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Investigation of rice husk derived activated carbon for removal of nitrate contamination from water

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Abstract

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Development of porous carbons with high specific surface area (more than 1200 m g⁻¹) targeted at 14 nitrate removal from aqueous solutions is investigated by chemical activation of carbonized rice husk. 15 Potassium carbonate is used as activating and desilicating agent. The effect of post-synthetic treatment by 16 gas phase ammoxidation with ozone/ammonia or oxidation with concentrated nitric acid followed by 17 nitrification with urea on main physicochemical properties and on the effectiveness of the activated carbons 18 in nitrate removal is compared with those determined for a pristine activated carbonized rice husk sample. 19 The two-fold enhancement of nitrate removal by the urea-modified activated carbon in comparison with 20 pristine and ammoxidated sample is in direct correlation with the development of surface basic groups. 21

Highlights:

• Carbonized rice husk was activated yielding activated carbons with high surface area;

• Activated carbons (AC) were nitrified through ammoxidation or modification with urea;

• The nitrified AC surface basicity act as adsorption sites towards nitrate ions.

Keywords: biomass, rice husk carbonization, activated carbon, surface modification, nitrogen functionalization, nitrate ions adsorption.

1. Introduction

Nitrate is a potentially hazardous ion that is naturally present in groundwater. In addition, municipal, 33 agricultural and industrial wastewater seepage enhance nitrate groundwater contamination. Water pollution 34 due to excessive discharge of nitrate has become a significant environmental problem worldwide [1, 2]. The 35 effect of nitrate on human health is described as *primary toxicity* [3]. Excess nitrate in drinking water may 36 cause severe health problems such as infant cyanosis syndrome ("blue baby syndrome") due to 37 methemoglobinemia [4, 5], central nervous system birth defect [6], and lead to possible formation of 38 nitrosamines, which can cause various types of human cancer and other diseases [7-10]. An excess level of 39 40 nitrate in groundwater can lead to eutrophication (harmful algae bloom, fish hypoxia and toxin production) 41 [11].

The water denitrification is a complex and expensive process considering the solubility and stability 42 of nitrate ion as well as its low adsorption affinity [12]. Current denitrification technologies include 43 biologically-assisted [13] and physicochemical treatments. Biological methods require a large-scale 44 installation and have slow ion removal kinetics. As for physicochemical methods, adsorption on activated 45 carbon (AC) and ion-exchange resins were recognised as most effective for drinking water denitrification 46 [14, 15]. However commercial activated carbons and synthetic resins are costly and the guest for other 47 adsorbing materials has continued. Recently, biomass, as a renewable source, has attracted attention for the 48 49 preparation of porous carbons [16]. Traditionally activated carbons have been produced from agricultural waste products such as coconut shells, hazelnuts, cherry stones or apricot kernels which has kept production 50 costs to a minimum [17-23]. Nevertheless, the literature on the nitrate uptake by ACs developed from 51

agricultural waste is quite poor [24] and only few studies have reported the adsorption of nitrates from water 1 2 using such ACs (See Supplementary information, SI, Table S1). Rice husk (RH), as an abundant by-product of rice milling industry, with low combustion value and potential damage to the environment, brings the 3 issue of its sustainable utilization to the research hotspot [25-28]. Rice husk is composed of lignin, cellulose 4 5 and SiO₂, hence it can be considered as a natural organic-inorganic composite material [29]. Carbonization/activation of rice husk in inert atmosphere yields highly porous carbons with large specific 6 surface area [30, 31]. Recently, porous carbons prepared from rice husk attracted attention in the field of 7 energy storage [32], solar technology [16] and environmental protection [33]. Further research has been 8 9 directed towards post-carbonization modification methods such as acid treatment [34] and introduction of positively charged basic functional groups [35, 36] that have capacity for removing oxyanions such as 10 nitrate. 11

In 2017 Kazakhstan (KZ) produced 291,000 metric tons of rice, 20% of which is RH [37]. The commercial value of RH does not exceed £60 per metric ton, and in fact most KZ rice producers offer it for free since it reduces their costs of waste disposal; therefore, one of the research objectives has been to investigate the possibility of producing novel porous carbons from locally available rice husk biomass.

In this work, rice husk was carbonized to produce carbonized rice husk (denoted as CRH). CRH was further subjected to potassium carbonate etching in order to desilicate the initial carbonized material and prepare a porous carbon (denoted as CRH-I). At the next step, CRH-I was nitrified via ammoxidation or modification with urea (samples denoted as CRH-II and CRH-III, correspondingly). All three samples of porous carbons (CRH-I, CRH-II and CRH-III) have large specific surface area and well developed porosity, and CRH-III also demonstrated elevated anion adsorption capacity.

2. Experimental

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2.1. Preparation of activated carbons

AC samples were obtained from rice husk collected in the Bakanas Village, Balkhash District of Almaty Region via a procedure described elsewhere [38, 39]. Briefly, the RH material was initially washed and air dried at 105°C for 24 h in an oven and then pyrolyzed in air at 475 °C, in a spherical rotary steel reactor (SI, Fig. S1) for 30 min to yield 41 wt% of carbonized rice husk (CRH).

For the chemical alkaline activation and simultaneous removal of silica from the carbonized rice 30 husk, a sample of 175 g of CRH was impregnated in solution of potassium carbonate (K₂CO₃) for 24 h, in 31 order to facilitate the access of the alkaline inside the carbonized particles. The impregnation ratio of 1:4 32 $(g CRH / g K_2CO_3)$ was used producing the sample CRH-K₂CO₃. After impregnation the sample was 33 pyrolyzed in a cylindrical steel reactor placed inside a vertical electric furnace at 475 °C (SI, Fig. S2) and 34 activated at 950 °C for 1 hour in an inert atmosphere of argon. The resulting material was washed with 35 copious amount of distilled water until neutral pH, to achieve a complete removal of potassium silicate from 36 the carbonized sample, and then air dried overnight at 110°C. The weight of the obtained AC sample CRH-I 37 was 55.4 g giving a yield of 31.7 wt%. CRH-I was further surface functionalized by two different routes. 38 route 1 via mild ammoxidation yielding carbon CRH-II or route 2 in the two-step procedure yielding carbon 39 40 CRH-III as shown in Fig. 1.

Mild-temperature ammoxidation of CRH-I was carried out in a quartz tube reactor inside a horizontal electric furnace (SI, Fig. S3). The sample placed in a quartz boat was oxidized with ozone/air mixture at 130 °C for 24 h (at gas flow rate of 1 L/min), immediately followed by the reaction with gaseous ammonia in the same reactor (one-pot synthesis) at 350 °C for 5 h (at gas flow rate of 5-6 mL/min) to yield 93.1 wt% of the ammoxidized product CRH-II.

Scheme of Rice Husk Conversion into Activated Carbon

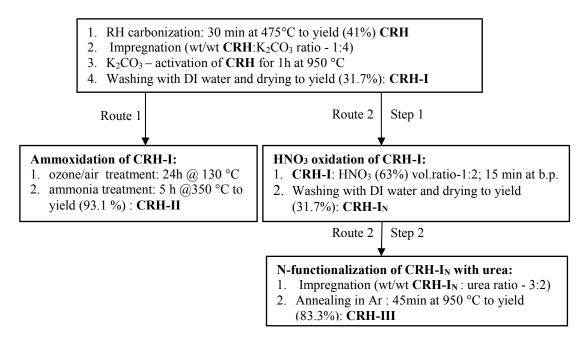


Figure 1. Block diagram of the activated carbon CRH-I modification routes.

5 Functionalization of activated carbon CRH-I by route 2 was carried out via oxidation with nitric acid, subsequent impregnation with urea and high-temperature treatment in inert atmosphere. Nitric acid 6 oxidation was performed according to the procedure described elsewhere [40]; briefly, CRH-I sample was 7 mixed with concentrated nitric acid (63 wt%), at the ratio of 1 g of AC per 5 mL of nitric acid solution, 8 which corresponds to 1:2 volume ratio) and the suspension was refluxed under mild agitation at 90 °C for 15 9 min. After oxidation, the samples were washed thoroughly with an excess of distilled water until neutral pH, and dried in an oven with a vield of 96 wt%; the resulting sample was denoted CRH-I_N. In a typical procedure, 15 g of oxidized AC product CRH-I_N was mixed/saturated with 10 g of urea (5.0 wt% in ethanol) at the AC : urea impregnation ratio of 3:2 (wt/wt), then dried and annealed at 950 °C in a quartz tube reactor placed in a horizontal electric furnace under Ar flow (at the flow rate of 5 L/h) for 45 min. Then sample was rapidly cooled to room temperature and the final product CRH-III was dried in an oven at 105 °C to constant weight of 12.5 g, corresponding to a yield of 83.3 wt%.

2.2 Characterization of activated carbons

2.2.1. Textural characteristics

The particles' surface morphology of the activated carbons was studied using field emission scanning 21 electron microscope (FE-SEM, Zeiss Sigma) at high vacuum and 5.00 kV EHT equipped with an energy-22 dispersive X-ray spectroscopy (EDX) detector to confirm the elemental composition of samples. The AC 23 samples were mounted onto aluminum stubs using black carbon tapes and sputter coated with platinum 24 (Sputter Coater Q150TES, Quorum, Italy). The Brunauer-Emmett-Teller (BET) surface area and porosity of 25 AC were determined from N₂ adsorption measurements. N₂ adsorption/desorption isotherms at 77K were 26 recorded using Autosorb-1 (Quantachrome, UK) in the range of relative pressures, p/p_0 , from 0.005 to 0.991. 27 Samples were outgassed for 10h at 150 °C prior to the measurements to remove any moisture or 28 contaminants adsorbed. The Autosorb-1 software was used to calculate BET specific surface area (SBET) of 29 samples by applying the BET equation to the adsorption data. The pore volume and size distribution were 30 calculated using the QS Density Functional Theory (QSDFT) for slit/cylindrical pores [41, 42]. The total 31 pore volume $V_{\rm T}$ was evaluated by summation of micropore ($V_{\rm mi}$) and mesopore ($V_{\rm me}$) volumes. The mean 32

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pore diameter, D_P , was calculated from equation $D_P=4V_T/S_{BET}$ [43]. The mercury porosimetry data were collected with a Poremaster (Quantachrome, UK) instrument.

2.2.2. Thermal analysis

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The thermal analysis was performed using a TGA analyzer Q500 (TA Instruments) coupled with mass
spectrometer (TGA-MS); measurements were carried out on 5 mg samples at a heating rate of 10°C/min in
nitrogen atmosphere at gas flow rate of 120 mL/min; weight losses and corresponding temperatures were
recorded and analyzed. Thermal decomposition data were obtained using differential scanning calorimetry
(DSC, Perkin Elmer 6000 simultaneous thermal analyzer) in the heating range from 50 to 500°C at a heating
rate of 10 °C/min.

2.2.3. X-ray photoelectron spectroscopy

15 X-ray photoelectron spectroscopy (XPS) was performed using an ESCALAB 250 Xi system (Thermo 16 Scientific) with a monochromated X-ray source of Al K α X-rays (1486.6 eV), a hemispherical electron 17 energy analyser, magnetic lens and video camera for viewing the analysis position. Wide scans (step size 1 18 eV, pass energy 200 eV) and narrow scans (step size 0.1 eV, pass energy 20 eV) of N1s (binding energy, BE 19 ~399 eV), C1s (BE ~285 eV), O1s (BE~531 eV), and Si 2p (BE ~100 eV) regions were acquired from 20 three separate areas for each carbonaceous sample. The XPS data were transmission function corrected and 21 analysed using Thermo Avantage Software (Version 5.952) with a smart background.

2.2.4. Boehm titration

Surface functionality of AC samples was assessed via modified Boehm titration procedure [44-47] using 25 pH meter Orion 410A (Thermo Scientific) equipped with a Glass Combination Electrode (BDH) and a 26 titration step of 0.1 titrant volume. In each case 0.100 g of activated carbon was shaken with 20 mL solution 27 of 0.025 M NaOH or 0.05 M HCl in 0.1 M NaCl overnight. Then the supernatants were back-titrated with 28 HCl or NaOH, respectively, to determine the surface functionality; analysis was performed in triplicates for 29 each sample. The total number of acidic (carboxylic, phenolic and lactone) groups was determined by their 30 neutralization with sodium hydroxide. The number of basic sites was calculated from the amount of HCl 31 required for their neutralization. The quantity of surface functional groups was expressed as milliequivalents 32 (meq) per gram of sample. 33

2.2.5. Adsorption experiments

37 To analyse the adsorption capacity of the synthesised samples analytical grade sodium nitrate (NaNO₃, Sigma-Aldrich) and DI water were used to prepare stock solutions of the nitrate anion (NO₃⁻). Nitrate anion 38 uptake was studied in batch reactor experiments at room temperature (25±0.5°C). A known quantity of 39 40 sample (0.05 g) was added to 50 mL aqueous solutions (5 or 15 ppm) and shaken continuously at 100 rpm, for 24 h. The ion chromatography system, Dionex ICS-1100 (Thermo Fisher Scientific) was employed to 41 determine nitrate concentration, a standard method for analysis of inorganic anions in water and wastewater 42 [48]. Prior to the analysis all eluents were filtered through the regenerated cellulose syringe filter (0.2 μ m) to 43 prevent the chromatographic column clogging. Each experiment was triplicated under identical conditions. 44 The amount adsorbed at time t, q_t (mg/g), was calculated using the following equation: 45

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$$q_t = \frac{(C_0 - C_t) \times V}{m} \tag{1}$$

47 where C_o and C_t (mg/L) were the initial and equilibrium NO₃⁻ concentrations in the aqueous solution; V (L) 48 was the volume of the experimental solution and m (g) was the mass of dry adsorbent used.

50 3. Results and Discussion

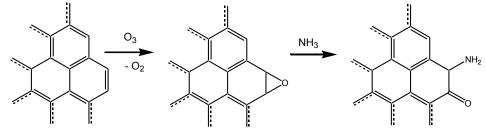
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Since higher RH carbonization temperature causes the reduction of yield and C-content thus increasing SiO₂-content in the resulting CRH, the temperature for RH carbonization most suitable for increasing the carbon content was experimentally determined at 475 °C. The pyrolytic regime has an effect on the successive K_2CO_3 -activation step, which comprises both processes of carbonization and desilication/activation (Equations 1.1 – 1.3) [49].

 $K_2CO_3 + SiO_2 \rightarrow K_2SiO_3 + CO_2\uparrow$ (1.1) $CO_2 + C \rightarrow 2CO\uparrow$ (1.2) $K_2CO_3 + 2C \rightarrow 2K + 3CO\uparrow$ (1.3) 10

The reactions (1.1) and (1.2) effectively start ca. 891° C, the melting point of K₂CO₃. The evolution of gaseous CO₂ and CO vacating the system facilitates the diffusion and soaking of the molten K₂CO₃ in the carbon matrix. Moreover, at such a high temperature CO₂ is able to react with the carbon matrix causing socalled "physical" activation or "burn-off" according to the reaction (1.2). Finally, the net balance reaction (1.3) summarizes a cascade of the complex processes leading to potassium reduction and its eventual intercalation into the carbon matrix. All of these processes greatly contribute to the development of microand mesoporosity of the resulting AC.

The mechanism of nitrogen incorporation into the carbon matrix during AC surface ammoxidation via ozonolysis, hypothesized here, occurs via the Criegee mechanism of cycloaddition [50] followed by amination. According to this mechanism, formation of epoxide groups occurs in the first place with subsequent transformation of the unstable three-membered oxirane ring attacked by the strong nucleophilic molecule of ammonia resulting in creation of hydroxylic and amide groups (Fig. 2).



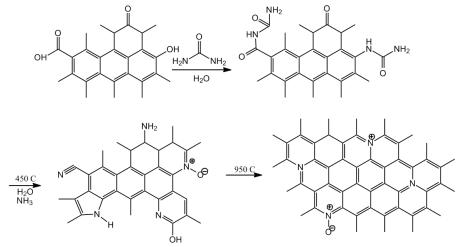


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Figure 2. Scheme of the ammoxidation reaction on the AC surface.

A high temperature nitrogen functionalization of CRH-I probably involves at least five stages of nitrogen transformation over the carbonaceous surface (Fig. 3). Prior to the reaction with urea the surface of CRH-I was activated with nitric acid resulting in generation of oxygen-containing functional groups. A reaction scheme of nitrogen incorporation into the carbon matrix using high-temperature treatment of nitric acid oxidized AC impregnated with urea in order to form pyrrole/pyridine functionalities is shown in Fig. 3 (adapted from [51]).



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Figure 3. The proposed mechanism of chemical transformations of the oxidized AC surface functional groups during the high temperature reaction with urea.

The resulting activated carbon structures strongly depend on such experimental parameters as the ratio of precursor to modifier, onset temperature of the rice husk decomposition and the temperature of final annealing. The morphology of the carbon materials obtained from rice husk was characterized by SEM (Fig. 4, a-c). The formation of distinctive plate-like structures was observed. These structural features start to appear at 950 °C (Fig. 4, a) and remain intact following amination with ammonia (Fig. 4, b) or urea treatment (Fig. 4, c).

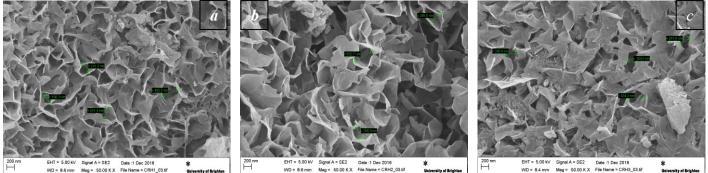


Figure 4. SEM images of activated carbon materials obtained from rice husk: CRH-I@950 (*a*), CRH-II (*b*) and CRH-III (*c*).

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The analysis of CRH treated with nitric acid (CRH-III sample) shows that whilst there was no indication of structure etching or significant changes in the structure at macroscale, there was a change in the surface roughness (Fig. 4, c) and chemical functionality confirmed by XPS and SEM/EDX analysis discussed later.

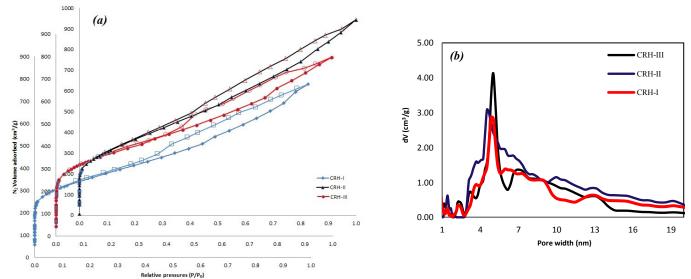


Figure 5. Low temperature nitrogen adsorption on CRH materials. Nitrogen adsorption/desorption
 isotherms at 77K (a) and their corresponding PSDs (b).

The nitrogen adsorption/desorption isotherms at 77K are shown in Fig. 5(a). All isotherms can be classified as type IV, according to IUPAC classification [52] with an uptake in the low pressure region $(p/p_0 \le 0.1)$ characteristic of microporous materials. However, each isotherm also shows a distinct hysteresis loop at intermediate to high relative pressures, which is characteristic of the presence of large micropores and mesopores (type IV); this is clearly shown on the graph of the pore size distribution (PSDs) in carbon samples (Fig. 5, b). A plate-like bulk with slit-like porous structure, observed by SEM (Fig. 4, a-c) is

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confirmed by **H3** type of hysteresis loop [52] in isotherms for all three CRH samples. The observed phenomenon of a low pressure onset of the hysteresis loop in adsorption isotherms of CRH samples is often reported for carbon materials [53, 54] and commonly ascribed to intercrystalline swelling. Still, the hysteresis loop closure evidences the extent at which after given adsorption cycle the carbon returns to its original state.

6 The textural properties of the CRH samples shown in Table 1, as determined from the analysis of the 7 nitrogen adsorption isotherms using BET and QSDFT methods (Fig. 5). The micropore volume, V_{mi} , 8 corresponds to the volume adsorbed at $p/p^0=0.1$ and the mesopore volume, V_{me} , was estimated as difference 9 between the volume of nitrogen adsorbed at $p/p^0=0.95$ and $p/p^0=0.1$. The pore size distribution was 10 calculated by QSDFT method assuming a slit pore model [55].

Sample	S _{BET}	S _{QSDFT}	V _{mi}	V _{me}	V_{T}	$V_{\rm mi}/V_{\rm T}$	D_{P}
	(m^2/g)	(m^2/g)	(cm^3/g)	(cm^3/g)	(cm^3/g)	(%)	(nm)
CRH-I	1260	1430	0.266	0.851	1.117	23.8	3.54
CRH-II	1320	1430	0.257	0.876	1.133	22.7	3.44
CRH–III	1330	1360	0.193	1.144	1.337	14.4	4.02

Table 1. Textural properties of activated carbons.

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As expected for this adsorption isotherm type, all samples are predominantly mesoporous and a 14 microporous contribution to the adsorption is in the range of ~14.4-23.8%. The ammoxidation did not have a 15 significant effect on the carbon porosity, and CRH-I and CRH-II samples have a very similar porous 16 structure (Fig. 4 and Table 1). It is interesting to note that although the nitric acid oxidation has little impact 17 on the morphology of CRH-III (Fig. 4, c), its effect on the porous structure of the activated carbon is 18 significant. The micropore volume is reduced whereas the mesopore volume is increased, probably owing to 19 a burn-off of micropores and converting them into mesopores. The mercury intrusion porosimetry data are 20 presented in Fig. 6. The low pressure part of the graph (Fig. 6, a) reflects the intrusion of mercury in the 21 interparticle space characterized by voids larger than 1 µm (the right side of the graph in Fig. 6, b). The 22 difference between samples CRH-II and CRH-III at high pressures corresponding to mercury intrusion into 23 sub-micron pores is small but both materials show higher volume of mercury intrusion than the original 24 sample CRH-I. These results which differ from low temperature nitrogen adsorption data suggest that 25 subjecting samples at high pressures may lead to crushing and fracturing of pore walls, which is known for 26 carbonaceous materials [56]. The synthetic approach described above also may lead to the formation of large 27 macropores. Thus high pressure mercury porosimetry data show larger pores than those detected by nitrogen 28 adsorption. 29



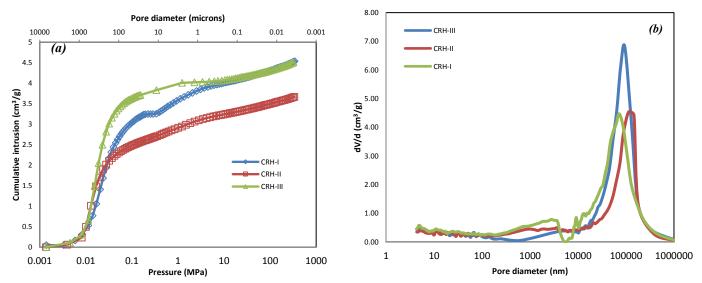


Figure 6. Mercury porosimetry intrusion data (a), and pore size distributions in CRH samples from mercury porosimetry data (b).

2 The MS evolution profiles (Fig. 7, a-c) provide a record of the ion current generated by different gases or molecular fragments with different molecular mass. All profiles were plotted vs the pyrolysis temperature (Fig. 7). The present study was focused on the main volatile and combustion products evolving in the temperature range from 25 to 800 °C. According to their m/z values, some of the gaseous products can be identified as H₂O (m/z = 18), hydrocarbons such as CH_3^+ , $C_2H_2^+$ and $C_2H_6^+$ (m/z = 15, 26 and 30 respectively); carbon oxides CO^+ , CO_2^{++} and CO_2^+ (m/z = 28, 22 and 44 respectively) and nitrogen containing species CN^+ (m/z = 26) and NCO⁺ (m/z = 42). The mass loss curves measured using TG analysis are shown in Fig. 7 (d).

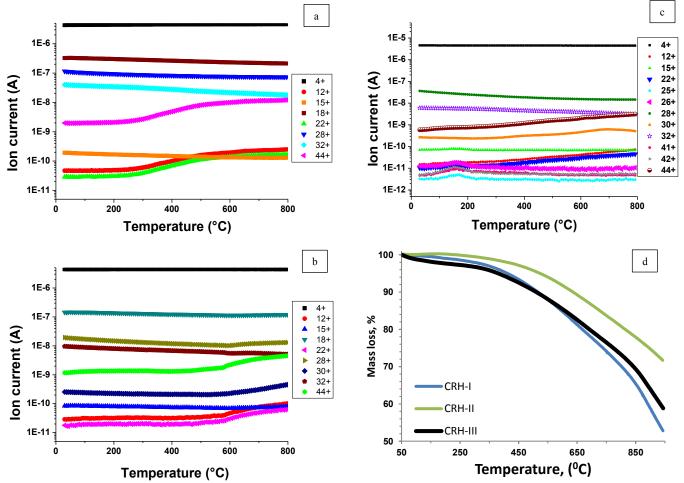


Figure 7. MS curves, an overview (y axis in logarithmic scale): CRH I (a), CRH II (b), CRH III (c) and TGA curves (d).

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11 Fig. 7(a) shows that CO₂ (m/z: 22 and 44) begins to evolve from CRH-I in the region between 300 and 500 °C, which corresponds to the change in the mass loss slope on TGA curve (Fig. 7(d)). The evolution 12 of carbon dioxide is caused by the destruction of oxygen-containing functional groups. MS spectrum for 13 CRH-III sample (Fig. 7(c)) has only one peak of volatile product evolution around 160 °C, together with 14 corresponding mass loss on TGA curve (Fig. 7(d)). This mass loss is attributed to the decomposition of urea 15 $(m/z: 22^+ - CO_2^{++}, 26^+ - CN^+ \text{ and } 42^+ - NCO^+)$, which fully decomposes at 250 °C [57]. Evolution of CO₂ 16 from CRH-II intensifies above 600 °C. 17

XPS analysis was performed to obtain a deeper understanding of the degree of chemical modification 18 of the CRH surface. The results of XPS survey scan analysis giving the atomic percentage of elements are 19 presented in Table 2 and supplementary information (SI, Fig. S4). The XPS results show that CRH-III and 20 CRH-II have a significant amount of nitrogen. The deconvolution of N 1s peak revealed the presence of four 21 different types of nitrogen-containing species N1, N2, N3 and N4 at 398.0, 399.4, 400.7 and 402.2 eV 22 23 respectively in samples CRH-II and CRH-III. These peaks could be assigned to certain types of nitrogen

structures, such as pyridine (N1), amine/imine/amide (N2), pyrrole (N3) and quaternary nitrogen (N4) [58]. 1 2 In CRH-II the N2 peak has the highest intensity, whereas in CRH-III peak N1 has the highest intensity 3 (Fig. S4). The starting CRH-I material also has some small amount of two types of nitrogen-containing species, N1 and N3 which could be attributed to nitrogen in pyridine and pyrrole rings. The distribution of 4 5 N-containing groups in CRH-III is different from CRH-II, which was expected owing to the different synthetic routes of these two carbons. As per sample description, nitrogen was not expected in sample 6 CRH-I. The amount of nitrogen present in CRH-I is below the detection limit of the instrument, i.e. below 7 0.1 atomic%. However, a small quantity of nitrogen was detected while carrying narrow scan analysis. All 8 samples have shown the presence of oxygen-containing functional groups and Si, the latter originating from 9 the raw material, rice husk. Si was not completely removed in the process of synthesis of CRH-II and 10 CRH-III. 11

Name	CRH-I	CRH-II	CRH-III	
Carbon (C) at. [%]	88.6 ± 0.8	90.3 ± 0.8	94.1 ± 0.6	
Oxygen (O) at. [%]	8.8 ± 0.4	6.7 ± 0.5	3.5 ± 0.3	
Silicon (Si) at. [%]	1.6 ± 0.2	1.1 ± 0.2	1.1 ± 0.2	
Nitrogen (N) at. [%]	0.3 ± 0.1	1.3 ± 0.1	1.3 ± 0.1	

Table 2. Chemical composition and relative atomic ratio determined by XPS of untreated CRH-I and
 nitrogen-functionalised CRH-II and CRH-III activated carbons.

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XPS data showed similar nitrogen content in CRH-II and CRH-III, which means that both 16 ammoxidation and impregnation with urea introduced N-containing functional groups. The results of XPS 17 (Table 2) and EDX analyses (Table 3) regarding the oxygen content are contradictory; according to EDX 18 analysis, sample CRH-III has the highest oxygen content whereas XPS data suggest the opposite. The other 19 discrepancy between the two methods is related to the silicon content. EDX analysis has found no silicon in 20 CRH-II, whereas, according to XPS data, its content is comparable in CRH-II and CRH-III. The reason for 21 these discrepancies is not clear, however it may be related to the difference in the chemical composition of 22 carbons on the surface and in the bulk, since these methods probe materials to different depths. This 23 explanation is indirectly confirmed by the fact that according to EDX analysis, no nitrogen was detected in 24 all three samples, whereas XPS showed its presence in both samples which were obtained by surface 25 modification with N-containing reagents, ammonia (CRH-II) and urea (CRH-III). Most likely, nitrogen 26 27 containing groups are located on the uppermost surface of carbon undetectable by EDX. In EDX analysis, the content of Si and O are correlated. It possibly means that they are bound as silica which is distributed 28 differently between the bulk and the uppermost layer of CRH samples. 29

The elemental composition of the carbon samples presented in Table 3, shows that all samples have a high carbon content, as expected for carbonized rice husk. The carbon content increased after ammoxidation, in agreement with the proposed reaction scheme (Fig. 2); in comparison with urea functionalization, ammoxidation completely removed silica and potassium from carbon (CRH-II). A small increase in the oxygen content from 4.6 wt% in CRH-I to 5.16 wt% in CRH-III was observed; this growth of the O/C atom ratio in CRH-III is related to an increase in the total amount of acidic groups created by oxidative nitric acid treatment [59].

Table 3. EDX elemental analysis of CRH-I	I, CRH-II and CRH-III samples.

A ativated earhour	Wt.% of the elements			
Activated carbons	С	0	Si	K
CRH-I	93.1	4.6	1.05	1.03
CRH-II	99.25	0.75	-	-
CRH-III	92.3	5.16	1.45	1.06

The functionalization of the activated carbons by the methods used in this work introduced O-containing groups along with N-containing groups [44].

4	Table 4. Total amount of acidic/basic groups on the surface of CRH carbons and their adsorption capacity
5	towards NO_3^- in solutions with initial concentration of 5 and 15 ppm (mean n=3, ±sd). Temperature 25 °C.
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Sample	Boehm acidic	Boehm basic	NO ₃ ⁻ adsorption, mg/g	
	groups, meq/g	groups, meq/g	5 ppm	15 ppm
CRH-I	1.20	1.65	0.87±0.053	3.76±0.047
CRH-II	1.36	1.90	0.90±0.036	3.80±0.033
CRH-III	0.90	3.40	2.20±0.032	8.11±0.031

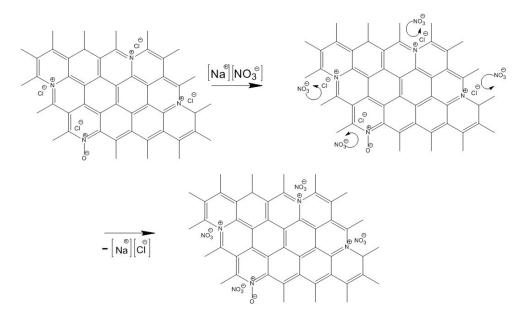
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Nitrate adsorption from aqueous solutions by samples CRH-I and CRH-II is lower than that of CRH-8 III (Table 4) and similar to other activated carbon materials [60]. The adsorption effectiveness of CRH-III 9 for nitrate was two times higher in the concentration range of 5-15 ppm studied in this work. A relatively 10 minor difference in the textural properties of activated carbons (Table 1) could not have such a strong effect 11 on the nitrate adsorption capacity of CRH-III. The results of acid-base titration (Table 4) suggest that urea 12 modification added more positively charged basic functional groups which according to XPS data could be 13 assigned to the pyridine-like nitrogen state (Fig. 3). Based on the acid-base titration data a schematic of the 14 interaction of nitrate ion with surface of CRH-III is proposed (Fig. 8). Since the ammonolysis of CRH-I did 15 not lead to an increased content of basic groups in CRH-II, it is likely that the dominant nitrogen species in 16 CRH-II is likely to be an amide-like state rather than amine-like state, the former having weak basic 17 properties. 18

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Figure 8. Proposed mechanism of nitrate ion removal by CRH-III as ion exchange.

The results presented show the impact of the activated carbon surface functionality on nitrate adsorption. Particularly, basic nitrogen containing groups on the surface introduced by modification with urea could act as adsorption sites for nitrate ions enhancing the adsorption capacity of CRH-III carbon towards nitrate.

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28 Conclusions and future perspectives

Effective porous carbons for removal of nitrate anions from aqueous solutions have been developed by 1 2 chemical activation of carbonized rice husk using potassium carbonate as activating and desilicating agent, 3 and further post-treated by gas phase ammoxidation with ozone/ammonia or oxidation with concentrated nitric acid followed by nitrification with urea. It has been demonstrated that introducing basic nitrogen-4 containing groups in the pyridine-like state into carbonized rice husk enhanced anion exchange properties of 5 activated carbon CRH-III which correlated with enhanced nitrate adsorption. Introduction of nitrogen-6 containing groups in the amine/amide-like state via low-temperature treatment (CRH-II) did not alter its 7 8 adsorption capacity towards nitrate compared to the initial sample (CRH-I). As revealed from literature, only 9 a few agricultural wastes have been investigated for nitrate removal and urea modified carbonized rice husk shows an appreciable potential for denitrification (Supplementary Table S1). Thus, it would be worth to 10 investigate a more wide range of nitrate concentrations and effectiveness in a presence of adsorptive 11 competing ions in real industrial effluents. 12

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