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## Phase stabilities of MgCO<sub>3</sub> and MgCO<sub>3</sub>-II studied by Raman spectroscopy, X-ray diffraction and DFT

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#### Abstract

Carbonates are the major hosts of carbon on Earth's surface and their fate during subduction needs to be known in order to understand the deep carbon cycle. Magnesite (MgCO<sub>3</sub>) is thought to be an important phase participating in deep Earth processes, but its phase stability is still a matter of debate for the conditions prevalent in the lowest part of the mantle and at the core mantle boundary. Here, we have studied the phase relations and stabilities of MgCO<sub>3</sub>, at these P, T-conditions using Raman spectroscopy at high pressures (~148 GPa) and after heating to high temperatures (~3600 K) in laser-heated diamond anvil cell experiments (LH-DAC). The experimental Raman experiments were supplemented by X-ray powder diffraction data, obtained at a pressure of 110 GPa. Density functional theory-based model calculations were used to compute Raman spectra for several MgCO<sub>3</sub> high pressure polymorphs, thus allowing an unambiguous assignment of Raman modes. By combining the experimental observations with the DFT-results, we constrain the phase stability field of MgCO<sub>3</sub> (magnesite) with respect to the high pressure polymorph, MgCO<sub>3</sub>-II. We further confirm that Fe-free MgCO<sub>3</sub>-II is a tetracarbonate with monoclinic symmetry (space group C2/m), which is stable over the entire P, T-range of the Earth's lowermost mantle geotherm.

#### I. INTRODUCTION

The Earth's mantle is believed to experience a carbon influx on the order of teragrams per year, 2 due to the subduction of oceanic lithosphere [1, 2]. Carbonate minerals are considered to con-3 stitute the major carbon source during subduction processes [2–6]. In the mantle, decomposition 4 reactions of carbonates and reactions with silicates are thought to lead to the formation of other 5 carbon containing phases [7-10]. As silicates can only incorporate very minor amounts of carbon 6 at mantle conditions [11, 12], the most prevalent carbon bearing phases are believed to be diamond 7 [13], high P, T phases of CO<sub>2</sub> [14], or metal carbides [15, 16]. However, carbonates may survive 8 in the Earth's mantle in cold oxidized subducting slabs [17, 18] some of which might penetrate 9 into the lowermost mantle [19]. This model is supported by the presence of carbonate inclusions 10  $(CaCO_3, MgCO_3, CaMg(CO_3)_2)$  in mantle xenoliths and in super deep diamonds [20–25]. 11 Magnesite (MgCO<sub>3</sub>) is thought to be stable under P, T-conditions of the Earth's mantle [9, 12

 <sup>13</sup> 26, 27]. While other carbonates undergo several phase transitions (e.g. calcite (CaCO<sub>3</sub>)) [28],
 <sup>14</sup> or decompose at *P*, *T*-conditions of the Earth's mantle (e.g. dolomite (CaMg(CO<sub>3</sub>)<sub>2</sub>) and siderite <sup>\*</sup> binck@kristall.uni-frankfurt.de (FeCO<sub>3</sub>)) [29, 30], magnesite remains stable up to at least 80 GPa and 3000 K [26, 31, 32]. Studies of reactions of MgCO<sub>3</sub> with SiO<sub>2</sub> implied that high pressure, high temperature polymorphs of MgCO<sub>3</sub> might coexist with silicates in super cold slabs in the lower mantle [9]. Model calculations of reactions of MgCO<sub>3</sub>/CaCO<sub>3</sub> with MgO/MgSiO<sub>3</sub> under *P*, *T*-conditions of the Earth's lower mantle imply the possible existence of oxidized carbon in the form of MgCO<sub>3</sub> in the absence of iron [33, 34].

<sup>21</sup> Based on in-situ powder X-ray diffraction, Isshiki *et al.* [26] showed that magnesite ( $R\bar{3}c$ , <sup>22</sup> Z = 6) undergoes a phase transition to 'magnesite-II' at deep lower mantle conditions (>115 GPa <sup>23</sup> and 2200 K). Shortly after these findings, DFT-based predictions reported the possibility of the <sup>24</sup> existence of very high pressure (e.g., at pressures >80 GPa) carbonate structures that are charac-<sup>25</sup> terized by  $sp^3$ -hybridized bonding environments within CO<sub>4</sub><sup>4-</sup> tetrahedrons instead of triangular <sup>26</sup>  $sp^2$ -hybridized CO<sub>3</sub><sup>3-</sup> groups [35–39]. Within the last decade, the existence of so-called 'tetracar-<sup>27</sup> bonates' has been confirmed by several experimental studies [30, 40–45].

Several theoretical structures have been reported for  $sp^3$ -MgCO<sub>3</sub> polymorphs [35, 38, 39]. 28 Combined results from DFT and X-ray powder diffraction suggested a MgCO<sub>3</sub> structure with 29 space group C2/m and Z = 12 formula units to be the most stable phase at pressures between 30 82 - 138.1 GPa [38]. However, a comparison of powder X-ray diffraction data of the C2/m-phase 31 with powder patterns published by Isshiki et al. [26] gave an unsatisfactory match. Further exper-32 imental evidence for the formation of the C2/m-phase around ~82 GPa was given by Boulard 33 et al. [40] and Maeda et al. [9]. However, their data analyses relied on Le Bail fits only. While 34 X-ray powder diffraction data were fitted with the C2/m-phase between 85–152 GPa in the study 35 by Maeda et al. [9], a lower symmetric structure with space group  $P2_1/c$  was proposed to be better 36 suited for fitting X-ray diffraction patterns by Boulard et al. [40] at 82 GPa. 37

Using a sample with an initial composition of  $Fe_{0.15}Mg_{0.85}CO_3$  at ambient conditions, Chariton *et al.* [46] have been able to solve the crystal structure of  $Fe_{0.4}Mg_{2.6}C_3O_9$  at 98 GPa and after heating to 2500 K from single-crystal data. This structure was shown to be identical to the C2/mphase proposed by Oganov *et al.* [38].

While the existence of the C2/m-phase seems now to be established for pressures <100 GPa and temperatures <2500 K [46], significant gaps and inconsistencies remain in our understanding of the high P, T behaviour of Fe-free MgCO<sub>3</sub> [9, 26, 38, 39]. Currently, all experimental data above 100 GPa rely on poorly constrained indexing of X-ray powder diffraction patterns [9, 26, 38], while theoretical calculations suggest several possible low energy structures [35, 38, 39]. The strength of vibrational spectroscopy for the detection of high pressure, high temperature phases in the diamond anvil cell has recently been exploited for high pressure studies on different carbonates [28, 29, 42, 45, 47–49]. Raman spectroscopy is a very sensitive method, which may provide additional structural information for the particular high pressure, high temperature polymorphs of MgCO<sub>3</sub> thus complementing previous X-ray diffraction studies. However, studies that have been employing Raman spectroscopy on MgCO<sub>3</sub> are limited to pressures <55 GPa [32, 47, 50, 51].</p>

In this study, we have investigated the MgCO<sub>3</sub>-system in the entire pressure and temperature range reaching to the uppermost part of the Earth's outer core corresponding to pressures up to  $\sim$ 148 GPa and temperatures up to  $\sim$ 3600 K. We combined Raman spectroscopy in the laser heated diamond anvil cell with DFT-based model calculations. Supplementary X-ray powder diffraction data were obtained at high pressures, which support our observations further.

#### 58 II. METHODS

Syntheses, preparations of experiments, sample characterizations at ambient conditions, as well
 as Raman spectroscopy in the LH-DAC have been carried out at the Institute of Geosciences at
 the Goethe Universität Frankfurt, Germany. The high pressure X-ray diffraction experiments have
 been carried out at P02.2 at PETRA III (DESY) in Hamburg, Germany.

#### 63 A. Synthesis

Single crystals of magnesite were synthesized according to the method described by Ni et al. 64 [52]. All chemicals (magnesium acetate tetrahydrate, hexamethylenetetramine and sodium sul-65 fate) were analytical grade reagents purchased from Merck KGaA (Darmstadt) and used as re-66 ceived without further purification. 3 mmol magnesium acetate, 3 mmol hexamethylenetetramine, 67 and 0.6 g sodium sulfate were dissolved in 40 ml bidistilled water. After stirring for 30 min, the 68 obtained transparent solution was transferred into a 60 ml Teflon cup, which was filled to 60 %69 of its volume. Subsequently, the cup was put into a stainless steel autoclave and sealed tightly. 70 After reaction at  $160^{\circ}$  for 48 h, the autoclave was slowly cooled down ( $160 - 100 \,^{\circ}$ C in 48 h, then 71 100 – 25 °C in 12 h). The precipitate was filtered under vacuum, washed with distilled water re-72 peatedly and dried at 60 °C in an oven. The slow cooling allowed us to grow crystals with edge 73 lengths up to 60 µm. 74

#### 75 **B.** Characterization

The phase purity of the synthesized batch of single crystals has been characterized at ambient 76 conditions by X-ray powder diffraction. Therefore, we separated the majority of single crystals 77 from the synthesized batch and grounded a fine sample powder in an agate mortar. The grounded 78 sample powder was measured using an X'Pert Pro diffractometer equipped with a linear position-79 sensitive detector from PANanytical (PIXcel<sup>3D</sup>) and a Johannson monochromator (Ge 111) using 80  $CuK\alpha_1$  radiation ( $\lambda = 1.5418$  Å) generated at 40 kV and 30 mA. A Rietveld refinement was car-81 ried out using the GSAS-II software [53] and a reported structure of magnesite [54] as a starting 82 model (see appendix Fig. 1 and Tab. 1-2). The refined lattice parameters were a = b = 4.6375(1) Å, 83 c = 15.0268(2) Å and V = 279.876(1) Å<sup>3</sup>, which are consistent with the parameters of the estab-84 lished structure of magnesite [54]. Additionally, Raman spectroscopy was carried out at ambient 85 conditions on single crystals. The spectra confirmed the excellent sample quality (Fig. 1). 86

#### <sup>87</sup> C. Preparation of high-pressure, high-temperature experiments

High-pressure, high-temperature experiments were carried out using Boehler-Almax diamond 88 anvil cells (DAC) [55]. Type Ia and IIas diamonds with low birefringence and ultra low fluores-89 cence having either regular culets (200, 250 and 350 µm diameter), or beveled culets (9° bevel, 90 100 µm inner- and 220 µm outer diameter) were inserted in WC seats. The opening angles of the 91 cells were  $48^{\circ}$  or  $70^{\circ}$ . Depending on the culet size, sample chambers of  $45 - 160 \,\mu\text{m}$  in diameter 92 were laser drilled in Re gaskets pre-indented to  $\sim 40 \,\mu$ m. Before loading, the sample material was 93 dried at 150 °C for 24 h in an oven. Immediately after drying, three diamond anvil cells were 94 loaded with single crystals and one cell with a compacted powder. The single crystals had edge 95 lenghts of  $\sim 15 \,\mu$ m. All cells were loaded with a ruby pressure marker. Neon served as a pressure 96 transmitting medium for the single crystal cells, which was loaded by a custom built gas loading 97 system. In order to obtain X-ray powder diffraction data without any interference by the pressure 98 transmitting medium, no Neon gas was used for the DAC loaded with the compacted powder sam-99 ple. No further thermal insulation was added to the loadings, in order to prevent possible chemical 100 reactions. 101

#### 102 **D. Raman spectroscopy in the LH-DAC**

Raman spectra were measured in 0.5 - 4 GPa steps upon compression and decompression cov-103 ering a range between ambient pressure and ~148 GPa. A frequency doubled 532.14 nm Nd:YAG 104 Oxxius laser (LCX-532S) was focused on the sample with a spot size of 6 µm. Spectra were col-105 lected in backscattering geometry, using a grating spectrometer (Acton, SP-2356) equipped with 106 a CCD detector (Pixis 256E) and a microscope objective (Mitutoyo). The spectral resolution of 107 the spectrometer is 3 cm<sup>-1</sup> [28]. The laser power was set to 430 mW and spectra were collected 108 for 50 s in a frequency window of  $100 - 1500 \text{ cm}^{-1}$ , using a grating of 1800 grooves/mm. The 109 estimated laser power on the sample was around  $\sim$ 350 mW. For measurements up to 95 GPa, 110 the pressure was determined before and after the Raman measurement, using the ruby reference 11 scales for non- [56] and quasi-hydrostatic conditions [57]. Pressures were further determined for 112  $P \ge 45$  GPa using the diamond edge reference scale [58]. For P > 95 GPa only the diamond 113 edge reference scale was employed. According to Dewaele et al. [59], the uncertainty in pressures 114 determined by Mao et al. [56] increases from 0.05 GPa at 1 GPa up to 2 GPa at 150 GPa. The 115 accuracy of our pressure determination by the ruby and diamond reference scales was  $\leq 2$  GPa, 116 while pressure gradients may have caused uncertainties up to 4 GPa during and after laser heating. 117

The sample was heated from both sides with a pulsed CO<sub>2</sub> laser (Diamond K-250 from Coher-118 ent,  $\lambda = 10.6 \,\mu\text{m}$ ) [28]. For the spectroradiometric temperature determination we used the same 119 set-up as for the Raman measurements, while the grating was set to 150 grooves/mm. In order to 120 achieve coupling of the CO<sub>2</sub> heating laser from both sides of the sample, the laser power was typ-121 ically set in a range between 1 - 6 W, depending on the pressure and the loading of the diamond 122 anvil cell. The heating laser was focused on the sample so that the diameter of the heated area 123 was around 25 µm, which nearly covered all of the single crystals. However, due to the poor to 124 moderate coupling of the laser with the samples, heating was inhomogeneous. Hence, we moved 125 the heating laser across the sample, while typically heating for about 5 min per position. The po-126 sition of the Raman laser with respect to the heated areas on the sample was controlled using an 127 optical camera. The thermal emission of the sample, as well as the Raman signal was measured 128 with a spatial resolution of around  $5-6 \mu m$ , i.e. the areas for the measurements were significantly 129 smaller than the heating spots. The temperatures during laser heating were determined by the two-130 colour pyrometer method, employing Planck and Wien fits [60]. We assume a typical uncertainty 131 associated with radiometric temperature measurements in LH-DACs of  $\sim 10$  %. 132

#### 133 E. High-pressure X-ray diffraction

High pressure X-ray powder diffraction measurements have been carried out at the extreme 134 conditions beamline P02.2 at PETRA III (Desy, Hamburg, Germany). X-ray diffraction data were 135 collected for a MgCO<sub>3</sub> powder sample, which at first was pressurized to 110 GPa in a diamond 136 anvil cell and temperature quenched after laser-heating to  $\sim$ 2500 K. The diffraction patterns were 137 acquired using a wavelength of 0.2898 Å, a beam focused to  $8 \times 3 \,\mu\text{m}^2$  (full width at half max-138 imum), compound reflective lenses, and a Perkin Elmer XRD 1621 flat-panel detector. A grid 139 of  $5 \times 5$  points with spacings of 2 µm between each point was measured. The DAC was rotated 140 by  $\pm 10^{\circ}$  with an exposure time of 40 s for every data point collection. The sample-to-detector 141 distance of 402.78 mm and the wavelength were determined employing a  $CeO_2$  reference sample. 142 The diffraction data were processed with the Dioptas software [61]. Rietveld refinements were car-143 ried out using the GSAS-II software [53] and the structure model based on an earlier description 144 of the C2/m-phase [38]. 145

#### 146 **F. Density functional theory**

In order to obtain theoretical Raman spectra, density functional perturbation theory (DFPT) 147 calculations were performed employing the CASTEP code [62]. The code is an implementation 148 of Kohn–Sham DFT based on a plane wave basis set in conjunction with pseudopotentials. The 149 plane wave basis set allows to achieve numerically converged results in a straightforward manner, 150 as the convergence is controlled by a single adjustable parameter, the plane wave cut-off, which we 151 set to 1020 eV. For calculations for pressures <100 GPa, the norm-conserving pseudopotentials 152 were generated 'on the fly' from the information provided in the CASTEP data base. These pseu-153 dopotentials have been tested extensively for accuracy and transferability [63]. For calculations at 154 pressures >100 GPa norm-conserving pseudopotentials with smaller core radii were constructed, 155 in order to avoid overlap of ionic cores. The descriptors of these pseudopotentials are given in 156 the appendix. All calculations employed the GGA-PBE exchange-correlation functional [64]. The 157 Brillouin zone integrals were performed using Monkhorst–Pack grids [65] with spacings between 158 grid points of less than 0.037 Å<sup>-1</sup>. Geometry optimizations were defined as being converged 159 when the energy change between iterations was  $<0.5 \times 10^{-6}$  eV/atom, the maximal residual force 160 was < 0.01 eV/Å, and the maximal residual stress was < 0.02 GPa. Phonon frequencies were ob-161

tained from density functional perturbation theory (DFPT) calculations. Raman intensities were computed using DFPT in the 2n + 1 theorem approach [66].

#### 164 III. RESULTS AND DISCUSSION

### A. MgCO<sub>3</sub> (magnesite) at high-pressures and high temperatures measured by Raman spectroscopy

The characteristic Raman phonon frequencies of MgCO<sub>3</sub> (magnesite) were measured in four 167 different experimental runs, covering a pressure range between ambient pressure and  $\sim 107$  GPa 168 (Fig. 1 and 2). According to group theory, the following Raman and infrared active modes are 169 expected for magnesite at ambient conditions:  $\Gamma = A_{1g}(R) + 3A_{2u}(IR) + 5E_u(IR) + 4E_g(R).$  All 170 Raman active modes were observed at ambient conditions. Starting at ambient pressure with a fre-171 quency of 1444 cm<sup>-1</sup>, the  $E_g(\nu_3)$  mode is obscured by the first order Raman mode of the diamond 172 anvils in the DAC experiments. The frequencies of the symmetric stretching mode and the sym-173 metric in-plane bend allow a straightforward identification of the  $CO_3^{2-}$ -groups [50, 67, 68]. We 174 observed the  $A_{1g}(\nu_1)$  and  $E_g(\nu_4)$  modes up to ~107 GPa in a range between 1095 – 1304 cm<sup>-1</sup> 175 and 740 – 893 cm<sup>-1</sup> respectively. The two low frequency  $E_g$  modes ( $\nu = 213$  and 331 cm<sup>-1</sup> at 176 ambient pressure) were observed up to 45.5 GPa. At higher pressures these modes displayed a 177 significant broadening, or disappeared completely. A similar behaviour for the low frequency  $E_g$ 178 modes was reported by Williams et al. [50] and Gillet [51], who observed the 213 cm<sup>-1</sup> Raman 179 band at pressures between 13 - 20 GPa and the 331 cm<sup>-1</sup> Raman band up to 26 GPa. 180

We were able to detect all Raman active modes that are observable in a diamond anvil cell up 181 to 87.8 GPa by carefully thermally annealing the single crystal at nearly each pressure step up to 182 82 GPa and measuring the Raman signal after temperature quenching (Fig. 1). The temperatures 183 during the thermal annealing by the laser were estimated to be <1000 K, since no visible thermal 184 radiation was observed. An offset for especially the low frequency Raman modes may be observed 185 for some of the high pressure Raman spectra due to the non-hydrostatic pressure on the sample. 186 These effects disappeared after heating was applied and hydrostatic conditions on the sample were 187 increased. 188



FIG. 1. High pressure Raman spectra of MgCO<sub>3</sub> (magnesite) from ambient pressure to  $\sim$ 88 GPa as obtained for a single run. For most of the pressure steps between 15.9 and 82.1 GPa, the crystal was slightly annealed at temperatures <1000 K. All Raman measurements were conducted after quenching the sample to ambient temperature.



FIG. 2. Pressure dependence of the characteristic Raman modes of magnesite (black symbols) and MgCO<sub>3</sub>-II (yellow symbols). Triangles and circles correspond to single crystal and powder samples respectively. Open symbols correspond to data obtained under cold-decompression. DFT-calculated phonon frequencies of MgCO<sub>3</sub>-II at 85 and 140 GPa are shown for comparison (green diamond symbols). The black dashed line marks the phase boundary of the MgCO<sub>3</sub> polymorphs at 85 GPa. Data points were fitted using linear or quadratic fits. (color online)

# B. Phase transition of MgCO<sub>3</sub> (magnesite) to MgCO<sub>3</sub>-II identified by Raman spectroscopy, XRD and density functional theory

At 83 GPa and after heating to  $\sim$ 2400 K, our Raman spectra still indicate the presence of 191 MgCO<sub>3</sub> (magnesite) at these conditions, while a drastic change in the spectra is observed at around 192 85 GPa and after heating to  $\sim$ 2000 K, where new characteristic Raman bands were observed be-193 sides those of MgCO<sub>3</sub> (magnesite) (appendix Fig. 2). At 87.9 GPa and after heating to maximum 194 temperatures of  $\sim$ 3100 K, Raman spectra yielded at least 23 strong intense modes covering a fre-195 quency range between 250 and 1250  $\text{cm}^{-1}$  (Fig. 3). These changes are due to a phase transition 196 from MgCO<sub>3</sub> (magnesite) to a second phase, which we have labeled MgCO<sub>3</sub>-II here. We com-197 puted the Raman spectra for the monoclinic C2/m-phase [38, 46] and for a hypothetical triclinic 198  $P\bar{1}$ -phase [39], which has been suggested to be stable between 85 - 101 GPa. The experimental 199 Raman spectrum at 87.9 GPa can very satisfactorily be explained by a combination of the theoret-200 ical Raman spectra of magnesite and the C2/m-phase, while no indication for the  $P\bar{1}$ -phase was 201 found (Fig. 3). 202

We continued measuring Raman spectra up to 148 GPa (Fig. 2 and 4). For most of the 203 pressure steps, the sample was heated up to maximum temperatures of  $\sim$ 3600 K prior to the 204 measurement, in order to achieve a hydrostatic pressure distribution on the sample. A com-205 parison of theoretical spectra with experimental data at 115 and 140 GPa shows an excellent 206 match of frequencies and intensities (Fig. 5). Characteristic modes of MgCO<sub>3</sub> (magnesite) are 207 no longer observed in the experimental spectrum at these conditions. According to group the-208 ory, the irreducible representations of the C2/m-phase for the Raman and infrared-modes are 209  $\Gamma = 25 A_g(R) + 18 A_u(IR) + 20 B_g(R) + 24 B_u(IR)$ . The tetrahedral  $CO_4^{4-}$  groups of the C2/m-210 phase are polymerized and form corner sharing  $C_3O_9^{6-}$  rings [38]. Between 85 and 148 GPa, 211 characteristic vibrations of those rings are frequencies in the ranges 1023 - 1095 cm<sup>-1</sup>, 1026 -212 1128 cm<sup>-1</sup>, 1050 – 1146 cm<sup>-1</sup>, and 1065 – 1173 cm<sup>-1</sup>, respectively. Further characteristic fea-213 tures are the intense  $A_g$  and  $B_g$  modes at 733 – 781 cm<sup>-1</sup> and 444 – 463 cm<sup>-1</sup>, respectively. Both 214 modes are due to relative movements between the  $C_3 O_9^{6-}$  rings and the Mg<sup>2+</sup> cations. Two modes 215 in MgCO<sub>3</sub>-II at 189 cm<sup>-1</sup> ( $\leq$ 105 GPa) and 1050 cm<sup>-1</sup> ( $\geq$ 120 GPa) have no correspondence in the 216 DFT-calculations. The origin of these modes is currently unexplained. 217

There have been suggestions that the C2/m polymorph transforms into another phase at  $P \ge 138$  GPa. For this phase, structures with space group  $P2_1$  [38], or  $P2_12_12_1$  [39] have been

<sup>220</sup> suggested. Theoretical Raman spectra of the hypothetical  $P2_1$ -phase [38] at 120 and at 140 GPa <sup>221</sup> were computed and compared to our experimental observations (see appendix Fig. 3). Although <sup>222</sup> the comparison between experimental and theoretical spectra exhibits some similarities at higher <sup>223</sup> frequencies, significant differences are observed in the lower frequency range. Hence, we con-<sup>224</sup> clude that this phase has not been formed in our experiments. Computation of the Raman spectra <sup>225</sup> of the  $P2_12_12_1$ -phase [39] was beyond the available computation resources.



FIG. 3. Experimental Raman spectra of MgCO<sub>3</sub> (magnesite) and a mixture of MgCO<sub>3</sub> (magnesite) and MgCO<sub>3</sub>-II (C2/m-phase [38]) are shown in black. DFT-calculated Raman spectra of magnesite, the C2/m-phase and the  $P\bar{1}$ -phase [39] are shown in green. DFT frequencies were calculated with a FWHM broadening of 5 cm<sup>-1</sup>. The calculated frequencies were multiplied by a scaling factor of 1.02. The characteristic Raman modes of MgCO<sub>3</sub> (magnesite) at 85 GPa are indicated by dashed grey lines. (color online)

Our combined results from Raman spectroscopy and DFT-calculations are supported by synchrotron X-ray powder diffraction measurements, which have been conducted on MgCO<sub>3</sub> powder at 110 GPa and after heating to  $\sim$ 2500 K. We were able to successfully carry out a Rietveld refinement, which allowed the identification of the C2/m-phase (Fig. 6 and appendix Tab. 1-2). Since



FIG. 4. Raman spectra of MgCO<sub>3</sub>-II (C2/m-phase) [38] in a pressure range from 87 to 148 GPa for a single run. For most of the pressure steps, the sample was heated to maximum temperatures between 3000 and 3600 K and measured after quenching to ambient temperature. For the present data, the grating of the Raman spectrometer was centered towards higher Raman shifts, which resulted in obscuration of the CO<sub>3</sub>-stretching mode ( $\nu = 1278 \text{ cm}^{-1}$ ) by the diamond anvils.

Rietveld refinements are usually hard to conduct for high pressure data especially after heating, 230 intensities of a grid of 25 diffraction images were summed at each particular 2 theta angle, in order 231 to achieve an accurate ratio of intensities for the refinement (appendix Fig. 4-5). Profile parameters 232 including scaling, Gaussian and Lorentzian terms, as well as the unit cell were initially refined. 233 The background was manually fitted, using a Chebychev function with twenty terms. In order to 234 reduce the number of parameters, we constrained the isotropic atomic displacement parameters 235 to be the same for symmetrically independent atoms of the same chemical species. Further, we 236 employed restraints on the atomic distances and refined the atomic positions. Refined lattice pa-237 rameters were a = 8.117(4) Å, b = 6.510(1) Å, c = 6.911(2) Å,  $\beta = 103.858(9)^{\circ}$ , V = 354.64(5) Å<sup>3</sup>. 238 Our refined structural model is in excellent agreement with the structure of the C2/m-phase re-239



FIG. 5. Comparison between Raman bands of experimental data (black) and theoretical data (green) of MgCO<sub>3</sub>-II (C2/m-phase [38]) at 115 and 140 GPa. Experimental data have been obtained on the temperature quenched sample after heating to ~3000 and ~3200 K, respectively. The experimental spectrum reveals the complete transformation of MgCO<sub>3</sub> (magnesite) to MgCO<sub>3</sub>-II at these conditions. DFT frequencies were calculated with a FWHM broadening of 5 cm<sup>-1</sup>. The calculated frequencies were multiplied by a scaling factor of 1.02. (color online)

ported by Oganov *et al.* [38] and lattice parameters are well in agreement with those from Le Bail
refinements for the same pressure range as reported by Maeda *et al.* [9].

Raman spectra of MgCO<sub>3</sub> (magnesite) and MgCO<sub>3</sub>-II (C2/m-phase) were measured upon pressure release (Fig. 7 and 2). The pressure unexpectedly dropped during the first step of the release from 85 GPa down to 64 GPa. The pressure was then released in small steps down to ambient conditions, while Raman spectra were measured. During pressure release, the characteristic Raman bands of MgCO<sub>3</sub> (magnesite) and those of MgCO<sub>3</sub>-II (C2/m-phase) could concomitantly be observed in the pressure range between 39.6 – 85 GPa. At lower pressures, only the A<sub>1g</sub>( $\nu_1$ ) and E<sub>g</sub>( $\nu_4$ ) modes of MgCO<sub>3</sub> (magnesite) remained. The low frequency E<sub>g</sub> modes reappeared at



FIG. 6. Rietveld refinement of diffraction data collected at 110 GPa. The structural model by Oganov *et al.* [38] for MgCO<sub>3</sub>-II (C2/m-phase) was used for the refinement ( $\lambda = 0.2898$  Å). Refined structural parameters are listed in the appendix (Tab. 1-2). (color online)

around 5 GPa. MgCO<sub>3</sub> (magnesite) was eventually recovered at ambient conditions, unequivocally
showing that no decomposition occured after laser heating at high pressures.



FIG. 7. Raman spectra of MgCO<sub>3</sub> (magnesite) + MgCO<sub>3</sub>-II (C2/m) obtained during decompression down to ambient conditions. Frequencies of the  $(C_3O_9)^{6-}$  ring bending (light grey) are observed down to ~40 GPa. The concomitant presence of a characteristic mode ( $CO_3^{2-}$ -stretching mode) of MgCO<sub>3</sub> (magnesite) indicated in dark grey shows the coexistence of magnesite and metastable MgCO<sub>3</sub>-II. MgCO<sub>3</sub> (magnesite) is recovered at low pressures and ambient conditions. (color online)

#### **C.** Phase diagram of MgCO<sub>3</sub>

Based on the results from this study in conjunction with data from the literature [9, 26, 31, 32, 252 40, 69] we revise the phase diagram of MgCO<sub>3</sub> (Fig. 8). A first phase diagram of MgCO<sub>3</sub> was 253 given by Isshiki et al. [26] in which phase boundaries were drawn for the magnesite to 'magnesite-254 II' phase transition and the decomposition behaviour of MgCO<sub>3</sub>, determined by Fiquet *et al.* [31], 255 was extended to higher P, T-conditions. Another phase diagram at lower P, T-conditions was 256 reported by Solopova et al. [32], where the melting- and decomposition behavior of magnesite 257 was described. The slope of the melting curve was discussed to be less steep than that reported by 258 Katsura and Ito [69]. Also, for pressures below 50 GPa the decomposition of MgCO<sub>3</sub> (magnesite) 259 was shown to happen at significantly lower temperatures than reported by Fiquet et al. [31], while 260 at higher pressures both curves are approaching one another. We combined the contents of the 261 phase diagrams from both studies [26, 32] and added data points obtained in this study (large filled 262 circles in Fig. 8) along with datapoints of the available high pressure high temperature studies 263 on MgCO<sub>3</sub> beyond 50 GPa [9, 26, 32, 40]. It should be borne in mind that our data and those 264 by Solopova et al. [32] were obtained after temperature quenching and at high pressures, while 265 all other data from the literature were measured in-situ at high pressures and high temperatures 266 [9, 26, 40]. 267

In the present phase diagram, data points of MgCO<sub>3</sub> (magnesite) are indicated by black circles 268 (Fig. 8). According to Solopova et al. [32] our data points of heated MgCO<sub>3</sub> (magnesite) fall 269 within the conditions for magnesite as a solid phase. Our data points of MgCO<sub>3</sub>-II (yellow circles) 270 and MgCO<sub>3</sub>-II associated with MgCO<sub>3</sub> (black-yellow cirlces) are in very good agreement with 271 the observations by Maeda et al. [9] and Boulard et al. [40]. Hence, we present the MgCO<sub>3</sub>-272 MgCO<sub>3</sub>-II phase boundary (yellow dashed line) with a negative slope of dT/dP = -940 K GPa<sup>-1</sup> 273 towards higher temperatures. Due to a lack of data, the phase boundary is only drawn up to the 274 maximum available P, T-conditions [40]. Our data points reveal that no decomposition is to be 275 expected for MgCO<sub>3</sub>-II at P, T-conditions suggested by the decompositon lines of the previous 276 phase diagrams [26, 32] (see appendix Fig. 6). As a consequence, the decomposition boundary 277 of MgCO<sub>3</sub>-II (C2/m-phase) by Isshiki *et al.* [26] is not shown in the revised phase diagram (Fig. 278 8). Further, MgCO<sub>3</sub> (magnesite) is observed to transform into MgCO<sub>3</sub>-II rather than decomposing 279 into MgO + C +  $O_2$  at pressures and temperatures above 85 GPa and  $\sim$ 3000 K, respectively. 280

In our experiments, no other phase transition was found for pressures and temperatures up to

<sup>282</sup> 148 GPa and  $\sim$ 3600 K. Our conclusion is consistent with the interpretation of Le Bail fits by <sup>283</sup> Boulard *et al.* [40] and Maeda *et al.* [9] who conducted in-situ powder X-ray diffraction in the <sup>284</sup> same *P*, *T*-range. These combined observations suggest that the magnesite-II phase, found by <sup>285</sup> Isshiki *et al.* [26], was very likely the *C*2/*m*-phase (MgCO<sub>3</sub>-II).



FIG. 8. Phase relations in the MgCO<sub>3</sub> system with respect to the depth profile of the Earth's mantle and outtermost core. Markers correspond to studies, while colours denote distinct phases (black = magnesite, yellow = C2/m-phase [38], white = magnesite + MgO). Data from this study are shown as large circles and were obtained after temperature quenching at high pressures. Data obtained by Solopova *et al.* [32] are shown as hexagons, which have also been obtained after temperature quenching at high pressures. Squares, triangles, and diamonds correspond to the in situ data obtained in the studies by Isshiki *et al.* [26], Maeda *et al.* [9], and Boulard *et al.* [40], respectively. A typical mantle geotherm is shown as grey solid line [70]. The solid (1) and dashed (3) lines represent liquidus and decomposition for magnesite as reported by Solopova *et al.* [32]. The dotted line (2) represents the liquidus for magnesite as reported by Katsura and Ito [69]. The dash-dotted line (4) represents the decomposition of MgCO<sub>3</sub> as reported by Fiquet *et al.* [31]. The dashed yellow band presents a boundary of magnesite into the C2/m-phase derived here. The broad dashed yellow band presents a boundary above which we have not observed decomposition, but solid or liquid MgCO<sub>3</sub>-II. (color online)

#### 286 IV. CONCLUSIONS

This study provides the first Raman spectra of pure  $MgCO_3$  at pressures and temperatures up to 287 148 GPa and 3600 K. Our spectra allowed the identification of MgCO<sub>3</sub>-II-tetracarbonate and the 288 location of the phase boundary between MgCO<sub>3</sub> (magnesite) and MgCO<sub>3</sub>-II (C2/m-phase). Based 289 on our observations from Raman spectroscopy, XRD and density functional theory calculations, 290 we propose a revised phase diagram for  $MgCO_3$ . At P, T-conditions of Earth's upper mantle and 291 upper part of the lower mantle (e.g. pressures up to  $\sim$ 80 GPa and temperatures up to  $\sim$ 2500 K), 292 MgCO<sub>3</sub> is stable as magnesite  $(R\bar{3}c)$  [26, 32]. At P, T-conditions of Earth's lowermost mantle 293 and outermost core (e.g. 85-148 GPa and at temperatures above 2500 K) the stable polymorph 294 MgCO<sub>3</sub>-II is a monoclinic C2/m-tetracarbonate as predicted by Oganov *et al.* [38]. In the present 295 study we observe that MgCO<sub>3</sub>-II can exist on pressure release down to  $\sim$ 40 GPa. This is similar 296 to an earlier observation of the existance of CaCO<sub>3</sub>-tetracarbonate on pressure release down to 297 57 GPa [45]. 298

#### 299 V. ACKNOWLEDGEMENTS

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<sup>308</sup> See Supplemental Material at [URL will be inserted by publisher].

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