

**Supplementary Materials**  
**of**  
**Polyhydride CeH<sub>9</sub> with an atomic-like hydrogen clathrate**  
**structure**

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### **Determination of stoichiometries of Ce polyhydrides.**

We have determined the stoichiometries and crystal structures of new phases in three aspects in main paper: 1) X-ray diffraction data give the possible phase transitions and symmetry and match the structures of prediction; 2) The volume changes obtained from XRD data indicate the stoichiometry of new phases; 3) Theoretical calculations help us to finally determine the crystal structure and stoichiometry through the calculated enthalpy difference, pressure-volume relation and phase stability. Particularly, the Rietveld refinement combines the calculated and experimental XRD results. We would like to expound detailed distinction of  $\text{CeH}_3$ ,  $\text{CeH}_{3+x}$  and  $\text{CeH}_4$ , and  $\text{CeH}_{9-\delta}$  and  $\text{CeH}_9$ .

#### **The determination of $\text{CeH}_3$ -*Fm-3m***

In Supplementary Fig. 3, we found that the patterns of fcc phase match the known structures of  $\text{CeH}_3$  and  $\text{CeH}_2$ . Both  $\text{CeH}_3$  and  $\text{CeH}_2$  have the same space group *Fm-3m*, but  $\text{CeH}_3$  has one more hydrogen atom than  $\text{CeH}_2$  at interstitial sites  $4b$  (0,0,1/2). They have very similar lattice parameters and compression  $P$ - $V$  curves under pressure. Here we propose fcc phase as  $\text{CeH}_3$  rather than  $\text{CeH}_2$ , because the  $\text{H}_2$  is excessive in the chamber and the enthalpy calculations indicate that  $\text{CeH}_3$  has a lower enthalpy value than  $\text{CeH}_2$  (Supplementary Fig. 5).

#### **The stoichiometries of $\text{CeH}_{3+x}$ .**

From 40 GPa to 72 GPa, Rietveld structural refinement and calculated enthalpy difference (Fig. 1b and Supplementary Fig. 5) prefer the structure of  $\text{CeH}_3$ -*Pm-3n*. Because of the low atomic scattering power of hydrogen, we couldn't determine the content of hydrogen directly through XRD data. In fact, the hydrogen content of cerium hydrides was changing with increasing pressure in the experiment. In order to avoid the confusion and guarantee the accuracy, here we defined the second cubic phase as  $\text{CeH}_{3+x}$  ( $0.7 \leq x \leq 1.1$ ) and the reasons are as follows. As seen in Supplementary Fig. 6, the volume of  $\text{CeH}_{3+x}$  is close to  $\text{CeH}_3$ -*Fm-3m* at 40 GPa, however the volume becomes close to  $\text{CeH}_4$ -*I4/mmm* at 72 GPa. In this pressure range, we can also intuitively see that the compression curve of  $\text{CeH}_{3+x}$  is approaching the mixed line of  $\text{Ce}+2\text{H}_2$  and connects the curves of  $\text{CeH}_3$ -*Fm-3m* and  $\text{CeH}_4$ -*I4/mmm* smoothly, which also indicates a process of hydrogen absorption. At 40 GPa,  $V(\text{CeH}_{3+x}) = 30.54 \text{ \AA}^3/\text{f.u.}$ ;  $V(\text{Ce}) = 17.98 \text{ \AA}^3/\text{f.u.}$  and  $V(\text{H}) = 3.4 \text{ \AA}^3/\text{f.u.}$  at this pressure. Thus the stoichiometry of cerium hydrides  $\text{CeH}_{3+x}$  is  $x=0.7$ . At 72 GPa,  $V(\text{CeH}_{3+x})$  is  $27.21 \text{ \AA}^3/\text{f.u.}$ ;  $V(\text{Ce})$

16.01 Å<sup>3</sup>/f.u. and  $V(\text{H})$  is 2.7 Å<sup>3</sup>/f.u., which gives  $x=1.1$ .

### **The distinguish of CeH<sub>9-δ</sub> and CeH<sub>9</sub>.**

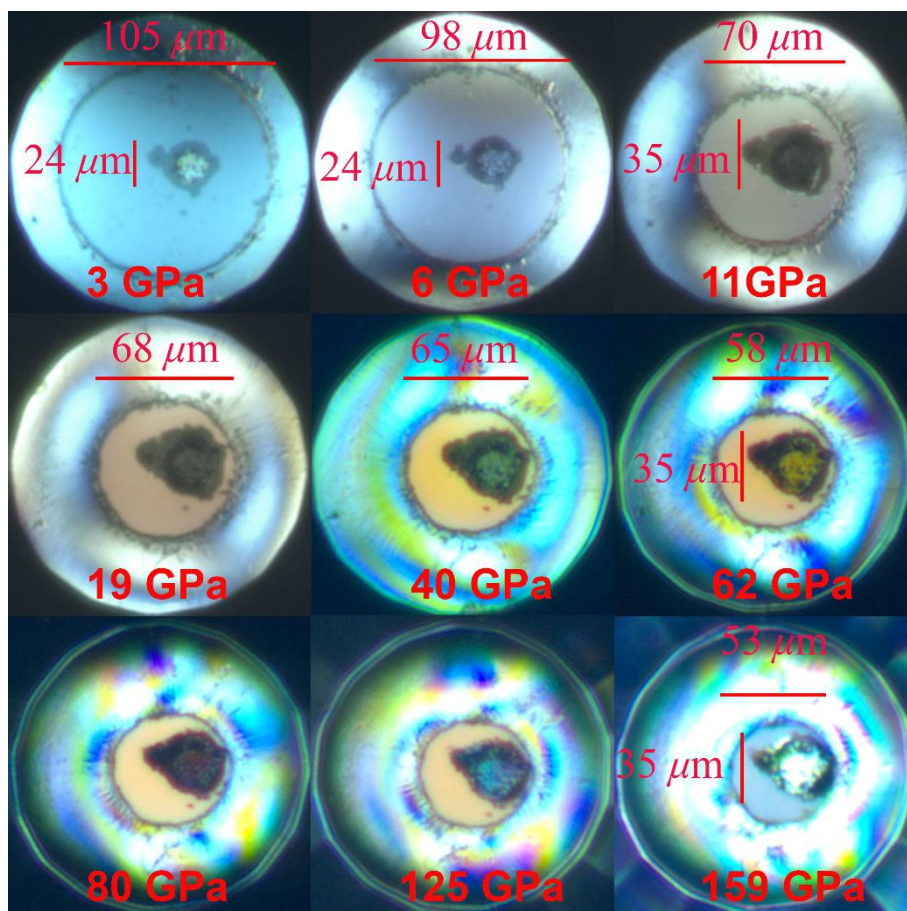
For CeH<sub>9</sub>: We determine the phase of CeH<sub>9</sub>-*P6<sub>3</sub>/mmc* through Rietveld refinement of experimental XRD data (Fig. 1b), good agreement between experimental and calculated *P-V* relation (Fig. 2) and advantage of calculated enthalpy (Supplementary Fig. 5).

For CeH<sub>9-δ</sub>: Our XRD data match the predicted CeH<sub>8</sub>-*P6<sub>3</sub>mc*. We exclude this structure because of the imaginary vibration modes in phonon dispersion curves. CeH<sub>9</sub>-*P6<sub>3</sub>/mmc* also matches XRD pattern profile, but we distinguish them by the *P-V* relationship (Fig. 2). We have determined the phase of CeH<sub>9</sub>-*P6<sub>3</sub>/mmc* above 103 GPa, however, in the pressure range of 84-103 GPa, we observe abnormal movement of peaks from CeH<sub>9-δ</sub> (Supplementary Fig. 13) which indicates the increase of lattice parameters, i.e. the volume expansion of unit cell. Continuously significant expansion of volume is attributed to absorption of hydrogen (Fig. 2 and Supplementary Fig. 13), indicating a gradual hydrogenation process of forming CeH<sub>9</sub>, thus we assign the phase in this pressure range noted as CeH<sub>9-δ</sub>. At 88 GPa, the volumes of Ce and CeH<sub>9</sub> (extrapolation through BM equation) are  $V(\text{Ce}) = 15.37 \text{ Å}^3/\text{f.u.}$  and  $V(\text{CeH}_9) = 33.91 \text{ Å}^3/\text{f.u.}$ , then the estimation of hydrogen expansion is  $V(\text{H}) = 2.06 \text{ Å}^3/\text{atom}$ . Therefore,  $V(\text{CeH}_{9-\delta})$  of 32.17 Å<sup>3</sup>/f.u. gives the value range  $\delta \leq 0.85$ .

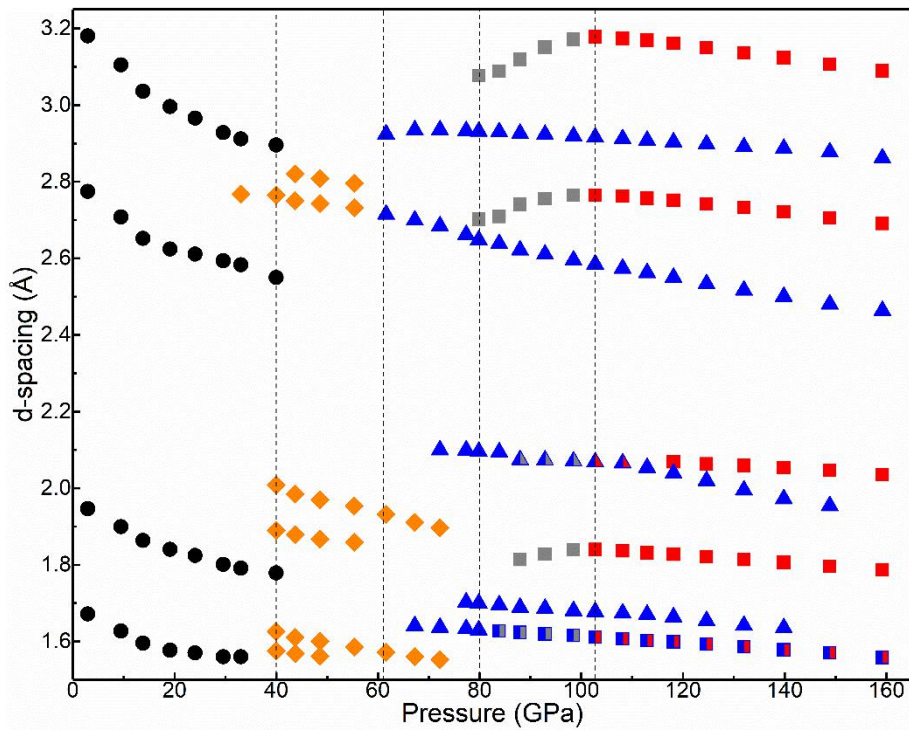
### **Calculation**

Here we have used DFT calculations with GGA+U to obtain the lattice parameters and cell volume at different pressures. As reported in previous literatures, Ce atoms with *f* electrons are an extremely delicate system for DFT calculations with the use of standard approximations. We have carefully checked the calculated data, and found that the used GGA+U reproduced the experimental data well and the volume difference between experimental and calculated data is below 3% (Supplementary Fig. 11). In this regard, the current theoretical GGA+U approximations give the correct EOS, which is in agreement with the experimental one. Besides, the GGA+U calculations have been done on band structure, DOS, ELF and phonon spectrum.

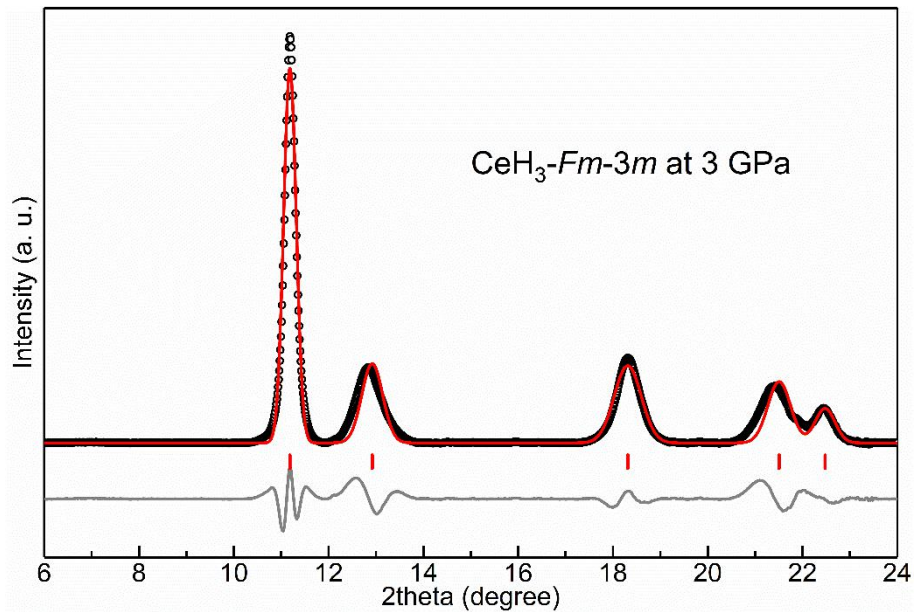
It is reported that the anharmonic effects may affect the vibrational frequencies, thermodynamics and lattice stability of superhydrides. Errea *et al.* have found that the quantum nature of the proton fundamentally changes the superconducting phase diagram of H<sub>3</sub>S. But among these properties, the current work is mainly involved with the calculated phase stability or phase diagram. By comparing with the theoretical and experimental phase stability (Supplementary Fig. 5 and Supplementary Fig. 14), we can find that the range of corresponding phase stability is slightly different with the transition pressure, which maybe contributed by the anharmonic effects. Compared with high symmetry *Im-3m* of H<sub>3</sub>S, the lower symmetry *P6<sub>3</sub>/mmc* CeH<sub>9</sub> makes the calculation with anharmonic effect difficult. The calculation with anharmonic effects is also extremely hard for the compounds with multi-atoms such as ten-atoms CeH<sub>9</sub>. A more delicate calculation is further required for theorist in the future study.



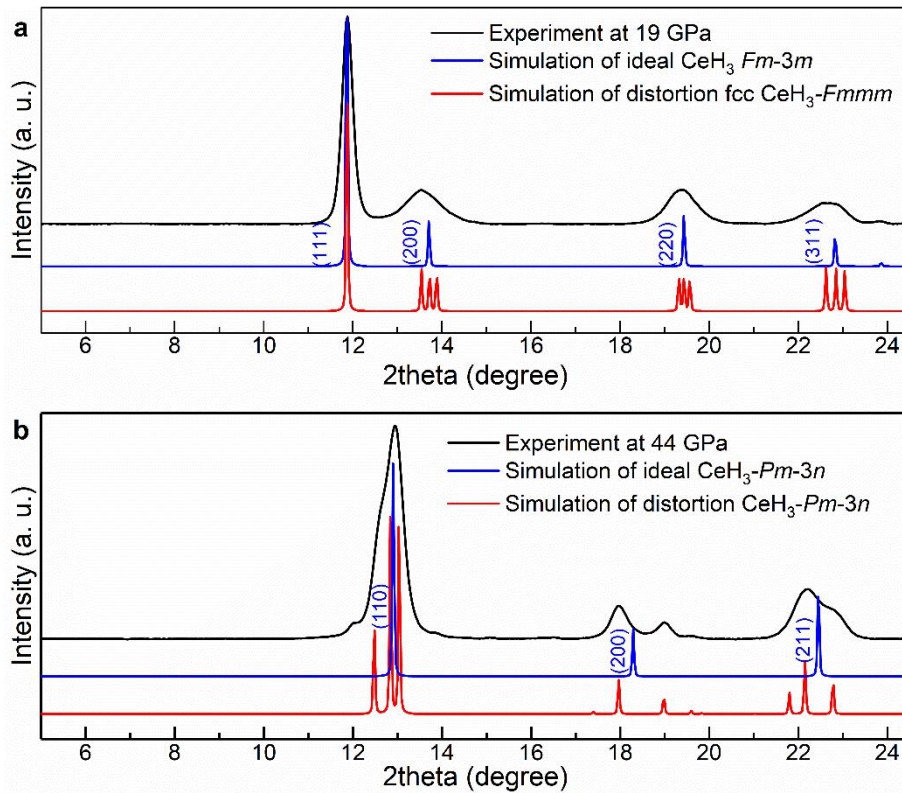
**Supplementary Fig. 1 | The microphotoes of the sample chamber at selected pressure.** After loading hydrogen, the sealed pressure is 3 GPa with about 24  $\mu\text{m}$  Ce sample. With increasing pressure, the volume of black solid sample increases about 46% accompanied with the decrease of hydrogen sample. Above 34 GPa, the sample chamber remains unchanged, and the excess solid hydrogen remained transparent up to the highest pressure 159 GPa.



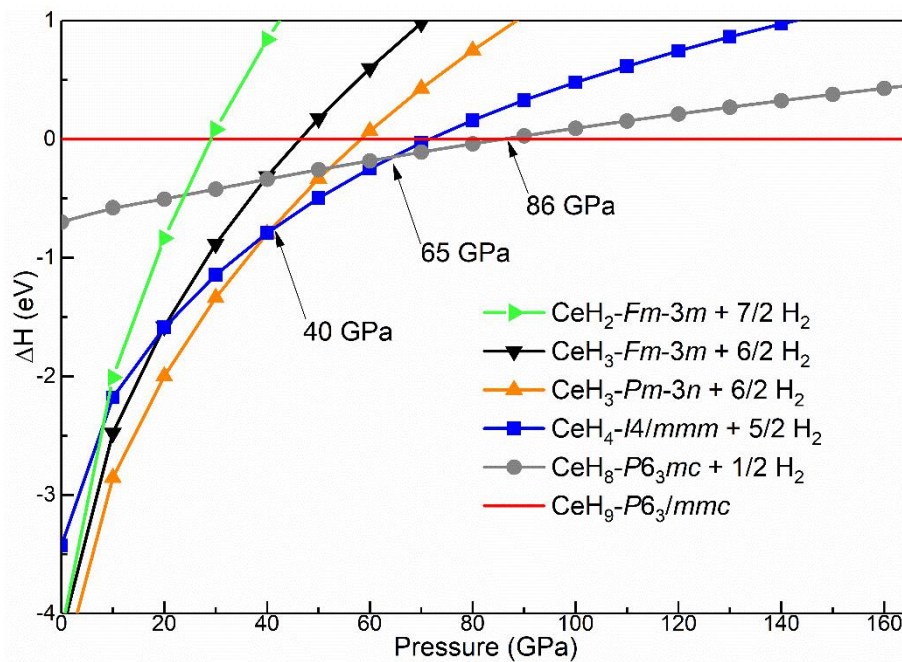
**Supplementary Fig. 2 | Evolution of  $d$ -spacing of Ce polyhydrides with pressure.** The vertical dashed lines indicate the phase transitions in Ce polyhydrides.



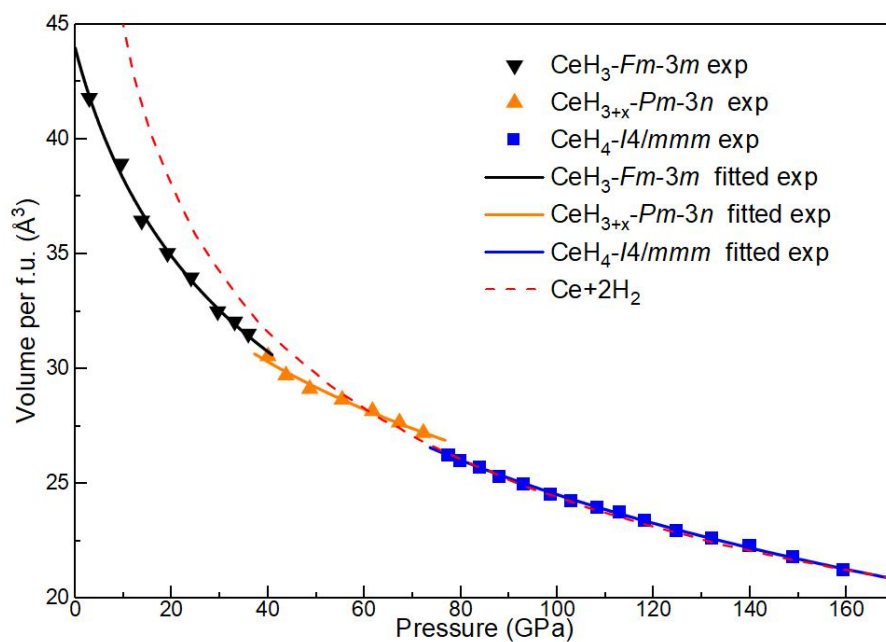
**Supplementary Fig. 3 | The Rietveld refinement on  $\text{CeH}_3\text{-Fm-3m}$  at 3 GPa.** The  $R$  factors are  $R_{\text{wp}} = 18.58\%$  and  $R_{\text{p}} = 12.68\%$  for the refinement.



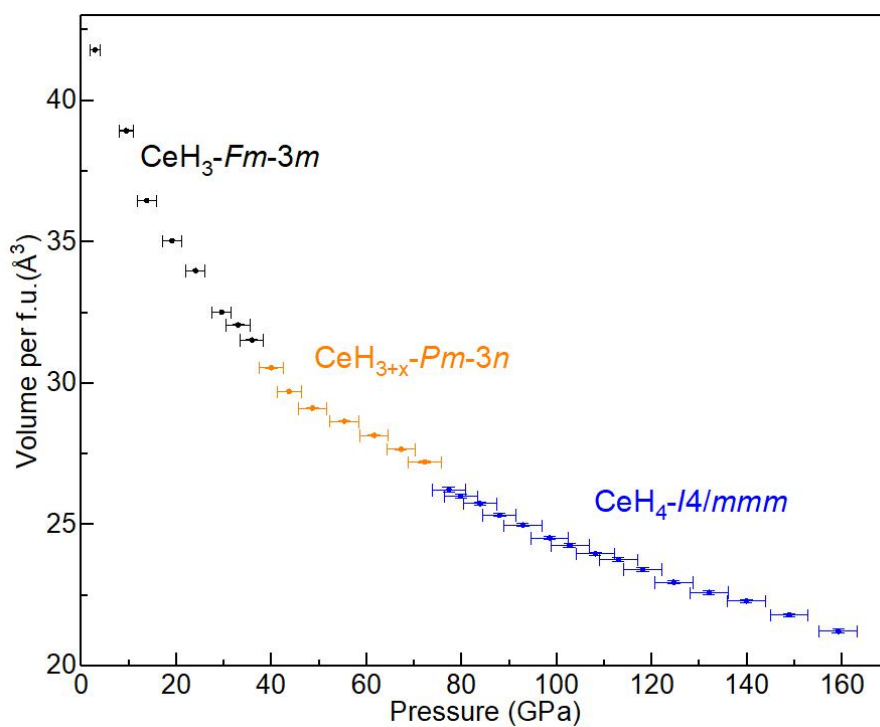
**Supplementary Fig. 4 | Distortion of Ce sublattice at 19 GPa and 44 GPa. a,** Comparison between ideal and distorted lattice of  $CeH_3$ - $Fm-3m$  at 19 GPa. The diffraction peaks (200),(220),(311) have obviously anomalous broaden and shift, indicating a distortion in the fcc Ce sublattice. **b,** Comparison between ideal and distorted  $CeH_3$ - $Pm-3n$  at 44 GPa.



**Supplementary Fig. 5 | The enthalpy of Ce polyhydrides as a function of pressure.**  $\Delta H = H(CeH_n) + (9-n)/2H(H_2) - H(CeH_9)$ . The enthalpy of  $CeH_9$  is used as the reference energy.

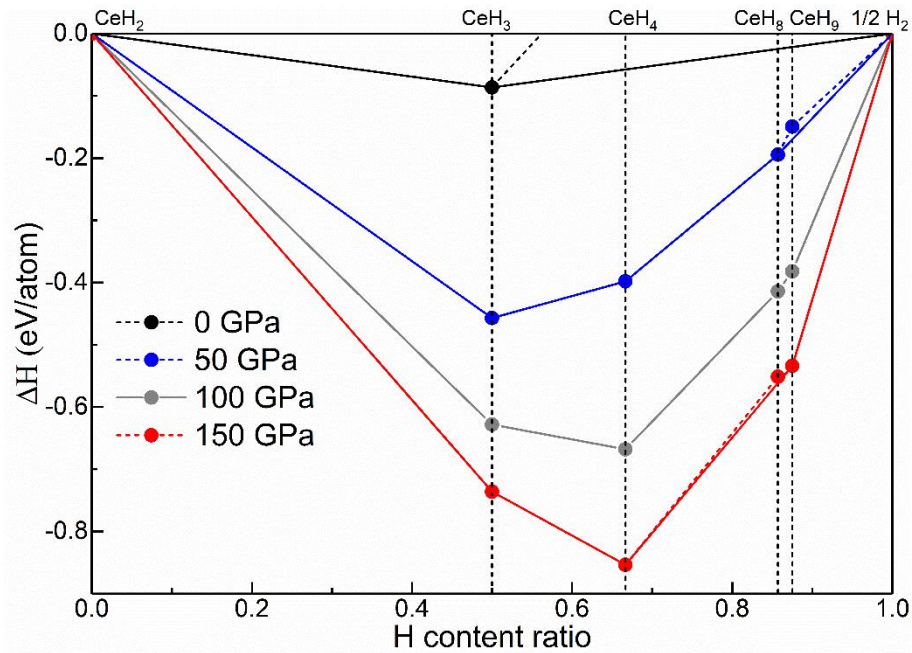


**Supplementary Fig. 6 | The experimental volume of  $\text{CeH}_3$ ,  $\text{CeH}_{3+x}$  and  $\text{CeH}_4$  at different pressures.**

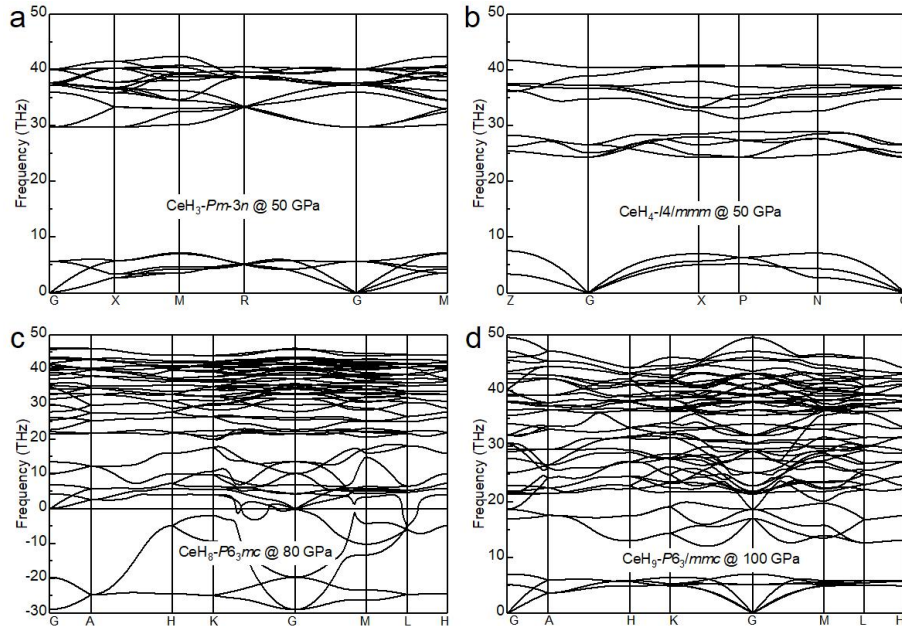


**Supplementary Fig. 7 | The experimental volume of  $\text{CeH}_3$ ,  $\text{CeH}_{3+x}$  and  $\text{CeH}_4$  with error bars.**

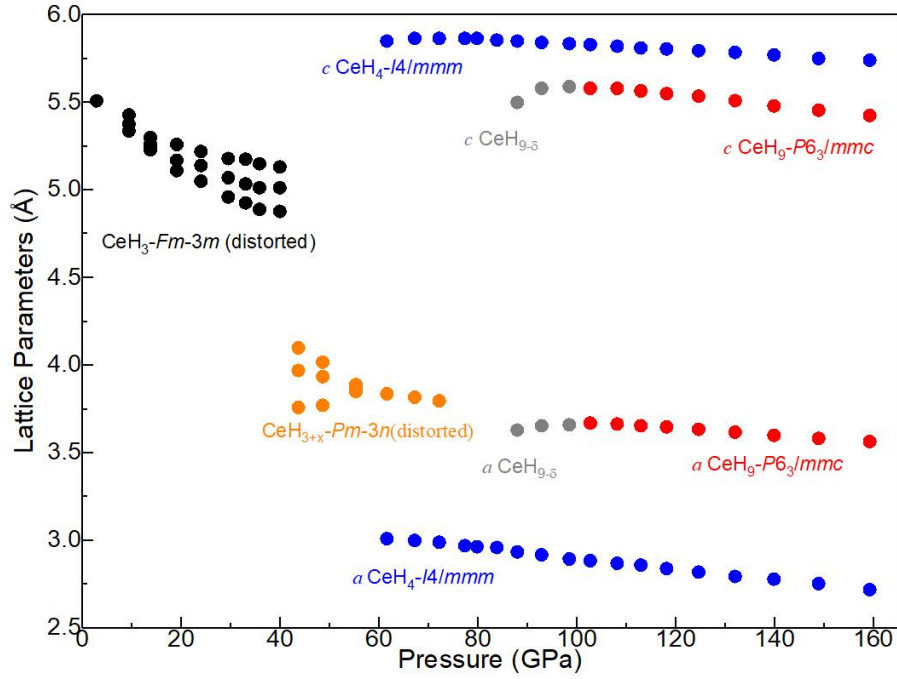




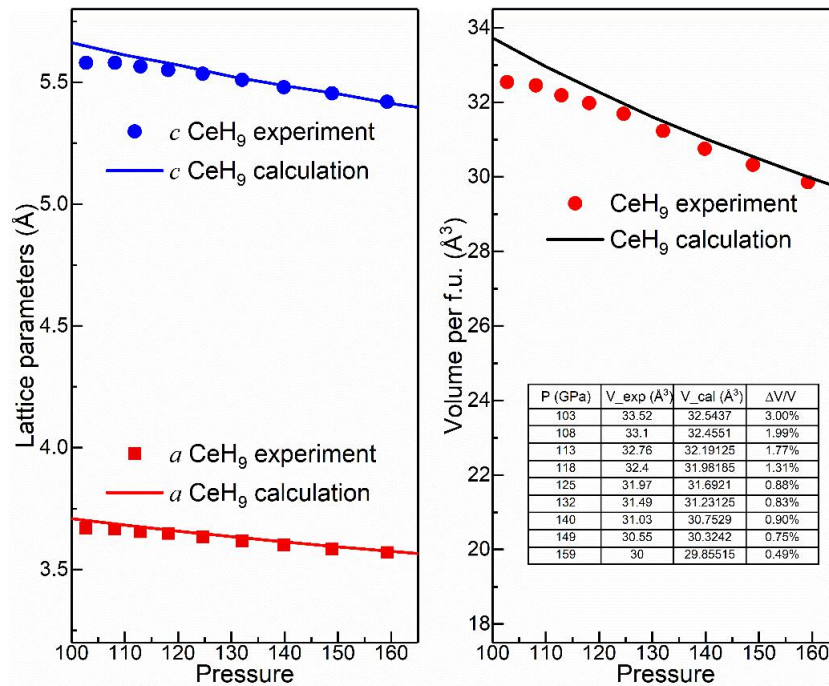
**Supplementary Fig. 8 | The convex hull diagram of Ce polyhydrides at selected pressure.** The structures of  $\text{CeH}_2\text{-}Fm\text{-}3m$ ,  $\text{CeH}_3\text{-}Pm\text{-}3n$ ,  $\text{CeH}_4\text{-}I4\text{-}mmm$ ,  $\text{CeH}_8\text{-}P6_3mc$ ,  $\text{CeH}_9\text{-}P6_3/mmc$  and  $\text{H}_2\text{-}P6_3/m$  and  $\text{H}_2\text{-}C2/c$  are considered respectively.



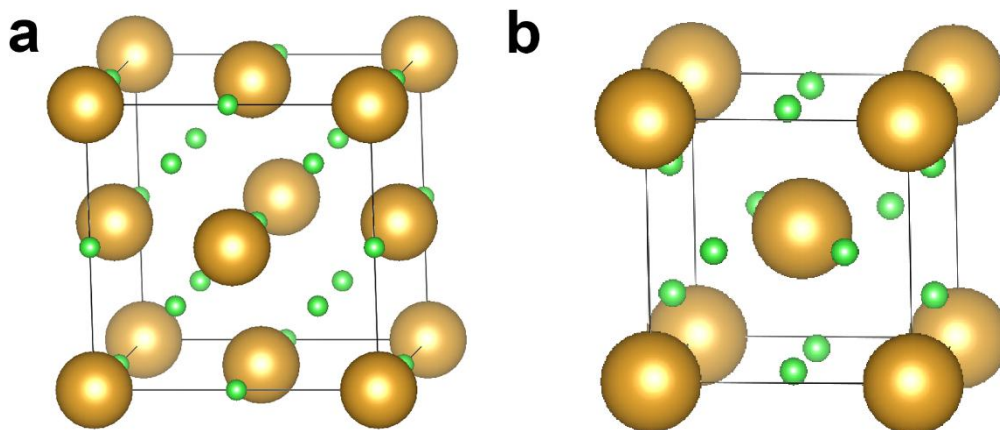
**Supplementary Fig. 9 | (a-d) Phonon dispersion curves for  $\text{CeH}_3\text{-}Pm\text{-}3n$ ,  $\text{CeH}_4\text{-}I4\text{-}mmm$ ,  $\text{CeH}_8\text{-}P6_3mc$  and  $\text{CeH}_9\text{-}P6_3/mmc$  at 50, 50, 80 and 100 GPa, respectively.** Imaginary phonons observed in  $\text{CeH}_8\text{-}P6_3mc$ .



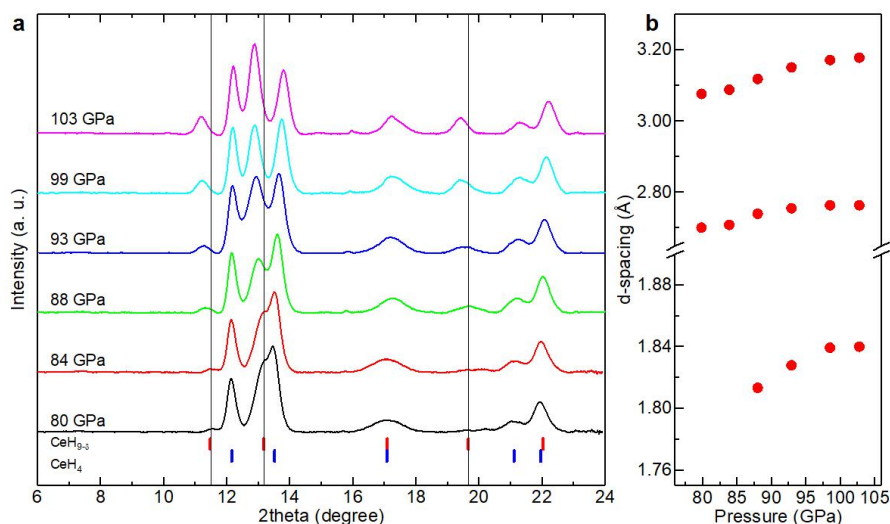
**Supplementary Fig. 10 | Lattice parameters of Ce polyhydrides as a function of pressure.** The evolution of lattice parameters of  $\text{CeH}_3$ - $Fm-3m$ ,  $\text{CeH}_{3+x}$ ,  $\text{CeH}_4$ - $I4/mmm$ ,  $\text{CeH}_9$ - $\delta$ , and  $\text{CeH}_9$ - $P6_3/mmc$ .



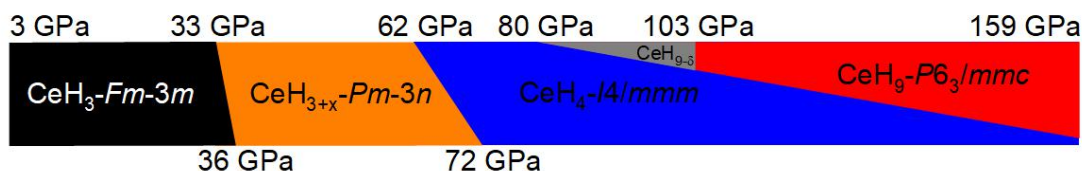
**Supplementary Fig. 11 | Experimental and calculated lattice parameters and volume of  $\text{CeH}_9$  at different pressures.** a, The comparison between experimental and theoretical lattice parameters of  $\text{CeH}_9$  at pressures, and b, The comparison between experimental and theoretical volume of  $\text{CeH}_9$  at pressures. Inset is the volume difference under different pressures.



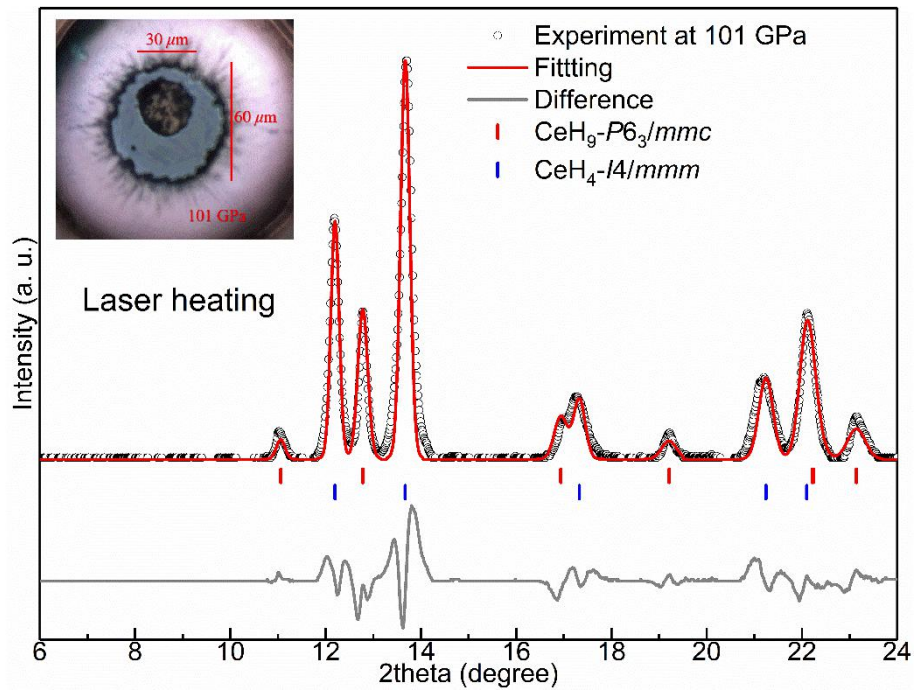
**Supplementary Fig. 12 | The crystal structures of Ce polyhydrides. a,** The crystal structure of  $\text{CeH}_3\text{-Fm-}3m$ . **b,** The crystal structure of  $\text{CeH}_3\text{-Pm-}3n$ .



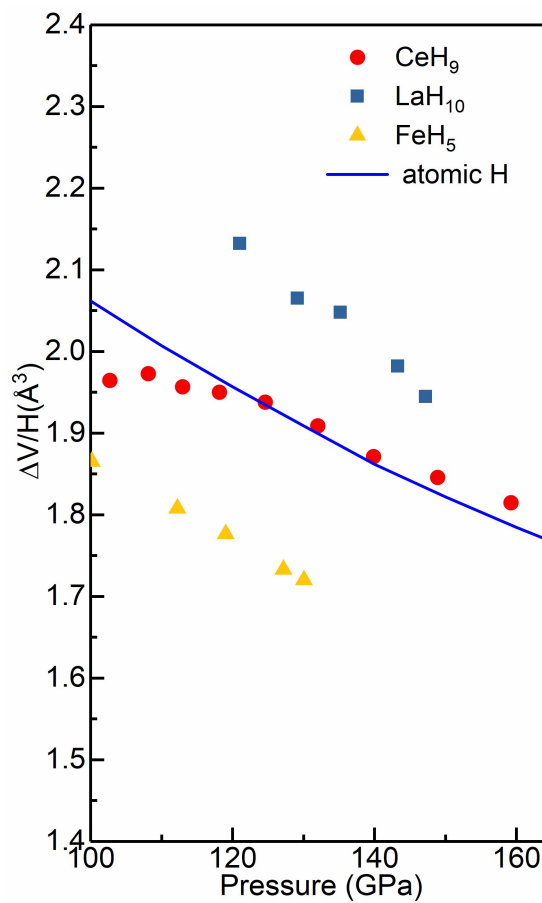
**Supplementary Fig. 13 | The XRD patterns and evolution  $d$ -spacing of  $\text{CeH}_{9-\delta}$ .** a, The XRD patterns in the pressure range 80-103 GPa. At the bottom, the patterns are contributed from  $\text{CeH}_{9-\delta}$  and  $\text{CeH}_4\text{-I4/mmm}$ . With increasing pressure, the pattern shows the abnormal shift to lower angles, indicating the expansion of the volume and new phase with higher stoichiometry. b, Evolution of  $d$ -spacing of  $\text{CeH}_{9-\delta}$  with pressure.



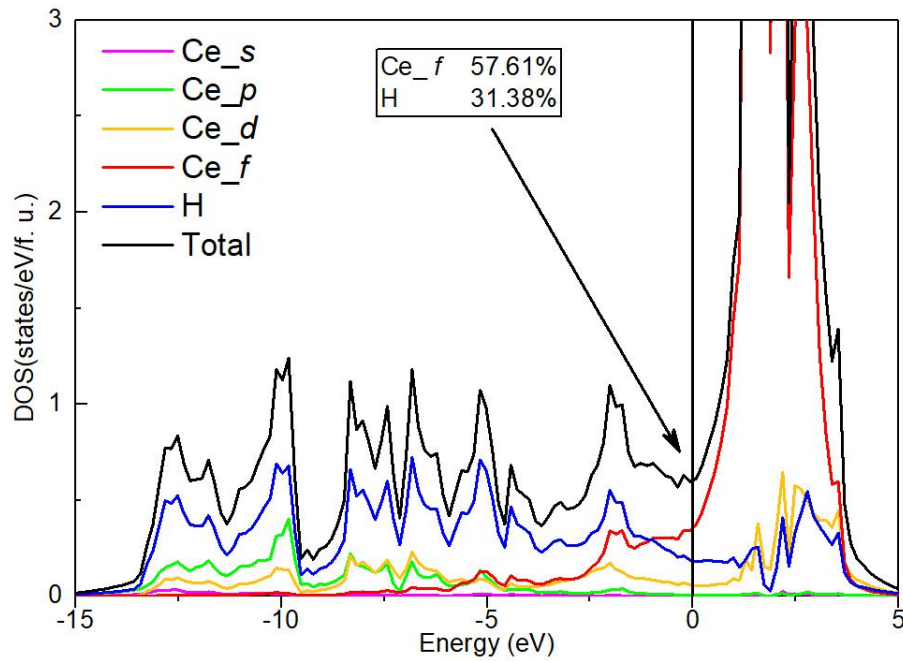
**Supplementary Fig. 14 | Experimental phase diagram of Ce polyhydrides.** The stable regions were confirmed via our experimental XRD patterns. The stoichiometry of Ce hydrides would increase with pressure.



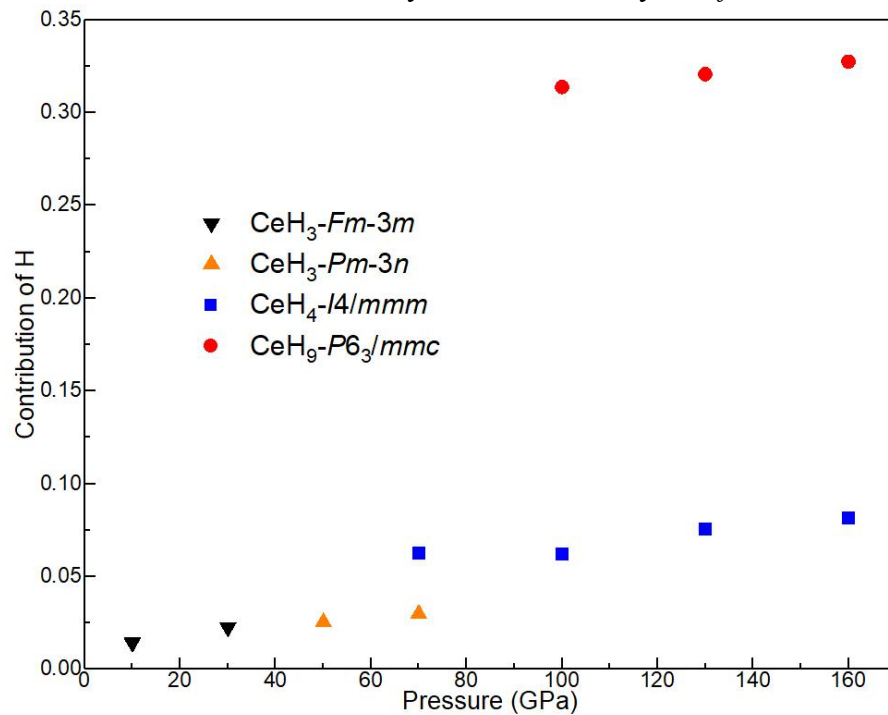
**Supplementary Fig. 15 | The Rietveld refinements on new Ce polyhydrides at 101 GPa after laser-heating.** The  $R$  factors are  $R_{wp} = 19.28\%$  and  $R_p = 13.04\%$  for the refinement.



**Supplementary Fig. 16 | The volume expansion per H atom.**



**Supplementary Fig. 17 | The partial density of electronic states of CeH<sub>9</sub> at 100 GPa.** The substantial contribution of Ce *f* electrons and H *s* electrons to the DOS at the Fermi level of CeH<sub>9</sub> with 31.38% by H and 57.61% by Ce 4*f* states.



**Supplementary Fig. 18 | The contribution of Hydrogen to the density of states at Fermi level in each phase.**

**Supplementary Table 1. Refined lattice parameters of Ce polyhydrides**

Pressure	Space group	Lattice Parameters	Atoms	Atomic coordinates			
				x	y	z	
CeH <sub>3</sub>	3 GPa	<i>Fm-3m</i>	<i>a</i> =5.509 Å	Ce(4 <i>a</i> )	0	0	0
				H(4 <i>b</i> )	1/2	0	0
				H(8 <i>c</i> )	1/4	1/4	1/4
CeH <sub>3</sub>	57 GPa	<i>Pm-3n</i>	<i>a</i> =3.811 Å	Ce(2 <i>a</i> )	0	0	0
				H(6 <i>d</i> )	0	1/4	1/2
CeH <sub>4</sub>	76 GPa	<i>I4/mmm</i>	<i>a</i> =2.970 Å <i>b</i> =5.865 Å	Ce(2 <i>a</i> )	0	0	0
				H(4 <i>d</i> )	0	1/2	1/4
				H(4 <i>e</i> )	0	0	0.365
CeH <sub>9</sub>	159 GPa	<i>P6<sub>3</sub>/mmc</i>	<i>a</i> =3.565 Å <i>b</i> =5.510 Å	Ce(2 <i>d</i> )	2/3	1/3	1/4
				H(2 <i>b</i> )	0	0	1/4
				H(4 <i>f</i> )	1/3	2/3	0.149
				H(12 <i>k</i> )	0.156	0.312	0.062