



# Review 3D-MID Technology for Surface Modification of Polymer-Based Composites: A Comprehensive Review

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Received: 27 May 2020; Accepted: 18 June 2020; Published: 23 June 2020



**Abstract:** The three-dimensional molded interconnected device (3D-MID) has received considerable attention because of the growing demand for greater functionality and miniaturization of electronic parts. Polymer based composite are the primary choice to be used as substrate. These materials enable flexibility in production from macro to micro-MID products, high fracture toughness when subjected to mechanical loading, and they are lightweight. This survey proposes a detailed review of different types of 3D-MID modules, also presents the requirement criteria for manufacture a polymer substrate and the main surface modification techniques used to enhance the polymer substrate. The findings presented here allow to fundamentally understand the concept of 3D-MID, which can be used to manufacture a novel polymer composite substrate.

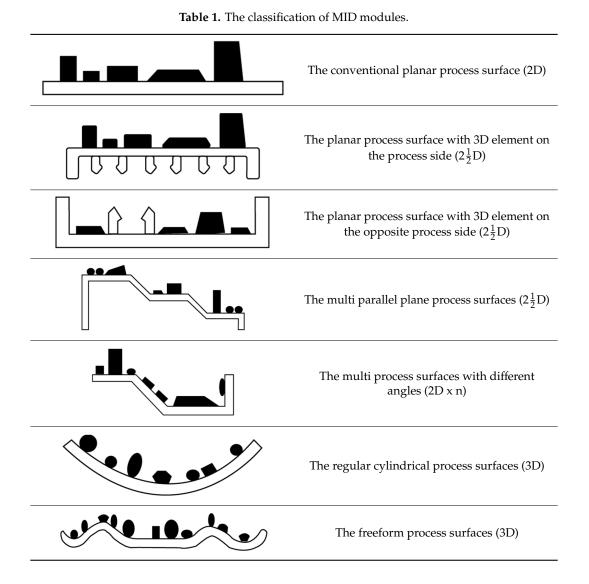
Keywords: 3D-MID technology; surface modification; polymer; composites

# 1. Introduction

Molded interconnected device (MID) represents the versatility of injection molding process that incorporates a conductive circuit pattern with structured metallization. It has ideal mechanical and electrical functions. This is commonly referred to as "3D-MID" and can be integrated directly into plastics structures which include the three-dimensional circuits [1–5].

MID technology has become well-known because of its freedom of geometric design in combination with selective structuring and metallization of 3D layout. It is able to define the angle between components, stacking of chips and foaming of cavities in contrast to the traditional printed circuit boards (PCBs) that are two-dimensional one [6–11]. Moreover, MIDs not only reduce the number of components and the cost by the embedding parts such as connector within a single device but also may save space and shorten assembly time. Therefore, MIDs are employed in numerous applications (i.e., sensor technology, medical technology, automotive, telecommunications, and antennas) [12–15].

The MID modules can be classified into class  $2\frac{1}{2}D$ , n × 2D and 3D categories as shown in Table 1. The conventional circuit board is a flat module with 2D planar process surface. Class  $2\frac{1}{2}D$  permits to produce sample that have flat or plane-parallel process surfaces in Z direction and on the reverse side of processed surfaces two or more plane-parallel. On the other hand, the class n  $\times$  2D and 3D is an interconnected device that consist of multiple process surfaces intersected at the angles or have freeform surfaces [16–20].



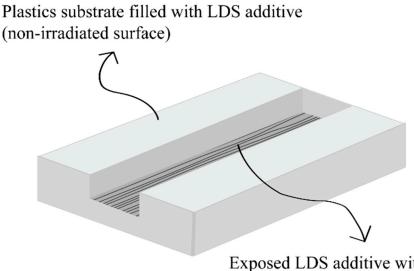
So far, there are reported several methods for manufacturing the MID substrate. Some of them reported the polymer substrate and surface modification in order to develop polymer-based composites. Therefore, this review presents a detailed and a meaningful insight of each processing routes. The main objective of this work is to highlight the benefits of using the MID technology for polymer-based composites by evaluating the state of the art and to provides detail of surface modification of polymer-based composites which leads to the manufacturing of a novel polymer composite substrate.

#### 2. Single-Shot Injection Molding

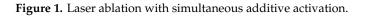
Single-shot injection molding is preferred for high-volume production with short cycle time. Generally, the granulate of plastics are mixed and conveyed by a screw feeder from a hopper unit to the injection unit, then the liquefied plastic is injected into a mold part. Single-shot injection molding for MID technology can be sub-divided into two methods [21–27].

Laser direct structuring (LDS) was noted as an essential process for MID production over the last decades. LDS offers high level of versatility, the possibility for prototyping product with low tool

costs and 3D design freedom. In order to work with LDS method, usable plastics for LDS required specific additives such as Cu<sub>2</sub>O, CuO or CuCl<sub>2</sub>, which can be added in a very high concentration to the blend during compounding step. The characteristic of these additives is extreme heat resistance and great potential to prevent nucleation during injection molding process [28–38]. However, it is also possible to prepare plastics for LDS without specific additives. The LDS method consists of four-step process, injection molding, laser structuring, metallization and surface finish. LDS is based on the principle of ablation and nucleation by using laser irradiation. Therefore, the laser patterning creates a microscopically rough surface and simultaneously activates the specific additives that are necessary for the metallization step [28–38]. The schematic of LDS method is illustrated in Figure 1.



Exposed LDS additive with laser beam (irradiated surface)



Hot embossing is a common method used in MIDs production. It is a rapid process, clean and economical fully additive structuring that can add benefits for limited number of process steps and reduce cost of investment. In this process, a specially coated copper film is pressed into a thermoplastic substrate by a heated die with the negative conductor layout. The pressure force for embossing which is designed depends on the thickness of the copper film and substrate. There, the copper film is cut to form a positive bond to locally melted plastic close to the surface of the blank [39–45].

# 3. Two-Shot Injection Molding

Two-shot injection molding consists of two separate molding cycles. It generally uses different type of plastic that are plate-able (with plating catalyst) and non-plate able (without platting catalyst). The major advantage of two-shot injection molding is related to geometric design freedom which enable to create a very complex three-dimensional circuitry. Also, this is highly suitable for high-volume production with uniform precision. In this process, the first shot of non-plate able plastic is injected into a mold cavity to form the main component. Afterwards, the cavity in the first shot is filled by the second shot of plate able plastic on its locally surface where circuit tracks is required prior to metallization of the conductive structures [46–50]. The schematic of two-shot injection molding is showed in Figure 2.

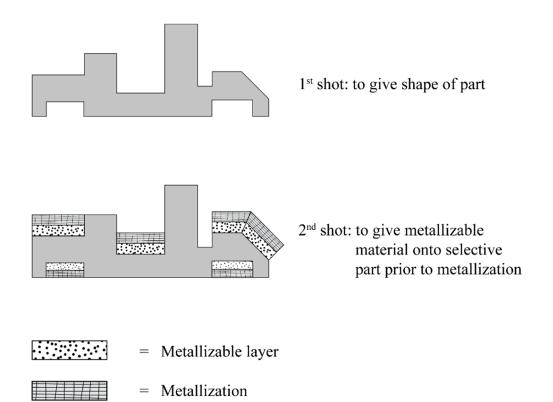


Figure 2. Processing step in two-shot injection molding.

# 4. Material Properties and Characteristics for MID Substrate

The substrate materials used to manufacture MID are made of plastic. The plastics not only offer more production flexibility from macro to micro-MID products but also has high fracture toughness when is subjected to mechanical loading. Moreover, they have good mechanical properties and light weight. The plastic substrates employed in the production of MID may vary depending on the manufacturer or supplier. Therefore, some important parameters must be considered when selecting the material.

- Assembly temperature
- Rheological properties
- Degradation
- Shrinkage and tolerances
- Anisotropy properties
- Tensile and flexural properties
- Metallization capability
- Electrical properties
- Environmental concerns
- Cost

In order to select a substrate material from a various type of plastics with different properties profiles, some requirement criteria of plastic materials used for MIDs are necessary. The main criteria are illustrated in Table 2 [51–54].

In order to search for more and higher capability plastic materials, there was extended the range of standard plastics and engineering plastics which includes high-specification polymers. Obviously, the requirements which apply to any base material impinge on decision-making process at several different levels. Alongside purely technical aspect such as intrinsic load ability, compatibility, the basic workability properties, ecological factors (i.e., suitability for return to the natural materials cycle) and economic factors (i.e., procurement and processing costs) influences the materials development and selection [55–58]. Therefore, the requirement outlined in Table 2 can be very diverse. In order to satisfy these requirements, the characteristics values have to line up with certain basics requirements:

- Materials have to be meaningful
- Materials have to be comparable
- Materials should be rationally measurable

The selection of the right MID substrate is often a very challenge process that should be a balance between electrical, thermal and mechanical properties and the manufacturing cost [59–62].

	1-9
Thermal properties	- Heat distortion resistance - Thermal expansion - Melting and crystallization
Mechanical properties	- Strength and yield strength - Fracture elongation - Stiffness (E modulus)
Electrical properties	- Dielectricity - Electrical puncture resistance
Workability properties	- Flowability - Shrinkage - Distortion
Compatibility	- Plastic/plastic - Plastic/metal
Environmental compatibility	- Recycling - Scarcity

Table 2. Requirement criteria for polymer materials used in MIDs.

#### 5. Thermoplastics for MID

There are many different polymers used to manufacture the MID substrate. The choice of material and production process are primarily linked to mechanical, thermal and electrical requirement of the purpose of MID application. Thermoplastics materials are the most used to fabricate the MID substrate [63–65]. They can be subdivided on the basis of heat distortion resistance, long-term service temperature and the price of raw material. The plastics pyramid depicted in Figure 3 describe the three major subgroups of thermoplastic materials [66,67]. In the plastics pyramid, the presence of high-performance plastics is very scarce, but extremely interesting in terms of engineering applications. However, high-performance plastics are associated to high prices and costly processing which may entail limitations for industrial use. On the other hand, many MIDs are made of engineering-grade thermoplastics, because they are available at low cost and present virtually no difficulties for manufacturing in the existing production facilities. Typically, the characterization of an MID substrate materials depends primarily on the material properties which are responsible for indicating how the material behaves during metallization along with the chemical resistance. The pyramid overview is considered a guideline to select materials for MID [68,69].

#### 5.1. Polypropylene (PP)

Polypropylene (PP) is a standard plastic that can be used in MID. It is largely nonpolar and has a crystallinity approximately 60–70%. The glass transition ( $T_g$ ) is about 0 °C and melting temperature is in the range of 155 to 160 °C thus, PP is not suitable for standard soldering process. PP has a high chemical resistance on account of its nonpolar character, and it has no inherent flame-retardant properties. Moreover, reinforced PP by glass fiber or talc can reduce the isotropic shrinkage and heat distortion resistance [70–79].

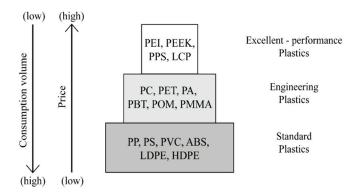


Figure 3. Plastics pyramid for worldwide consumption.

#### 5.2. Syndiotactic Polystyrene (sPS)

Syndiotactic polystyrene (sPS) is partially crystalline plastic that can withstand to temperatures up to 265 °C for short periods. Long-term service temperature can reach up to 150 °C. The maximum crystallinity is approximately 52% and it depend on processing conditions. It has a melting point of 275 °C and the glass transition of 105 °C. sPS is a polymer with extremely high chemical resistance but can be easily damaged by oxidization with ozone and chorine particularly at high temperature. Moreover, sPS can be metallized by a wet-chemical or catalytic process [80–89].

#### 5.3. Polyphenyl Ether (PPE)

Polyphenyl Ether (PPE) or also known as polyphenyl oxide (PPO) is an engineering plastic which is produced by blend of PPE and PS. It belongs to the group of amorphous plastics which has a good mechanical property. Its electrical properties are virtually unaffected by the temperature and frequency. The long-term service temperature is in the range of 100–110 °C. It can be chemically metallized. PPE is not only resistant to hydrocarbons but also susceptible to stress cracking on account of its styrene content [90–99].

# 5.4. Polycarbonate (PC)

Polycarbonate (PC) is an amorphous, transparent plastics that combines high strength, heat distortion resistance, and impact roughness even at low temperature. It also has good electrical insulating properties that are virtually independent of temperature and moisture. The glass transition temperature is between 140 and 150 °C [100–109].

#### 5.5. Polybutylene Terephthalate (PBT)

Polybutylene terephthalate (PBT) is a partially crystalline engineering plastic. It is used primarily on account of its good dimensional stability, temperature resistance and electrical properties. PBT exhibits high resistance to many solvents, but it is not inherently flame retardant [110–119].

#### 5.6. Acrylonitrile Butadiene Styrene (ABS)

Acrylonitrile butadiene styrene (ABS) belongs to the group of amorphous plastics. It exhibits good impact resistance, hardness and scratch resistance. The maximum use temperature of ABS is up to 90 °C which makes them unsuitable for standard soldering process. The glass transition temperature is about 85–100 °C. The proportion of its three constituent components can be varied which permits to adapt this material to very widely differing requirement [120–129].

# 5.7. Polyamides (PA)

Polyamides (PA) are found in an exceptionally wide range. The polyamides constituent is one of the most important groups of materials within the partially crystalline engineering plastics. Some of

polyamides show properties on those of the high temperature plastics in particular PPA, PA6T/X, PA46 and PA6/6T. High mechanical loading, high damping capability, and high wear resistance are the characteristics of PA, regardless of structure. The water absorption is approximately 10%. Generally, PAs are resistant to solvents, fuels, and lubricants. However, they are not inherently flame retardant [130–139].

# 5.8. Polyphenylene Sulfide (PPS)

Polyphenylene sulfide (PPS) belongs to the group of partially crystalline and nonpolar high temperature plastics. Its structure lacks branches thus, it is highly crystalline. It is a very hard and stiff material which is suitable for process at high temperature up to 240 °C. The melting point is about 285 °C, and glass transition temperature is between 85 and 100 °C. PPS exhibits excellent chemical resistance, very low water absorption, and inherent flame resistance [140–146].

#### 5.9. Liquid Crystal Polymer (LCPs)

Liquid crystal polymers (LCPs) consist of rigid, rod-shaped macromolecules that self-parallel in melt and form liquid crystalline structure. It can be used to manufacture extremely delicate molding with thin wall sections and long runner channels. Other characteristic properties besides high strength and rigidity in the flow direction includs the long-term service temperature of 185 to 250 °C and the melting temperature between 280 and 355 °C. LCPs are solderable with common methods and galvano-workable types also are available [147–153].

#### 5.10. Polyetherimide (PEI)

Polyetherimide (PEI) is an amorphous high temperature plastic. It exhibits high strength rigidity and hardness even without reinforcement. The maximum long-term service temperature is about 170 °C and its glass transition temperature is 210 °C. PEI can easily solderable by reflow soldering process. It can be chemically metallized, and it is not resistant to ketones, chloroform, ethyl acetate, and methyl ethyl ketone. Moreover, it is inherently flame retardant [154–160].

#### 5.11. Polyethersulfone (PES)

Polyethersulfone (PES) belongs to the group of amorphous, transparent and polar high temperature plastics. It is available in both reinforced and non-reinforced type. It exhibits high strength, rigidity and hardness across a wide temperature range from –100 to +200 °C. Their Tg is about 225 °C. PES can be chemically metalized by surface treatment printing and metallization under vacuum or galvanization after appropriate pretreatment process. Moreover, PES is not only chemically resistant to ketones, esters, hydrocarbons, and aromatics, but also can include highly polar solvents [161–167].

#### 6. Modified Thermoplastics for MID

The high temperatures requirement is a changeover related to lead-free solder process imposed by the constituted law. This is the major reason why high-performance thermoplastics are virtually the only unmodified products that can be used adequately as substrate materials [168–170]. Engineering plastics such as PA6 and PA66 are often used as base materials on the account of their good metallization and service behaviour. Therefore, searching for approaches that are economically and relatively practically for engineering thermoplastics is still a major challenge. It is especially for assembly and connection technology process such as lead-free reflow soldering. Nowadays, these thermoplastics are enriched with up to 40% by weight of reinforcing with additive which allows to increase their mechanical limits. Glass fiber, glass balls, and a wide of variety of mineral substances are used as reinforcing additive. As far as extending their thermal limits is a concern, the MID research conducted in the recent years has brought to light two main approaches, namely electron-beam crosslinking and the use of the filler

system for modifying thermoplastics, which is capable of satisfying higher requirements, particularly those deriving from the production process [171–174].

#### 7. Radiation of Cross-Linked Thermoplastics

The radiation cross-linked thermoplastics is a technique which was investigated since 1950. In the past, its use was mainly restricted to the group of polyolefins. Recently, the endeavor to transfer experience gained in the radiation crosslinking to injection molding and extrusion films made of engineering plastics are a new departure into relatively unknown territory [175,176]. One of the success has been achieved by cross-linked PA6 and PBT molding. Despite their excellent mechanical and electrical properties, engineering plastics have not been adopted for high thermal load process such as reflow soldering primarily on account of their low heat distortion resistance and poor temperature resistance. This thermal weakness can be overcome by radiation crosslinking [177–179]. In the process of crosslinking, plastics are not only cross-linked chemically but also by radiation chemistry for example in the presence of peroxides. In principle the effect of radiation-chemical reactions can be induced by electromagnetics wave such as X-ray or gamma rays [5,180].

Radiation induced crosslinking has numerous advantages. This is because the main thermoplastics material's properties can be improved after modified by radiation induced crosslinking as showed in Table 3 [181,182].

Mechanical Properties	Thermal Properties	<b>Chemical/Physical Properties</b>
- Strength	- Temperature resistance	- Chemical resistance
- Moduli	- Reduction in thermal expansion	- Reduction in solubility
- Abrasion resistance	- Solderability	- Increase adhesion
- Creep behavior	- Flame retardation	- Hydrolysis resistance

Table 3. Improvement in thermoplastics properties by radiation induced crosslinking.

#### 8. Thermoplastics Composites

Compounding the thermoplastics with special fillers or additives as composites materials has proven to be an effective way of integrating functions such as mechanical, thermal and electrical properties. These plastic compounds benefit from material synergies derived from the good workability of the plastics and the additional properties which are gains from the fillers. Tailoring the properties of plastic compounds such as thermally conductive in order to meet superior requirements would entail conforming with the following criteria for selection the composition of the fillers [183–186]:

- Maximization of filling
- Filler shape and size
- Filler mixtures
- Additivity

Generally, fillers such as ceramic materials are thermally conductive but electrically insulating. On the other hand, the metallic materials are thermally and electrically conductive. The high proportion of fillers can alter the behaviour radically compared to unmodified thermoplastics. On the other hand, the fracture and strain generally diminish, whereas rigidity decreases. There, the strength depends largely on the bonding of filler with the embedded matrix. Moreover, some special additives and other fillers such as glass fiber can be applied to improve the bonding between matrix and filler which result in strength and fracture strain enhancement.

Typically, the plastic compounds for MID can be manufactured from a very wide variety of fillers from the group of metallic or ceramic materials. Recently, some of fillers are coming into widespread use on account of their versatility, excellent thermal properties and because they offer considerable benefits for MID production. The preferred fillers are graphite, carbon black, copper, aluminium oxide and boron nitride. The geometry of filler is one of the main factors that influences

the mechanics of reinforcement. Nowadays, they can be classified into three groups, one-dimensional filler (i.e., fibers), two-dimensional fillers (i.e., platelets) and three-dimensional fillers (i.e., powder). For one-dimensional fillers, glass fibers are widely used, because of their geometry and superior improvement in the mechanical characteristic values. The proportion of glass fiber reinforcement is generally between 15% and 50% by weight. Below 15% by weight, there is no reinforcing effect while above 50% by weight it becomes difficult to wet the fillers with the uniformity necessary for smooth surfaces. For three-dimensional filler such as glass balls usually increase only the elastic modulus. The classification of fillers by geometry is also crucial in term of thermal conductivity. Anisometric one- or two-dimensional fillers lead to higher thermal conductivities than three-dimensional fillers and spherical fillers. Therefore, the thermal conductivity is the process dependent and component dependent material property, particularly when the fillers used are anisometric. Moreover, highly filler filled, thermal conductive plastics require an adaptation of the process parameters for dependable injection molding. This is because of the higher melt viscosity and rapid cooling of the plastics melt while the mold cavity is filling. Consequently, MID design for highly filled thermally conductive compounds is always trade-off between the modified properties and workability of the materials [187–191].

#### 9. LDS Materials for MID

The LDS is one of method used to produce MID product by using laser activation of special additive filled plastics. Therefore, the special additives that enable the LDS process must be finely distributed in body part and transferred into catalytically agent during laser activation. The requirement of these additives are good chemical compatibility and homogeneous distribution in matrix, no impairing of electrical and mechanical of substrate, excellent thermal stability and no catalytic activities in deactivated state. In this process, the preferred additives are from the group of metal oxide or mixed metal oxide especially, copper compound such as Cu<sub>2</sub>O, CuO or CuCl<sub>2</sub>. This is because its temperature resistance is extremely high thus, there no nucleation can take place while the plastic melt. LDS materials have found in the range from standard thermoplastics through engineering thermoplastics and high-performance thermoplastics which are suitable for reflow soldering process. The currently available LDS materials for industrial scale are listed in Table 4 [192–197].

Polymer Matrix	Supplier Company	Commercial Grade Name
	RTP	699 X 113386 B
ABS	Trinseo	MAGNUM LDS/ABS 3453
	RTP	399 X 113385 B
	MEP	XANTAR LDS 3750
PC	Blustar Chengrand	SUNPLAS LDS C0040
	Trinseo	EMERGE LDS/PC 8900
	Kingfa	Vismid SOL 2100 LDS
PA/PPA	BASF	Ultramid T4381 LDS
	DSM	ForTii LDS 85
	EMS	Grilamid 1SBVX-50H LDS
	Evonik	Vestamid HT plus LDS 1031
	Kingfa	Vismid SOL 65250 LDS
PA/PPA	RTP	RTP 299 X 113399 H
	MEP	Reny XHP 1351L
	RTP	1099 X 127271 C
PBT	Evonik	Vestodur X9423

Table 4. The list of LDS materials.

Polymer Matrix	Supplier Company	Commercial Grade Name
PPE	Premix	Preperm 260 LDS
	RTP	RTP 3499-3 X 113393 A
LCP	Ticona	Vectra E840i LDS
PEI	RTP	2199 X 127272 A
PPS	DIC	LP-150-LDS

Table 4. Cont.

#### 10. Thermoset Plastic for MID

Thermoset plastic is of interest for MID substrate because it offers some high heat distortion temperature. The thermoset compound can be mixed with a wide variety of fillers, which allows to modify their properties to meet the requirements for a given applications [198–202]. The addition of suitable proportion of fillers can produce the coefficients of thermal expansion, heat distortion and elongation behaviour virtually identical to those of copper. Once thermoset is cured, their three-dimensional molecular network structure exhibits a very high level of dimensional stability. The important thermal properties of the current commercially available thermoset compound are listed in Table 5 [203–206].

Table 5. List of current commercial thermoset compound and thermal properties.

Polymer Name	Elongation (10 <sup>-6</sup> /K) (ISO 11359)	Heat Distortion Temperature (°C) HDT-A (1.8 N/mm <sup>2</sup> ) (ISO 75-2)
Phenolic resin	16–24	>250
Epoxy resin	15–25	>250
UP resin	10–20	>250
DAP resin	10–20	>250

#### 11. Laser-Assisted Metallization for Polymer Materials

Regarding the MID parts, the surface properties of polymer substrate is one of the crucial factors that significantly affect metallization process, adhesion strength, and overall quality of the deposited metallic layer. Generally, the surface modification of polymer substrate can be divided into two methods, chemical and physical modification. In chemical modification, the polymer substrate is modified on selective surface by using chemical solution as solvent such as potassium manganite (KMnO<sub>4</sub>), nitric acid (HNO<sub>3</sub>), and ethyl alcohol [207–209]. On the other hand, laser modification is one of the popular physical surface treatment methods used in order to prepare the polymer surface for metallization. In the last decades, laser-assisted electroless metallization for polymer materials has been reported in numerous works [210,211]. Laser treatment represent a group of advanced engineering tool useful in selective modification of small surface areas of complex shapes. They have great importance in manufacturing of printed circuit boards and other small electronic devices. Laser treatment is able to change the surface geometrical structure, degradation or crosslinking of molecular chains of polymer without changing bulk properties. Laser wavelength, laser power and laser mode of operation (continuous or pulsed laser) are the factors that influences the surface properties of polymer materials. Typically, the polymers are not only absorbing well ultraviolet (UV) but also infrared (IR), whereas they are mostly transparent to visible light (Vis). Basically, UV radiation causes photolytic breaking of molecular bond within polymer materials while, the heat effects are limited. It results in the formation of free radicals that enable to initiate photochemical reactions. In the IR region, most of polymer functional groups are excited to higher vibrational and rotational energetic states that is accompanied by heat generation. The accumulation of heat is able to break the molecular bonds and

thus the thermally activated reactions occur. Moreover, the polymers which are modified in the range of Vis spectrum, are commonly doped with absorbing agent (photoinitiators) which decompose and initiate various reactions. The interaction between laser irradiation and polymer are greatly depending on applied wavelength of the radiation. In the laser surface modification method, a very important parameter is energy of radiation per unit area, called laser fluence for a given polymer. The ablation of material starts when a certain value of laser fluence, the so-called ablation threshold is reached or exceeded. Therefore, the laser surface modification of materials is classified as below or above the ablation threshold [212–216].

#### 12. Influence of Laser Irradiation on Polymer

In the laser modification of surface layer, there are two important material parameters that influence the surface properties of the polymer material. Firstly, absorption coefficient ( $\alpha$ ), associated with absorption of light wave energy which can be evaluated by measuring of the energy attenuation within that materials. Secondly, the refractive index (n), that is connected with the coefficient of reflection (R) of the radiation at the material-medium phase boundary [217,218]. The light wave dissipation is associated with the light absorption that can be drawn as an absorption coefficient in the following Equation (1) [219].

$$\alpha = \frac{4\pi\kappa}{\lambda} \tag{1}$$

where  $\kappa$  is extinction coefficient and  $\lambda$  is the light wavelength.

As per Equation (1), it is clear that the light absorption coefficient of material depends on both the light wavelength and extinction coefficient. The laser radiation intensity within an examined material can be described by the empirical Beer's law Equation (2) [220,221].

$$I(Z) = (1 - R)I_0 e^{-az}$$
(2)

where I(z) is the laser radiation intensity within materials at a vary depth (*z*), measured from the material surface, *R* is coefficient of light reflection from material surface, and  $I_0$  is intensity of incident laser radiation.

The coefficient of light reflection (R) is also an important factor for modification of the material surface layer. It can be described by Maxwell's equation system in the case of the perpendicular incidence of light on the material surface which yields the value R [222,223].

$$R = \frac{(n-1)^2 + \kappa^2}{(n+1)^2 + \kappa^2}$$
(3)

According to Equation (3), *R* value depends on both the refractive index (*n*) and the extinction coefficient ( $\kappa$ ). Thus, the laser radiation intensity within material at a depth (*z*) can be rewritten by taking Equation (3) into Equation (2) which leads to following relationship [224,225] as shown in Equation (4):

$$I(Z) = \left[\frac{4n + 2\kappa^2}{\left((n+1)^2 + \kappa^2\right)}\right] I_0 e^{-az}$$
(4)

#### 13. The Absorption Coefficient for Laser Radiation

The polymer materials may have high absorption coefficient for laser radiation especially, in the UV range. Penetration depth  $(1/\alpha)$  of laser light at 147 nm for example, polyethylene (PE) is about 34 nm, whereas its gamma radiation is about 15 cm [226]. This implies that the energy is absorbed in a thin surface layer of polymeric material irradiated even by using ultraviolet radiation. In addition, the surface layer properties of polymeric materials also related to the laser ablation. Basically, the laser ablation is based on physicochemical changes induced by the laser radiation and resulted in tearing

off fragments of the irradiated surface layer in the ablation process [227–231]. In the laser ablation of polymeric materials, there are two main process, photochemical and photothermal that may proceed simultaneously. The photochemical ablation consists in photolytic breaking of chemical bond. It was done by electron excitation in macromolecule segments to high-energy electron states. The essential phase of the ablation process begins when a large number of chemical bonds simultaneously undergo breaking upon irradiation by high-energy laser pulses [232]. The ablation threshold ( $E_{th}$ ), and the number of broken chemical bonds (n) can be drawn as in Equation (5) [233,234].

$$E_{th} = n \bigg[ \frac{h\nu}{\phi \alpha (1-R)} \bigg] \tag{5}$$

where, *R* is coefficient of laser radiation reflection of the polymeric surface,  $\phi$  is the quantum yield of bond breaking (0 to 1),  $\alpha$  is coefficient of radiation absorption, and hv is photon energy. On the other hand, in the photothermal ablation process is also assumed that the laser radiation strongly is absorbed by the material excites molecules to high-energy state. Due to mutual collisions, the molecules relax from this state to ground state thus, heat generated in this way causes an increase in temperature being sufficient for breaking of chemical bonds in polymeric materials [235]. When the material temperature exceeds a certain value, called threshold ablation temperature ( $T_D$ ), the process of thermal ablation of polymeric materials begins. The relationship between  $E_{th}$  and  $T_D$  can be expressed as in Equation (6) [236,237].

$$E_{th} = Cw \bigg[ \frac{T_D - T_R}{\alpha (1 - R)} \bigg]$$
(6)

where  $T_R$  and  $C_w$  are the initial temperature and specific heat of polymeric materials, respectively.

#### 14. Photochemical and Photothermal Ablation

Photochemical and photothermal ablation proceed works simultaneously and are difficult to separate. The thickness of material fragment torn off the surface layer i.e., ablation depth, depend on both photochemical and photothermal ablation mechanisms. Hence, the ablation depth (L) of materials can be expressed as a sum of the two components as presented in Equations (7) and (8) [238–240].

$$L = \frac{1}{\alpha_{eff}} \ln\left(\frac{E_j}{E_{th}}\right) + Aexp\left(\frac{-E_a}{\kappa_B T}\right)$$
(7)

$$L = L_{chem} + L_{therm} \tag{8}$$

where  $E_j$  is the energy per laser pulse,  $E_a$  is activation energy for ablation, and  $\alpha_{eff}$  is effective coefficient of absorption. The first component in Equation (7) represents the photochemical ablation model which is derived from the Beer's law, whereas the second component represents the photothermal ablation model based on the Arrhenius's law associated to Equation (8) [241–244]. The ablation rate of the surface layer of polymeric materials is illustrated in Figure 4 [245,246]. The ablation rate is defined as a quotient of the total ablation depth (*L*) and the number of laser pulses (*N*) as a function of  $E_j$ when is neglected the thermal component. The rapid tearing off fragments of polymeric materials starts when the laser pulse energy exceeds the ablation threshold. During that time, some changes in surface geometric structure appear while no significant chemical change occur. Moreover, the ablation rate depends also on the nature of material to be modified,  $E_j$  value, laser radiation wavelength, laser pulse duration (pulse width) and ambient condition. Because so many variable factors, including the two-step absorption and attenuation of radiation beam by the material being torn off, there are a number of models for the mechanism of the ablation process [247–251].

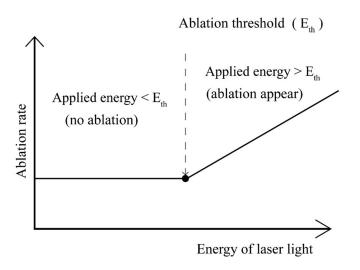


Figure 4. Laser ablation rate as a function of a unit energy of laser light.

#### **15. Selection Criteria of Laser for Polymers**

There are many factors that affect surface layer properties such as absorption coefficient, laser beam angle of incidence, and energy distribution of laser beam. Therefore, a suitable criterion for the selection of laser for polymer is required. For instance, the coefficients of absorption of the laser UV radiation for poly(ethylene terephthalate) (PET) and polystyrene (PS) are significantly higher than those for polyethylene (PE) and polypropylene (PP). PET and PS absorb the laser UV radiation mostly in the surface layer of thickness not exceeding 0.1 and 0.4 micrometer, respectively. This is in contrast to PP and PE which the absorption occurs within much thicker layer. Some of additives such as polyolefin and benzophenone are being used for the improvement of UV and UV-Vis light absorption [252–255]. The specific heat and thermal diffusion constant of the polymeric material can be a concern because they are related to minimization of thermal damages into material. In the recent years, the number of lasers operating with different ranges of wavelength and various mode (pulsed or continuous) are rapidly growing. They can be classified according to various criteria. One of the most important criteria is the type of active medium or gain medium. Basically, there are three types of active medium which are used in present [256,257]:

- Gas laser (excimer laser, nitrogen or carbon dioxide laser)
- Solid-state lasers (neodymium, Nd:YAG laser)
- Dye laser

Table 6 presents a summary with the list of available lasers and their capacity [258,259].

Wavelength (nm)	Active Medium	Operation Mode	Average Power (W)
193	ArF excimer	pulsed mode	1–100
248	KrF excimer	pulsed mode	1–100
308	XeCl excimer	pulsed mode	1–100
351	XeF excimer	pulsed mode	1–100
337	N <sub>2</sub>	pulsed mode	0.1
351–1092	Ar <sup>+</sup>	pulsed/continuous	0.001–0.1
262/355/532/1064	Nd:YAG	pulsed/continuous	100
697	Al <sub>2</sub> O <sub>3</sub> (ruby laser)	pulsed/continuous	1
9000-11,000	CO <sub>2</sub>	pulsed/continuous	10,000
808/940/980	GaAs	continuous mode	1000

Table 6. The list of available lasers in industry.

#### 16. Metallization

In order to create complete MID parts integrated within the circuit track, plastic parts must be coated with metal ion by a process called metallization. The metallization or perhaps called plating is a process to plate or deposit a metal ion (i.e., Cu<sup>2+</sup>, Ni<sup>2+</sup>) into the conductive or non-conductive substrate. The goal of this process is to enhance the specific properties of plastics including reflectivity, abrasion resistance, and electrical conductivity. Therefore, plating on plastics has received considerable attention in the manufacturing of printed circuit boards (PCBs), automotive parts and electromagnetic interference (EMI) shielding applications [260–263]. Over the last decades, the plastics was successfully metallized with gold, silver, nickel and copper [264,265]. Generally, metallization of plastics is classified into two groups that are primary and secondary metallization. In primary metallization, plastics can be done by depositing a thin layer of metal having thickness approximately 10–50 µm. On the other hand, the secondary metallization is conducted after primary metallized parts to increase the thickness of metallic layer up to 180 µm or above. There are many methods for the metallization of plastics as following [266,267]

- Dipping in a metal paint
- Sputtering
- Vapour deposition technique
- Electro plating
- Electroless plating

#### 17. Electro and Electroless Plating

The electro and electroless plating are the most important methods for producing functional and decorative finishes in the requirement for MID applications. Both electro and electroless represent a chemical reduction between the reducing agent, in the presence of solution, and the metal ion. During the plating process, a thin layer of catalyst is applied to localized surface. For instance, a layer of metal (i.e., copper or nickel) is plated as a result of the metallic phase that appears on the solid surface. For the electro plating or perhaps called galvanic plating, it is necessary to use battery or rectifier. The applied electric current is combined with a chemical solution to reduce the metal cations. Basically, there are two main components, the parts that is plated called cathode and the metal which is plated on the parts is called anode. On the other hand, electroless plating, also known as autocatalytic plating, is a process which uses purely chemical reduction process without any electrical energy dispersal [268–273]. It is possible to obtain metalize on dielectric surface at ambient temperatures by using some simple aqueous solutions. Therefore, electroless plating is widely used in modifying the surface of plastics. Especially it was noted a significant growth on the printed circuit market. The schematic of electro and electroless plating are illustrated in graphical abstract, and the various advantageous and disadvantageous between electro and electroless plating are presented in Table 7 [274–276].

Electro Plating	Electroless Plating
Plating thickness: 40–50 µm	Plating thickness 8–10 μm
Controlled electro plating reaction	Complicated chemical process
Electrical conduction	No conduction problem
High quality surface finish	Rough surface finish

 Table 7. Electro and electroless various advantageous and disadvantageous.

# **18. Electroless Plating Procedure**

Electroless plating on plastic is a technologically and chemically complex process. Generally, electroless plating consists of the following steps [277–281].

#### (i) Cleaning step:

This step is not only used for cleaning the surface of plastics to remove oil, dirt but also used to produce roughness which leads on improving surface area. The cleaner typically belongs to alkaline group which include neutral or acidic materials. The most important factors for a superior cleaning solution are: temperature of cleaner, concentration of cleaner and cleanliness of cleaner after cleaning.

# (ii) Etching steps:

Etching is the key stage in plating process for achieving a good metal-plastic bonding. The employed surface is not only chemically etched but also physically etched resulting in the development of pores that can increase the surface area thereby providing the opportunity for superior contact between metallic layer of plating. During the chemical etching, the plastic parts are immersed in an oxidant solution which is either chromic acid in aqueous sulphuric acid or hydrofluoric acid containing sulphuric acid. It is applied on the selective surfaces to be plated. The etching time is normally for 5–15 min with temperature in the range of 60–65 °C. In addition, a laser can be applied for physically etching the plastic parts. The quality of the etching surface depends on laser mode (pulse or continuous), laser wavelength, and energy of laser beam.

# (iii) Neutralization steps:

Neutralization is necessary for removing the residual of oxidant that remains during the etching steps. The reducing agent such as ferrous ions aids to prevent the inhibition of the catalyst. This is because even trace may completely inhibit electroless deposition of metallic layer on the plastic surface.

(iv) Activation steps:

In this process, the modified surface is contacted with an activator or catalysts usually in colloidal suspension as catalyst powder. The catalyst deposited in the surface micro-cavities is formed during conditioning for subsequently the electroless plating. The activation process is typically carried out at 40-45 °C for 3-5 min. Higher concentration of activator or too long immersion time lead to improper activation which probably cause poor metallic bonding. There are many commercial catalysts that can be used in the activation process as showed in Table 8 [282–285]. Generally, the electroless plating is regarded as a dehydrogenation reaction since the hydrogen may develops simultaneously with metal reduction. According to Table 8, palladium (Pd) is the most active from the catalyst list. It is because its exchange current density for hydrogen gas evolution reaction  $(-\log i_0)$  is the lowest. However, palladium is unable to form itself on the plastics surface. Therefore, the combination of stannous (Sn) with palladium is preferred because stannous is an excellent reducing agent for palladium when is used for coating of non-conductive surface. Stannous used in form of stannous chloride serves to wet the surface. The uniform layer of reducing agent afterward is subjected to intermediate water rinse which then is converted into insoluble, hydrous stannous oxychloride coating. When palladium chloride solution is dipping in, the palladium chloride may be reduced to metallic palladium (Pd<sup>0</sup>) and bound to the desired surface. Moreover, the concentration of palladium and stannous is normally  $5 \times 10^{-6}$  g/cm<sup>2</sup> and  $20 \times 10^{-6}$  g/cm<sup>2</sup> respectively, which usually is used for plating of plastics [286–289].

#### (v) Acceleration steps:

The acceleration is a process which permits to activate the catalyst and during the activation step the activity is more intense. The accelerator can be acidic or alkaline solution. The activated surface is typically washed and immersed in an acceleration bath at 30–35 °C for 3–5 min. There palladium and stannous mat be used as catalysts too. The excess of stannous on the activated surface allows stabilizing stannous ions by the accelerator solution consequently, the Pd<sup>2+</sup> as a catalyst is reacted and formed into Pd<sup>0</sup>. The chemical reduction of palladium (II) in the presence of stannous (II) after acceleration can be drawn by [290–295]:

$$Pd^{2+} + Sn^{2+} \xrightarrow{HCl} Pd^0 + Sn^{4+}$$
(9)

Metal	–log i <sub>0</sub> (A/cm <sup>2</sup> ) (Exchange Current Density for Hydrogen Evolution Reaction)
Palladium	3.0
Platinum	3.1
Rhodium	3.6
Iridium	3.7
Nickel	5.2
Gold	5.4
Silver	5.9
Copper	6.3

Table 8. Comparison of catalytic activity of metals.

#### (vi) Deposition or plating steps:

Deposition or plating is the final process for electroless metallization. In this stage, the activated surface which is prepared throughout the solution usually contains metal-salts as a reducing agent. The plating layer forms by oxidation-reduction reaction on the activated surface. Electroless plating bath typically contains metal-salts, reducing agent (i.e., formaldehyde), alkaline hydroxide (i.e sodium hydroxide), chelating agents (i.e., EDTA, Rochelle salts), stabilizer and brightener. This process is conducted at bath temperature in the range 45–65 °C [296–299]. A versatile explanation of mechanism of electroless plating process based on electrochemical reactions is presented [300–302]. The reducing agents are anodically oxidized on the catalyst surface while the electrons obtained are transferred to metal ions by a cathodically reduction. An excellent example is electroless plating of copper. The copper ions are reduced by formaldehyde process at room temperature (30–35 °C) in alkaline solutions (pH  $\cong$  12–14). At this stage copper ions are bounded into a complex. The most suitable Cu<sup>2+</sup> ligands for electroless copper plating solutions contain polyhydroxy compounds from tertiary amine groups and hydroxy groups in most common practice used K-Na tartrate, Na<sub>2</sub>EDTA, NaOH and formaldehyde. The chemical reduction for copper plating is described by [303–305]:

$$2CH_2O + 40H^- \to Cu^0 + 2HCOO^- + H_2 + 2H_2O$$
(10)

The measurement of polarization resistance ( $R_p$ ) is a method that can provide information about the mechanism of plating process. The polarization resistance is inversely proportional to the process rate (*i*). Here, the relationship between  $R_p$  and *i* is written as described in Equations (11) and (12) [306–308], respectively.

$$i = \frac{b_a b_c}{R_p (b_a + b_c)} \tag{11}$$

$$R_P = \left(\frac{dE}{di}\right)_{i=0} \tag{12}$$

where  $b_a$  and  $b_c$  are Tafel coefficients ( $b \approx 1/\alpha nf$ ),  $\alpha$  is the transfer coefficient, n is the number of electrons involved in the reaction for one molecule of reactant and f is equal to F/RT (F = Faraday number).

Another factor which should be considered during plating process is deposition rate. Deposition rate is expressed in unit of micrometer per hour. When the concentration of reducing substance is not maintained at a constant level, the deposition rate will start to decrease. The deposition rate is often

given as average rates that depend on the ratio of the surface to be plated and the volume of solutions  $(dm^2/L)$ . The dependence of deposition rate (v) for the specific concentration of reducing substance is described by an empirical equation as follow [309–314]:

$$v = k \left[ M e^{n+} \right]^a \left[ R e d \right]^b \left[ H^+ \right]^c \left[ L \right]^d \tag{13}$$

where *k* is the rate constant, *L* is the concentration of free ligand;  $Me^{n+}$  and Red, are metal ions and reducing agent respectively. The exponents *a* and *b* are usually smaller than unity, *c* is a negative value and *d* is usually close to zero when the ligand is substituted. However, this relationship is for a general case. There, the electroless deposition rate is normally about 2–5 µm/h.

# **19. Future Recommendation**

The strategies for the development of polymer-based composites undoubtedly strive to address the surface modification and selection of catalytic activity within nowadays challenging metals in order to achieve the requirement of MID technology. The surface modification is paramount important to address this challenge. However, only a catalytic activity can offer a simultaneous effect to all modification techniques. Accordingly, a possible solution forward to accelerating the research and development of polymer-based composites for MID, is to bridge the research gap between surface modification techniques of polymer-based composites and catalytic activity of metals agent. However, the efficiency of chemical modification surface is limited due to its chemical reaction.

On the other hand, laser-assisted modifications have successfully embarked to modify polymer surface. Similarly, a number of fabrications of electroless plating solution with palladium have been reported. Hence, the way forward, is to produce a combination of laser-assisted modification and electroless plating solution which incorporated palladium. This can extend the scope of polymer-based composites and MID technology to shorten the gap between lab scale research and industrial applications in future.

# 20. Conclusions

The higher requirements of three-dimensional electronic parts associated to better functioning and continuum miniaturization have encouraged the scientific community to focus their research and develop novel method such as 3D-MID technology. The 3D-MID represents the versatility of injection molding process that incorporates a conductive circuit pattern with structured metallization. It can reach ideal mechanical and electrical functions and can be effectively integrated directly into the plastics substrate which include the three-dimensional circuits for automotive parts, mobile phones, and medical devices. This review was devoted to providing a robust understanding of 3D-MID modules, to present the requirement criteria for manufacture polymer substrate, and the main surface modification techniques used to enhance the polymer substrate. The findings presented here allows to fundamentally understand the concept of 3D-MID which can be used to create novel polymer composite substrate. The importance of selection polymer substrate including surface modification, which play a crucial role was emphasized. By applying an appropriate laser treatment on a polymer substrate, it will be possible to obtain better surface characteristics which enable to improve the metallization process. The laser treatment applied to a polymer substrate could be considered a very versatile technique because its effectiveness in surface modification permits the use of polymer composites within multiple critical applications.

**Author Contributions:** J.T., M.R.S., and S.S.: performed the data analysis. J.T., M.R.S., and C.I.P.: wrote the draft of the manuscript. J.T., M.R.S., and S.S., reviewed the draft and pointed out the shortcomings. C.I.P.: revised the manuscript. All authors have read and agreed to the published version of the manuscript.

Funding: This research was partly supported by the King Mongkut's University of Technology North Bangkok.

Conflicts of Interest: No conflicts of interests to declare

# Nomenclature

Glass transition temperature [°C]
Laser absorption coefficient
Effective coefficient of laser absorption
Reflective index
Coefficient of reflection
Polarization resistance $[\Omega/cm^2]$
Extinction coefficient
Light wavelength [nm]
Laser radiation intensity [W/sr]
Intensity of incident laser radiation [W/sr]
Plating process rate [s]
Ablation threshold [J/cm <sup>2</sup> ]
Energy per laser pulse [J/cm <sup>2</sup> ]
Activation energy for ablation [J/cm <sup>2</sup> ]
Quantum yield of bond breaking (0 to 1)
Photon energy [eV]
Threshold ablation temperature [°K]
Initial temperature [°K]
Specific heat of polymeric materials [J/kg·K]
Ablation depth [mm]
Photochemical ablation depth [mm]
Photothermal ablation depth [mm]
Tafel equation coefficients
Dependence of deposition rate [dm <sup>2</sup> /L]

# References

- 1. Islam, A.; Hansen, H.N.; Tang, P.T.; Sun, J. Process Chains for the Manufacturing of Molded Interconnect Devices. *Int. J. Adv. Manuf. Technol.* **2009**, *42*, 831–841. [CrossRef]
- 2. Feldmann, K.; Brand, A. Analytical and Experimental Research on Assembly Systems for Molded Interconnection Devices (3D-MID). *CIRP Ann.* **1994**, 43, 15–18. [CrossRef]
- Amend, P.; Pscherer, C.; Rechtenwald, T.; Frick, T.; Schmidt, M. A Fast and Flexible Method for Manufacturing 3D Molded Interconnect Devices by the Use of a Rapid Prototyping Technology. *Phys. Procedia* 2010, *5*, 561–572. [CrossRef]
- 4. Schmidt, B.; Hirsch, S.; Leneke, T. A Multilayer Process for the Connection of Fine-pitch-devices on Molded Interconnect Devices (MIDs). *Circuit World* **2009**, *35*, 23–29. [CrossRef]
- Three-Dimensional Molded Interconnect Devices (3D-MID): Materials, Manufacturing, Assembly and Applications for Injection Molded Circuit Carriers; Franke, J. (Ed.) Carl Hanser Verlag GmbH & Co. KG: München, Germany, 2014. [CrossRef]
- 6. Rong, C.; Tao, X.; Lu, C.; Hu, Z.; Huang, X.; Zeng, Y.; Liu, M. Analysis and Optimized Design of Metamaterials for Mid-Range Wireless Power Transfer Using a Class-E RF Power Amplifier. *Appl. Sci.* **2019**, *9*, 26. [CrossRef]
- 7. Yang, Y.; Gao, Z.; Xue, H.; Zhang, L.; He, M.; Yang, Z.; Singh, R.; Chong, Y.; Zhang, B.; Chen, H. Realization of a Three-Dimensional Photonic Topological Insulator. *Nature* **2019**, *565*, 622. [CrossRef]
- Yu, X.; Wang, D.; Zhao, Z. Research Progress of Semi-Physical Verification Technology Based on Photoelectric Sensing. In *Semi-physical Verification Technology for Dynamic Performance of Internet of Things System*; Springer: Berlin/Heidelberg, Germany, 2019; pp. 1–43.
- 9. Ushimaru, Y.; Takahashi, T.; Souma, Y.; Yanagimoto, Y.; Nagase, H.; Tanaka, K.; Miyazaki, Y.; Makino, T.; Kurokawa, Y.; Yamasaki, M.; et al. Innovation in Surgery/Operating Room Driven by Internet of Things on Medical Devices. *Surg. Endosc.* **2019**. [CrossRef]
- 10. Venkatraman, P.; Sureka, C.S. An In-Vitro Study for Early Detection and to Distinguish Breast and Lung Malignancies Using the Pcb Technology Based Nanodosimeter. *Sci. Rep.* **2019**, *9*, 380. [CrossRef]

- Choi, D.S.; Zhang, Z.; Pachauri, N. Nanoengineered Material Applications in Electronics, Biology, and Energy Harnessing. *Microelectron. Nanoelectron. Mater. Devices Manuf.* 2012, 350–369.
- 12. Ahmad, S. Organic Semiconductors for Device Applications: Current Trends and Future Prospects. J. Polym. Eng. 2014, 34, 279–338. [CrossRef]
- Nadeem, A.; Hussain, M.A.; Owais, O.; Salam, A.; Iqbal, S.; Ahsan, K. Application Specific Study, Analysis and Classification of Body Area Wireless Sensor Network Applications. *Comput. Netw.* 2015, *83*, 363–380. [CrossRef]
- Becker, K.; Braun, T.; Neumann, A.; Ostmann, A.; Koch, M.; Bader, V.; Aschenbrenner, R.; Reichl, H. Duromer MID Technology for System-in-Package Generation. *IEEE Trans. Electron. Packag. Manuf.* 2005, 28, 291–296. [CrossRef]
- 15. Schneider, M. Radio Frequency Identification (RFID) Technology and its Applications in the Commercial Construction Industry. Professur Informatik im Bauwesen. Available online: https://www.db-thueringen.de/receive/dbt\_mods\_00035602 (accessed on 27 May 2020).
- 16. Huang, Y.; Wu, H.; Xiao, L.; Duan, Y.; Zhu, H.; Bian, J.; Ye, D.; Yin, Z. Assembly and Applications of 3D Conformal Electronics on Curvilinear Surfaces. *Mater. Horiz.* **2019**. [CrossRef]
- 17. Chtioui, I.; Bossuyt, F.; Vanfleteren, J.; Bedoui, M.H. 2.5/3D Dynamically Stretchable and Permanently Shaped Electronic Circuits. *Microsyst. Technol.* **2018**, *24*, 831–853. [CrossRef]
- Kurnosenko, A.E.; Arabov, D.I. Optimization of Electronic Components Mounting Sequence for 3D MID Assembly Process. *KnE Eng.* 2018, *3*, 311–321. [CrossRef]
- 19. Tacken, R.; De Zwart, R.; Bolt, P. New Methods for Selective Metallization of 3-D Polymer Microparts. Available online: http://citeseerx.ist.psu.edu/viewdoc/summary?doi=10.1.1.515.1814 (accessed on 27 May 2020).
- 20. Shaw, M.; Zhan, H.; Elmi, M.; Pawar, V.; Essmann, C.; Srinivasan, M.A. Three-Dimensional Behavioural Phenotyping of Freely Moving C. Elegans Using Quantitative Light Field Microscopy. *PLoS ONE* **2018**, *13*, e0200108. [CrossRef]
- 21. Peng, H.; Wang, K.; Huang, Z. An Injection Molding Method to Prepare Chitosan-Zinc Composite Material for Novel Biodegradable Flexible Implant Devices. *Mater. Manuf. Process.* **2019**, *34*, 256–261. [CrossRef]
- 22. Oh, J.W.; Park, J.M.; Shin, D.S.; Noh, J.; Park, S.J. Comparative Study of Nanoparticle Effects on Feedstock Behavior for Injection Molding. *Mater. Manuf. Process.* **2019**, *34*, 414–421. [CrossRef]
- 23. Azad, R.; Shahrajabian, H. Experimental Study of Warpage and Shrinkage in Injection Molding of HDPE/RPET/ Wood Composites with Multiobjective Optimization. *Mater. Manuf. Process.* **2019**, *34*, 274–282. [CrossRef]
- Lu, Y.; Chen, F.; Wu, X.; Zhou, C.; Zhao, H.; Li, L.; Tang, Y. Precise WEDM of Micro-Textured Mould for Micro-Injection Molding of Hydrophobic Polymer Surface. *Mater. Manuf. Process.* 2019, 34, 1342–1351. [CrossRef]
- 25. Chaitanya, S.; Singh, I. Processing of PLA/Sisal Fiber Biocomposites Using Direct-and Extrusion-Injection Molding. *Mater. Manuf. Process.* 2017, 32, 468–474. [CrossRef]
- 26. Liparoti, S.; Sorrentino, A.; Titomanlio, G. Temperature and Pressure Evolution in Fast Heat Cycle Injection Molding. *Mater. Manuf. Process.* **2019**, *34*, 422–430. [CrossRef]
- 27. Tian, Y.-C.; Gao, F. Injection Velocity Control of Thermoplastic Injection Molding via a Double Controller Scheme. *Ind. Eng. Chem. Res.* **1999**, *38*, 3396–3406. [CrossRef]
- Kim, K.; Lee, J.; Ryu, S.; Kim, J. Laser Direct Structuring and Electroless Plating Applicable Super-Engineering Plastic PPS Based Thermal Conductive Composite with Particle Surface Modification. *RSC Adv.* 2018, *8*, 9933–9940. [CrossRef]
- 29. Balzereit, S.; Proes, F.; Altstädt, V.; Emmelmann, C. Properties of Copper Modified Polyamide 12-Powders and Their Potential for the Use as Laser Direct Structurable Electronic Circuit Carriers. *Addit. Manuf.* **2018**, 23, 347–354. [CrossRef]
- Mahanty, S.; Gouthama. Surface Nanostructuring of TIMETAL 834 Using Excimer Laser Treatment. *Surf. Eng.* 2016, 32, 916–922. [CrossRef]
- Bachy, B.; Süß-Wolf, R.; Wang, L.; Fu, Z.; Travitzky, N.; Greil, P.; Franke, J. Novel Ceramic-Based Material for the Applications of Molded Interconnect Devices (3D-MID) Based on Laser Direct Structuring. *Adv. Eng. Mater.* 2018, 20, 1700824. [CrossRef]
- 32. Petit, E. Laser Processing of Metallic and Intermetallic Surfaces. *Mater. Manuf. Process* **1993**, *8*, 299–314. [CrossRef]

- Kromer, R.; Danlos, Y.; Aubignat, E.; Verdy, C.; Costil, S. Coating Deposition and Adhesion Enhancements by Laser Surface Texturing—Metallic Particles on Different Classes of Substrates in Cold Spraying Process. *Mater. Manuf. Process.* 2017, *32*, 1642–1652. [CrossRef]
- 34. Cheng, C.-W.; Chen, J.-S.; Chen, H.-H. Patterning of Crystalline ITO Using Infrared Nanosecond Fiber Laser Pulses. *Mater. Manuf. Process.* **2010**, *25*, 684–688. [CrossRef]
- 35. Nattapat, M.; Marimuthu, S.; Kamara, A.; Esfahani, M.N. Laser Surface Modification of Carbon Fiber Reinforced Composites. *Mater. Manuf. Process.* **2015**, *30*, 1450–1456. [CrossRef]
- 36. Fang, S.; Pérez, V.; Salán, N.; Baehre, D.; Llanes, L. Surface Patterning of Cemented Carbides by Means of Nanosecond Laser. *Mater. Manuf. Process.* **2020**, *35*, 123–129. [CrossRef]
- Sankaranarayanan, S.; Guo, W.; Kar, A. Characteristics of Laser-Fabricated Metal Structures. *Mater. Manuf. Process.* 1998, 13, 537–554. [CrossRef]
- Armeyev, V.Y.; Chapliev, N.; Chistyakov, I.; Konov, V.; Ralchenko, V.; Strelnitsky, V.; Volkov, V.Y. Direct Laser Writing of Microstructures in Diamond-like Carbon Films. *Mater. Manuf. Process.* 1993, *8*, 9–17. [CrossRef]
- Roslizar, A.; Dottermusch, S.; Vüllers, F.; Kavalenka, M.N.; Guttmann, M.; Schneider, M.; Paetzold, U.W.; Hölscher, H.; Richards, B.S.; Klampaftis, E. Self-Cleaning Performance of Superhydrophobic Hot-Embossed Fluoropolymer Films for Photovoltaic Modules. *Sol. Energy Mater. Sol. Cells* 2019, 189, 188–196. [CrossRef]
- 40. Gao, X.; Huang, Y.; He, X.; Fan, X.; Liu, Y.; Xu, H.; Wu, D.; Wan, C. Mechanically Enhanced Electrical Conductivity of Polydimethylsiloxane-Based Composites by a Hot Embossing Process. *Polymers* **2019**, *11*, 56. [CrossRef]
- 41. Hu, M.; Xie, J.; Li, W.; Lu, K. Study on Non-Isothermal Hot-Embossing of Polymer Micro-Prism Array with Efficiency and Accuracy. *J. Mater. Process. Technol.* **2019**, *266*, 675–686. [CrossRef]
- 42. Li, J.; Zhou, Y.; Yang, J.; Ye, R.; Gao, J.; Ren, L.; Liu, B.; Liang, L.; Jiang, L. Fabrication of Gradient Porous Microneedle Array by Modified Hot Embossing for Transdermal Drug Delivery. *Mater. Sci. Eng. C* 2019, *96*, 576–582. [CrossRef]
- Yi, P.; Deng, Y.; Shu, Y.; Peng, L. Experimental Studies on a Novel Roll-to-Roll Powder Hot Embossing for Large-Area Fabrication of Micropyramid Arrays on Polymers. *J. Micromech. Microeng.* 2018, 28, 085007. [CrossRef]
- 44. Lin, T.-Y.; Do, T.; Kwon, P.; Lillehoj, P.B. 3D Printed Metal Molds for Hot Embossing Plastic Microfluidic Devices. *Lab. Chip* **2017**, *17*, 241–247. [CrossRef]
- 45. Çoğun, F.; Yıldırım, E.; Sahir Arikan, M. Investigation on Replication of Microfluidic Channels by Hot Embossing. *Mater. Manuf. Process.* **2017**, *32*, 1838–1844. [CrossRef]
- Ong, N.S.; Koh, Y.H. Experimental Investigation into Micro Injection Molding of Plastic Parts. *Mater. Manuf.* Process. 2005, 20, 245–253. [CrossRef]
- Chen, C.-C.; Su, P.-L.; Chiou, C.-B.; Chiang, K.-T. Experimental Investigation of Designed Parameters on Dimension Shrinkage of Injection Molded Thin-Wall Part by Integrated Response Surface Methodology and Genetic Algorithm: A Case Study. *Mater. Manuf. Process.* 2011, 26, 534–540. [CrossRef]
- Ali, M.; Ahmad, F. Influence of Powder Loading on Rheology and Injection Molding of Fe-50Ni Feedstocks. *Mater. Manuf. Process.* 2020, 35, 579–589. [CrossRef]
- 49. Ong, N.S.; Zhang, H.; Woo, W.H. Plastic Injection Molding of High-Aspect Ratio Micro-Rods. *Mater. Manuf. Process.* **2006**, *21*, 824–831. [CrossRef]
- 50. Islam, A.; Hansen, H.; Tang, P.; Jørgensen, M.; Ørts, S. Two Component Injection Molding for MID Fabrication. *Laser* **2009**, *2*, 3.
- 51. Tenteris-Noebe, A. Review of: "Introduction to Plastics and Composites Mechanical Properties and Engineering Applications" Edited by Edward Miller. *Mater. Manuf. Process.* **1996**, *11*, 891–892. [CrossRef]
- 52. Wissbrock, H.; Naundorf, G. A Fundamentally New Mechanism for Additive Metallization of Polymeric Substrates in Ultra Fine Line Technology Illustrated for 3D-MIDs. *Galvanotechnik* **2000**, *91*, 2449–2451.
- 53. Hoerber, J.; Glasschroeder, J.; Pfeffer, M.; Schilp, J.; Zaeh, M.; Franke, J. Approaches for Additive Manufacturing of 3D Electronic Applications. *Procedia CIRP* **2014**, *17*, 806–811. [CrossRef]
- 54. Lomakin, K.; Pavlenko, T.; Ankenbrand, M.; Sippel, M.; Ringel, J.; Scheetz, M.; Klemm, T.; Gräf, D.; Helmreich, K.; Franke, J.; et al. Evaluation and Characterization of 3-D Printed Pyramid Horn Antennas Utilizing Different Deposition Techniques for Conductive Material. *IEEE Trans. Compon. Packag. Manuf. Technol.* 2018, *8*, 1998–2006. [CrossRef]

- Shin, I.J.; Park, M.S. Direct Conductive Patterning on 3D Printed Structure Using Laser. *Phys. Status Solidi A* 2018, 215, 1700597. [CrossRef]
- 56. Berkelmann, L.; Martinelli, T.; Friedrich, A.; Manteuffel, D. Design and Integration of a Wearable Antenna System for On and Off-Body Communication Based on 3D-MID Technology. In Proceedings of the 12th European Conference on Antennas and Propagation (EuCAP 2018), London, UK, 9–13 April 2018. [CrossRef]
- 57. Yoshida, Y.; Wada, H.; Izumi, K.; Tokito, S. Three-Dimensional Interconnect Layers Inkjet Printed on Plastic Substrates Using Continuous-Wave Xenon Light Sintering. *Jpn. J. Appl. Phys.* **2018**, *58*, 016507. [CrossRef]
- 58. Lu, B.; Lan, H.; Liu, H. Additive Manufacturing Frontier: 3D Printing Electronics. *Opto-Electron. Adv.* 2018, 01, 170004. [CrossRef]
- Kilian, A.; Weinzierl, J.; Schmidt, L. Investigation of the Hot Embossing Technology for Low-Cost Antennas Printed on Polymer Substrates. In Proceedings of the 2008 38th European Microwave Conference, Amsterdam, The Netherlands, 27–31 October 2008; pp. 1–4. [CrossRef]
- Battermann, A.; Gunther, B.; Schafer, H. UV-Curable Conductive Adhesives for 3-D MID Application. In Proceedings of the First International IEEE Conference on Polymers and Adhesives in Microelectronics and Photonics. Incorporating POLY, PEP Adhesives in Electronics. Proceedings (Cat. No.01TH8592), Potsdam, Germany, 21–24 October 2001; pp. 181–184. [CrossRef]
- Lu, Y.; Yun, H.-Y.; Vatani, M.; Kim, H.-C.; Choi, J.-W. Direct-Print/Cure as a Molded Interconnect Device (MID) Process for Fabrication of Automobile Cruise Controllers. *J. Mech. Sci. Technol.* 2015, 29, 5377–5385. [CrossRef]
- Polzinger, B.; Matic, V.; Liedtke, L.; Keck, J.; Hera, D.; Günther, T.; Eberhardt, W.; Kück, H. Printing of Functional Structures on Molded 3D Devices. Available online: www.scientific.net/AMR.1038.37 (accessed on 21 April 2019).
- 63. Petrovskaya, T.; Toropkov, N.; Mironov, E.; Azarmi, F. 3D Printed Biocompatible Polylactide-Hydroxyapatite Based Material for Bone Implants. *Mater. Manuf. Process.* **2018**, *33*, 1899–1904. [CrossRef]
- 64. Goud, V.; Ramasamy, A.; Das, A.; Kalyanasundaram, D. Box-Behnken Technique Based Multi-Parametric Optimization of Electrostatic Spray Coating in the Manufacturing of Thermoplastic Composites. *Mater. Manuf. Process.* **2019**, *34*, 1638–1645. [CrossRef]
- 65. Tamrin, K.; Nukman, Y.; Sheikh, N. Laser Spot Welding of Thermoplastic and Ceramic: An Experimental Investigation. *Mater. Manuf. Process.* **2015**, *30*, 1138–1145. [CrossRef]
- Jakus, A.E. Chapter 1—An Introduction to 3D Printing—Past, Present, and Future Promise. In 3D Printing in Orthopaedic Surgery; Dipaola, M., Wodajo, F.M., Eds.; Elsevier: Amsterdam, The Netherlands, 2019; pp. 1–15.
   [CrossRef]
- 67. Hüske, M.; Kickelhain, J.; Müller, J.; Eßer, G. Laser supported activation and additive metallization of thermoplastics for 3D-MIDs. In Proceedings of the 3rd LANE, Erlangen, Germany, 28–31 August 2001.
- 68. Vanfleteren, J.; Chtioui, I.; Plovie, B.; Yang, Y.; Bossuyt, F.; Vervust, T.; Dunphy, S.; Vandecasteele, B. Arbitrarily Shaped 2.5D Circuits Using Stretchable Interconnections and Embedding in Thermoplastic Polymers. *Procedia Technol.* **2014**, *15*, 208–215. [CrossRef]
- 69. Vanfleteren, J.; Bossuyt, F.; Plovie, B. A New Technology for Rigid 3D Free-Form Electronics Based on the Thermoplastic Deformation of Flat Standard PCB Type Circuits. In Proceedings of the 2016 12th International Congress Molded Interconnect Devices (MID), Wurzburg, Germany, 28–29 September 2016; pp. 1–4. [CrossRef]
- 70. Chinnadurai, T.; Arungalai Vendan, S.; Rusu, C.; Scutelnicu, E. Experimental Investigations on the Polypropylene Behavior during Ultrasonic Welding. *Mater. Manuf. Process.* **2018**, *33*, 718–726. [CrossRef]
- 71. Srinivasan, T.; Palanikumar, K.; Rajagopal, K.; Latha, B. Optimization of Delamination Factor in Drilling GFR–Polypropylene Composites. *Mater. Manuf. Process.* **2017**, *32*, 226–233. [CrossRef]
- 72. Li, T.-T.; Chen, A.-P.; Hwang, P.-W.; Pan, Y.-J.; Hsing, W.-H.; Lou, C.-W.; Chen, Y.-S.; Lin, J.-H. Synergistic Effects of Micro-/Nano-Fillers on Conductive and Electromagnetic Shielding Properties of Polypropylene Nanocomposites. *Mater. Manuf. Process.* **2018**, *33*, 149–155. [CrossRef]
- 73. Mudhukrishnan, M.; Hariharan, P.; Palanikumar, K.; Latha, B. Tool Materials Influence on Surface Roughness and Oversize in Machining Glass Fiber Reinforced Polypropylene (GFR-PP) Composites. *Mater. Manuf. Process.* **2017**, *32*, 988–997. [CrossRef]
- 74. Long, X.; He, L.; Zhang, Y.; Yu, S.; Ge, M. Surface Modification of Polypropylene Non-Woven Fabric for Improving Its Hydrophilicity. *Surf. Eng.* **2018**, *34*, 818–824. [CrossRef]

- Huang, P.; Wu, M.; Pang, Y.; Shen, B.; Wu, F.; Lan, X.; Luo, H.; Zheng, W. Ultrastrong, Flexible and Lightweight Anisotropic Polypropylene Foams with Superior Flame Retardancy. *Compos. Part Appl. Sci. Manuf.* 2019, 116, 180–186. [CrossRef]
- 76. Davis, A.M.; Hanzly, L.E.; DeButts, B.L.; Barone, J.R. Characterization of Dimensional Stability in Flax Fiber Reinforced Polypropylene Composites. *Polym. Compos.* **2019**, *40*, 132–140. [CrossRef]
- 77. Chen, Y.; Guo, X.; Peng, Y.; Cao, J. Water Absorption and Mold Susceptibility of Wood Flour/Polypropylene Composites Modified with Silane-Wax Emulsions. *Polym. Compos.* **2019**, *40*, 141–148. [CrossRef]
- 78. Kaymakci, A.; Birinci, E.; Ayrilmis, N. Surface Characteristics of Wood Polypropylene Nanocomposites Reinforced with Multi-Walled Carbon Nanotubes. *Compos. Part B Eng.* **2019**, *157*, 43–46. [CrossRef]
- Dai, Z.; Su, J.; Zhu, X.; Xu, K.; Zhu, J.; Huang, C.; Ke, Q. Multifunctional Polyethylene (PE)/Polypropylene (PP) Bicomponent Fiber Filter with Anchored Nanocrystalline MnO<sub>2</sub> for Effective Air Purification. *J. Mater. Chem. A* 2018, *6*, 14856–14866. [CrossRef]
- López Nava, J.; Méndez González, J.; Ruelas Chacón, X.; Nájera Luna, J. Assessment of Edible Fungi and Films Bio-Based Material Simulating Expanded Polystyrene. *Mater. Manuf. Process.* 2016, *31*, 1085–1090. [CrossRef]
- 81. Guo, Y.; Yang, L.; Wang, D. Preparation and Hydrophobic Behaviours of Polystyrene Composite Coating. *Surf. Eng.* **2016**, *32*, 95–101. [CrossRef]
- Kausar, A.; Wajid-Ullah; Muhammad, B.; Siddiq, M. Influence of Processing Technique on the Physical Properties of Modified Polystyrene/Exfoliated Graphite Nanocomposites. *Mater. Manuf. Process.* 2015, 30, 346–355. [CrossRef]
- 83. Sato, S.; Yamamoto, Y.; Kaneko, F. Cocrystal Formation between Syndiotactic Polystyrene and Polyethylene Glycol from Binary Solute Solutions. *Chem. Lett.* **2019**, *48*, 177–180. [CrossRef]
- Abdeldaym, A.; Elhady, M.A. Investigation of the Effect of Aluminum Oxide Nanoparticle Additives and Gamma Irradiation on the Thermal Behavior of Syndiotactic Polystyrene. *Polym. Compos.* 2019, 40, 3588–3600. [CrossRef]
- Chen, P.-H.; Lan, Y.-K.; Lin, S.-J.; Tsai, J.-C.; Jeng, U.-S.; Su, A.-C. Crystallization of α versus β Phases in Syndiotactic Poly(Styrene-Stat-3-Methylstyrene) and Poly(Styrene-Stat-4-Methylstyrene). ACS Appl. Polym. Mater. 2019, 1, 251–258. [CrossRef]
- 86. Naga, N.; Sakurai, T.; Hashimoto, T.; Noguchi, K. Crystalline Structure and Phase Transition of Syndiotactic Styrene-based Copolymers. *Polym. Int.* **2019**, *68*, 71–78. [CrossRef]
- 87. Liu, C.; Kremer, K.; Bereau, T. Polymorphism of Syndiotactic Polystyrene Crystals from Multiscale Simulations. *Adv. Theory Simul.* **2018**, *1*, 1800024. [CrossRef]
- 88. Shi, Z.; Guo, F.; Li, Y.; Hou, Z. Synthesis of Amino-Containing Syndiotactic Polystyrene as Efficient Polymer Support for Palladium Nanoparticles. *J. Polym. Sci. Part Polym. Chem.* **2015**, *53*, 5–9. [CrossRef]
- 89. Itagaki, H.; Sano, T.; Okabe, T.; Sano, S.; Ebihara, H.; Tomono, F.; Dohra, H. Polymerization of Aniline in Tubular Cavities of the Crystalline Phase of Syndiotactic Polystyrene: Proposal of a Preparation Method of Sophisticated Polymer Composites. *ACS Macro Lett.* **2017**, *6*, 1099–1103. [CrossRef]
- 90. Li, Y.; Huang, X.; Zeng, L.; Li, R.; Tian, H.; Fu, X.; Wang, Y.; Zhong, W.-H. A Review of the Electrical and Mechanical Properties of Carbon Nanofiller-Reinforced Polymer Composites. *J. Mater. Sci.* **2019**, *54*, 1036–1076. [CrossRef]
- 91. Bin Rusayyis, M.A.; Schiraldi, D.A.; Maia, J. Property/Morphology Relationships in SEBS-Compatibilized HDPE/Poly(Phenylene Ether) Blends. *Macromolecules* **2018**, *51*, 6513–6523. [CrossRef]
- 92. Wang, J.; Tsou, A.H.; Passino, H.L.; Favis, B.D. PPE-g-HDPE in High-Performance Poly(p-Phenylene Ether)/Polyethylene Blends: Synthesis and Compatibilization Effects. *Polymer* **2018**, *138*, 92–102. [CrossRef]
- 93. Jasinska-Walc, L.; Bouyahyi, M.; Lorenc, P.; Heeneman, A.L.; Duchateau, R.; Różański, A.; Bernaerts, K.V. Synthesis of Isotactic Polypropylene-Block-Polystyrene Block Copolymers as Compatibilizers for Isotactic Polypropylene/Polyphenylene Oxide Blends. *Polymer* 2018, 147, 121–132. [CrossRef]
- Wang, Y.; Tao, Y.; Zhou, J.; Sun, J.; Fang, Q. Biobased Anethole-Functionalized Poly(Phenylene Oxides): New Low Dielectric Materials with High Tg and Good Dimensional Stability. ACS Sustain. Chem. Eng. 2018, 6, 9277–9282. [CrossRef]
- 95. Seike, Y.; Okude, Y.; Iwakura, I.; Chiba, I.; Ikeno, T.; Yamada, T. Synthesis of Polyphenylene Ether Derivatives: Estimation of Their Dielectric Constants. *Macromol. Chem. Phys.* **2003**, *204*, 1876–1881. [CrossRef]

- 96. Stack, S.; O'Donoghue, O.; Birkinshaw, C. The Thermal Stability and Thermal Degradation of Blends of Syndiotactic Polystyrene and Polyphenylene Ether. *Polym. Degrad. Stab.* **2003**, *79*, 29–36. [CrossRef]
- 97. Hou, H.; Vacandio, F.; Vona, M.L.D.; Knauth, P. Sulfonated Polyphenyl Ether by Electropolymerization. *Electrochim. Acta* 2012, *81*, 58–63. [CrossRef]
- Nunoshige, J.; Akahoshi, H.; Shibasaki, Y.; Ueda, M. Efficient Oxidative Coupling Polymerization for Synthesis of Thermosetting Poly(Phenylene Ether) Copolymer with a Low Dielectric Loss. J. Polym. Sci. Part Polym. Chem. 2008, 46, 5278–5282. [CrossRef]
- Kim, I.G.; Hong, S.Y.; Park, B.O.; Choi, H.J.; Lee, J.H. Polyphenylene Ether/Glycol Modified Polyethylene Terephthalate Blends and Their Physical Characteristics. J. Macromol. Sci. Part B 2012, 51, 798–806. [CrossRef]
- 100. Durante, M.; Formisano, A.; Boccarusso, L.; Langella, A. Influence of Cold-Rolling on Incremental Sheet Forming of Polycarbonate. *Mater. Manuf. Process.* **2020**, *35*, 328–336. [CrossRef]
- 101. Małachowska, A.; Winnicki, M.; Stachowicz, M.; Korzeniowski, M. Metallisation of Polycarbonates Using a Low Pressure Cold Spray Method. *Surf. Eng.* **2018**, *34*, 251–258. [CrossRef]
- Sabry, R.S.; Al-Mosawi, M.I. Novel Approach to Fabricate a Stable Superhydrophobic Polycarbonate. Surf. Eng. 2018, 34, 151–157. [CrossRef]
- 103. Ul Haq, A.; Boyd, A.; Acheson, J.; McLaughlin, J.; Meenan, B.J. Corona Discharge-Induced Functional Surfaces of Polycarbonate and Cyclic Olefins Substrates. *Surf. Coat. Technol.* 2019, 362, 185–190. [CrossRef]
- Hazer, S.; Coban, M.; Aytac, A. Effects of Ammonium Polyphosphate and Triphenyl Phosphate on the Flame Retardancy, Thermal, and Mechanical Properties of Glass Fiber–Reinforced PLA/PC Composites. *Fire Mater.* 2019, 43, 277–282. [CrossRef]
- 105. Zimmerer, C.; Matulaitiene, I.; Niaura, G.; Reuter, U.; Janke, A.; Boldt, R.; Sablinskas, V.; Steiner, G. Nondestructive Characterization of the Polycarbonate—Octadecylamine Interface by Surface Enhanced Raman Spectroscopy. *Polym. Test.* 2019, *73*, 152–158. [CrossRef]
- 106. Pedreño-Rojas, M.A.; Morales-Conde, M.J.; Pérez-Gálvez, F.; Rubio-de-Hita, P. Influence of Polycarbonate Waste on Gypsum Composites: Mechanical and Environmental Study. J. Clean. Prod. 2019, 218, 21–37. [CrossRef]
- 107. Abenojar, J.; Torregrosa-Coque, R.; Martínez, M.A.; Martín-Martínez, J.M. Surface Modifications of Polycarbonate (PC) and Acrylonitrile Butadiene Styrene (ABS) Copolymer by Treatment with Atmospheric Plasma. *Surf. Coat. Technol.* 2009, 203, 2173–2180. [CrossRef]
- Aquino, F.W.B.; Paranhos, C.M.; Pereira-Filho, E.R. Method for the Production of Acrylonitrile–Butadiene– Styrene (ABS) and Polycarbonate (PC)/ABS Standards for Direct Sb Determination in Plastics from e-Waste Using Laser-Induced Breakdown Spectroscopy. J. Anal. At. Spectrom. 2016, 31, 1228–1233. [CrossRef]
- 109. Xiao, C.; Leng, X.; Zhang, X.; Zheng, K.; Tian, X. Improved Thermal Properties by Controlling Selective Distribution of AlN and MWCNT in Immiscible Polycarbonate (PC)/Polyamide 66 (PA66) Composites. *Compos. Part Appl. Sci. Manuf.* 2018, 110, 133–141. [CrossRef]
- 110. Standau, T.; Hädelt, B.; Fafara, M.; Altstädt, V. Development of a Bead Foam Based on the Engineering Polymer Polybutylene Terephthalate. *AIP Conf. Proc.* **2019**, 2055, 060004. [CrossRef]
- 111. Gemmeke, N.; Feldmann, M.; Heim, H.-P. Processing and Characterization of Engineering Biocomposites Based on Polybutylenterephthalat (PBT) and Polytrimethylentherephthalat (PTT) with Regenerated Cellulose Fibers Modified with Maleic Anhydride Grafted Polyethylene as a Processing Agent. *Compos. Part Appl. Sci. Manuf.* 2019, 118, 327–335. [CrossRef]
- 112. Kchaou, M.; Torres, E.; Ylla, N.; Colella, M.; Da Cruz-Boisson, F.; Cassagnau, P.; Espuche, E.; Bounor-Legaré, V. Enhanced Hydrophobicity and Reduced Water Transport Properties in Alkylalkoxysilane Modified Poly(Butylene Terephthalate) Using Reactive Extrusion. *Mater. Chem. Phys.* 2019, 223, 597–606. [CrossRef]
- Zhao, Y.; Ma, X.; Xu, T.; Salem, D.R.; Fong, H. Hybrid Multi-Scale Thermoplastic Composites Reinforced with Interleaved Nanofiber Mats Using in-Situ Polymerization of Cyclic Butylene Terephthalate. *Compos. Commun.* 2019, 12, 91–97. [CrossRef]
- De Almeida, A.; Nébouy, M.; Baeza, G.P. Bimodal Crystallization Kinetics of PBT/PTHF Segmented Block Copolymers: Impact of the Chain Rigidity. *Macromolecules* 2019, 52, 1227–1240. [CrossRef]
- 115. Slot, H.M.; IJzerman, R.M.; le Feber, M.; Nord-Varhaug, K.; van der Heide, E. Rain Erosion Resistance of Injection Moulded and Compression Moulded Polybutylene Terephthalate PBT. Wear 2018, 414–415, 234–242. [CrossRef]

- 116. Dechet, M.A.; Gómez Bonilla, J.S.; Lanzl, L.; Drummer, D.; Bück, A.; Schmidt, J.; Peukert, W. Spherical Polybutylene Terephthalate (PBT)—Polycarbonate (PC) Blend Particles by Mechanical Alloying and Thermal Rounding. *Polymers* 2018, *10*, 1373. [CrossRef] [PubMed]
- 117. Rosenboom, J.-G.; De Lorenzi, L.; Storti, G.; Morbidelli, M. Reaction Kinetics and Simulations of Ring-Opening Polymerization for the Synthesis of Polybutylene Terephthalate. *Polymer* **2018**, *146*, 120–132. [CrossRef]
- Chiou, K.-C.; Chang, F.-C. Reactive Compatibilization of Polyamide-6 (PA 6)/Polybutylene Terephthalate (PBT) Blends by a Multifunctional Epoxy Resin. J. Polym. Sci. Part B Polym. Phys. 2000, 38, 23–33. [CrossRef]
- Davoodi, M.M.; Sapuan, S.M.; Ahmad, D.; Aidy, A.; Khalina, A.; Jonoobi, M. Effect of Polybutylene Terephthalate (PBT) on Impact Property Improvement of Hybrid Kenaf/Glass Epoxy Composite. *Mater. Lett.* 2012, 67, 5–7. [CrossRef]
- 120. Kuram, E.; Timur, G.; Ozcelik, B.; Yilmaz, F. Influences of Injection Conditions on Strength Properties of Recycled and Virgin PBT/PC/ABS. *Mater. Manuf. Process.* **2014**, *29*, 1260–1268. [CrossRef]
- 121. Harris, M.; Potgieter, J.; Ray, S.; Archer, R.; Arif, K.M. Preparation and Characterization of Thermally Stable ABS/HDPE Blend for Fused Filament Fabrication. *Mater. Manuf. Process.* **2020**, *35*, 230–240. [CrossRef]
- 122. Li, J.; Li, C.; Liao, Q.; Xu, Z. Environmentally-Friendly Technology for Rapid on-Line Recycling of Acrylonitrile-Butadiene-Styrene, Polystyrene and Polypropylene Using near-Infrared Spectroscopy. J. Clean. Prod. 2019, 213, 838–844. [CrossRef]
- 123. Selvamani, S.K.; Samykano, M.; Subramaniam, S.R.; Ngui, W.K.; Kadirgama, K.; Sudhakar, K.; Idris, M.S. Preliminary Investigation of Acrylonitrile Butadiene Styrene (ABS) Properties. *AIP Conf. Proc.* 2019, 2059, 020040. [CrossRef]
- Huang, B.; Meng, S.; He, H.; Jia, Y.; Xu, Y.; Huang, H. Study of Processing Parameters in Fused Deposition Modeling Based on Mechanical Properties of Acrylonitrile-Butadiene-Styrene Filament. *Polym. Eng. Sci.* 2019, 59, 120–128. [CrossRef]
- Wang, Y.; Li, Y.; Wang, W.; Lv, L.; Li, C.; Zhang, J. Recycled Polycarbonate/Acrylonitrile–Butadiene–Styrene Reinforced and Toughened through Chemical Compatibilization. J. Appl. Polym. Sci. 2019, 136, 47537. [CrossRef]
- 126. Jung, D.-J.; Cheon, J.; Na, S.-J. Effect of Surface Pre-Oxidation on Laser Assisted Joining of Acrylonitrile Butadiene Styrene (ABS) and Zinc-Coated Steel. *Mater. Des.* **2016**, *99*, 1–9. [CrossRef]
- Jyoti, J.; Basu, S.; Singh, B.P.; Dhakate, S.R. Superior Mechanical and Electrical Properties of Multiwall Carbon Nanotube Reinforced Acrylonitrile Butadiene Styrene High Performance Composites. *Compos. Part B Eng.* 2015, *83*, 58–65. [CrossRef]
- 128. Wang, J.; Li, Y.; Song, J.; He, M.; Song, J.; Xia, K. Recycling of Acrylonitrile–Butadiene–Styrene (ABS) Copolymers from Waste Electrical and Electronic Equipment (WEEE), through Using an Epoxy-Based Chain Extender. *Polym. Degrad. Stab.* 2015, 112, 167–174. [CrossRef]
- Li, Y.; Wu, X.; Song, J.; Li, J.; Shao, Q.; Cao, N.; Lu, N.; Guo, Z. Reparation of Recycled Acrylonitrile-Butadiene-Styrene by Pyromellitic Dianhydride: Reparation Performance Evaluation and Property Analysis. *Polymer* 2017, 124, 41–47. [CrossRef]
- 130. Wang, N.; Liu, Y.; Liu, Y.; Wang, Q. A Shear Pan Mill for Preparation of Ultrafine Polyamide 66 Powder Using Sodium Sulfate Ionic Crystals as Grinding Aid. *Mater. Manuf. Process.* **2017**, *32*, 115–120. [CrossRef]
- 131. Negi, S.; Sharma, R.K. Influence of Processing Variables on Dynamic Mechanical Response of Laser-Sintered Glass-Filled Polyamide. *Mater. Manuf. Process.* **2015**, *30*, 1431–1441. [CrossRef]
- 132. Rytlewski, P. Influence of Glass Fibre Content on Catalytic and Adhesion Properties of Laser Irradiated Polyamide Composites. *Surf. Eng.* **2013**, *29*, 713–719. [CrossRef]
- 133. Zheng, R.; Zhang, X.; Zhang, Z.; Niu, H.; Wang, C.; Wang, W. Preparation and Multifunction of Electrochromic Polyamides Containing Flexible Backbone Chains with Electrochemical, Fluorescence and Memory Properties. *Appl. Surf. Sci.* 2019, 478, 906–915. [CrossRef]
- 134. Krause, B.; Kroschwald, L.; Pötschke, P. The Influence of the Blend Ratio in PA6/PA66/MWCNT Blend Composites on the Electrical and Thermal Properties. *Polymers* **2019**, *11*, 122. [CrossRef] [PubMed]
- 135. Zou, G.; Wang, P.; Feng, W.; Ren, Z.; Ji, J. Bio-Based Transparent Polyamide 10T/10I/1012 with High Performance. J. Appl. Polym. Sci. 2019, 136, 47305. [CrossRef]
- Nagel, K.; Spange, S. Polyamide/Silica Hybrid Materials by Anionic Melt Polymerization of Lactam-Substituted Silane Monomers with ε-Caprolactam. *Eur. Polym. J.* 2019, 113, 385–394. [CrossRef]

- Winnacker, M.; Rieger, B. Biobased Polyamides: Recent Advances in Basic and Applied Research. *Macromol. Rapid Commun.* 2016, 37, 1391–1413. [CrossRef] [PubMed]
- 138. Battegazzore, D.; Salvetti, O.; Frache, A.; Peduto, N.; De Sio, A.; Marino, F. Thermo-Mechanical Properties Enhancement of Bio-Polyamides (PA10.10 and PA6.10) by Using Rice Husk Ash and Nanoclay. *Compos. Part Appl. Sci. Manuf.* 2016, *81*, 193–201. [CrossRef]
- 139. Van Velthoven, J.L.J.; Gootjes, L.; Noordover, B.A.J.; Meuldijk, J. Bio-Based, Amorphous Polyamides with Tunable Thermal Properties. *Eur. Polym. J.* **2015**, *66*, 57–66. [CrossRef]
- Wu, J.; Mizuno, Y.; Nakamura, K. Enhancement in Mechanical Quality Factors of Poly Phenylene Sulfide under High-Amplitude Ultrasonic Vibration through Thermal Annealing. *Ultrasonics* 2019, *91*, 52–61. [CrossRef]
- 141. Zhao, L.; Huang, Z.; Xiong, S.; Peng, J.; Sun, J.; Yin, X.; Guo, G.; Xiong, B.; Song, X.; Huang, L.; et al. Polyphenylene Sulfide Composite Laminate from Flexible Nonwovens and Carbon Fiber Fabrics Prepared by Thermal Lamination and Thermal Treatment. *Polym. Bull.* **2019**. [CrossRef]
- 142. Huang, H.; Li, Y.; Zhao, L.; Yu, Y.; Xu, J.; Yin, X.; Chen, S.; Wu, J.; Yue, H.; Wang, H.; et al. A Facile Fabrication of Chitosan Modified PPS-Based Microfiber Membrane for Effective Antibacterial Activity and Oil-in-Water Emulsion Separation. *Cellulose* 2019, 26, 2599–2611. [CrossRef]
- 143. Xing, J.; Ni, Q.-Q.; Deng, B.; Liu, Q. Morphology and Properties of Polyphenylene Sulfide (PPS)/Polyvinylidene Fluoride (PVDF) Polymer Alloys by Melt Blending. *Compos. Sci. Technol.* **2016**, *134*, 184–190. [CrossRef]
- 144. Gu, J.; Guo, Y.; Yang, X.; Liang, C.; Geng, W.; Tang, L.; Li, N.; Zhang, Q. Synergistic Improvement of Thermal Conductivities of Polyphenylene Sulfide Composites Filled with Boron Nitride Hybrid Fillers. *Compos. Part Appl. Sci. Manuf.* 2017, 95, 267–273. [CrossRef]
- 145. Luo, W.; Liu, Q.; Li, Y.; Zhou, S.; Zou, H.; Liang, M. Enhanced Mechanical and Tribological Properties in Polyphenylene Sulfide/Polytetrafluoroethylene Composites Reinforced by Short Carbon Fiber. *Compos. Part B Eng.* 2016, *91*, 579–588. [CrossRef]
- 146. Huang, H.; Liu, M.; Li, Y.; Yu, Y.; Yin, X.; Wu, J.; Chen, S.; Xu, J.; Wang, L.; Wang, H. Polyphenylene Sulfide Microfiber Membrane with Superhydrophobicity and Superoleophilicity for Oil/Water Separation. *J. Mater. Sci.* 2018, 53, 13243–13252. [CrossRef]
- 147. Chen, D.; Kang, Z.; Hirahara, H.; Aisawa, S.; Li, W. Adsorption Behaviors of Deposition-Targeted Metallic Ions onto Thiol-Containing Silane Modified Liquid Crystal Polymer Surfaces. *Appl. Surf. Sci.* 2019, 479, 368–374. [CrossRef]
- 148. Li, X.; Ma, S.; Hu, J.; Ni, Y.; Lin, Z.; Yu, H. Photo-Activated Bimorph Composites of Kapton and Liquid-Crystalline Polymer towards Biomimetic Circadian Rhythms of Albizia Julibrissin Leaves. *J. Mater. Chem. C* 2019, *7*, 622–629. [CrossRef]
- 149. Gantenbein, S.; Masania, K.; Woigk, W.; Sesseg, J.P.W.; Tervoort, T.A.; Studart, A.R. Three-Dimensional Printing of Hierarchical Liquid-Crystal-Polymer Structures. *Nature* **2018**, *561*, 226. [CrossRef]
- 150. Xu, F.; Xin, Y.; Li, T. Friction-Induced Surface Textures of Liquid Crystalline Polymer Evaluated by Atomic Force Microscopy, Spectroscopy and Nanoindentation. *Polym. Test.* **2018**, *68*, 146–152. [CrossRef]
- 151. Fan, W.; Tong, X.; Li, G.; Zhao, Y. Photoresponsive Liquid Crystalline Polymer Single-Chain Nanoparticles. *Polym. Chem.* **2017**, *8*, 3523–3529. [CrossRef]
- 152. Li, C.; Yun, J.-H.; Kim, H.; Cho, M. Light Propagation and Photoactuation in Densely Cross-Linked Azobenzene-Functionalized Liquid-Crystalline Polymers: Contribution of Host and Concerted Isomerism. *Macromolecules* **2016**, *49*, 6012–6020. [CrossRef]
- Xiang, Z.; Chen, S.; Luo, Y.; Li, P.; Zhang, H. Influence of the Side-Chain Structure and Molecular Weight on the Re-Entrant Behaviors of Mesogen-Jacketed Liquid Crystalline Polymers. *RSC Adv.* 2016, *6*, 78516–78527. [CrossRef]
- 154. Guo, D.; Khan, A.U.; Liu, T.; Zhou, Z.; Liu, G. Sub-10 Nm Domains in High-Performance Polyetherimides. *Polym. Chem.* **2019**, *10*, 379–385. [CrossRef]
- Vega, J.; Andrio, A.; Lemus, A.A.; Díaz, J.A.I.; del Castillo, L.F.; Gavara, R.; Compañ, V. Modification of Polyetherimide Membranes with ZIFs Fillers for CO<sub>2</sub> Separation. *Sep. Purif. Technol.* 2019, 212, 474–482. [CrossRef]
- 156. Cacciotti, I.; Rinaldi, M.; Fabbrizi, J.; Nanni, F. Innovative Polyetherimide and Diatomite Based Composites: Influence of the Diatomite Kind and Treatment. *J. Mater. Res. Technol.* **2019**. [CrossRef]

- 157. Liu, J.; Su, J.; Li, R.; Zhao, L.; Lu, Y.; Liu, X. Characterization of Cavities in Polyetherimide and the Influence on Insulation Performance of Ultra-High Voltage Pulse. *Mater. Lett.* **2019**, *240*, 153–156. [CrossRef]
- 158. Jiao, Y.; Shi, H.; Tang, J.; Li, F.; Wang, S. Improvement of Radio Frequency (RF) Heating Uniformity on Low Moisture Foods with Polyetherimide (PEI) Blocks. *Food Res. Int.* **2015**, *74*, 106–114. [CrossRef]
- 159. Karisma, D.; Febrianto, G.; Mangindaan, D. Polyetherimide Thin Film Composite (PEI-TFC) Membranes for Nanofiltration Treatment of Dyes Wastewater. *IOP Conf. Ser. Earth Environ. Sci.* 2018, 195, 012057. [CrossRef]
- 160. Blanco, I.; Cicala, G.; Ognibene, G.; Rapisarda, M.; Recca, A. Thermal Properties of Polyetherimide/ Polycarbonate Blends for Advanced Applications. *Polym. Degrad. Stab.* **2018**, 154, 234–238. [CrossRef]
- 161. Zhao, Z.; Wang, H.; Liu, Z.; Zhang, X.; Zhang, W.; Chen, X.; Zhu, Y. Durable Fluorine-Free Superhydrophobic Polyethersulfone (PES) Composite Coating with Uniquely Weathering Stability, Anti-Corrosion and Wear-Resistance. *Prog. Org. Coat.* 2019, 127, 16–26. [CrossRef]
- 162. Zangeneh, H.; Zinatizadeh, A.A.; Zinadini, S.; Feyzi, M.; Bahnemann, D.W. Preparation and Characterization of a Novel Photocatalytic Self-Cleaning PES Nanofiltration Membrane by Embedding a Visible-Driven Photocatalyst Boron Doped-TiO<sub>2</sub>SiO<sub>2</sub>/CoFe<sub>2</sub>O<sub>4</sub> Nanoparticles. *Sep. Purif. Technol.* **2019**, 209, 764–775. [CrossRef]
- 163. Ballout, W.; Van Velthem, P.; Magnin, D.; Henry, E.; Sclavons, M.; Pardoen, T.; Bailly, C. Specific Influence of Polyethersulfone Functionalization on the Delamination Toughness of Modified Carbon Fiber Reinforced Polymer Processed by Resin Transfer Molding. *Polym. Eng. Sci.* 2019, *59*, 996–1009. [CrossRef]
- 164. Khorshidi, B.; Hosseini, S.A.; Ma, G.; McGregor, M.; Sadrzadeh, M. Novel Nanocomposite Polyethersulfone-Antimony Tin Oxide Membrane with Enhanced Thermal, Electrical and Antifouling Properties. *Polymer* 2019, 163, 48–56. [CrossRef]
- 165. Lin, J.; Ye, W.; Zhong, K.; Shen, J.; Jullok, N.; Sotto, A.; Van der Bruggen, B. Enhancement of Polyethersulfone (PES) Membrane Doped by Monodisperse Stöber Silica for Water Treatment. *Chem. Eng. Process. Process Intensif.* 2016, 107, 194–205. [CrossRef]
- 166. Rajabi, H.; Ghaemi, N.; Madaeni, S.S.; Daraei, P.; Astinchap, B.; Zinadini, S.; Razavizadeh, S.H. Nano-ZnO Embedded Mixed Matrix Polyethersulfone (PES) Membrane: Influence of Nanofiller Shape on Characterization and Fouling Resistance. *Appl. Surf. Sci.* 2015, 349, 66–77. [CrossRef]
- 167. Nasrollahi, N.; Vatanpour, V.; Aber, S.; Mahmoodi, N.M. Preparation and Characterization of a Novel Polyethersulfone (PES) Ultrafiltration Membrane Modified with a CuO/ZnO Nanocomposite to Improve Permeability and Antifouling Properties. *Sep. Purif. Technol.* **2018**, *192*, 369–382. [CrossRef]
- Chen, D.; Pi, C.; Chen, M.; He, L.; Xia, F.; Peng, S. Amplitude-Dependent Damping Properties of Ramie Fiber-Reinforced Thermoplastic Composites with Varying Fiber Content. *Polym. Compos.* 2019, 40, 2681–2689. [CrossRef]
- 169. Baena, J.; Jiménez, G.; López-Ruiz, E.; Antich, C.; Griñán-Lisón, C.; Perán, M.; Gálvez-Martín, P.; Marchal, J. Volume-by-Volume Bioprinting of Chondrocytes-Alginate Bioinks in High Temperature Thermoplastic Scaffolds for Cartilage Regeneration. *Exp. Biol. Med.* **2019**, 244, 13–21. [CrossRef]
- 170. Mamalis, D.; Obande, W.; Koutsos, V.; Blackford, J.R.; O Brádaigh, C.M.; Ray, D. Novel Thermoplastic Fibre-Metal Laminates Manufactured by Vacuum Resin Infusion: The Effect of Surface Treatments on Interfacial Bonding. *Mater. Des.* 2019, 162, 331–344. [CrossRef]
- 171. Jose, A.J.; Alagar, M.; Thomas, P.S. Preparation and Characterization of Organoclay Filled Polysulfone Nanocomposites. *Mater. Manuf. Process.* **2012**, *27*, 247–254. [CrossRef]
- 172. Barbe, A.; Bökamp, K.; Kummerlöwe, C.; Sollmann, H.; Vennemann, N.; Vinzelberg, S. Investigation of Modified SEBS-Based Thermoplastic Elastomers by Temperature Scanning Stress Relaxation Measurements. *Polym. Eng. Sci.* 2005, 45, 1498–1507. [CrossRef]
- 173. Kashi, S.; Hadigheh, S.A.; Varley, R. Microwave Attenuation of Graphene Modified Thermoplastic Poly(Butylene Adipate-Co-Terephthalate) Nanocomposites. *Polymers* **2018**, *10*, 582. [CrossRef] [PubMed]
- 174. Peters, E.N. 1—Engineering Thermoplastics—Materials, Properties, Trends. In *Applied Plastics Engineering Handbook*, 2nd ed.; Kutz, M., Ed.; Plastics Design Library; William Andrew Publishing: Norwich, NY, USA, 2017; pp. 3–26. [CrossRef]
- 175. Rao, Y.S.; Mohan, N.S.; Shetty, N.; Shivamurthy, B. Drilling and Structural Property Study of Multi-Layered Fiber and Fabric Reinforced Polymer Composite-a Review. *Mater. Manuf. Process.* 2019, 34, 1549–1579. [CrossRef]

- Natarajan, S. Fundamental Principles of Polymeric Materials by Christopher S. Brazel and Stephen L. Rosen. Mater. Manufact. Process. 2015, 30. [CrossRef]
- 177. Li, R.; Wu, G.; Hao, Y.; Peng, J.; Zhai, M. Chapter Five Radiation Degradation or Modification of Poly(Tetrafluoroethylene) and Natural Polymers. In *Radiation Technology for Advanced Materials*; Wu, G., Zhai, M., Wang, M., Eds.; Academic Press: Cambridge, MA, USA, 2019; pp. 141–182. [CrossRef]
- 178. Ghazali, Z.; Johnson, A.F.; Dahlan, K.Z. Radiation Crosslinked Thermoplastics Natural Rubber (TPNR) Foams. *Radiat. Phys. Chem.* **1999**, *55*, 73–79. [CrossRef]
- 179. Voit, W.; Ware, T.; Gall, K. Radiation Crosslinked Shape-Memory Polymers. *Polymer* **2010**, *51*, 3551–3559. [CrossRef]
- Gehring, J.; Zyball, A. Radiation Crosslinking of Polymers-Status, Current Issues, Trends and Challenges. *Radiat. Phys. Chem.* 1995, 46, 931–936. [CrossRef]
- Shi, X.; Wang, X.; Fu, C.; Ran, X. Dual-Shape Memory Effect in Radiation Crosslinked Thermoplastic Blends: Fabrication, Optimization and Mechanisms. *RSC Adv.* 2015, *5*, 61601–61611. [CrossRef]
- Leisen, C.; Seefried, A.; Drummer, D. Post-Crosslinking Behavior of Radiation Crosslinked Polyamide 66 during Vibration Welding. *Polym. Eng. Sci.* 2016, 56, 735–742. [CrossRef]
- Mohsin, M.A.A.; Iannucci, L.; Greenhalgh, E.S. Fibre-Volume-Fraction Measurement of Carbon Fibre Reinforced Thermoplastic Composites Using Thermogravimetric Analysis. *Heliyon* 2019, 5, e01132. [CrossRef]
- Cousins, D.S.; Suzuki, Y.; Murray, R.E.; Samaniuk, J.R.; Stebner, A.P. Recycling Glass Fiber Thermoplastic Composites from Wind Turbine Blades. J. Clean. Prod. 2019, 209, 1252–1263. [CrossRef]
- Gaugler, M.; Luedtke, J.; Grigsby, W.J.; Krause, A. A New Methodology for Rapidly Assessing Interfacial Bonding within Fibre-Reinforced Thermoplastic Composites. *Int. J. Adhes.* 2019, 89, 66–71. [CrossRef]
- Arhant, M.; Briançon, C.; Burtin, C.; Davies, P. Carbon/Polyamide 6 Thermoplastic Composite Cylinders for Deep Sea Applications. *Compos. Struct.* 2019, 212, 535–546. [CrossRef]
- Unal, H.; Mimaroglu, A. Comparison of Tribological Performance of Some Neat Polymer and Polymers Composites. *Surf. Eng.* 2013, 29, 455–461. [CrossRef]
- Kumar, S.; Chauhan, S.; Rakesh, P.; Singh, I.; Davim, J. Drilling of Glass Fiber/Vinyl Ester Composites with Fillers. *Mater. Manuf. Process.* 2012, 27, 314–319. [CrossRef]
- Perisic, S.D.; Radovic, I.; Petrovic, M.; Marinkovic, A.; Stojanovic, D.; Uskokovic, P.; Radojevic, V. Processing of Hybrid Wood Plastic Composite Reinforced with Short PET Fibers. *Mater. Manuf. Process.* 2018, 33, 572–579. [CrossRef]
- Singh, K.; Nanda, T.; Mehta, R. Processing of Polyethylene Terephthalate Fiber Reinforcement to Improve Compatibility with Constituents of GFRP Nanocomposites. *Mater. Manuf. Process.* 2018, 33, 165–173. [CrossRef]
- 191. Jiang, M.; Li, H.; Fang, D.; Liu, L.; Tai, Q.; Li, L.; Dong, L.; Xie, H.; Xiong, C. Structure–Property Relationship in Injection-Molded Polypropylene/Clay Composite Foams. *Mater. Manuf. Process.* 2014, 29, 160–165. [CrossRef]
- 192. Kaci, S.; Rahmoune, R.; Kezzoula, F.; Boudiaf, Y.; Keffous, A.; Manseri, A.; Menari, H.; Cheraga, H.; Guerbous, L.; Belkacem, Y.; et al. Impact of Porous SiC-Doped PVA Based LDS Layer on Electrical Parameters of Si Solar Cells. *Opt. Mater.* **2018**, *80*, 225–232. [CrossRef]
- 193. Zhang, Q.; Zheng, R.; Ding, J.; Wei, W. Excellent Luminous Efficiency and High Thermal Stability of Glass-in-LuAG Ceramic for Laser-Diode-Pumped Green-Emitting Phosphor. *Opt. Lett.* 2018, 43, 3566–3569. [CrossRef]
- Hardy, M.T.; Feezell, D.F.; DenBaars, S.P.; Nakamura, S. Group III-Nitride Lasers: A Materials Perspective. *Mater. Today* 2011, 14, 408–415. [CrossRef]
- 195. Nakamura, S. Materials Issues for InGaN-Based Lasers. J. Electron. Mater. 1998, 27, 160–165. [CrossRef]
- 196. Sanner, N.; Huot, N.; Audouard, E.; Larat, C.; Huignard, J.-P. Direct Ultrafast Laser Micro-Structuring of Materials Using Programmable Beam Shaping. *Opt. Lasers Eng.* **2007**, *45*, 737–741. [CrossRef]
- 197. Ovsianikov, A.; Malinauskas, M.; Schlie, S.; Chichkov, B.; Gittard, S.; Narayan, R.; Löbler, M.; Sternberg, K.; Schmitz, K.-P.; Haverich, A. Three-Dimensional Laser Micro- and Nano-Structuring of Acrylated Poly(Ethylene Glycol) Materials and Evaluation of Their Cytoxicity for Tissue Engineering Applications. *Acta Biomater.* 2011, 7, 967–974. [CrossRef] [PubMed]

- Faraz, A.; Heymann, T.; Biermann, D. Experimental Investigations on Drilling GFRP Epoxy Composite Laminates Using Specialized and Conventional Uncoated Cemented Carbide Drills. *Mater. Manuf. Process.* 2011, 26, 609–617. [CrossRef]
- 199. Xu, J.; Zhou, L.; Chen, M.; Ren, F. Experimental Study on Mechanical Drilling of Carbon/Epoxy Composite-Ti6Al4V Stacks. *Mater. Manuf. Process.* **2019**, *34*, 715–725. [CrossRef]
- Li, C.; Dong, H.; Zhang, W. Low-Temperature Plasma Treatment of Carbon Fibre/Epoxy Resin Composite. Surf. Eng. 2018, 34, 870–876. [CrossRef]
- Garg, M.; Sharma, S.; Mehta, R. Processing of Functionalized and Pristine Carbon Nanotube Epoxy Composites with Silane-Treated Glass Fiber. *Mater. Manuf. Process.* 2016, 31, 2044–2056. [CrossRef]
- 202. Poorzeinolabedin, M.; Golzar, M. Improving the Woven Glass/Epoxy Composite for Automobile Exterior Body Cover. *Mater. Manuf. Process.* 2011, *26*, 562–566. [CrossRef]
- 203. Kordass, T.; Bachy, B.; Weisser, M.; Franke, J. Laser-Assisted Selective Activation of Injection Molded Chip Packaging Devices with Thermoset Substrate Materials for Intelligent Connectivity Systems in Automobiles. *Procedia CIRP* 2017, 63, 101–106. [CrossRef]
- 204. Nishida, H.; Carvelli, V.; Fujii, T.; Okubo, K. Thermoplastic vs. Thermoset Epoxy Carbon Textile Composites. *IOP Conf. Ser. Mater. Sci. Eng.* 2018, 406, 012043. [CrossRef]
- 205. Jony, B.; Thapa, M.; Mulani, S.B.; Roy, S. Repeatable Self-Healing of Thermosetting Fiber Reinforced Polymer Composites with Thermoplastic Healant. *Smart Mater. Struct.* **2019**, *28*, 025037. [CrossRef]
- 206. Sobkowicz-Kline, M.; Budhlall, B.M.; Mead, J.L. Synthetic Resins and Plastics. In *Handbook of Industrial Chemistry and Biotechnology*; Kent, J.A., Bommaraju, T.V., Barnicki, S.D., Eds.; Springer International Publishing: Cham, Switzerland, 2017; pp. 1397–1462. [CrossRef]
- 207. Xia, Y. Hole Accumulation Effect on Laser-Assisted Field Evaporation of Insulators. *Ultramicroscopy* **2019**, 196, 121–128. [CrossRef] [PubMed]
- 208. Baranauskaite, V.E.; Novomlinskii, M.O.; Tumkin, I.I.; Khairullina, E.M.; Mereshchenko, A.S.; Balova, I.A.; Panov, M.S.; Kochemirovsky, V.A. In Situ Laser-Induced Synthesis of Gas Sensing Microcomposites Based on Molybdenum and Its Oxides. *Compos. Part B Eng.* 2019, 157, 322–330. [CrossRef]
- 209. Xu, J.; Li, X.; Zhong, Y.; Qi, J.; Wang, Z.; Chai, Z.; Li, W.; Jing, C.; Cheng, Y. Glass-Channel Molding Assisted 3D Printing of Metallic Microstructures Enabled by Femtosecond Laser Internal Processing and Microfluidic Electroless Plating. *Adv. Mater. Technol.* **2018**, *3*, 1800372. [CrossRef]
- Ratautas, K.; Jagminienė, A.; Stankevičienė, I.; Norkus, E.; Račiukaitis, G. Laser Assisted Fabrication of Copper Traces on Dielectrics by Electroless Plating. *Procedia CIRP* 2018, 74, 367–370. [CrossRef]
- 211. Ninz, P.; Kern, F.; Ermantraut, E.; Müller, H.; Eberhardt, W.; Zimmermann, A.; Gadow, R. Doping of Alumina Substrates for Laser Induced Selective Metallization. *Procedia CIRP* **2018**, *68*, 772–777. [CrossRef]
- 212. Waugh, D.; Lawrence, J. CO<sub>2</sub> Laser Surface Patterning of Nylon 6, 6 and Subsequent Effects on Wettability Characteristics and Apatite Response. *Surf. Eng.* **2011**, *27*, 724–728. [CrossRef]
- 213. Bellmann, M.; Gerhard, C.; Haese, C.; Wieneke, S.; Viöl, W. DBD Plasma Improved Spot Repair of Automotive Polymer Surfaces. *Surf. Eng.* **2012**, *28*, 754–758. [CrossRef]
- 214. Ahmed, N.; Darwish, S.; Alahmari, A.M. Laser Ablation and Laser-Hybrid Ablation Processes: A Review. *Mater. Manuf. Process.* 2016, *31*, 1121–1142. [CrossRef]
- 215. Sathiyamurthy, R.; Duraiselvam, M. Selective Laser Ablation of CFRP Composite to Enhance Adhesion Bonding. *Mater. Manuf. Process.* **2019**, *34*, 1296–1305. [CrossRef]
- 216. Mittal, K.L.; Bahners, T. Laser Surface Modification and Adhesion; John Wiley & Sons: Hoboken, NJ, USA, 2014.
- 217. Konieczkowska, J.; Schab-Balcerzak, E.; Libera, M.; Mihaila, I.; Sava, I. Surface Relief Gratings in Azopolyimides Induced by Pulsed Laser Irradiation. *Eur. Polym. J.* **2019**, *110*, 85–89. [CrossRef]
- Ponnamma, D.; Sivakumar, V.; Popelka, A.; Hussein, Y.H.A.; Al Ali Al-Maadeed, M. Laser Induced Periodic Surface Structures on Nano Metal Oxide Filled Polyvinylidene Fluoride Nanocomposites. *Optik* 2019, 176, 372–383. [CrossRef]
- Arai, S.; Tsunoda, S.; Yamaguchi, A.; Ougizawa, T. Effect of Anisotropy in the Build Direction and Laser-Scanning Conditions on Characterization of Short-Glass-Fiber-Reinforced PBT for Laser Sintering. *Opt. Laser Technol.* 2019, 113, 345–356. [CrossRef]
- 220. Curylofo-Zotti, F.A.; Scheffel, D.L.S.; Macedo, A.P.; de Souza-Gabriel, A.E.; Hebling, J.; Corona, S.A.M. Effect of Er:YAG Laser Irradiation and Chitosan Biomodification on the Stability of Resin/Demineralized Bovine Dentin Bond. *J. Mech. Behav. Biomed. Mater.* 2019, *91*, 220–228. [CrossRef]

- 221. Liu, H.; Tang, Y.; Xie, Y.; Lu, L.; Wan, Z.; Tang, W.; Yang, L.; Yang, D. Effect of Pulsed Nd:YAG Laser Processing Parameters on Surface Properties of Polyimide Films. *Surf. Coat. Technol.* **2019**, *361*, 102–111. [CrossRef]
- 222. Sionkowska, A.; Wisniewski, M.; Kaczmarek, H.; Skopinska, J.; Chevallier, P.; Mantovani, D.; Lazare, S.; Tokarev, V. The Influence of UV Irradiation on Surface Composition of Collagen/PVP Blended Films. *Appl. Surf. Sci.* 2006, 253, 1970–1977. [CrossRef]
- 223. Serafetinides, A.A.; Makropoulou, M.I.; Skordoulis, C.D.; Kar, A.K. Ultra-Short Pulsed Laser Ablation of Polymers. *Appl. Surf. Sci.* 2001, *180*, 42–56. [CrossRef]
- 224. Rebollar, E.; Hernández, M.; Sanz, M.; Pérez, S.; Ezquerra, T.A.; Castillejo, M. Laser-Induced Surface Structures on Gold-Coated Polymers: Influence of Morphology on Surface-Enhanced Raman Scattering Enhancement. J. Appl. Polym. Sci. 2015, 132. [CrossRef]
- 225. Michaljaničová, I.; Slepička, P.; Heitz, J.; Barb, R.A.; Sajdl, P.; Švorčík, V. Comparison of KrF and ArF Excimer Laser Treatment of Biopolymer Surface. *Appl. Surf. Sci.* **2015**, *339*, 144–150. [CrossRef]
- 226. Li, T.; Hou, Z.; Fu, Y.; Yu, J.; Gu, W.; Wang, Z. Correction of Self-Absorption Effect in Calibration-Free Laser-Induced Breakdown Spectroscopy (CF-LIBS) with Blackbody Radiation Reference. *Anal. Chim. Acta* 2019, 1058, 39–47. [CrossRef]
- 227. Shukla, P.; Lawrence, J. Distribution of Temperature during Fibre Laser Radiation and Effects Thereon Phase Transformation of ZrO<sub>2</sub> Engineering Ceramic. *Surf. Eng.* **2011**, *27*, 742–748. [CrossRef]
- 228. Yilbas, B. Laser Ablation of Phosphor Bronze for Superhydrophobic Surface. *Surf. Eng.* **2016**, *32*, 885–892. [CrossRef]
- 229. Yilbas, B.; Arif, A.; Karatas, C. Laser Treatment of Silicon at Nitrogen Ambient: Thermal Stress Analysis. *Surf. Eng.* 2011, 27, 436–444. [CrossRef]
- 230. Tyurin, A.; Nagavkin, S.; Malikov, A.; Orishich, A. Microstructure of WC–Co Hard Alloy Surface after Laser Treatment. *Surf. Eng.* **2015**, *31*, 74–77. [CrossRef]
- 231. Hashim, M.; Sarath Raghavendra Babu, K.; Duraiselvam, M.; Natu, H.; Muthupandi, V.; Sivaprasad, K.; Rajith, K. Tribological Studies on Laser Surface Melted Hastelloy C-276. *Surf. Eng.* 2013, 29, 531–535. [CrossRef]
- 232. Peng, W.Y.; Cassady, S.J.; Strand, C.L.; Goldenstein, C.S.; Spearrin, R.M.; Brophy, C.M.; Jeffries, J.B.; Hanson, R.K. Single-Ended Mid-Infrared Laser-Absorption Sensor for Time-Resolved Measurements of Water Concentration and Temperature within the Annulus of a Rotating Detonation Engine. *Proc. Combust. Inst.* 2019, *37*, 1435–1443. [CrossRef]
- 233. Jansen, E.D.; van Leeuwen, T.G.; Motamedi, M.; Borst, C.; Welch, A.J. Temperature Dependence of the Absorption Coefficient of Water for Midinfrared Laser Radiation. *Lasers Surg. Med.* 1994, 14, 258–268. [CrossRef]
- Meier, M.; Romano, V.; Feurer, T. Material Processing with Pulsed Radially and Azimuthally Polarized Laser Radiation. *Appl. Phys. A* 2007, *86*, 329–334. [CrossRef]
- 235. Frenz, M.; Pratisto, H.; Konz, F.; Jansen, E.D.; Welch, A.J.; Weber, H.P. Comparison of the Effects of Absorption Coefficient and Pulse Duration of 2.12-/Spl Mu/m and 2.79-/Spl Mu/m Radiation on Laser Ablation of Tissue. *IEEE J. Quantum Electron.* **1996**, *32*, 2025–2036. [CrossRef]
- 236. Angelsky, O.V.; Bekshaev, A.Y.; Maksimyak, P.P.; Maksimyak, A.P.; Hanson, S.G.; Kontush, S.M. Controllable Generation and Manipulation of Micro-Bubbles in Water with Absorptive Colloid Particles by CW Laser Radiation. *Opt. Express* **2017**, *25*, 5232–5243. [CrossRef] [PubMed]
- 237. Torrisi, L.; Calcagno, L.; Giulietti, D.; Cutroneo, M.; Zimbone, M.; Skala, J. Laser Irradiations of Advanced Targets Promoting Absorption Resonance for Ion Acceleration in TNSA Regime. *Nucl. Instrum. Methods Phys. Res. Sect. B Beam Interact. Mater. At.* **2015**, 355, 221–226. [CrossRef]
- 238. Sun, S.; Chen, J.; Jiang, K.; Tang, Z.; Wang, Y.; Li, Z.; Liu, C.; Wu, A.; Lin, H. Ce6-Modified Carbon Dots for Multimodal-Imaging-Guided and Single-NIR-Laser-Triggered Photothermal/Photodynamic Synergistic Cancer Therapy by Reduced Irradiation Power. ACS Appl. Mater. Interfaces 2019, 11, 5791–5803. [CrossRef] [PubMed]
- 239. Du, Q.; Chen, T.; Liu, J.; Zeng, X. Surface Microstructure and Chemistry of Polyimide by Single Pulse Ablation of Picosecond Laser. *Appl. Surf. Sci.* 2018, 434, 588–595. [CrossRef]
- 240. Neelgund, G.M.; Oki, A. Photothermal Effect of Ag Nanoparticles Deposited over Poly(Amidoamine) Grafted Carbon Nanotubes. *J. Photochem. Photobiol. Chem.* **2018**, *364*, 309–315. [CrossRef]

- Hatanaka, K.; Kawao, M.; Tsuboi, Y.; Fukumura, H.; Masuhara, H. Switching from Photochemical to Photothermal Mechanism in Laser Ablation of Benzene Solutions. *J. Appl. Phys.* **1997**, *82*, 5799–5806. [CrossRef]
- 242. Melancon, M.P.; Lu, W.; Yang, Z.; Zhang, R.; Cheng, Z.; Elliot, A.M.; Stafford, J.; Olson, T.; Zhang, J.Z.; Li, C. In Vitro and in Vivo Targeting of Hollow Gold Nanoshells Directed at Epidermal Growth Factor Receptor for Photothermal Ablation Therapy. *Mol. Cancer Ther.* 2008, *7*, 1730–1739. [CrossRef]
- 243. Yingling, Y.G.; Garrison, B.J. Photochemical Induced Effects in Material Ejection in Laser Ablation. *Chem. Phys. Lett.* **2002**, *364*, 237–243. [CrossRef]
- 244. Sato, H.; Nishio, S. Polymer Laser Photochemistry, Ablation, Reconstruction, and Polymerization. *J. Photochem. Photobiol. C Photochem. Rev.* **2001**, *2*, 139–152. [CrossRef]
- 245. Abdelaziem, A.; El-Khatib, K.M.; Hafez, M.A.; Badr, Y. Effect of Annealing on La<sub>0.8</sub>Sr<sub>0.2</sub>MnO<sub>3</sub> Thin Films Prepared by Pulsed Laser Deposition. *Spectrochim. Acta A. Mol. Biomol. Spectrosc.* **2019**, 211, 100–107. [CrossRef]
- 246. Davari, S.A.; Gottfried, J.L.; Liu, C.; Ribeiro, E.L.; Duscher, G.; Mukherjee, D. Graphitic Coated Al Nanoparticles Manufactured as Superior Energetic Materials via Laser Ablation Synthesis in Organic Solvents. *Appl. Surf. Sci.* 2019, 473, 156–163. [CrossRef]
- 247. Van Geenen, F.A.M.G.; Franssen, M.C.R.; Miikkulainen, V.; Ritala, M.; Zuilhof, H.; Kostiainen, R.; Nielen, M.W.F. TiO<sub>2</sub> Photocatalyzed Oxidation of Drugs Studied by Laser Ablation Electrospray Ionization Mass Spectrometry. J. Am. Soc. Mass Spectrom. 2019, 30, 639–646. [CrossRef] [PubMed]
- 248. Censabella, M.; Torrisi, V.; Boninelli, S.; Bongiorno, C.; Grimaldi, M.G.; Ruffino, F. Laser Ablation Synthesis of Mono- and Bimetallic Pt and Pd Nanoparticles and Fabrication of Pt-Pd/Graphene Nanocomposites. *Appl. Surf. Sci.* **2019**, 475, 494–503. [CrossRef]
- Sinha, S.K. Effect of Temperature on Structural, Optical and Electrical Properties of Pulsed-Laser Deposited W-Doped V<sub>2</sub>O<sub>5</sub> Thin Films. *Superlattices Microstruct.* 2019, 125, 88–94. [CrossRef]
- 250. Mikhailov, V.A.; Liko, I.; Mize, T.H.; Bush, M.F.; Benesch, J.L.P.; Robinson, C.V. Infrared Laser Activation of Soluble and Membrane Protein Assemblies in the Gas Phase. *Anal. Chem.* 2016, *88*, 7060–7067. [CrossRef] [PubMed]
- 251. Zhang, T.; Simonds, B.; Nomoto, K.; Puthen Veettil, B.; Lin, Z.; Perez Wurfl, I.; Conibeer, G. Pulsed KrF Excimer Laser Dopant Activation in Nanocrystal Silicon in a Silicon Dioxide Matrix. *Appl. Phys. Lett.* 2016, 108, 083103. [CrossRef]
- 252. Ahluwalia, J.; Avram, M.M.; Ortiz, A.E. Outcomes of Long-Pulsed 1064 Nm Nd:YAG Laser Treatment of Basal Cell Carcinoma: A Retrospective Review. *Lasers Surg. Med.* **2019**, *51*, 34–39. [CrossRef]
- 253. Abedi, H.R.; Hoseinpour Gollo, M. An Experimental Study of the Effects of Surface Roughness and Coating of Cr<sub>2</sub>O<sub>3</sub> Layer on the Laser-Forming Process. *Opt. Laser Technol.* **2019**, *109*, 336–347. [CrossRef]
- 254. Hopkins, P.E.; Serrano, J.R.; Phinney, L.M.; Kearney, S.P.; Grasser, T.W.; Harris, C.T. Criteria for Cross-Plane Dominated Thermal Transport in Multilayer Thin Film Systems During Modulated Laser Heating. *J. Heat Transf.* 2010, 132, 081302–081310. [CrossRef]
- 255. Eltas, S.D.; Gursel, M.; Eltas, A.; Alptekin, N.O.; Ataoglu, T. Evaluation of Long-Term Effects of Diode Laser Application in Periodontal Treatment of Poorly Controlled Type 2 Diabetic Patients with Chronic Periodontitis. *Int. J. Dent. Hyg.* **2019**, *17*, 292–299. [CrossRef]
- 256. Bernhardt, A.F.; Rasmussen, P. Design Criteria and Operating Characteristics of a Single-Mode Pulsed Dye Laser. *Appl. Phys. B* **1981**, *26*, 141–146. [CrossRef]
- 257. Brenier, A.; Boulon, G. New Criteria to Choose the Best Yb3+-Doped Laser Crystals. *EPL Europhys. Lett.* **2001**, *55*, 647. [CrossRef]
- 258. Gaddam, S.; Mathur, S.C.; Singh, M.; Arora, J.; Wani, S.B.; Gupta, N.; Overhiser, A.; Rastogi, A.; Singh, V.; Desai, N. Erratum: Novel Probe-Based Confocal Laser Endomicroscopy Criteria and Interobserver Agreement for the Detection of Dysplasia in Barrett's Esophagus. *Am. J. Gastroenterol.* **2013**, *108*, 460. [CrossRef]
- Malafeyev, O.A.; Neverova, E.G.; Nemnyugin, S.A.; Alferov, G.V. Multi-Criteria Model of Laser Radiation Control. In Proceedings of the 2014 2nd 2014 2nd International Conference on Emission Electronics (ICEE), St. Petersburg, Russia, 30 June–4 July 2014; pp. 1–5. [CrossRef]
- Kudtarkar, K.; Iglesias, P.; Smith, T.W.; Schertzer, M.J. Effect of Metallization on the Electromechanical Properties of Microfluidically Synthesized Hydrogel Beads. J. Fluids Eng. 2018, 141, 031303–031306. [CrossRef]

- Valov, I.; Waser, R.; Jameson, J.R.; Kozicki, M.N. Electrochemical Metallization Memories—Fundamentals, Applications, Prospects. *Nanotechnology* 2011, 22, 254003. [CrossRef]
- 262. Le, X.T.; Poirier, J.-S.; Michel, S. Completely Aqueous Route for Metallization of Structural Polymeric Materials in Micro-Electro-Mechanical Systems. *Appl. Surf. Sci.* **2019**, *478*, 717–724. [CrossRef]
- 263. Zhao, J.; An, T.; Fang, C.; Bie, X.; Qin, F.; Chen, P.; Dai, Y. A Study on the Effect of Microstructure Evolution of the Aluminum Metallization Layer on Its Electrical Performance during Power Cycling. *IEEE Trans. Power Electron.* 2019. [CrossRef]
- Murarka, S.P.; Hymes, S.W. Copper Metallization for ULSL and Beyond. *Crit. Rev. Solid State Mater. Sci.* 1995, 20, 87–124. [CrossRef]
- 265. Shacham-Diamand, Y.; Lopatin, S. Integrated Electroless Metallization for ULSI. *Electrochim. Acta* **1999**, 44, 3639–3649. [CrossRef]
- 266. Rekha, S.; Jeeva, P.; Karthikeyan, S.; Srinivasan, K.; Ramkumar, D.; Arivazhagan, N.; Narayanan, S. Metallization of Iron Powders by Autocatalytic Copper Plating Process. *Mater. Manuf. Process.* 2013, 28, 905–909. [CrossRef]
- Nelyub, V.; Tarasov, V. Ion-Beam Processing of Metallized Carbon Fiber-Reinforced Plastics. *Mater. Manuf. Process.* 2020, 35. [CrossRef]
- 268. Pujari, M.; Agarwal, A.; Uppaluri, R.; Verma, A. Combinatorial Electroless Plating Characteristics for Dense Pd–PSS Composite Membrane Fabrication. *Mater. Manuf. Process.* **2016**, *31*, 6–11. [CrossRef]
- Pujari, M.; Agarwal, A.; Uppaluri, R.; Verma, A. Effect of Pd Concentration on Electroless Dense Pd-PSS Membrane Fabrication. *Surf. Eng.* 2015, *31*, 209–213. [CrossRef]
- 270. Pujari, M.; Agarwal, A.; Uppaluri, R.; Verma, A. Efficacy of Novel Electroless Plating Process for Dense Pd/Cr<sub>2</sub>O<sub>3</sub>/PSS Membrane Fabrication. *Mater. Manuf. Process.* **2016**, *31*, 1–5. [CrossRef]
- 271. Agarwal, A.; Pujari, M.; Uppaluri, R.; Verma, A. Efficacy of Palladium Solution Concentration on Electroless Fabrication of Dense Metal Ceramic Composite Membranes Coupled with Surfactant and Sonication. *Mater. Manuf. Process.* 2016, *31*, 18–23. [CrossRef]
- Zhang, Y.; Zong, B.; Jin, J.; Cao, X. Electroless Copper Plating on Particulate Reinforcements and Effects on Mechanical Properties of SiCp/Fe Composite. *Surf. Eng.* 2015, *31*, 232–239. [CrossRef]
- 273. Lu, Z.-L.; Wang, Z.-C.; Luo, L.-M.; Huang, X.-M.; Tan, X.-Y.; Chen, J.-B.; Cheng, J.; Wu, Y.-C. Electroless Plating of Copper on Al<sub>2</sub>O<sub>3</sub> and Its Heat Treatment Behaviour. *Surf. Eng.* **2015**, *31*, 240–244. [CrossRef]
- 274. Vaškelis, A.; Norkus, H.J.; Rozovskis, G.; Vinkevičius, H.J. New Methods of Electroless Plating and Direct Electroplating of Plastics. *Trans. IMF* **1997**, *75*, 1–3. [CrossRef]
- 275. Dos Santos, F.S.; Martins, L.F.O.; D'Ajello, P.C.T.; Pasa, A.A.; Hasenack, C.M. Electroless and Electro-Plating of Cu on Si. *Microelectron. Eng.* **1997**, *33*, 59–64. [CrossRef]
- 276. Shacham-Diamand, Y.; Osaka, T.; Okinaka, Y.; Sugiyama, A.; Dubin, V. 30years of Electroless Plating for Semiconductor and Polymer Micro-Systems. *Microelectron. Eng.* 2015, 132, 35–45. [CrossRef]
- 277. Genova, V.; Paglia, L.; Marra, F.; Bartuli, C.; Pulci, G. Pure Thick Nickel Coating Obtained by Electroless Plating: Surface Characterization and Wetting Properties. *Surf. Coat. Technol.* **2019**, *357*, 595–603. [CrossRef]
- 278. Nagireddi, S.; Golder, A.K.; Uppaluri, R. Role of EDTA on the Pd(II) Adsorption Characteristics of Chitosan Cross-Linked 3-Amino-1,2,4-Triazole-5-Thiol Derivative from Synthetic Electroless Plating Solutions. *Int. J. Biol. Macromol.* **2019**, 127, 320–329. [CrossRef]
- 279. Alique, D.; Martinez-Diaz, D.; Sanz, R.; Calles, J.A. Review of Supported Pd-Based Membranes Preparation by Electroless Plating for Ultra-Pure Hydrogen Production. *Membranes* **2018**, *8*, 5. [CrossRef] [PubMed]
- Zoppas, F.M.; Marchesini, F.A.; Devard, A.; Bernardes, A.M.; Miró, E.E. Controlled Deposition of Pd and In on Carbon Fibers by Sequential Electroless Plating for the Catalytic Reduction of Nitrate in Water. *Catal. Commun.* 2016, 78, 59–63. [CrossRef]
- Jones, C.G.; Mills, B.E.; Nishimoto, R.K.; Robinson, D.B. Electroless Deposition of Palladium on Macroscopic 3D-Printed Polymers with Dense Microlattice Architectures for Development of Multifunctional Composite Materials. J. Electrochem. Soc. 2017, 164, D867–D874. [CrossRef]
- Tang, J.; Ou, Q.; Zhou, H.; Qi, L.; Man, S. Seed-Mediated Electroless Deposition of Gold Nanoparticles for Highly Uniform and Efficient SERS Enhancement. *Nanomaterials* 2019, 9, 185. [CrossRef]
- 283. Atli, A.; Simon, S.; Aires, F.J.C.S.; Cardenas, L.; Ehret, E.; Lourdin, P. A New Strategy to Activate Liquid Crystal Polymer Samples for Electroless Copper Deposition. *J. Appl. Polym. Sci.* **2017**, *134*. [CrossRef]

- 284. Muench, F.; Solomonov, A.; Bendikov, T.; Molina-Luna, L.; Rubinstein, I.; Vaskevich, A. Empowering Electroless Plating to Produce Silver Nanoparticle Films for DNA Biosensing Using Localized Surface Plasmon Resonance Spectroscopy. ACS Appl. Bio Mater. 2019, 2, 856–864. [CrossRef]
- 285. Aixiang, Z.; Weihao, X.; Jian, X. Electroless Ni-P Coating of Cenospheres Using Silver Nitrate Activator. *Surf. Coat. Technol.* **2005**, 197, 142–147. [CrossRef]
- Horkans, J.; Kim, J.; McGrath, C.; Romankiw, L.T. A TEM Study of the Effect of Accelerators on Pd-Sn Colloidal Catalysts and on the Initiation of Electroless Cu Deposition on Epoxy. J. Electrochem. Soc. 1987, 134, 300–304. [CrossRef]
- 287. Ghosh, S. Electroless Copper Deposition: A Critical Review. Thin Solid Films 2019, 669, 641–658. [CrossRef]
- 288. Xiang, S.; Li, W.; Qian, Z.; Zhu, L.; Liu, H. The Effect of 2-Mercaptobenzothiazole on Laser-Assisted Electroless Copper Plating. *RSC Adv.* **2016**, *6*, 38647–38652. [CrossRef]
- 289. Shi, S.; Wang, L.; Wang, A.; Huang, R.; Ding, L.; Su, R.; Qi, W.; He, Z. Bioinspired Fabrication of Optical Fiber SPR Sensors for Immunoassays Using Polydopamine-Accelerated Electroless Plating. *J. Mater. Chem. C* 2016, 4, 7554–7562. [CrossRef]
- 290. Qi, K.; Xie, Y.; Wang, R.; Liu, S.; Zhao, Z. Electroless Plating Ni-P Cocatalyst Decorated g-C3N4 with Enhanced Photocatalytic Water Splitting for H2 Generation. *Appl. Surf. Sci.* **2019**, *466*, 847–853. [CrossRef]
- 291. Simpson, D.E.; Johnson, C.A.; Roy, D. Pulsed Galvanostatic Electrodeposition of Copper on Cobalt Using a PH-Neutral Plating Bath and Electroless Seeds. *J. Electrochem. Soc.* **2019**, *166*, D3142–D3154. [CrossRef]
- Cheng, J.; Gan, X.; Chen, S.; Lai, Y.; Xiong, H.; Zhou, K. Properties and Microstructure of Copper/Nickel-Iron-Coated Graphite Composites Prepared by Electroless Plating and Spark Plasma Sintering. *Powder Technol.* 2019, 343, 705–713. [CrossRef]
- 293. Kobayashi, Y.; Salgueiriño-Maceira, V.; Liz-Marzán, L.M. Deposition of Silver Nanoparticles on Silica Spheres by Pretreatment Steps in Electroless Plating. *Chem. Mater.* **2001**, *13*, 1630–1633. [CrossRef]
- 294. Hanna, F.; Hamid, Z.A.; Aal, A.A. Controlling Factors Affecting the Stability and Rate of Electroless Copper Plating. *Mater. Lett.* 2004, *58*, 104–109. [CrossRef]
- 295. Kobayashi, Y.; Tadaki, Y.; Nagao, D.; Konno, M. Deposition of Gold Nanoparticles on Silica Spheres by Electroless Metal Plating Technique. *J. Colloid Interface Sci.* 2005, 283, 601–604. [CrossRef]
- 296. Stanley, M.L.; Henne, P.; Yang, B.W.; De Brigard, F. Resistance to Position Change, Motivated Reasoning, and Polarization. *Polit. Behav.* 2019. [CrossRef]
- 297. Choi, S.M.; An, H.; Yoon, K.J.; Kim, B.-K.; Lee, H.-W.; Son, J.-W.; Kim, H.; Shin, D.; Ji, H.-I.; Lee, J.-H. Electrochemical Analysis of High-Performance Protonic Ceramic Fuel Cells Based on a Columnar-Structured Thin Electrolyte. *Appl. Energy* **2019**, *233–234*, 29–36. [CrossRef]
- 298. Subardi, A.; Liao, K.-Y.; Fu, Y.-P. Oxygen Transport, Thermal and Electrochemical Properties of NdBa0.5Sr0.5Co2O5+δ Cathode for SOFCs. *J. Eur. Ceram. Soc.* **2019**, *39*, 30–40. [CrossRef]
- 299. Abdel-Gawad, S.A.; Shoeib, M.A. Corrosion Studies and Microstructure of Mg–Zn–Ca Alloys for Biomedical Applications. *Surf. Interfaces* **2019**, *14*, 108–116. [CrossRef]
- 300. Brown, D.; Darr, D.; Morse, J.; Laskowski, B.; Betti, R. Experimental Validation of a Micro-Sized Polarization Resistance Corrosion Sensor for Structural Health Management Applications. In Proceedings of the 6th European Workshop onStructural Health Monitoring, Dresden, Germany, 3–6 July 2012.
- 301. Montero, M.A.; Marozzi, C.A.; Gennero de Chialvo, M.R.; Chialvo, A.C. A Simple Method to Establish the Relationship between the Equilibrium Polarization Resistance and the Elementary Kinetic Parameters of an Electrocatalysed Reaction. *Electrocatalysis* **2018**, *9*, 409–415. [CrossRef]
- 302. Huan, D.; Wang, W.; Xie, Y.; Shi, N.; Wan, Y.; Xia, C.; Peng, R.; Lu, Y. Investigation of Real Polarization Resistance for Electrode Performance in Proton-Conducting Electrolysis Cells. J. Mater. Chem. A 2018, 6, 18508–18517. [CrossRef]
- 303. Samson, G.; Deby, F.; Garciaz, J.-L.; Perrin, J.-L. Alternative Methodology for Linear Polarization Resistance Assessment of Reinforced Concrete Structure. *MATEC Web Conf.* 2018, 199, 06009. [CrossRef]
- 304. Setevich, C.; Prado, F.; Caneiro, A. Study of the Electrode Polarization Resistance of Cobaltites with High Ba Content as Cathode for IT-SOFC. *J. Electrochem. Soc.* **2017**, *164*, F759–F767. [CrossRef]
- Wang, J.; Zhou, J.; Wang, T.; Chen, G.; Wu, K.; Cheng, Y. Decreasing the Polarization Resistance of LaSrCoO<sub>4</sub> Cathode by Fe Substitution for Ba(Zr<sub>0.1</sub>Ce<sub>0.7</sub>Y<sub>0.2</sub>)O<sub>3</sub> Based Protonic Ceramic Fuel Cells. *J. Alloys Compd.* 2016, 689, 581–586. [CrossRef]

- 306. Iatalese, M.; Coluccio, M.L.; Onesto, V.; Amato, F.; Fabrizio, E.D.; Gentile, F. Relating the Rate of Growth of Metal Nanoparticles to Cluster Size Distribution in Electroless Deposition. *Nanoscale Adv.* 2019, 1, 228–240. [CrossRef]
- Li, J.-M.; Hu, C.-C.; Wu, T.-H.; Hsu, Y.-J. Electroless Deposition of RuO 2 -Based Nanoparticles for Energy Conversion Applications. *RSC Adv.* 2019, *9*, 4239–4245. [CrossRef]
- 308. Abdel-Gawad, S.A.; Sadik, M.A.; Shoeib, M.A. Preparation and Properties of a Novel Nano Ni-B-Sn by Electroless Deposition on 7075-T6 Aluminum Alloy for Aerospace Application. J. Alloys Compd. 2019, 785, 1284–1292. [CrossRef]
- Xu, M.L.; Kang, S.M.; Chen, T.T. Coating Properties of Electroless Ni-P Plating on Magnesium Alloy with Cerium Chloride. *Metalurgija* 2019, 58, 106–108.
- 310. Porter, L.A.; Choi, H.C.; Ribbe, A.E.; Buriak, J.M. Controlled Electroless Deposition of Noble Metal Nanoparticle Films on Germanium Surfaces. *Nano Lett.* **2002**, *2*, 1067–1071. [CrossRef]
- 311. Eom, K.; Cho, K.; Kwon, H. Effects of Electroless Deposition Conditions on Microstructures of Cobalt–Phosphorous Catalysts and Their Hydrogen Generation Properties in Alkaline Sodium Borohydride Solution. J. Power Sources 2008, 180, 484–490. [CrossRef]
- 312. Touir, R.; Larhzil, H.; EbnTouhami, M.; Cherkaoui, M.; Chassaing, E. Electroless Deposition of Copper in Acidic Solutions Using Hypophosphite Reducing Agent. *J. Appl. Electrochem.* **2006**, *36*, 69–75. [CrossRef]
- Zhang, W.X.; Huang, N.; He, J.G.; Jiang, Z.H.; Jiang, Q.; Lian, J.S. Electroless Deposition of Ni–W–P Coating on AZ91D Magnesium Alloy. *Appl. Surf. Sci.* 2007, 253, 5116–5121. [CrossRef]
- 314. Ohno, I. Electrochemistry of Electroless Plating. Mater. Sci. Eng. A 1991, 146, 33–49. [CrossRef]



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