

A novel mitigation mechanism for photo-induced trapping in an anthradithiophene derivative using additives

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We report a novel trap mitigation mechanism using molecular additives, which relieves a characteristic early turn-on voltage in a high-mobility *p*-type acene-based small-molecule organic semiconductor, when processed from hydrous solvents. The early turn-on voltage is attributed to photo-induced trapping, and additive incorporation has been found to be very effective in suppressing this effect. Remarkably, the molecular additive does not disturb the charge transport properties of the small-molecule semiconductor, but rather intercalates in the crystal structure. This novel technique allows for the solution-processing of small molecular semiconductors from hydrous solvents, greatly simplifying manufacturing processes for large-area electronics. Along with various electric and spectroscopic characterisation techniques, simulations have given a deeper insight into the trap mitigation effect induced by the additive.

Over the last 20 years organic semiconductors have experienced impressive performance improvements. As field-effect mobilities have exceeded values of 1 cm²/Vs, organic semiconductor based field-effect transistors (FETs) have emerged as a viable thin film electronic technology to realize displays, electronic circuits and sensors on low temperature plastic substrates¹⁻⁴. To enable such applications a detailed understanding of the reliability of organic semiconductors under typical operational and environmental conditions and the

34 mechanisms that can lead to device degradation during operation is required⁵⁻⁸. There have
35 been reports in the literature on instabilities of small-molecule based organic transistors such
36 as pentacene to light and oxygen⁹⁻¹⁶. Other instabilities have been investigated, arising from
37 trace gases in the atmosphere¹⁷. Few studies have been concerned with finding solutions to
38 mitigate these instabilities. We present a simple and generally applicable additive-based
39 technique that reduces the light-induced degradation of small-molecule based organic FETs.

40 2,8-difluoro-5,11-bis(triethylsilylethynyl)anthradithiophene (diF-TES ADT) - a *p*-type
41 small molecule (SM) organic semiconductor - is a prototypical molecular semiconductor that
42 has been widely studied owing to its high crystallinity and reproducible high charge mobilities
43¹⁸. Its high solubility in a broad range of solvents renders it easy to process by solution-based
44 coating.

45 We fabricate our devices in a top-gate bottom-contact (TGBC) transistor architecture
46 using CYTOP as gate dielectric in a nitrogen glove box environment. Air exposure during
47 fabrication is minimal. The devices are measured in an inert atmosphere glovebox and under
48 orange lighting (>590nm). When using ordinary mesitylene as a solvent, i.e. when no attempt
49 was made to reduce the residual water content of the solvent, we noticed that as-fabricated
50 diF-TES ADT FETs initially exhibit a positive turn-on voltage > 10 V, when they were first tested
51 in the glovebox (Fig. 1a). This is observed in other literature on this small-molecule as well
52^{19,20}. When the transistor is placed and tested in the dark and in vacuum, the turn-on voltage
53 recovers to 0V, with the majority of the recovery taking place in the first 5 hours. Whilst in
54 vacuum, the positive turn-on voltage is reinstated upon subsequent light illumination and
55 removed again in the dark. This light-induced instability, which has been widely reported
56 when exposing acene-based small-molecule FETs is seen when our transistors are exposed to
57 orange, white and UV light (Supporting Material 1). Our optical setup does not have sufficient
58 spectral resolution to determine whether with orange light, which is close to the bandgap of
59 diF-TESADT, the process involves regular HOMO-LUMO excitations or specific trap states
60 within the bandgap, but the light-induced device instability exhibit similar qualitative
61 characteristics independent of excitation wavelength. This instability has been attributed in
62 literature to trapping of light-generated electrons in deep trap states within the organic
63 semiconductor bandgap. The associated trapped electrons act as a fixed negative charge near
64 the semiconductor-dielectric interface and result in a positive shift of the turn on/threshold

65 voltage. In the dark the electrons slowly recombine with injected mobile holes and the
66 positive turn-on voltage shift recovers⁹⁻¹⁶. The instability is undesirable for display and
67 electronic applications - though it may be useable for light sensing applications^{21,22} - and
68 techniques are needed to minimize it.

69 We noticed that when processing diF-TES ADT from an anhydrous solvent, the turn-
70 on in the as-fabricated transistors is close to 0V when measured under the same orange
71 lighting and inert environments. Figure 1b shows the difference between a transistor
72 fabricated from hydrous mesitylene and another fabricated from anhydrous tetralin,
73 measured in an inert atmosphere glovebox and under orange lighting. The mesitylene device
74 exhibits a distinct subthreshold bump and a positive turn-on voltage of around +25V in the
75 transfer characteristics, while the tetralin device has a more regular transfer characteristic
76 with a turn-on closer to 0V. Unfortunately, it is laborious to obtain anhydrous mesitylene,
77 and we were unable to purchase it commercially. For a more direct comparison and to
78 eliminate the effect the solvent itself may have, Figure 1c contrasts transistors processed
79 from hydrous and anhydrous toluene, showing a similar difference, i.e. the device fabricated
80 from the hydrous solvent has a more positive turn-on voltage. After the exposure of both
81 preparations to a day in air, it is evident that the turn-on begins to drift to more positive values
82 in both cases. This finding, as well as the one from hydrous mesitylene (Fig. 1a) provide an
83 indication that the water/oxygen content of the film is a contending culprit to the mechanism.

84 We have recently demonstrated a simple method to drastically improve the
85 operational and environmental stability of a wide range of polymer semiconductor systems,
86 and alleviate the effect of water-induced trapping through the incorporation of small
87 molecular additives into the semiconducting polymer films⁵. Here we adopted a similar
88 approach by analogy to investigate whether the incorporation of a small molecular additive
89 into the diF-TES ADT films may reduce the concentration or effectiveness of water impurities
90 in the film and in this way improve the positive turn-on instability illustrated in Figure 1.

91 7,7,8,8-Tetracyanoquinodimethane (TCNQ) and 2,3,5,6-Tetrafluoro-7,7,8,8-
92 tetracyanoquinodimethane (F4TCNQ) were used as additives in the hydrous mesitylene
93 solution of diF-TES ADT. They are added in solution at 2 wt.%. In the case of F4TCNQ, the
94 electron affinity is deep enough to induce *p*-type doping of diF-TES ADT, which results in a
95 large positive turn-on voltage shift and difficulty in turning-off the devices (Fig. 2a). However,

96 in contrast, the addition of TCNQ successfully suppresses the positive turn-on voltage, when
97 measured in an inert glovebox under orange lighting conditions. The literature value of the
98 electron affinity of TCNQ (2.8-3.3 eV)²³ is significantly shallower than the HOMO of diF-TES
99 ADT (5.3 eV)²⁴, and hence very little to no charge transfer doping occurs. The addition of
100 TCNQ clearly leads to a device with near perfect turn-on near 0V even when using a hydrous
101 solvent.

102 Photothermal Deflection Spectroscopy (PDS) (Fig. 2b) carried out on the preparations
103 in Fig. 2a shows a broad polaronic absorption background signal below 2.1 eV in the case of
104 the F4TCNQ additive (blue) due to *p*-type charge transfer doping. This is not evident in the
105 case of TCNQ (orange), which shows no such doping effect. This is fully consistent with the
106 electrical measurements and the energy levels of TCNQ and diF-TESADT. However, both
107 preparations show a similar sub-bandgap absorption feature with a peak around 1.1 eV and
108 a high energy tail between 1.4-1.6 eV. The origin of this spectral feature is discussed below.

109 It is remarkable to see that the addition of the additives at these small concentrations
110 does not hinder charge transport in the films. The same level of current is achieved as that of
111 a pure diF-TES ADT film. θ -2 θ X-ray Diffraction (XRD) measurements (Fig. 2c) revealed that the
112 (00L) lamellar stacking distance between the molecular layers is slightly increased by around
113 0.4Å, whereas Grazing Incidence Diffraction (GID) measurements confirmed that the crystal
114 packing of diF-TES ADT in the direction of π - π stacking remains unaffected (Supporting
115 Material 2). This suggests that some of the additive intercalates into the film causing the
116 lamellar distance to increase. The TCNQ most likely inserts itself in the vicinity of the alkyne
117 side chains of diF-TES ADT creating an anisotropic lattice expansion. This does not exclude the
118 possibility that the additive is also incorporated into grain boundaries in the film, which we
119 are not sensitive to in X-ray measurements. We were unable to measure directly the
120 concentration of TCNQ in the films; it could be different from the 2 wt. % that is present in
121 solution, but the XRD lattice expansion provides clear evidence for some level of TCNQ
122 incorporation into the films. The full width half maximum (FWHM) of the Bragg-peak at 0.4\AA^{-1} of is
123 reduced in the films with additives compared to the neat films, which suggest an improvement in
124 crystallinity upon additive incorporation.

125 Figure 2d shows that the additive completely passivates the effect of the ambient
126 environment on the small-molecule. After exposing a transistor to 24 hours of ambient air

127 environment under orange lighting, complete stability is observed without any increase in the
128 turn-on voltage to more positive values (orange line, solid to dashed). Likewise, the black
129 transfer characteristics show the stability before and after 24 hours of exposure to orange
130 light in a nitrogen-filled glovebox (black line, solid to dashed). This perfect stability is in
131 contrast to the instability shown in Figure 1c, where pure transistors develop a pronounced
132 positive turn-on voltage after similar air exposure.

133 **Discussion**

134 There are several sources of impurities that can be present in the thin film of the small-
135 molecule. It is synthesized and stored in ambient air environments, which contribute
136 oxygenated species to the film ¹⁷, whether they are permanent bonds or charge transfer
137 complexes. Also, since the diF-TES ADT films are annealed for only 2 minutes at 100°C, we
138 assume there is still a solvent content. In films spun from hydrous solvents, this content will
139 also constitute hydrated-oxygenated species, which from hereon will be referred to as
140 hydrated species.

141 It is apparent from Figure 1, that the effect observed is a photo-induced effect,
142 whereby charge is excited from the HOMO to the LUMO of diF-TES ADT when exposed to
143 light. This excitation is affected by the existence of hydrated species in the film, as films spun
144 from anhydrous solvents do not exhibit the light induced effect, until they are exposed to
145 ambient environments whereby ambient oxygen and humidity begin to induce a similar effect
146 as those films spun from hydrous solvents (Fig. 1c).

147 It is also concluded that it is not a requirement for the additive to undergo charge
148 transfer with the semiconducting molecule. In the case of TCNQ addition, where no (or very
149 little) doping is induced, an early turn-on suppression is elegantly evident in the transfer
150 characteristic.

151 The mechanism of photo-induced trapping has been reported for thiophene-based
152 semiconducting polymers ²⁵. We hereby report a similar effect in highly-crystalline small-
153 molecule semiconductors. The incorporation of additives into polymer films is significantly
154 less detrimental to charge transport due to the amorphous/semi-crystalline nature of
155 polymer systems. We propose a mitigation method for this photo-induced trapping
156 mechanism through the incorporation of additives into the crystalline small-molecule film,

157 without any hindrance to the crystalline packing of the thin film. This in turn can induce no
158 detrimental effect on the charge transport properties of the fabricated transistors,
159 considering an optimized amount of additive.

160 In the absence of hydrated species in a pure thin film, any excited charge would
161 recover back to the HOMO once the light is removed. However as illustrated schematically in
162 Figure 3, the presence of hydrated species forms a trap level aligned within the band-gap of
163 the small-molecule, where the electrons favor a relaxation to this level from the LUMO,
164 causing them to become trapped. In the case where dark environments and vacuum are
165 introduced, Figure 1 shows that the charges de-trap with prolonged time.

166 When an additive such as TCNQ is introduced to the thin film, the charge favorably
167 relaxes onto the LUMO of TCNQ as it is lower in energy (Fig. 3, ungated). Under the influence
168 of an electric field, such as that present in operating a transistor (Fig. 3, gated), the charge on
169 the additive recombines with the excess of holes present in the accumulation region of the
170 transistor, allowing the turn-on to recover to an ideal behavior. This was previously not
171 possible in the absence of the additive, where the charge was trapped more permanently on
172 the hydrated species. The energy level of this electron state localized on the TCNQ molecule
173 is indicated in Fig. 3 at an energy around 4.3eV. The theoretical calculations indicate that in
174 the presence of water and oxygen molecules this energy level can be below what would be
175 expected from the literature value of the TCNQ electron affinity. However, we note that the
176 value of the energy of this state depends on the detailed arrangements of the cluster being
177 considered in the calculations and should be considered an approximate value only.

178 In PDS, the sub-bandgap absorption seen between 0.8 eV and 1.6 eV in thin films with
179 TCNQ consists of two assignable signatures: the peak centered at around 1.1 eV corresponds
180 to a localised polaron-induced absorption (PIA) of diF-TES ADT and the broad high energy
181 shoulder can be assigned to two peaks at 1.4 eV and 1.6 eV that have been attributed in the
182 literature to the anionic species of TCNQ/F4TCNQ²⁶⁻²⁸ (Supporting Material 5). These peaks
183 are synonymous and we believe this to be the optical signature of a trapped electron (that
184 was created by photoexcitation during the PDS measurement) and a localized hole polaron
185 induced on diF-TES ADT to compensate for the negative electron charge prior to
186 recombination.

187 One can imagine a scenario where the TCNQ is closely packed between the diF-TES
188 ADT side chains, and the hydrated species are partially solvated in the residual solvent in the
189 films. In this case, one can see that the effect of hydrous mesitylene on transistors (Fig. 1a)
190 will be much more pronounced than that of hydrous toluene (Fig. 1c), due to toluene having
191 much higher water solubility than that of mesitylene (Supporting Material 5), providing a
192 better shielding against charge trapping from the diF-TES ADT to the hydrated species.

193 We have performed density functional theory calculations to gain more insights on
194 the proposed mechanism. All computational details and details of model systems considered
195 are given in Supporting Material 6. All geometries of these complexes were fully optimized
196 without any constraints. The optimized geometries were used to carry out the excited state
197 analysis. The hole and electron wavefunctions were obtained from the natural transition
198 orbital analysis.

199 The results obtained from these calculations clearly show that that hydrated species
200 introduce an energy level aligned to the band gap of diF-TES ADT (Supporting Material 6, Fig.
201 6-1). This is in agreement with the work done by Zhuo et al.²⁵. In the absence of TCNQ, in the
202 lowest excited state, the electron density is concentrated mostly on the oxygen molecule, and
203 the hole density is predominantly on the small molecule (Figure 4). Charged oxygen molecules
204 could react readily with water giving, OH anions, OH radicals and OOH radicals (Supporting
205 Material, Fig. 6-3). Hydrated OH radicals and OH anion/OOH radicals have energies deeper
206 than diF-TES ADT's LUMO and hence are likely to be the culprits of charge trapping (Figure 4).

207 The incorporation of TCNQ near the side chains of diF-TES ADT - as indicated by our
208 XRD/GID measurements - yielded no charge transfer between the molecules (Supporting
209 Material 6, Fig. 6-2), confirming our experimental results. The calculations show that in the
210 presence of TCNQ molecules, the light-induced excited electron density resides wholly on the
211 TCNQ molecule, whilst the hole density remains on the small molecule (Figure 4). The
212 electrons now reside at an energy much closer to the HOMO of the small molecule
213 (Supporting Material 6, Fig. 6-1). As a result, the trapping effect of the hydrated species is
214 mitigated.

215 **Operational Stability**

216 diF-TES ADT has shown to develop a more pronounced shoulder in the off state with
217 increased positive gate bias stress (PGBS) (Fig. 5a) when performed under orange lighting and
218 in a nitrogen glovebox, exhibiting a threshold voltage shift (ΔV_{th}) of 6.1V over 12 hours of
219 stress. When the transistor was left to de-trap in the dark for 5 hours prior to stress, and the
220 stress was performed in the dark (Fig. 5b), the effect of PGBS was minimized to a ΔV_{th} of
221 1.7V, with a reduced increase in the shoulder feature. Threshold voltage extractions can be
222 seen in Supporting Material 7. It can also be seen in Figure 5b that the transistor recovers
223 completely to its initial state within the first hour of post-stress recovery, which is not the
224 case in Figure 5a, even after 3 hours of recovery.

225 This is an indication that the light-induced trapping mechanism described in this study
226 influences PGBS and is hampered by slow recombination rates from the hydrated impurities
227 back into the HOMO of the semiconductor. Unfortunately, our TCNQ additive approach,
228 which imparts improved stability of the threshold voltage under environmental and light
229 exposure, is not effective to improve the PGBS stability. Figure 5c shows that the PGBS–
230 induced threshold voltage shift of devices with TCNQ additive is of a similar magnitude as of
231 devices without the additive. At this stage it is unclear why this is the case. In a regime of bias
232 stress, where charge trapping is influenced by prolonged applied electric fields the
233 incorporation of TCNQ may be amplifying other trapping mechanisms, such as trapping of
234 electrons injected from the source-drain contacts. Whilst experimenting with different
235 additives is beyond the scope of this study, future experimentation is required to understand
236 these to discover the right additive that imparts both good light stability and operational
237 stability.

238 **Conclusions**

239 We have shown that diF-TES ADT shows a pronounced positive gate turn-on voltage,
240 when fabricated from hydrous solvents, and when exposed to ambient environments. This
241 has been attributed to a photo-induced effect whereby excited charges are trapped onto a
242 trap level introduced by hydrated impurities in the semiconducting film.

243 We have identified that the addition of small molecular additives at small
244 concentrations (around 2 wt.%) do not hinder the crystal packing or charge transport in the
245 small molecular crystal lattice, but rather intercalate in the film structure forming an

246 anisotropic lattice expansion in the out-of-plane direction. This has the beneficial effect of
247 alleviating the photo-induced trapping mechanism without hindrance to charge transport
248 properties.

249 The method by which this occurs is by a favorable relaxation of charge onto the
250 electron-accepting TCNQ molecule, which in the presence of an electric field, donates the
251 charge back to the HOMO level of the semiconducting molecule.

252 This effect has important implications on stabilizing organic semiconductor
253 performance against the several degrading environmental factors that they are susceptible
254 to in their pure form, making them suitable for reproducible integration into state-of-the-art
255 applications.

256 From this study, we are able to begin understanding the design features of an optimal
257 additive molecule. To achieve light stability, the additive molecule must have an electron-
258 accepting nature, with a LUMO level deeper than the trap level introduced by the impurity in
259 the thin film in order to introduce a favorable transition. In order to prevent a charge transfer
260 doping effect, it is advisable that the LUMO level is also shallower than the HOMO of the
261 organic semiconductor. Further experimentation can determine more detailed design
262 features (shape, size, etc...) of the additive so as to generally suit a host of different small-
263 molecule semiconductor packing motifs, and to mitigate the effects of different sources or
264 traps in the films.

265 **Experimental Methods**

266 All fabrication methods were carried out under orange lighting conditions (filter
267 transmission above 590nm). Measurements done in inert atmospheres were conducted using
268 a nitrogen-filled glovebox.

269 Transistors fabricated for this study are of top-gate bottom-contact (TGBC)
270 architecture. Apart from Figure 1a and Supporting Material 1, all transistors have a channel
271 length (L) of 20 μm and channel width (W) of 1 mm. The substrates are fabricated on 1737
272 low alkali alumino-silicate glass substrates purchased from 'Präzisions Glas & Optik GmbH',
273 with a spin-coated 50nm film of polyimide (spun and from 1-Methyl-2-pyrrolidone) and cured
274 (1hr at 160 °C, then for 3 hrs at 300 °C in N₂ glovebox) for better semiconductor film formation

275 properties. Chromium/gold Source and drain electrodes were patterned using
276 photolithography, and thermally evaporated at 1.5nm/25nm thicknesses, respectively. The
277 electrodes were treated with a pentafluorobenzenethiol (PFBT) self-assembled monolayer
278 (SAM) for enhanced contact resistance and crystallinity into the channel region²⁹.

279 Solvents used to dissolve diF-TES ADT ³⁰ were stored in air/nitrogen for
280 hydrous/anhydrous analogues, respectively. However, all solutions were made inside a
281 nitrogen-filled glovebox. diF-TES ADT was spin-coated (20 wt.% solutions, 1200 rpm for 60
282 seconds) and annealed at 100°C for 2 minutes inside a nitrogen-filled glovebox. CYTOP was
283 used as a gate dielectric, and spun to achieve films of 500nm thickness. Aluminium gate
284 electrodes were thermally evaporated at a thickness of 35nm.

285 PDS samples were spun on 13mm diameter water-free boro-silicate substrates
286 (spectrosil). Measurements were carried out in a dark environment. The PDS set-up used is
287 home-made³¹, with a high level of accuracy.

288 X-ray Diffraction (XRD) and Grazing Incidence Diffraction (GID) spectroscopy samples
289 were spun on similar glass/polyimide substrates as those used for transistors. Half of the
290 substrates were covered with a PFBT-treated 6nm thin film of chromium/gold, to mimic the
291 preferential nucleation effects of the gold electrodes in transistors. The measurements were
292 taken on the glass/polyimide side. The sample cross-section and measurement conditions are
293 available in Supporting Material 2. Coplanar XRD measurements on glass were performed at
294 beamline ID3 at the European Synchrotron Research Facility (ESRF) in Grenoble, France, using
295 a beam energy of 15.0keV (wavelength 0.83Å) with dimensions of 500um x 30um. The
296 detector was a Maxipix 2D pixel detector with 500 pixel x 500 pixel. The distance between
297 the sample and the detector was 77.2cm.

298 Optical microscope images (Supporting Material 3) were taken using an Olympus
299 microscope with cross-polarizers on the same samples used for XRD and GID.

300 Charge Accumulation Spectroscopy was carried out using a home-made set-up, and is
301 best described in Ref. ^{32,33}.

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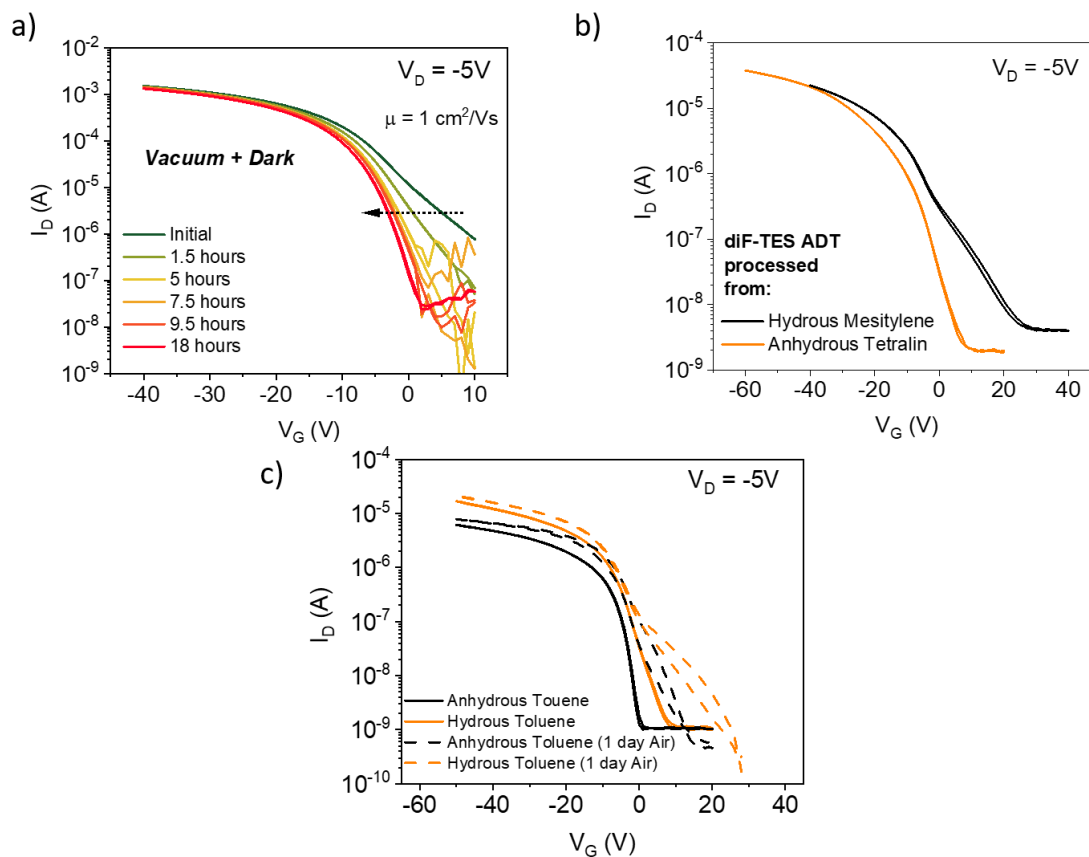
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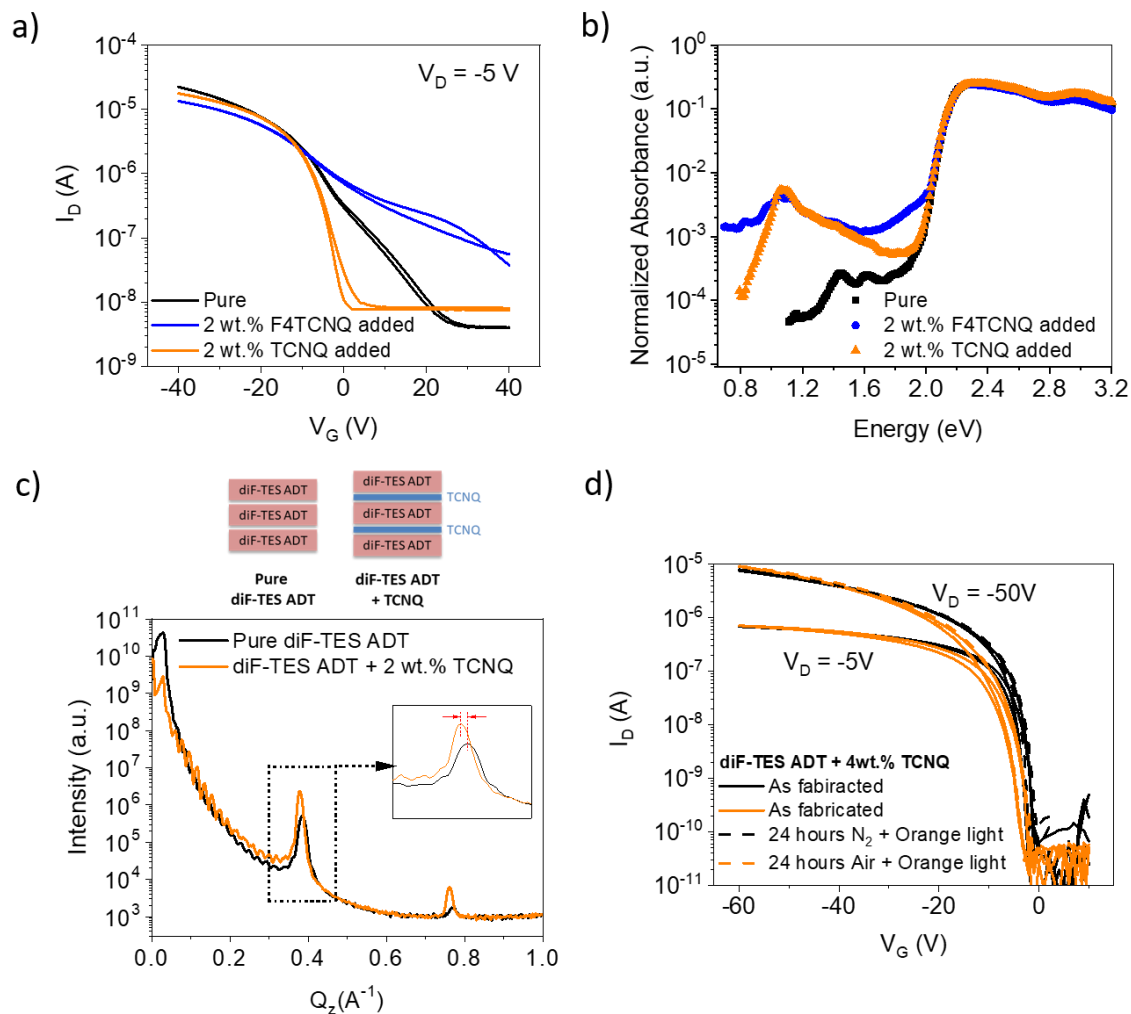
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400 **Figure 1** a) The decrease in the gate turn-on voltage of a diF-TES ADT transistor spun from
 401 hydrous mesitylene (Channel Width/Length=190 nm/80 μm), stored in the dark and in
 402 vacuum. b) The difference in transfer characteristic between as-fabricated diF-TES ADT
 403 transistors (Channel Width/Length=1000/20 μm) fabricated from hydrous mesitylene (black)
 404 and anhydrous tetralin (orange). c) A comparison of diF-TES ADT transistors (Channel
 405 Width/Length=1000/20 μm) fabricated from hydrous (orange) and anhydrous (black) toluene
 406 solution, as fabricated (solid) and after a day of ambient air exposure (dashed).

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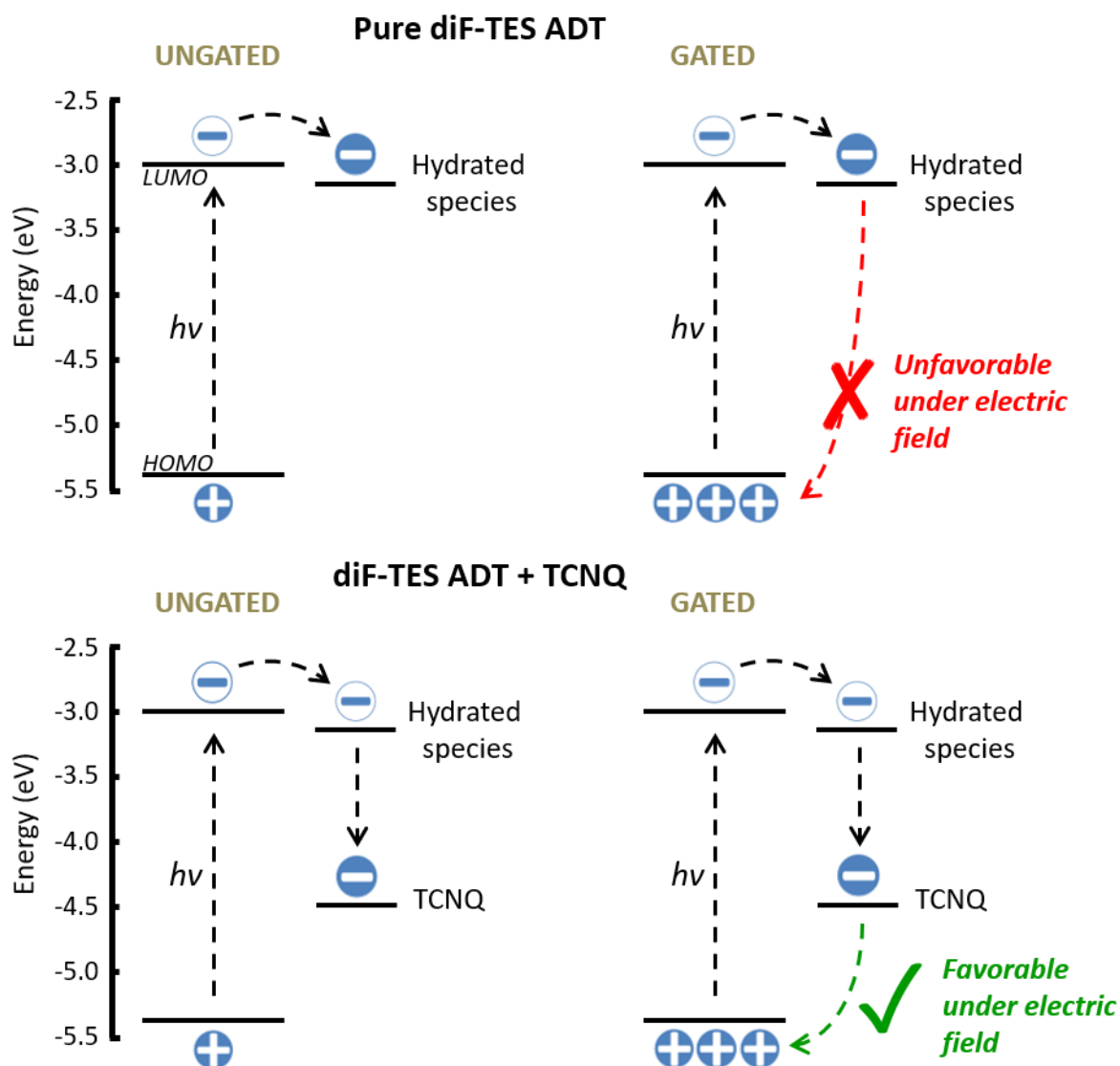
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410 **Figure 2** a) A comparison of transfer characteristics of diF-TES ADT transistors (Channel
 411 Width/Length=1000/20 μm) spun from hydrous mesitylene, pure (black), with 2 wt.% TCNQ
 412 (orange) and 2 wt.% F4TCNQ (blue). b) PDS measurements for the same preparations as those
 413 in a). c) X-ray Diffraction (XRD) curves along Q_z for pure diF-TES ADT and 2 wt.% TCNQ
 414 addition, highlighting the small z increase in the vertical spacing of diF-TES ADT packing
 415 through the addition of TCNQ. The cartoon shows the intercalation motif of TCNQ into the
 416 diF-TES ADT packing. d) Linear ($V_D = -5V$) and saturation ($V_D = -50V$) transfer characteristics
 417 (Channel Width/Length=1000/20 μm) of two transistors with 4 wt.% TCNQ. The transistor
 418 shown in orange was exposed for 24 hours to ambient air under orange lighting, while the
 419 transistor shown in black was exposed for 24 hours to orange light in a nitrogen-filled
 420 glovebox. Solid lines show the transistor before exposure, and dashed lines show the
 421 transistor after exposure, highlighting the environmental stability that TCNQ additive induces
 422 in diF-TES ADT transistors.

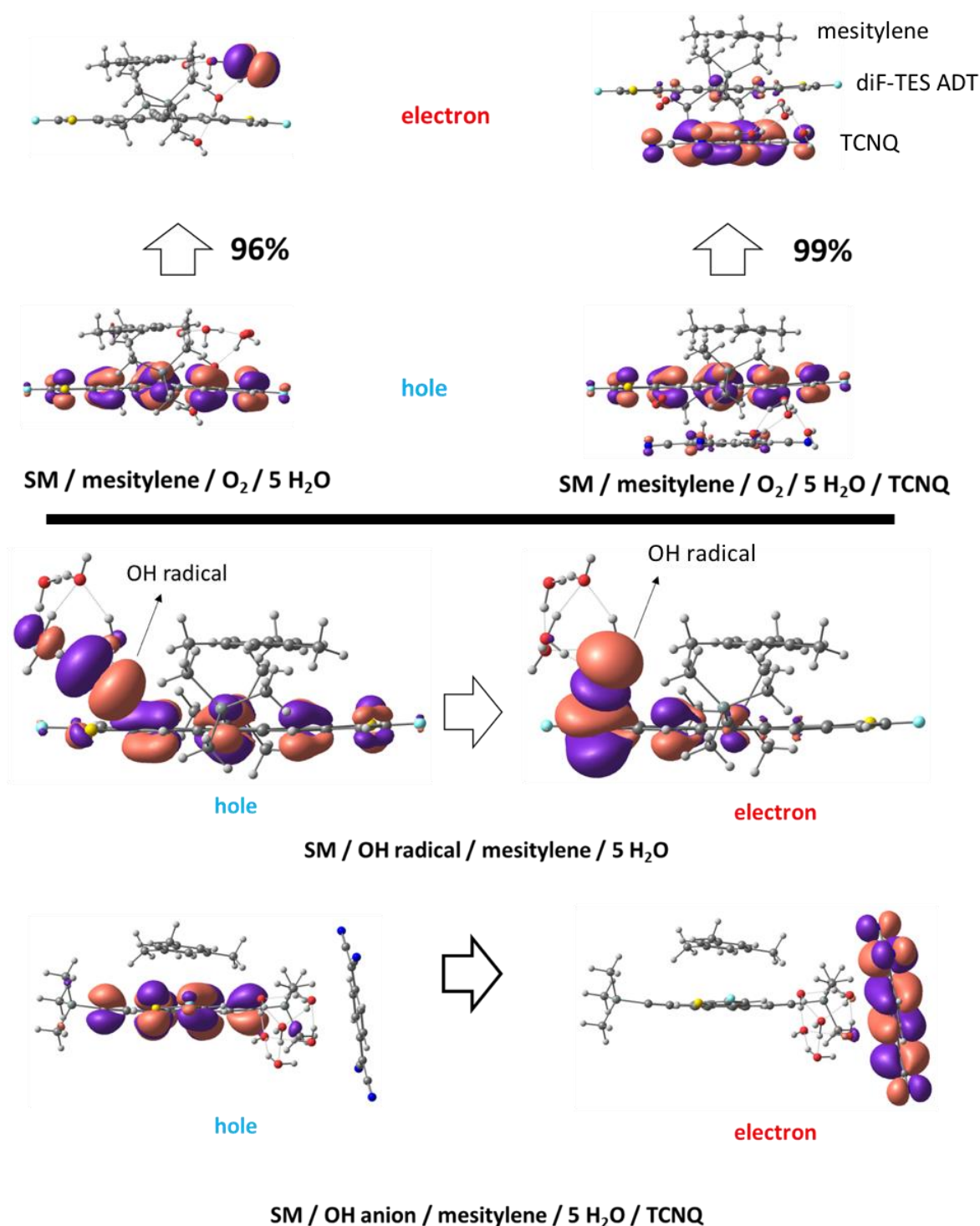
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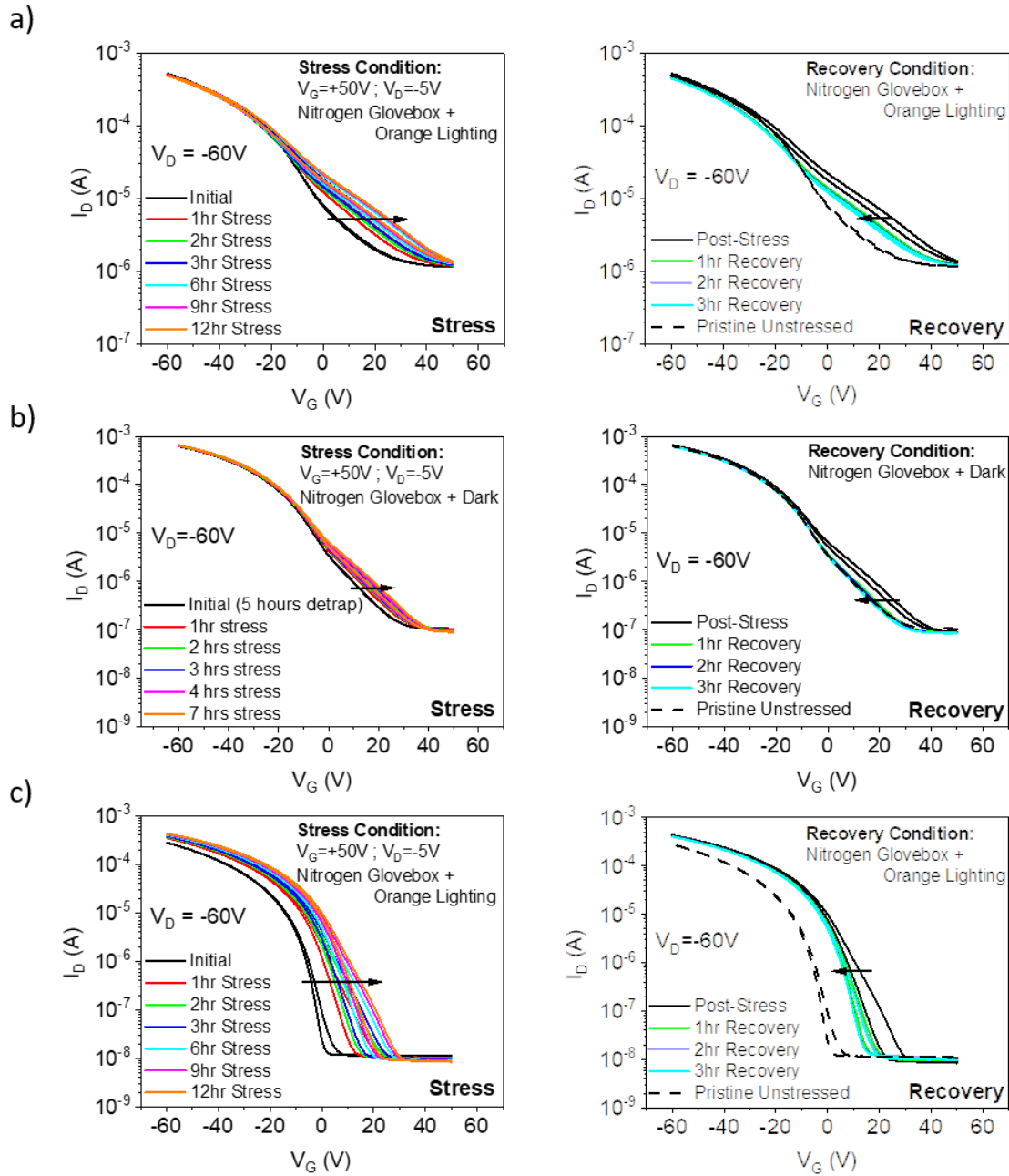
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426 **Figure 3** The proposed mechanism by which TCNQ successfully mitigates the charge trapping
 427 effect. The left two quadrants present energy level diagrams in the absence and presence
 428 (top and bottom quadrant, respectively) of the TCNQ additive. The right two quadrants show
 429 energy level diagrams of corresponding systems when the device is gated, and the transistor
 430 channel is experiencing an electric field. The arrows denote the transfer of an electron.



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432 **Figure 4** Pictorial representation of hole and electron wavefunctions in various clusters of the
 433 small molecule (SM) with O₂, H₂O, TCNQ and mesitylene in their lowest excited states as
 434 determined at the TD-OT(PCM)- ω B97XD/6-31G(d,p) level of theory. (Top panel) The electron
 435 density is completely on oxygen causing it to get charged. In the presence of TCNQ, the
 436 electron density is preferentially on the TCNQ. (Bottom Panel) Taking into account OH anions
 437 as a product of charged oxygen reacting with water in the film, the electron density now is on
 438 the OH anions in the absence of TCNQ, and on TCNQ when it is present. The TCNQ molecule
 439 here is placed near the side chains, as suggested by XRD.



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Figure 5 Transistor (Channel Width/Length=1000/20 μm) transfer characteristics showing PGBS and subsequent post-stress recovery behavior of pure diF-TES ADT in a nitrogen glovebox a) under orange lighting, b) detrapped in the dark for 5 hours prior to PGBS, with PGBS performed in the dark as well. c) shows the PGBS and post-stress recovery of diF-TES ADT with 2 wt.% TCNQ performed under orange lighting.