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1 **Graphene-based supercapacitor with carbon nanotube film as**
2 **a highly efficient current collector**

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10

11 **Abstract**

12 Flexible graphene-based thin film supercapacitors were made using carbon nanotubes
13 (CNTs) films as current collectors and graphene films as electrodes. The graphene sheets
14 were produced by simple electrochemical exfoliation, while the graphene films with
15 controlled thickness were prepared by vacuum filtration. The solid-state supercapacitor
16 was made by using two graphene/CNT films on plastic substrates to sandwich a thin layer
17 of gelled electrolyte. We found that the thin graphene film with thickness $<1 \mu\text{m}$ can
18 greatly increase the capacitance. Using only CNT films as electrodes, the device
19 exhibited a capacitance as low as $\sim 0.4 \text{ mF/cm}^2$, whereas by adding a 360-nm-thick
20 graphene film to the CNT electrodes led to a $\sim 4.5 \text{ mF/cm}^2$ capacitance. We
21 experimentally demonstrated that the conductive CNT film is equivalent to gold as

22 current collector while it provides a stronger binding force to the graphene film.
23 Combining the high capacitance of the thin graphene film and the high conductivity of
24 the CNT film, our devices exhibited high energy density (8-14 Wh/kg) and power density
25 (250-450 kW/kg).

26 **Keywords**

27 Supercapacitor; Graphene; Electrochemical exfoliation; Carbon nanotube film; Current
28 collector.

29 **1. Introduction**

30 Electrical energy storage plays an important role in the high tech industry as the energy
31 components are indispensable to portable devices and hybrid vehicles [1, 2].
32 Supercapacitors have attracted much interest in the last decade and their technology
33 advanced with the use of carbon nanomaterials. The unique features of supercapacitors,
34 such as high power density, ability to charge and discharge within seconds, long lifetime,
35 and no maintenance, make them superior to traditional batteries in practical applications
36 [3, 4]. At the current stage, the energy density of a commercial supercapacitor is still
37 much lower compared to that of a standard battery, however the combination of a
38 supercapacitor with a battery has been the design of choice to power electrical vehicles,
39 exploiting the short duration power boost of a supercapacitor for vehicle acceleration.
40 Today, most supercapacitor manufacturers use coconut shell activated carbon as their
41 active material. Commercial supercapacitors are bulky and require high standard
42 encapsulation to avoid the leakage of liquid electrolyte. The development of gel polymer
43 electrolytes allows for the fabrication of solid-state supercapacitors, simplifying the

44 fabrication process and making this technology very appealing for several applications [5,
45 6]. The use of polymer electrolyte has led to paper-like supercapacitors that can be bent
46 and twisted [7]. Other authors used titania nanotubes as substrate and loaded NiO on top
47 of the nanotube network by anodizing a Ni-Ti binary alloy foil, reporting a high value of
48 areal capacitance ($1-10 \text{ F/cm}^2$) [8]. They suggest that the faster charge/discharge kinetics
49 of NT electrodes at high scan rates is due to the ordered NT film architecture.
50 Presumably, the direct conducting pathways for electrons in the NT walls and for ions
51 through the pore - system facilitate the
52 discharge reactions.

53 Carbon nanomaterials like carbon nanotubes (CNTs) and graphene have been proposed
54 as electrode materials in supercapacitors because of their extraordinary large specific
55 surface area, remarkable chemical stability, high electric conductivity, thermal and
56 mechanical properties [9-13]. They are in fact valid candidates to replace activated
57 carbons that present a low specific capacitance due to a great portion of micropores,
58 leading to low electrode-electrolyte accessibility [14]. The main advantage of CNT and
59 graphene is that they can provide higher conductivity than other carbonaceous materials,
60 leading to a more simple and lightweight device. The combination of gelled electrolyte
61 and thin films of carbon nanomaterials opens opportunities for developing flexible
62 supercapacitors that can be combined with bendable displays and touch-pads.

63 Supercapacitor electrodes based on CNTs have been made by using either CNT
64 arrays grown on a substrate or network films processed from a CNT suspension [12-15].
65 For individual CNTs only the outer surface can be used to absorb ions, whereas for a
66 graphene sheet both sides can be exploited to trap ions, making graphene more suitable

67 for making high-capacitance electrodes. Xu *et al.* prepared a 3D porous structure of
68 reduced graphene oxide (rGO) for electrodes by pressing such thick films onto an Au-
69 coated plastic substrate to make supercapacitors [16]. However, the high cost of using Au
70 as the current collector has to be considered if devices are produced on an industrial
71 scale. El-Kady *et al.* developed a novel technique to thermally reduce GO using the laser
72 of a DVD optical drive and assembling laser-scribed graphene films with gelled
73 electrolyte to make supercapacitors [17, 18]. The conductivity of their graphene film is
74 1738 S/m, equivalent to 75 Ω/sq (the thickness of laser-irradiated graphene film is 7.6
75 μm), leading to an increase of the resistance of the device if used as both the electrode
76 and current collector. High internal resistance not only decreases the power density of the
77 device, but also leads to heat problems when charged/discharged at large currents. Hence,
78 a current collector with high conductivity and the ability to bind with the graphene
79 electrode can be helpful in improving the capacitive performance and facilitating the
80 device fabrication. Metal foils such as Al, Cu, and stainless steel with thicknesses
81 between 20 μm and 80 μm are largely used as current collectors in commercial
82 supercapacitors. The metal foil suffers from corrosion due to the use of acidic electrolyte,
83 thus requiring special technique to passivate the metal surface [19]. Zhou *et al* used
84 cross-stacking aligned CNT film as current collector to support metal oxide nanoparticles
85 and made supercapacitors using liquid electrolyte [20]. In their work the sheet resistance
86 of CNT film is 50 Ω/sq .

87 In this paper we report the use of double-wall CNT (DWCNT) films with very low
88 sheet resistance as current collectors and graphene films as electrodes to make flexible
89 and solid-state supercapacitors. The function of the CNT film is investigated by

90 comparing it with a gold film used to support the graphene electrode. We studied the
91 dependence of the capacitance on the graphene film thickness, ranging from 50 nm to
92 360 nm.

93 **2. Experiment**

94 **2.1 Preparation of CNT Films**

95 DWCNTs with an average length of 10 μm and average diameter of 2.4 nm were
96 purchased from Continental Carbon Nanotechnologies, Inc. (batch X647H) and mixed
97 with chlorosulfonic acid at 2000 rpm with a speed-mixer (DAC 150.1 FV-K, Flak Tek
98 Inc) for 20 min. In order to make CNTs films, the solution was diluted to 20 ppm and
99 then vacuum filtered by using alumina membranes (Whatman Co., pore size 200 nm).
100 The CNT film on the membrane was washed by chloroform to remove most of the acid
101 (film coagulation) and then dried in air at room temperature overnight [21]. The alumina
102 membrane was etched away by floating it onto an aqueous solution of NaOH (0.5 M) in a
103 petri dish. After draining off the NaOH solution and refilling with DI water, the CNT
104 film was transferred onto a polyethylene naphthalate (PEN) paper and then dried at 50 $^{\circ}\text{C}$
105 for 5 hours. The CNT film was characterized by Raman spectroscopy and its thickness
106 was optically determined to be 520 nm (Supporting Information), weighting 29 $\mu\text{g}/\text{cm}^2$.
107 The sheet resistance of the CNTs film measured by a four-point probe meter was 5 Ω/sq .

108 **2.2. Preparation of Graphene Films**

109 Graphene was produced by electrochemically exfoliating highly oriented pyrolytic
110 graphite ($1 \times 1 \text{ cm}^2$) in a 150 mL aqueous solution containing 0.15 M Na_2SO_4 and 0.01
111 M sodium dodecyl sulfate. A Pt wire was used as a cathode and the voltage was set to be

112 5-6 V. The setup was placed in a sonicator during the electrochemical exfoliation
113 process, which results in a final product consisting of mostly bilayer graphene. By using
114 atomic force microscopy to scan a number of graphene sheets randomly dispersed onto a
115 Mica substrate, statistics of the graphene flake thickness was obtained. Details of the
116 experiment and characterization of graphene samples can be found in our previous work
117 [22, 23].

118 The graphene sheets dispersed in dimethylformamide (DMF) were vacuum filtered
119 onto a porous alumina membrane (Whatman Co., pore size 100 nm) in order to form a
120 uniform film. The film thickness was controlled by measuring the volume of suspension
121 for filtration. After drying, the membrane covered with the graphene film was etched
122 away with the same process described above for the CNT film. The graphene film was
123 then transferred onto the PEN-supported CNT film and dried at 50 °C in air. For
124 comparison, a graphene film transferred onto Au-coated PEN film was also used to make
125 a device.

126 **2.3. Fabrication of solid-state supercapacitors**

127 Methods for preparing the gel electrolyte are reported by Kaempgen *et al.* [24]. In our
128 case, 1 g of poly(vinyl alcohol) (PVA) was dissolved into 10 mL DI water at 90 °C by
129 continuous stirring. After cooling to room temperature, 0.8 g of H₃PO₄ (85% solution in
130 water) was mixed with the gel by stirring. The plastic substrate coated with carbon
131 nanomaterials was cut into rectangular pieces to make devices. The electrolyte gel was
132 applied on the two electrodes and dried at room temperature for 5 h before stacking the
133 two substrates together face-to-face. The device was then left at room temperature for
134 other 10 h to assure the solidification of the gel electrolyte, which glues the two electrode

135 films together. The thickness of electrolyte layer is about 15 μm and the capacitor
136 working area is 1 cm^2 .

137 We made three types of supercapacitors (D1, D2, and D3) with different electrodes,
138 as illustrated in figure 1. In D1, the CNT film was used as both current collector and
139 electrode. In D2, the graphene film attached to the CNT film was used as electrode. In
140 D3, the Au film was used to replace the CNT film as current collector.

141 **2. Results and discussions**

142 The CNT films were well attached to the PEN plastic substrates without any binder.
143 Figure 2(a) is an optical photograph showing the plastic substrate coated with carbon
144 nanomaterials used to make the supercapacitors. The graphene film coated onto the CNT
145 film can be easily recognized by naked eye. A tilted-view field-emission scanning
146 electron microscopy (FE-SEM, Zeiss Sigma) image in figure 2(b) shows that the CNT
147 film is felt-like and the graphene film on top has wrinkles. Figure 2(c) shows a free-
148 standing graphene film peeled off the substrate. CNTs stuck to the bottom face indicate a
149 strong binding force at the graphene/CNTs interface, though no use has been made of
150 bonder additives. Presumably, the strong bonding is due to the functional groups of the
151 graphene and the residual sulphuric acid present in the CNT films due to the reaction of
152 chlorosulfonic acid with the moisture of the air. In fact, our graphene, produced by
153 electrochemical exfoliation, contains 13 at. % of C-O and a 3 at. % of C=O functional
154 groups as revealed by X-ray photoelectron spectroscopy [22]. On the other hand, it has
155 been demonstrated that CNTs and graphene sheets in solution are mutually attracted due
156 to the π - π interactions between basal planes [25]. Hence the Van der Waals force is also a
157 factor making the graphene film adhere to the CNT film. Figure 2(d) shows the plane-

158 view SEM image of a CNT film. Bundles of CNTs are interlaced to form a mat-like film
159 used as current collector.

160

161 Devices D1 and D2 were tested and their performance results are compared in figure
162 3. Both devices had 520 nm-thick CNT films coated onto plastic sheets, however D1 had
163 the CNT film as both electrode and current collector, while D2 had an 85-nm-thick
164 graphene film coated onto the CNT film as electrode. Figures 3(a) and 3(b) show the
165 cyclic voltammetry (CV) curves of D1 and D2, respectively. The CV curves measured
166 from -1 V to +1V are quite symmetric and exhibit quasi-rectangular shape. The
167 galvanostatic charge-discharge (CD) curves of D1 and D2, measured at different current
168 values, are shown in figures 3(c) and 3(d), respectively. The CV curves are closer to a
169 rectangular shape and the CD curves are in a symmetric triangle shape. However, when
170 graphene was used as electrode (D2), the CV and CD curves were more distorted
171 compared to those of D1, but gained a much larger current range for the CV loop and
172 longer duration for a CD cycle. This means that the capacitance of D2 is higher than that
173 of D1, due to the graphene film electrode.

174 We made device D3 with a graphene film as electrode and a Au film as current
175 collector. D2 and D3 had identical graphene film electrodes with a thickness of 85 nm,
176 weighing $\sim 35 \mu\text{g}/\text{cm}^2$. We found that the adhesion force between the graphene sheets and
177 the Au film is relatively weak as the graphene film can be easily peeled off with a
178 tweezer. The Nyquist impedance spectra of D1, D2, and D3, measured from 0.05 Hz to 1
179 MHz, are shown in figure 4(a) with the high frequency region magnified in the inset. For
180 an ideal capacitor, the impedance plot would be a vertical line and the intercept on $\text{Re}(Z)$

181 axis would represent the internal resistance of the device. Therefore D1 with only the
182 CNT film on the substrate had the best capacitive behaviour. From the inset, the internal
183 resistance values for D1, D2, and D3 were found to be 78 Ω , 79 Ω , and 40 Ω ,
184 respectively. The relative high resistance of D1 and D2 can be attributed to the relatively
185 high sheet resistance of the CNT film (5 Ω /sq) and the contact resistance between the
186 CNT film and the metal clip. For D3, with a highly conductive Au film (in the order of
187 10^{-4} Ω /sq) as current collector, the internal resistance can be attributed to the graphene
188 film composed of slightly oxidized graphene sheets [22]. The sheet resistance of the CNT
189 film is higher than that of the Au film but lower than that of commercial ITO glass (8-60
190 Ω /sq) [26]. The variations of phase angle with respect to the scanning frequency are
191 shown in figure 4(b) for the three devices. At low frequencies, the D1 phase angle change
192 was up to -84° , close to the ideal value of -90° . The other two devices, D2 and D3, were
193 relatively inferior in the capacitive behavior, while their capacitance values were much
194 higher due to the use of graphene film, as shown in figure 4(c). The capacitance values
195 were extracted from the CV curves and the dependence of specific capacitance against
196 voltage scan rate was plotted for each device. The decrease of capacitance with the
197 increase of the voltage scan rate is related to the diffusion and response time of ions in
198 the electrolyte. D2 and D3 were almost identical in capacitance, indicating that: (i) the
199 CNT film is as efficient as the Au film for collecting charges from the electrode and (ii)
200 the CNT film does not contribute to the capacitance. At a scan rate of 10 mV/s, the areal
201 capacitance of D1 was 0.6 mF/cm², while that of D3 was 2.1 mF/cm². This means that the
202 capacitance of an 85-nm-thick graphene film is about 3.5 times larger than that of a ~520-
203 nm-thick CNT film. The areal density of the graphene film was 35 $\mu\text{g}/\text{cm}^2$, and that of the

204 CNT film was $29 \mu\text{g}/\text{cm}^2$. Using these values, the mass specific capacitance (F/g) for the
205 three devices were obtained (figure 4(d)).

206 In order to investigate the relationship of graphene film thickness and capacitance,
207 we made D2 type devices by varying the graphene film thicknesses. From this group of
208 devices we obtained the dependence of capacitance against graphene film thickness, as
209 shown in figure 5. Overall, the thicker the graphene film, the higher the areal specific
210 capacitance. In fact, by adding more graphene sheets, more ions can be stored in the
211 electrode. However, the mass specific capacitance decreases when the graphene film
212 thickness is increased from 100 nm to 360 nm. Charging the capacitor means driving ions
213 from the electrolyte into the stacked graphene sheets. Graphene sheets at the bottom of a
214 thick electrode are not fully exploited to trap ions, thus the mass specific capacitance of
215 thick graphene films is low when compared to that of a thin graphene films. The laser-
216 scribe graphene film produced by El-Kady *et al.* was about $7.6 \mu\text{m}$ -thick and the areal
217 capacitance was $\sim 5 \text{ mF}/\text{cm}^2$ [17]. In our work, the $0.36 \mu\text{m}$ -thick graphene film shows a
218 capacitance of $4.3 \text{ mF}/\text{cm}^2$. The areal capacitance could be increased by increasing the
219 graphene film thickness. However, as described in the experimental section, the graphene
220 film is transferred from the water surface to the CNT film; graphene films over a certain
221 thickness will contract and detach off the CNT film during the drying process. Having a
222 thin graphene film with high capacitance is important to achieve high volumetric energy
223 density, especially for a paper-like supercapacitor made by stacking a number of thin film
224 supercapacitor units. Figure 5(b) shows the stability of a D2 device with 85 nm-thick
225 graphene films, which was tested by repeatedly charging and discharging the device at 25
226 $\mu\text{A}/\text{cm}^2$ for five days. The device retained 89% of its initial capacitance after the stability

227 test. In fact, the device stability is related to the charge-discharge rate and the electrolyte.
228 The capacitance would be even more stable if the device was tested with a higher
229 charge/discharge current, which corresponds to shorter cycle period. Furthermore, the
230 stability could be improved if gelled ionic liquid is used as electrolyte [27, 28].

231 Energy and power densities of three devices, one D1 with only CNT films and two
232 D2 with graphene films in different thicknesses, are shown in figure 6. For comparison,
233 also the power and energy densities of Li ion thin film battery and previously reported
234 supercapacitor based on laser-scribed graphene are shown. In figure 6(a), the device with
235 360-nm-thick graphene film has the highest volumetric energy density ($\sim 10^{-4}$ Wh/cm³),
236 which is close to the energy density of Li ion film battery (10^{-4} - 10^{-3} Wh/cm³) [28]. Mass
237 energy and power densities of the three devices are shown in figure 6(b). The device with
238 85-nm-thick graphene films has the highest values in terms of mass power and energy
239 densities. Its power density is in the range of 250-450 kW/kg and the energy density
240 range is 8-14 Wh/kg. Xu *et al.* used ~ 120 - μ m-thick porous graphene film and gelled
241 electrolyte to make supercapacitors which showed very high areal specific capacitance up
242 to 400 mF/cm² [16]. The values of energy and power densities as a function of mass,
243 reported in that work, are represented by the green region in figure 6(b). The power
244 density (0.5-5.0 kW/kg) is about 100 times lower than that of our D2 device with 85 nm-
245 thick graphene films. The combination of a CNTs film and a graphene film, with total
246 thickness < 1 μ m when stacked together, has the potential to make thin film
247 supercapacitors with volumetric energy densities comparable to that of a Li film battery.
248 In our work, the electrochemically produced graphene sheets are compactly stacked to

249 form thin films. We believe that the capacitance can be further improved if the film is
250 processed to be rough and porous.

251

252 **3. Conclusion**

253 We fabricated solid-state electrochemical supercapacitors using graphene films as
254 electrodes and CNTs films as current collectors. Thin films of carbon nanomaterials were
255 supported by plastic papers which make the device bendable. By comparing different
256 types of devices, we concluded that thin graphene films, made by filtration of
257 electrochemically-produced graphene sheets, have much higher capacitance than the mat-
258 like CNT films. For supercapacitor application, CNT films are comparable to Au films
259 when used as current collectors and are superior to Au films in providing strong adhesion
260 to the graphene electrodes. We investigated the relationship of specific capacitance and
261 graphene film thickness. The areal specific capacitance can be improved by increasing
262 the electrode thickness, and 4.3 mF/cm^2 was achieved by using a 360-nm-thick graphene
263 film as electrode. Our study suggests that the combination of CNT films and graphene
264 films is very suitable for making paper-like supercapacitors with the graphene/CNT
265 bilayer film $< 1 \mu\text{m}$ in thickness while maintaining high energy and power densities.

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278 **References**

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362 **Figure captions**

363

364 **Figure 1.** Cartoon of the three devices. (a) D1: CNT films as both electrodes and current
365 collectors; (b) D2: CNT films as current collectors and graphene films as electrodes; (c)
366 D3: Au films as current collectors and graphene films as electrodes.

367 **Figure 2:** (a) Photograph of a typical electrode made by transferring a graphene film onto
368 a PEN-supported CNT film; (b) A tilted-view SEM image of the graphene/CNT double
369 layers; (c) A SEM image of a freestanding graphene film peeled off the substrate. CNTs
370 sticking to the graphene film indicate strong binding force between CNTs and graphene
371 sheets; (d) A plane-view image of the CNT film.

372 **Figure 3:** (a) and (b) CV curves for D1 and D2 devices measured at different scan rates,
373 respectively; (c) and (d) Galvanostatic charge-discharge curves of D1 and D2 devices
374 measured at different current densities, respectively.

375 **Figure 4:** (a) Nyquist impedance plots measured from 0.05 Hz to 1 MHz for the three
376 devices; (b) Dependence of phase angle change against frequency for the three devices;
377 (c) Specific areal capacitance extracted from the CV curves. The plots of specific mass
378 capacitance against scan rate for the three devices are shown in (d).

379 **Figure 5:** (a) Relation between specific capacitance and graphene electrode thickness for
380 D2; (b) Cycling stability of D2 measured by repeatedly galvanostatic charge-discharge.

381 **Figure 6:** (a) and (b) Ragone plots (energy density vs power density) for the three
382 devices calculated in terms of volume (cm^3) and weight (kg).

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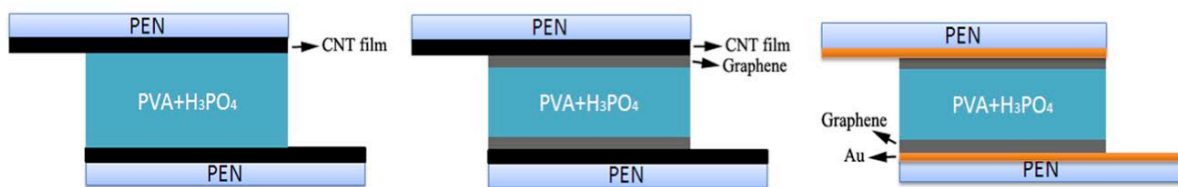
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Figures and Captions



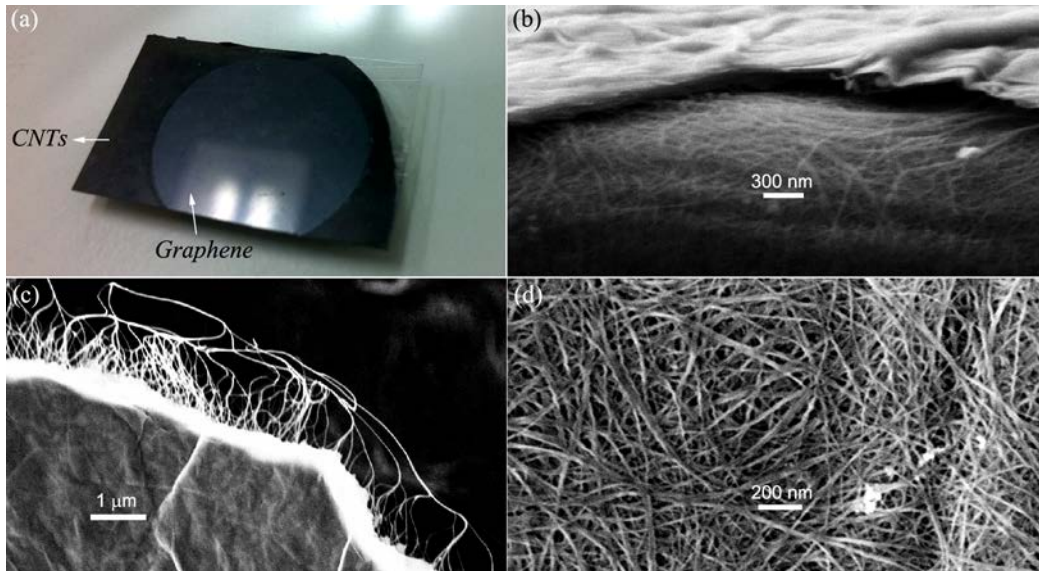
393

394 **Figure 1.** Cartoon of the three devices. (a) D1: CNT films as both electrodes and current

395 collectors; (b) D2: CNT films as current collectors and graphene films as electrodes; (c)

396 D3: Au films as current collectors and graphene films as electrodes.

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401 **Figure 2:** (a) Photograph of a typical electrode made by transferring a graphene film onto

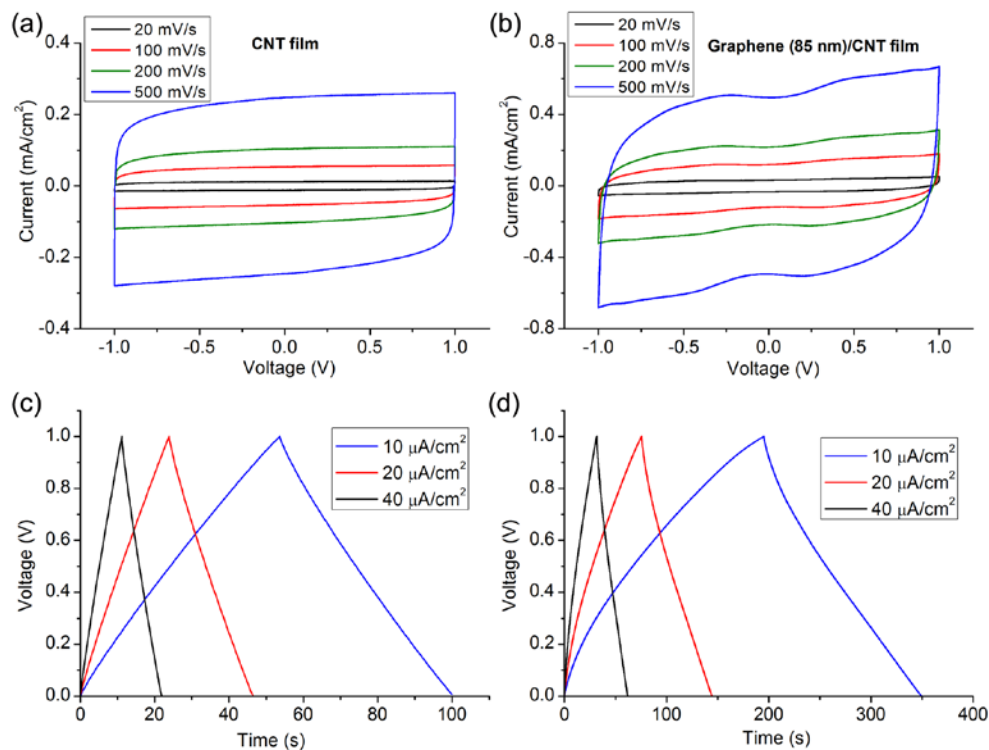
402 a PEN-supported CNT film; (b) A tilted-view SEM image of the graphene/CNT double

403 layers; (c) A SEM image of a freestanding graphene film peeled off the substrate. CNTs

404 sticking to the graphene film indicate strong binding force between CNTs and graphene

405 sheets; (d) A plane-view image of the CNT film.

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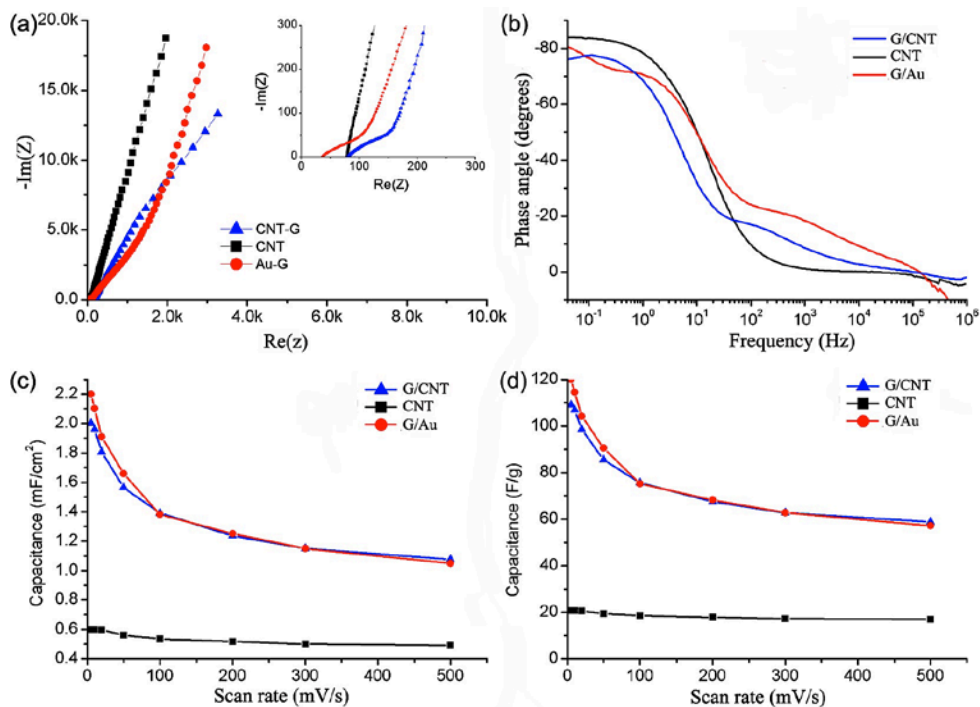


407

408 **Figure 3:** (a) and (b) CV curves for D1 and D2 devices measured at different scan rates,
 409 respectively; (c) and (d) Galvanostatic charge-discharge curves of D1 and D2 devices
 410 measured at different current densities, respectively.

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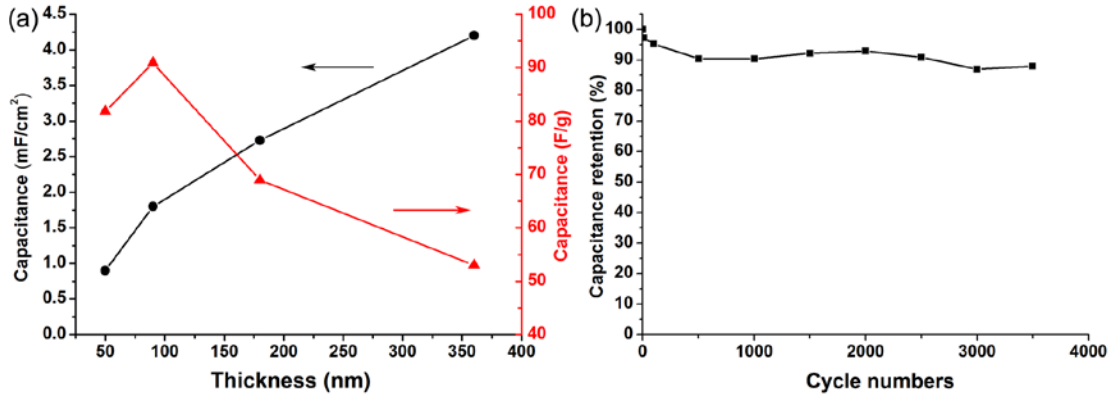
414 **Figure 4:** (a) Nyquist impedance plots measured from 0.05 Hz to 1 MHz for the three
 415 devices; (b) Dependence of phase angle change against frequency for the three devices;
 416 (c) Specific areal capacitance extracted from the CV curves. The plots of specific mass
 417 capacitance against scan rate for the three devices are shown in (d).

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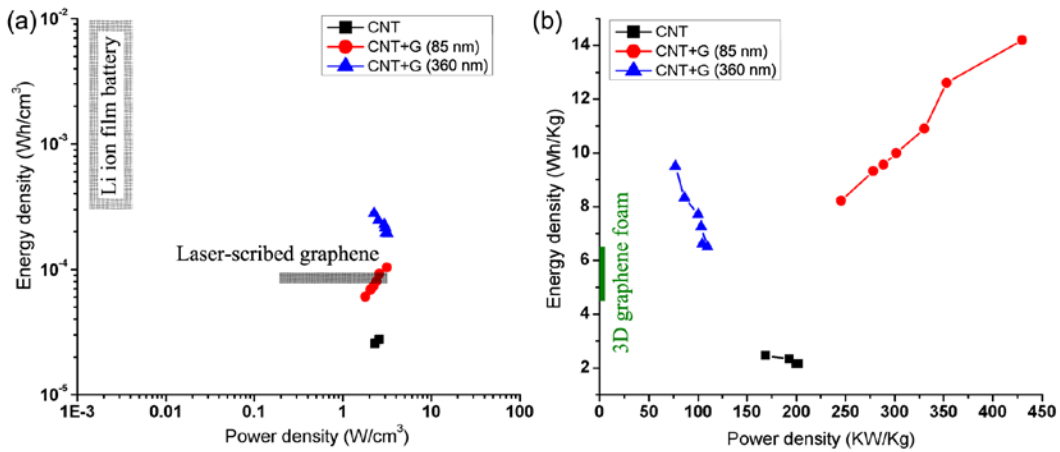


422

423 **Figure 5:** (a) Relation between specific capacitance and graphene electrode thickness for

424 D2; (b) Cycling stability of D2 measured by repeatedly galvanostatic charge-discharge.

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426

427 **Figure 6:** (a) and (b) Ragone plots (energy density vs power density) for the three

428 devices calculated in terms of volume (cm³) and weight (kg).

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Supporting Information

434

Graphene-based supercapacitor with carbon nanotube film as highly efficient

435

current collector

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443 **1. Raman analysis**

444

Figure S1 shows the Raman spectrum (Reinshaw *InVia*) of the DWCNT film measured

445

using the 534 nm laser source. The G band Raman mode at 1593 cm^{-1} corresponds to the

446

tangential movement of atoms on the nanotube surface. The D band, which is observed at

447

1345 cm^{-1} , is related to defects and disorder.

448

Raman analysis of the graphene film has been reported previously [1]. The intensity

449

ratio I_D/I_G of graphene can be used to assess the quality. In this regard, our

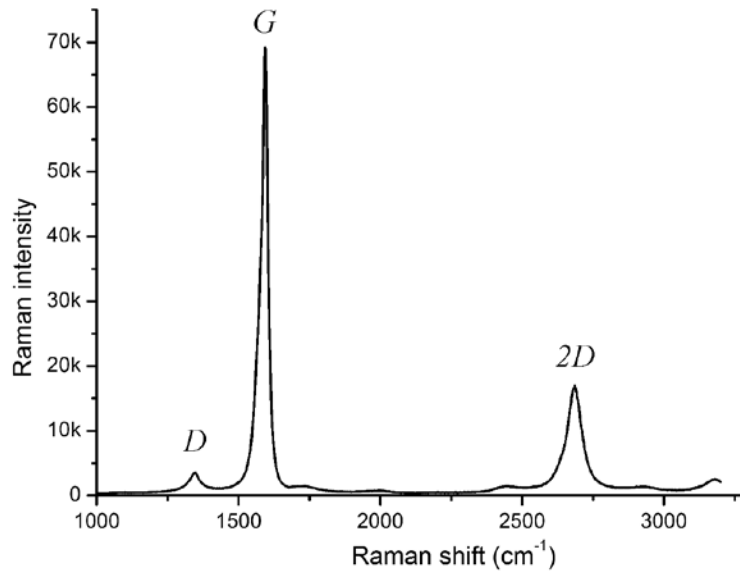
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electrochemically produced graphene sheets have less defects and higher quality than GO

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made by the Hummer's method.

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Figure S1: Raman spectrum of a DWCNT film.

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456 **2. Strong adhesion between CNTs and graphene sheets**

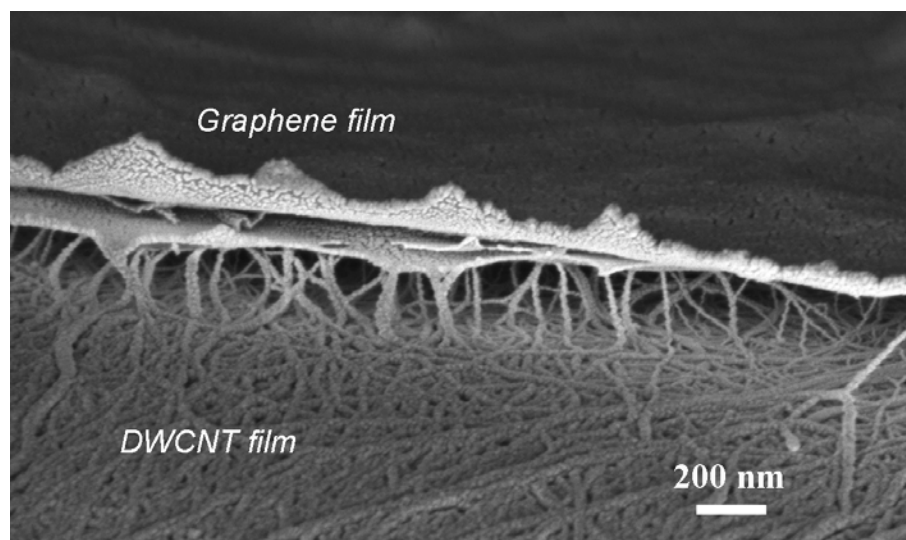
457 The carbon-nanomaterial-coated plastic substrate was cut to observe the interface

458 between the CNT film and graphene film. The SEM image in figure S2 shows that the

459 graphene film was separated at the edge from the CNT film base due to blade cutting.

460 However those CNTs linking to the bottom face of the graphene film indicate a strong

461 binding force between CNTs and graphene sheets.



462

463 **Figure S2:** SEM image showing the interface between CNT and graphene films.
464 Strong binding force between CNTs and graphene sheets makes the graphene film
465 attach to the CNT film. The nanoparticle feature is caused by the deposition of Au
466 layer for SEM imaging.

467

468 **3. Determination of the CNTs film thickness**

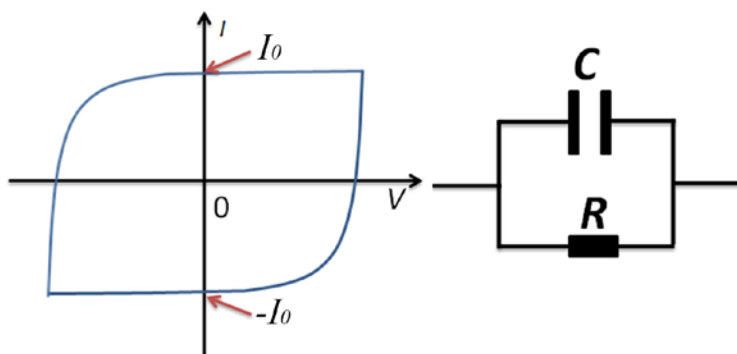
469 The film thickness was determined by the Beer-Lambert law: $T = e^{-\alpha h}$, where α is the
470 extinction coefficient, T is the optical transmittance and h is the film thickness. The
471 extinction coefficient α was found to be 0.0054 nm^{-1} by measuring h and T for several
472 films by Atomic Force Microscopy [AFM (Digital Instrument Nanoscope IIIA Atomic
473 force microscope)] and UV-vis spectrometry (Shimadzu UV-1800).

474 **4. Device Performance Calculations**

475 The CV curves are measured in the voltage range of -1 V to +1 V. By treating the
476 graphene supercapacitor as a RC circuit (figure S3), the total current through the

477 capacitor is $I = C \times dV/dt + V/R$, where dV/dt is the voltage scan speed. At $V=0$,
 478 $I_0 = C \times dV/dt$. Hence the capacitance can be obtained as $C=I_0/(dV/dt)$.

479 To calculate the capacitance using galvanostatic CD curves, the current I and the
 480 slope of the discharge curve, dV/dt , are used: $C=I/(dV/dt)$. Note that the capacitance
 481 range calculated from the CD curves is similar to that from the CV curves.



482

483 **Figure S3:** The use of CV curve to deduce the capacitance, by taking the capacitor as a
 484 RC circuit.

485

486 The power of a single supercapacitor is calculated from the galvanostatic curves at
 487 different charge/discharge current densities using the following formula:

488
$$P = (\Delta V)^2 / 4R_{ESR}Ad \quad (1)$$

489 Where P is the power (W/cm^3), ΔV is the operating window (measured in volts and
 490 obtained from the discharge curve excluding the IR drop), A is the area and d is the
 491 thickness of gelled electrolyte layer. R_{ESR} is the internal resistance of the device that can
 492 be estimated from the voltage drop at the beginning of the discharge and from the
 493 discharge current I by $R_{ESR} = V_{drop}/2I$.

494 The energy density of the device is obtained by using Eq. (2):

495
$$E = C \times (\Delta V)^2 / (2 \times 3600 \times d) \quad (2)$$

496 Where E is the energy density in Wh/cm³, C is the areal capacitance, and d is the gap
497 between two opposite electrodes.

498