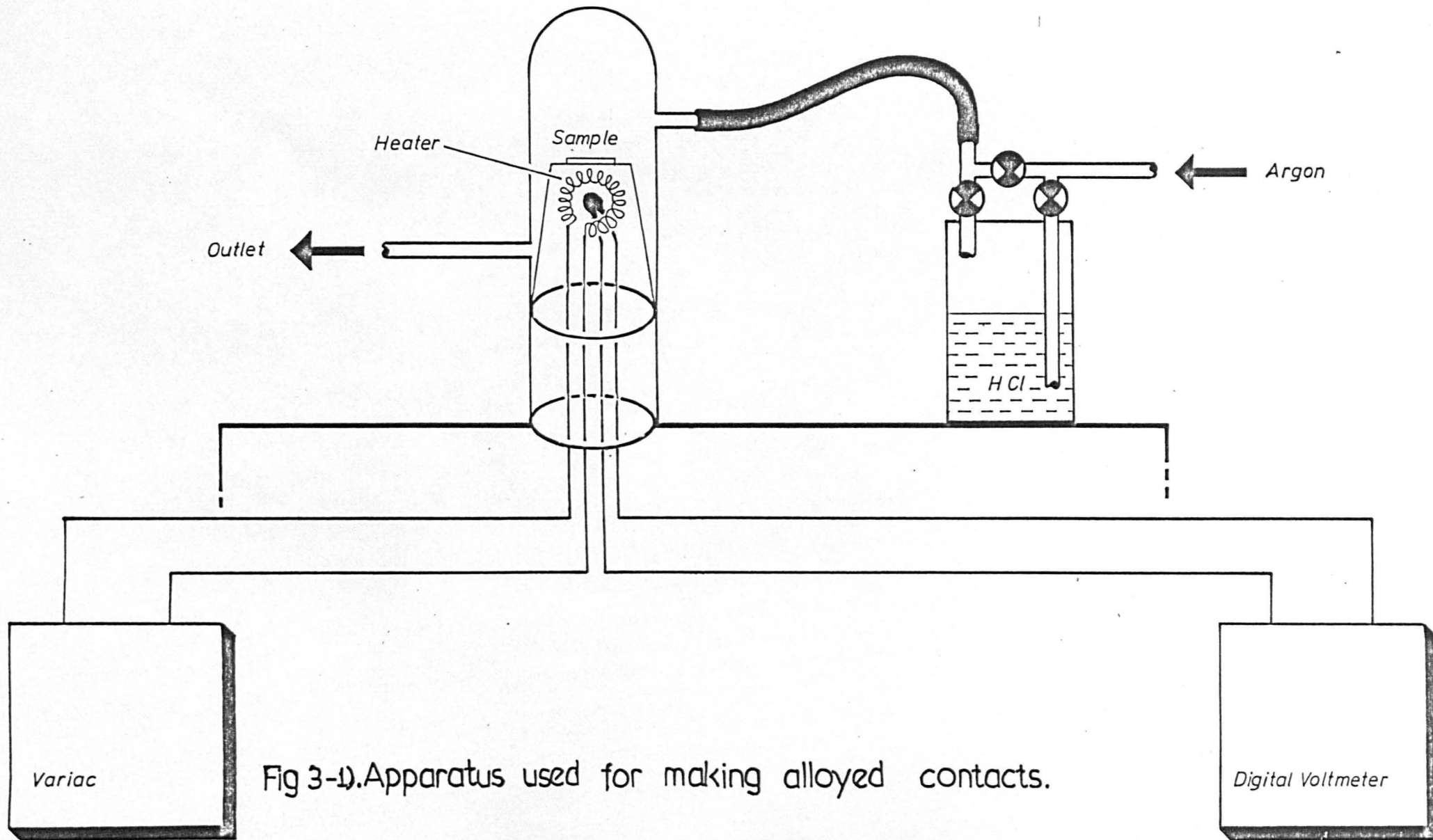


Fig 3-1).Apparatus used for making alloyed contacts.

through HCl acid. When the temperature reached 350 °C it was held constant for about a minute before cooling the specimens.

In the second method, the sample is alloyed in a vacuum in the bell jar of an evaporator. With this method, In, Sn, and Pb contacts were made to SnS. Small (0.8 mm diameter) pellets of the contact materials were prepared and etched as previously described in diluted HNO<sub>3</sub> followed by an identical rinsing procedure. The pressed contacts were placed on the semiconductor and the Bell jar was evacuated 10<sup>-5</sup> torr or better before heating. Alternatively the pumping was stopped at 10<sup>-5</sup> torr and pure argon gas introduced into the bell jar before heating as in the previous technique. The temperature was increased at a rate of about 10 °C per minute up to 350 °C and then (in the case of In contacts) held constant for approximately 20 minutes. For Sn and Pb contacts, however, the temperature is raised to 500 °C, because of the higher melting points of these materials.

With these methods, contacts could be made that were stable and reproducible. Both ohmic and non-ohmic contacts were made as seen in Figs. 3.2 - 3.5.

### 3.3 Barrier Height Measurements

Barrier height measurements were made at the range of temperature 30 - 150 °C for In-SnS-In and In-SnS-Ag systems. The specimens were prepared as

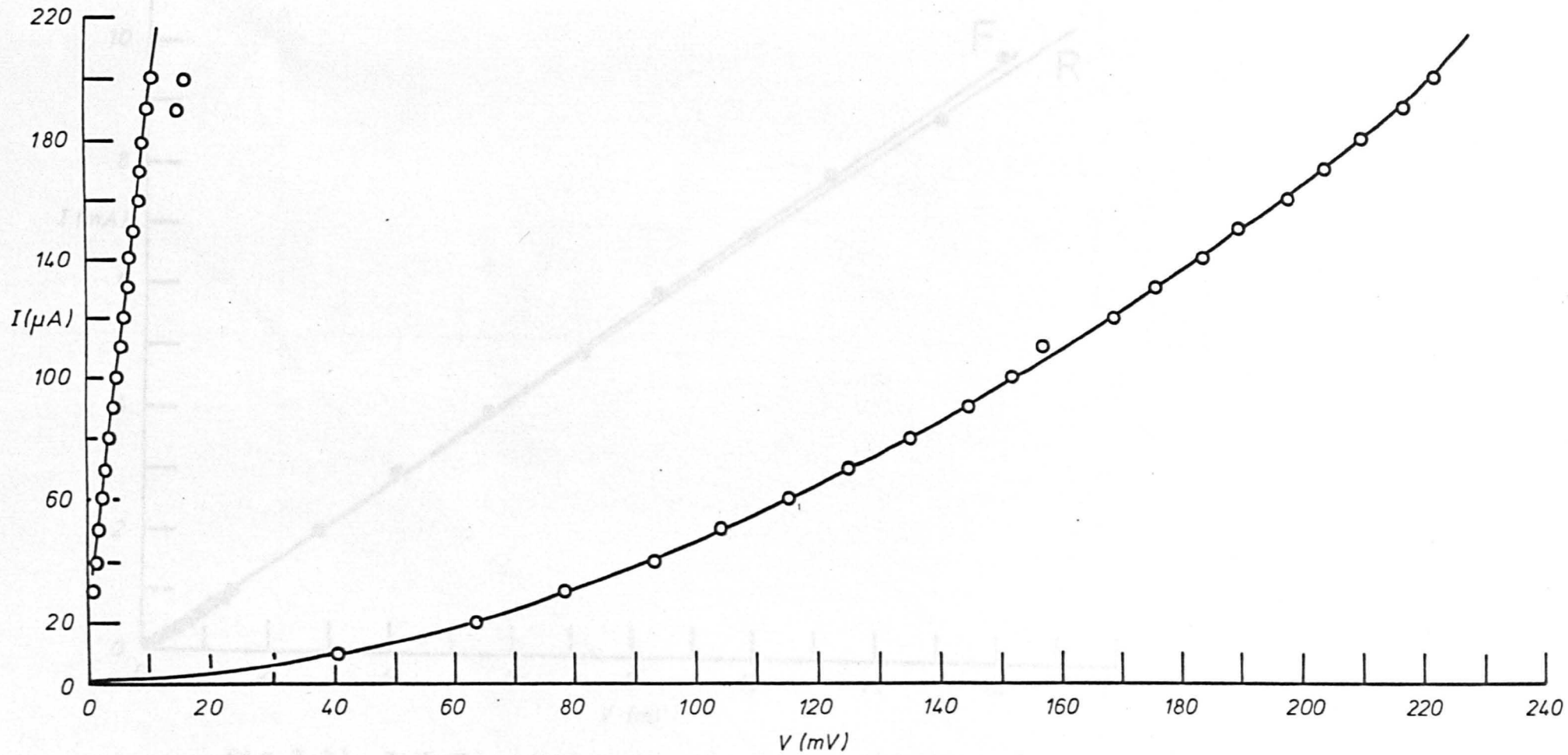


Fig.3-2): I-V Characteristics of alloyed In contacts onto n-type SnS.



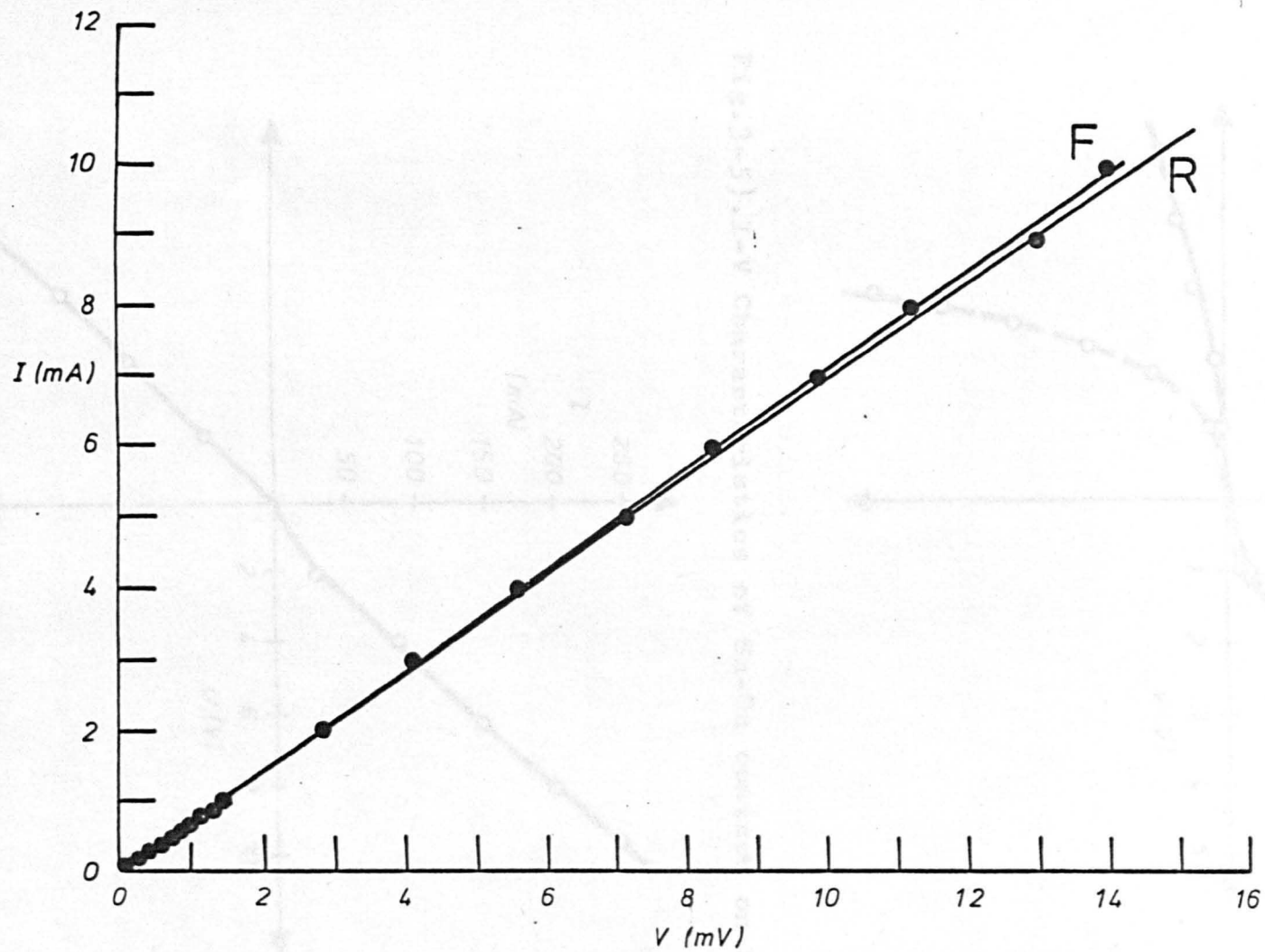


Fig.3-3). I-V Characteristics of alloyed In contacts onto p-type SnS.

Fig.3-4). I-V Characteristics of In-In contacts on SnS.

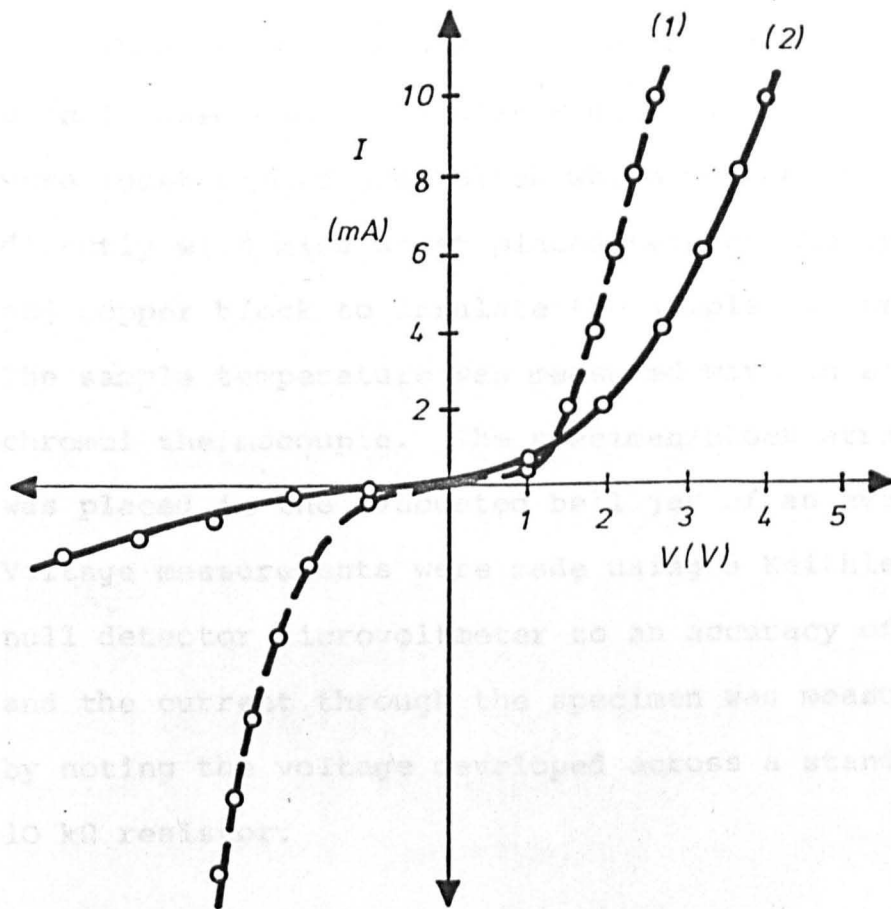
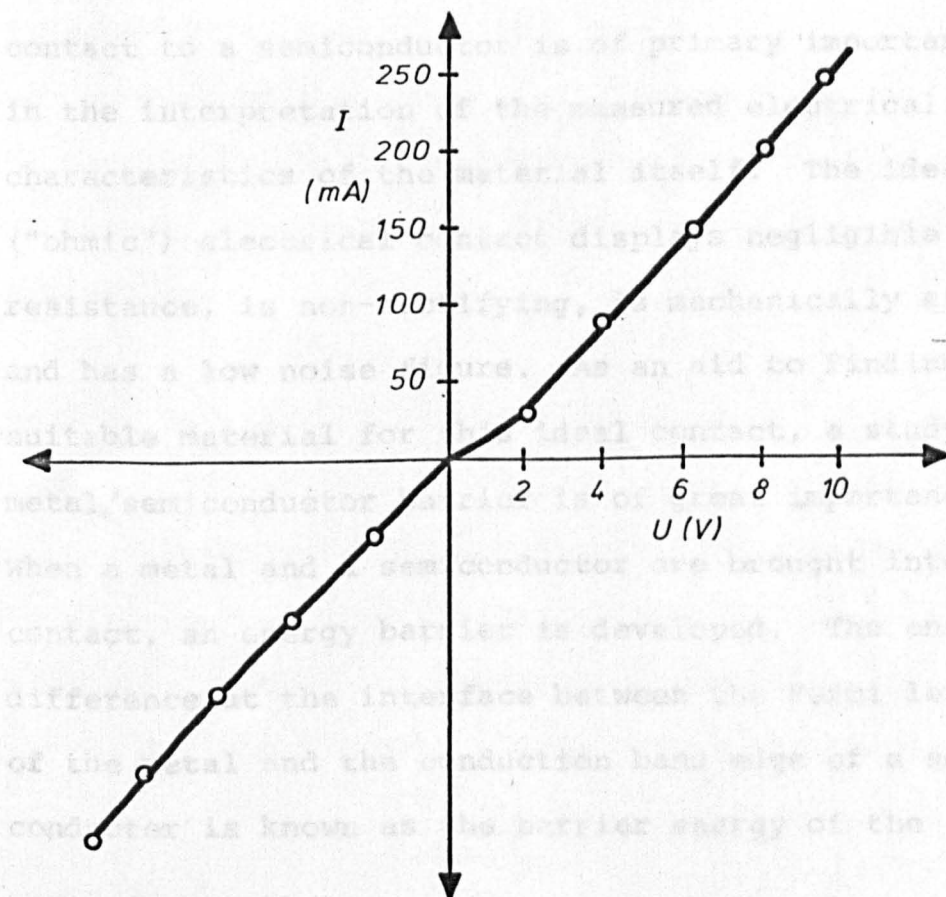


Fig.3-5). I-V Characteristics of Sn-Sn contacts on SnS.



described in section 3.2. The general lay-out of circuit used can be seen in Fig. 3.6. Specimens were located on a copper block which was heated indirectly with mica sheet placed between the specimen and copper block to insulate the sample electrically. The sample temperature was measured with an alumel-chromel thermocouple. The specimen/block arrangement was placed in the evacuated bell jar of an evaporator. Voltage measurements were made using a Keithley 155 null detector microvoltmeter to an accuracy of  $\pm 1\mu\text{V}$ , and the current through the specimen was measured by noting the voltage developed across a standard  $10\text{ k}\Omega$  resistor.

### 3.3.1 Barrier Height Theoretical Considerations

An understanding of the nature of the electrical contact to a semiconductor is of primary importance in the interpretation of the measured electrical characteristics of the material itself. The ideal ("ohmic") electrical contact displays negligible resistance, is non-rectifying, is mechanically strong and has a low noise figure. As an aid to finding a suitable material for this ideal contact, a study of metal/semiconductor barrier is of great importance. When a metal and a semiconductor are brought into contact, an energy barrier is developed. The energy difference at the interface between the Fermi level of the metal and the conduction band edge of a semiconductor is known as the barrier energy of the

particular metal/semiconductor interface. The study of electrical interface barriers starts in 1874 with Braun who observed rectification at an interface formed by a metal and a PbS crystal. Since then many experimental and theoretical investigations of barrier heights at metal/insulator interfaces have been carried out. The physics of metal/semiconductor interfaces, and the general properties of ohmic contacts have been analysed and reviewed by Mead (1969), and Bube (1960) has discussed the formation of ohmic contact to semiconductors. It is an experimentally observed fact that the size of the barrier height  $\phi_B$  for the covalent semiconductors is with few exceptions approximately 2/3 of the band gap for n-type materials and approximately 1/3 of the band gap for p-type materials (Mead 1964, 1969, Rideout 1975). Although the barrier height for covalent semiconductors is approximately independent of the contact metal used (Rideout, 1975; McGill, 1974), for highly ionic materials such as most of the group 11-VI compound semiconductors the barrier height is strongly dependent on the work function of the metal. In this case the barrier height is given by

$$\phi_B = \phi_w - \chi_s$$

where  $\phi_w$  is the work function of metal and  $\chi_s$  is the electron affinity of the semiconductor. Due to difficulties in measuring  $\phi_w$  accurately, we make use of the electronegativity (Pauling, 1960) of the metal which is related to the work function through the expression  $\chi_m = 0.44\phi_w - 0.15$  (Bube, 1960). Thus the barrier

height can be written (McGill, 1974) for ionic semiconductors:

$$\phi_B \approx \chi_m - \chi_s + \text{constant}$$

One can conclude from above equation that the barrier energy increases linearly with the electronegativity of the contact metal used, as has been shown by Rideout (1975). So we might write that

$$\phi_B \approx SX_m + \text{constant}$$

where S is the so called Fermi-level stabilization parameter

$$S = \frac{d\phi_B}{dX_m}, \text{ (Kurtin, McGill and Mead 1969).}$$

The relationship of S to the electronegativity difference between the constituent elements of a compound semiconductor,  $\Delta X$ , is shown in Fig. 3.13.

Several methods have been used to determine the barrier height  $\phi_B$ , but the results are usually consistent (Mead, 1966). The methods include analysing the I-V characteristics of the contact/semiconductor junction, investigating the metal/semiconductor interface capacitance as a function of frequency, determining the photoresponse threshold (Spitzer and Mead, 1961) and observing the behaviour of tunneling electrons at liquid helium temperature (Guinet, 1974). The simplest measurement technique (measuring the I-V curves) was chosen in this case. Since the analysis used here is not given in any standard texts or publications we present it in full.

### 3.3.2 Measurement Technique

Consider an n-type material with two contacts which have unequal barrier heights and assume there is only one type of carrier as shown in Fig. 3.7.

The Richardson current over a barrier B is given by

$$J_B = AT^2 \exp(-B/kT)$$

where A is the Richardson constant. When an electric field is applied, a current I, passes through the sample. If the voltage between the terminals is V, and across the bulk of the semiconductor is  $V_R$ , then the current I will be written as:

$$I = \frac{V_R}{R}$$

where R is the resistance of bulk.

If the contact connected to the negative terminal of the battery has area  $a_1$  and barrier  $b_1$  the Richardson current for this contact will be:

$$I_{b_1} = a_1 J_{b_1} = a_1 AT^2 \exp(-b_1/kT)$$

If the applied current  $I \ll I_{b_1}$ ; then  $V_R = V$ . Similarly if an applied field is reversed:

$$V_R = V \quad \text{if} \quad I \ll I_{b_2}$$

In general, if the current passing through the semiconductor is much less than the Richardson current (which may be of the order of tens of milliamps), the voltage drop will be symmetrical no matter what the barrier height is.

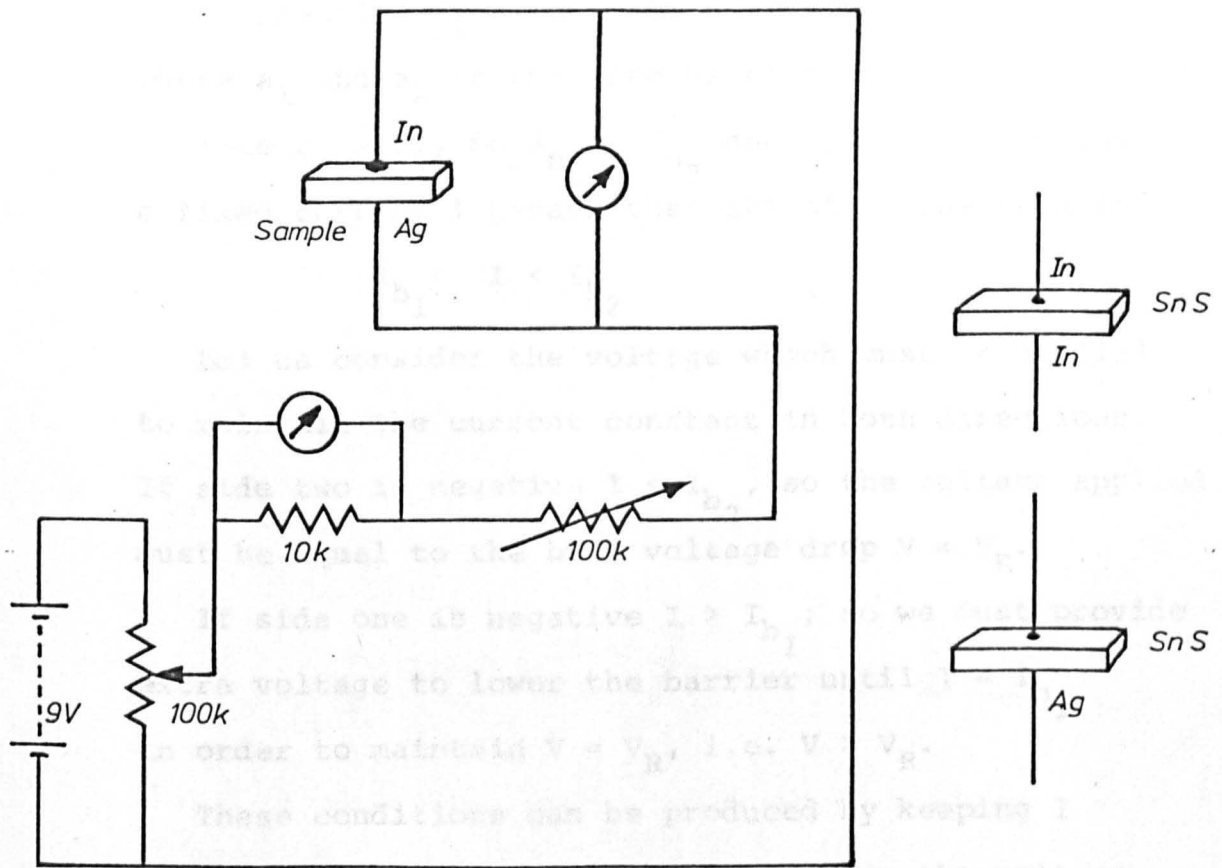


Fig.3-6). Circuit for barrier height measurements.

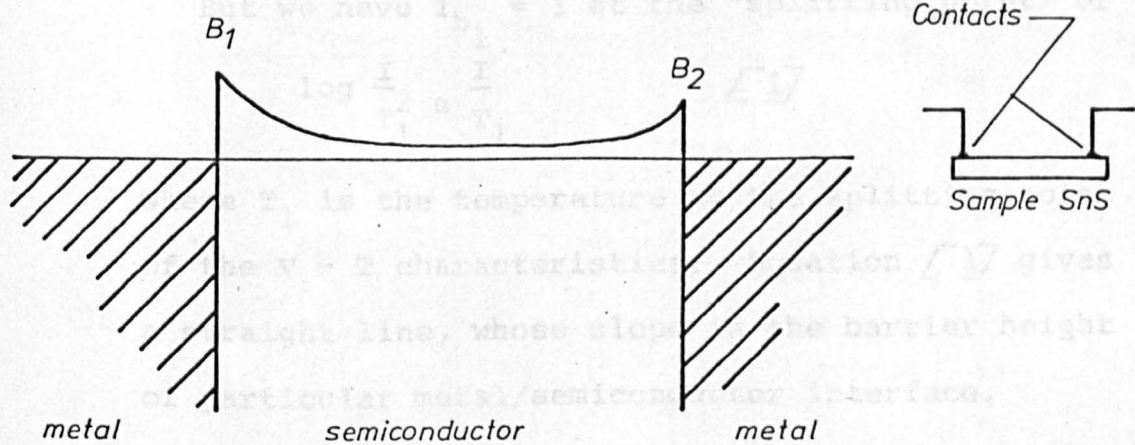


Fig.3-7). Schematic representation of two Schottky barrier diodes.

In practice  $a_1 \approx a_2$  but the barrier height  $b_1 \neq b_2$ , where  $a_1$  and  $a_2$  is the area of contact made.

Take  $b_1 > b_2$ , so  $J_{b_1} < J_{b_2}$  and  $I_{b_1} < I_{b_2}$ . Suppose a fixed current  $I$  passes that satisfies the relation

$$I_{b_1} < I < I_{b_2}$$

Let us consider the voltage which must be applied to maintain the current constant in both directions. If side two is negative  $I < I_{b_2}$ , so the voltage applied must be equal to the bulk voltage drop  $V = V_R$ .

If side one is negative  $I > I_{b_1}$ ; so we must provide extra voltage to lower the barrier until  $I = I_{b_1}$  in order to maintain  $V = V_R$ , i.e.  $V > V_R$ .

These conditions can be produced by keeping  $I$  constant and varying temperature. When the voltage between the terminals is plotted as a function of the temperature of the sample for several constant current values, one obtains the type of the graph shown in Fig. 3.8.

But we have  $I_{b_1} = I$  at the "splitting point" or

$$\log \frac{I}{T_1^2} \propto \frac{I}{T_1} \quad [1]$$

where  $T_1$  is the temperature at the splitting point of the  $V - T$  characteristics. Equation [1] gives a straight line, whose slope is the barrier height of particular metal/semiconductor interface.

In practice the small voltage levels invoked in this determination can present measurement problems. For example, thermoelectric voltages will appear



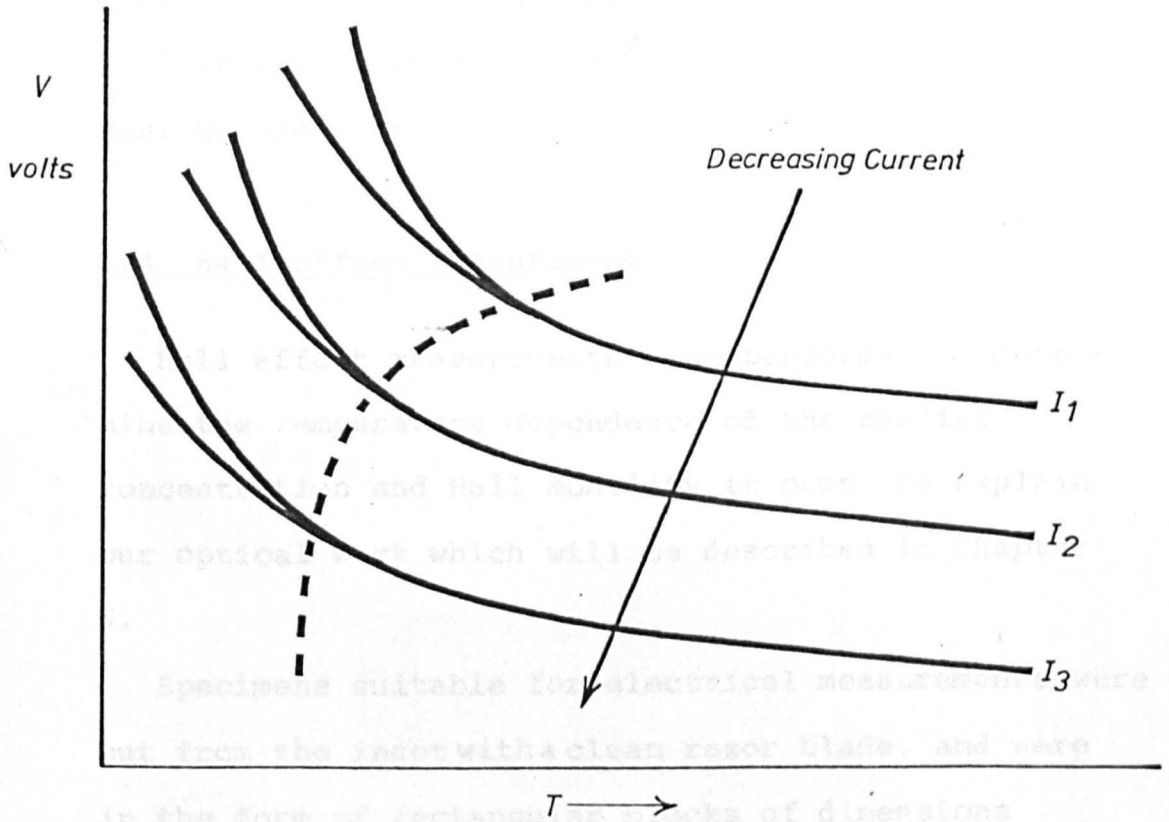


Fig.3-8). Typical voltage-temperature variation of Schottky barrier diode.

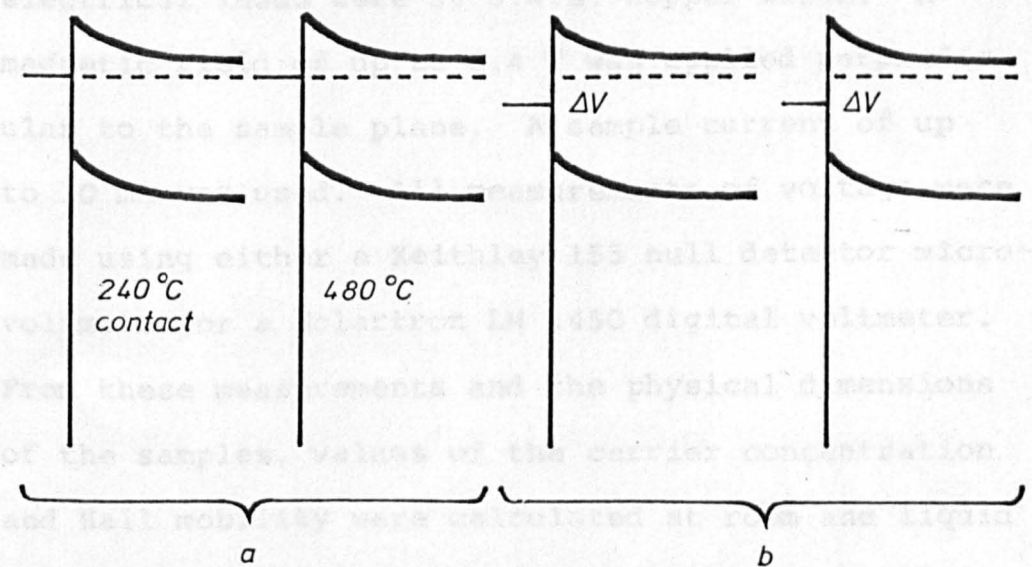


Fig.3-9). Energy diagrams of Schottky barrier diodes before and after applied electric field.

across the temperature drop along the pin seals used in the evacuated chamber, and due allowance must be made for this.

### 3.4 Hall Effect Measurements

Hall effect measurements were performed to determine the temperature dependence of the carrier concentration and Hall mobility in order to explain our optical work which will be described in Chapter 6.

Specimens suitable for electrical measurements were cut from the ingot with a clean razor blade, and were in the form of rectangular blocks of dimensions 10 x 4 x 0.3 mm. Four and six contacts were made to the sample as described in section 3.2.4. The specimens were then stuck onto a sample holder with Durefix. The sample holder was made of teflon; the electrical leads were 36 S.W.G. copper wires. A magnetic field of up to 0.4 T was applied perpendicular to the sample plane. A sample current of up to 10 mA was used. All measurements of voltage were made using either a Keithley 155 null detector micro-voltmeter or a Solartron LM 1450 digital voltmeter. From these measurements and the physical dimensions of the samples, values of the carrier concentration and Hall mobility were calculated at room and liquid nitrogen temperature using the equations below for a single one-carrier system.

The usual precautions of reversing magnetic field and current were carried out in order to minimise errors of interpretation due to contact misalignment and sample magnetoresistance.

The simple theory of the Hall effect is well known. Our data here analysed in the one carrier approximation for which:

$$R = - \frac{r}{ne} \quad ; \quad r = 1$$

$$R = \frac{V_H \cdot t}{B_Z \cdot I_X}$$

$$\sigma = ne\mu_H \quad ;$$

$$\mu_H B_Z = \frac{E_H}{E_X}$$

where we have considered a sample oriented until a current  $I_X$  passing down its major axis:  $E_H$  is the Hall electric field;  $V_H$  the Hall voltage,  $B_Z$  the applied field and  $t$  the sample thickness. The carrier concentration is  $n$ , the Hall mobility  $\mu_H$  and, the Hall coefficient ( $r$ ) is assigned at unity. Other symbols have their usual meanings.

### 3.5 Results and Discussions

Specimens of single crystal SnS were prepared as described in Chapter 2 (see also Nikolić et al. 1976) using the Bridgman technique. The crystals can be easily cleaved into thin plates perpendicular to the c-axis due to its layer-type structure. X-ray analysis,

laue-photography and X-ray diffractograms confirmed that the ingots were single crystal. We do, however, have reason to believe that inhomogenities (local regions of impurity excess) exist in some samples, and this may explain the enormous variety of electrical contact behaviour observed.

### 3.5.1 Contact Properties; discussion

The contacts made by silver paint or lead-tin soldering were electrically noisy although sometimes mechanically strong. We therefore concentrate on the reproducible results obtained with alloyed metal contacts which as shown in Figs. 3.2-5. Figure 3.2 shows a non-linear characteristic which is typical of many obtained for n-type specimens, whereas Fig. 3.3 displays typical results for alloyed In-contacts on p-type SnS, showing apparently ohmic behaviour. The contacts on both materials were made at 240 °C, and the characteristics measured by the same d.c. procedure. The contacts to the n-type material usually became noisy and the characteristics un<sup>re</sup>producible for higher currents than displayed in Fig. 3.2. The sample of Fig. 3.3 was extensively used for transport measurements. Fig. 3.4 shows two curves for another n-type sample with alloyed In contacts made at 450 °C (curve 1) and 400 °C (curve 2). At low currents and voltages both curves show the non-linear behaviour exhibited by the forward portion of

Fig. 3.2, but the curves are symmetric in this case. The entire curve (1) is symmetric up to the maximum voltage ( $\pm 5V$ ), whereas this is not the case for the curve (2).

In attempting to explain the observed contact properties of indium alloyed onto SnS, we suggest that in the alloying process the indium diffuses a small distance into the material and acts as an acceptor impurity, the distance of diffusion being a function of alloying temperature. This is deduced from thermopower measurements made on the small area around the In contacts on n-SnS, which indicate a p-type region. This introduction of the acceptor impurity would then lower  $E_F$  in the p-type material, producing an ohmic contact as seen in Fig. 3.3. This behaviour is in accordance with the well known 'rule of thumb' that a diffused contact impurity must have the same conductivity characteristic as the host semiconductor to produce 'ohmic' contact i.e., acceptor for p-type material and a donor for n-type material. The non-linear behaviour frequently observed as in Fig. 3.2 can be represented by an equivalent circuit of a diode and resistor in series, i.e., the contact made at this temperature ( $240^\circ C$ ) to n-type material has produced one rectifying and one non-rectifying contact.

One possible explanation for this behaviour would be the existence of some gross sample inhomogeneity

e.g. one end (the 'ohmic' contact end) being p-type and the other end being n-type due, presumably, to sulphur excess [Anderson and Morton, 1945; Albers et al., 1961].

This seems unlikely, however, as the behaviour is noted for a number of n-type samples with Indium contacts alloyed at this temperature. Fig. 3-4 shows a symmetrical I-V curve for Indium contacts made at a higher temperature (450 °C), the low voltage (high resistance) portions of which are identical with the non-linear section of curve 3.2 (this is not shown explicitly in Fig. 3.4). About  $|V_c| = 1.2$  V, however, the conductivity characteristic changes significantly, and for  $|V| > |V_c|$  a lower resistance is symmetrically displayed by the material. One can explain this low voltage behaviour of Fig. 3.4 by two 'back to back' diodes, i.e., two non-ohmic contacts to n-type material as found in the reverse characteristic of Fig. 3.2. We suggest that the curves for  $|V| > |V_c|$  may be explained by increased carrier injection from one (or possibly two) deeper acceptor states of Indium in SnS, which are only created by diffusion of Indium at the higher temperature of contact formation (450 °C). We invoke the presence of these levels in the observed near infrared photoresponse (see Chapter 4) and remark that similar are found for CdS (Kröger, Vink and Volger, 1955; Bube and Thomsen, 1955) and Ge (Morton, Hahn and Schultz 1956; Bube 1960). Above about 1.2 volts we envisage that ionisation of the deeper acceptor by energetic electrons proceeds by an

impact ionisation process, leading to an increased electron population and increased sample conductivity. The behaviour is displayed by each contact. This process may be alternatively thought of in terms of the measured Fermi levels in 'doped' and 'undoped' samples, i.e., n-type samples with contacts formed at 450 °C and 240 °C. Fig. 3-9 (a) shows the two situations before a forward bias is applied and Fig. 3-9 (b) after the application of a bias voltage. It may be seen that the barrier for flow into the metal is reduced significantly for the 'doped' specimen under bias, thus giving the higher conductivity as observed experimentally. The curve (2) evidently displays a greater  $V_c$  for the onset of this behaviour in the forward direction, which is compatible with a smaller population of acceptors being created in the semiconductor by the lower diffusion rate observed at the lower contact temperature (400 °C). This curve, however, exhibits a non-symmetric characteristic for  $|V| > |V_c|$ . We suggest therefore that in the case of contacts formed by alloying of Indium at lower temperatures, the existence of sample inhomogeneities becomes more significant, yielding unsymmetric contact behaviour.

The properties of alloyed tin and tin/Indium contacts were also studied as it was felt that these

might produce non-rectifying behaviour in n-type specimens. A typical curve produced for an n-type specimen with Sn contacts alloyed at 450 °C is shown in Fig. 3.5. The characteristics obtained with this contact material were noise free and repeatable up to the larger voltages applied, and display linear behaviour in one current direction with the same slope ( $40 \Omega$ ) as found for the linear section of Fig. 3.2. This value is evidently significant as it is obtained for a number of contact materials, and is to be interpreted as the bulk conductance of the n-type samples at room temperature. We would suggest that the change of slope in the forward characteristic of Fig. 3.5 is accounted for by the presence of a rectifying surface oxide layer between the tin contact and the semiconductor. At low voltages ( $< 2V$ ) this leads to an increased resistance, but above 2V breakdown occurs and the conductivity is that of the bulk material i.e. the slopes of each side of the characteristic are the same.

Other alloy contact materials investigated were cadmium, lead and antimony. Lead sometimes formed ohmic contacts onto n-type SnS, which is understandable if one assumes that lead diffuses as a donor into SnS. However, the procedure was not always reliable, it is being noted that frequently SnS would melt before the 5-nines pure lead contact. Similar problems were



encountered with cadmium and antimony. Some success was obtained with evaporated antimony and cadmium contacts, but here problems arose with fixing suitable connecting wires to these metals.

The study of electrical contacts to a new semiconductor such as SnS could occupy a thesis in itself, so this brief investigation is far from exhaustive. The object was to determine experimentally how low noise, low resistance contacts may be obtained to n- and p-type material which are mechanically sound down to at least 77 °K. The effects of contact ageing have not been thoroughly examined, neither have the surface conditions been as carefully controlled as would be necessary in such a study. However, we conclude that contacts approaching this performance criterion may be repeatedly obtained for p-SnS by alloying with Indium at 250 °C under vacuum, and for n-SnS with a lead alloy, but that the repeatability for lead contacts is less good.

### 3.5.2 Barrier Height Discussion

Barrier height measurements were made as described in section 3.3. In Figs. 3.10 and 3.11 the log V versus T graphs are shown for silver paint and for both alloyed and soldered In contacts made to p-SnS. The barrier heights are obtained as explained in section 3.3.2 from the slope of the  $\log \frac{I}{T^2}$  versus  $\frac{I}{T}$  graphs for each type of contact. These values are 0.1 eV for alloyed In contacts, and approximately

0.6 eV for both soldered In and silver paint contacts. Barrier height measurements are not easy to make, and we assign an uncertainty of  $\pm 0.1$  eV to these values.

The very low value of the barrier measured for alloyed contacts confirms the behaviour noted from the linear I-V characteristics. This is understandable if we imagine the Indium alloy contacts diffusing a short way into the p-SnS and acting as acceptors, as has been discussed in the previous sections. The high and approximately equal barriers observed for the silver (paint) and soldered Indium contacts may well represent the height over an oxide layer present in both systems between the metal and the semiconductor. This oxide layer would not be destroyed as it might be in the alloy-contact forming process, where a higher sample surface temperature and an acid flux are used.

A full study of barrier heights for a range of contact materials was not possible for SnS. This is because of the relatively low melting point ( $1881.5^{\circ}\text{C}$ ) which precluded evaporation of many metals as contact materials. Thus, had it been possible to produce gold and magnesium contacts, one would have been able to obtain a graph of barrier height ( $\phi_B$ ) versus electronegativity (X) for four contact materials (i.e. Mg, In, Ag, Au, for which  $X = 1.2, 1.7, 1.9$  and  $2.4$  respectively). This would have then enabled a value for the Fermi level stabilisation parameter

$\frac{d\phi_B}{dx_m}$  and the electronegativity difference  $\Delta X$  of the species of a compound semiconductor ( $\Delta X = 0.2$  in this case). This value is not inconsistent with another interpretation of the present barrier height measurements for silver paint and (soldered) Indium contacts, namely that the values  $\phi_B = 0.5$  and  $\phi_B = 0.7$  eV for Indium and silver respectively, yielding

$$\frac{d\phi_B}{dx_m} \approx \frac{0.7 - 0.5}{1.9 - 1.7} = 1$$

as a maximal value, i.e.

$$0 < \frac{d\phi_B}{dx_m} < 1.$$

In conclusion, barrier height measurements for SnS have been made by analysing I - V characteristics in the temperature range 20 - 150 °C. The analysis assumes current flow over the barriers by a Richardson emission process, but this must remain an approximation since other processes (e.g. tunneling and field emission) are known to occur. Nevertheless, the major conclusion that reduced barrier heights may be obtained with Indium contacts to p-SnS is confirmed by observation of the I-V curves, and tends to support the basic correctness of the assumption of thermionic emission as being the dominant barrier crossing mechanism.

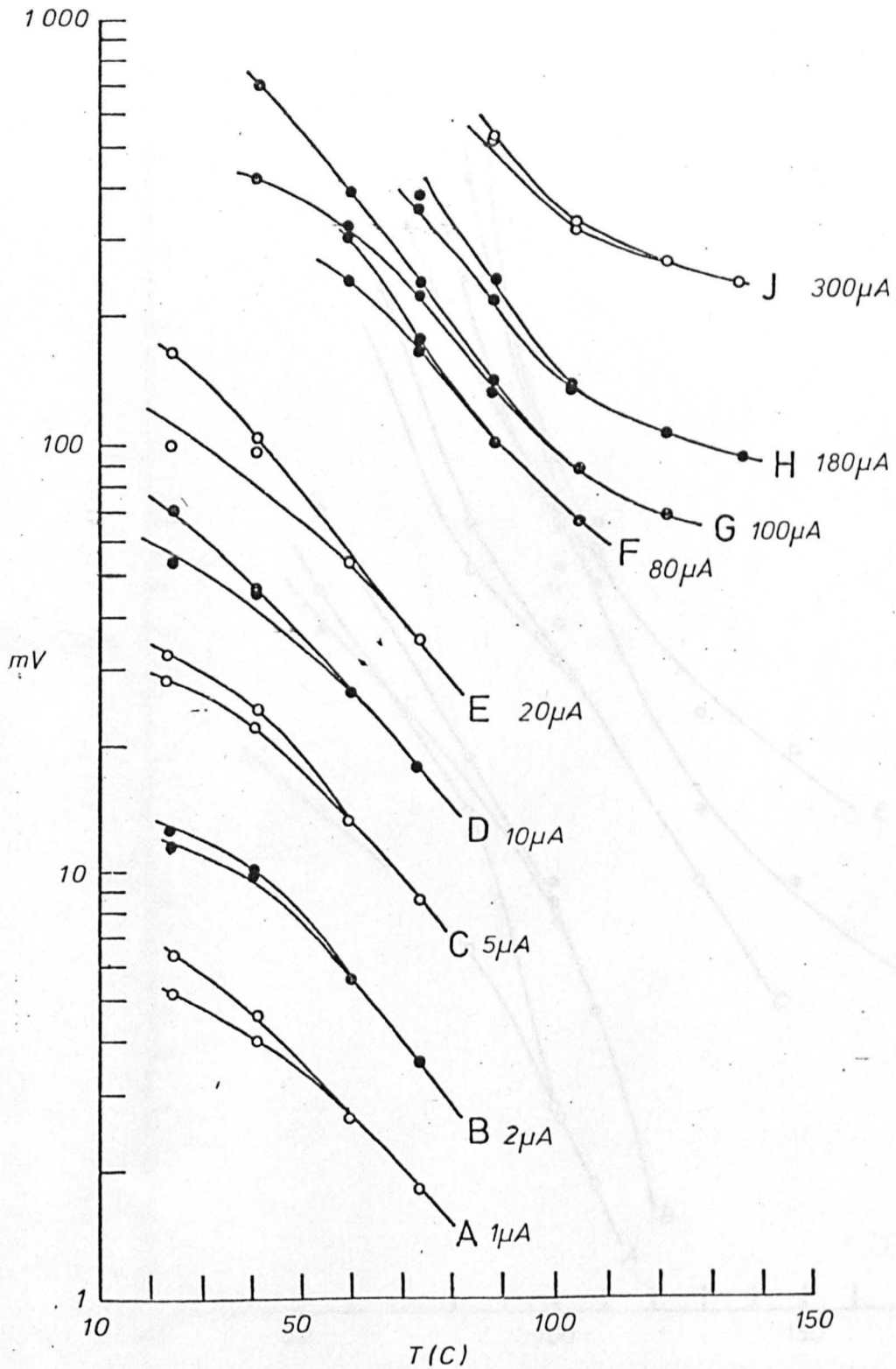


Fig.3-10). Voltage-Temperature variation of alloyed  
In contact on p-type SnS.

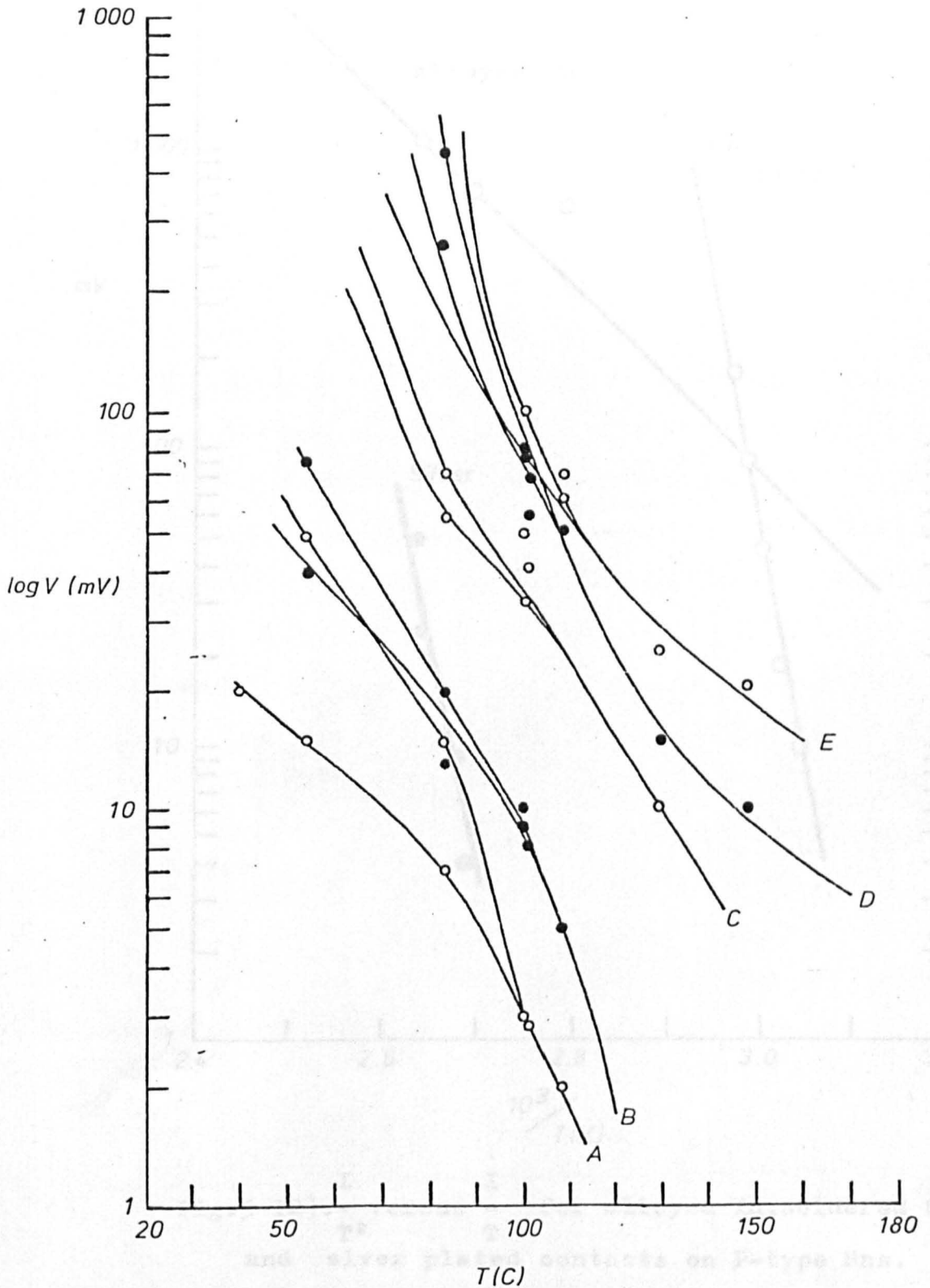


Fig.3-11). Voltage-Temperature variation of silver plated contact on p-type SnS.

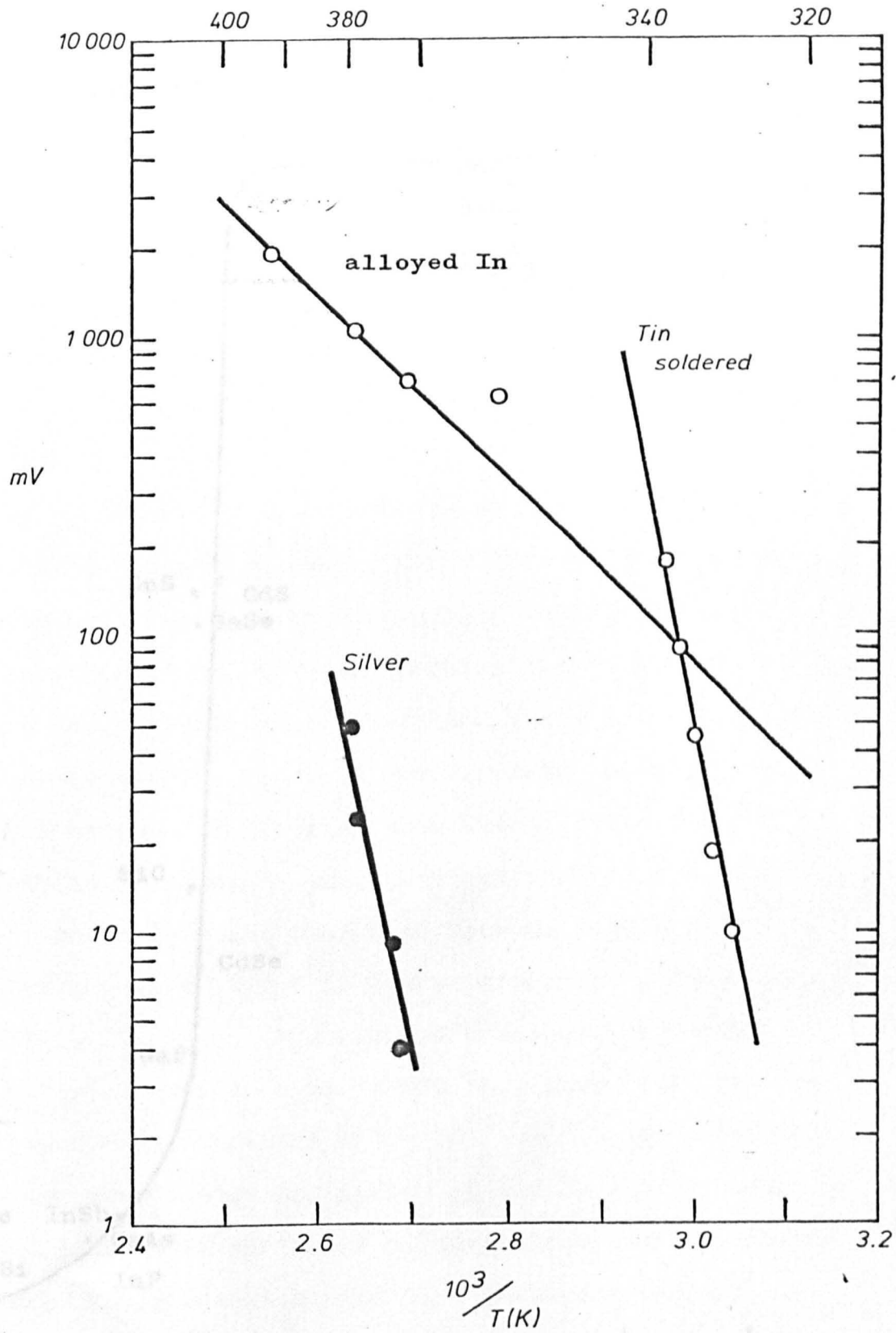


Fig.3-12). - versus  $\frac{I}{T^2}$  for alloyed In, soldered Sn and silver plated contacts on P-type Sns.

Fig.3-13). Index of interface behavior,  $\alpha$ , as a function of electronegativity difference,  $\chi$ , between the species of a compound semiconductor (after Harris et al, 1969)

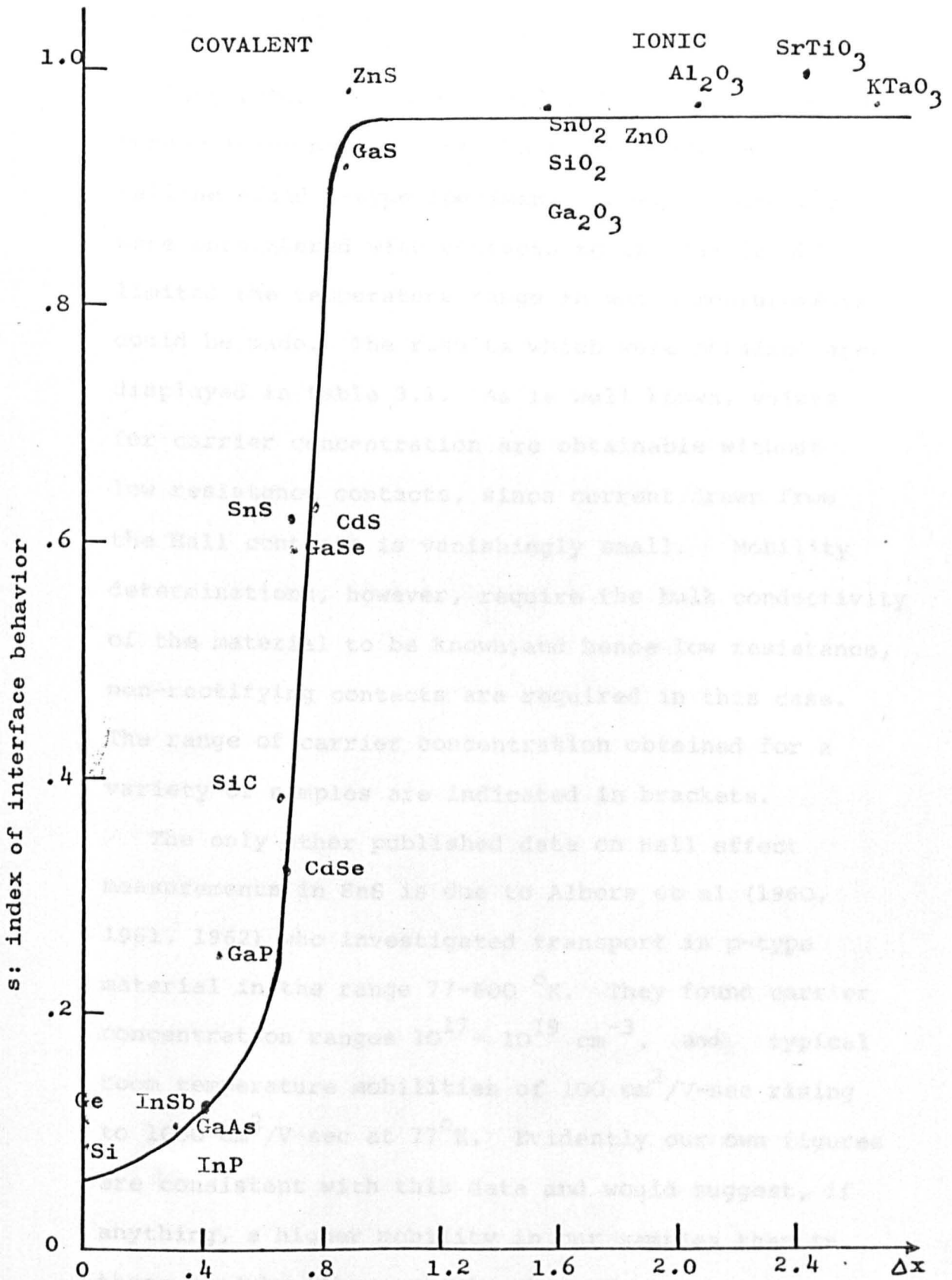


Fig.3-13). Index of interface behavior,  $s$ , vs electronegativity difference,  $x$ , between the species of a compound semiconductor (after Kurtin et al, 1969)

### 3.5.3 Hall Effect and Mobility Measurements

Temperature dependent Hall effect and carrier conductivity measurements have been made on crystalline n- and p-type specimens. Serious problems were encountered with contacts to the sample which limited the temperature range in which measurements could be made. The results which were obtained are displayed in table 3.1. As is well known, values for carrier concentration are obtainable without low resistance contacts, since current drawn from the Hall contacts is vanishingly small. Mobility determinations, however, require the bulk conductivity of the material to be known, and hence low resistance, non-rectifying contacts are required in this case. The range of carrier concentration obtained for a variety of samples are indicated in brackets.

The only other published data on Hall effect measurements in SnS is due to Albers et al (1960, 1961, 1962) who investigated transport in p-type material in the range 77-600 °K. They found carrier concentration ranges  $10^{17} - 10^{19} \text{ cm}^{-3}$ , and typical room temperature mobilities of  $100 \text{ cm}^2/\text{V-sec}$  rising to  $1000 \text{ cm}^2/\text{V-sec}$  at 77°K. Evidently our own figures are consistent with this data and would suggest, if anything, a higher mobility in our samples than in those used by Albers et al. Presumably the variation of carrier mobility with temperature is best explained by the usual judicious mixture of (high temperature) ionised impurity scattering, although the lack of



comprehensive low temperature data (due to contact problems) limits these conclusions in our case.

In the case of n-type material the contacts gave extreme problems, and it was only possible to extract carrier concentration data at 77 °K and 300 °K for a small number of samples. We remark, however, that in the same temperature range, the freeze-out in the two materials is essentially the same  $((n/p)_{300} / (n/p)_{77} \sim 2.5)$ , indicating approximately equal depths for substituted acceptors and donors. In view of the obvious crystallographic assymetries in this material it was decided to investigate mobilities in a, b and c directions for a p-type sample. The preparation of a suitable sample for c-axis work presented such materials problems that it was finally abandoned, leaving measurements in the a and b directions which yielded identical values for mobility within the limits of experimental error. Evidently one may expect significant variation in mobility between the c-axis and the other two in view of the differences in physical properties in this direction, and it is regrettable that this measurement could not be made.

In conclusion, the present transport measurements may be considered to indicate the range of sample mobilities and carrier concentrations used in our experiments. They represent a limited study, however, which must await further improvements in contact fabrication before further conclusions can be made.

TABLE 3 - 1

Sample No:	Temperature (°K)	Carrier Concentration (cm <sup>-3</sup> )	Hall Mobility (cm <sup>2</sup> /V.Sec)
A(p-type)	300	$1.15 \times 10^{19}$	146
	77	$6.54 \times 10^{18}$	1512
B ,,	300	$7.49 \times 10^{18}$	169
	77	$5.50 \times 10^{18}$	1500
C ,,	300	$5.62 \times 10^{19}$	-
	77	$4.85 \times 10^{18}$	-
	50	$1.25 \times 10^{18}$	-
D(n-type)	300	$6.25 \times 10^{18}$	236
E ,,	300	$3.83 \times 10^{18}$	354

## CHAPTER FOUR

### PHOTOCONDUCTIVITY AND RELATED PHENOMENON ON CRYSTALLINE SnS

#### 4.1 Introduction

A study of photoconductivity and the photovoltaic effect in semiconductors allow us to explore semiconductor band structure, recombination processes, impurity levels and the properties of metal-semiconductor interfaces.

In the present work, photoconductivity and photovoltaic effects were investigated for the first time as a function of sample temperature, ( $40^{\circ}\text{K} - 300^{\circ}\text{K}$ ) energy of incident radiation ( $0.4 - 2 \text{ eV}$ ), sample bias and illumination intensity for a variety of both n- and p-type crystalline SnS samples. The determination of the temperature and energy variation of photoconductivity and photovoltaic effect involved three sets of apparatus. One, for determining the room temperature photoconductivity and photovoltaic spectrum covering the region approximately  $0.99-1.50 \text{ eV}$ , was used for n-type crystalline SnS samples. Two further systems were used for measurements on p-type material; one for  $77^{\circ}\text{K}$  and  $300^{\circ}\text{K}$  measurements in the range  $0.99-2.10 \text{ eV}$ , and one for  $40-300^{\circ}\text{K}$  measurements in the spectral range  $0.4-1.3 \text{ eV}$ . We confine the details of

these arrangements to the appendix to this chapter, A.

#### 4.2 Photoconductivity Measurement Technique

In each case a fairly conventional photoconductivity measurement was used; a wide band quartz-halogen lamp illuminates the entrance slit of a monochromator, the output of which is focussed by a quartz-lens or aluminium mirror train onto the specimen which is placed in the cryostat. A mechanical chopper and reference assembly (10 hz - 800 hz) is used and phase-sensitive detection of the photo-response is employed. The results are displayed on a chart recorder as the monochromator is scanned automatically. By varying the chopper frequency, the signal can be measured away from the region of mechanical and flicker noise frequencies, specific contact noise frequencies and even noise from fluctuations in the density of photocurrent carriers. (Pose 1955).

Specimens suitable for photoconductive measurements were cut from the ingot. Each time four contacts were made on freshly cleaved and oriented samples as described in chapter 3. On most specimens two-contact and four-contact measurements were attempted, and in addition a number of simple combinations of biasing circuits and transformer/preamplifier arrangements were used to optimise sample photosignal. In all cases extensive shielding precautions were taken. Simple masks were also made up to shield or expose various parts of the sample and contacts. The analogue records of the photosignal which were ultimately obtained

were ratioed to the monochromator instrument function. This was obtained by measuring the monochromator output with a thermopile (Hilger-Schwarz FT - 9008) and pre-amplifier arrangement.

A certain amount of confusion exists in the literature over the terms 'photoconductive' and 'photovoltaic' effects. In a perfectly homogeneous sample provided with two barrier contacts the action of illuminating radiation  $h\nu > E_g$  is to create electron-hole pairs in the conduction and valence band respectively, thereby changing the sample resistance. Evidently this change may be observed as a change in voltage across the specimen, or across a resistance in series with it, when the specimen is placed in a simple bias circuit. This, strictly speaking, is the photoconductive effect. An alternative method of measuring this phenomena of bulk-generated photoconductivity is to measure the voltage change across the inner pair of four contacts, the outer pair being connected to a constant-current supply. In many practical cases, however, an inhomogeneous sample, or a sample with partially or totally rectifying contacts implies the existence of space charges at barriers. Photoproduced carriers will under these circumstances migrate, ultimately producing an e.m.f. This, strictly speaking, is the photovoltaic effect, and is measured in a two-contact geometry. In most of our experiments, it is this effect which is measured, since it is often orders of magnitude greater than pure photoconductivity. The a.c. photovoltages generated are of the order of 10  $\mu$ v. Evidently the measurement of such small signals

requires careful impedance matching of amplifiers and transformers and the extensive use of screening.

### 4.3 Results and Discussion

#### 4.3.1 n-SnS for $0.9 < h\nu < 1.5$ eV

Figures 4-1 to 4-5 inclusive show typical photo-conductive characteristics obtained for n-type samples at 300 °K in this frequency range. Two photo-conductive response characteristics for a sample with Sn-Sn contacts (1) and Sn-In contacts (2) are shown in Figure 4.1. Although the scales of these graphs are different, the shape of the response is broadly similar. For curve (1) illumination of the bulk (contacts shielded) and illumination of the contacts produced essentially the same results. With In-Sn mixed contacts, however, the photo-current intensity was much higher if the indium contact was exposed. Using the Moss criteria for obtaining the excitation gap from curve (2) we obtained  $1.13 \pm 0.02$  eV, ( $1.09\mu$ ) and if we regard the curve (1) as essentially curve (2) at long-wavelengths (i.e.  $\lambda > 0.9\mu$ ) the same result is obtained. We justify this procedure later, but remark now that  $E_g = (1.13 \pm 0.02)$  eV almost certainly corresponds to the indirect transition observed in absorption by Lambros et.al (1974). The higher intensity of curve (2) compared to (1) is explained by carrier injection from the acceptor level created in n-SnS by indium diffusion at the contact. It is not, perhaps, therefore surprising to find that two indium contacts

('B', Fig. 4.2) give an even larger photoresponse, as may be seen from the relevant scales on these diagrams. The long-wavelength edge of this curve again provides us with a value for the indirect gap, but the curve peaks at  $0.84\mu$  with a point of inflexion at  $0.95\mu$ . We suggest that the peak in our curve at  $0.84\mu$  may be interpreted as a transition to another level at 1.46 eV. Lambros et al (1974) have invoked the presence of a phonon-involved transition at 1.42 eV to explain their transmission data, and this is certainly in good agreement with the present data. We remark that the difference in shape of curves 4.1 and 4.2 (B) are almost certainly attributable to different surface recombination properties in these samples. This type of variance is not unknown in, for example, CdS (e.g. Görlich, photoconductivity of solids p.48).

Curves 4.2 (A), 4.3, 4.4 and 4.5 were all obtained with rather thin specimens,  $t < L$  where  $L$  is the diffusion length. Under these conditions we may expect moderate to high surface recombination, giving a sharp peak in the region of the gap. An investigation of this type of response was made, for example, in GaAs by Frederikse and Blunt (1955). Moss (1965) has also analysed the photoresponse under these conditions, and has shown that the short-wavelength sensitivity is inversely proportional to the surface recombination velocity. The sharp peak observed in these samples provides a value  $1.09 \pm 0.02$  eV for the indirect transition energy which is consistent with that obtained from the previous curves. In addition, a small peak is observed at  $0.84\mu$  (i.e. 1.46 eV) which

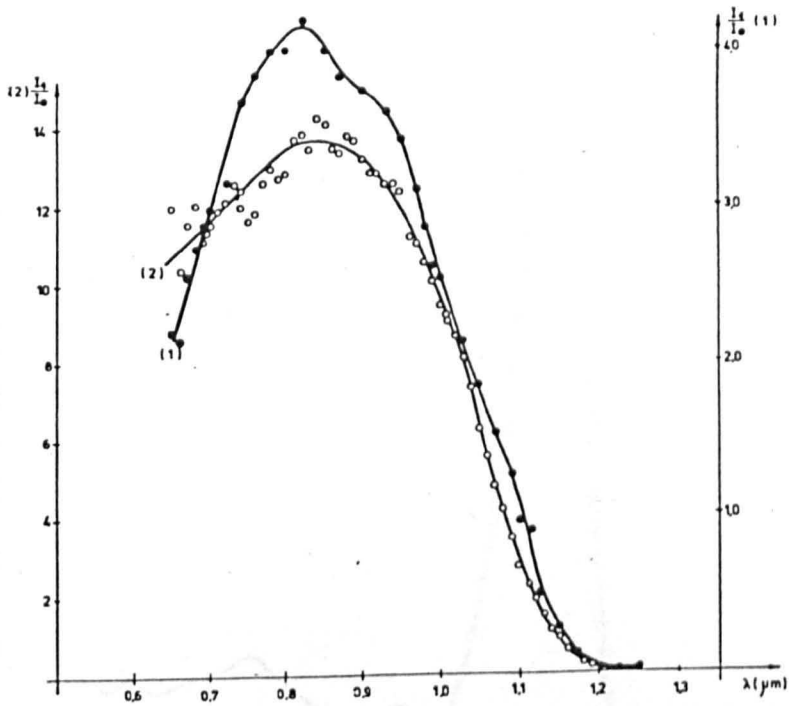


Fig.4-1). Photo response curve of n-type SnS.

Fig.4-3). Photo response curve of n-type SnS.

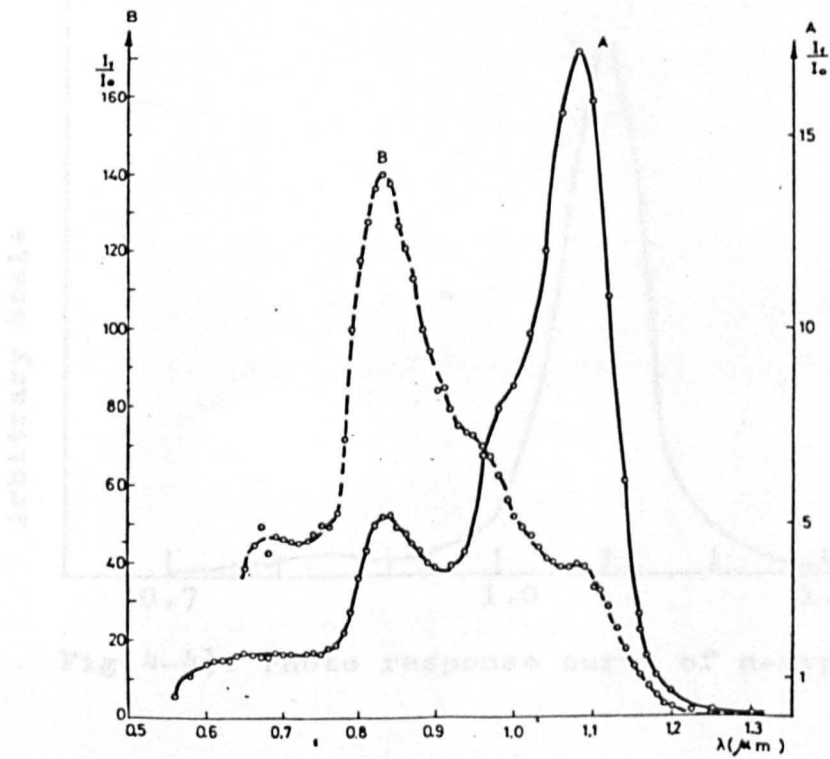


Fig.4-2). Photo response curve of n-type SnS.



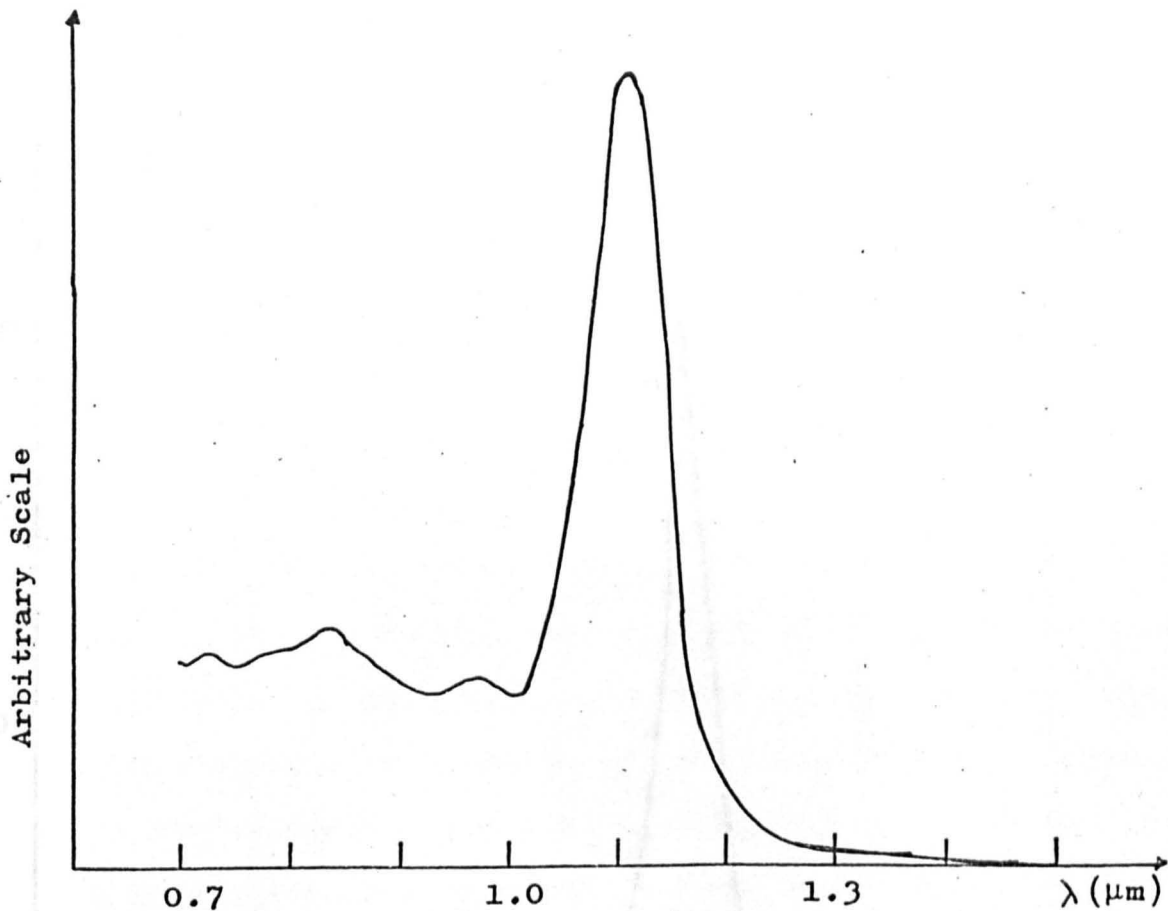


Fig.4-3). Photo response curve of n-type SnS.

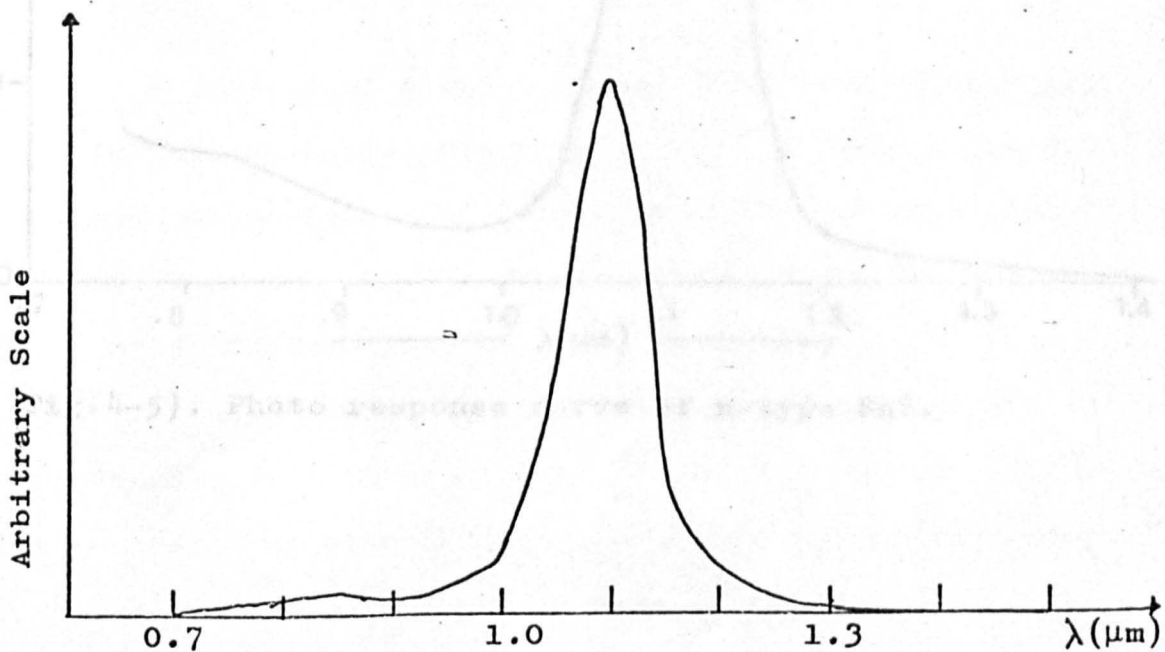


Fig.4-4). Photo response curve of n-type SnS.

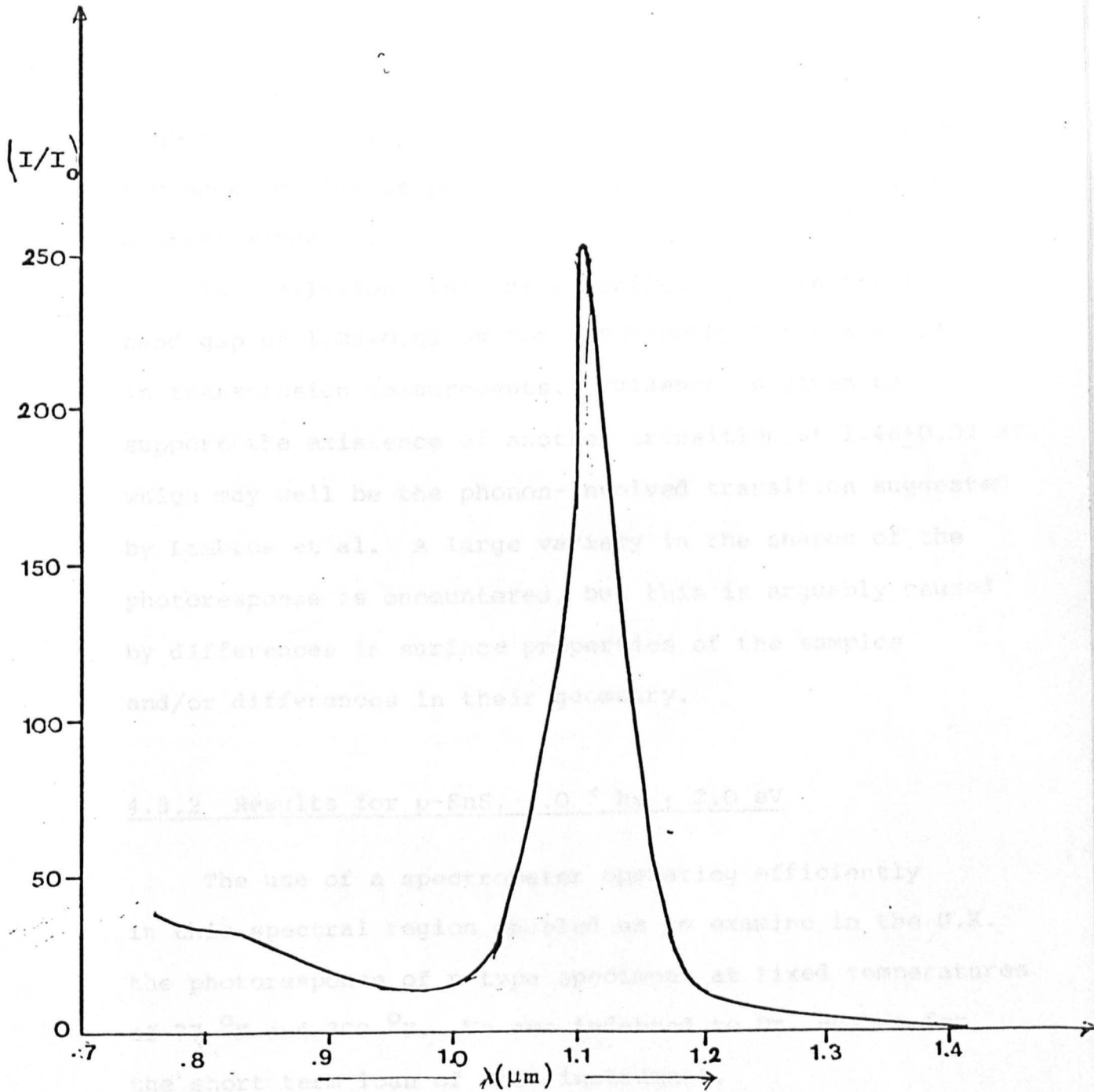


Fig.4-5). Photo response curve of n-type SnS. (means as

described in chapter three. The bias current (dark current) passing through the sample was of the order of the Richardson barrier current, for which values the contacts were non-rectifying. Both the room temperature data, and the 77°K data of necessity, were taken with

is again explained in terms of another transition in the material. The relative size of the response for Figures 4.3 (bulk only illuminated) and 4.4 (bulk plus contacts) is again explained in terms of injection from the acceptor levels provided by the diffusing of indium contact material.

In conclusion, this data confirms the (indirect) band gap of  $1.09 \pm 0.02$  eV for SnS found by Lambros et al. in transmission measurements. Evidence is given to support the existence of another transition at  $1.46 \pm 0.02$  eV which may well be the phonon-involved transition suggested by Lambros et al. A large variety in the shapes of the photoresponse is encountered, but this is arguably caused by differences in surface properties of the samples and/or differences in their geometry.

#### 4.3.2 Results for p-SnS, $1.0 < h\nu < 2.0$ eV

The use of a spectrometer operating efficiently in this spectral region enabled us to examine in the U.K. the photoresponse of p-type specimens at fixed temperatures of  $77^\circ\text{K}$  and  $300^\circ\text{K}$ . We are indebted to Dr. Hughes for the short term loan of this instrument.

Indium contacts were made to these specimens as described in chapter three. The bias current (dark current) passing through the sample was of the order of the Richardson barrier current, for which values the contacts were non-rectifying. Both the room temperature data, and the  $77^\circ\text{K}$  data of necessity, were taken with

the specimen in vacuo. Effects of the ambient atmosphere, producing changes in the surface recombination velocity which have been mentioned in the previous section in connection with the short-wavelength response should not, therefore, be of importance in this case (Bock et al, 1958; Rube 1953, 1960).

We analyse the photoresponse band edge in terms of an indirect transition and a direct transition. The inflexion point  $E_2$  occurring in Fig. 4.6 at  $1.13 \pm 0.02$  eV and  $1.22 \pm 0.02$  eV for  $300^\circ\text{K}$  and  $77^\circ\text{K}$  respectively is associated with the indirect transition and the temperature independent peak  $E_1$  in Fig. 4.7 ( $1.43 \pm 0.02$  eV) is, in our opinion, an indication of a direct valence-band to conduction band transition. We suggest that one of these transitions is direct and the other indirect from the difference in the slopes  $\theta_1$  and  $\theta_2$  of the photoresponse vs. energy curves. An analysis by the author for this material suggests that for indirect transitions the slope of this graph is greater than for direct transitions. Finally, we remark that the shift of the indirect band edge from  $(1.13 \pm 0.02)$  eV at  $300^\circ\text{K}$  to  $(1.22 \pm 0.02)$  eV at  $77^\circ\text{K}$  is consistent with the shift of  $(0.18 \pm 0.05)$  eV found for the shift of the absorption edge between  $300^\circ\text{K}$  and  $100^\circ\text{K}$  by Lambros et al. The photoresponse for  $h\nu > E_g$  in semiconductors is not frequently investigated. As has been remarked earlier the surface properties are of great importance in this régime, and effects due to atmospheric change (Bube, 1953, 1960; Rock, et al. 1958) or the disposition of contacts (Bock) are not unknown. As yet, no band structure is available for SnS and

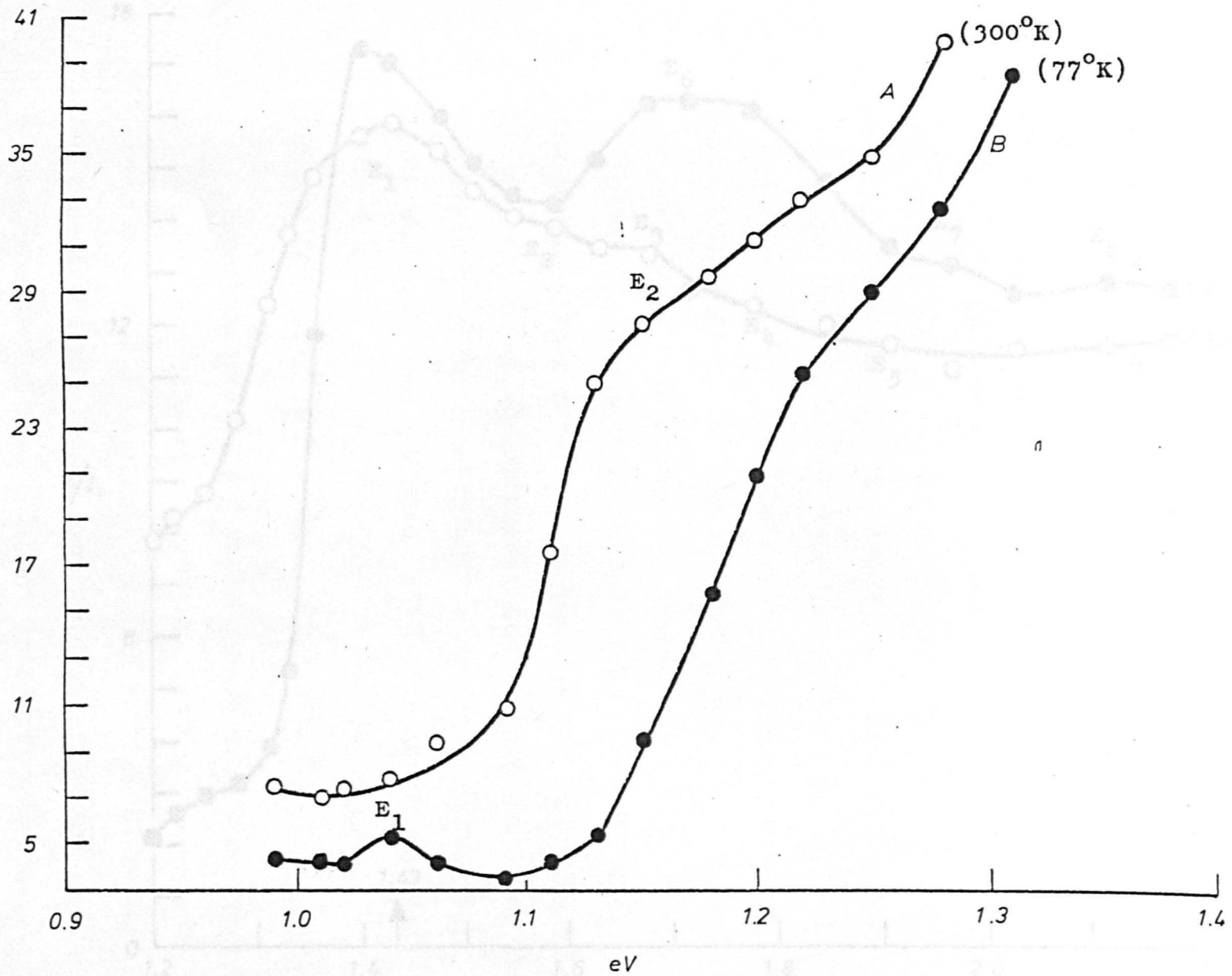


Fig.4-6). Photo response curve of p-type SnS.

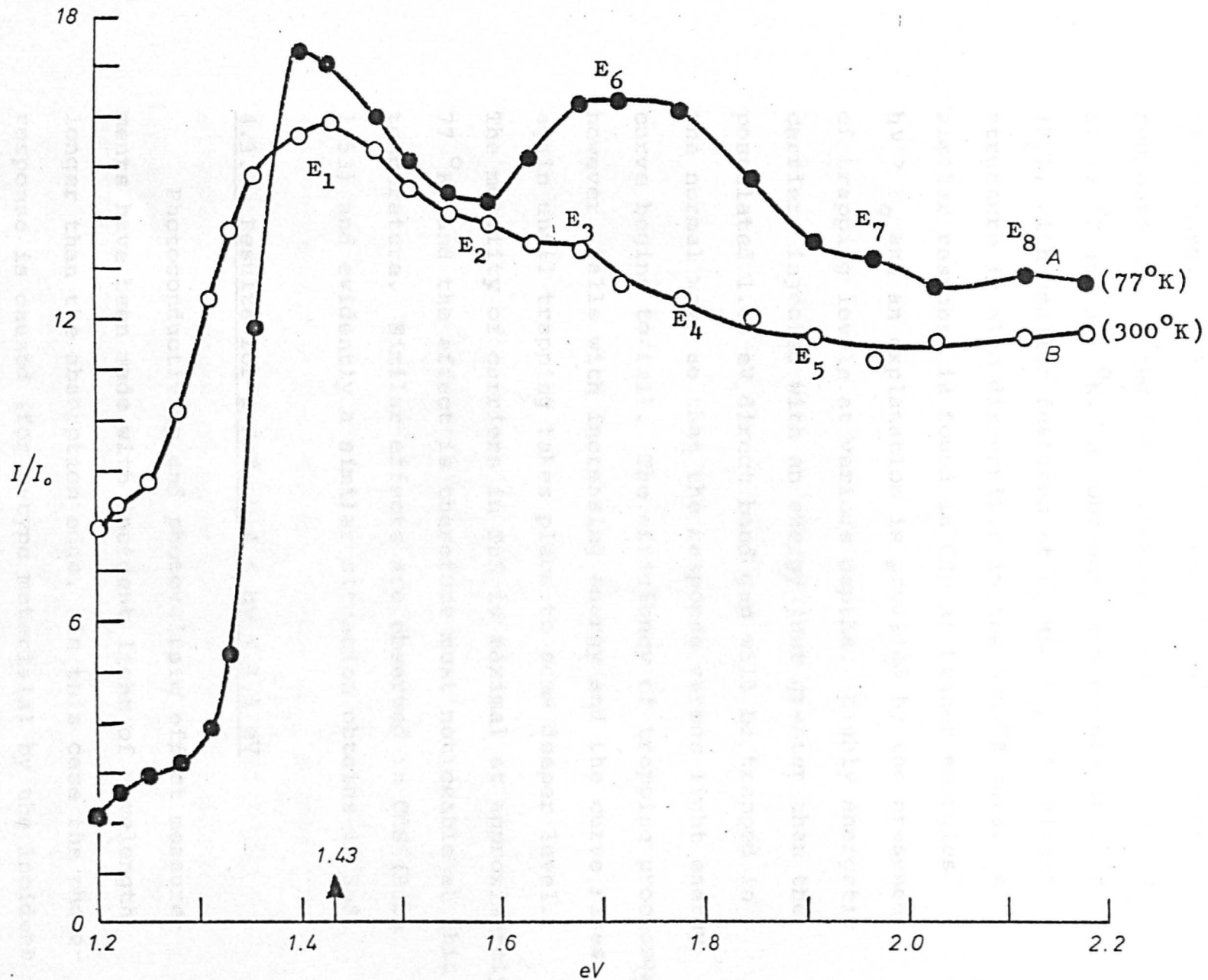


Fig.4-7). Photo response curve of p-type SnS.

therefore no theoretical evidence for other bands which might show up in the high energy photoresponse is to hand. In Figure 4.7, the typical high-energy response is plotted for a freshly cleaved p-type SnS at 77 °K and 300 °K. An obvious bump occurs at 77 °K ( $F_6$ ), with smaller features at  $F_7$  and  $F_8$ ; some slight structure is also discernible in the 300 °K data. A similar response is found in CdS at higher energies  $h\nu > E_g$  and an explanation is provided by the presence of trapping levels at various depths. Highly energetic carriers injected with an energy just greater than the postulated 1.43 eV direct band gap will be trapped in the normal way, so that the response versus light energy curve begins to fall. The efficiency of trapping processes, however, falls with increasing energy and the curve rises again until trapping takes place to some deeper level. The mobility of carriers in SnS is maximal at approximately 77 °K, and the effect is therefore most noticeable at this temperature. Similar effects are observed in CdS (Bube 1953) and evidently a similar situation obtains in SnS.

#### 4.3.3 Results for p-SnS, $0.4 < h\nu < 1.1$ eV

Photoconductivity and photovoltaic effect measurements have been made with incident light of wavelength longer than the absorption edge. In this case the photoresponse is caused (for p-type materials) by the incident photon producing a free hole and an electron bound in the neighbourhood of the acceptor impurity. The photo-

current density will be determined, in the normal way, by the total excess carrier concentrations and their mobilities

with  $\Delta p = \tau \cdot f$ ;

where  $\Delta p$  is the steady state increase in the density of free carriers generated at an excitation rate  $f$ /second per unit volume, with  $\tau$  the free-state lifetime of the carriers. This is the simplest and most general photoconductivity relation.

In this section, we review the photoresponse of p-type SnS samples in the temperature range 40-300 °K for incident radiation of energy less than the band gap. Although several measurements were made down to 10 °K in the cold finger cryostat system, severe contact noise problems were encountered. The 'below gap' photoresponse of semiconductors is not widely investigated, except for the notable exceptions of Ge, Si and CdS. This is perhaps surprising as the technique forms an invaluable complement to absorption spectroscopy and photoluminescence in investigating semiconductor impurities. This ability to resolve structure to a greater extent than is possible with, for example, the absorption spectroscopy technique is reviewed by Kogan and Lifshits (1977) and Collins and Lightowers (1968). We remark that in the case of SnS the present photoresponse data are significantly better than the only absorption data to hand in this spectral region.

Figure 4.8 shows averaged traces of the photoresponse from a typical p-type SnS sample between 40 °K and 300 °K.



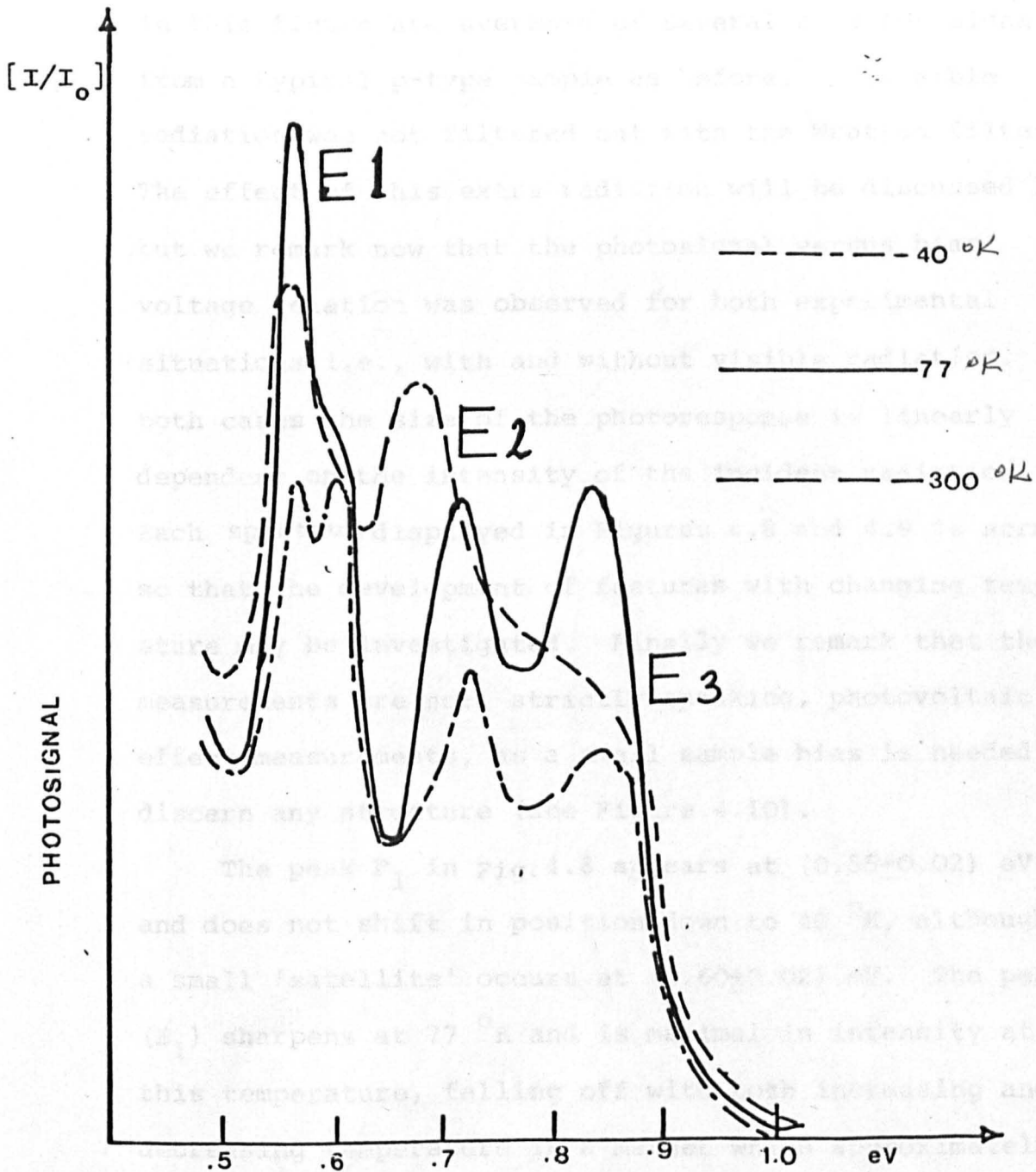


Fig.4-8). Photo response curve of p-type SnS.

A Wratten 70 filter was used in these experiments, which had the effect of cutting out visible ( $\lambda < 700$  nm) light. For low sample bias voltage a linear relation is found between the size of the photosignal and the sample bias, as may be seen in Figure 4.9. The traces in this figure are averages of several runs for signals from a typical p-type sample as before, but visible radiation was not filtered out with the Wratten filter. The effect of this extra radiation will be discussed later but we remark now that the photosignal versus bias voltage relation was observed for both experimental situations i.e., with and without visible radiation. In both cases the size of the photoresponse is linearly dependent on the intensity of the incident radiation. Each spectrum displayed in Figures 4.8 and 4.9 is normalised so that the development of features with changing temperature may be investigated. Finally we remark that these measurements are not, strictly speaking, photovoltaic effect measurements, as a small sample bias is needed to discern any structure (see Figure 4.10).

The peak  $E_1$  in Fig. 4.8 appears at  $(0.55 \pm 0.02)$  eV and does not shift in position down to  $40^\circ\text{K}$ , although a small 'satellite' occurs at  $(0.60 \pm 0.02)$  eV. The peak ( $E_1$ ) sharpens at  $77^\circ\text{K}$  and is maximal in intensity at this temperature, falling off with both increasing and decreasing temperature in a manner which approximately follows the Hall mobility versus temperature behaviour noted by ourselves and Albers (1961). We discuss the

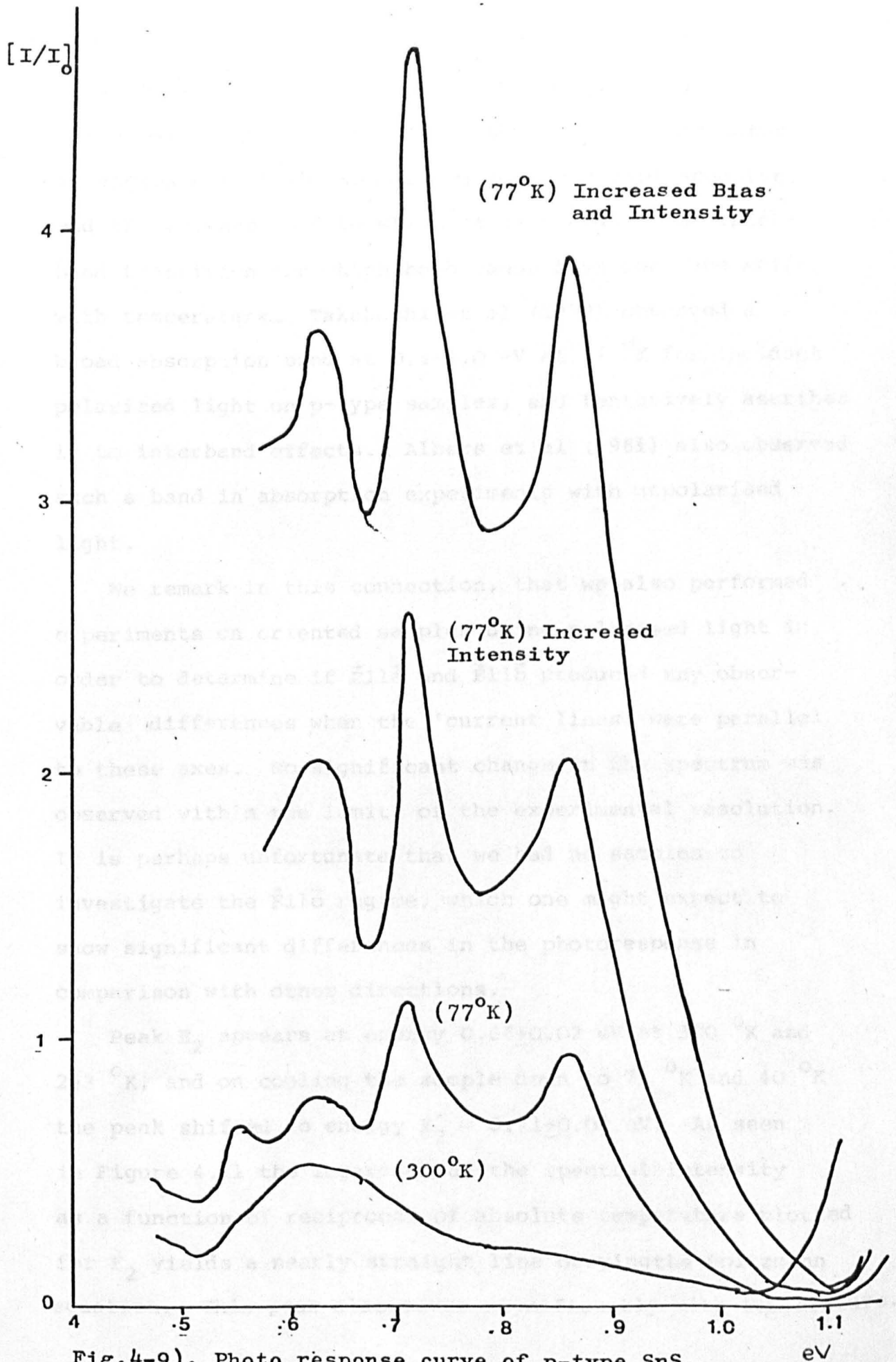


Fig.4-9). Photo response curve of p-type SnS.  
(Visible radiations unfiltered)

possible origins of this peak later, but remark now that the apparent lack of a temperature shift implies a transition between two states moving with temperature at approximately the same rate, e.g. a (deep) acceptor and the valence band to which it is bound, or an interband transition for which both bands have the same shift with temperature. Takahashi et al (1972) observed a broad absorption band at 0.4-1.0 eV at 77 °K for incident polarised light on p-type samples, and tentatively ascribes it to interband effects. Albers et al (1961) also observed such a band in absorption experiments with unpolarised light.

We remark in this connection, that we also performed experiments on oriented samples using polarised light in order to determine if  $\vec{E}11\vec{a}$  and  $\vec{E}11\vec{b}$  produced any observable differences when the 'current lines' were parallel to these axes. No significant change in the spectrum was observed within the limits of the experimental resolution. It is perhaps unfortunate that we had no samples to investigate the  $\vec{E}11\vec{c}$  regime, which one might expect to show significant differences in the photoresponse in comparison with other directions.

Peak  $E_2$  appears at energy  $0.66 \pm 0.02$  eV at 300 °K and 263 °K, and on cooling the sample down to 77 °K and 40 °K the peak shifted to energy  $E_2' = 0.71 \pm 0.02$  eV. As seen in Figure 4.11 the logarithm of the spectral intensity as a function of reciprocal of absolute temperature plotted for  $E_2$  yields a nearly straight line obeying the Boltzmann equation. This peak also moves significantly with temperature.

Fig.4-11).  $\log(I/I_0)$  versus  $I/T$  characteristic of photoconductor.

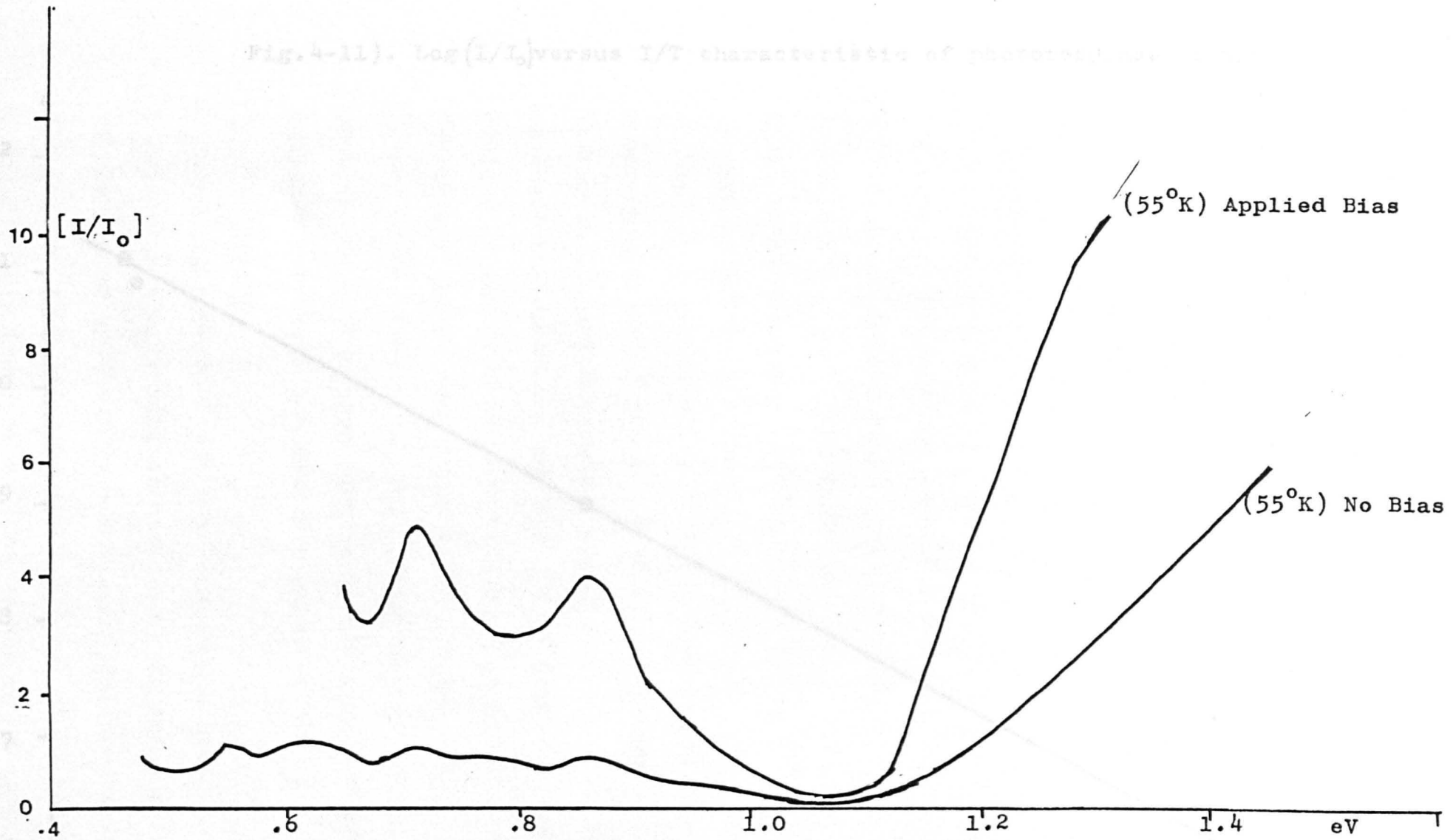
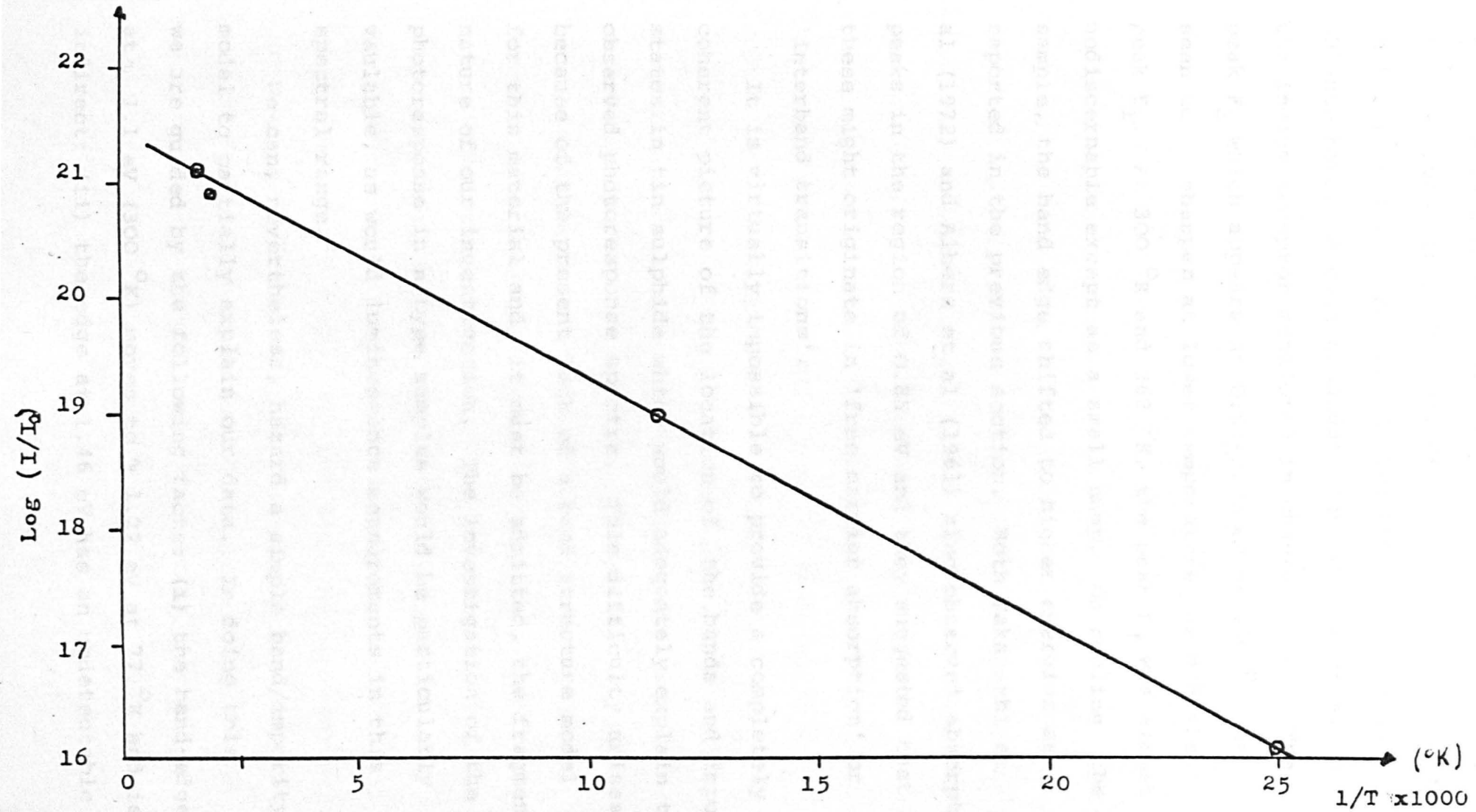


Fig.4-10). Photo response curve of p-type SnS.(Visible radiations unfiltered)

Fig.4-11).  $\text{Log}(I/I_0)$  versus  $I/T$  characteristic of photoresponse of SnS.



This peak is therefore probably due to the transition of an electron from the valence band to localized levels in the forbidden band gap leaving behind a hole in valence band. A good candidate for this level may be the indium acceptor mentioned in chapter three. The peak  $F_3$  which appears at  $0.83 \pm 0.02$  eV at  $300^\circ\text{K}$  was seen to sharpen at lower temperatures as did the peak  $F_1$ . At  $300^\circ\text{K}$  and  $263^\circ\text{K}$ , the peak  $F_3$  was almost undiscernable except as a small hump. On cooling the sample, the band edge shifted to higher energies as reported in the previous section. Both Takahashi et al (1972) and Albers et al (1961) also observed absorption peaks in the region of 0.85 eV and they suggested that these might originate in 'free carrier absorption' or 'interband transitions'.

It is virtually impossible to provide a completely coherent picture of the location of the bands and impurity states in tin sulphide which would adequately explain the observed photoresponse spectra. This difficulty arises because of the present lack of a band structure model for this material and, it must be admitted, the fragmentary nature of our investigation. The investigation of the photoresponse in n-type samples would be particularly valuable, as would luminescence measurements in this spectral range.

We can, nevertheless, hazard a simple band/impurity model to partially explain our data. In doing this we are guided by the following facts: (i) the band-edge at  $\sim 1.1$  eV ( $300^\circ\text{K}$ ) moves to  $\sim 1.22$  eV at  $77^\circ\text{K}$  and is indirect; (ii) the edge at 1.46 eV has an undetectable

shift with temperature and is direct; (iii) the low energy peaks at 0.55 eV and 0.83 eV (77 °K) do not shift significantly with temperature (although they change in size); (iv) the peak at 0.66 eV (300 °K) moves to 0.71 eV (77 °K); (v) the effect of additional irradiation with intrinsic radiation (i.e.,  $F_g < h\nu$ , no wratten 70 filter) is to accentuate some of the low energy peaks and diminish others. The theoretical information that can be brought to bear on the problem is limited; Albers et al (1961) in an analysis of density-of-states masses obtained from free carrier absorption data suggests that there must be at least four equivalent maxima in the valence band. As has been remarked earlier, no band structures are available for the GeSe, GeS, SnS, SnSe members of the IV-VI family. For the other members of the group (e.g. SnTe) the band structure models are legion (see for example, Ota and Rabii, 1974) but a common feature of many of these is the presence of a 'second' valence band, which is not continuous with the first. Such a band may have several equivalent maxima.

In any band/impurity level scheme we must also remember that absorption data imply the presence of broad bands at approximately 0.4 eV and 0.85 eV in contactless samples. These samples were p-type (as were ours) due to the presence of a doubly charged acceptor caused by Sn vacancies and S excess. Our own samples, however, had indium contacts which diffused a little way into the bulk, probably resulting in an indium acceptor in this material. We therefore tentatively suggest that the photoresponse



peaks at 0.55 eV and 0.85 eV may be associated with the doubly charged acceptor, whereas the feature at 0.66 eV is a result of transitions involving the indium impurity. A diagram showing a suggested impurity level/hand model is shown in Figure 4.12. The transitions A(0.83 eV) and B(0.55 eV) are shown between a deep level which we believe may be the second charged state of the doubly charged acceptor, the first charged state occurring at very small energies possibly in the far infrared régime. A similar mechanism has been suggested to explain the photoconductivity of CdS; in this case the presence of a 'deep' (0.9 eV) second charged state of the  $A_2$  doubly-charged acceptor (due to S excess) is invoked to account for the photoresponse both with and without additional radiation with  $h\nu > E_g$ . This radiation (Honig, Moscow conference p.1170, 1968) will clear (or partially clear) the deep trap state of electrons, with the consequent reduction in size of the low energy peak compared with the high energy peak (see Fig. 4.9). The band gap deduced from this scheme is  $1.38 \pm 0.04$  eV which is consistent with  $1.43 \pm 0.02$  eV observed from the photoresponse at higher energies. Furthermore, the lack of a shift in peak positions is consistent with the constancy of the edge which is also found experimentally. Further experiments with n-type samples (which should not display these doubly-charged acceptor transitions) and with p-type samples in the far infrared régime at low temperatures are clearly indicated in order to test the doubly-charged acceptor hypothesis. In the latter case, the photothermalisation technique would be used to investigate the level of the first state

of the doubly-charged acceptor; to perform such experiments, however, would require better contacts to the specimens than are presently obtainable.

The central peak at 0.66 eV (300 °K) moves significantly with temperature, as does the indirect band edge at 1.08 eV. We therefore associate this peak with a transition from an indirect valence band to an indium acceptor state. The other possible transitions from this level to the 'main' valence and conduction bands are either too low (0.4 eV) for our present instrumentation or they lie in the energy range of the indirect edge at 1.1 eV. These are many shortcomings of this model; for example, the small shoulder peak at 0.6 eV is not explained, neither is the shift of peaks  $F_1$  and  $F_3$  (but not  $E_2$ ) in the presence of additional visible light. In the absence of a suitable band structure, however, we propose this scheme as a reasonable possibility.

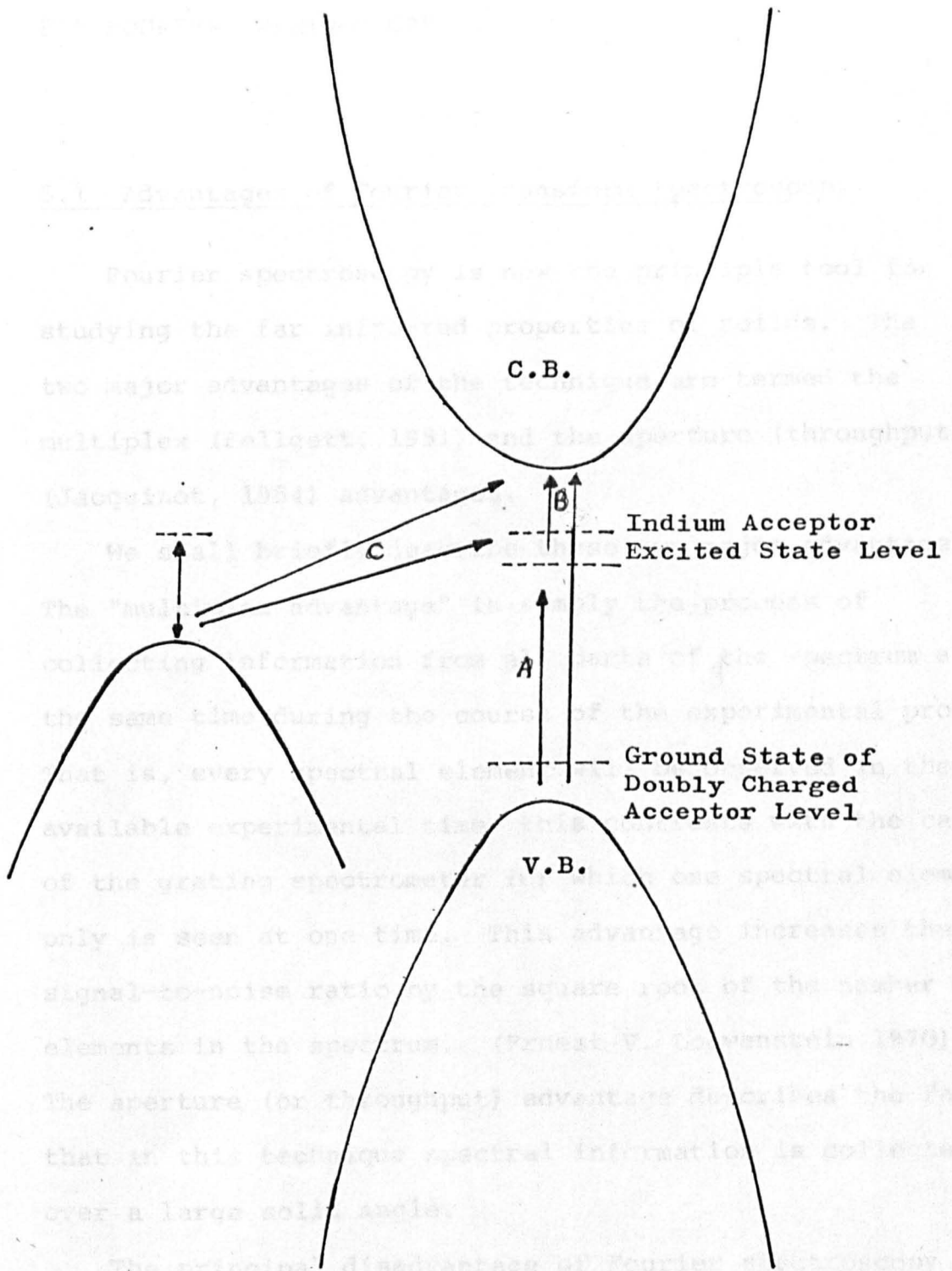


Fig. 4-12). Suggested Band Structure of SnS.

## CHAPTER FIVE

### EXPERIMENTAL TECHNIQUE AND DESCRIPTION OF APPARATUS FOR FOURIER SPECTROSCOPY

#### 5.1 Advantages of Fourier Transform Spectroscopy

Fourier spectroscopy is now the principle tool for studying the far infra-red properties of solids. The two major advantages of the technique are termed the multiplex ( Fellgett, 1951) and the aperture (throughput) (Jacquinot, 1954) advantages.

We shall briefly describe these two major advantages. The "multiplex advantage" is simply the process of collecting information from all parts of the spectrum at the same time during the course of the experimental process. That is, every spectral element will be observed in the available experimental time; this contrasts with the case of the grating spectrometer for which one spectral element only is seen at one time. This advantage increases the signal-to-noise ratio by the square root of the number of elements in the spectrum. (Ernest V. Loewenstein 1970). The aperture (or throughput) advantage describes the fact that in this technique spectral information is collected over a large solid angle.

The principal disadvantage of Fourier spectroscopy is that the spectrum of interest is not immediately observable, but the "interferogram" has to be analysed by a computer to produce the required spectrum. This delay in computing the final spectrum makes it difficult

to monitor the progress of an experiment and constitutes a significant disadvantage. Nevertheless, with a little experience one can deduce the final spectrum, at least approximately, by careful examination of the recorded-interferogram that is obtained in the time domain. In our own laboratory a small computer is available on a time-sharing basis and this enables us to analyse the spectrum while the experiment is going on.

## 5.2 Simple Derivation of Fourier Transform Integrals

Consider first the simple case of incident monochromatic radiation. Let the incident wave be:

$$y = A \exp i(\omega t - 2\pi\sigma_0 x) \quad (5.1)$$

where  $\sigma_0$  is the wavenumber of monochromatic wave. The radiation from the source is split into two beams, either by the beam divider in the Michelson interferometer or by the two sets of mirrors in the Lameller grating interferometer. Before recombination, the beams travel distances  $x_1$  and  $x_2$ , where  $x_1$  and  $x_2$  are the "round trip" distances from the beam divider to the mirrors  $M_1$  and  $M_2$  respectively. (See figure 5.1).

The resultant amplitude in the direction of the detector will be:

$$A_{\text{det}} = A(rt) \left[ \exp i(\omega t - 2\pi\sigma_0 x_1) + \exp i(\omega t - 2\pi\sigma_0 x_2) \right] \quad (5.2)$$

where  $r$  and  $t$  are the reflection and transmission coefficients (of the beam splitter) respectively.

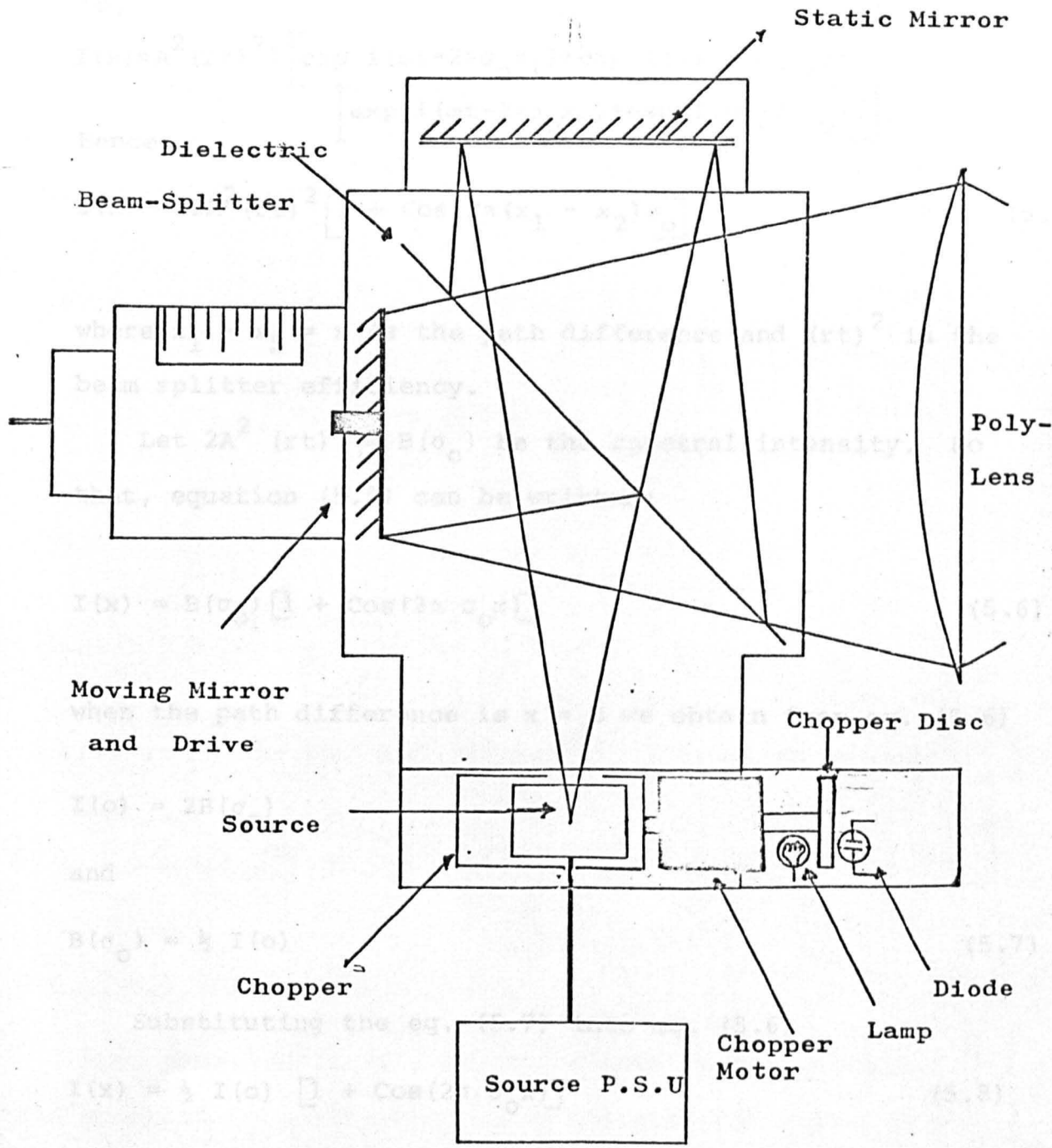


Fig.5-1). A Schematic diagram of Interferometer

The resultant intensity can thus be written:

$$I(x) = A_{\text{det}}^2 / r^2 = A^2 / r^2 \left[ \exp i(\omega t - 2\pi\sigma_0 x_1) + \exp i(\omega t - 2\pi\sigma_0 x_2) \right]^2 \quad (5.3)$$

So,

$$I(x) = A^2 (rt)^2 \left\{ \left[ \exp i(\omega t - 2\pi\sigma_0 x_1) + \exp i(\omega t - 2\pi\sigma_0 x_2) \right] \right. \\ \left. \left[ \exp i(\omega t - 2\pi\sigma_0 x_1) + \exp i(\omega t - 2\pi\sigma_0 x_2) \right] \right\}$$

Hence,

$$I(x) = 2A^2 (rt)^2 \left[ 1 + \cos 2\pi(x_1 - x_2)\sigma_0 \right] \quad (5.5)$$

where  $x_1 - x_2 = x$  is the path difference and  $(rt)^2$  is the beam splitter efficiency.

Let  $2A^2 (rt)^2 = B(\sigma_0)$  be the spectral intensity. So that, equation (5.5) can be written:

$$I(x) = B(\sigma_0) \left[ 1 + \cos(2\pi \sigma_0 x) \right] \quad (5.6)$$

when the path difference is  $x = 0$  we obtain from eq. (5.6)

$$I(0) = 2B(\sigma_0)$$

and

$$B(\sigma_0) = \frac{1}{2} I(0) \quad (5.7)$$

Substituting the eq. (5.7) into eq. (5.6)

$$I(x) = \frac{1}{2} I(0) \left[ 1 + \cos(2\pi \sigma_0 x) \right] \quad (5.8)$$

Up to here, the radiation has been considered to be monochromatic. In fact, the beam is not monochromatic (being provided by a mercury arc source), and thus expression (5.6) becomes in a more general form:

$$I(x) = \int_{-\infty}^{\infty} B(\sigma) [1 + \text{Cos}(2\pi\sigma x)] d\sigma \quad (5.9)$$

which can be re-written as

$$I(x) = \int_{-\infty}^{\infty} B(\sigma) d\sigma + \int_{-\infty}^{\infty} B(\sigma) \text{Cos}(2\pi\sigma x) d\sigma \quad (5.10)$$

or

$$I(x) = \frac{1}{2} I(0) + \int_0^{\infty} B(\sigma) \text{Cos}(2\pi\sigma x) d\sigma$$

hence

$$2 [I(x) - \frac{1}{2} I(0)] = \int_0^{\infty} B(\sigma) \text{Cos}(2\pi\sigma x) d\sigma$$

Finally using the Fourier integral transformation we have

$$B(\sigma) = (\text{Const}) \int_0^{\infty} [I(x) - \frac{1}{2} I(0)] \text{Cos}(2\pi\sigma x) dx \quad (5.11)$$

### 5.3 Computing Techniques

From equation (5.11), it can be seen that to obtain the spectral intensity  $B(\sigma_1)$  from the recorded output  $I(x)$  (the "interferogram") one must perform a Fourier transformation of the form

$$B(\sigma_1) = \int_0^{\infty} [I(x) - \frac{1}{2} I(0)] \text{Cos}(2\pi\sigma_1 x) dx \quad (5.12)$$



In practice, interferograms  $I(x)$  are recorded against the sampled output at regular intervals  $\Delta x$  between the zero path position and some finite maximum path difference  $L_1$  (which is equal to twice the moving mirror displacement in a Michelson interferometer) and thus equation (5.12) is approximated by a summation

$$B(\sigma) = \sum_{\sigma=\sigma_1}^L \left[ I(x) - \frac{1}{2}I(0) \right] \text{Cos} (2\pi\sigma_1 x) \Delta x \quad (5.13)$$

where  $x=h.\Delta x$  and  $h=0,1,2,\dots,\frac{L}{\Delta x}$

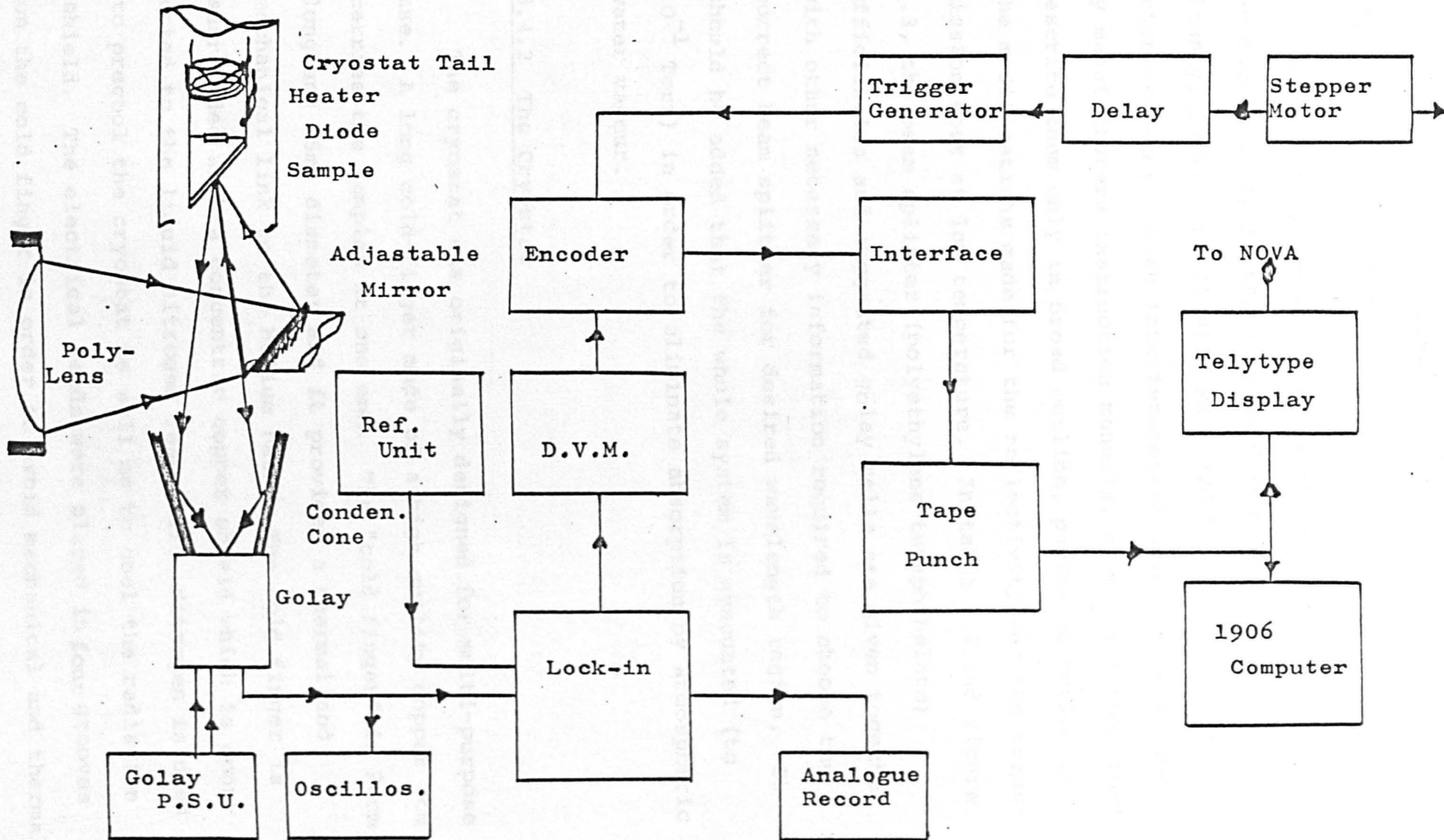
and so on.

We do not discuss in here how digital computers are programmed to analyse the Fourier transformation of digitized interferograms from equation (5.13). These methods are standard and can be supplied by the manufacturers of interferometer. We remark, however, that the interferometer has to be aligned very accurately to get symmetric interferograms, otherwise artefacts are frequently obtained in the computed spectrum, particularly when "single sided" cosine transforms are performed. It cannot be emphasised too strongly that the computed spectrum is sensitive to even one misrecorded data point.

#### 5.4 Description of Apparatus used for Fourier Spectroscopy

In this section the apparatus used for obtaining the far infrared reflectivity and transmission spectrum in the range of temperature 10-300 °K will be described. Figure 5.2 shows a block diagram of the system as a whole, and each stage is described in detail in the subsequent sections.

Figure. 5.2: A Block Diagram of the System for Reflectivity Measurements



#### 5.4.1 The Interferometer

The optical spectra in this work were measured on two machines, 1) An NPL - Grubb Parson modular Michelson (Chanty, 1971), and 2) R11C FS - 720 Fourier transform interferometer. Both interferometers have been described by manufacturers instruction manuals, so we will therefore describe them only in broad outline, paying attention to the modifications made for the reflectivity and the transmission work at low temperature. In table 5.1 and figure 5.3, the beam splitter (polyethylene terephthalate) efficiencies and suggested golay cells are given together with other necessary information required to choose the correct beam splitter for desired wavelength region. It should be added that the whole system is evacuated (to  $10^{-1}$  Torr) in order to eliminate absorption by atmospheric water vapour.

#### 5.4.2 The Cryostat

The cryostat was originally designed for multi-purpose use. A long cold-finger made of a high quality copper rod carries the samples at one end. The "cold finger" is 20cm long and 15mm diameter and it provides a thermal and mechanical link to the helium bath. The cold finger is surrounded with a concentric copper shield which is connected to the liquid nitrogen reservoir. Nitrogen is used to precool the cryostat as well as to cool the radiation shield. The electrical leads were placed in four grooves on the cold finger in order to avoid mechanical and thermal contact with the outer shield. The temperature is measured with two thermocouples (copper against constantan and

T A B L E 5.1

GENERAL RANGE WORKING CONDITIONS

Range	Beamsplitter Thickness	Golay Detector Type
10 - 50cm <sup>-1</sup>	50μ	Quartz or Diamond *
20 - 100cm <sup>-1</sup>	25μ	Quartz or Diamond *
40 - 200cm <sup>-1</sup>	12.5μ	Quartz or Diamond *
80 - 400cm <sup>-1</sup>	6.25μ	Diamond
120 - 660cm <sup>-1</sup>	3.75μ	Diamond
325 - 425cm <sup>-1</sup>	12.5μ (second peak)	Diamond
600 - 800cm <sup>-1</sup>	6.25μ (second peak)	Diamond

\* If a diamond Golay is used here it is recommended that a special wedged quartz filter is interposed in the beam together with the other filtration.

TAPE DATA FOR 5μ PATH DIFFERENCE STEPPING INTERVAL

Resolution cm <sup>-1</sup>	Path Diff. (1 side) mm	Mirror Movement (1 side) mm	Points (1 side)	Tape length (1 side) 3 digit	Tape length (1 side) 4 digit	Time (1 side) mins.
0.1	100	50	20,000	666' 8"	833' 4"	333.30 x t
0.2	50	25	10,000	333' 4"	416' 8"	166.65 x t
0.25	40	20	8,000	266' 8"	333' 4"	133.34 x t
0.5	20	10	4,000	133' 4"	166' 8"	66.67 x t
1.0	10	5	2,000	66' 8"	83' 4"	33.33 x t
2.0	5	2.5	1,000	33' 4"	41' 8"	16.67 x t
2.5	4	2	800	26' 8"	33' 4"	13.33 x t
5.0	2	1	400	13' 4"	16' 8"	6.67 x t
10.0	1	0.5	200	6' 8"	8' 4"	3.33 x t

NOTE: (1) t (column 7) is Pulse Interval in secs.  
 (2) In general, time/mm of mirror movement =  $6\frac{2}{3} \times t$  mins.

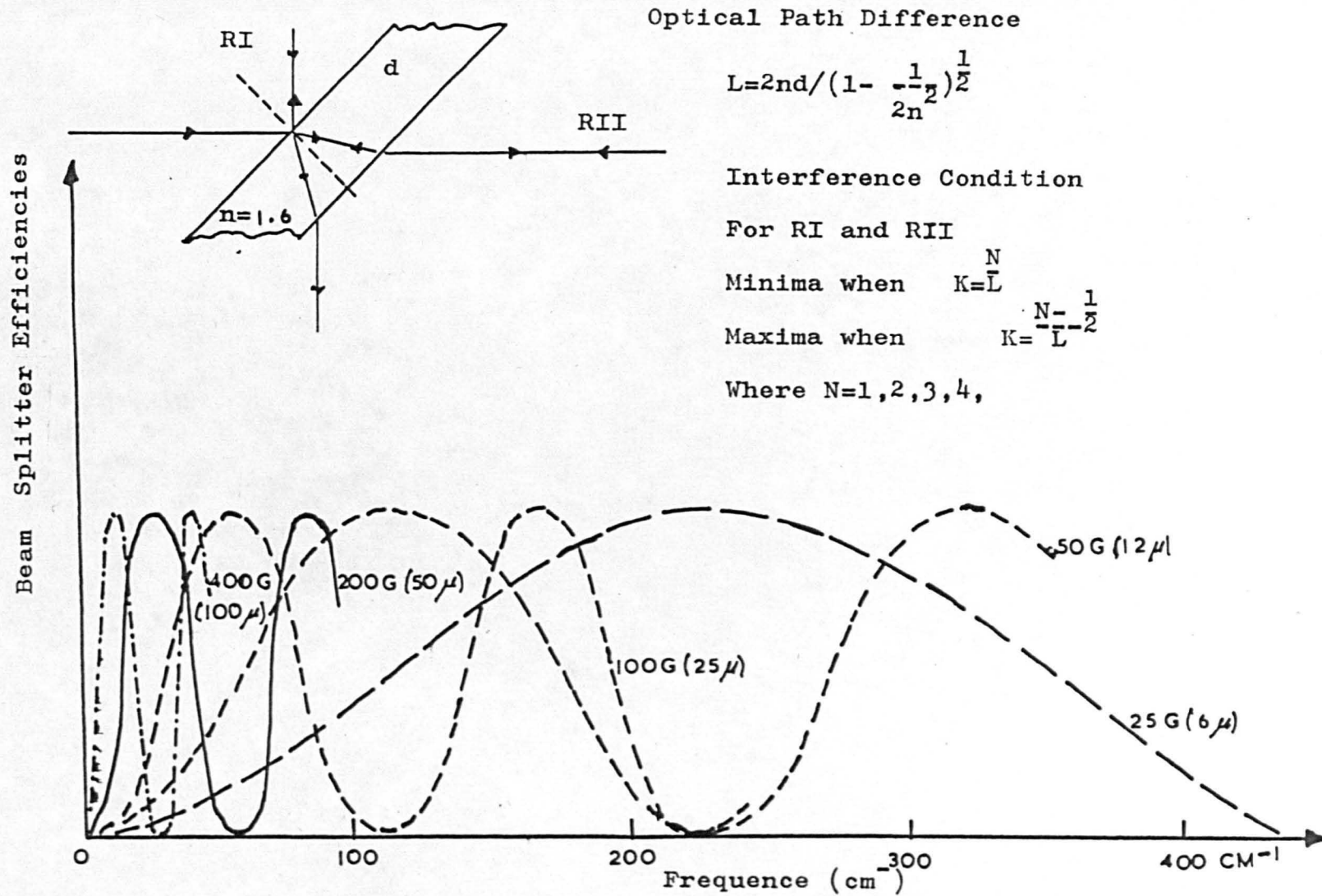


Figure -5.3 : Relative Beam Splitter Efficiencies

Au + 0.03 at o/o Fe against chromel) and one reverse-biased diode (1S44) which is also used as a temperature sensor in the temperature controller. A heater is placed between the specimen and helium bath, made of 30 SWG cotton covered chromel wire of about 40 ohm resistance. The temperature at the end of the cold finger was no more than 5 °K warmer than that of the helium bath. The far end of the cold finger is cut to make a 22.5° angle for reflection measurements. The samples under investigation were mounted on extended cold finger of cryostat using a single blob of durefix. It is to be noted that no collimating optics are employed in the interferometer, so that the divergent light output was focussed on to the specimen using a suitable polyethylene lens (mounted at the interferometer exit) and a freshly aluminised plane mirror.

To avoid unwanted (visible and near infrared) radiation falling on the detector a sufficient thickness of white and black polyethylene windows were used as filters. These filters also acted as windows holding the samples under the desired cryostat vacuum ( $\sim 10^{-5}$  torr or better). Black polyethylene serves to filter out visible and U.V. radiation and white polyethylene acts as a near infrared filter. However, these materials have disadvantages, due to lattice absorptions in the far infrared which are mentioned in chapter six. The losses in the sample chamber and the guiding system (condensing cone) are negligible compared with the other losses occurring in windows and lenses.

In order to minimise the mechanical vibrations arising from external sources such as pumps the cryostat together with the sample chamber is placed on a soft plastic foam, while the interferometer is placed on a heavy table

supported by four "mini" valve springs.

### 5.4.3 Temperature Controller

The precise control of sample temperature using a heater working against a thermal link to a cryostat containing a cryogenic fluid requires complicated electronic circuitry.

Precise temperature controllers have been described previously by Rohlin (1970); Pries and Moore (1970); Głuyas, Hughes and James (1970). These have resistance sensors in one arm of an AC bridge. Temperature controllers using the DC method are described by Fletcher (1971); Plesiewicz and Swiercz (1974); and Jones (1974). Temperature controllers are available commercially, and most of these utilise the DC method.

The system described here is very simple to construct and operate and less expensive than conventional units. It can be used over a wide temperature range, although the control is more precise over some temperature intervals than others.

The heart of the system is the temperature sensing diode (1S44); the forward voltage drop on the diode varies inversely with the surrounding temperature. The voltage across it goes to the input of the first "operational amplifier" 741. (This operational amplifier 741 had a gain unity for simplicity). The inverted output of this 741 is compared with a reference voltage (set externally, according to the desired temperature) at the second (inverting) 741. The output of the second inverting operational amplifier 741 biases an NPN BD121 type trans-





istor which switches on the heater as required. The heater is thus automatically controlled, and switches on and off as necessary.

The heater is wound out of 40 ohm wire and is located on the cold finger between the sample and the cryogenic fluid bath. The power supplies for the heater (24 V) and the IC's (+5 V) are entirely conventional and are shown in figure 5.4.

The main drawback arises due to the variation of the characteristic from diode to diode, so that recalibration is needed every time a diode is changed. Calibration is straightforward and is done by immersing the sensor in appropriate cryogenic fluids. A typical calibration curve is shown in figure 5.5

### 5.5 The Detection System Electronics

Figure 5.2 gives a block diagram of the detection system used for reflectivity and transmission measurements. The reflected or transmitted radiation from the sample is focussed on the window of the Golay detector by a conical light pipe closely attached to the cell. The modulated output signal from the Golay is applied to a lock-in amplifier and also displayed on an oscilloscope. The output of the lock-in amplifier is then fed to a digital voltmeter (DVM) as well as being displayed on a chart recorder. The digitized signal from the DVM is then triggered at the appropriate intervals of optical path difference by command pulses sent from the step drive control unit. These digitized interferogram values are then transferred by means of a suitable interface circuit

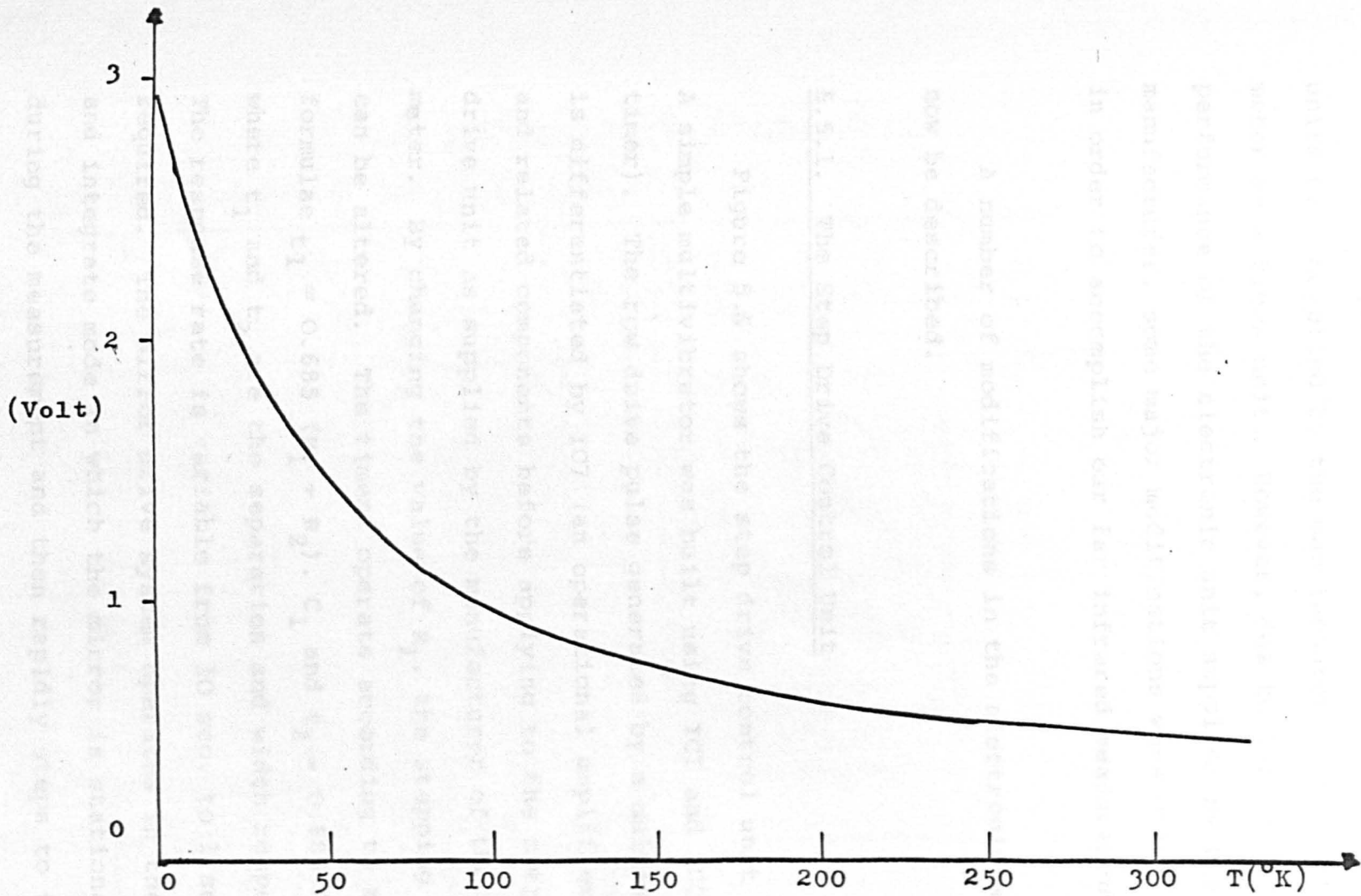


Fig.5-5). A typical calibration curve for diode used for temperature sensor

unit to a punched paper tape to be read into a digital computer.

Originally, as mentioned earlier, all the electronic units were supplied by the manufacturer of the interferometer as a block unit. However, due to lack of adequate performance of the electronic unit supplied by the manufacturer, some major modifications were undertaken in order to accomplish our far infrared measurements.

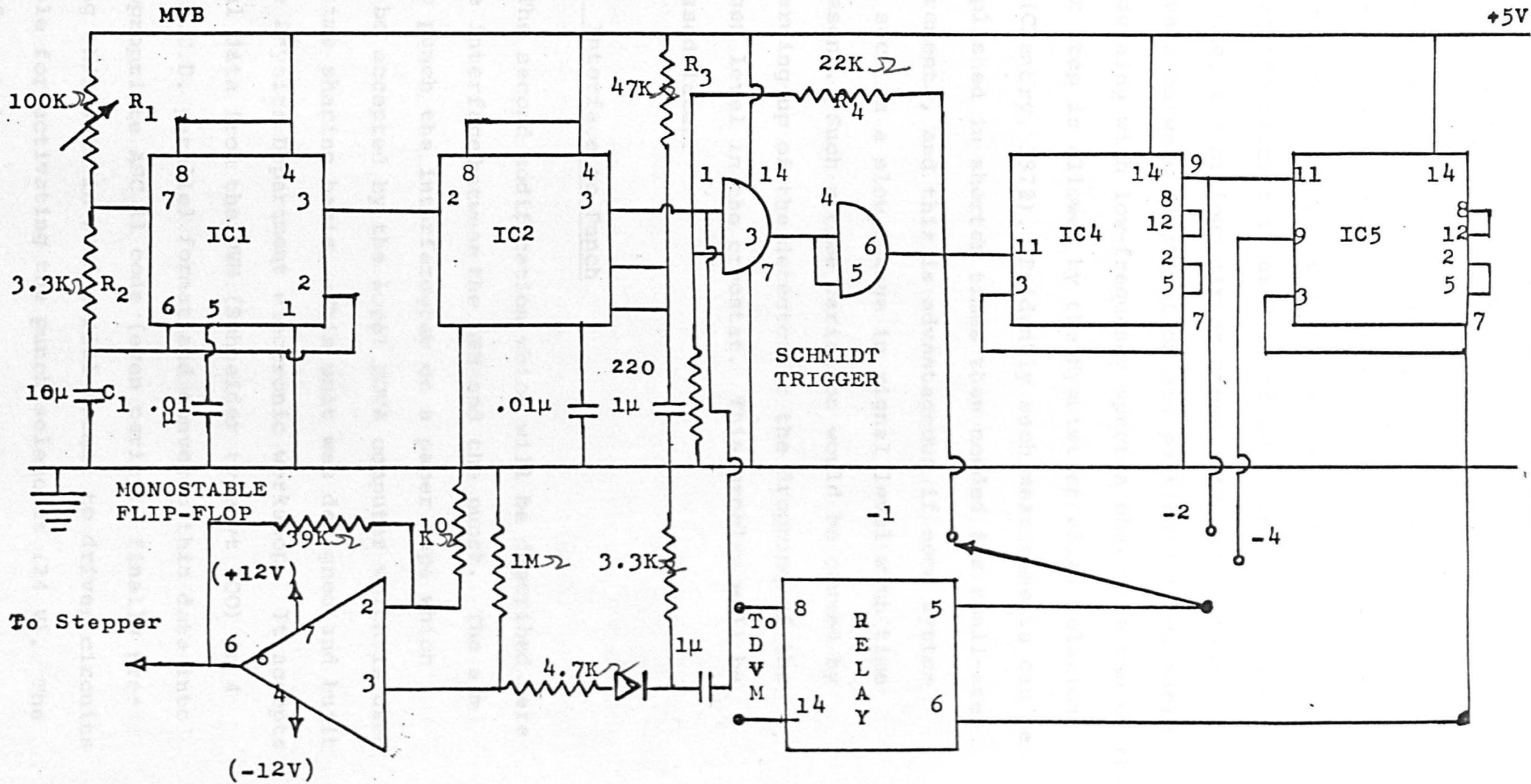
A number of modifications in the electronic system will now be described.

#### 5.5.1. The Step Drive Control Unit

Figure 5.6 shows the step drive control unit circuit. A simple multivibrator was built using IC1 and IC2 (555 timer). The row drive pulse generated by a multivibrator is differentiated by IC7 (an operational amplifier 741) and related components before applying to the stepper motor drive unit as supplied by the manufacturer of the interferometer. By changing the value of  $R_1$ , the stepping rating can be altered. The timers operate according to the formulae  $t_1 = 0.685 (R_1 + R_2) \cdot C_1$  and  $t_2 = 0.685 R_2 \cdot C_1$ , where  $t_1$  and  $t_2$  are the separation and width respectively. The response rate is variable from 10 sec. to 1 sec. as required. The mirror drive system operates in the step and integrate mode in which the mirror is stationary during the measurement and then rapidly steps to the next position. The mirror step is 2.5 m.

The second part of the circuit consists of a schmidt trigger which provides a well-defined square wave pulse,

FIG. 5.6). STEP DRIVE CONTROL UNIT



and two IC4 (7474) integrated circuit chips for delaying the signal provided by the schmidt trigger together with a relay IC6 which triggers the digital voltmeter. This part of the circuit is used to trigger the digitized signal after one, two or four mirror steps. Longer spatial intervals between digitisation are particularly valuable when delaing with low-frequency spectra where the use of the longer step is allowed by the Nyquist or alias relationship (Chantry, 1972). Evidently such measurements can be accomplished in shorter times than needed for small-step measurements, and this is advantageous if some system fault such as a slow change in signal level with time is present. Such a time variation would be caused by the warming-up of the detector or the dropping of the nitrogen level in the cryostat. This problem will be discussed later.

### 5.5.2 Interface to Punch

The second modification which will be described here is the interface between the DVM and the punch. The aim was to punch the interferogram on a paper tape which could be accepted by the local NOVA computer which is used on a time sharing basis. This unit was designed and built in the Physics Department electronic workshop. It accepts digital data from the DVM (Schneider type vt 300) in 4 digit B.C.D. parallel format and converts this data into the appropriate ASC 11 code (even parity), finally presenting this data in word serial format to driver circuits suitable for activating the punch selenoids (24 V). The punch format is:

NULL

1st data digit (most significant magnitude digit)

2nd data digit (next " " " )

3rd " " (next " " " )

4th " " (least " " " )

CR carriage return

LF line feed

This unit provides the facility that after approximately 50 data points are obtained, in a run the tape which has been produced is fed to the NOVA through our own teletype. The resultant spectrum is then immediately analysed and displayed on the teletype. Thus one can see the resultant spectrum roughly and decide at what point to terminate the experiment.

### 5.5.3 Reference Signal

The third modification in the electronic unit which will be described here is the reference signal supply to the lock-in amplifier. Originally, the reference signal was driven from a small generator (as supplied by the manufacturer of the interferometer) driven by the chopper motor, but slipping of the linkage between the motor and generator caused a changing phase shift between the chopper and reference signal, which - if it occurred in the middle of an experimental run - could upset whole sets of experimental results. Wheeler and Hill, (1966) also reported the same problem with the Grubb Parson's unit, so they also modified the original reference signal drive. In our case, the generator was removed and a reference signal for the lock-in amplifier was generated by using the same chopper motor to interrupt a light beam

provided by a small lamp illuminating a high sensitive RS photo-diode which has been placed in the chopper unit housing of the interferometer. A schematic diagram of the reference signal supply is given in figure 5.7; this provides a convenient square-wave signal to our lock-in amplifier reference channel.

#### 5.5.4 Power Supply to the Golay Cell

For all measurements a Golay cell with a diamond window was used as the detector; although, in our opinion, a more sensitive low temperature detector was really needed for some of the reflectivity and transmission experiments which were carried out. Originally, the Golay cell power supply unit was housed in the lock-in amplifier unit supplied by the manufacturers. This lock-in was replaced by a more sensitive (Brookdeal type 9501) unit offering better signal to noise ratios. The Golay cell power supply was thus rebuilt in a separate unit and this is shown in figure 5.8. The +4V power unit is a highly regulated supply obtained from Coutant Instruments. The +90V power unit is home-made and the simple circuit diagram is shown in figure 5.8. The +90V is derived from bridge rectification (Rec 50, RS) of the 125V secondary of a transformer. Simple zener (IS 3082) stabilisation is used; only sufficient current for the Golay cathode follower is needed and this is provided by the circuit shown.

#### 5.6 Optics

Several different sample cell and focussing arrange-

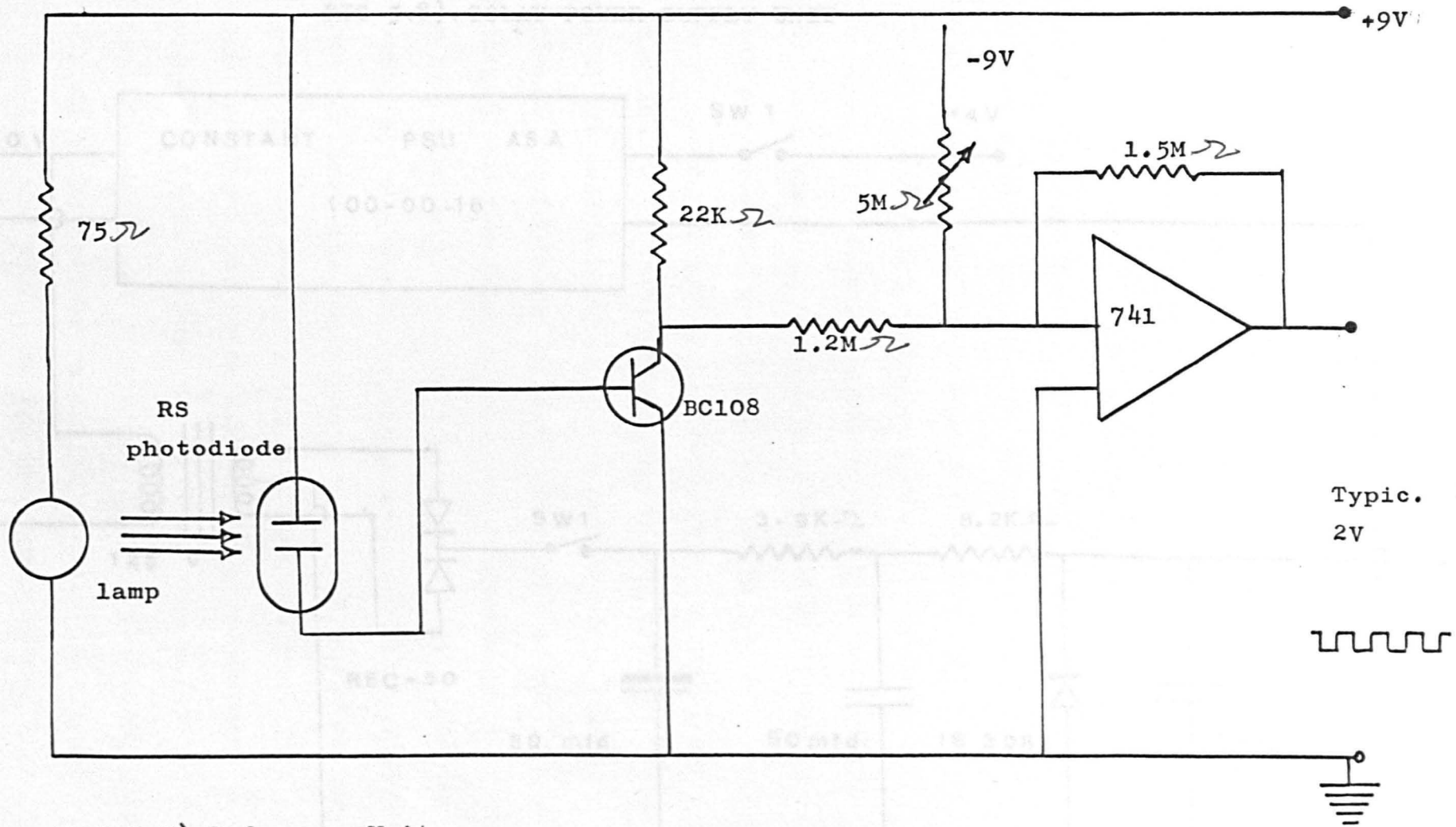
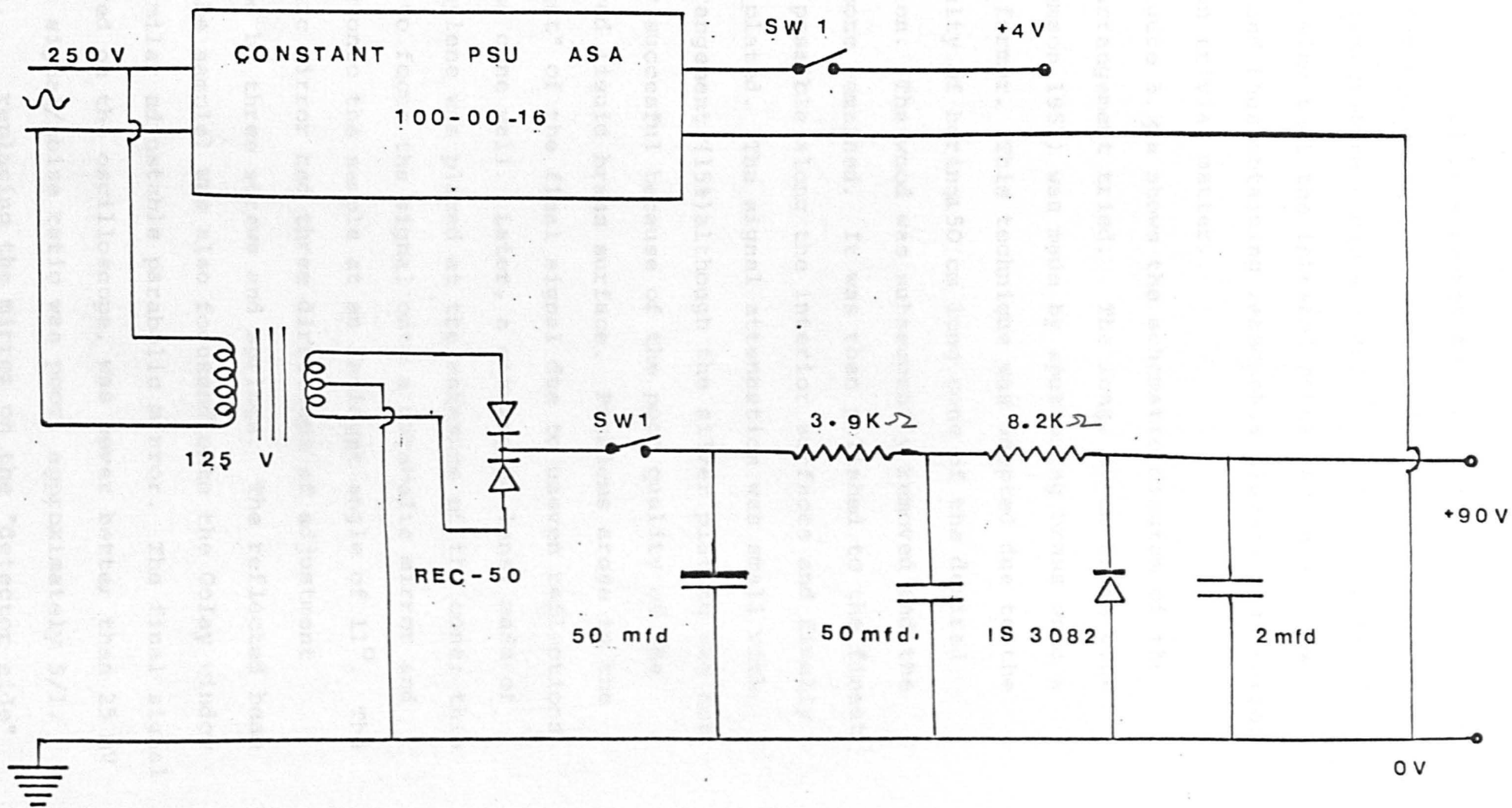


Fig.5-7).Reference Unit.

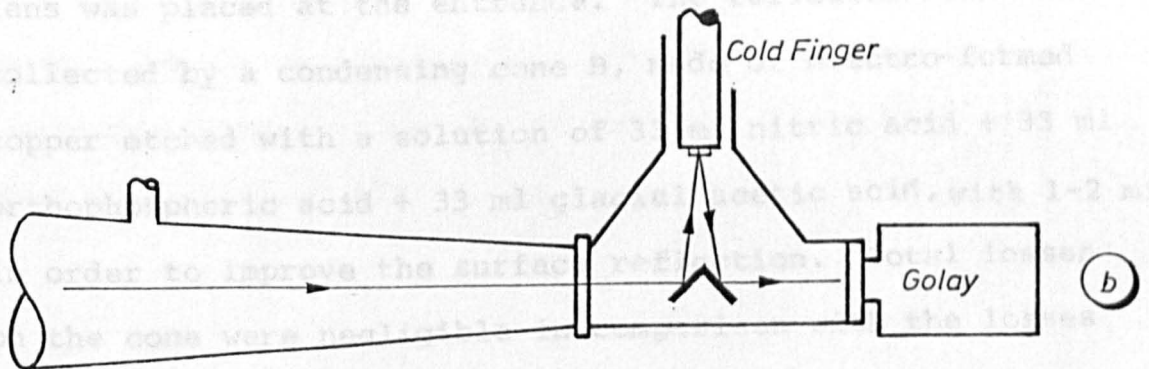
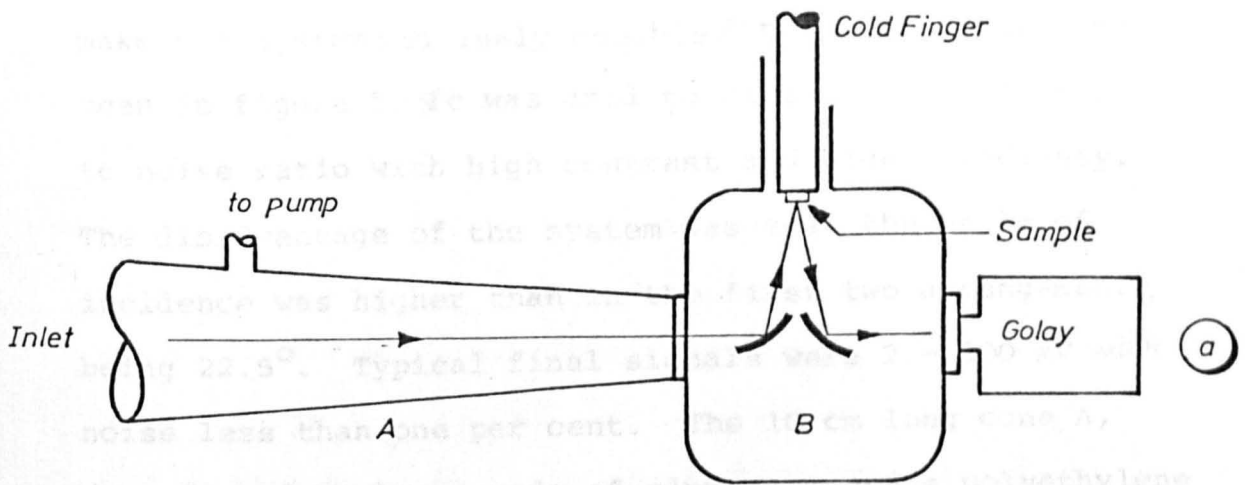


FIG 5-8). GOLAY POWER SUPPLY UNIT

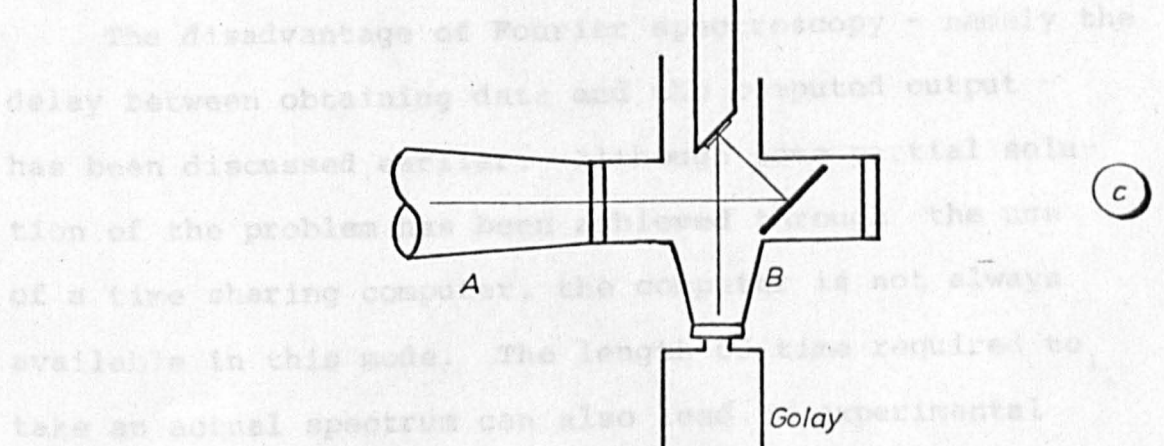


ments were tried in a search for an efficient optical system for measuring sample transmission and reflection in the temperature range 4 - 300 °K. It is essential to bear in mind that the infrared power levels are low ( $\mu$ W's) and thus obtaining reasonable signal:noise ratios is a non trivial matter.

Figure 5.9.a shows the schematic diagram of the first arrangement tried. The long, 50 cm brass cone (Williamson 1952) was made by sputtering brass onto a wooden former. This technique was adopted due to the difficulty of boring a 50 cm long cone of the desired dimension. The wood was subsequently removed and the metal cone remained. It was then polished to the finest finish possible along the interior surfaces and finally silver plated. The signal attenuation was small with this arrangement (15%) although the silver plating was not totally successful because of the poor quality of the sputtered liquid brass surface. Problems arose in the "contrast" of the final signal due to uneven reflections from the cone wall. Later, a converging lens, made of polyethylene was placed at the entrance of the cone; this helped to focus the signal onto a parabolic mirror and finally onto the sample at an incident angle of  $11^\circ$ . The parabolic mirror had three directions of adjustment provided by three screws and springs. The reflected beam (from the sample) was also focussed onto the Golay window by a similar adjustable parabolic mirror. The final signal displayed on the oscilloscope, was never better than 25 mV and the signal/noise ratio was poor, approximately 5/1. However, replacing the mirror on the "detector side" with a torroidal mirror (Fletcher, 1975) slightly



5.3 Problems



not to scale

Fig.5-9).Sample cell arrangements.

improved the signal to noise ratio but not enough to make the system routinely useable. Finally an arrangement seen in figure 5.9c was used to give a better signal to noise ratio with high contrast and high efficiency. The disadvantage of the system was that the angle of incidence was higher than in the first two arrangements, being  $22.5^\circ$ . Typical final signals were 2 - 300 mV with noise less than one per cent. The 10 cm long cone A, (see fig. 5.9c) was made of aluminium and a polyethylene lens was placed at the entrance. The reflected beam was collected by a condensing cone B, made of electro-formed copper etched with a solution of 33 ml nitric acid + 33 ml orthophosphoric acid + 33 ml glacial acetic acid, with 1-2 min. in order to improve the surface reflection. Total losses on the cone were negligible in comparison with the losses on the windows.

### 5.7 Problems

The disadvantage of Fourier spectroscopy - namely the delay between obtaining data and the computed output - has been discussed earlier. Although some partial solution of the problem has been achieved through the use of a time sharing computer, the computer is not always available in this mode. The length of time required to take an actual spectrum can also lead to experimental problems. Typically this can be three hours in high resolution ( $< 1 \text{ cm}^{-1}$ ) work, and during this time a 10 o/o decrease can occur in the signal, presumably due to the warming up of the Golay cell. In addition this time lag can create problems if the cryogenic fluid is used up

during the experimental time. An attempt was made to cool the Golay cell with running water using a tank which was fitted under the Golay base. This created additional noise, however. In order to improve signal to noise ratios, the use of a longer time constant is advisable; however, this also implies an increased sampling interval and a longer experimental run with the concomitant problems discussed above.

The raw data on the punched tape was heavily modulated with the instrumental function. The digitized spectra had to be corrected for this by ratioing to a background tape obtained by replacing the sample with a freshly aluminised mirror.

## CHAPTER SIX

### OPTICAL MEASUREMENTS ON SnS

#### 6.1 Introduction

The IV-VI semiconductors SnS, SnSe, GeSe and GeS form an interesting class of isomorphous materials which are in many ways intermediate between two-dimensional (layer-type) systems and three dimensional crystals. SnS is a fairly typical member of this series: it crystallises into distorted rock-salt structure units arranged in layers with a strong intra-layer (mostly covalent) and weak interlayer coupling (the nature of the weak interlayer forces is poorly understood for all layered crystals, although the assumption of Van der Waals forces is widely made in the absence of better information). These crystals have the appearance of shiny platelets which are easily cleaved in a direction perpendicular to the c-axis.

We have prepared single-crystal specimens of SnS as described in chapter two and performed polarised far infrared reflection and transmission measurements in the temperature range 10-300 °K in order to deter-

mine the fundamental lattice vibrations, and to compare our findings with recent group theoretical predictions (Wiley et al 1976). In addition, far infrared reflection measurements have also been made using partially polarised light (due to the interferometer design) down to 10 °K in order to determine the existence and magnitude of any possible plasma effects. No direct measurements of the phonon frequencies have been previously made in this material. We note, however, that Lambros et al (1974) suggest a phonon frequency of approximately  $320 \text{ cm}^{-1}$  which is involved in an indirect band gap transition obtained optically.

## 6.2 Experimental Details

A study of the infrared reflectivity and transmission spectra of semiconductors allows us to explore, amongst other things, their fundamental lattice vibrations and free carrier dispersion. The first measurement of the far infrared reflectivity spectrum in the range  $50 - 350 \text{ cm}^{-1}$  of SnS was that of Nikolic et al (1974). These authors measured the room temperature reflectivity spectrum of p-type crystalline SnS at normal incidence and calculated the optical constants of SnS using Kramers-Krönig analysis. The present investigation adds to this earlier study by extending the measurements to lower temperatures using both partially polarised (due to the interferometer design) and

polarized light (Chamberlain et al 1976). We now have a fairly clear interpretation of our results in terms of a straightforward lattice mode reflection.

Single crystals of SnS were grown by the Bridgman technique. The details of these preparations have been given elsewhere (Nikolic et al 1974, and in chapter two in this thesis). Samples in this work had a surface area  $0.5 - 1 \text{ cm}^2$ . Our samples were analysed by x-ray techniques and confirmed to be single crystal. Subsequent electrical characterization measurements indicated that our specimens were both n- and p-type, with carrier concentrations in the range  $10^{17} - 10^{19} \text{ cm}^{-3}$  (see chapter three). Measurements with  $\vec{E} \parallel \vec{a}$  and  $\vec{E} \parallel \vec{b}$  were performed on (040) cleavage faces, the cleavage being done simply by peeling off with sellotape one of the layers from SnS plates which had been cut from the ingot, thereby exposing a fresh face. Samples of 2 - 5  $\mu\text{m}$  thickness for transmission experiments were prepared by bonding the crystal with a cyanoacrylate ester adhesive to a polythene substrate, and then peeling off successive layers with sellotape. In this way relatively strain-free specimens were obtained.

Near normal incidence polarized far infrared reflectivity and transmission measurements were made in the frequency range  $10 - 500 \text{ cm}^{-1}$  using conventional Fourier transform techniques. A commercial



optical cryostat DN 70, (Oxford Instruments Ltd.) was used for the measurements in the range 77 - 300 °K, and the home made cold-finger cryostat was used for both polarized and partially polarized light studies in the lower temperature range 10 - 77 °K. In the case of the polarized light measurements wire-grid polariser (Auton, 1967) was used to polarise the light falling on the specimen:

Samples were oriented to locate  $\vec{a}$  and  $\vec{b}$  axes before insertion into the spectrometer by X-ray technique. The specimens were simply mounted using Durofix. A block diagram of the apparatus used for the reflectivity and transmission measurements has been given in chapter five together with a detailed description of the experimental technique.

### 6.3 Results

The reflection spectrum of the n-type SnS samples was measured from 50 to 350  $\text{cm}^{-1}$  with  $\vec{E} \parallel$  to  $\vec{a}$ ,  $\vec{b}$  and  $\vec{c}$ , axis at room temperature, displayed in figure 6.1. These measurements were performed on either FS - 720 Beckman or a NPL - Grubb Parsons modular Michelson interferometer (Chantry et al 1969). The frequency resolution was 2.5  $\text{cm}^{-1}$ , sufficient to resolve all the structure observed in the spectra, and the data presented are averaged over many spectra. The spectra  $\vec{E} \parallel \vec{a}$  and  $\vec{E} \parallel \vec{b}$  were taken on an oriented

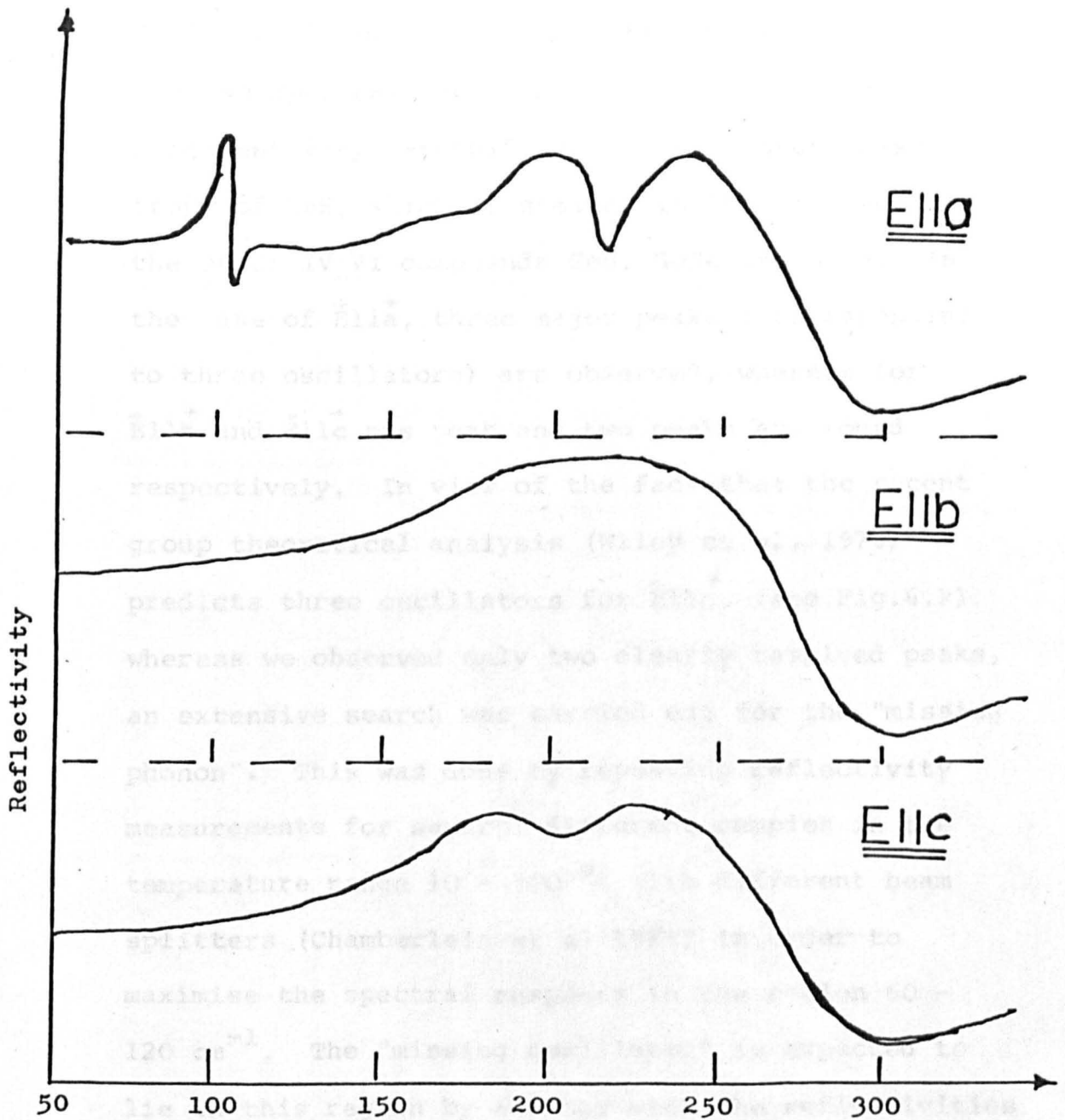


Fig.6-1). Averaged reflectivity versus wavenumber curves for  $E||a$ ,  $E||b$  and  $E||c$  for an n-type SnS sample at 300K.

cleavage face using a wire-grid polarizer (Auton, 1967) which could be rotated by  $90^\circ$  for recording the two spectra. It is necessary to perform this alignment very carefully due to the large anisotropy of SnS, which is similar in this respect to the other IV-VI compounds GeS, GeSe and SnSe. In the case of  $\vec{E}11a$ , three major peaks (corresponding to three oscillators) are observed, whereas for  $\vec{E}11b$  and  $\vec{E}11c$  one peak and two peaks are found respectively. In view of the fact that the recent group theoretical analysis (Wiley et al, 1976) predicts three oscillators for  $\vec{E}11c$ , (see Fig.6.2) whereas we observed only two clearly resolved peaks, an extensive search was carried out for the "missing phonon". This was done by repeating reflectivity measurements for several different samples in the temperature range 10 - 300 °K with different beam splitters (Chamberlain et al 1966) in order to maximise the spectral response in the region 60 - 120  $\text{cm}^{-1}$ . The "missing oscillator" is expected to lie in this region by analogy with the reflectivities of the isomorphs GeSe and GeS (Wiley et al 1976, Chandrasekhar and Zwick 1976). However, despite careful sample surface preparation, interferometer optimization and (in addition) transmission measurements over the 50 - 350  $\text{cm}^{-1}$  range, we cannot conclusively observe the presence of an oscillator in this region with a strength greater than one-hundredth of the strength of the other two oscillators

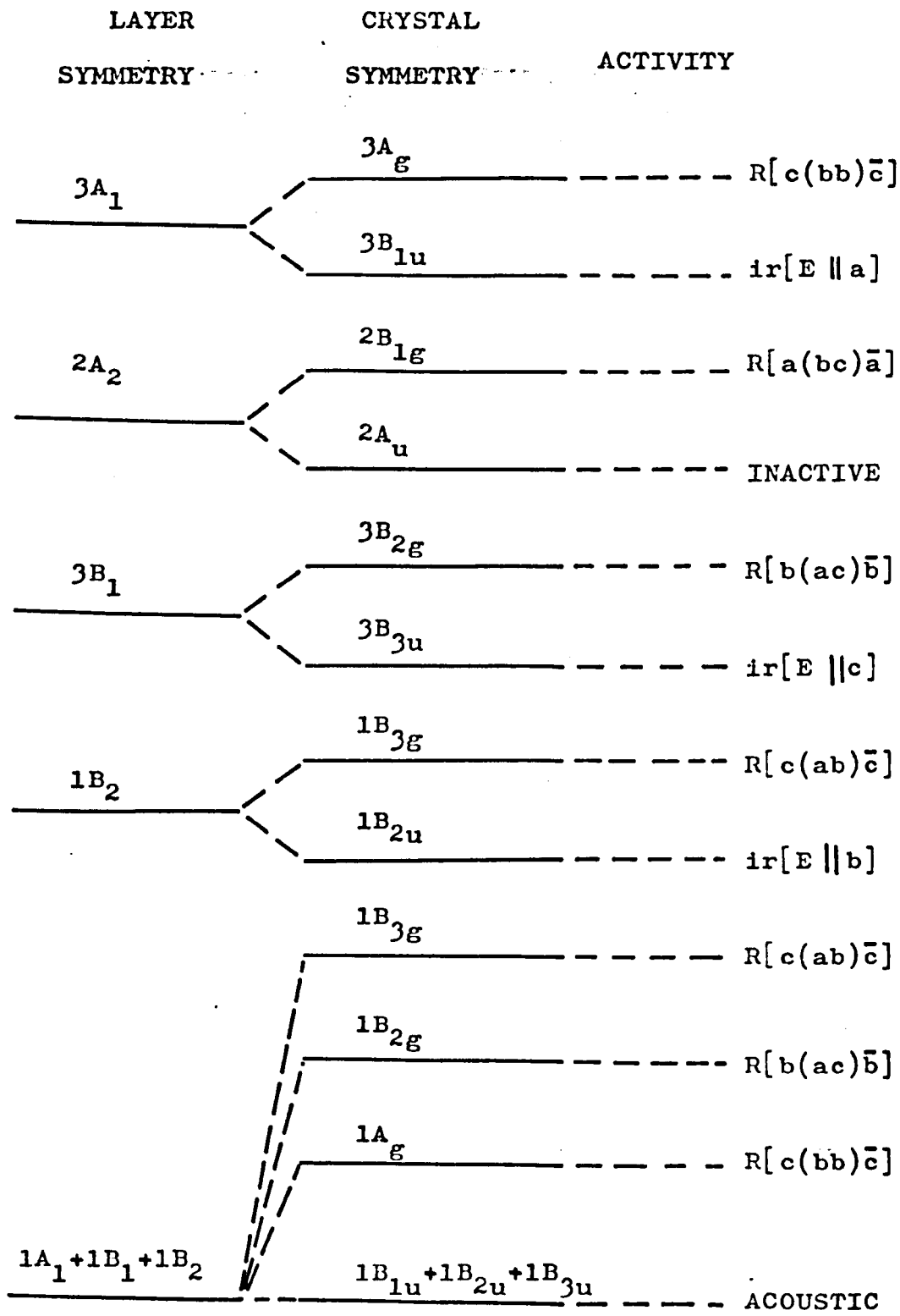


Fig.6-2). Diagram showing the relations among phonons for individual layers and three dimensional crystals of GeS. (After Wiley et al ,1976)

observed for  $\vec{\Gamma}11\vec{C}$ . In order to obtain the optical constants of SnS, the experimental reflectivity curves measured by Fourier transform spectroscopy for  $\vec{\Gamma}11\vec{a}$  and  $\vec{\Gamma}11\vec{b}$  directions were analysed by means of Kramers-Krönig techniques: the results of this analysis are displayed in Figure 6.3 and in table one, which also displays the phonon mode assignments based on Wiley's recent group theoretical treatment. The computer programs used in this laboratory for the Fourier transform and Kramers-Krönig analysis has been given elsewhere (Doran, 1974). Table two displays the oscillator parameters and dielectric constants for the three polarization directions which have been deduced from a least-squares computer fit to the experimental reflectivity data. This program (together with an explanation of its physical basis) is given as an appendix B to this chapter. Table three shows the changes in transverse optical phonon frequencies  $\omega_T$  (for the  $\vec{\Gamma}11\vec{a}$  and  $\vec{\Gamma}11\vec{b}$  directions) in the temperature range 10 - 300 °K, as deduced from the shift in the reflectivity peak positions over this temperature range, and from the results of the room temperature Kramers-Krönig analysis.  $\vec{\Gamma}11\vec{C}$  data values of the frequency shifts are not displayed because of the uncertainty in this direction of polarization. As seen from the numerical values of the transverse optical phonons there is a negligible change down to 50°K. One might say, that transverse optical

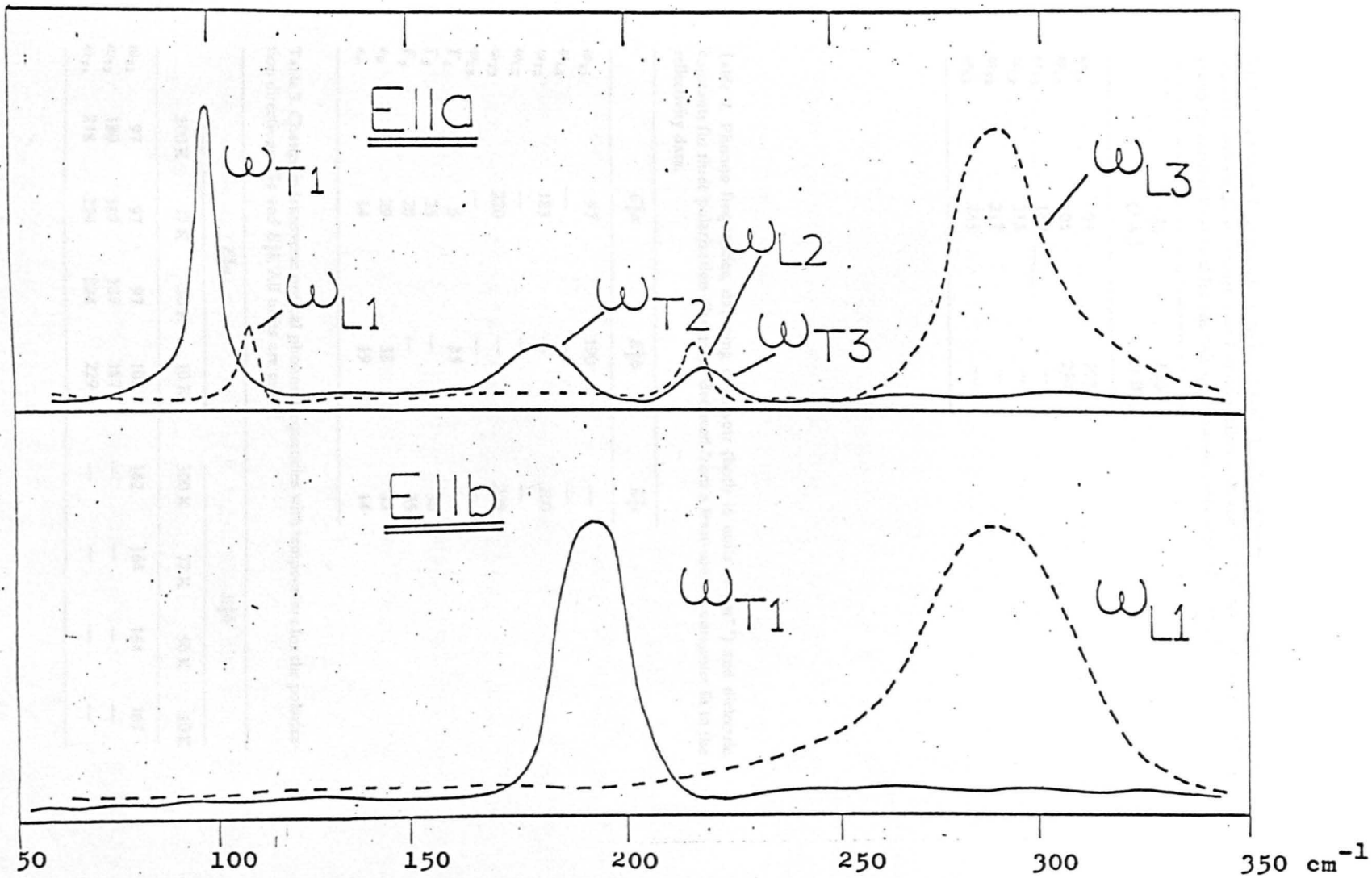


Fig.6-3). Kramers-Kronig analysis of the reflectivity spectra.

Table 1. Phonon frequencies (in units of  $\text{cm}^{-1}$ ) deduced from Kramers-Krönig analysis, together with mode assignments and multiplicities deduced from Wiley's (Wiley *et al* 1976) group theoretical treatment for the two polarization directions  $E\parallel a$  and  $E\parallel b$ .

	$E\parallel a$ (3 $A_1$ )	$E\parallel b$ (1 $B_{2u}$ )
$\omega_{T1}$	97	192
$\omega_{L1}$	105	290
$\omega_{T2}$	180	—
$\omega_{L2}$	215	—
$\omega_{T3}$	217	—
$\omega_{L3}$	285	—

Table 2. Phonon frequencies, damping constants (both in units of  $\text{cm}^{-1}$ ) and dielectric constants for three polarization directions deduced from a least-squares computer fit to the reflectivity data.

	$E\parallel a$	$E\parallel b$	$E\parallel c$
$\omega_{T1}$	97	190	—
$\omega_{L1}$	—	—	—
$\omega_{T2}$	183	—	200
$\omega_{L2}$	—	—	—
$\omega_{T3}$	220	—	229
$\omega_{L3}$	—	—	—
$\Gamma_1$	5	35	—
$\Gamma_2$	25	—	30
$\Gamma_3$	20	—	35
$\epsilon_0$	29	38	33
$\epsilon_\infty$	14	19	14

Table 3. Change in transverse optical phonon frequencies with temperature for the polarization directions  $E\parallel a$  and  $E\parallel b$ . All units are  $\text{cm}^{-1}$ .

	$E\parallel a$				$E\parallel b$			
	300 K	77 K	50 K	10 K	300 K	77 K	50 K	10 K
$\omega_{T1}$	97	97	97	102	162	164	164	167
$\omega_{T2}$	180	187	187	187	—	—	—	—
$\omega_{T3}$	218	224	224	229	—	—	—	—

phonon frequencies of SnS are temperature-independent in the range 50 - 300 °K.

A typical room temperature absorption curve for a thin n-type sample with  $\bar{\epsilon}_{11}^+$  (suitably corrected for surface reflectivity changes) is shown in Figure 6.4. Three oscillator peaks corresponding to the transverse optical frequencies are displayed clearly. A peak appearing about at  $100 \text{ cm}^{-1}$  is sharper than those of two others; these results are very sensitive to orientation of the sample. In addition, transmission data were also taken at room temperature on this specimen in the spectral range  $500 - 4000 \text{ cm}^{-1}$  using a Perkin-Elmer monochromator system. This data showed interference fringes, from which a value of the high-frequency dielectric constant could be deduced as a check on the value obtained from Kramers-Krönig and curve-fitting analysis of the reflectivity data. The thickness of this sample was measured to be  $2.5 \mu\text{m}$ , using a talystep gauge. The reflection spectra for n-type samples are essentially similar in the temperature range 50 - 300 °K to those obtained at room temperature, and are not therefore displayed separately. In this temperature range the p-type samples also show the same characteristic features.



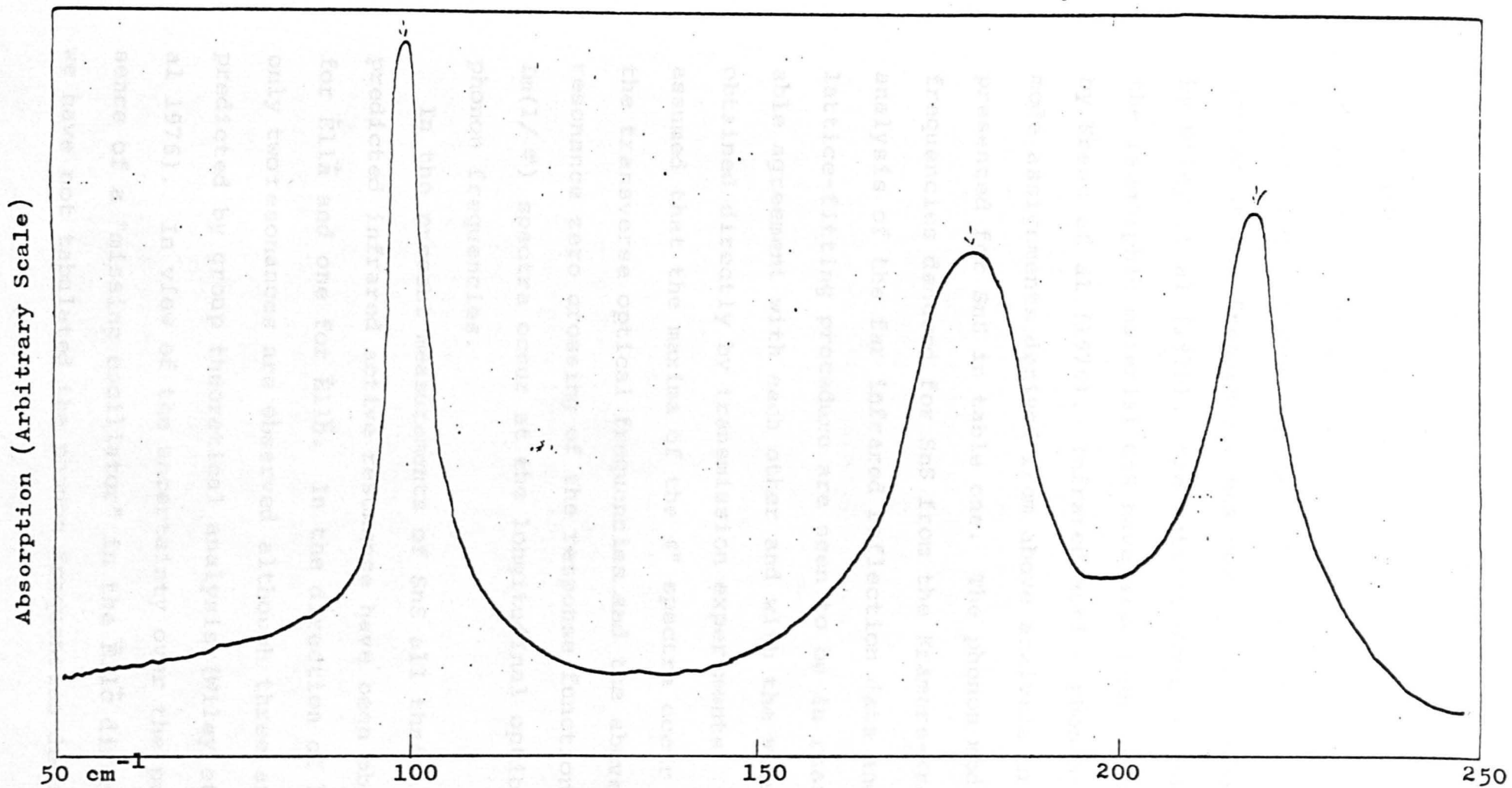


Fig.6-4). Absorption versus wavenumber for a 2.5 $\mu$ m thick sample of n-SnS at 300K with E||a.

## 6.7 Discussions

A factor-group analysis of the  $D_{2h}^{16}$  space group of SnS, together with a discussion of the effects of interlayer forces on the infrared and Raman-active phonon frequencies has been given recently by Wiley et al (1976). The lattice dynamics of the isomorphic material GeS have also been reported by Kress et al (1976). Infrared active phonon-mode assignments derived from above analysis are presented for SnS in table one. The phonon mode frequencies deduced for SnS from the Kramers-Krönig analysis of the far infrared reflection data and lattice-fitting procedure are seen to be in reasonable agreement with each other and with the values obtained directly by transmission experiments. We assumed that the maxima of the  $\epsilon''$  spectra occur at the transverse optical frequencies and the above resonance zero crossing of the response function  $\text{Im}(1/\epsilon')$  spectra occur at the longitudinal optical phonon frequencies.

In the present measurements of SnS all three predicted infrared active resonance have been observed for  $\vec{E}11\vec{a}$  and one for  $\vec{E}11\vec{b}$ . In the direction of  $\vec{E}11\vec{c}$  only two resonances are observed although three are predicted by group theoretical analysis (Wiley et al 1976). In view of the uncertainty over the presence of a "missing oscillator" in the  $\vec{E}11\vec{c}$  direction, we have not tabulated the phonon frequencies deduced

from the Kramers-Krönig analysis data in this polarization direction. We have, however, applied the lattice fitting procedure to our  $\vec{E}||\vec{c}$  reflectivity spectra assuming the presence of two oscillators in order to furnish approximate phonon frequency and dielectric constant values. We notice in this connection that Wiley et al (1976) have similarly failed to observe a predicted phonon in the  $\vec{E}||\vec{a}$  polarization direction for GeS, although recently Nikolic and Hughes (1976) have been able to observe the presence of this oscillator in their reflection spectra. Evidently, very careful sample surface preparation techniques are required for specimens cut perpendicular to the  $\vec{c}$  faces because of scattering effects from irregularities in the exposed layer ends.

It is of interest to compare the SnS optical phonon frequencies with those recently obtained for the isomorphic systems GeSe and GeS (Wiley et al 1976, Chandrasekhar and Zwick 1976, Siapakas et al 1976). (See table four). In the case of SnS the transverse optical phonon modes lie in all cases between the two corresponding values for the Germanium chalcogenides. That is, the ratio of the corresponding frequencies of GeSe and GeS are related by a simple scaling factor for most of the high frequency phonons in the layer plane. This frequency ratio,

$$\frac{\gamma(\text{GeSe})}{\gamma(\text{GeS})} \approx 0.73 \pm 0.06 \quad \text{and} \quad \frac{\gamma(\text{GeSe})}{\gamma(\text{SnS})} \approx 0.88 \pm 0.07.$$

We may assume that SnS is more layer-like than GeS and less layer-like than GeSe.

This conclusion was also reached by Chandrasekhar and Zwick (1976) using a similar dynamical argument.

We also attempted to perform Raman Scattering experiments on our samples at Leicester University using a Kr 6371 Å laser system. The combination of Raman and Infrared spectroscopy would here allow us to obtain complete information about the ratio of interlayer to intralayer coupling, as has been done for GeSe (Chandrasekhar and Zwick 1976) and GeS (Wiley et al, 1976). Unfortunately, these experiments were not a success and this was, in our opinion, caused by the poor signal/noise ratios obtainable in the detection system. Finally, we have appended in table four a compendium of room temperature phonon frequencies for the isomorphic systems GeSe, SnS, GeS.

In view of the well-known hazards of using Kramers-Krönig analysis for heavily damped systems, such as SnS (Barker, 1964), we have not presented the dielectric constant values obtained by this technique in table one. The values for this quantity displayed in table two are, however, believed to be reliable in view of the fact that this procedure yields values of  $\epsilon_{\infty}$  in the  $\vec{E}11\vec{a}$  and  $\vec{E}11\vec{b}$  directions which are consistent with those obtained from an analysis of the interference fringe data. For example, with partially polarized light incident on the c face of a 2  $\mu\text{m}$  thick sample, interference fringes were observed in transmission above  $700\text{ cm}^{-1}$ . The spacing of these fringes was  $500\text{ cm}^{-1}$  at the highest frequencies

of incident light. The value of  $\epsilon_\infty = 16$  deduced from this observation is consistent with the values of  $\epsilon_\infty = 14$  and  $\epsilon_\infty = 19$  obtained by a fitting procedure from the  $\vec{E}11a$  and  $\vec{E}11b$  data respectively.

One of the original objectives of this work was to search for plasma reflection effects in our specimens in order to determine the carrier susceptibility mass. We see no evidence in our n-type samples of any "plasma edge" (Bradley, 1973) in the wavelength and temperature regions investigated. In the case of our p-type specimens, we are almost equally convinced that the same conclusions hold. Slight doubts, however, are cast by order of magnitude estimates of the plasma-edge position and by certain experimental artefacts that will be discussed later. This order of magnitude estimates will now be considered: if we take the early estimates of hole effective mass ratios for the symmetry directions of SnS (Haas and Corbay, 1961), we obtain

$$\left(\frac{m^*}{m_0}\right)_a \sim \left(\frac{m^*}{m_0}\right)_b \sim 0.2 \text{ and } \left(\frac{m^*}{m_0}\right)_c \sim 1$$

from free carrier absorption data. Combining these in the usual way we obtain a hole susceptibility mass of  $\sim 0.3m_0$ , which taken with an averaged value of  $\epsilon_\infty$  over the three symmetry directions of  $\langle\epsilon_\infty\rangle \sim 16$  yields a plasma frequency of  $\omega_p \sim 80 \text{ cm}^{-1}$  for typical samples with  $N_A - N_D \sim 5 \times 10^{17} \text{ cm}^{-3}$ . In conclusion, therefore, complex plasmon-phonon coupling effects should be seen for our p-type samples as this frequency

( $\omega_p$ ) is in the region of the phonon frequencies measured in our experiments. However, no clear indication is given in our data for such effects. In some of our spectra, it initially appeared that there was an increase in the non-resonant contribution to the reflectivity below  $200 \text{ cm}^{-1}$  for samples at  $10^\circ \text{K}$ . However, this observation is inconsistent with the normal movement of plasma edges with temperature in view of the fact that higher temperature samples do not show this response. Careful analysis of the experimental findings leads us to think that this effect is due to "icing" of the polyethylene windows over the course of a day's experimentation. Broadband absorption by the ice at frequencies less than  $\sim 150 \text{ cm}^{-1}$  would not then be "ratioed out", giving rise to these pseudo-plasma effects. Another critical problem encountered in the measurements of far infrared reflectivity and transmission of SnS is caused by movement with temperature of the polyethylene absorption features. It is well-known that the spectroscopist suffers from a scarcity of components in the far infrared, such as windows and lenses. Polyethylene is widely used as a window material but a problem arises due to its intrinsic absorption. The first feature is observed at  $71 \text{ cm}^{-1}$  (Davies and Haigh, 1974), (a sharp single phonon absorption) and another at  $96 \text{ cm}^{-1}$  (Krimm and Bank, 1965, Fleming et al. 1972) the latter being a weak absorption at room temperature shifting to  $106 \text{ cm}^{-1}$  at liquid nitrogen temperature (Davies and Haigh, 1974). The third feature of note is the

absorption maximum found at  $190 \text{ cm}^{-1}$  at room temperature, which on cooling to  $77 \text{ }^\circ\text{K}$  is observed to shift to  $215 \text{ cm}^{-1}$  for high density and  $220 \text{ cm}^{-1}$  for the low density material. All three absorptions of polyethylene coincide with the fundamental lattice vibrations of SnS, and it is therefore particularly important to correct the effect of polyethylene on SnS spectrum.

To conclude: six of the seven infrared active modes of SnS have been observed in the reflection in the region  $50$  to  $350 \text{ cm}^{-1}$ ; in general their positions agree with recent group theoretical analysis by Wiley et al. (1976).

Table 4

	GeS			GeSe			SnS		
	E II a	E II b	E II c	E II a	E II b	E II c	E II a	E II b	E II c
$W_{T1}$	117.5	201	105	88 88	150 155		97	192	
$W_{T1}'$	117.5	205	105	88	154		97	190	
$W_{T2}$	257.5		237	175 173			180		
$W_{T2}'$	257.5		237.5	173			183		200
$W_{T3}$			280	186 181			217		
$W_{T3}'$			280	182			220		229
$W_{L1}$	123.5	298	107	91.5 92	210.5 207		105	290	
$W_{L1}'$				91	208				
$W_{L2}$	325		275	178 178			215		
$W_{L2}'$				179					
$W_{L3}$			320	224 225			285		
$W_{L3}'$				224					
$E_{O0}$	25.1	29.5	30	21.9 24.5	30.4 23.5				
$E_{O0}'$	22.7	27.9	20.9				29	38	33
$E_{O00}$	14.8	12	10	18.7 13.6	21.9 13				
$E_{O00}'$	13.5	13	10				14	19	14

obtained in SnS, at room temperature only by Nikolic et al. (1976).  
 Lattice mode frequencies of GeS ( Wiley et al 1976 ),  
 GeSe ( Chandrasekhar and Zwick 1976, Siapkas et al 1976  
 the bottom reference ) and SnS.  
 (Lattice mode frequencies  $\omega_T, \omega_L$  in  $cm^{-1}$ )  
 following the analysis of Zallen and Sipe (1971, 1974).  
 The parameter  $\epsilon_{\infty}$  representing the strength of the



Foot Note: In any comparison of the strengths of the interlayer forces amongst the members of the SnS subset, it is of great importance to take account of any variations from stoichiometry. This is apparent from the fact that a slightly non stoichiometric material will not cleave as easily as one with perfect stoichiometry.

In this chapter we have compared the corresponding far infra-red lattice mode frequencies of the members of the SnS subset. We have simply assumed that our SnS samples were more layer-like than GeS and less layer-like than GeSe; the data for GeS and GeSe used in this comparison were obtained by Wiley et al(1976) and Chandrasekhar and Zwick(1976). Our arguments will only be true under certain assumptions; namely that the stretching forces between Sn-S and Ge-S are of the same order. However this is not strictly true. Furthermore in previous chapters we discussed the electrical and opto-electrical properties of our samples together with the effects on these properties of the sample quality. In this connection we do not know the sample quality and opto-electrical properties of the GeS and GeSe samples reported by the above authors.

After the preparation of this chapter it has learnt from the literature that Raman spectra had been obtained in SnS, at room temperature only by Nikolic et al(1977). These authors compared their Raman frequencies with the present infra-red data(Chamberlain et al 1976) to obtain a value for the interlayer coupling factor following the analysis of Zallen and Slade(1971,1974). The parameter  $(\nu/\Delta)^2$  representing the strength of the

interlayer coupling is found from this work to be of the order 6. This value indicates that SnS is less layer-like compared with, for example, GeS. This statement is in conflict with our conclusions. We now suggests two possible explanations for this discrepancy. Firstly, some of the crystal growing processes were carried out at Belgrade University(as reported in the acknowledgement section) but there is no evidence that our sample and the Raman sample(Nikolic et al 1977) have the same physical properties. These workers have taken our infra-red data for comparison purposes, and it is almost certain that their sample were not cut from the same ingot as these used by us. Secondly Nikolic et al have made no attempts to calculate the bond stretching force in SnS, which is, of course, given by the expression

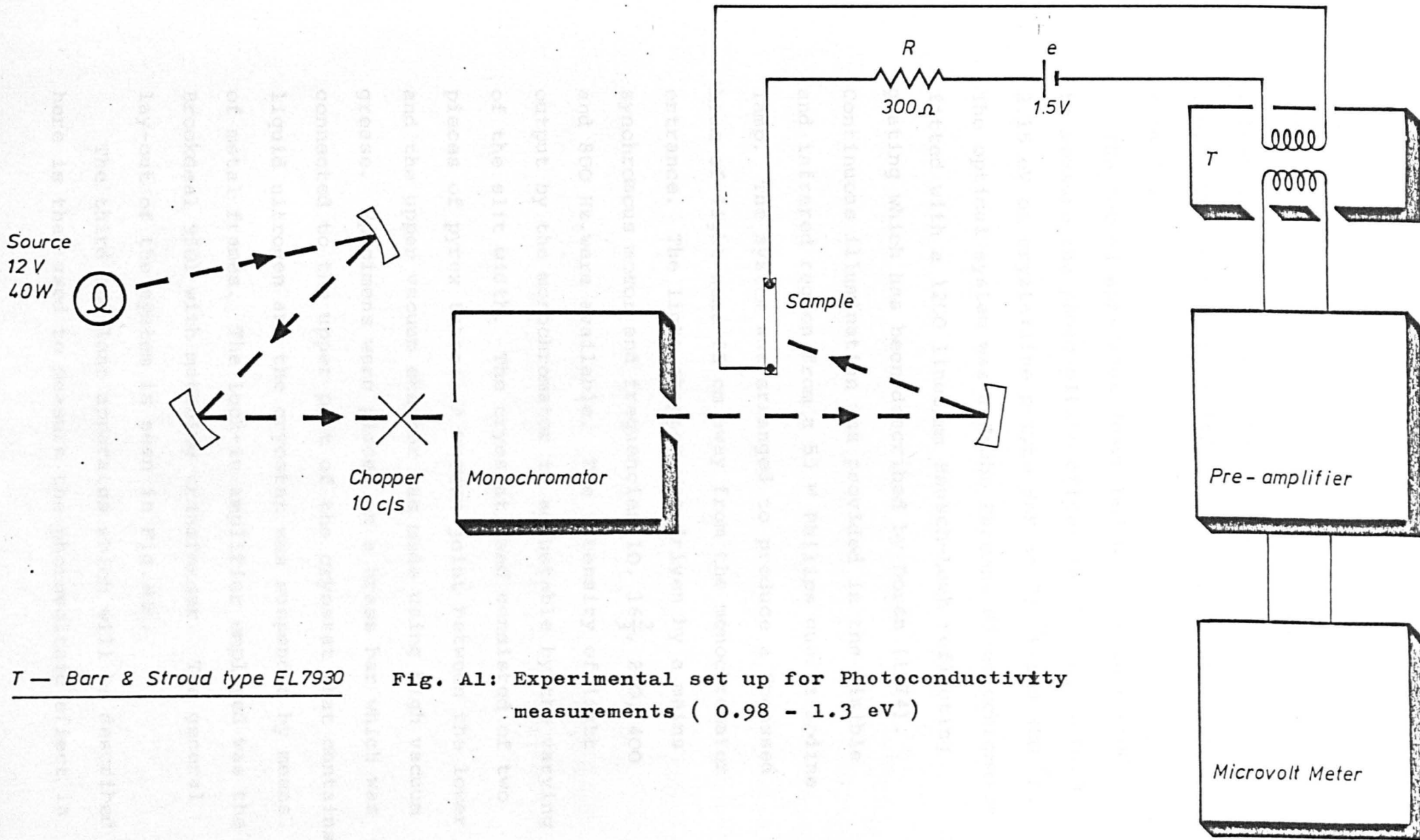
$$k_r(\text{Sn,S}) = [k_r(\text{Sn,Sn}) k_r(\text{S,S})]^{\frac{1}{2}} (X_{\text{Sn}} - X_{\text{S}})^2$$

[ Lucovsky et al 1974]. Where  $X_{\text{Sn}}$  and  $X_{\text{S}}$  are the electronegativity of Sn and S respectively. It is apparent that using the standard data and the above expression the bond stretching forces in SnS and GeS are fairly different.

## APPENDIX A

### DESCRIPTION OF APPARATUS USED FOR PHOTOCONDUCTIVITY MEASUREMENTS.

The first apparatus to be described here is that used to measure the photoconductivity and the photo-voltaic effect in the region (0.99-1.5) eV on crystalline n-type SnS at 300 °K. A general layout of the apparatus used is shown in Fig. A.1. A Carlzeiss SPM2 prism spectrometer with a "Unipan" preamplifier and a selective microvoltmeter type 203 were the main part of the set-up, which has been described in detail by Nikolić (1969). The source was a 12V, 40W, single filament car spot light bulb acting as a broad band source in the visible and infrared regions. The incident radiation was mechanically chopped at a frequency of 10 Hz, and focussed into the entrance slit of the monochromator by means of combination of aluminised mirrors. The intensity of the radiation from the monochromator was adjusted with an adjustable exit slit and focussed onto the sample using an aluminised mirror as seen in Fig. A.1. Bias to the specimen was supplied from a 1.5V battery and no attempt made to stabilise the current further. An a.c. signal created on the sample by illuminating it with single wavelength radiation was fed with a quadruply screened



T — Barr & Stroud type EL7930

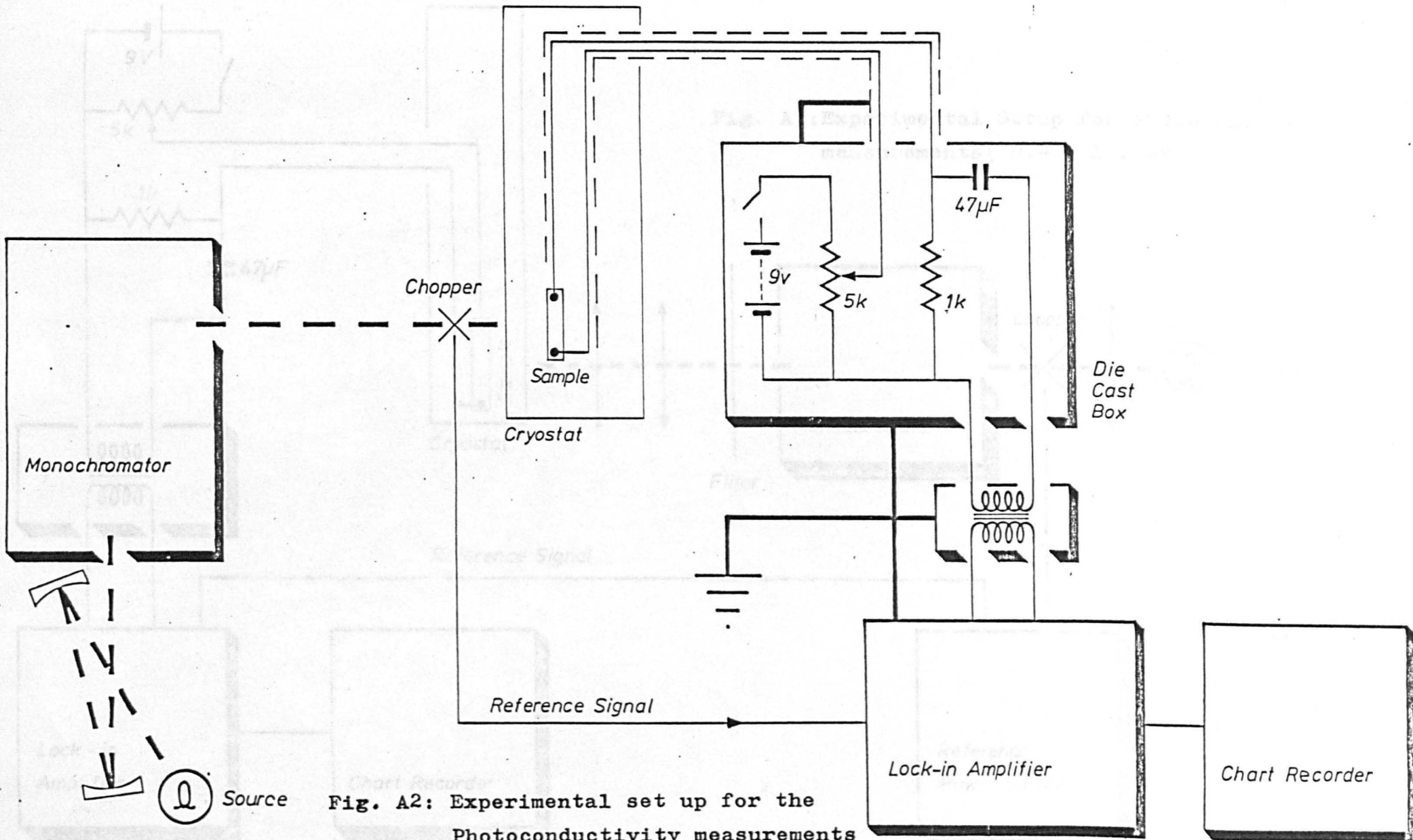
**Fig. A1: Experimental set up for Photoconductivity measurements ( 0.98 - 1.3 eV )**

matching transformer. (Barr and Stroud, 60  $\Omega$  type 7930) to a "Unipan" preamplifier and selective micro-voltmeter with a gain factor of  $10^9$  and band width of 5 Hz. The output of amplifier was read directly on a meter.

The second apparatus described here is that used to measure the photovoltaic effect in the region 0.99-2.15 eV on crystalline p-type SnS at 77  $^{\circ}\text{K}$  and 300  $^{\circ}\text{K}$ . The optical system was a Grubb Parsons M2 monochromator fitted with a 1200 lines/mm Bausch-Lomb reflecting grating which has been described by Doran (1974). Continuous illumination was provided in the visible and infrared region from a 55 W Philips quartz-iodine lamp. The system was arranged to produce a focussed beam of light some 25 cm away from the monochromator entrance. The light chopper was driven by a mains synchronous motor and frequencies 10,  $16\frac{2}{3}$ , 200, 400 and 800 Hz. were available. The intensity of light output by the monochromator is adjustable by the varying of the slit width. The cryostat used consisted of two pieces of pyrex tubes. A vacuum joint between the lower and the upper vacuum chamber was made using high vacuum grease. Specimens were placed on a brass bar which was connected to the upper part of the cryostat that contains liquid nitrogen and the cryostat was suspended by means of metal frames. The lock-in amplifier employed was the Brookdeal 9501 with matching transformer. The general lay-out of the system is seen in Fig. A2.

The third and last apparatus which will be described here is that used to measure the photovoltaic effect in

the region 0.48-1.3 eV on crystalline p-type SnS at lower temperatures (40-77) °K. The monochromator employed was a Metrospec high radiance monochromator fitted with a 2000 nm blaze grating with 295 lines/mm. Both entrance and exit slit widths were adjustable. A quartz halogen lamp was used as a source. The filament current and voltage were controlled continuously. The light was focussed on to the entrance of monochromator and on to the sample by means of movable quartz lenses. The wavelength drum on the monochromator was driven automatically by a motor which is engaged to a clutch, and could be driven at different speeds. The spectrum was obtained with and without a Kodak wratten 70 filter (Doran 1974, and manufacturer) for the infrared region. This has a low-pass response with cut off at 1.8 eV as well as high transmission in the pass-band. Electrical connections from the sample inside the cryostat were made by covered 36 SWG copper leads. It was found that to avoid standing noise it was necessary to carefully screen all the electrical leads.



**Fig. A2: Experimental set up for the Photoconductivity measurements (0.98- 2.1 eV)**

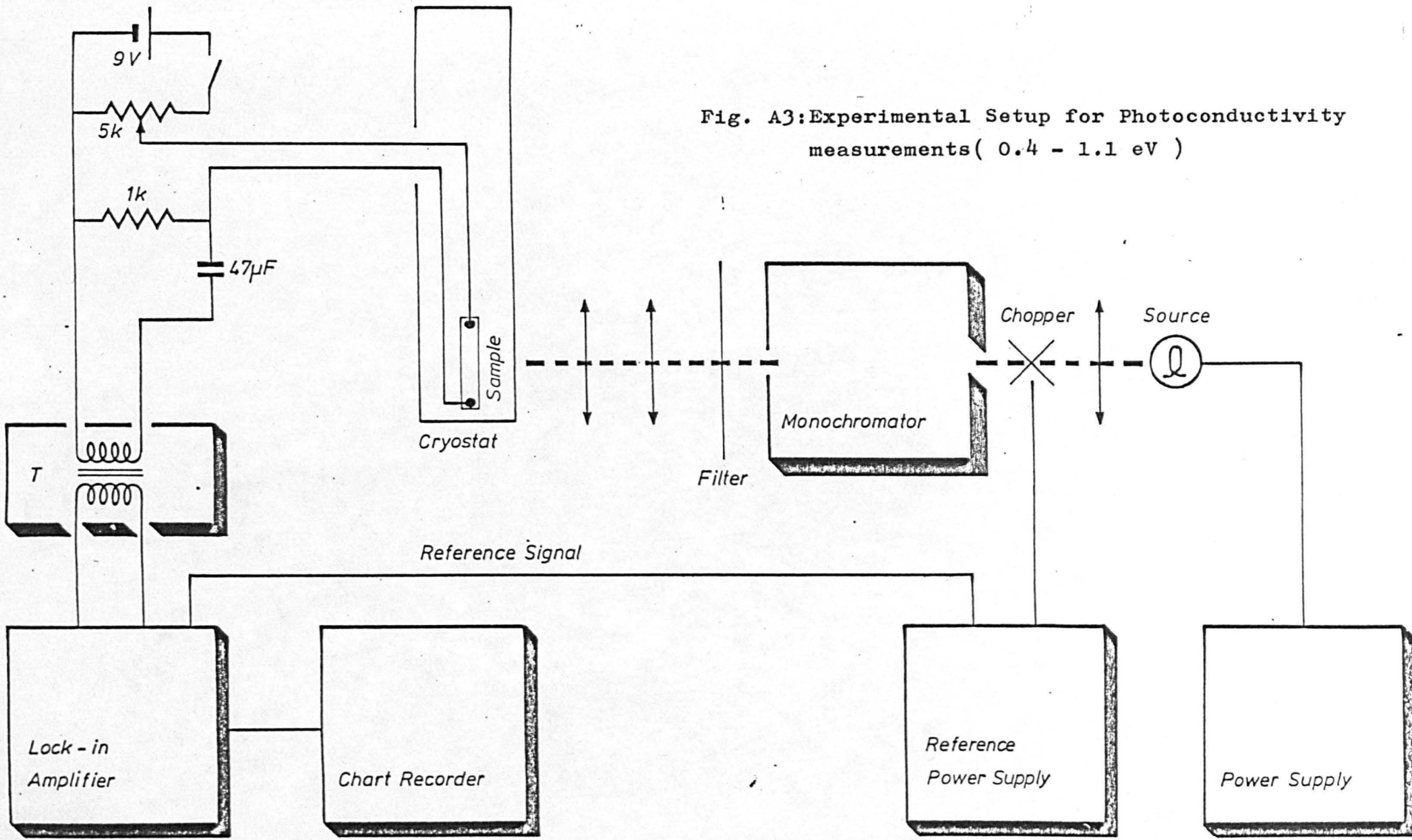


Fig. A3: Experimental Setup for Photoconductivity measurements ( 0.4 - 1.1 eV )



## APPENDIX B

### OSCILLATOR FITTING PROGRAM

#### Introduction

The reflection at normal incidence is related to the real and imaginary parts of the dielectric response function  $\epsilon(\omega)$  through the usual equation

$$R = \frac{[(n-1)^2 + k^2]}{[(n+1)^2 + k^2]} \quad [1]$$

where  $n$  and  $k$  are the real and imaginary parts of the refractive index, i.e.  $(n+ik)^2 = \epsilon(\omega) = \epsilon^1(\omega) + i\epsilon^{11}(\omega)$ .

We use the dynamic argument (Balkanski/Abeles, p.546) to obtain

$$\epsilon^1(\omega) = n^2 - k^2 = \epsilon_\infty + \frac{(\epsilon_0 - \epsilon_\infty) \omega_{TO}^2 \omega_{TO}^2}{(\omega_{TO}^2 - \omega^2)^2 + \Gamma_a^2 \omega^2} \quad [2]$$

and

$$\epsilon^{11}(\omega) = 2nk = \frac{(\epsilon_0 - \epsilon_\infty) \omega_{TO}^2 \omega \Gamma_a}{(\omega_{TO}^2 - \omega^2)^2 + \Gamma_a^2 \omega^2} \quad [3]$$

where  $\omega_{TO}$  is the transverse optic phonon frequency.

$\epsilon_0$  is the "low frequency" dielectric constant;

$\epsilon_\infty$  is the "high frequency" dielectric constant,

and  $\Gamma_a$  refers to the damping of the mode. We extend

the analysis to several lattice modes on the assumption

that these modes are independent and therefore we can

write for n oscillators

$$\epsilon^l(\omega) = \epsilon_\infty + \sum_{j=1}^n \frac{(\epsilon_0 - \epsilon_\infty) (\omega_j^2 - \omega^2) \omega_j^2}{(\omega_j^2 - \omega^2)^2 + \Gamma_j^2 \omega^2} \quad [4]$$

$$\epsilon^{ll}(\omega) = \sum_{j=1}^n \frac{(\epsilon_0 - \epsilon_\infty) \omega_j^2 \omega \Gamma_j}{(\omega_j^2 - \omega^2)^2 + \Gamma_j^2 \omega^2} \quad [5]$$

where the  $\omega_j$ 's and  $\Gamma_j$ 's refer to each mode.

#### Computation Procedure

The experimental reflectivity data points (R versus  $\sigma$ ) are fitted to one, two or three lattice oscillators (as seems appropriate) using a least-squares computer fit. The routine used is Peckhams method which is implemented on the Cripps Computer Centre routine FO4FAF. The program used by us is rather cumbersome and can, no doubt, be streamlined as far as input and output is concerned. A compilation for fitting 150 data points to three oscillators is appended. The first segment inputs the data points and notes the "initial guesses" ( $\theta_1$  to  $\theta_{10}$ ) of the variables to be fitting and the accuracy parameters ( $V_1$  to  $V_{10}$ ) required by the NAG routine. In the second segment the full expression is given which is to be fitted to the data points provided. This is equation [1] with n and k subtracted from equation [2] and [3] (or in the case of several oscillators

from equation  $\langle 4 \rangle$  and  $\langle 5 \rangle$ ). The smooth working of the routine FO4FAF requires that the physical variables  $\epsilon_\infty$ ,  $(\epsilon_0 - \epsilon_\infty)_j$ ,  $\omega_j$  and  $\Gamma_j$  be of the same numerical order. The computer variables  $\theta(1)$ ,  $\theta(2)$  etc. used in our program are therefore chosen in accordance with this requirement using the scheme:

$$\begin{aligned} \theta_1 &= \epsilon_\infty \\ \theta_2 &= (\epsilon_0 - \epsilon_\infty) \\ \theta_3 &= \omega_1 / 10 \\ \theta_4 &= \frac{\Gamma_1}{\omega_1} \times 10^3 \quad \text{(Contribution from} \\ & \quad \text{first oscillator. )} \end{aligned}$$

and same  $\theta_5$ ,  $\theta_6$ ,  $\theta_7$  and  $\theta_8$ ,  $\theta_9$ ,  $\theta_{10}$  for second and third oscillators.

The third segment calls up the NAG routine FO4FAF which accomplishes the fitting. In this routine we give details of the output format. The final output lists the values of the computer parameters  $\theta_1$ ,  $\theta_2$ ,  $\theta_3$  etc. from which the parameters of the oscillators ( $\epsilon_\infty$ ,  $\epsilon_0$ ,  $\omega_1$ ,  $\Gamma_a$  etc.) may be deduced.

Reference - see CCC NAG library manual on routine FO4FAF.

FORTRAN COMPILATION BY XFAT MK 5A DATE 09/08/76 TIME 14/22/46

```
0001 LIST
0002 LIBRARY(ED,L(1))
0003 LIBRARY(ED,L(2))
0004 MAP
0005 PROGRAM(GENF)
0006 EXTENDED
0007 INPUT5,20=CRO
0008 OUTPUT6,30=LPO
0009 TRACE2
0010 END
0011 MASTER FIDDLE
0012 DIMENSION R(150),Y(150),THETA(10),W(20,000),V(10)
0013 DIMENSION Z(150)
0014 COMMON Y,Z
0015 EXTERNAL FUNCT,MONIT
0016 N=10
0017 M=141
0018 IP=N+3+N/3
0019 IW=2*M+4*N+M*N+(N**2+N)/2+IP*(M+2*N+2)
0020 I FAIL=1
0021 THETA(1)=10
0022 THETA(2)=5
0023 THETA(3)=9
0024 THETA(4)=50
0025 THETA(5)=4
0026 THETA(6)=19
0027 THETA(7)=125
0028 THETA(8)=4
0029 THETA(9)=22
0030 THETA(10)=140
0031 V(1)=5
0032 V(2)=2
0033 V(3)=2
0034 V(4)=15
0035 V(5)=2
0036 V(6)=9
0037 V(7)=20
0038 V(8)=2
0039 V(9)=2
```

```

0040      V(10)=20
0041      ALF=1.0E-5
0042      EPS=1.0E-6
0043      I PRINT=1
0044      MAXIT=1000
0045      READ(5,200) ZN,STEP
0046 200  FORMAT(F6,3,F5,3)
0047      DO 250 I=1,141
0048 250  Z(I)=(ZN+(I-1)*STEP)/10
0049      READ(5,500) (Y(I),I=1,141
0050 500  FORMAT(141F0.0)
0051      WRITE(6,900)
0052 900  FORMAT(1H1)
0053      CALL EO4FAF(M,N,THETA,R,S,EPS,ALF,V,W,IW,FUNCT,MONIT,
0054      1 I PRINT,MAXIT,I FAIL)
0055 300  FORMAT(44HO FINAL LEAST SQUARES ESTIMATES OF THETA ARE,7E1
0056      WRITE(6,400) I FAIL
0057 400  FORMAT(9H I FAIL=,I1)
0058      WRITE(6,800) EPS
0059 800  FORMAT(7H EPS=,F4.0)
0060      STOP
0061      END

```

END OF SEGMENT, LENGTH 334, NAME FIDDLE

```

0062      SUBROUTINE FUNCT(M,N,THETA,R)
0063      DIMENSION THETA(N),R(M),Y(150),Z(150)
0064      DIMENSION ALPHA(150),BETA(150)
0065      DIMENSION COMPA(150,10),COMPB(150,10)
0066      DIMENSION DENOM(150,10)
0067      COMMON Y,Z
0068      DO 10 I=1,140
0069      DO 20 J=2,5,8
0070      DENOM(I,J)=(1-(Z(I)/THETA(J+1))**2)**2
0071      1 +(1.0E-6)*(THETA(J+2)**2)*((Z(I)/THETA(J+1))**2)
0072      COMPA(I,J)=THETA(J)*(1-Z(I)/THETA(J+1))**2/DENOM(I,J)
0073      COMPB(I,J)=(1.0E-3)*Z(I)*THETA(J)/(THETA(J+2)*DENOM(I,J))
0074      ALPHA(I)=THETA(1)*COMPA(I,2)+COMPA(I,5)+COMPA(I,8)
0075      BETA(I)=COMPB(I,2)+COMPB(I,5)+COMPB(I,8)
0076 20  CONTINUE
0077 10  R(I)=(SQRT(ALPHA(I)**2+BETA(I)**2)+1-1.414214*SQRT(SQRT
0078      1 (ALPHA(I)**2+BETA(I)**2)+ALPHA(I)))/

```

```

0079      2(SQRT(ALPHA(I)**2+BETA(I)**2)+1+1.414214*SQRT(SQRT(ALPHA(I)
0080      3**2+BETA(I)**2)+ALPHA(I)))-Y(I)
0081      RETURN
0082      END

```

END OF SEGMENT, LENGTH 358, NAME FUNCT

```

0083      SUBROUTINE MONIT(M,N,THETA,S,I TERC,SING,LIM)
0084      LOGICAL SING, LIM
0085      DIMENSION THETA(N)
0086      WRITE(6,100) I TERC
0087      WRITE(6,200) S
0088      WRITE(6,300) (THETA(I), I=1,N)
0089      IF(SING) WRITE(6,400)
0090      IF(LIM) WRITE(6,500)
0091      RETURN
0092      100  FORMAT(11H0 ITERATION ,I2)
0093      200  FORMAT(16H SUM OF SQUARES ,E14.6)
0094      300  FORMAT(21H VALUES OF THETA ARE ,3F14.6)
0095      400  FORMAT(9H SINGULAR)
0096      500  FORMAT(8H LIMITED)
0097      END

```

END OF SEGMENT , LENGTH 111, NAME MONIT

Marmatite is formed containing Fe partially substituting for Zn. The iron concentration is always less than 26 per cent, and no 100% Fe member of the sphalerite group is known [Palache et al., 1966, pp. 210].

The colour of Marmatite varies with the Fe content from white-brown (in which the ratio of Zn:Fe is 10:1) to brown-black in which Fe content is more than 10 per cent. The range in the latter case is Zn:(Fe) 5:1 with a maximum of 26 per cent Fe.

Marmatite is brittle and has a perfect cleavage plane developed parallel to the dodecahedron  $d(110)$ ; with care, a large fragment can be cleaved in this form [Mikhankou, 1964, pp. 294].

The pale coloured specimens are transparent through 2

## APPENDIX C

### OPTICAL STUDIES ON MARMATITE

#### Introduction

Marmatite (Zn,Fe)S crystallizes predominately in the tetrahedral structure (Fig. C1) which is often developed into forms resembling octahedra. It seldom appears with sharp or smooth faces, but is usually curved, twinned, striated, etc. (Fig. C2) [p.294, Sinkankas 1964, Mineralogy]. Simple and repeated twinning is very common on 0(111) as manifested by narrow striations and bands of slightly differing lustre on crystal surfaces and cleavage planes.

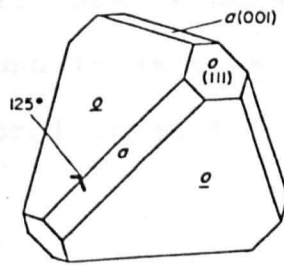
Marmatite is a typical member of the sphalerite (ZnS) group [Boussingault, 1829]. The space group is  $F\bar{4}3m$  and the unit cell contains 8 atoms [Palache et al, 1966].

Marmatite is formed containing Fe partially substituting for Zn. The iron concentration is always less than 26 per cent, and no 100% Fe member of the sphalerite group is known [Palache et al, 1966, pp.210].

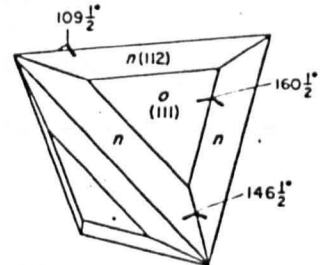
The colour of Marmatite varies with the Fe content from white-brown (in which the ratio of Zn:Fe  $> 5:1$ ) to brown-black in which Fe content is more than 10 per cent. The range in the latter case is  $6:5 > \text{Zn:Fe} > 5:1$  with a maximum of 26 per cent Fe.

Marmatite is brittle and has a perfect cleavage plane developed parallel to the dodecahedron  $d(110)$ ; with care, a large fragment can be cleaved in this form [Sinkankas, 1964, pp.294]

The pale coloured specimens are transparent through 2"

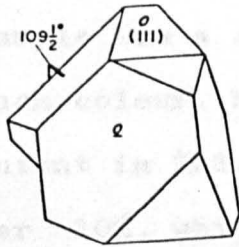


POSITIVE & NEGATIVE OCTAHEDRONS & CUBE

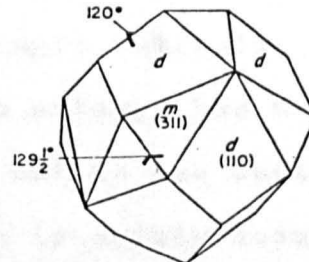


POSITIVE OCTAHEDRON & TRISTETRAHEDRON  $n/112/$

SPHALERITE



SPINEL-TYPE TWIN



DODECAHEDRON & TRISTETRAHEDRON  $m/311/$

Fig C1: Crystal Structure of Sphalerite.

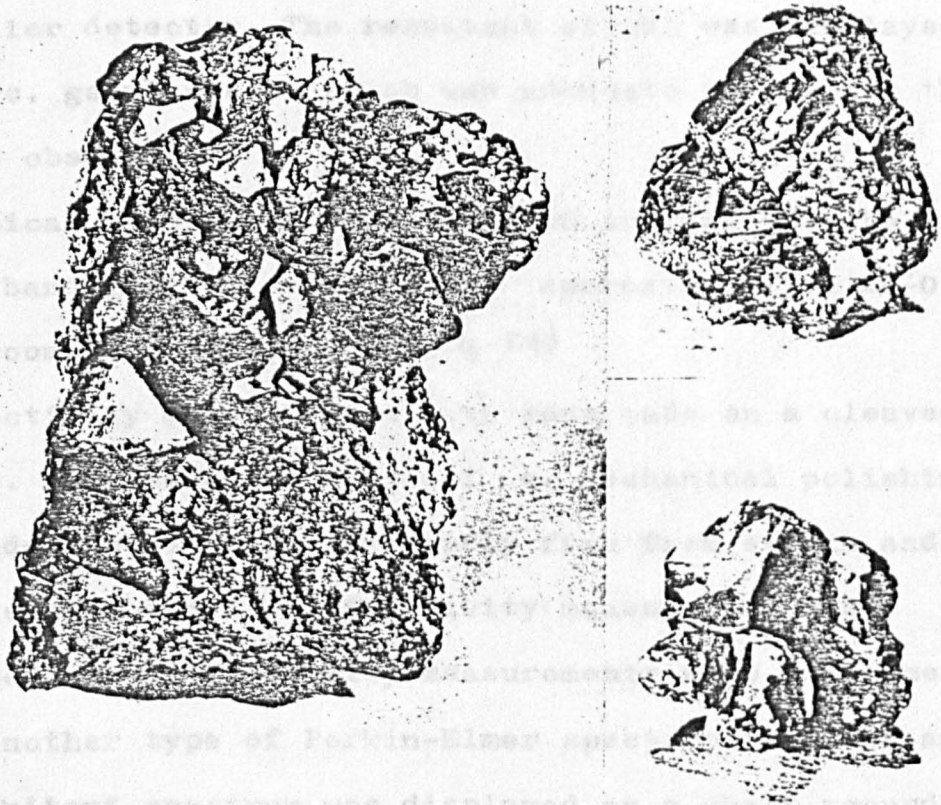


FIG. C2: Sphalerite. Left: Rude twinned black crystals on chert from southwest Missouri. Size  $5\frac{1}{2}$ " by  $3\frac{1}{2}$ ". Top right: Transparent greenish-yellow curved and twinned crystals from Picos de Europa, Santander, Spain. Size  $1\frac{3}{4}$ " by  $1\frac{3}{4}$ ". Bottom right: pale green cluster of transparent crystals from Franklin, Sussex County, New Jersey. Size  $1\frac{1}{2}$ " by  $1\frac{3}{8}$ ".



of material, while dark coloured specimens are opaque.

It has been found in various colours with various Fe content as a natural crystal in different parts of the world.

#### Experimental Technique and Results.

The marmatite sample was a single naturally occurring crystal with a black colour. No attempt has been made to analyse the Fe content in ZnS, but it was certain that the Fe content was over 10%, which is safely assumed from its colour as discussed previously.

The sample for the transmission experiment was of 400 $\mu$  thickness stuck onto sellotape. The spectra were obtained using a Perkin-Elmer type spectrophotometer with a photomultiplier detector. The resultant signal was displayed on a d.c. galvanometer which was adequate to resolve the feature observed.

A typical spectra obtained is shown in fig. C3. The indirect band gap is assumed to be approximately  $1.28 \pm 0.05$  eV at room temperature. (See, fig. C5)

Reflectivity measurements have been made on a cleaved surface. No attempt at chemical or mechanical polishing were made. The surface was fairly free from strain and striatae on which the reflectivity measurement was performed. The reflectivity measurements were performed using another type of Perkin-Elmer spectrophotometer and the resultant spectrum was displayed on a chart recorder. The spectrum was corrected for the apparatus characteristic by replacing the sample with a freshly aluminized mirror. A typical reflectivity curve is displayed in fig. C3.

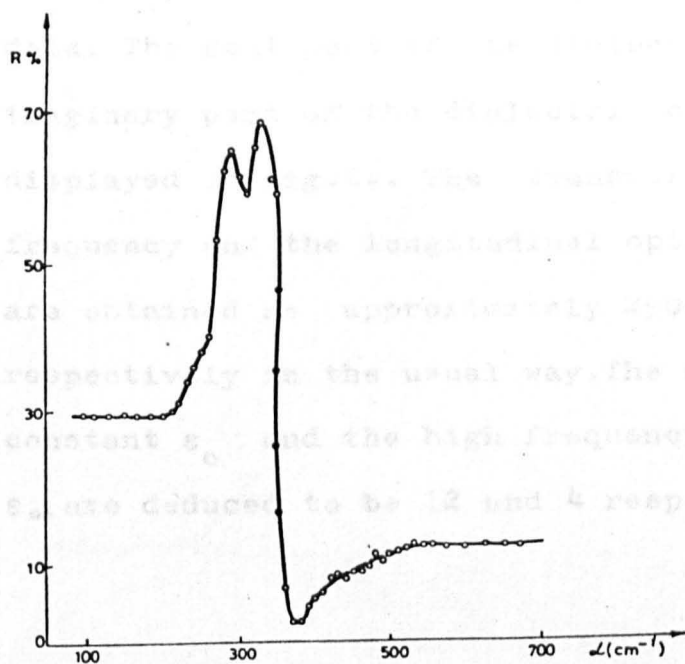


Fig. C3: Reflectivity Curve of Marmatite

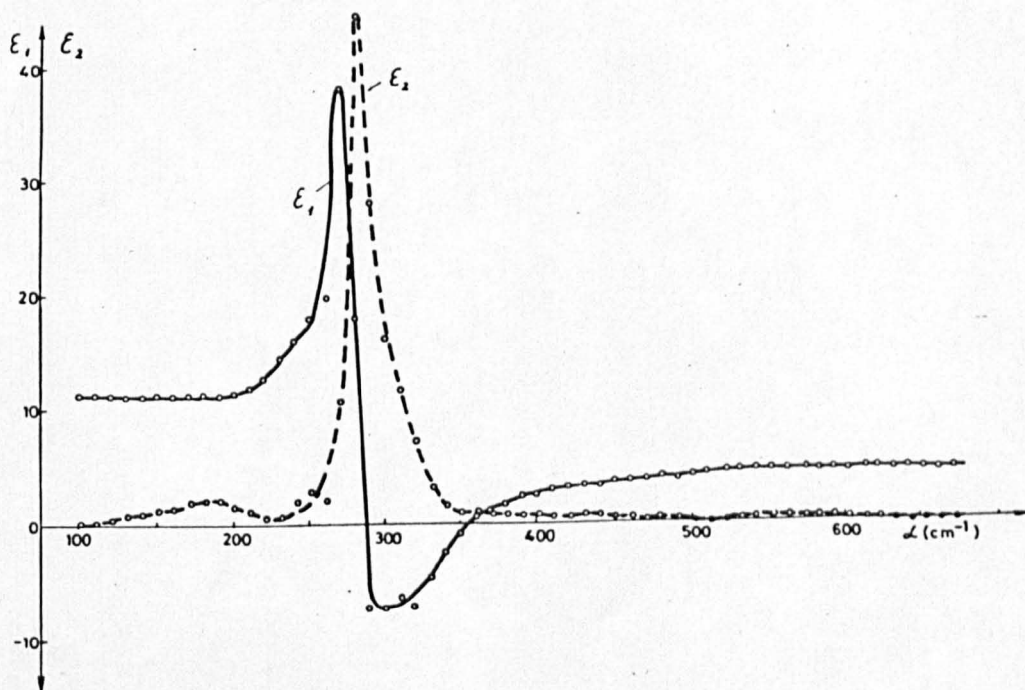


Fig. C4: Kramers-Kronig analysis of the reflectivity spectra.

Optical constants were obtained by performing Kramers-Kronig analysis and oscillator fits on the reflectivity data. The real part of the dielectric constant  $\epsilon'$  and the imaginary part of the dielectric constant  $\epsilon''$  spectrum are displayed in fig.C4. The transverse optical phonon frequency and the longitudinal optical phonon frequencies are obtained as approximately  $290\text{cm}^{-1}$  and  $358\text{cm}^{-1}$  respectively in the usual way. The static dielectric constant  $\epsilon_0$  and the high frequency dielectric constant  $\epsilon_\infty$  are deduced to be 12 and 4 respectively.

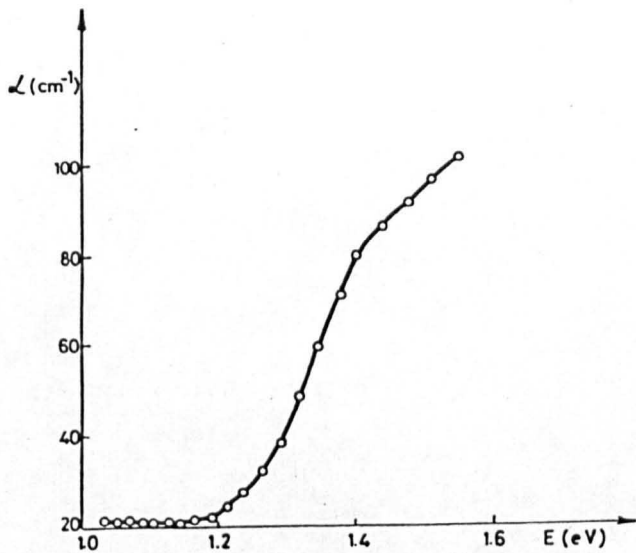


Fig. C5: Absorption of Marmatite

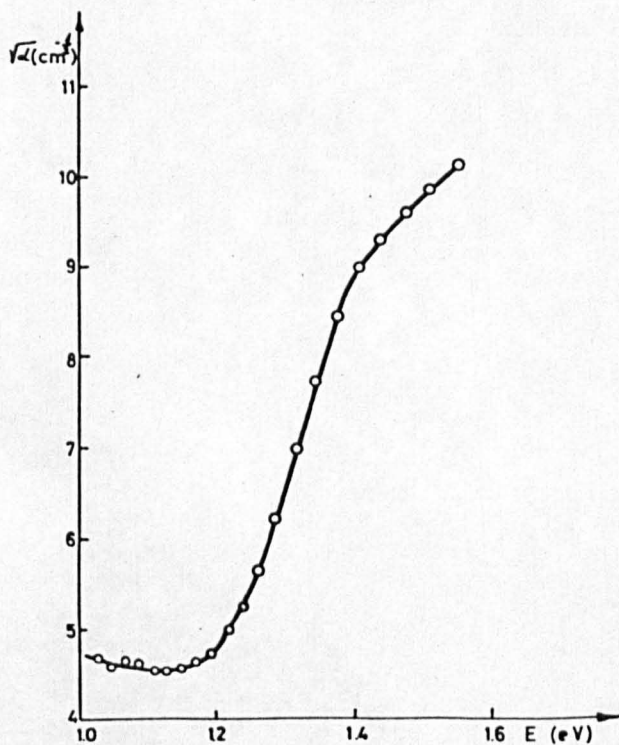


Fig. C6:  $\sqrt{\alpha}$  versus Energy variations of Marmatite

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