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Letter

Double-shell CeO₂:Yb, Er@SiO₂@Ag upconversion composite nanofibers as an assistant layer enhanced near-infrared harvesting for dye-sensitized solar cells

A B S T R A C T

Keywords:

CeO₂:Yb,Er@SiO₂@Ag
Upconversion luminescence
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Double-shell CeO₂:Yb,Er@SiO₂@Ag upconversion composite nanofibers are synthesized by electrospinning and subsequent process. CeO₂:Yb,Er@SiO₂@Ag nanofibers show high upconversion luminescence property due to the coating of amorphous SiO₂ and the surface plasmon resonance effect of Ag nanoparticles. CeO₂:Yb,Er@SiO₂@Ag nanofibers act as an assistant layer in dye-sensitized solar cells (DSSCs) and enhance the photoelectric conversion efficiency (PCE) to 8.17%. The photocurrent-voltage characteristic is obtained under 980 nm laser as illumination light source. In addition, the absorption of the incident photon-to-current conversion efficiency curve in 900–1000 nm near-infrared light confirms that the introduction of the upconversion nanomaterial broadens the absorption range, improves the utilization rate of the sunlight and increases the PCE of DSSCs.

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1. Introduction

The dye-sensitized solar cells (DSSCs), owing to their relatively good efficiency, easy manufacturing processing, and low cost, have garnered much attention since 1991 [1]. The main problem that limits the photoelectric conversion efficiency (PCE) of DSSCs is low absorption for the infrared and near-infrared (NIR) light, which occupies 43% of the solar spectrum. Photon upconversion provides an alternative approach to resolve this limitation utilizing NIR light absorber [2–4]. Therefore, one straightforward way to increase the PCE of DSSCs is adopted the suitable dye to broaden the sunlight harvesting range, but this kind of dye always is expensive and poor stability [5–8]. Moreover, the inorganic rare-earth doped NaYF₄ and Y₂O₃ nanomaterials are usually used as the upconversion materials, which are introduced into the DSSCs for adsorbing NIR light [9–14]. Cerium dioxide (CeO₂) has the stable crystalline field for the upconversion progress. Furthermore, it has a suitable energy band gap matching with TiO₂ and good light scattering effect, which lead to the wide-spread use of CeO₂ as an assistant component in DSSCs. A few research focus on rare-earth ion doped CeO₂ as an upconversion assistant layer in DSSCs [15–18].

In this work, we designed and synthesized double-shell CeO₂:Yb,Er@SiO₂@Ag upconversion composite nanofibers (UCNFs). Under the assistance of SiO₂ layer and Ag nanoparticles (NPs), CeO₂:Yb,Er@SiO₂@Ag UCNFs display the excellent upconversion luminescence performance. The use of CeO₂:Yb,Er@SiO₂@Ag UCNFs as assistant materials for improvement of the NIR light harvesting in DSSCs reach a PCE as high as 8.17%.

2. Materials and methods

In our work, the agents were acquired from Sinopharm Chemical Reagent Co. Ltd. (China). All reagents were analytical grade and received without further purification. The details for synthesizing CeO₂:Yb,Er@SiO₂@Ag UCNFs, sensitized DSSCs and characterization are shown in [Supporting Information \(SI\)](#).

3. Results and discussions

Fig. 1a illustrates the preparation progress of CeO₂:Yb,Er@SiO₂@Ag UCNFs. Firstly, CeO₂:Yb,Er nanofibers as fluorescence center are synthesized *via* electrospinning and calcination. Secondly, the SiO₂ layers are coated on the surface of CeO₂:Yb,Er nanofibers by hydrolysis of TEOS to form the core-shell CeO₂:Yb,Er@SiO₂ nanofibers. Lastly, Ag NPs *in situ* grow on the surface of SiO₂ by microwave irradiation to obtain double-shell CeO₂:Yb,Er@SiO₂@Ag UCNFs. SEM and TEM images of the intermediate or final products were shown in **Fig. 1b–j**. The composite nanofibers have a diameter about 150 nm with the about 15 nm thickness of SiO₂ and the Ag NPs is about 10 nm. The HRTEM images (**Fig. 1k–l**), SAED pattern (**Fig. 1m**), and element mapping images (**Fig. 1n–s**) prove the well crystallinity, the composite structure and the hypodispersion of every elements for CeO₂:Yb,Er@SiO₂@Ag UCNFs. As shown in **Fig. S1a**, XRD presents the characteristic peaks at 28.5°, 33.1°, 47.5°, and 56.5° with respective lattice planes (111), (200), (220), and (311), which are consistent with the cubic CeO₂. XPS spectra (**Fig. S1b–d**) indicated the existence of Er and Yb in the form of trivalent ions in CeO₂, cerium as all attributed to Ce⁴⁺ state [19,20], and a small amount of Ag simple substance.

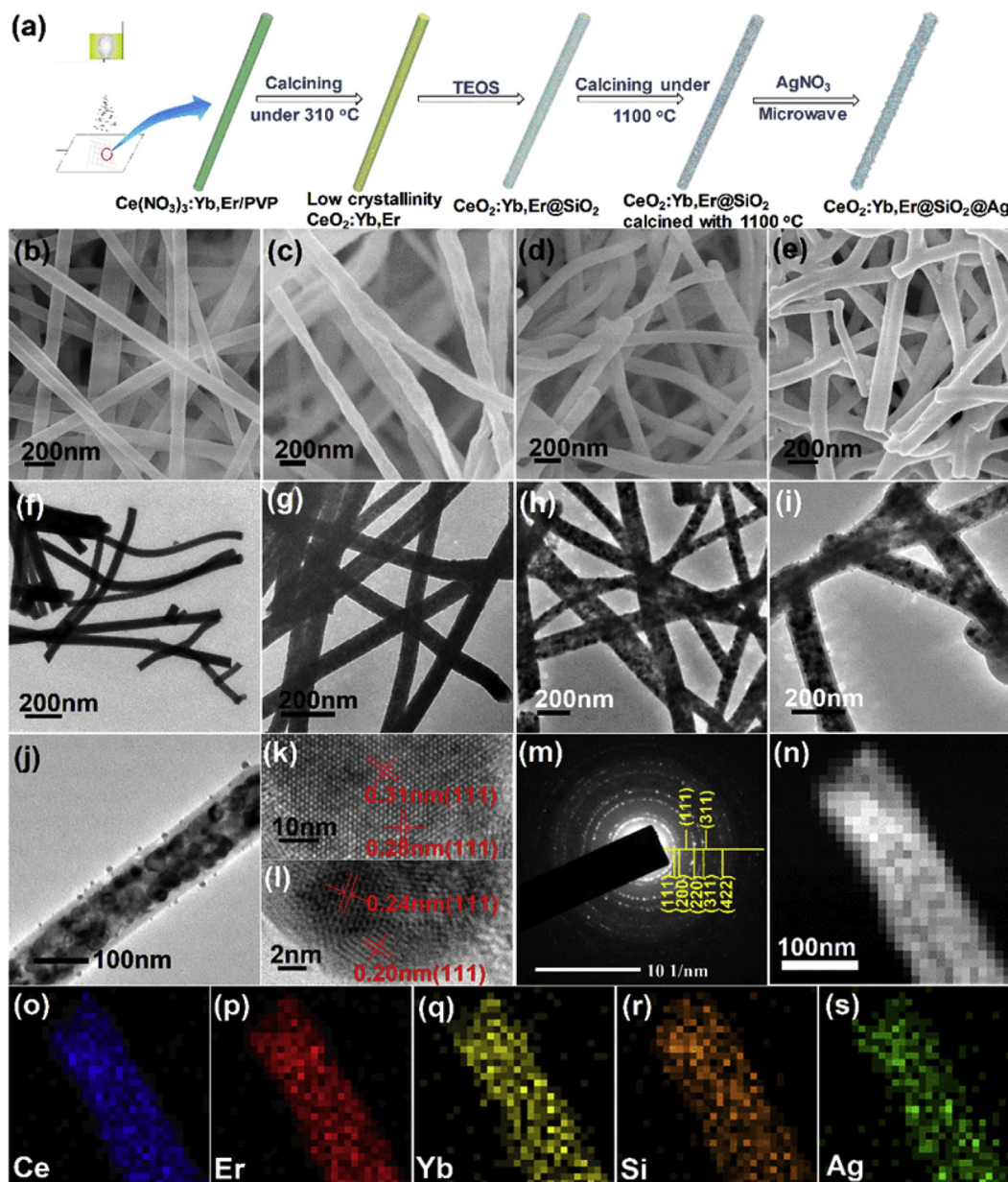


Fig. 1. (a) Illustration for the preparation of double-shell $\text{CeO}_2:\text{Yb,Er}@SiO_2@Ag$ UCNFs. SEM and TEM images of (b) (f) precursor of $\text{CeO}_2:\text{Yb,Er}$ nanofibers, (c) (g) $\text{CeO}_2:\text{Yb,Er}$ nanofibers after 310°C calcination, (d) (h) $\text{CeO}_2:\text{Yb,Er}@SiO_2$ nanofibers after 1100°C calcination, and (e) (i) (j) $\text{CeO}_2:\text{Yb,Er}@SiO_2@Ag$ UCNFs. (k) (l) HRTEM images, (m) SAED pattern, and (n) HAADF-STEM of $\text{CeO}_2:\text{Yb,Er}@SiO_2@Ag$ UCNFs, element mapping images (o) Ce, (p) Er, (q) Yb, (r) Si, and (s) Ag.

In this case, the double-shell design could easily adjust the distance between fluorescence center and the outer layer of Ag NPs by controlling the thickness of the SiO_2 layer to achieve the maximum luminescence. When changing the amounts of TEOS, the different thickness of SiO_2 coating from 0 to 50 nm (Fig. S2) was synthesized. It is found that the bare $\text{CeO}_2:\text{Yb,Er}$ nanofibers display faint upconversion luminescence. On the contrary, the $\text{CeO}_2:\text{Yb,Er}@SiO_2$ (15 nm) nanofibers show the highest emission intensity (Fig. 2), indicating the luminescence regulating effect of SiO_2 layer. Additionally, under the confinement effect of SiO_2 coating, the size of $\text{CeO}_2:\text{Yb,Er}$ nanoparticles is about 40–50 nm (Fig. S2). However, it is clearly shown that the size of $\text{CeO}_2:\text{Yb,Er}$ nanoparticles increases to 150 nm without the protection of SiO_2 shell (Fig. S3). The SiO_2 layers play two important roles in the $\text{CeO}_2:\text{Yb,Er}@SiO_2@Ag$ UCNFs. On the one hand, the SiO_2 layers confine the growth and fusion of

$\text{CeO}_2:\text{Yb,Er}$ nanoparticles during the calcination process, which can enhance the scattering and collection of light; on the other hand, the crystal symmetry of CeO_2 is broken locally by binding of CeO_2 and SiO_2 , and the asymmetry allows the f-f transitions in favor of upconversion luminescence property improvement [21,22].

Fig. S4a–b show the UV–vis absorption and Raman of $\text{CeO}_2:\text{Yb,Er}$, $\text{CeO}_2:\text{Yb,Er}@SiO_2$, and $\text{CeO}_2:\text{Yb,Er}@SiO_2@Ag$ UCNFs. The coating of SiO_2 and loading of Ag NPs all improved the absorption for the whole wavelength and Raman effect. Under illumination of 980 nm laser (Fig. 3), the luminescence intensity of $\text{CeO}_2:\text{Yb,Er}@SiO_2$ (15 nm) nanofibers and $\text{CeO}_2:\text{Yb,Er}@SiO_2@Ag$ UCNFs is 15 and 19 times stronger than $\text{CeO}_2:\text{Yb,Er}$ at 550 nm in green emission spectra. The enhancement of luminescence intensity depends on the distance between fluorescence center and Ag NPs. The suitable distance between $\text{CeO}_2:\text{Yb,Er}$ and Ag NPs can

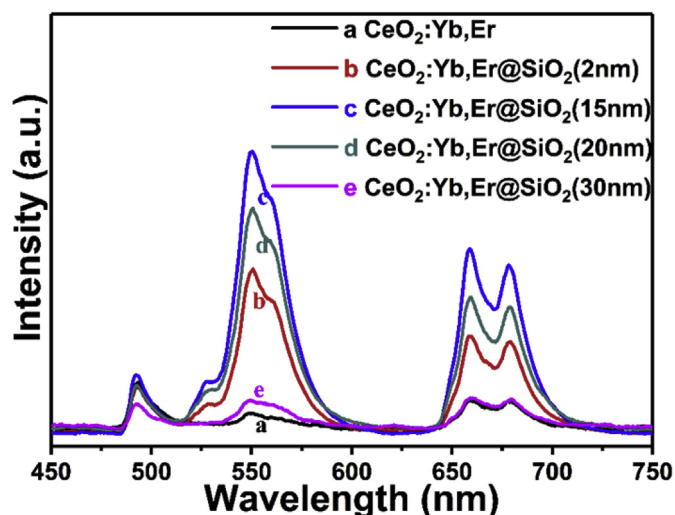


Fig. 2. Upconversion photoluminescence spectra of $\text{CeO}_2:\text{Yb,Er}/\text{SiO}_2$ nanofibers with different silica thickness: (a) without SiO_2 , (b) ~ 2 nm, (c) ~ 15 nm, (d) ~ 20 nm, and (e) ~ 30 nm.

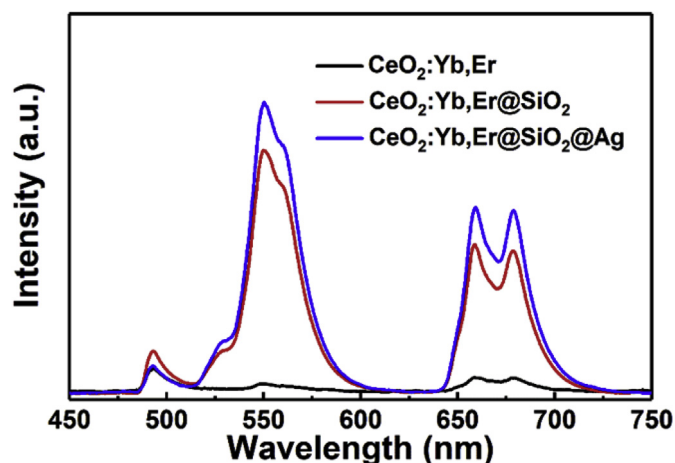


Fig. 3. Upconversion photoluminescence spectra of $\text{CeO}_2:\text{Yb,Er}$, $\text{CeO}_2:\text{Yb,Er}/\text{SiO}_2$ nanofibers, and $\text{CeO}_2:\text{Yb,Er}/\text{SiO}_2@\text{Ag}$ UCNFs.

prevent the luminescence quenching. The nanoscale fluorophore-metal interactions give rise to the process called metal-enhanced fluorescence, which is caused by the interaction between excited fluorophores ($\text{CeO}_2:\text{Yb,Er}$) and surface plasmon resonance (SPR) of metals (Ag NPs).

To investigate the influence of UCNFs on DSSCs, we fabricated four kinds of DSSCs. $\text{CeO}_2:\text{Yb,Er}$, $\text{CeO}_2:\text{Yb,Er}/\text{SiO}_2$ nanofibers, and $\text{CeO}_2:\text{Yb,Er}/\text{SiO}_2@\text{Ag}$ UCNFs were used as assistant layers, respectively. They are named as C1, C2 and C3, respectively. The control cell is normal DSSC (C4). Fig. 4a shows the photocurrent-voltage (J - V) characteristic of four DSSCs and the photovoltaic parameters listed in Table S1. The control cell displays a short current density (J_{sc}) of 13.99 mA cm^{-2} , $V_{oc} = 0.76 \text{ V}$, $FF = 0.63$, and $PCE = 6.65\%$. C3 shows a J_{sc} of 16.24 mA cm^{-2} and PCE of 8.17%, corresponding to enhancement for the 16.08% and 22.86%. C2 possess a J_{sc} of 15.32 mA cm^{-2} and PCE of 7.88%, the increase for the 9.5% and 18.49%; and C1 have a J_{sc} of 14.30 mA cm^{-2} and PCE of 6.89%.

To confirm the utilization of NIR light of the upconversion layer in DSSCs, the J - V curves of three DSSCs with upconversion assistant layer and normal DSSCs under 980 nm light were measured, as

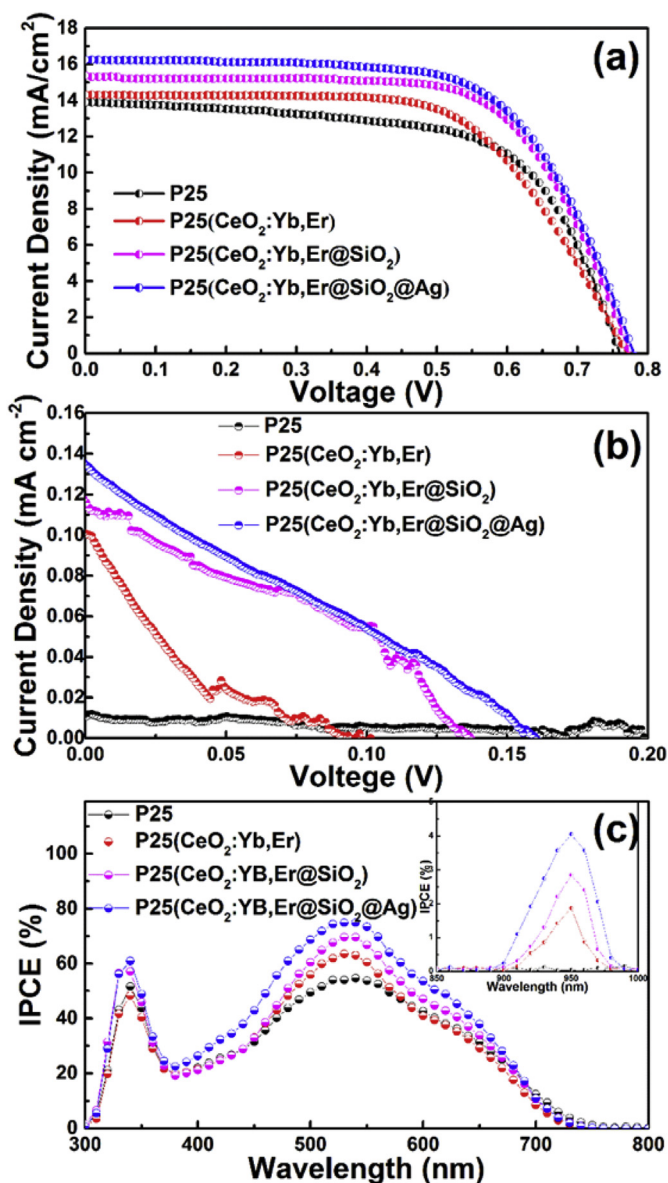


Fig. 4. The current density-voltage (J - V) characteristics of different fabrication DSSCs (a) under AM1.5G simulated sunlight irradiation (100 mW cm^{-2}) and (b) under 980 nm NIR laser, (c) IPCE curves of different fabrication DSSCs (inset is the IPCE curves of different fabrication DSSCs at the near-infrared wavelength from 850 to 1000 nm).

shown in Fig. 4b. It is clearly shown that the photocurrent for C4 is about close to zero, however, the DSSCs fabricated by $\text{CeO}_2:\text{Yb,Er}$ based composite materials exhibit some photoresponse. Especially, $\text{CeO}_2:\text{Yb,Er}/\text{SiO}_2@\text{Ag}$ UCNFs generated 0.135 mA cm^{-2} photocurrent. The above results confirmed that the enhancement of power conversion efficiency by double-shell $\text{CeO}_2:\text{Yb,Er}/\text{SiO}_2@\text{Ag}$ UCNFs was predominantly due to the upconversion progress for broadband harvesting of NIR sunlight.

The incident photon-to-current conversion efficiency (IPCE) was investigated to further analyze the photovoltaic performance of four DSSCs, as shown in Fig. 4c. IPCE values of four DSSCs are increased in the order of $\text{P25}(\text{CeO}_2:\text{Yb,Er}/\text{SiO}_2@\text{Ag}) > \text{P25}(\text{CeO}_2:\text{Yb,Er}/\text{SiO}_2) > \text{P25}(\text{CeO}_2:\text{Yb,Er}) > \text{P25}$. Under the assistance of double shells, i.e. SiO_2 layer and Ag NPs, C3 shows the highest IPCE value, which is ascribed to the upconversion ability of $\text{CeO}_2:\text{Yb,Er}/\text{SiO}_2@\text{Ag}$ UCNFs. More interestingly, C1, C2, and C3 all showed an

improved broad-band NIR response at 900–1000 nm wavelength range (the inset of Fig. 4c), clearly corresponding to C4 without NIR response. Therefore, the upconversion assistant layers treated cell display broader light responses and higher IPCE values owing to superior upconversion luminescence ability.

4. Conclusion

In summary, double-shell $\text{CeO}_2:\text{Yb,Er}@/\text{SiO}_2@\text{Ag}$ upconversion composite nanofibers were successfully synthesized by a multi-step process. $\text{CeO}_2:\text{Yb,Er}@/\text{SiO}_2@\text{Ag}$ UCNFs show the excellent upconversion luminescence property. The increase of luminescence intensity should be attributed to the following three factors: (1) the SiO_2 coatings act the role of light enrichment, and break the crystal symmetry of CeO_2 and allow f-f transitions, which result in the improved upconversion property; (2) Ag NPs display the SPR effect, which further increases upconversion property of $\text{CeO}_2:\text{Yb,Er}$; (3) the SiO_2 layers provide an adjustable distance between fluorescence center and Ag NPs to prevent the luminescence quenching. When $\text{CeO}_2:\text{Yb,Er}@/\text{SiO}_2@\text{Ag}$ were used as an assistant layer in DSSCs, the PCE was higher than normal cells. This work widens the light absorption range of DSSCs, improves the utilization ratio of solar light, and provides a feasible scheme for increasing the efficiency of DSSCs photoelectric conversion.

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Appendix A. Supplementary data

Supplementary data related to this article can be found at <https://doi.org/10.1016/j.jallcom.2018.07.225>.

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Rongfang Zhao

School of Chemistry and Chemical Engineering, Yangzhou University,
Yangzhou 225002, PR China

Yantai Institute of Coastal Zone Research, Chinese Academy of
Sciences, Yantai 264003, PR China

Qianhui Wu

School of Chemistry and Chemical Engineering, Yangzhou University,
Yangzhou 225002, PR China

Dongmei Tang

School of Chemistry and Chemical Engineering, Yangzhou University,
Yangzhou 225002, PR China

Wenlong Li

School of Chemistry and Chemical Engineering, Yangzhou University,
Yangzhou 225002, PR China

Xiue Zhang

School of Chemistry and Chemical Engineering, Yangzhou University,
Yangzhou 225002, PR China

Ming Chen*

School of Chemistry and Chemical Engineering, Yangzhou University,
Yangzhou 225002, PR China

Rong Guo

School of Chemistry and Chemical Engineering, Yangzhou University,
Yangzhou 225002, PR China

Guowang Diao**

School of Chemistry and Chemical Engineering, Yangzhou University,
Yangzhou 225002, PR China

* Corresponding author.

** Corresponding author.

E-mail address: chenming@yzu.edu.cn (M. Chen).

E-mail address: gwdiao@yzu.edu.cn (G. Diao).

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