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# **Lithium-Ion Supercapacitor Using Vertically-aligned Carbon Nanotubes from Direct Growth Technique, and its Electrochemical Characteristics**

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## **Abstract**

This paper reports the fabrication of a lithium ion supercapacitor from vertically-aligned carbon nanotubes (VACNTs) directly grown on a conductive substrate (SUS 310S alloy), using alcohol catalytic chemical vapour deposition technique. CNTs direct growth technique on an electrically conducting foil simplifies the electrode assembly, thus reducing the fabrication process, because the foil can directly act as a current collector. With the VACNT direct growth technique, the supercapacitor electrode was easily prepared and assembled with a non-aqueous 1 M LiPF<sub>6</sub> electrolyte. Experimental results show that CNTs (multi-walled type structures of good quality) were perpendicularly grown to the substrate. This device demonstrates a specific capacitance of up to 101 F g<sup>-1</sup> (at a scan rate of 1 mVs<sup>-1</sup>), and a high-rate capability, up to a scan rate of 1000 mVs<sup>-1</sup>. The VACNT electrode electrochemical performance was also measured by galvanostatic charge-discharge and electrochemical impedance spectroscopy. The effect of free standing CNTs direct growth on the current collector makes insulating binder material unnecessary, thus producing better ion accessibilities to its surface. This also contributes to the good and reliable electrochemical supercapacitor performance.

**Keywords:** vertically-aligned CNT; alcohol catalytic CVD; direct growth technique and Li- ion supercapacitor.

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## **Introduction**

In modern engineering, electrochemical capacitors (EC), also known as electrochemical double layer capacitors (EDLC) or supercapacitors, are able to store electrical charge in a growing variety of applications in many fields [1], such as portable electronics, digital communications, hybrid electric vehicles, etc.

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[2]. EC, including lithium (Li)-ion capacitor (LIC), fills the critical performance gap between batteries high energy density and conventional capacitors high power density [3]. ECs' main advantages are that they can provide high power density, high energy density, long life cycle and fast charge-discharge rate [4, 5, 6 and 7]. EDLC's energy storage mechanism arises from electrostatic charge or physical separation at the electrode-electrolyte interface [8]. Pseudocapacitor charge storage ability depends on the fast redox reactions [9].

Nowadays, nanostructured carbon materials, such as CNTs, have been focused as a highly potential electrode material for EC, due to their remarkable electrical charge storage ability, large specific surface area, chemical stability, and excellent electrical conductivity [10 and 11]. Nowadays, researchers are focusing on the vertically aligned CNTs (VACNTs) as an electrode material, due to their advantages for specific surface areas, and to the ion accessibility that is more efficient and better, compared to entangled CNT [12]. Alcohol catalytic chemical vapour deposition (ACCVD) is a promising technique to produce VACNTs, being well-known for its economical merit, wide selectivity of substrates, and highly yielding catalytic reaction for CNTs growth [13]. The used alcohol, typically ethanol ( $C_2H_5OH$ )-grown nanotubes, is nearly free from any amorphous carbon, due to the  $OH^\cdot$  radical etching action [14].

In addition, the electrode preparation technique is another critical issue of EC electrochemical performance [15]. In the conventional electrode preparation technique, such as in activated carbon or entangled CNTs, the electrode might have large internal resistance, due to the inclusion of insulating binder or adhesive materials, which become barriers in the access paths for electrons and electrolyte ions [16]. Hence, direct growth technique of aligned CNTs on a conducting substrate is an attractive electrode preparation technique, due to the direct electrical contact between the VACNTs electrode and the current collector. The conducting substrate can directly act as current collector, simplifying the conventional electrode assembly process. Thereby, this technique could reduce equivalent series resistance (ESR), and enhance capacitance performance [17].

Herein, the primary goal was to work on the direct growth of VACNTs metallic alloy (SUS 310S foil) substrates, using ACCVD technique to fabricate the supercapacitor electrode. The supercapacitor was constructed from the as-grown VACNT electrodes and a non-aqueous 1 M lithium hexafluorophosphate ( $LiPF_6$ ) electrolyte. The as-grown (AG)-VACNT electrode samples' morphological and structural properties were examined by electron microscopic and Raman spectroscopy techniques. The secondary goal was to determine the VACNT electrode electrochemical properties in a Li-ion electrolyte, from the viewpoint of cycling behaviors, charge discharge ability and frequency response analysis.

## **Experimental**

### ***Preparation of VACNT electrodes***

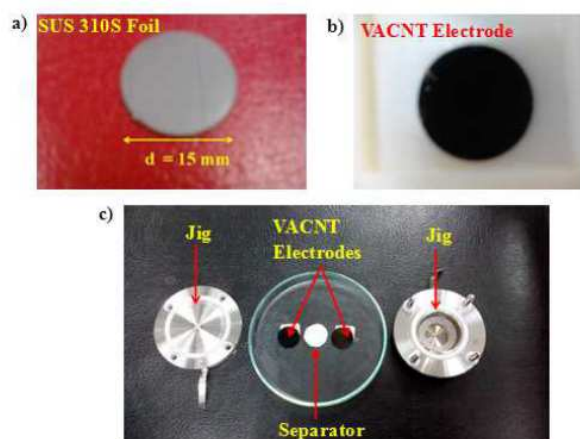
For the conducting substrate to be used as current collector, SUS 310S foils with 0.2 mm thickness were cut into 15 mm diameter round substrates. The average weight for two SUS 310S foils with this dimension is 0.2761 g. Thin films of Al

(25 nm) and, subsequently, of Co (7–9 nm) were deposited on the SUS 310S foils, using radio frequency (RF) magnetron sputtering deposition. After the thin films deposition, the sample weight (two thin films) was measured; the average weight of thin films (Co/Al-O) was 0.0060 g. For the formation of thermally-oxidized aluminium oxide (Al-O), the substrate was transferred to the CVD furnace (RTP-1000D4), and baked at 500 °C in static air, for 10 min, just before the CVD process. As a pre-treatment, argon gas was purged at a pressure of 350 mTorr, concurrently with a 5 min rapid heating up to CVD temperature of 725 °C. The argon gas was continuously flowed for another 5 min of the annealing process. For CNT growth, the CVD processing parameter was fixed at 725 °C, with ethanol flow rates of 100–130 sccm, 10-15 Torr internal pressure, and 15 min CVD processing time. The samples weight was again measured after the CVD process, and the average weight of as-grown VACNT/ Co/Al-O was 0.0062 g. Therefore, CNTs' net mass loading, as an active material (0.0002 g or 0.20 mg), was obtained by subtraction of the foil weight increase, with only Co/Al-O films, from that with CNTs.

A field emission scanning electron microscope (FESEM, 20 kV) and a transmission electron microscope (TEM, 120 kV) were used to characterize the AG-VACNTs morphology. Raman spectroscopy was used to further investigate the VACNTs structural and graphitic properties, using the 632.8 nm wavelength laser excitation: a HeNe laser.

### ***Electrochemical measurements***

The supercapacitor was built with symmetric AG-VACNT electrodes cells, separated by 25- $\mu$ m polypropylene (PP; Tonen) and a 1 M LiPF<sub>6</sub> solution as electrolyte. The capacitor was assembled using a battery jig (Takumi Giken), as shown in Fig. 1.



**Figure 1.** Construction of a Li-ion capacitor using VACNT electrodes. (a) digital image of the blank SUS 310S foil, (b) digital image of grown VACNTs on a SUS 310S foil, and (c) VACNT electrode, and the components for the supercapacitor cell.

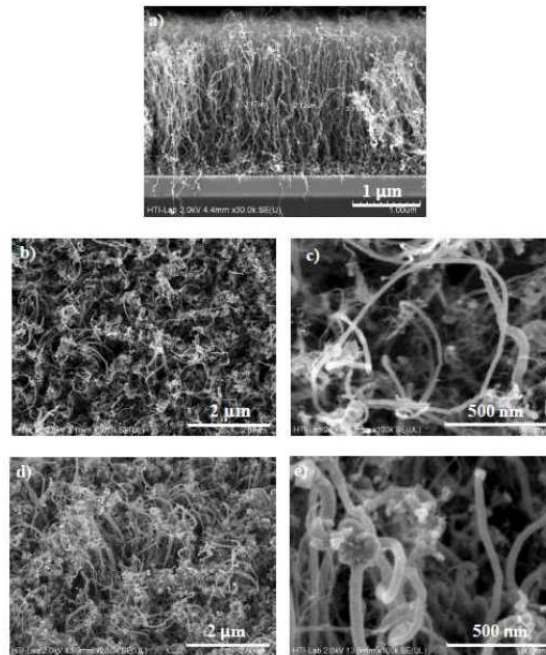
Cell assembly was carried out in a glove box, with controlled oxygen and moisture concentration. All electrochemical measurements were carried out at room temperature (25 °C).

CV and galvanostatic charge-discharge measurements were performed using Wonatech: WBCS3000 potentiostat/galvanostat. CV measurement was conducted to measure the electrode cyclic behaviour, as well as the specific gravimetric capacitance ( $C_{sp}$ ). The potential was set from 0.0 to 3.0 V, due to the Li electrolyte's thermodynamic window, which can actually be up to 4.6 V. The scan rate was set to 1, 5, 10, 50, 100, 500, and 1000  $\text{mV s}^{-1}$ . In addition, charge-discharge measurements were conducted to measure the charging and discharging behaviour with the different applied currents, namely 1, 5 and 10 mA. The EIS measurement was then conducted using a frequency response analyzer (Metrohm Autolab), ranging from 100 KHz to 0.01 Hz, at the amplitude of 10 mV.

## Results and discussion

### *Morphological and structural properties of the VACNT electrode*

Fig. 2 shows FESEM images of VACNTs that were directly grown on a SUS 310S foil.



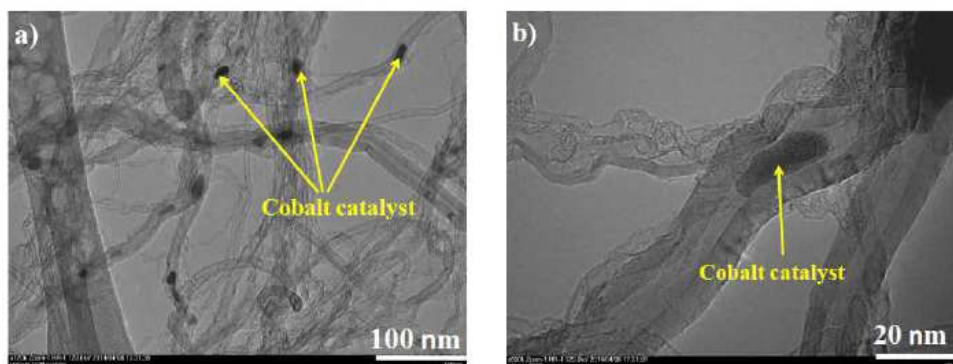
**Figure 2.** FESEM images of VACNTs directly grown on a SUS 310S foil. (a) cross-section view, (b) top view (low magnification), (c) top view (high magnification), (d) 45° tilted (low magnification), and (e) 45° tilted (high magnification).

Observations were directly made from the above substrate, at a tilt angle of 45 °, and cross-sectionally, to confirm the vertical structure of the CNTs forest on the substrates. The CNTs were found to be vertical aligned, perpendicular to the substrate, where most of the substrate's area was covered by VACNTs (Fig. 2 (a)). From the cross section image, the VACNTs height is approximately 2.10-μm. It was experimentally proved that the CVD-grown VACNTs height can increase with the extended CVD processing time, and some adjustments on other

parameters. In this work, and with the present CVD system setup, 15 min could be a short time. However, the effect of extended CVD processing time (i.e., more than 30 min) might also damage the VACNTs structure.

Meanwhile, TEM image in Fig. 3 reveals that the VACNT diameter grown by ACCVD technique varies from 2 to 20 nm, and that the VACNTs grown on the SUS 310S foil are multi-walled CNTs. There are also some Co particles in the TEM image, which may be the cobalt catalyst that maintains the CNTs growth.

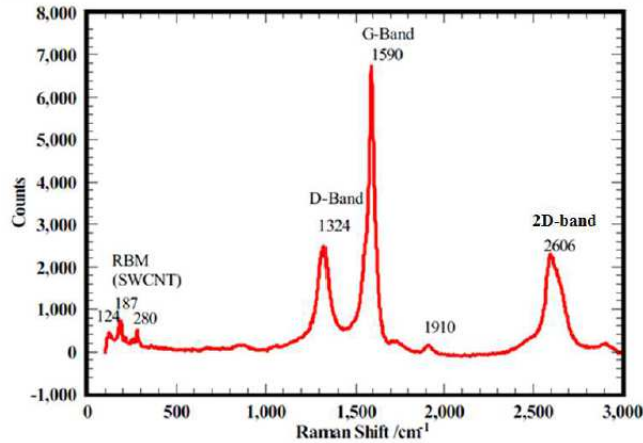
The VACNTs were catalytically grown on SUS 310S foils from a Co catalyst and catalyst-supports, respectively. Thermally oxidized Al-O, used as a catalyst-support layer, plays an important role in CNTs vertical growth, due to its pore structures and roughened surface [12]. The porous structure of the metal oxide layer will trap catalyst nanoparticles during the annealing process, and produce strong adhesion force, due to the obtuse contact angle amid them. During CNT growth, the metal oxide layer will act as a core to support CNT alignment, and simultaneously prevent the catalyst nanoparticle lifts off the substrate, during initial hydrocarbon decomposition and carbon diffusion [18]. The Al-O layer also supposedly prevents catalyst nanoparticles aggregation [11]. Therefore, Al-O used as buffer layers will enhance the aligned CNT growth efficiency [19]. It is also important to note that the CNT diameter varied due to some non-uniformities of the catalyst thin film evolution during the CVD annealing process.



**Figure 3.** TEM images of VACNTs after being cut from the SUS foil, where Co catalyst particles can be observed inside, and at the edge of the CNTs. (a) lower magnification, and (b) higher magnification.

Raman spectroscopy is a useful non-destructive method for the structural characterization of different carbon materials [20]. From Raman spectroscopy analysis, CNTs quality is according from G-band to D-band peak intensity ratio,  $I_G/I_D$ . Large G-band to D-band intensity ratio generally corresponds to CNTs of high quality [21]. From Raman spectra measured by 633 nm laser excitation (Fig. 4), the intensity ratio ( $I_G/I_D$ ) was found to be 2.70. The Raman spectra also exhibit peaks of the radial breathing mode (RBM) from 100 to 280  $\text{cm}^{-1}$ , which reveals the presence of single-walled CNTs, a D-band (defect, amorphous carbon) at approximately 1324  $\text{cm}^{-1}$ , and a G-band (graphite, CNTs) at approximately 1590  $\text{cm}^{-1}$ . The peak at 2606  $\text{cm}^{-1}$  represents the 2D-band peak, which can indicate the presence of graphene. However, we should note that CNT

has the structure of a rolled-up graphene sheet, which explains the 2D-band peak appearance. High G-band intensity confirms the fact that CNTs are dominantly grown. Also, it is possible that some of the CNTs were opened up (broken), and that the Raman laser has detected that particular graphene sheet/film. Raman spectra analysis also showed that VACNTs were common or dominant in multi-walled CNTs. This result was in good agreement with the TEM observation, and the low peak of RBM region. Therefore, this confirmed that CNTs could also effectively be grown on metal alloys, using ACCVD system.



**Figure 4.** Structural and graphitic properties of VACNTs directly grown from Co/Al-O/SUS 310S, confirming a relatively good quality of the CNTs

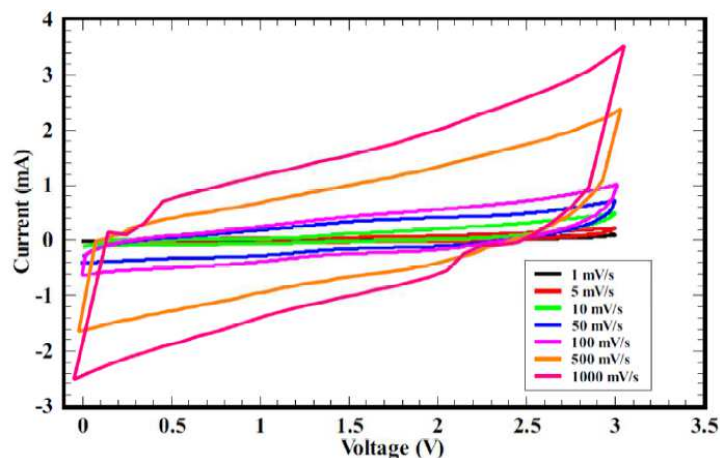
#### ***Electrochemical performances of the Li-ion capacitor***

The performance and charge storage behaviour of VACNTs electrode in a 1 M LiPF<sub>6</sub> electrolyte were investigated by CV analysis at various scan rates, using 1, 5, 10, 50, 100, 500 and 1000 mV s<sup>-1</sup> scan rates (Fig. 5). The Li-ion electrolyte, which acts as an electric current carrier, is another important factor in the electrochemical performance enhancement [22]. In this work, LiPF<sub>6</sub> was used as an electrolyte, with the potential window range from 0.0 to 3.0 V. From the CV measurement with various scan rates, it was confirmed that the VACNT electrode has double-layer capacitance behaviour, when used as an electrode material in ECs application. At 1 mV s<sup>-1</sup> of CV scan rate, the calculated C<sub>sp</sub> was 101 Fg<sup>-1</sup> (0.0 - 3.0 V potential). The C<sub>sp</sub> of the VACNT electrode was calculated from equation 1, using the recorded CVs at a scan rate of 1, 5, 10, 50, 100, 500 and 1000 mV s<sup>-1</sup>.

$$C_{sp} = \left( \int_{E_1}^{E_2} i(E) dE \right) \div 2(E_2 - E_1)mv \quad (1)$$

where E<sub>1</sub> and E<sub>2</sub> are the cutoff potentials in CV, i (E) is the current and the total voltammetric charges obtained by integration of positive (charge) and negative (discharge) sweeps in CV, (E<sub>2</sub> - E<sub>1</sub>) is the cell potential window width, m is the CNT mass per electrode, and v is the scan rate [23]. In our previous work in reference [23], we have discussed that VACNTs optimum height is critical for a supercapacitor to have a good performance. VACNTs optimum height must not

exceed 60- $\mu\text{m}$ . Thus, a 2- $\mu\text{m}$  height of VACNT forest is considered acceptable to achieve its maximum capacitance value during the device characterization.

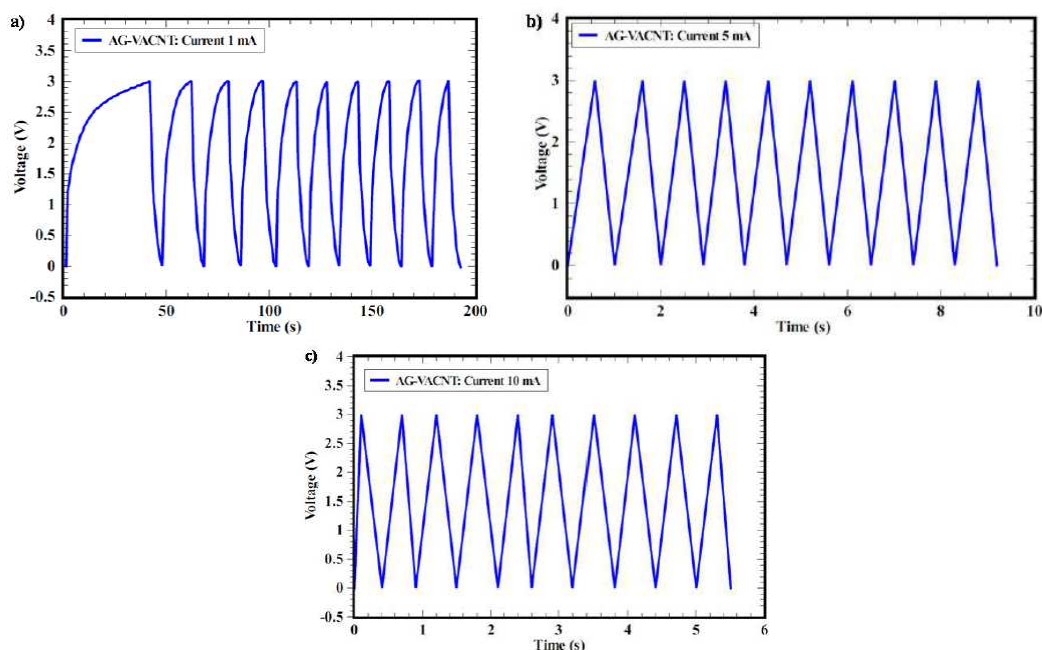


**Figure 5.** Charge-discharge performance of VACNT electrode cooperated with Li-ion electrolyte from the CV curves measured at various scan rates (1, 5, 10, 50, 100, 500, and 1000  $\text{mV s}^{-1}$ ).

The CV curves also show near rectangular shapes, even at the scan rate of 1000  $\text{mV s}^{-1}$ , with no redox peak, indicating a supercapacitive behaviour, and becoming the typical shape of EDLC. This result also confirms that the AG-VACNT electrodes have high reversibility and high power density [24 and 25]. These significant results from  $C_{\text{sp}}$  value and CV curve shape revealed that the AG-VACNTs electrode had good electrical contact with the SUS 310S substrate. CV near rectangular shape performed at high scan rates of 1000  $\text{mV s}^{-1}$ , and showed a typical ideal capacitive charging and discharging characteristic across the potential range from 0.0 to 3.0 V, suggesting that the integrated VACNT-current collector systems have very rapid current response on voltage reversal with low equivalent series resistance (ESR) [17 and 20].

The VACNT electrode performance in 1 M  $\text{LiPF}_6$  was further evaluated using galvanostatic charge-discharge analysis at 1, 5, and 10 mA (Fig. 6 (a), (b), (c)), from 0.0 to 3.0 V, and back to 0.0 V. It was continuously measured up (but not limited) to 10 cycles, and was very stable in terms of charging and discharging behaviour.

Various constant currents were applied to observe the ability of the electrodes to work from lower to higher current densities. The results showed only small IR (ohmic) drop when using 1 mA current, while IR drop at 5 mA and 10 mA was not obvious at all. This indicates very small internal resistance and excellent power characteristics of the electrode. All the charge-discharge curves for this supercapacitor are near triangular in shape, and without significant IR drop, indicating that the electrode has ideal capacitive charging-discharging behaviour, with less faradaic reaction and excellent electrochemical reversibility [26 – 28]. The success of the working device neglects the effect of the Co nanoparticles that incorporated the as-grown CNTs.



**Figure 6.** Charge discharge behaviour of the Li-ion capacitor measured at different applied currents. (a) 1 mA, (b) 5 mA, and (c) 10 mA.

Small and/or no IR drop for this capacitor show direct connection between the aligned CNT electrode and the current collector, accelerating the ion transport for the charge storage mechanism. The charge-discharge curve showed that the EC had very rapid charge-discharge performance (in second), which is a well-known EC advantage. On the other hand, the charge-discharge cycle duration was found to decrease with increasing current. The results show that the charge-discharge duration for 10 cycles was 190 s, 9.2 s and 5.5 s, at different 1 mA, 5 mA, and 10 mA applied currents, respectively. At low current, the charging/discharging process took more time, which is attributed to the sufficient insertion or release of electrolyte ions [29].

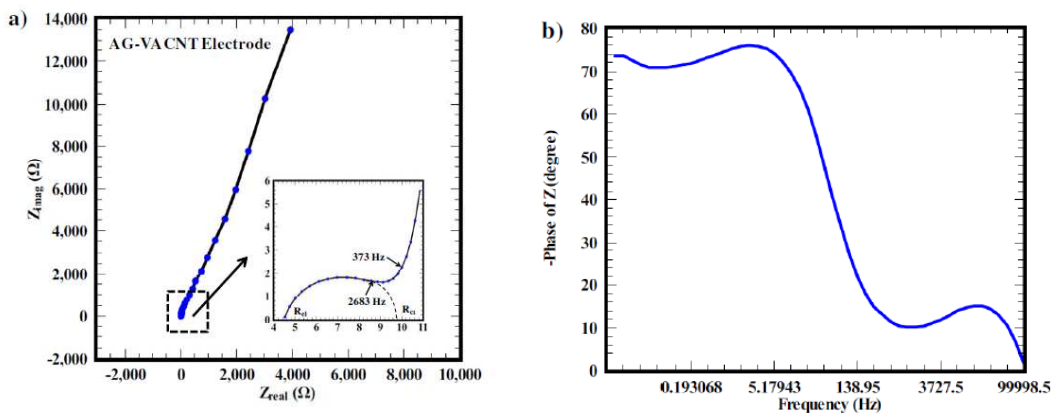
Further evaluation of the VACNT electrode's electrochemical properties was done using EIS analysis. EIS is a valuable technique to investigate energy devices' resistivity performance, such as charge transfer and electrode resistance. Fig. 7(a) illustrates the Nyquist plot of VACNT electrodes in a 1 M LiPF<sub>6</sub> electrolyte. The inset shows the spectrum enlargement in high frequency region. The impedance behaviour was composed of three major processes occurred in the high, medium and low frequency regions, respectively [30]. Theoretically, features of a semicircle over the high frequency region, and a vertical line parallel to the low frequency range, show an ideal Nyquist plot [29].

Nyquist plots show almost vertical lines parallel to the imaginary axis in the low frequency region, which identifies an ideal capacitive behaviour of the VACNTs electrode, with a relatively low ESR value. This indicates that the CNT supercapacitors behave like pure capacitors at low frequencies [26]. At higher frequencies (inset of Fig. 7 (a)), the presence of a small semicircular shape of the curve represents the time constant related to the electrical charge transfer process at the electrode/electrolyte interface, or the diffusion of electrolyte ions to the VACNTs electrode. Also, the arc in the high frequency part is probably due to



the VACNTs direct contact with the substrate [17]. The electrolyte resistance ( $R_{el}$ ) and the charge transfer resistance ( $R_{ct}$ ) measured from EIS data were found to be  $4.5 \Omega$  and  $8.6 \Omega$ , respectively. The semicircle diameter ( $R_{ct} - R_{el}$ ) represents ESR value [31]. ESR value for this AG-VACNT was found to be  $4.1 \Omega$ .

$Li^+$  ions from  $LiPF_6$  electrolyte started to diffuse at  $2683 \text{ Hz}$  ( $8.6 \Omega$ ), and the diffusion stopped at  $373 \text{ Hz}$  ( $10 \Omega$ ), where the maximum capacitance was reached and became dominant. The AG-VACNTs electrode also possessed a frequency dependent phase angle around  $74^\circ$  (Figure 7 (b)). Therefore, the results strongly suggest that the directly grown VACNTs on the SUS 310S foils possessed low device resistivity, due to good electrical and ionic conductivity. This also indicates that the electron and electrolyte ions easily diffused and/or moved through the VACNTs electrode. The excellent performance highlights the important role of the VACNTs in storing and accumulating charge, thereby boosting both power density and energy density [32].



**Figure 7.** VACNT frequency response analysis in a  $LiPF_6$  electrolyte. (a) Nyquist plot showing the imaginary versus real part of impedance. The inset is the enlargement of the plot in the high frequency region, and (b) Bode plot of the supercapacitor with a frequency dependent phase angle of  $74^\circ$ .

## Conclusions

A lithium-ion supercapacitor was fabricated from vertically-aligned multi-walled carbon nanotube (VA-MWCNT) electrodes, and  $1 \text{ M } LiPF_6$  as an electrolyte. From the electron microscopy observations, CNTs have a diameter in the range of  $2$  to  $20 \text{ nm}$ , and a height of approximately  $2.10\text{-}\mu\text{m}$ . On this type of symmetric capacitor electrode, CV analysis showed the typical rectangular shaped curves, suggesting the double layer capacitance behaviour of the electrode. This capacitor has a maximum gravimetric capacitance of  $101 \text{ F g}^{-1}$ , calculated from the CV curve at a scan rate of  $1 \text{ mV s}^{-1}$ . During charge-discharge measurements, a small IR drop was confirmed, and a near triangular shape of the curve indicates low series resistance and a high power characteristic. This was consistent with the EIS Nyquist plot that exhibited a close to vertical line along the imaginary axis, which again revealed the ideal capacitive properties of directly-grown CNTs on the conducting substrate.

Additionally, further advancements on preparing aligned CNT electrodes are highly desired. For example, research on the optimum vertical height of the CNTs, and the usage of nano imprinting technique, to design the catalyst films/particles display for a consistent arrangement of the aligned CNTs. By this, ion accessibility can be improved, which could be another interesting feature to explore, for higher energy density and power density devices.

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