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# Studies of Process Parameters on Scale up of Nd:YAG Nanopowder Synthesis by Sol-gel Process

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### ABSTRACT

Scale up of process from 6 g to 200 g batch for preparation of phase pure Neodymium doped Yttrium Aluminum Garnet (*Nd*:YAG) nanopowder by sol-gel process using nitrate-citrate route was carried out. Optimisation of process parameters was done to resolve problems associated with large batch such as evolution of large amount of acidic fumes, increased gelation and drying time and incomplete calcination. The problem of exhaust of excess acidic fumes with size scale up was handled by passing ammonia and optimized pH of the sol. Process of drying the sol was optimised for large batch size. On scaling up from 6 g to 200 g batch incomplete calcination with size scale up was identified and resolved to achieve completely crystalline particles in the range of ~25 nm to 50 nm.

Keywords: Phase-pure Nd: YAG, sol-gel process, nitrate-citrate sol, XRD, calcination

### 1. INTRODUCTION

Neodymium doped YAG ( $Y_3Al_5O_{12}$ ) is an important solidstate laser material. *Nd*:YAG lasers have defence applications as rangefinders, target designators, and target interrogators. Fabrication of polycrystalline *Nd*:YAG lasers comparable to single crystals has been reported in the literature<sup>1</sup>. Fabrication of laser quality Nd:YAG ceramics by solid state reaction from oxide precursors has been reported<sup>2</sup>. Using YAG nanopowder synthesized by sol-gel method, fabrication of transparent YAG ceramics has been reported<sup>3</sup>.

For achieving transparent ceramics the starting nanopowder should have some ideal characteristics such as sub-micron particle size (<100 nm), narrow size distribution and homogeneity throughout the bulk<sup>4</sup>. To carry out detailed studies from nanopowders to ceramics there is always a need to synthesize a large batch in one go to prevent batch to batch variation in nanopowders properties.

We report the scale up of the synthesis of phase pure Nd:YAG from 6 g to 200 g by nitrate-citrate sol-gel process. The evolution of a large amount of hazardous nitrous fumes was one of the major problems during scaling up. Another problem faced was significant increase in the gelation time, drying time and incomplete crystallization. In the present work optimisation of conditions on scale up to 200 g batch are being discussed.

### 2. EXPERIMENTAL WORK

Yttrium nitrate,  $Y(NO_3)_3.6H_2O$  (99.9 % purity Alfa Aesar), Neodymium nitrate,  $Nd(NO_3)_3.6H_2O$  (99.9 % purity Alfa Aesar) Aluminum nitrate,  $Al(NO_3)_3.9H_2O$  (99.9 % purity Alfa Aesar) and Citric acid anhydrous (99.5+ % purity Alfa Aesar) were used as received. Two sols were prepared by taking stoichiometric amounts of nitrates corresponding to the molar ratio of  $Y_{2.94}Nd_{0.06}Al_5$  for 6 g and 200 g YAG respectively, which corresponds to 2 at.% of *Nd* for both the batches.Citric acid was used as the chelating agent. The total cation to citric acid was taken in the ratio of 1:1. The process route is given in the flow chart as shown in Fig. 1. The nitrate-citrate sol in millipore water was prepared by ultrasonic stirring. The pH of the sols was adjusted to 1 by passing  $NH_3$  gas. Gelation took place at 80 °C within 16 h. Drying of the gel was carried out from 110 °C to 150 °C in an oven modified for scrubbing ammonium nitrates in dil. *NaOH* soln. For 6 g batch the precursor powder was carried out at 900 °C for 2 h. And calcination of precursor was carried out at 900 °C for 2 h with flow of oxygen @ 4 litres/min.

Fourier transform infra-red spectroscopy (FTIR) was carried out by Bruker, (Vector 22) to identify the temperature at which *Y*-O and *Al*-O bond formation was taking place. X-ray diffraction (XRD) was carried out with the help of Philips X-Ray Diffractometer, PW for characterisation of phase purity and particle size.  $Nd^{3+}$  doping was characterised by EDX using scanning electron microscope (ZEISS EVO Series, EVO 50). Transmission electron microscopy (TEM) was done on FEI Philips Morgagni 268. The TEM specimens were prepared on copper grids by dispersing the powder ultrasonically in acetone.

### 3. RESULTS AND DISCUSSION

The amount of water required to dissolve the solid reactants is an important parameter in scaling up the process as it is directly related to the consumption of electricity and time in gelation and drying in sol-gel process. Taking 6 g batch

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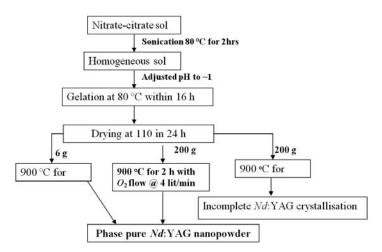


Figure 1. Flow chart of process route.

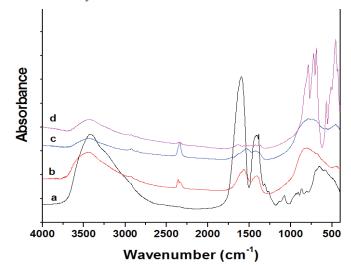
itself the water content was optimised by taking different w/w solid to total content percentage starting from 30 per cent, 40 per cent, and 50 per cent to 60 per cent till the solid contents dissolved easily without causing saturation. All these mixtures led to similar results in terms of phase purity and particle size. So 60 per cent was optimised for the larger batch as it was least time consuming during the process.

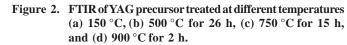
For 6 g batch total solid content was ~61 g and for 200 g it was ~2033 g. For 6 g batch from preparation of sol to drying of the gel reaction was carried out in a 1 litre beaker and optimised gelation and drying parameters as given in the process flow chart. For 200 g batch 5 litre beaker was taken for the same process but gelation and drying did not take place under those optimised parameters of 6 g batch. Gel was not formed at 80°C even within 48 hand drying also took place only in 3 days (72 h). The consumption of time and power on size scale up upto precursor stage was three times that of 6 g batch. To solve this problem the surface area of the apparatus was increased by transferring the sol into trays and carried out the process in the same oven under the optimised parameters of 6 g batch.

Further with increase in the batch size the increased continuous evolution of acidic fumes was tackled by converting the acidic byproducts into ammonium salt by passing  $NH_3$  to the sol. On passing  $NH_3$  even in slight excess [pH > 2] precipitation occurred which is supposed to create inhomogeneity in sol-gel

processes. So pH was optimised to 1 to prevent precipitation.

FTIR spectra depicted complete formation of YAG on scale up from 6 g batch to 200 g batch. FTIR was done of gels (Fig. 2) treated from 150 °C to 900 °C. In the spectrum of the gel at 150 °C there was no peaks in the M-O region, the bands at ~ 1562 cm<sup>-1</sup> and ~ 1384 cm<sup>-1</sup> can be assigned to the ionized carboxylates and carbonates resulting from the rupture of citrate network<sup>5</sup> and intensities of these peaks reduced significantly at 500 °C along with a broad band in the M-O region. At 750 °C the intensities of the peaks corresponding to the organic residues had further reduced and that in the M-O region had become more prominent. At 900 °C, the broad band in the region of 800 cm<sup>-1</sup> - 400 cm<sup>-1</sup> is replaced by several peaks at 784 cm<sup>-1</sup>, 719 cm<sup>-1</sup>, 688 cm<sup>-1</sup>, 566 cm<sup>-1</sup>, 511 cm<sup>-1</sup>, 455 cm<sup>-1</sup> and 428 cm<sup>-1</sup> which may be attributed to the M-O vibrations in YAG<sup>6</sup>. The broad peak at  $\sim$ 3300 cm<sup>-1</sup> in b, c, and d (Fig.2) corresponds to adsorbed moisture on the powders which may be removed by treating the powder at  $\sim 100$  °C before sampling for FTIR analysis.





For 6 g batch XRD patterns of *Nd*:YAG (Fig.3) calcined at 800 °C/4 h showed the characteristic peaks (JCPDS 33-40) but the background intensity was very high indicating incomplete

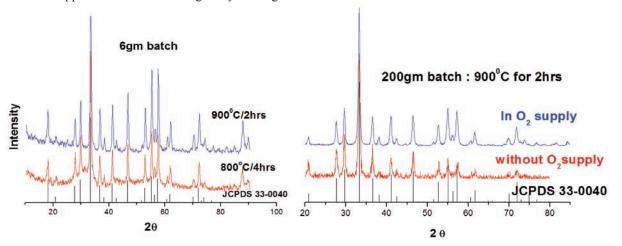


Figure 3. XRD depicting complete crystallisation to phase pure YAG.

crystallization. At 900 °C for 2 h complete crystallization to phase pure *Nd*:YAG was observed. But for 200 gm batch, on treatment at 900 °C for 2 h in the same furnace, due to the large amount of precursor, the atmospheric oxygen present inside the furnace chamber was not sufficient to burn out all the organic residues present in the precursor. Hence crystallization was not complete and presence of oxygen deficient phase i.e. *YAIO*<sub>3</sub> (at 20 = 44 °) was indicated in XRD pattern. However on treatment at 900 °C with flow of oxygen at the rate of 4 lit/min complete crystallization to phase pure cubic *Nd*:YAG structure was obtained. Particle size was calculated using Scherrer's equation<sup>7</sup>.

$$t = \frac{0.9\lambda}{\left(\beta_{sample}^2 - \beta_{inst}^2\right)^{1/2}\cos\theta}$$

where *t* is the crystallite diameter,  $\lambda = 1.54056$ Å,  $\theta$  is the diffraction angle,  $\beta_{\text{sample}}$  is the FWHM of the diffraction peak and  $\beta_{\text{inst}}$  is characteristic of the instrument. Three strong peaks were chosen (at  $2\theta = 29.6^{\circ}$ ,  $33.3^{\circ}$  and  $46.5^{\circ}$ ) to calculate the paticle size. The average primary particle size was ~ 25 nm for both the batches. The lattice parameter, *a* of cubic YAG is 12.008 Å<sup>8</sup>. For 2 at.% Nd doped YAG the lattice parameter was calculated to be 12.020. This increase in lattice parameter is indicative of  $Nd^{3+}$  replacing  $Y^{3+}$  because the effective radius<sup>9</sup> of  $Nd^{3+}$  is greater than that of  $Y^{3+}$ .

From EDX spectrum presence of *Nd* in the powder was detected for both the batches. Fig. 4 depicts the EDX spectrum which is representative of both the batches. When rare earth elements are involved the EDX is reliable for only qualitative analysis and accurate quantitative analysis of *Nd*: YAG has never being discussed in the literature. In the extensive work on *Nd* doping upto 24 at.% elemental composition is discussed only by the increase in the peak of *Nd* in EDX graphs and increase in lattice parameter<sup>10</sup>. TEM (Figs 5 & 6) showed particle size in the range of 25 nm-50 nm for both 6 g as well as 200 g batch indicating efficiency of scale up without distortion in particle morphology and size.

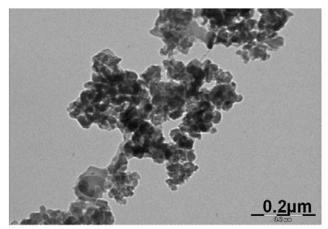


Figure 5. TEM of Nd:YAG nanopowder of 6 g batch.

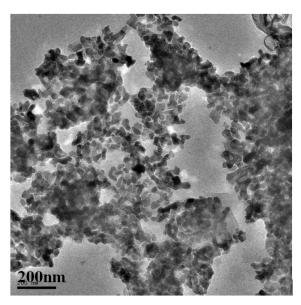


Figure 6. TEM of Nd:YAG nanopowder of 200 g batch.

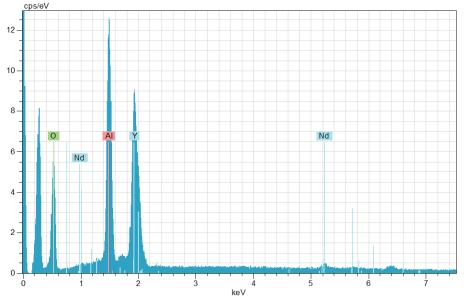


Figure 4. EDX showing presence of Nd.

### 4. CONCLUSION

A laboratory scale large batch (200 g) of phase pure *Nd*:YAG was synthesized by sol-gel process using nitratecitrate sol.

- Phase pure *Nd*:YAG was obtained for both 6 g and 200 g batch at 900 °C for 2 hunder oxygen supply.
- XRD and FTIR confirmed the formation of phase pure product in both the batch sizes.
- Particle sizes obtained were in the range of ~25 nm 50 nm by TEM.

By taking into consideration above discussion it can be concluded that the method has been optimised for large scale synthesis of *Nd*:YAG efficiently.

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