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1 **An injectable, self-healing and MMP-inhibiting hyaluronic acid gel via iron coordination**

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12

13 **Abstract**

14 Regulating the activity of matrix metalloproteinases (MMPs) is a potential strategy for  
15 osteoarthritis (OA) therapy, although delivering this effect in a spatially and temporally  
16 localised fashion remains a challenge. Here, we report an injectable and self-healing hydrogel  
17 enabling factor-free MMP regulation and biomechanical competence *in situ*. The hydrogel is  
18 realised within one minute upon room temperature coordination between hyaluronic acid  
19 (HA) and a cell-friendly iron-glutathione complex in aqueous environment. The resultant gel  
20 displayed up to 300% in shear strain and tolerance towards ATDC 5 chondrocytes, in line with  
21 the elasticity and biocompatibility requirements for connective tissue application.  
22 Significantly enhanced inhibition of MMP-13 activity was achieved after 12 hours *in vitro*,  
23 compared with a commercial HA injection (OSTENIL® PLUS). Noteworthy, 24-hour incubation  
24 of a clinical synovial fluid sample collected from a late-stage OA patient with the reported  
25 hydrogel was still shown to downregulate synovial fluid MMP activity ( $100.0 \pm 17.6\% \rightarrow$   
26  $81.0 \pm 7.5\%$ ), with at least comparable extent to the case of the OSTENIL® PLUS-treated SF  
27 group ( $100.0 \pm 17.6\% \rightarrow 92.3 \pm 27.3\%$ ). These results therefore open up new possibilities in the  
28 use of HA as both mechanically-competent hydrogel as well as a mediator of MMP regulation  
29 for OA therapy.

30 **Keywords:** Hyaluronic acid, iron-glutathione complex, injectable hydrogel, synovial fluid,  
31 osteoarthritis, MMP-13 inhibition.

32

33 **Introduction**

34 Osteoarthritis (OA) is a chronic and irreversible disease which results in continuous  
35 cartilage degradation, increased joint friction, and pain. The onset and progression of OA is  
36 closely linked to proteolytic imbalances, whereby upregulated activity of matrix

37 metalloproteinases (MMPs), particularly MMP-13 (collagenase), results in the pathological  
38 breakdown of articular cartilage (Yoshihara et al., 2000) (Burrage et al., 2006) (H. Li et al.,  
39 2017). MMP-13 concentration strongly correlates to vascular endothelial growth factor  
40 (VEGF) concentration, which plays an important role in angiogenesis and can serve as a  
41 biomarker for OA diagnosis and therapeutic monitoring (Kim et al., 2011). In addition, the  
42 overexpression of MMP-13 is found in advanced osteoarthritic synovial fluid (Heard et al.,  
43 2012). Injectable, non-cytotoxic and biomechanically viable materials that are able to inhibit  
44 MMP-13 are highly sought to restore tissue homeostasis and minimise the risks of knee  
45 replacement (M. Wang et al., 2013).

46 Injectable materials enable the delivery and localisation of therapeutic compounds at a  
47 target diseased site. In particular, injectable materials that mimic the features of the  
48 extracellular matrix (ECM) are ideal therapeutic scaffolds since they enable cell attachment,  
49 proliferation and temporally controlled mechanical function with minimal toxic effect  
50 following degradation (Stevens & George, 2005) (Blache et al., 2020). As such, they have been  
51 widely employed as carriers for improved mesenchymal stem cell (MSC) delivery for bone  
52 repair and OA management (M. Liu et al., 2017). Hydrogel systems that contain synthetic  
53 polymers have shown promise as materials for OA management due to their injectability and  
54 versatility in presenting bioactive functionalities that downregulate MMP activity and prolong  
55 the activity of encapsulated MSCs (Clark et al., 2020). Yet, the limited degradability of many  
56 synthetic polymers and the demands of polymer synthesis make their translation to  
57 commercial products challenging. The design of injectable hydrogels from ECM-derived  
58 polymers that can correct proteolytic imbalances may provide an alternative cell-free and  
59 regulatory-friendly strategy for OA management, which avoids non-biodegradable synthetic  
60 polymers.

61 Hyaluronic acid (HA) is an anionic non-sulfated glycosaminoglycan that constitutes one of  
62 the main components of cartilaginous ECM (Slepecky, 1967). Due to its polysaccharide  
63 backbone, a great deal of attention has been put into investigating HA functionalisation for  
64 targeted applications, aiming to accomplish tuneable physicochemical properties (Zamboni et  
65 al., 2020) and improved cell viability (Zamboni et al., 2017). However, many commercially  
66 available HA-based products are in the form of injectable materials, for instance OSTENIL®  
67 PLUS, which is routinely applied in the clinic for the treatment of osteoarthritic joints.  
68 Significantly improved knee function and pain relief were confirmed through the Visual Analog  
69 Scale (VAS) score and the Western Ontario and McMaster Universities Osteoarthritis Index  
70 (WOMAC) score (Kotevoglu et al., 2006)(Dernek et al., 2016). HA injections are usually  
71 suggested to be delivered every 1-2 weeks to the joint cavity, although they are unable to  
72 control OA-related MMP upregulation. Despite HA's capability to interact with and stimulate  
73 chondrocytes *in vivo*, these products are only designed to offer a palliative, short-lived  
74 biomechanical solution that is used as a last resort prior to joint replacement. Intelligent HA

75 formulations that include therapeutics for OA treatment through MMP-13 inhibition, and  
76 retain mechanical stability, are highly sought. To pursue this vision, a cell-friendly iron-  
77 glutathione ( $\text{Fe}^{3+}$ -GSH) complex recently reported by our group (Gao et al., 2020) was  
78 investigated for use as both a crosslinker of HA to yield an injectable hydrogel, and as a  
79 potential therapeutic to inhibit MMP-13 activity, exploiting the competitive metal-  
80 coordinating reaction between thiol complexed iron ( $\text{Fe}^{3+}$ ) and free sulfhydryl groups of active  
81 MMPs.

82 Although some effort afforded the creation of HA-containing gels via metal coordination,  
83 e.g. INTERGEL™, unpleasant side-effects and serious complications experienced by many  
84 patients call for new safer alternatives (Tang et al., 2006). To prevent tissue damage from  $\cdot\text{OH}$   
85 and peroxy-type radicals, which could be generated during hyaluronic acid degradation  
86 (Katarína Valachová et al., 2016; Katarína Valachová et al., 2015), it is important to involve  
87 reductive components into the HA-based therapeutic material, for example, thiol groups  
88 (Katarína Valachová et al., 2015). In this case, introducing cell friendly  $\text{Fe}^{3+}$ -GSH complex into  
89 HA hydrogels is worth investigating.

90 Hydrogel injectability has been pursued via dynamic covalent chemistries in biopolymer-  
91 based hydrogels for tissue engineering, including Schiff-base reactions (Huang et al., 2016; S.  
92 Li et al., 2020), Diels-Alder reactions (DA) click coupling reactions (Hu et al., 2019) (Spicer,  
93 2020), as well as via thermal gelation mechanisms (Zhang et al., 2019; Lee et al., 2020)  
94 compliant with injection-mediated delivery. On the one hand, the formation of covalently  
95 crosslinked hydrogels with appropriate mechanical properties in physiological conditions to  
96 reduce joint friction has up to now proven challenging. This is largely due to the fact that the  
97 presence of covalent crosslinks reduces hydrogel's dynamic tensile, compressive and shear  
98 strain, limiting hydrogel's ability to bear multiple load-bearing cycles, as in the case of articular  
99 cartilage. On the other hand, although thermosensitive polymer formulations have been  
100 developed, only a limited number have been made with HA formulations free of the synthetic  
101 polymer phase (Zhang et al., 2019).

102 Other than covalent networks, redox-based self-healable and injectable polymer hydrogels  
103 were achieved that can withstand relatively high shear strain (~50 %) (Chen et al., 2019) (L.  
104 Liu et al., 2019). Likewise, metal-coordinated hybrid materials have been reported serving as  
105 electroconductive materials (Shi et al., 2015), catalyst supports (Loynachan et al., 2019), and  
106 for magnetic resonance imaging (Paquet et al., 2011) (H. Wang et al., 2019). Ultimately,  
107 composite hydrogels have been made of multiple biopolymers and bioglass and ionically  
108 crosslinked by calcium dications (Yu et al., 2019). The composite material is able to withhold  
109 quercetin, an MMP inhibitor, so that 70% reduction in MMP-13 expression was reported after  
110 48 hours, which proved key to induce cartilage repair after 12 weeks *in vivo*. These studies  
111 provide novel design concepts that harness the functionalities of metals and peptides, aiming

112 to build simple ECM mimetics with flexible mechanical properties and MMP inhibition  
113 capability.

114 In this work, the straightforward creation of a non-toxic HA-based hydrogel that is  
115 injectable and self-healing is reported. HA combined with an iron (III)-glutathione ( $\text{Fe}^{3+}$ -GSH)  
116 complex results in the formation of a physical hydrogel upon co-injection. We hypothesised  
117 that hydrogel-induced MMP inhibition was accomplished by harnessing the metal-  
118 coordinating reaction between thiol-complexed iron(III) and the free sulfhydryl groups of  
119 active MMPs. Crucially, the  $\text{Fe}^{3+}$ -GSH complex has the dual function of being the crosslinker  
120 within the hydrogel, and also providing a therapeutic effect for inhibiting MMP activity, as  
121 confirmed with synovial fluid clinical samples collected from patients with late-stage OA.  
122 Consequently, the hydrogel may act as a self-healable scaffold that reduces joint friction and  
123 halts cartilage degradation, whilst boosting local cell function. Delivery of this system *in situ*  
124 has significant potential in OA therapy, aiming to prevent the degradation of cartilage whilst  
125 correcting growth factor concentrations and cellular activity towards cartilage repair.

126

## 127 **2. Materials and methods**

128 The hyaluronic acid sodium salt (molecular weight: 1,200 kDa, cosmetic grade) was  
129 purchased from Hollyberry Cosmetic. L-glutathione (reduced) was purchased Alfa Aesar.  
130 Alamar Blue assay kit was from ThermoFisher Scientific. Human recombinant Pro-MMP 13  
131 was purchased from Antibodies.com, and the MMP activity assay kit (Fluorometric Green,  
132 ab112146) was from ABCChem. All the other reagents were provided by Sigma-Aldrich.

### 133 **Rheology of HA solutions supplemented with $\text{Fe}^{3+}$ -GSH**

134 Different concentrations of  $\text{Fe}^{3+}$ -GSH complex were added to the HA solution (**Table S1**) to  
135 achieve the optimal, most stable, hydrogel. To exclude the influence of HA concentration on  
136 gel formation, the final concentration of HA in the gel-forming mixture was controlled to 1.33  
137 wt.% by addition of deionised water. All test group samples were named as “Fe xxx”, in which  
138 “xxx” corresponds to the volume of  $\text{Fe}^{3+}$ -GSH solution ( $\mu\text{L}$ ) in the HA solution (mL). All control  
139 samples were named as “Ctrl xxx”, in which “xxx” corresponds to the volume ( $\mu\text{L}$ ) of  $\text{Fe}^{3+}$ -GSH  
140 solvent (120 mM HCl) per mL of HA solution.

141 The  $\text{Fe}^{3+}$ -GSH-supplemented HA solution was injected onto an MCR 302 Rheometer (Anton  
142 Paar) and pressed by a 25 mm parallel plate (1.5 mm gap) at 37 °C with a variable shear rate  
143 to study the viscosity of hydrogels formed with different  $\text{Fe}^{3+}$ -GSH complex content.

### 144 **Preparation of $\text{Fe}^{3+}$ -GSH self-healing HA hydrogel (Fe 300)**

145 The  $\text{Fe}^{3+}$ -GSH complex was prepared using our previous method (Gao et al., 2020). Briefly,  
146 123 mg (0.4 millimoles) of GSH was added to 4 mL  $\text{FeCl}_3$  aqueous solution (0.1 M), and the  
147 mixture was mildly agitated by vortex mixing for 2 min until the solution became yellow. Then,

148 the complex was precipitated by adding 40 mL of ethanol ( $\times 3$ ) and collected by centrifugation  
149 at 10,000 rpm for 15 min. The  $\text{Fe}^{3+}$ -GSH complex was dried at 37 °C for further use.

150 10 mg of  $\text{Fe}^{3+}$ -GSH complex was dissolved in 1 mL HCl solution (120 mM). Each 300  $\mu\text{L}$   
151  $\text{Fe}^{3+}$ -GSH complex solution was added to 1 mL hyaluronic acid solution (2 wt.%) and stirred at  
152 room temperature for 1 min to obtain a self-healing hydrogel ( $\text{Fe}^{3+}$ -GSH gel). The self-healing  
153 behaviour of all hydrogels formed was characterised by determining the reversible viscosity  
154 from a low shear strain (0.01 %) for 200 s, followed by a high shear strain (500 %)  
155 measurement for 100 s at 37 °C. The testing frequency was fixed at a constant value of 5  
156  $\text{rad}\cdot\text{s}^{-1}$ . Ten low-to-high shear strain cycles were measured in this process using an Anton Paar  
157 MCR 302 rheometer.

### 158 **Determination of hydrogel shear modulus and shear strength**

159 The shear modulus (storage modulus  $G'$  and loss modulus  $G''$ ) of the  $\text{Fe}^{3+}$ -GSH crosslinked  
160 hydrogel (Fe 300) was measured via a frequency sweep using an MCR 302 rheometer (Anton  
161 Paar). This method was set with a 25 mm parallel plate at 37 °C, 1.5 mm gap, from 1-100 rad/s  
162 under 5 % amplitude.  $G'$  and  $G''$  were determined at 37 °C over a shear strain range of 0-500  
163 % with a constant angular frequency (5  $\text{rad}\cdot\text{s}^{-1}$ ). Every 1.0 mL volume of  $\text{Fe}^{3+}$ -GSH gel was  
164 injected onto the sample plate and slightly pressed by a 25 mm parallel plate geometry with  
165 a gap of 1.5 mm. Hyaluronic acid with the same amount of HCl solution only was measured as  
166 a control for both shear modulus and shear strain.

### 167 **Molecular mechanism study**

168  $^{57}\text{Fe}$  Mössbauer spectroscopy was applied to study iron chelation and valence.  
169 Measurements were carried out using acrylic absorber discs (area: 1.8  $\text{cm}^2$ ) loaded with a  
170 dried gel sample to achieve a Mössbauer thickness of 1. The 14.4 keV  $\gamma$ -rays were supplied by  
171 the cascade decay of 25 mCi  $^{57}\text{Co}$  in Rh matrix source, oscillated at constant acceleration by a  
172 SeeCo W304 drive unit, detected using a SeeCo 45431 Kr proportional counter operating with  
173 1.745 kV bias voltage applied to the cathode. All measurements were carried out at 293 K over  
174 a velocity range of  $\pm 6 \text{ mm}\cdot\text{s}^{-1}$ , and were calibrated relative to  $\alpha$ -Fe foil. Spectral data were  
175 fitted using the Recoil software package, using a single Lorentzian line shape necessitated by  
176 the low signal/noise ratio obtained for the sample (indicative of its low Fe content).

### 177 **Cellular tolerability study**

178 ATDC 5 chondrocytes were cultured (37 °C, 5%  $\text{CO}_2$ ) in a mixed medium of Dulbecco's  
179 modified Eagle's medium (DMEM) and Ham's F12 medium (1:1 in volume), supplemented  
180 with 5% fetal bovine serum (FBS), and 1 % penicillin-streptomycin. A defined amount of self-  
181 healing gel was transferred into individual wells of a 96-well-plate and diluted by cell culture  
182 medium to a final concentration of 0  $\mu\text{L}$  (tissue culture plastics, TCPs), 5  $\mu\text{L}$ , 10  $\mu\text{L}$ , 20  $\mu\text{L}$ , 30  
183  $\mu\text{L}$ , 40  $\mu\text{L}$  and 50  $\mu\text{L}$  per well, followed by addition of 100  $\mu\text{L}$  cell suspension ( $5\times 10^4$  cells/mL)

184 in each (n=4). The cell viability was quantified by Alamar blue assay after 1-day, 3-day, 5-day  
185 culture. Cells cultured on TCPs were set as the control group.

#### 186 **MMP-13 inhibition study with MMP-13-supplemented solution**

187 The self-healing gel, as well as an HA solution and a commercial HA gel for OA injection,  
188 OSTENIL® PLUS (both with the same HA concentration as the self-healing gel), were added to  
189 deionised water (×4). Then, 20 µL of each sample was added to individual wells of a 96-well  
190 plate, followed by adding 80 µL H<sub>2</sub>O per well. Pro-MMP 13 was activated following the  
191 manufacturer protocol. Briefly, 5 µL MMP-13 (10 µg MMP-13/20 µL sample) was dissolved in  
192 a *p*-aminophenyl mercuric acetate (AMPA) working solution (1 mM) to 1 µg/mL and then  
193 incubated at 37 °C for 40 min. Activated MMP-13 was diluted with AMPA solution (2 mM) to  
194 25 ng/mL and then immediately added into the sample wells (each containing 100 µL of the  
195 sample), corresponding to a final MMP-13 concentration of 12.5 ng/mL to cover the enzymatic  
196 concentration (6 ng/mL) recorded in synovial fluid samples of advanced OA patients (Heard et  
197 al., 2012). Deionised water with an equal volume of APMA solution (2 mM) was set as the  
198 blank, and deionised water with an equal volume of activated MMP-13 was set as the none  
199 treatment group. After 12-hour or 24-hour incubation, MMP-13 activity was quantified via  
200 fluorometric assay (Fluorometric Green, ab112146, Abcam) (Liang et al., 2018). 50 µL of each  
201 sample was pipetted into a new 96-well-plate, followed by 50 µL of MMP Green Substrate  
202 working solution. MMP 13-activity was recorded in fluorescence after 1-hour reaction in dark  
203 at 37 °C using a microplate reader (Thermo Scientific Varioskan® Flash, Ex/Em=490/525 nm).

#### 204 **MMP-13 regulation study with patient collected synovial fluid**

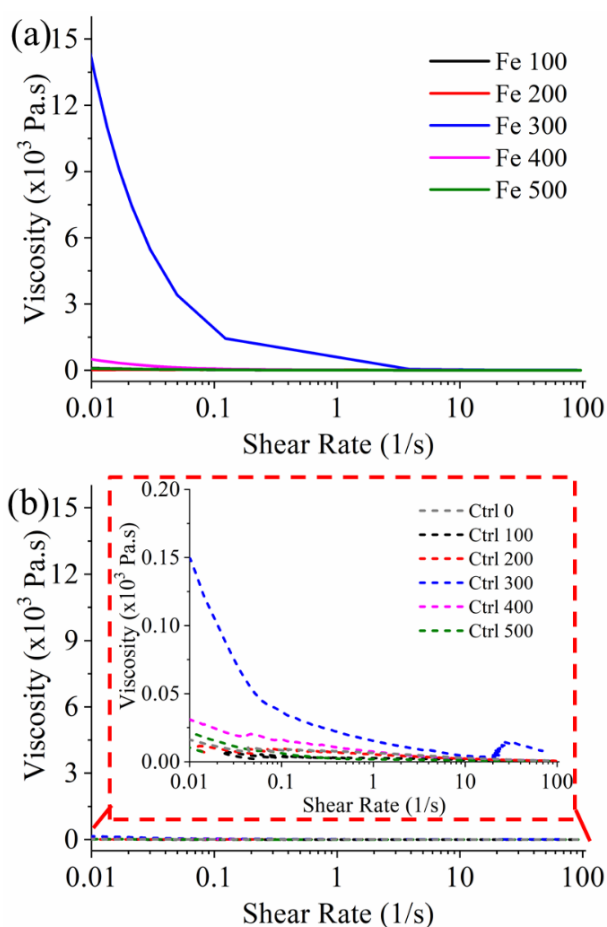
205 Synovial fluid (SF) samples were collected from late-stage osteoarthritic patients at Chapel  
206 Allerton Hospital (Leeds, UK) under ethical approval granted by the National Research Ethics  
207 Committee (ethical approval number: 07/Q1205/27). SF samples were stored at -80 °C until  
208 use. A fluorometric assay kit (Fluorometric Green, ab112146) was used to measure the total  
209 proteolytic activity in both SF and hydrogel-incubated SF samples. SF samples were diluted  
210 with the MMP assay buffer (×4), and the final Fe<sup>3+</sup>-GSH crosslinked gel dose was increased  
211 (×4). 50 µL of diluted SF were mixed with 40 µL of Fe<sup>3+</sup>-GSH crosslinked gel, and 10 µL of  
212 deionised water was supplemented in each well to achieve a final concentration of 100 µL/mL  
213 [Fe<sup>3+</sup>-GSH crosslinked gel/solution]. The fluorometric assay was conducted after 24-hour  
214 incubation following the same assay protocol reported for MMP-13 activity measurement.

#### 215 **Statistical analysis**

216 All the samples were tested with at least three replicates (n≥3) and presented as Mean±SD.  
217 Statistical significance level was calculated through one-way ANOVA with a p-value at 0.05.  
218 Final statistical results were presented as \*p≤ 0.05, \*\*p≤ 0.01, \*\*\*p≤ 0.001, \*\*\*\*p≤0.0001.

219 **Results and discussion**

220 Attempts to create hydrogels from HA (2 wt.%) and varying amounts of the Fe<sup>3+</sup>-GSH  
221 complex (10 mg/mL) were conducted, and the optimal hydrogel was formed from 300 μL Fe<sup>3+</sup>-  
222 GSH complex (10 mg/mL) and 1 mL HA solution (2 wt.%). A significant decrease in viscosity  
223 was observed with increasing shear rate from 0.01 Hz (14,400 Pa·s) to 4 Hz (37 Pa·s), whereas  
224 the viscosity remained constant at shear rates between 4 Hz and 100 Hz (**Fig. 1a**). Compared  
225 with the other materials created, the stability in hydrogel viscosity suggested a balanced  
226 coordination at a Fe<sup>3+</sup>-GSH crosslinker concentration of 300 μL per mL of HA solution. On the  
227 other hand, in the HA solution control groups, replacement of the Fe-GSH complex with the  
228 HCl solution resulted in significantly lower viscosity(**Fig. 1b**), whereby no significant viscosity  
229 variation was observed across the control groups.



230

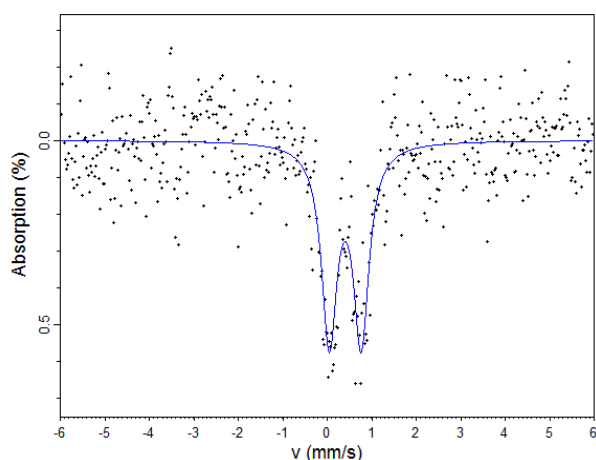
231 **Fig. 1** Flow curve of aqueous solutions supplemented with (a) either varied Fe<sup>3+</sup>-GSH complex/HA ratio  
232 or (b) varied concentration of HA, enlarged within the red box.

233

234 The iron oxidation state in the optimal hydrogel (Fe 300) was ferric (Fe<sup>3+</sup>) occupying  
235 octahedral coordination (Dyar et al., 2006)) ((Khalil et al., 2013), as determined by <sup>57</sup>Fe  
236 Mössbauer spectroscopy (**Fig. 2**), which also confirmed the chelation of Fe<sup>3+</sup> to HA. The  
237 confirmed Fe<sup>3+</sup> state in the hydrogel therefore speaks against a GSH-induced reduction to Fe<sup>2+</sup>

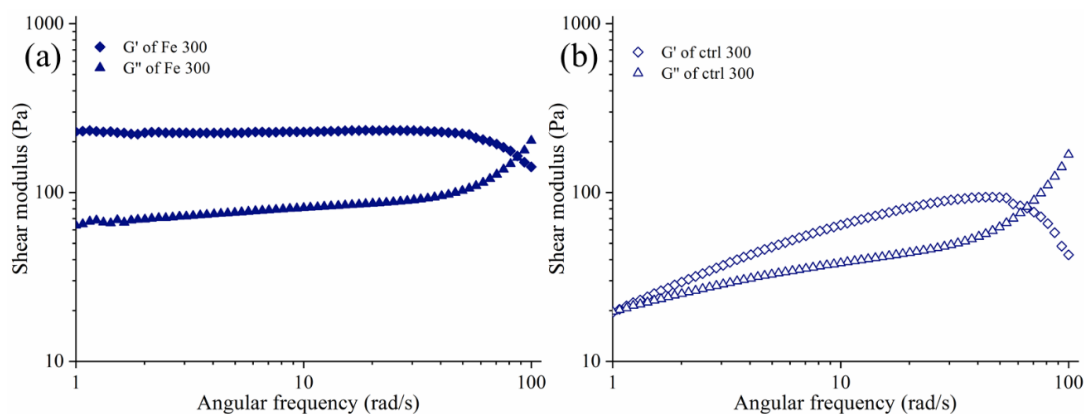


238 and the consequent generation of toxic reactive oxygen species, supporting the safe  
 239 injectability of the HA hydrogel in the OA site. In light of these characteristics, the  
 240 aforementioned hydrogel Fe 300 was chosen for further investigation.  
 241



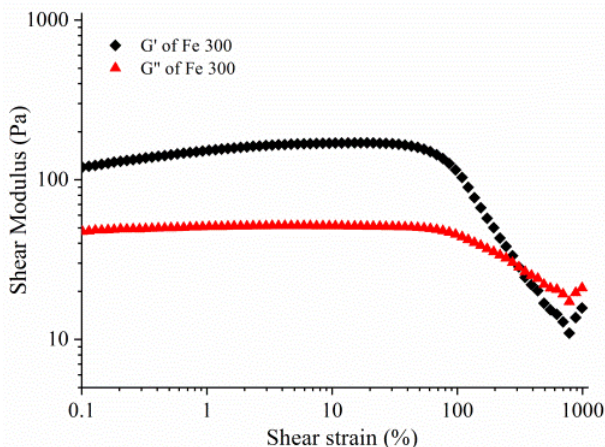
242  
 243 **Fig. 2** Fitted  $^{57}\text{Fe}$  Mössbauer spectrum of dry  $\text{Fe}^{3+}$ -GSH gel at 293 K, relative to thin  $\alpha\text{-Fe}$  foil. The clear  
 244 presence of a doublet attributable to paramagnetic  $\text{Fe}^{3+}$  can be observed, despite the low signal/noise  
 245 ratio due to the low abundance of  $\text{Fe}^{3+}$ -GSH content in the gel. Fitted centre shift ( $\delta$ ) =  $0.41 \pm 0.02$  mm  
 246  $\text{s}^{-1}$  and quadrupole splitting ( $\Delta$ ) =  $0.72 \pm 0.02$  mm  $\text{s}^{-1}$  with HWHM linewidth =  $0.21 \pm 0.02$  mm  $\text{s}^{-1}$ .  
 247

248 A much higher  $G'$  value (120 Pa) was recorded for the Fe 300 gel that contained the  $\text{Fe}^{3+}$ -  
 249 GSH crosslinker, compared to the HCl-HA control (10 Pa), again indicating that Fe-coordination  
 250 to HA enables gel formation. Constant storage ( $G' = 120$  Pa) and loss ( $G'' = 70$  Pa) moduli of the  
 251 self-healing gel were successfully measured in frequency sweep mode, confirming a  
 252 predominantly elastic behaviour in the range of 1-40  $\text{rad}\cdot\text{s}^{-1}$ , whilst the material elasticity was  
 253 found to decrease at the increased angular frequency (**Fig. 3a**). Although the storage modulus  
 254 is reduced compared to the chemically crosslinked HA hydrogel ( $G' = 300$  Pa), the elastic range  
 255 was much greater (angular frequency: 1-10  $\text{rad}\cdot\text{s}^{-1}$ ) compared to the latter care (Gao et al.,  
 256 2019). This behaviour illustrates the homogeneous nature of the gel. Conversely, the HCl-HA  
 257 control sample presented an obvious decrease in moduli from high to low frequency (**Fig. 3b**).



258  
 259 **Fig. 3** Shear modulus of  $\text{Fe}^{3+}$ -GSH hydrogel (a) and ctrl 300 samples (b) recorded during the frequency  
 260 sweep.

261 **Fig. 4** reveals the variability of dynamic shear modulus under shear strain (0.01-500 %) for  
262 the Fe<sup>3+</sup>-GSH crosslinked gel. A predominantly elastic gel response was observed up to 300 %  
263 shear strain, whereby both the storage and loss moduli remained constant when up to 80 %  
264 shear strain was applied with 5 rad/s (0.8 Hz) frequency.



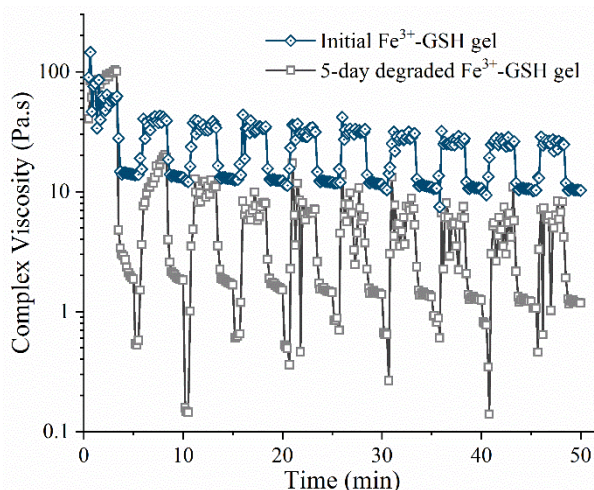
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266

**Fig. 4** Shear modulus of Fe<sup>3+</sup>-GSH gel measured via strain sweep.

267 These results demonstrate mechanical compliance of the hydrogel with the ranges of shear  
268 strain (up to 1 %) and frequency (0.5-2.0 Hz) observed *in vivo* in both connective and fatty  
269 tissues (Yoo et al., 2011). In line with previous results, the storage modulus of the Fe<sup>3+</sup>-GSH  
270 coordinated gel was found to be greater (105 Pa) than that of the hyaluronic acid control (70  
271 Pa, **Fig. S1**), demonstrating increased mechanical competence.

272 After 10 cycling tests from low shear strain to high strain, Fe<sup>3+</sup>-GSH crosslinked gels  
273 presented a stable complex viscosity in the range of 37-42 Pa·s and 12-16 Pa·s, respectively  
274 (**Fig. 5 blue**).



275

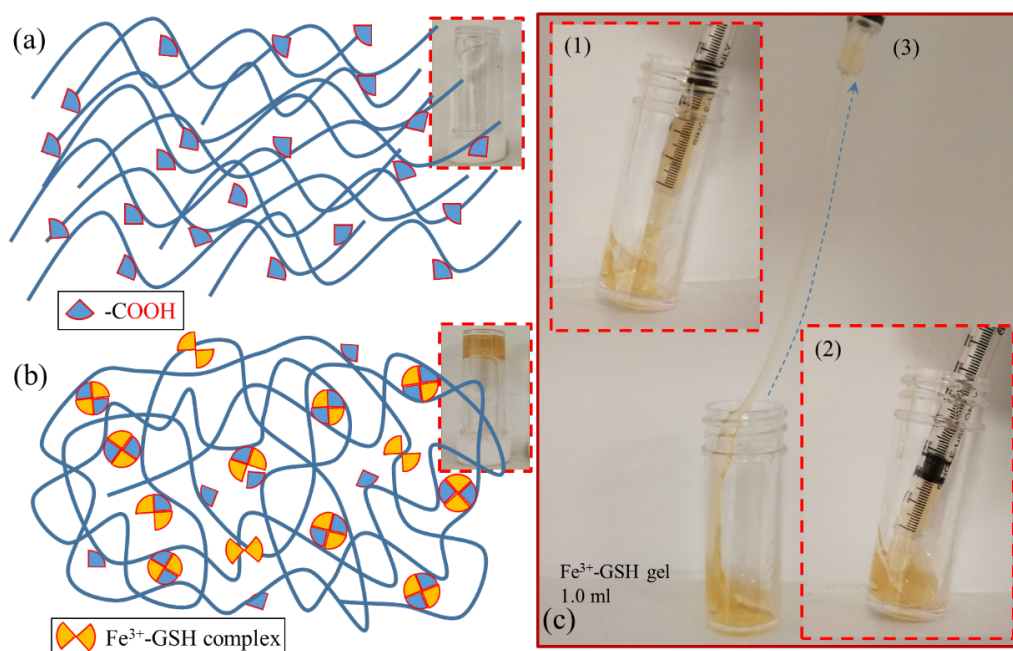
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**Fig. 5** Dynamic time-dependent viscosity measurement of the initial (blue) and degraded (grey)

277

Fe<sup>3+</sup>-GSH gel.

278 This dynamic reversible property confirms that Fe<sup>3+</sup>-GSH crosslinked gels are self-healing  
 279 materials. The profound degradability of Fe<sup>3+</sup>-GSH crosslinked hydrogel in aqueous solution  
 280 was confirmed by the decreased viscosity to 0.1-10 Pa·s after being incubated at 37 °C for 5  
 281 days (Fig. 5 grey). The transition from the HA solution to the Fe<sup>3+</sup>-GSH crosslinked self-healing  
 282 hydrogel was presented in Fig. 6a&b. Fig. 6c reveals the injectable property of this self-healing  
 283 hydrogel, and the fact that the material can be absorbed (step 1) by a syringe and then be  
 284 injected through the syringe tip (step 2), before undergoing extensive elongation (step 3).

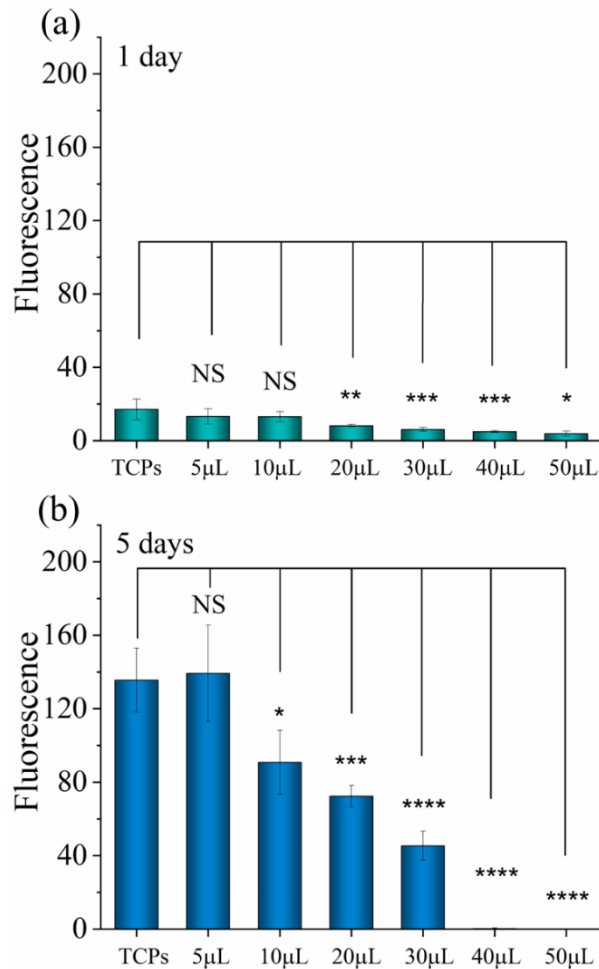


285  
 286 **Fig. 6** Illustration of Fe<sup>3+</sup>-GSH hydrogel formation. (a): Molecular configuration and physical  
 287 appearance of the HA solution; (b): Proposed coordination structure within, and physical appearance  
 288 of, the Fe<sup>3+</sup>-GSH hydrogel. (c): Macroscopic properties of Fe<sup>3+</sup>-GSH gel, being loaded up (step 1),  
 289 injected (step 2) and stretched (step 3).  
 290

291 We could also observe the sticky property of this self-healing hydrogel in step 3; in line with  
 292 previous viscosity analysis, the adhesive properties of HA were enhanced by Fe<sup>3+</sup>-GSH  
 293 induction. This feature is key to enable confined application and adhesion of the gel to  
 294 cartilage, aiming to stabilise the joint cavity and to reduce bone-to-bone friction, which is  
 295 essential to preserve the cartilage interface (Abubacker et al., 2018).

296 The dose of Fe<sup>3+</sup>-GSH crosslinked HA gel that is tolerated by ATDC 5 chondrocytes was then  
 297 determined *in vitro* via Alamar Blue assay (Fig. 7). As expected, the hydrogel reveals a dose-  
 298 dependent impact on cellular metabolic activity. At day 1, the lower dose (e.g. 5 and 10 µL)  
 299 of Fe<sup>3+</sup>-GSH crosslinked HA gel did not show significant effect compared to the case of the  
 300 TCPs control group (p > 0.05). However, the high dose groups (e.g. > 20 µL) significantly  
 301 reduced the metabolic activity of ATCD-5 cells compared to the control group (p ≤ 0.01, 0.001,  
 302 0.001, 0.05, respectively). Clearly, no significant difference in cellular activity was observed

303 following 1-day cell culture in either TCP or lower doses of Fe<sup>3+</sup>-GSH crosslinked hydrogel (with  
 304 both 5 μL and 10 μL dose). At day 5, only the 5 μL group was well tolerated (p > 0.05), but all  
 305 the other higher dose groups (e.g. > 10 μL) were significantly detrimental to the metabolic  
 306 activity of the cells compared to the control group (p ≤ 0.05, 0.001, 0.0001 respectively).



307

308 **Fig.7** ATDC 5 cells viability when growing with Fe<sup>3+</sup>-GSH gel after day 1 and 5. No significant  
 309 differences are labelled with “NS”. Significant differences are observed in each group with respect to  
 310 the TCPs group at the same time point (n=4). \* p < 0.05, \*\* p < 0.01, \*\*\* p < 0.001, \*\*\*\* p < 0.0001.

311

312 Furthermore, the increase in metabolic activity recorded from day 1 to day 5 in ATDC 5 cells  
 313 cultured with 5-30 μL hydrogel (**Table 1**) was similar to that measured in cells treated with the  
 314 TCPs control group (7.9 times). This observation indicates that decreased doses (e.g. ≤ 30 μL)  
 315 of Fe<sup>3+</sup>-GSH hydrogel did not affect the cell proliferation (e.g. cell doubling) in this time  
 316 window, in contrast to the case where higher doses (e.g. ≥ 40 μL) were applied. Given that the  
 317 initial cell seeding density (5,000 cells per well) was maintained across all hydrogel groups (5-  
 318 50 μL), the reduced cellular metabolic activity observed with increased gel volume (> 30 μL) is  
 319 likely attributed to the relatively small number of cells cultured with increased sample dosages.

320

321

**Table 1** Variation in ATDC 5 cellular activity over 5-day culture with varied hydrogel dosage.

Hydrogel dosage	Average cellular activity increase
0 $\mu$ l (TCP)*	7.9
5 $\mu$ L	10.5
10 $\mu$ L	7.0
20 $\mu$ L	8.8
30 $\mu$ L	7.4
40 $\mu$ L	0.1
50 $\mu$ L	0

322

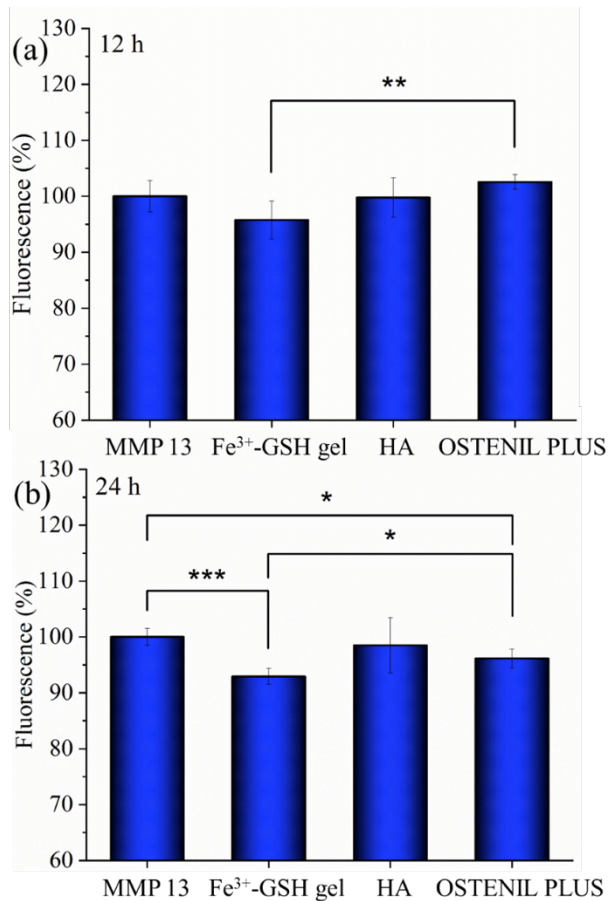
\* Cells cultured in hydrogel-free Tissue Culture Plastic (TCP).

323

324 This observation may suggest that the gels under 30  $\mu$ L dose were temporarily toxic after 1-  
 325 day. However, the proliferation of the remaining ATDC 5 cells was not affected, an explanation  
 326 which is supported by the optical microscope images of cells cultured for 1 (**Fig. S2**) and 5 days  
 327 (**Fig. S3**). In contrast, no cellular tolerability was observed in both 40 and 50  $\mu$ L hydrogel groups  
 328 over 5 days.

329 The capability of the Fe<sup>3+</sup>-GSH crosslinked hydrogels to inhibit proteolytic activity was then  
 330 assessed, whereby MMP-13 was selected as a well-known upregulated protease in late-stage  
 331 OA. By selecting MMP-13-supplemented aqueous solutions as a defined *in vitro* environment,  
 332 incubation of Fe<sup>3+</sup>-GSH hydrogel resulted in a reduction of MMP-13 activity after 12 hours  
 333 (95.7 $\pm$ 3.4 %). A significant reduction in MMP-13 activity (92.9 $\pm$ 1.4 %) was recorded after 24  
 334 hours, compared to the positive control group ( $p < 0.001$ ) (**Fig. 8**). On the other hand, no  
 335 significant activity difference was observed between MMP-13-supplemented solutions and  
 336 the same solutions following incubation with either soluble, complex-free GSH (103.1 $\pm$ 7.6 %)  
 337 (Gao et al., 2020) or native HA after 24 hours (98.5 $\pm$ 5.0 %). In OSTENIL® PLUS, no reduction in  
 338 MMP-13 activity was seen after 12 hours, but a significant reduction ( $p < 0.05$ ) in activity was  
 339 observed after 24 hours (96.1 $\pm$ 1.7 %), with respect to the pristine MMP-13 solution. A  
 340 comparison between the Fe<sup>3+</sup>-GSH crosslinked gel and OSTENIL® PLUS reveals that increased  
 341 MMP-13 inhibition occurred in the presence of the Fe<sup>3+</sup>-GSH crosslinked hydrogel after 12  
 342 hours ( $p < 0.01$ ), which was maintained after 24 hours ( $p < 0.05$ ). These results provide indirect  
 343 evidence that the hydrogel-induced MMP-13 inhibition was achieved via chelation of  
 344 respective iron sites with free sulfhydryl groups of active MMPs, rather than by complexation  
 345 of the free zinc sites of active MMPs (Liang et al., 2018) with either the hydrogel's or GSH's  
 346 sulfhydryl groups, on the one hand or HA's carboxylic groups on the other hand. These  
 347 observations support the key role played by the Fe<sup>3+</sup>-GSH complex in both hydrogel  
 348 crosslinking and MMP inhibition.

349



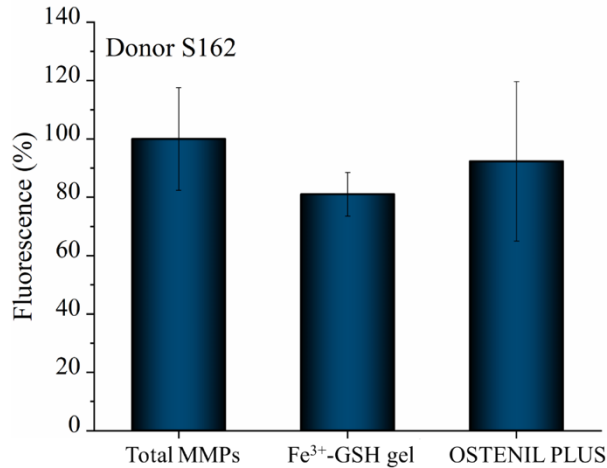
350

351 **Fig.8** Variation of MMP-13 activity in MMP-13-supplemented solutions after 12-hour (a) and 24-hour  
 352 (b) incubation with either the Fe<sup>3+</sup>-GSH crosslinked hydrogel, an HA solution or the OSTENIL® PLUS  
 353 commercial injection. Data are presented as Mean ± SD, statistical analysis was carried out between  
 354 each two groups and labelled as \*p≤ 0.05, \*\*p≤ 0.01, \*\*\*p≤ 0.001, otherwise means no significant  
 355 difference at p=0.05 level.

356

357 A sample of synovial fluid (S162) collected from patients with late-stage OA was used to  
 358 investigate the MMP-regulating capability of the Fe<sup>3+</sup>-GSH crosslinked gel in near-physiologic  
 359 conditions, and to further corroborate the previous findings obtained for hydrogel-mediated  
 360 MMP-13 inhibition in a defined *in vitro* environment, as the overall proteolytic activity,  
 361 including MMP-1, -2, -3, -7, -8, -9 and -13, were confirmed to have increased activity in  
 362 advanced OA (Yoshihara et al., 2000). **Fig. 9** reveals that lower overall MMP activity and  
 363 smaller standard deviations were observed for the Fe<sup>3+</sup>-GSH crosslinked gel (81.0±7.5 %)   
 364 compared to the native SF group (100.0±17.6 %), with a p-value of 0.0942. Although OSTENIL®  
 365 PLUS presented a lower average value of activity (92.3±27.3 %) compared to native SF  
 366 (p=0.6528), a larger standard deviation was recorded for this group versus both SF and the  
 367 Fe<sup>3+</sup>-GSH crosslinked gel.





368

369 **Fig.9** Variation of MMP activity recorded in a patient collected SF sample after 24-hour incubation  
 370 with either the Fe<sup>3+</sup>-GSH crosslinked hydrogel or the OSTENIL® PLUS commercial injection (n=4). The  
 371 SF sample was collected from a patient (donor S162) with late-stage OA.

372

373 The results obtained with the clinical SF sample in the absence of MMP activating reagents,  
 374 i.e. APMA, therefore confirm the new MMP inhibition functionality introduced in the Fe<sup>3+</sup>-GSH  
 375 crosslinked hydrogel. These results therefore support the use of this material as both a  
 376 mechanically-competent hydrogel and as a mediator of MMP regulation for OA therapy. The  
 377 confirmation of hydrogel performance with patient collected samples also lay down new  
 378 possibilities on the use of human synovial fluid for the preclinical evaluation of medical devices  
 379 intended for osteoarthritis management, yet minimising reliance on animal testing.

380

### 381 **Conclusions**

382 A drug-free Fe<sup>3+</sup>-GSH crosslinked injectable hydrogel was prepared with integrated self-  
 383 healing and MMP inhibition functionalities. The coordination mechanism to yield the hydrogel  
 384 was confirmed by shear frequency sweep tests, which revealed a storage modulus more than  
 385 ten times higher than the loss modulus. <sup>57</sup>Fe Mössbauer spectroscopy revealed that Fe was  
 386 present in the hydrogel as octahedrally-coordinated Fe<sup>3+</sup>, so that risks of Fe<sup>2+</sup>-mediated ROS  
 387 generation and ROS-mediated toxicity were minimised, supporting the hydrogel applicability  
 388 in biological environment. The hydrogel could hold up to 300% shear strain and presented a  
 389 stable complex viscosity (37-42 → 12-16 Pa·s) after 10 cycling tests from low to high strain. *In*  
 390 *vitro*, the gel proved to be well tolerated by ATDC 5 chondrocytes and to support cell  
 391 proliferation during a five day-culture. Furthermore, the gel demonstrated the inhibition of  
 392 MMP activity after 24 hour-incubation in both an MMP-13-supplemented aqueous solution  
 393 and a patient collected sample of synovial fluid, in light of the metal-coordinating reaction  
 394 between thiol-complexed iron(III) and free sulfhydryl groups of active MMPs is exploited to  
 395 induce MMP inhibition. These results therefore demonstrate that the hydrogel's

396 biomechanical competence was successfully integrated with drug-free MMP regulation  
397 capability. The simple material design, together with the hydrogel's injectability, and  
398 biochemical and self-healing functionalities support further development of this system for  
399 drug-free OA therapies.

400

#### 401 **Acknowledgements**

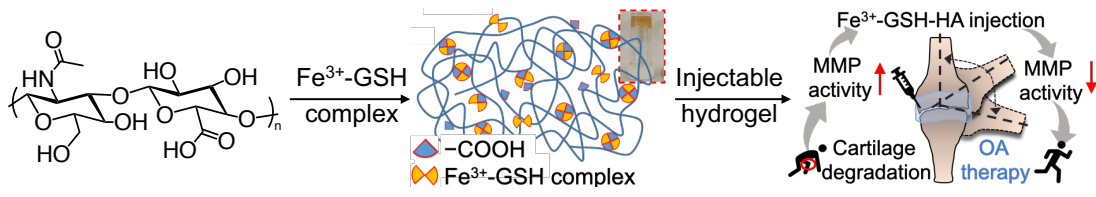
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404 Immortality.



405 **Graphical abstract**

406

407



408

409 Fe-coordination to HA in aqueous environment generates an injectable, self-healing and  
410 biomechanically viable gel and enables factor-free regulation of matrix metalloproteinases.

411

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