

Nd:GGG Nanopowders by Microwave Gel Combustion Route and Sinterability Studies

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ABSTRACT

Synthesis of Nd_{0.03}Gd_{2.97}Ga₅O₁₂ (Nd:GGG) nanopowder was carried out by microwave-assisted nitrate-citrate gel combustion technique. Various nitrate-to-citrate ratios from stoichiometric-to-fuel lean were explored. Gels were combusted by microwave heating and the combusted powders were calcined at 900 °C for 2 h. Fourier Transform Infra-red Spectroscopy (FTIR) and X-ray Diffraction (XRD) of calcined nanopowders showed phase pure Nd:GGG formation, from stoichiometric-to-fuel lean nitrate-to-citrate ratio of 1 to 0.416. Particles in the size range of 150 nm - 200 nm were obtained for stoichiometric ratio. Highly uniform, spherical morphology, with size range 90 nm - 100 nm, were obtained in fuel lean ratio of 1 to 0.416. Sintering of these nanopowders at 1550 °C for 2 h in air resulted in retention of phase purity as observed by XRD. Grain growth of less than 2 μm, for fuel lean ratio of 1 to 0.416, indicated formation of highly sinterable Nd:GGG nanopowders.

Keywords: Phase pure GGG, citrate-nitrate, microwave combustion, XRD

1. INTRODUCTION

Nd:GGG has potentially very important applications in solid state high capacity laser^{1,2} due to its excellent optical and mechanical properties³. GGG has high refractive index (n=2) and transparency in visible-to-far IR, making it a very useful material for solid state lasers. Nd:GGG has high fracture strength, good chemical stability, high thermal conductivity, large rare earth doping ratio, and is easy to grow⁴. This material is a better substitute to the most widely used solid state laser Nd:YAG due to certain reasons⁵. However laser quality single crystal Nd:GGG is very difficult to grow due to certain reasons⁶⁻⁸. But with the development of ceramic technology, Nd:GGG transparent ceramic for laser applications is a challenge and new research trend for the materials scientists worldwide^{9,3}. To achieve high density and transparent polycrystalline Nd:GGG, fine powders without or only with slight agglomeration are necessary¹⁰. GGG synthesis by co-precipitation has been reported using 2 per cent additional Ga³⁺ ions to compensate the loss of Ga³⁺ ions during washing and high temperature calcinations to GGG^{8,11,12}.

Synthesis of GGG for several gel combustion routes using citric acid have been reported^{13,14}. But detailed explorations of nitrate-citrate ratios to give highly sinterable nanopowders have not been reported. There are several reports on synthesis of Nd:YAG¹⁵ and Nd:Ytria¹⁶ nanopowders by citrate/alanine nitrate microwave gel combustion route, but there is no report on synthesis of GGG/Nd:GGG nanopowders by microwave

gel combustion route. Further sintering studies of Nd:GGG nanopowders with reference to retention of phase purity has not been explored. The ratio of complexing agent-to-metal cations is an important parameter because it decides the amount of organics to be removed during calcinations, which in turn affects the ceramic properties^{17,18}.

In the present work, synthesis of Nd:GGG nanopowders using different nitrate-to-citrate molar ratios by microwave gel combustion technique is being reported for the first time. The effects of different nitrate-to-citrate ratios on phase evolution, size and shape of the Nd:GGG particles were studied.

2. EXPERIMENTAL

Four sets of sols were prepared by taking Nd₂O₃: Gd₂O₃ and Ga₂O₃ in their stoichiometric ratios of Nd_{0.03}Gd_{2.97}Ga₅ and dissolved in HNO₃ and then mixed with citric acid to obtain metal nitrate-to-citrate ratios of 1: 0.208, 1: 0.416, 1: 0.833, and 1: 1.666. The ratios 1: 0.208, 1: 0.416 and 1: 0.833 are fuel lean and 1: 1.666 is stoichiometric by equivalence theory¹⁹. These reaction mixtures were kept in an ordinary oven for gelation at 110 °C for 10 h. Gels were further treated in domestic microwave modified for exhaust gases, operating at 2.45 GHz, with output power of 900 W. The gel underwent drying and subsequent rapid combustion within 5 min - 10 min for 10 gm batch in the microwave oven, giving fluffy powder precursors. The precursors formed were calcined at 900 °C for 2 h. Compaction of the powders was carried out by uniaxial

press at 34 MPa followed by cold isostatic press at 300 MPa. The compacts were sintered at 1550 °C for 2 h in air.

FTIR spectra were recorded on FTIR Spectrometer (Bruker, model Vector 22) using KBr matrix. XRD was carried out on PANalytical X'pert Pro from 2θ value 15° to 75° for characterisation of phase purity. Diffused reflectance measurement on Nd:GGG nanopowders was carried out on an ultraviolet–visible–near infrared (UV–VIS–NIR) spectrophotometer (Varian Cary 5000). Transmission electron microscopy (TEM) was done on 200 kV, JEOL TEM-2100. Scanning electron microscopy (SEM) of the gold-coated surface of ceramic was carried out on Zeiss EVO 40. The emission spectra of specimens were recorded using 808 nm 3 W diode laser as the excitation source and the emission was coupled to a monochromator (Acton SP2300) attached with a *InGaAs* detector. The spectral measurements were carried out at room temperature within the spectral region of 900 nm - 1200 nm with 0.1 nm resolution. Dynamic light scattering (DLS) analysis was done by taking approx- 0.1 gm sample dispersed in 15 ml ethanol and sonicated for 15 min, using Nanoplus Zeta/nano particle analyzer.

3. RESULTS AND DISCUSSION

FTIR spectra of microwave combusted precursors (Fig.1) showed weak absorption bands between 850 cm^{-1} and 900 cm^{-1} caused by the CO_3^{2-} produced during the process. The absorption peaks between 1350 cm^{-1} to 1400 cm^{-1} resulted from vibration of C-O and C-C bond. Broad absorption bands between 400 cm^{-1} and 800 cm^{-1} were observed which indicate that Nd:GGG formation has not taken place. FTIR spectra of the microwave precursor calcined at 900 °C for 2 h (Fig.2) showed characteristic peaks of Nd:GGG at 671 cm^{-1} , 615 cm^{-1} , and 578 cm^{-1} ^[20,3] for nitrate-to-citrate ratio of 1: 0.416, 1: 0.833, and 1: 1.666 but not for 1:0.208 which consists of least ratio of citric acid.

From XRD (Fig. 3) pattern of the samples calcined at 900 °C for 2 h, it can be seen that for nitrate-to-citrate ratio of 1: 0.416, 1: 0.833, and 1: 1.666 the positions of diffraction peaks are in accordance with those of standard GGG-PDF 880-574. But the diffraction peaks in nitrate-to-citrate ratio of 1:0.208 do not agree with those of standard GGG with additional peaks of Gd_2O_3 . It may be due to the evaporation of Ga_2O_3 during combustion because of less complexation of fuel with Ga^{2+} ions, leading to disturbance in the 3:5 ratio of $\text{Gd}^{3+}:\text{Ga}^{3+}$ ions,

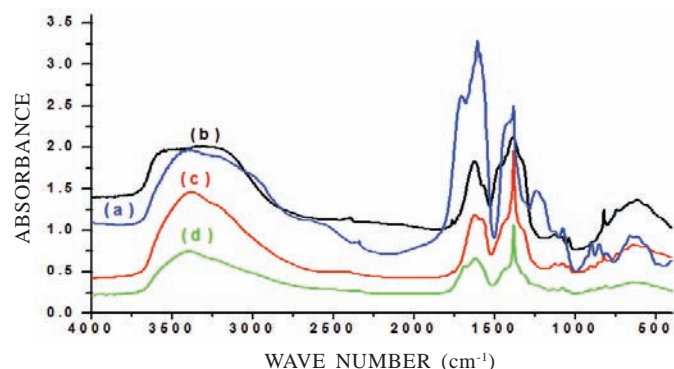


Figure 1. FTIR of microwave precursors with nitrate-to-citrate ratios (a) 1.666, (b) 0.833, (c) 0.416, (d) 0.208.

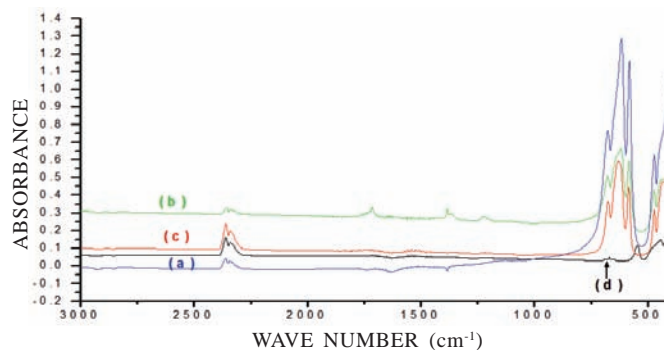


Figure 2. FTIR after calcination at 900 °C for 2 h with nitrate-to-citrate ratios (a) 1.666, (b) 0.833, (c) 0.416, (d) 0.208.

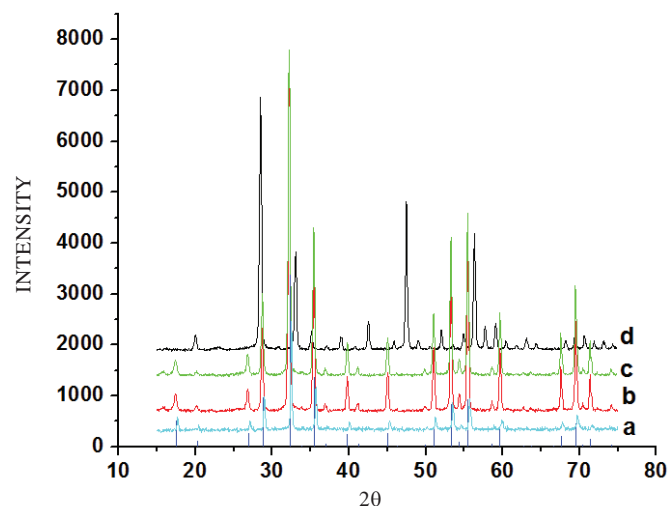


Figure 3. XRD after calcination at 900 °C for 2 h with nitrate-to-citrate ratios (a) 1.666, (b) 0.833, (c) 0.416, (d) 0.208.

resulting in non-formation of phase-pure GGG. Considering the cubic symmetry of Nd:GGG the lattice parameter, a , was calculated by least square method using the following equation:

$$a = d_{hkl} (h^2 + k^2 + l^2)^{1/2}$$

The lattice constant for Nd:GGG, as calculated, was found to be 12.3876 Å which is larger than 12.3835 Å for undoped GGG²¹. This indicates substitution of gadolinium by neodymium. The lattice parameter increases because neodymium ion has bigger size than gadolinium ion²². Diffused reflectance spectra of Nd:GGG in Fig. 4 showed all the transitions peaks of neodymium ion in GGG²³.

Room temperature emission spectra of Nd:GGG nanopowder measured under 808 nm excitation in Fig. 5 shows strongest fluorescence emission of Nd^{3+} ions at 1062 nm corresponding to ${}^4\text{F}_{3/2} \rightarrow {}^4\text{I}_{11/2}$ transition²⁴.

TEM images of calcined Nd:GGG powders at 900 °C with nitrate-to-citrate ratio 1: 1.666, 1: 0.833, and 1: 0.416 were shown in Figs 6(a) to 6(c). It was observed that particle size in stoichiometric ratio is between 200 nm - 300 nm with a strong necking between the particles. In case of fuel lean ratios, the size of particle has reduced up to 80 nm - 100 nm in 1:0.416. In citrate-to-nitrate ratio of 1: 0.416, particles are relatively well

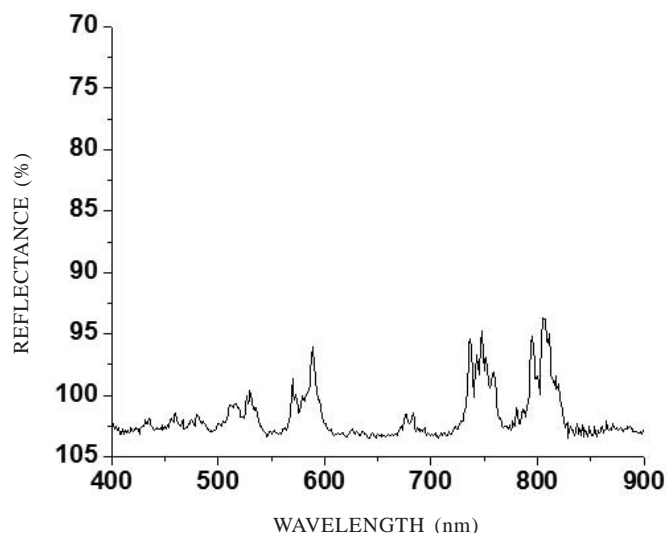


Figure 4. Diffused reflectance spectra of Nd:GGG after calcination at 900 °C for 2 h with nitrate-to-citrate ratio 1:0.416.

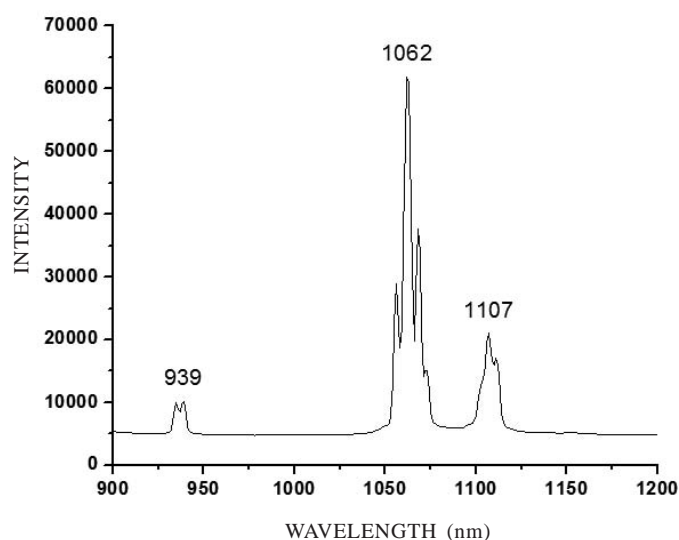


Figure 5. Photoluminescence of Nd:GGG nanopowder.

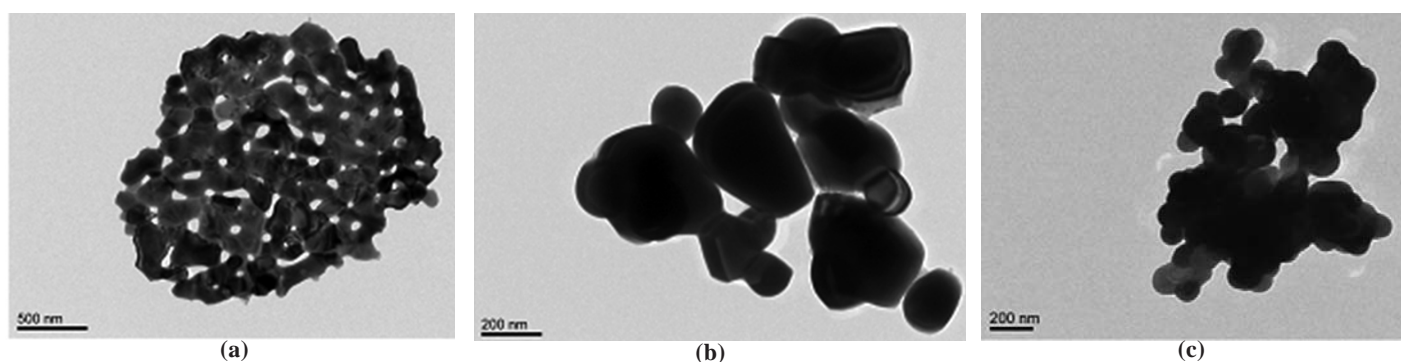


Figure 6. TEM of Nd:GGG after calcination at 900 °C for 2 h with nitrate-to-citrate ratio (a) 1.666, (b) 0.833, (c) 0.416.

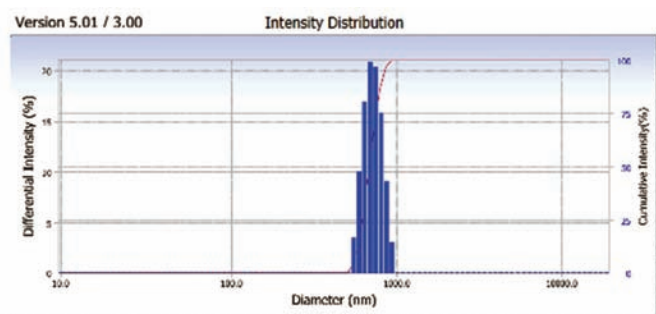


Figure 7. DLS showing agglomerate distribution.

and uniformly distributed, which is favourable for transparent ceramic fabrication. For stoichiometric nitrate-to-citrate ratio, large amount of carbon produced during microwave combustion led to big particle size during calcinations due to more localized heat compared to fuel lean ratios. Within fuel lean citrate-to-nitrate ratios, only for 0.416 best particle morphology and size range were obtained, indicating optimised ratio. Thus, metal nitrate-to-fuel ratio has direct influence on particle size. From DLS analysis, the agglomerate size was found to be less than 1 μm , indicating that nanosized particles of 80 nm - 100 nm tend to agglomerate due to van der Waals forces. Thus the agglomerate size has uniform distribution that led to uniform

sintering as observed from SEM of sintered ceramics.

Compaction and subsequent air sintering at 1550 °C for 5 h in air of phase pure Nd:GGG nanopowders with varied nitrate-to-citrate ratio of 1: 1.666, 1: 0.833, and 1: 0.416 showed good and homogeneous compaction with least porosity as observed by SEM in Fig.8. SEM showed highly homogeneous and least grain growth of less than 2 μm in case of nitrate-to-citrate ratio 1: 0.416. It complements with the TEM analysis in which this ratio showed smaller size and narrow size range, making these particles highly sinterable. XRD of the air-sintered ceramics showed phase purity retention as depicted in Fig. 9.

4. CONCLUSION

A detailed study on the phase purity and particle morphologies of Nd:GGG was carried out taking different ratios of metal nitrates-to-citric acid. Formation of phase pure Nd:GGG took place, from stoichiometric citrate-to-nitrate ratio up to fuel lean ratio of 1 : 0.416. However best particle properties, with regard to narrow particle size range and close to spherical and uniform particle morphology were achieved for citrate-to-nitrate ratio of 0.416. Least grain growth on air sintering at 1550 °C for 2 h was observed for citrate-to-nitrate ratio of 0.416. These highly sinterable powders are expected to give transparent ceramics.

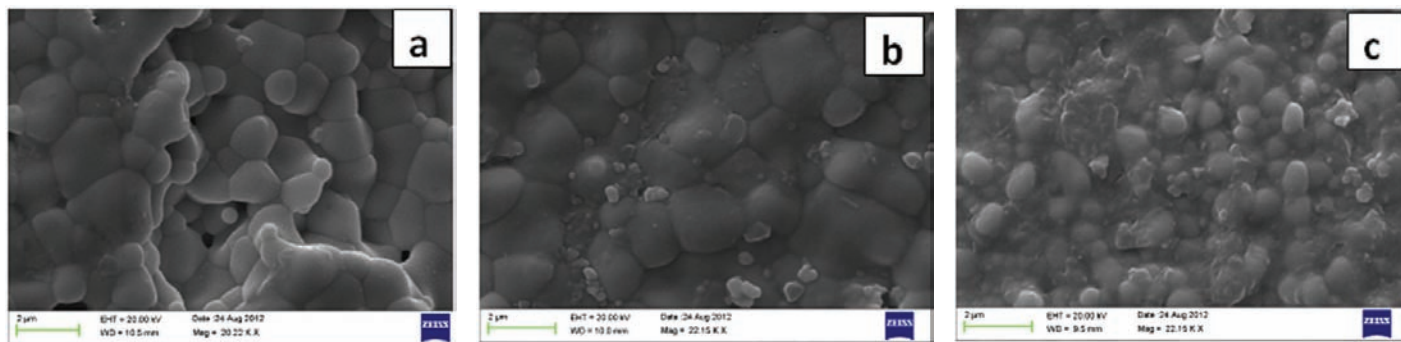


Figure 8. SEM after air sintering at 1550 °C for 2 h of Nd:GGG nanopowders compact with nitrate-to-citrate ratios (a) 1.666, (b) 0.833, (c) 0.416.

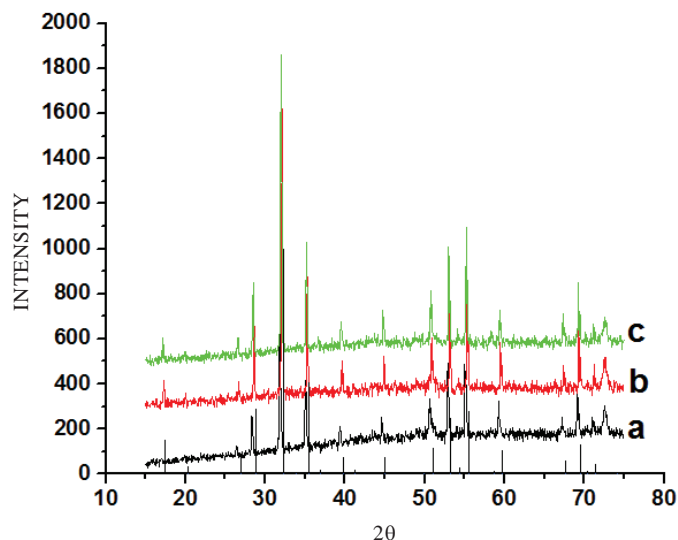


Figure 9. XRD after air sintering at 1550 °C for 2 hrs of Nd:GGG nanopowders compact with nitrate-to-citrate ratios (a) 1.666, (b) 0.833, (c) 0.416.

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