

1 Mechanisms of fault mirror formation and fault healing in carbonate rocks

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12  
13 Abstract

14 The development of smooth, mirror-like surfaces provides insight into the mechanical behaviour  
15 of crustal faults during the seismic cycle. To determine the thermo-chemical mechanisms of fault  
16 mirror formation, we investigated carbonate fault systems in seismically active areas of central  
17 Greece. Using multi-scale electron microscopy combined with Raman and electron energy loss  
18 spectroscopy we show that fault mirror surfaces do not always develop from nanogranular volumes.  
19 The microstructural observations indicate that decarbonation is the transformation process that  
20 leads to the formation of smooth surface coatings in the faults studied here. Piercement structures  
21 on top of the fault surface indicate calcite decarbonation, producing CO<sub>2</sub> and lime (CaO). Lime  
22 subsequently reacts to portlandite (Ca(OH)<sub>2</sub>) under hydrous conditions. Nanoscale imaging and  
23 electron diffraction reveals a thin coat of a non-crystalline material sporadically mixed with nano-  
24 clay, forming a complex-composite material that smooths the slip surface. Spectroscopic analyses  
25 reveal that the thin coat is non-crystalline carbon. We suggest that ordering (hybridisation) of

26 amorphous carbon led to the formation of partly-hybridised amorphous carbon but did not reach  
27 full graphitisation. Calcite nanograins, < 50 nm in diameter, are spatially associated with the carbon  
28 and indicate that the decomposition products acted as a crystallisation medium. Within this  
29 medium, portlandite back-reacted with CO<sub>2</sub> to form nanocrystalline calcite. Consequently, two  
30 types of calcite nanograins are present; nanograins formed by grain-size reduction (primary, > 100  
31 nm) and new nanograins (< 50 nm) formed by back-reaction (secondary). Hence, we suggest that  
32 the new, secondary nanograins are not the result of comminution during slip but originate from  
33 pseudomorphic replacement of calcite after portlandite. The continuous coverage of partly-  
34 hybridised amorphous carbon on all samples suggests that calcite decarbonation products may  
35 develop across the entire fault surface, controlling the formation of carbonate fault mirrors, and  
36 may facilitate slip on a decarbonation-product glide film.

## 37 1 Introduction

38 Brittle deformation of upper-crustal rocks can result in high-magnitude seismic events (Scholz,  
39 1998). Mirror slip surfaces (MSSs) along principal slip zones in carbonate rocks provide an  
40 excellent opportunity to investigate the deformation processes that occur in relation to slip events  
41 in the seismogenic zone. Carbonate-hosted faults can produce seismic events with magnitudes of  
42  $M_w = 6$  and greater. These events present a hazard to populated regions, such as the Mediterranean.  
43 The defining feature of MSSs is the high degree of visible-light reflectance resulting from a low  
44 surface roughness. MSSs can form at seismic slip velocities and, therefore, may indicate paleo-  
45 seismicity (Siman-Tov et al., 2013; Smith et al., 2013; Fondriest et al., 2013; Kirkpatrick et al.,  
46 2013; Spagnuolo et al., 2015). However, MSS have also been developed at sub-seismic slip  
47 conditions in deformation experiments (Tisato et al., 2012; Verberne et al., 2014) and, thus, they  
48 are not necessarily diagnostic of seismic slip.

49           The studies from Siman-Tov et al. (2013), Collettini et al. (2014) and Verberne et al. (2014)  
50 suggest that the low surface roughness of MMS may be the result of nano-sized grains ( $< 1 \mu\text{m}$ ),  
51 which constitute the uppermost layers of many principal slip surfaces. However, work by Fondriest  
52 et al. (2013) demonstrates that truncated, large grains can be part of mirror surfaces as well,  
53 illustrating that the formation of ultra-polished surfaces does not necessitate a nanograin coating  
54 along the slip surface. In addition, nanogranular coatings are also produced in deformation  
55 experiments under seismic conditions (Green et al., 2015; Spagnuolo et al., 2015). Therefore, the  
56 formation of natural MSSs has been associated with the development of a nanogranular slip-surface  
57 coating during seismic events. These nanogranular coatings may in addition control the frictional  
58 behaviour of the fault (Han et al., 2007a, b; Han et al., 2010; Di Toro et al., 2011; Bullock et al.,  
59 2014; De Paola et al., 2015; Smeraglia et al., 2017).

60           Current research suggests different formation mechanisms for carbonate mirror slip  
61 surfaces: (1) Formation of elongate, twin-derived beams by plasticity and subsequent brittle  
62 fracturing of the beams to form a nanograin coating (Siman-Tov et al., 2013) and (2) localised  
63 dynamic recrystallisation and static recrystallisation (Smith et al., 2013). Several studies have  
64 reported decarbonation products during or after experiments that produced fault mirrors (Han et  
65 al., 2007a and b, De Paola et al., 2011, Smith et al., 2013, Fondriest et al., 2013, Delle Piane et al.,  
66 2017). Pluymakers and Røyne (2017) found an increase in nanospherules (nanoparticles)  
67 responsible for reducing the surface roughness after heating, which demonstrates that existing  
68 mirror slip surfaces could be affected by heating events. These studies suggest that the formation  
69 of nanograins and fault mirrors are intrinsically linked, tying the formation of mirror slip surfaces  
70 to the formation of nanograins.

71           One of the main mechanisms of fault-gouge formation is grain-size reduction by  
72 comminution. Reduction of grain size commences by inducing fractures at grain-to-grain contacts,

73 which then break the grains. The grain-size dependence of the modified Hall-Petch relation  
74 proposed by Sammis and Ben-Zion (2008) illustrates that smaller grains require higher differential  
75 stresses to fracture. Therefore, local stress concentrations during co-seismic events may be  
76 favourable sites for the generation of nanograins by fracturing micron to millimetre-sized grains.  
77 Because of the high stress concentrations during earthquakes, nanograins may be evidence for  
78 paleo-seismic events.

79         Less commonly considered effects of fault-rock deformation in carbonates are thermo-  
80 mechanical processes, e.g., decarbonation. Products of decarbonation processes are reported from  
81 experiments (Han, et al., 2007a and b; De Paola et al., 2011; Verberne et al., 2014; Spagnuolo et  
82 al., 2015; Delle Piane et al., 2017) and from natural faults (Vigano et al., 2011; Collettini et al.,  
83 2013). The experimental studies of Verberne et al. (2014), Spagnuolo et al. (2015) and Delle Piane  
84 et al. (2017) report the presence of amorphous carbon as a deformation product and Han, et al.,  
85 (2007a) mention the presence of portlandite ( $\text{Ca}(\text{OH})_2$ ). Some natural faults also exhibit amorphous  
86 glass coatings around euhedral calcite crystals in potential carbonate pseudotachylites (Vigano et  
87 al., 2011). In addition, deformation experiments on silicates (Yund et al., 1990; Di Toro et al.,  
88 2004; Pec et al., 2012; Hirose et al., 2012; Toy et al., 2015) produce non-crystalline or partly-  
89 amorphous silicate materials (Pec et al., 2012).

90         We investigated the surface and the underlying few tens of micrometres of two natural  
91 carbonate fault-mirror structures to test the hypothesis of a nanogranular surface coating. Our high-  
92 resolution electron microscopy and Raman spectroscopy analyses demonstrate that a thin coating  
93 of decarbonation products is able to produce a MMS without a nanogranular surface coating,  
94 similar to the results of Fondriest et al. (2013). In addition, we suggest that the amorphous products  
95 may facilitate and lubricate seismic slip (De Paola et al., 2011; Di Toro et al., 2011) while being

96 produced or possibly re-strengthen the fault by post-seismic recrystallisation of the decomposition  
97 products during the inter-seismic period.

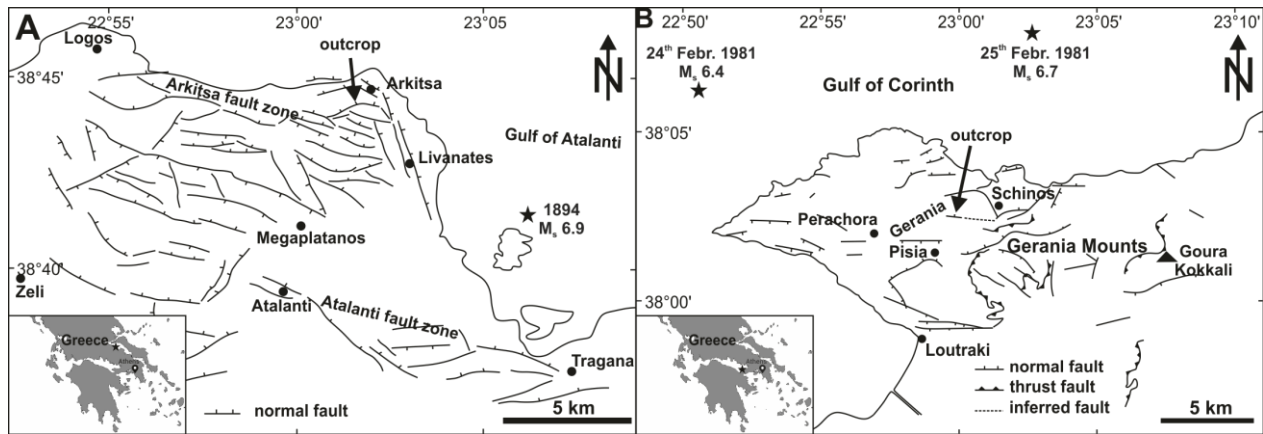
## 98 2. Geological Setting

### 99 2.1 Geology of the Arkitsa fault zone

100 The 700 m long Arkitsa fault surface exposure (Fig. 1A) belongs to the Kamena Vourla  
101 fault system; a northward-dipping, ESE-WNW striking, active normal fault zone of approximately  
102 50 km length along the southern coast of the Gulf of Evia. The left-stepping fault system consists  
103 of the Kamena Vourla, the Agios Konstantinos and the Arkitsa fault segments (Roberts and  
104 Jackson, 1991; Ganas et al., 1998). The study area (38°43'56.17"N, 23° 0'27.41"E) is situated  
105 within the Pelagonian zone of Central Greece. The Arkitsa fault planes juxtapose Late Triassic to  
106 Middle/Late Jurassic platform carbonates in the footwall with lower Pliocene-Pleistocene to  
107 Quaternary sediments in the hanging wall (Kokkalas et al., 2007). The Arkitsa fault scarp is an  
108 anthropogenically formed fault exposure first described by Jackson and McKenzie (1999).  
109 Quarrying for two decades removed a major part of the hanging-wall colluvium to reveal three  
110 large, smooth fault planes of up to 65 m height (Kokkalas et al., 2007).

111 Historical records of seismicity in the Gulf of Evia area indicate about 13 rupture events  
112 from 426 BC until the last major event in 1894 (Ganas et al., 1998; Ganas et al., 2006), with a  
113 magnitude  $M_s$  6.9 earthquake hosted within the Atalanti fault zone (Fig. 1A) (Ambraseys and  
114 Jackson, 1990). This number of events fit with the approximately 50 slip increments identified by  
115 Jackson and McKenzie (1999), where the authors calculated the slip increments from fault plane  
116 height and average earthquake slip distances. Evidence for Holocene seismic activity along the  
117 Arkitsa fault planes is recorded by the approximately 1 m unweathered fault scarp existing prior to  
118 quarrying (Jones et al., 2009).

119



120 *Figure 1: Tectonic maps of the study areas. Insets show the geographical location within central Greece. **A:** Location*  
121 *of the studied exposure of the Arkitsa fault zone (modified after Jones et al., 2009; earthquake data: NOAA). **B:***  
122 *Outcrop location of the Schinos fault segment north of Corinth (modified after Kaplanis et al., 2013 and Collier et al.,*  
123 *1998).*

124

## 125 2.2 Geology of the Schinos fault zone

126 The Schinos fault zone is located north-east of Corinth within the Gerania mountains, with

127 the studied exposure ( $38^{\circ} 2'14.40''N$ ,  $23^{\circ} 0'22.33''E$ ) belonging to the Gerania unit of the Internal

128 Hellenides (Kaplanis et al., 2013). Formation of the 120 m long fault exposure is from material

129 excavation for the adjacent dirt road. The fault plane is an east-west striking, northward-dipping

130 normal fault plane hosted within Upper Triassic limestones and dolomites. The stratigraphic

131 succession of the Gerania unit includes from bottom to top: Permian sedimentary and volcanic

132 rocks, Upper Triassic neritic limestones and dolomites, Lower Jurassic limestones, an Upper

133 Jurassic succession of rift-related sediments overlain by either Ammonitico Rosso (Bathonian or

134 Oxfordian age) or radiolarian sediments, and ophiolite-derived turbidites of the Beotian flysch

135 (Kaplanis et al., 2013).

136 Seismicity data record an earthquake sequence in the Corinth region with three main

137 shocks: 24<sup>th</sup> February 1981,  $M_s$  6.7; 25<sup>th</sup> February 1981,  $M_s$  6.4 and 4<sup>th</sup> March 1981,  $M_s$  6.4 (Collier

138 et al., 1998). Based on the epicentral location and a mismatch between the focal mechanism Collier

139 et al. (1998) interpreted that the 24<sup>th</sup> February event started with displacement on an offshore fault

140 and continued to rupture the Pisias fault, producing a surface break. In addition, Collier et al. (1998)

141 propose the same succession of events for the 25<sup>th</sup> February event, which caused a displacement  
142 on the Schinos fault, probably forming the most recent event on the fault plane that we study here.

### 143 3. Methods

144 We analysed drill-core samples from the Arkitsa and Schinos fault exposures using a range  
145 of micro-analytical techniques. In total, 40 drill cores, 2.54 cm in diameter, were collected from  
146 the Arkitsa fault zone and 18 were collected from the Schinos fault zone. The maximum retrievable  
147 drill-core lengths did not exceed 10 cm (Arkitsa) and 5 cm (Schinos) as the porosity of the fault  
148 rock increases and the material loses its cohesion with distance from the fault surfaces. Seven  
149 samples from the Arkitsa fault zone and four samples from the Schinos fault zone were chosen for  
150 further investigation based on the best-preserved mirror surfaces. Sample cores were sputter coated  
151 with an 8-nm thick layer of platinum/palladium for charge deduction.

152 From both faults, two representative samples were selected for electron-transparent foil  
153 preparation using a FEI Helios Nanolab G3 Dualbeam focused ion beam scanning electron  
154 microscope (FIB-SEM). Prior to ion beam deposition of the main platinum strip, a 200-nm layer  
155 of platinum was deposited using the electron beam (2kV, 0.4 nA) to prevent surface amorphisation  
156 by the ion beam as a preparation artefact. Transmission electron microscope (TEM) investigations  
157 of the foils were carried out with a FEI Talos F200X. TEM images were acquired with 200 kV  
158 acceleration voltage and 5 nA or 10 nA beam current depending on final sample thickness. The  
159 FEI Talos SuperEDX detector was used for chemical element mapping via energy dispersive X-  
160 rays (EDX) in scanning transmission electron microscope (STEM) mode. EDX maps were  
161 quantified post-acquisition with Bruker Esprit 1.9 software using the Cliff-Lorimer approximation  
162 and ImageJ.

163 Electron energy loss spectroscopy (EELS) data were acquired with a Zeiss Libra 200FE in  
164 TEM mode at 200 kV with an Omega in-column energy filter. The energy resolution of EELS

165 analyses was 0.7 eV, measured at the full width half maximum (FWHM) of the zero-loss peak.  
166 Energy loss spectra were obtained at 250,000x magnification with a 100  $\mu\text{m}$  filter-entrance aperture  
167 giving an effective aperture of about 40 nm on the sample. The convergence angle of EELS was  
168 about 0.1 milliradian (mrad), and the acceptance angle was defined by the 60  $\mu\text{m}$  diameter of the  
169 objective aperture giving a collection angle of 11.6 mrad. Measurement times of EEL spectra were  
170 set to 1–5 s, with 5–10 frames/spectrum, on a slow-scan CCD (model UltraScan 4000), with  
171 binning of 4x4 pixels (giving 1024 pixels), and an energy spread of 0.08 eV/pixel, resulting in an  
172 energy range of 82 eV in the acquired spectra. Spectra were deconvolved with the zero-loss peak  
173 to remove the effect of plural scattering and background-subtracted assuming a power law function  
174 with Gatan's Digital Micrograph software.

175 Raman spectroscopy was carried out with a WiTec ALPHA300 R confocal microscope to  
176 identify and characterize possible reaction or decomposition products on the fault surfaces. We  
177 used a laser with a wavelength of 532 nm and a spectral grating of 600 grooves/cm. Data  
178 acquisition, data post-processing and peak fitting were performed with WiTec ProjectFour 4.1  
179 software and Fityk 0.98 (Wojdyr, 2010) peak fitting software. In accordance with (Ferrari and  
180 Robertson, 2000) we determined the intensity ratios of the D to G peak,  $\frac{I(D)}{I(G)}$ , to obtain information  
181 about the degree of crystallinity within carbon phases. We used the approach by Ferrari and  
182 Robertson (2000) based on first principle considerations to determine the peak height ratios  
183 because the information about the less disordered aromatic rings and the clustering of the sp<sup>2</sup> phase  
184 is contained in the intensity maximum of the D peak and not in the width. Surface roughness  
185 measurements were executed within a Nanoscope atomic force microscope (AFM). Calculation of  
186 reaction enthalpy values were executed with SUPCRT92 (Johnson et al., 1992) at standard  
187 conditions (1 bar, 298 K).



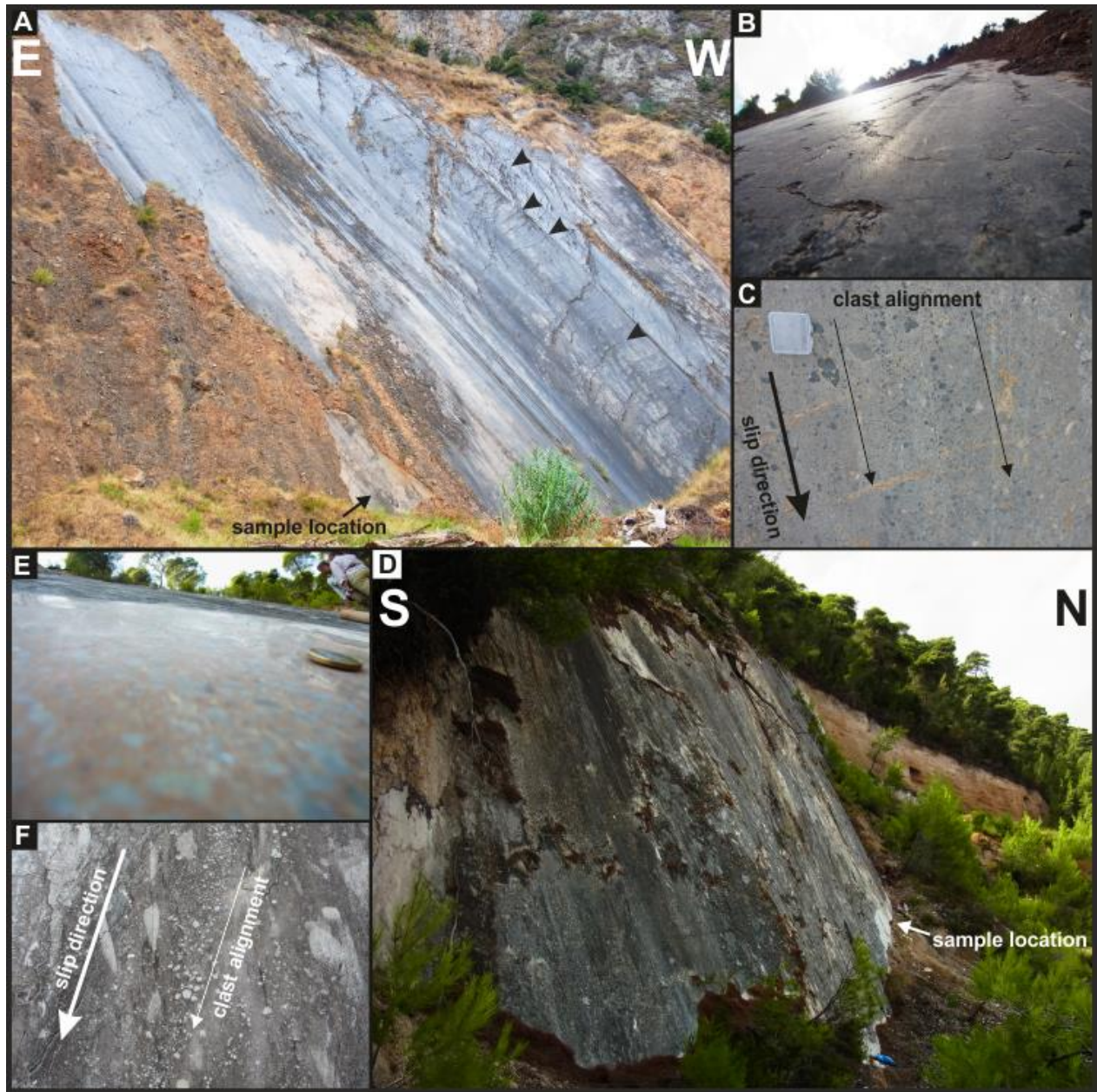
188 4. Results

189 4.1 Field results

190 The exposed surface of the Arkitsa fault (Fig. 2A) exhibits a range of slip-related structures.

191 At outcrop scale, the fault plane exposure steps over, which suggests the presence of several slip  
192 planes inside the fault damage zone. The fault surface contains pronounced slip grooves parallel to  
193 the direction of oblique slip. Fractures with a spacing on the order of one metre are oriented  
194 approximately perpendicular to the grooves (Fig. 2A). Parts of the slip plane are covered with  
195 residual reddish-brown hanging-wall breccia, incorporating fragments of dark host-rock carbonate  
196 up to several decimetres in size (Fig. 2A). The most prominent feature is the low roughness of the  
197 fault surface, which enables the reflection of sunlight (Fig. 2B). The fault rock is a matrix-  
198 supported, greyish cataclasite with dark carbonate clasts up to several centimetres in size (Fig. 2C).  
199 Slip-parallel alignment of clasts on the fault surfaces can be traced over several meters (Fig. 2C).

200 The Schinos fault surface (Fig. 2D) exhibits a range of structures formed during slip. We  
201 observe stepovers, indicating the presence of several fault planes inside the fault damage zone. The  
202 fault surface is extremely smooth with areas not only reflecting light but also mirroring the  
203 surrounding vegetation (Fig. 2E). Fragments of grey carbonate host rock are incorporated into the  
204 red-orange footwall cataclasite. The clasts are strongly aligned in the slip direction and usually do  
205 not exceed 1–2 cm in size (Fig. 2F). The fault surface contains pronounced slip grooves and a wavy  
206 surface morphology (Fig. 2D). The mirror surfaces of both faults are hosted inside a zone of high  
207 competence with average thicknesses of 10 cm and 5 cm for the Arkitsa and Schinos faults,  
208 respectively.



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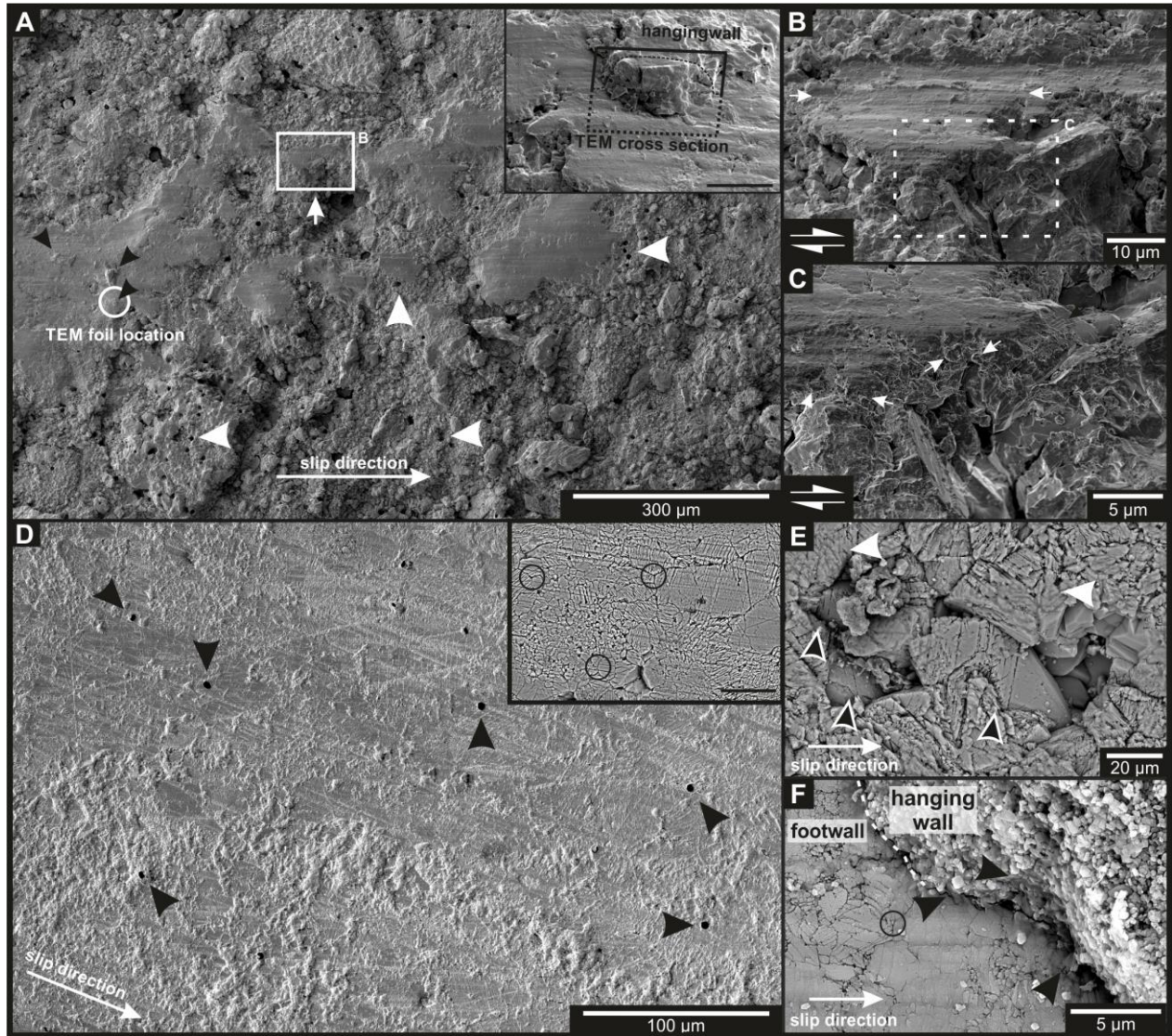
210 *Figure 2: Representative field structures from Arkitsa (A – C) and Schinos (D – F) fault exposures. **A:** Overview of*  
 211 *Arkitsa fault exposure showing one of the three main slip planes with reddish-brown hanging-wall breccia cover on*  
 212 *part of the slip surface. Black arrows indicate fractures perpendicular to slip direction. Person for scale, bottom right*  
 213 *corner. **B:** Fault surface reflecting sunlight. **C:** Alignment of host rock fragments parallel to slip direction. Box approx.*  
 214 *5 x 4 cm. **D:** Schinos fault surface exposure shows grey weathering of the slip plane. The slip plane is curved with the*  
 215 *salient pointing out of the figure. Fault scarp to the north is now covered by alluvium. **E:** Image demonstrating fault*  
 216 *surface reflectivity. Reflection of vegetation along the top edge and above the one-euro coin. **F:** Grey, weathered fault*  
 217 *plane showing a strong alignment of light grey host-rock clasts parallel to slip direction (one-euro coin for scale,*  
 218 *bottom edge).*

219 4.2 Slip surface microstructures

220 SEM analyses of the Arkitsa and Schinos slip surfaces reveal a low surface roughness at  
221 the microscale. The surfaces are well polished and, in some places, preserve parts of the hanging  
222 wall (Fig. 3A and F). Microgrooves are aligned parallel to the slip direction (Fig. 3B) and holes  
223 with a diameter of 6–8  $\mu\text{m}$  pierce the slip surfaces (Fig. 3A and D). The average surface roughness  
224 measured by AFM is 63.5 nm for the Arkitsa fault surface and 32.3 nm for the Schinos fault,  
225 respectively (see SF1 in the supplementary material). Localized erosion of the Arkitsa slip surface  
226 exposes the underlying fault rock, with a grain size of 2–5  $\mu\text{m}$  (Fig. 3A, B and C). The first 10–20  
227  $\mu\text{m}$  of the fault rock below the Arkitsa slip surface consist of a fine-grained deformation product,  
228 which is situated on top of a comparatively coarse-grained, less deformed material (Fig. 3B and  
229 C).

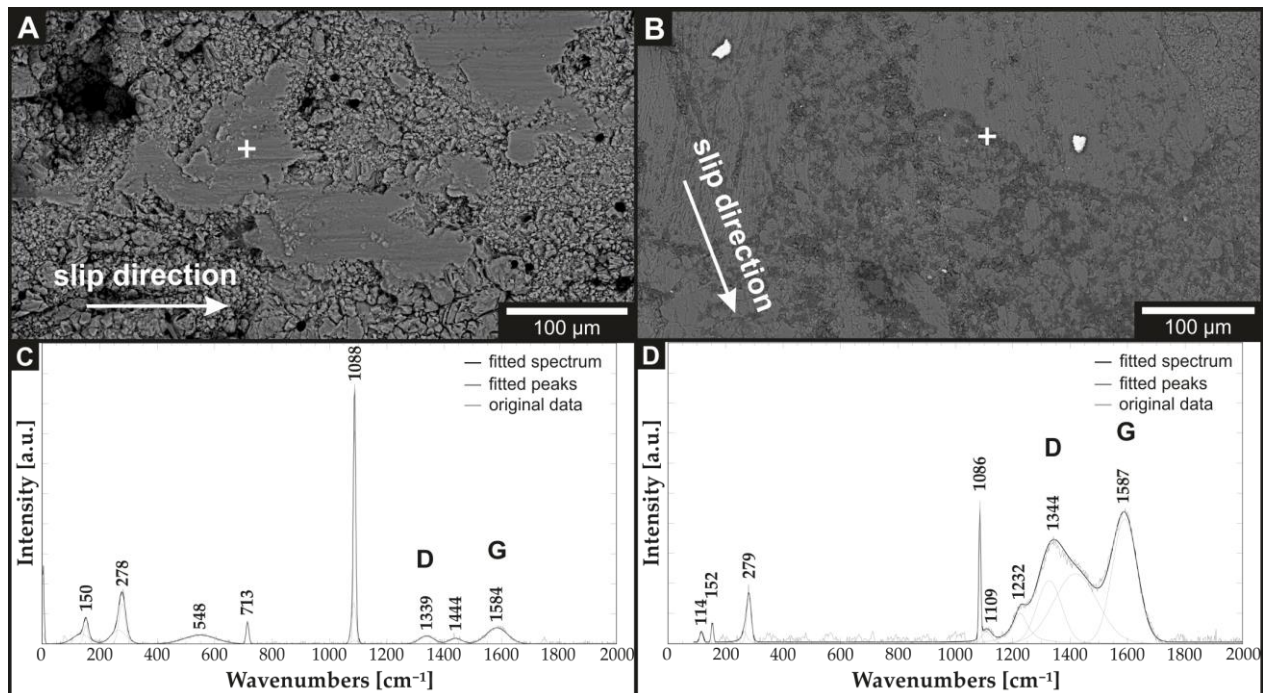
230 In contrast, the Schinos fault surface consists of large, truncated grains with boundaries that  
231 meet in triple junctions (Fig. 3D and inset). A fragmented layer lies on top of undeformed calcite  
232 crystals (Fig. 3E) and the damage extends about 10–20  $\mu\text{m}$  into the fault rock. Here also an  
233 amorphous material is present on the slip plane (Fig. 3E and F). On both faults, Arkitsa and Schinos,  
234 the amorphous material has a low contrast in backscattered-electron images (Fig 4A and B). The  
235 phase also coats grain edges and reduces the slip surface roughness (Fig. 3C, E and F). In addition,  
236 the amorphous material infiltrates interstitial regions of the hanging-wall breccia and holds residual  
237 pieces of hanging wall in place (Fig. 3F).





238  
 239 **A:** Secondary electron overview image of the Arkitsa slip surface. Smooth slip patches with microgrooves parallel to slip  
 240 direction. Uniform holes (white arrows) penetrate the slip surface. Residue of hanging wall on top of the slip surface  
 241 patch (black arrows, white circle) with TEM sample location. Inset: Detailed view of TEM foil location from figure  
 242 5A, white circle. Scale bar = 5  $\mu\text{m}$ . **B:** Close-up of slip surface from A. Fine-grained slip surface material situated on  
 243 top of coarse fault rock grains. Microscope stage tilted to 52°. **C:** Close-up of section from B. Amorphous material  
 244 covers roundish grains (black arrows). **D:** Secondary electron image of Schinos slip surface with microgrooves  
 245 parallel to slip direction. Uniform holes penetrate the principal slip surface (black arrows). Inset: Backscatter electron  
 246 image illustrating triple junction grain boundary contacts. Scale bar = 50  $\mu\text{m}$ . **E:** Backscatter electron image top view  
 247 onto the slip surface. Cracks in idiomorphic calcite crystals are visible a few micrometres into the crystal (black-white  
 248 arrows). Amorphous material with low backscatter contrast covers parts of the slip surface and reduces surface  
 249 roughness (white, dashed lasso) **F:** Backscatter electron image of the contact between footwall and hanging wall  
 250 (dashed line). Black arrows indicate the locations of an amorphous material with lower backscatter contrast.  
 251

252



253

254 *Figure 4: Backscatter electron images with locations of Raman spectroscopy measurements for Arkitsa (A and C)*  
 255 *and Schinos (B and D) slip surfaces. **A and B:** Backscatter electron images of the fault surfaces. A phase with low*  
 256 *backscatter contrast (darker grey) is visible in B. **C and D:** Raman spectra showing the D and G peak position of a*  
 257 *disordered carbon phase.*

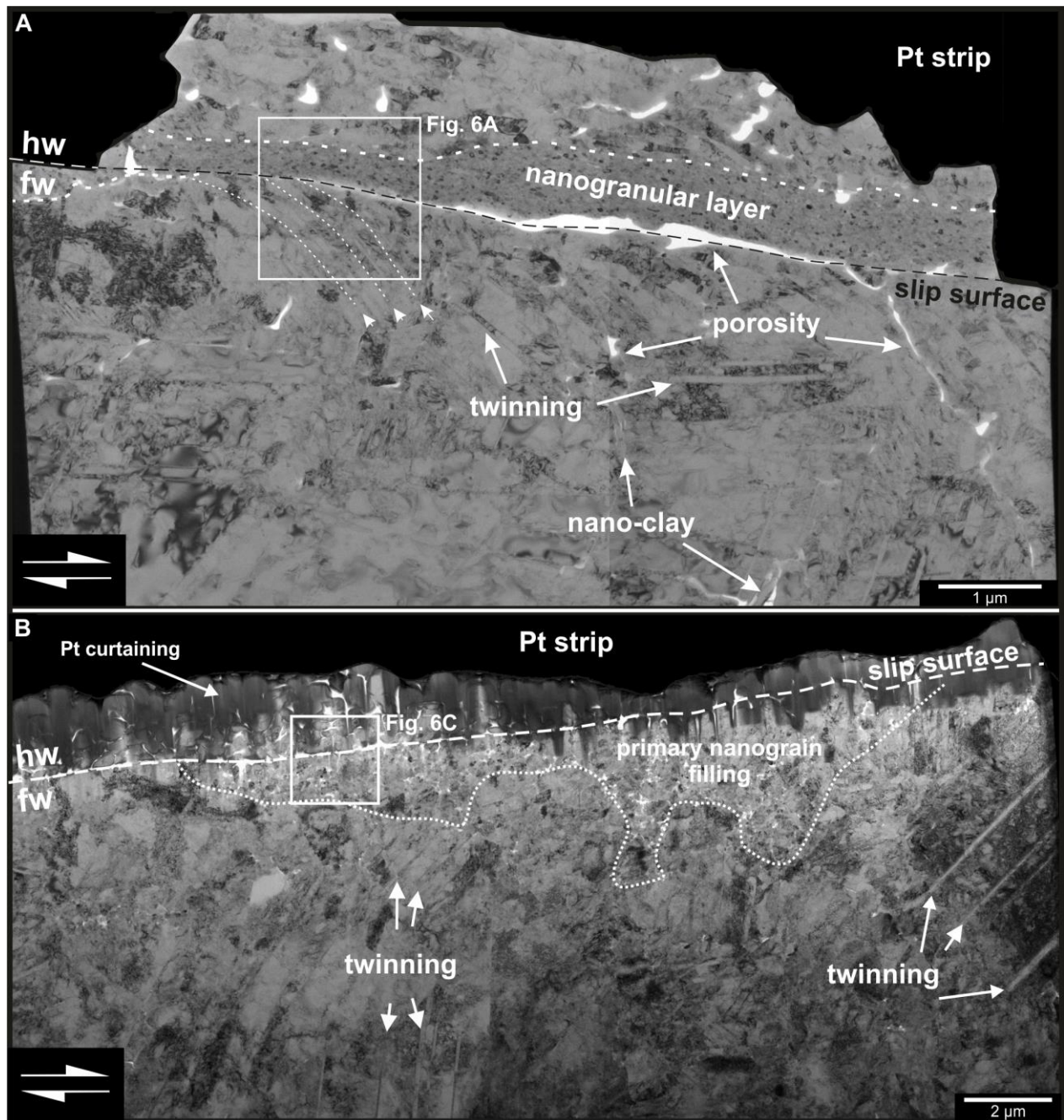
258 Figure 4 presents Raman spectra from the amorphous material. Two broad peaks between  
 259 1200–1700 cm<sup>-1</sup> indicate the presence of a disordered carbon phase, whilst a peak at 1086 cm<sup>-1</sup>  
 260 demonstrates the presence of crystalline calcite. The fitted spectrum from the Arkitsa fault plane  
 261 gives a  $\frac{I(D)}{I(G)}$  ratio of 0.543 at a D-peak position of 1339 cm<sup>-1</sup> and a G-peak position of 1584 cm<sup>-1</sup>.  
 262 The fitted spectrum from the Schinos fault plane gives a  $\frac{I(D)}{I(G)}$  ratio of 0.789 at a D-peak position of  
 263 1344 cm<sup>-1</sup> and a G peak position of 1587 cm<sup>-1</sup>.

264 4.3 Nanostructures

265 TEM analyses on FIB-SEM foils of both fault-surface exposures reveal a thin surface coat  
 266 (e.g. Fig. 8A and F, but also Fig. 6A–D). The uppermost layer of both fault-rock exposures is  
 267 defined by a principal slip surface with a thin, non-crystalline coating between hanging wall and  
 268 footwall (Fig. 5A and B, Fig. 6A–D). The coat exhibits homogenous diffraction contrast, shows no

269 lattice fringes (Fig. 6A to D, 8A and F) and EDX analyses indicate the presence of carbon (C), iron  
270 (Fe), silicon (Si) and aluminium (Al) (Fig. 7, Si content of Arkitsa ~ 40 area % estimated with  
271 ImageJ). A similar content of impurity elements has been mentioned by Pluymakers and Røyne  
272 (2017), Colletini et al. (2013) and Goldberg et al. (2016). The coat connects hanging-wall breccia  
273 with the footwall fault surface (Fig. 6B and D) but also extends into the hanging-wall breccia (Fig.  
274 6A and C). EELS measurements of the amorphous material are given in the inset of Fig. 6D. The  
275 onset of the  $\pi^*$  peak represents the measured carbon K-edge at 284 eV, which is followed by a  
276 'whaleback-shaped'  $\sigma^*$  region. In addition, the measurement of the Schinos sample exhibits an  
277 additional peak at 286.5 eV, possibly indicating C-O bonding.



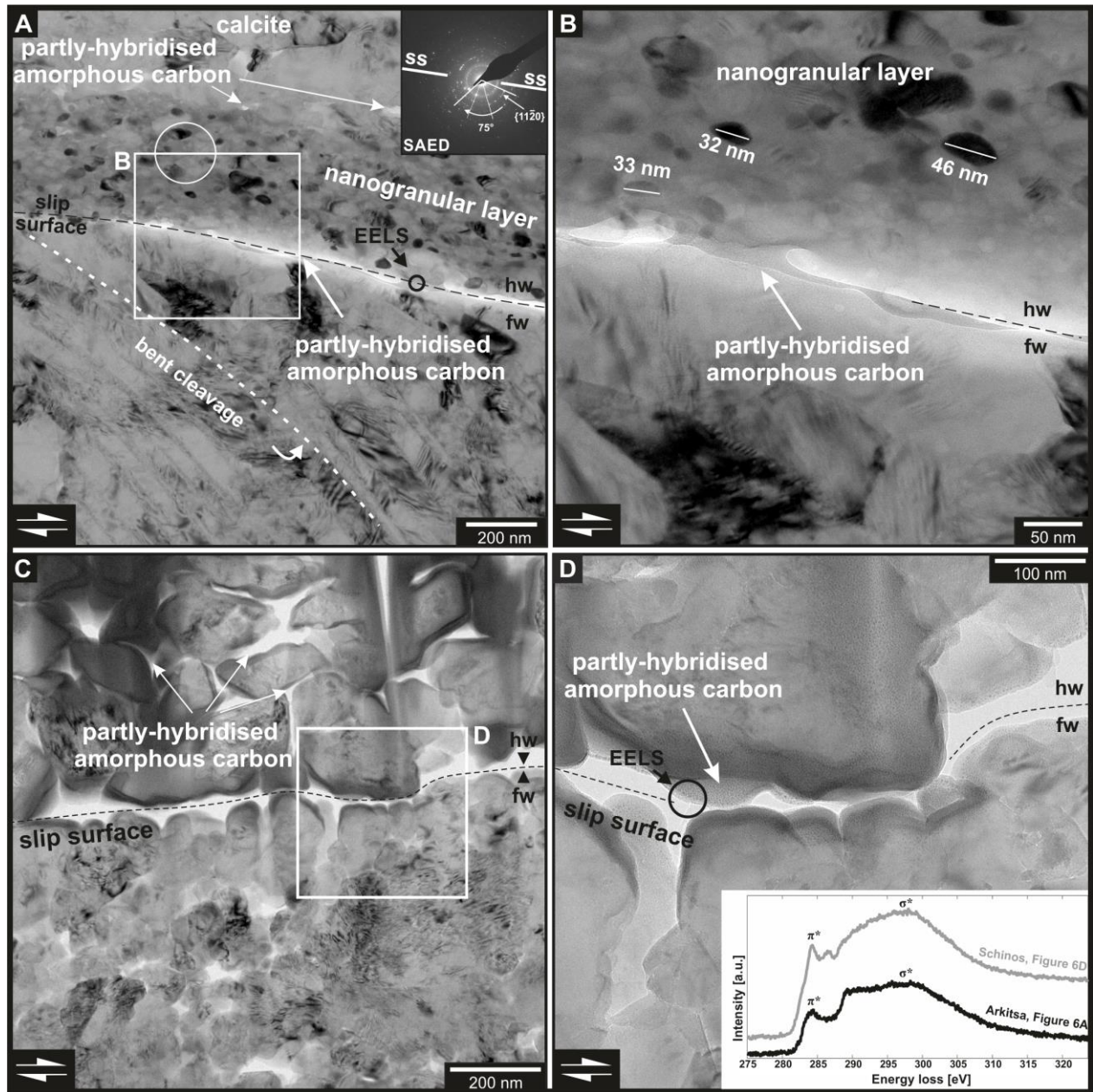


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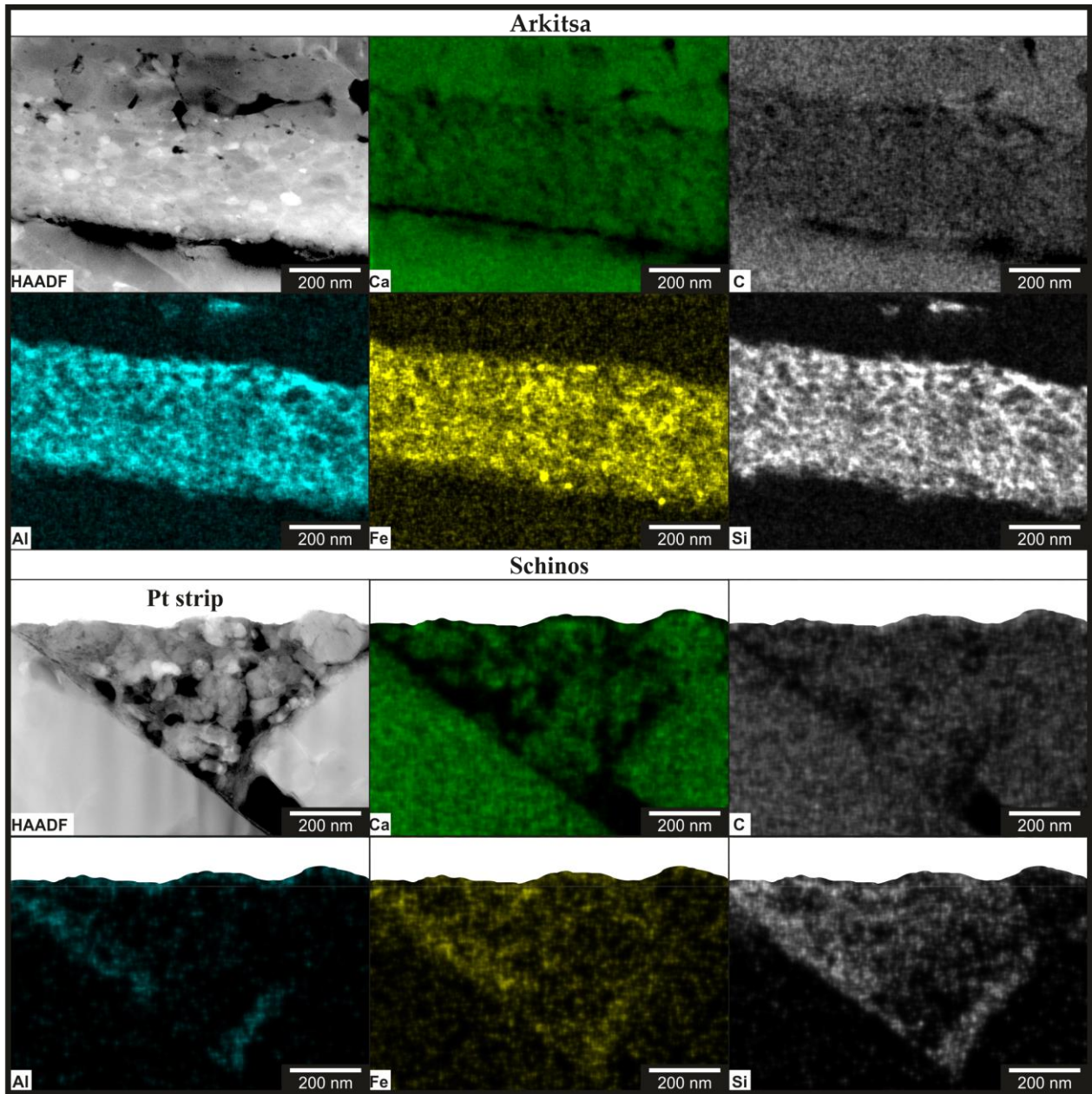
279 *Figure 5: Overview of TEM cross sections into the principal slip surfaces of Arkitsa and Schinos, including preserved*  
 280 *hanging-wall residue. **A:** Bright-field TEM image overview of representative Arkitsa nanostructures. The hanging-*  
 281 *wall breccia consists of small, elongated calcite fragments held in place by non-crystalline carbon. The nanogranular*  
 282 *layer is part of the hanging-wall breccia and lies on top of deformed calcite crystals. Fractures (dashed lines) bend*  
 283 *down from the principal slip surface into the deformed calcite crystals. Fractures and holes form porosity inside the*  
 284 *mirror slip volume and are filled either with non-crystalline carbon and/or nano-clay (see arrows with labelling). **B:***  
 285 *Bright-field TEM image with representative Schinos nanostructural hanging-wall to footwall relation. The hanging*  
 286 *wall consists of small grains of several hundred nanometres. Individual grains are surrounded by non-crystalline*  
 287 *carbon. The footwall hosts a discontinuous, primary nanograin filling. The grain size increases abruptly with distance*  
 288 *from the slip surface to grains of several micrometres in size. Larger grains contain twins. hw = hanging wall, fw =*  
 289 *footwall.*

290 We observe a direct relation between the spatial distribution of amorphous material and  
291 calcite nanograins. For the Arkitsa fault exposure, a nanogranular layer is present with rounded,  
292 elliptical calcite nanograins of ~50 nm size between the footwall and the hanging-wall breccia (Fig.  
293 5A, 6A, B and 7). Selected area electron diffraction (SAED) patterns of the Arkitsa nanogranular  
294 layer in Figure 5A demonstrate the polycrystalline nature of the material with clustered diffraction  
295 spots forming two arcs subtending an angle of about 75° (inset Fig. 6A). Figure 8A reveals that the  
296 thickness of the coat varies along the slip surface and locally incorporates calcite grains from the  
297 underlying nanostructure. We find that calcite nanograins are not restricted to the principal slip  
298 surface but also occur away from the displacement zone inside the fault rock (Fig. 8B). The  
299 nanograins do not contain any dislocations (Fig. 6B and 8B) and are not always in grain-to-grain  
300 contact in the plane of the section. Calcite nanograins of the Schinos exposure are mostly located  
301 between large grains along the principal slip surface (Fig. 8C and D). The gaps between large grains  
302 also contain an amorphous material that increases the overall smoothness of the fault surface. Inside  
303 the amorphous material, most nanograins have elongate shapes with no dislocation substructure  
304 (Fig. 8D). Nano-clay minerals often wrap around the calcite nanograins (Fig. 8D).





305  
 306 *Figure 6: TEM cross sections illustrating the relationship between hanging-wall (hw) breccia and footwall (fw) contact*  
 307 *for Arkitsa (A – B) and Schinos (C – D). **A:** TEM bright field close-up from Fig. 5A. Non-crystalline carbon establishes*  
 308 *the contact between hanging wall and footwall. The footwall consists of large, stressed calcite crystals with bent*  
 309 *fractures. Nanogranular layer with non-crystalline carbon and calcite nanograins. White circle indicates SAED*  
 310 *location (circle diameter = aperture diameter); black circle depicts EELS measurement (circle diameter = aperture*  
 311 *diameter). Inset: SAED showing two sets of crystal orientation. ss = slip surface. **B:** TEM bright field close-up from*  
 312 *A. Non-crystalline carbon connects hanging wall and footwall. The nanogranular layer produces Moiré fringes*  
 313 *indicative of overlapping crystal lattices. Nanograins are not always in grain-to-grain contact. **C:** TEM bright field*  
 314 *close-up image from Fig. 5B. The grain size of the hanging wall is 200 nm and larger. The grain size of the footwall*  
 315 *directly at slip surface is about 100 nm. **D:** TEM bright field close-up image from C. The hanging wall and footwall*  
 316 *are connected via non-crystalline carbon. Black circle depicts EELS measurement (circle diameter = aperture*  
 317 *diameter). Inset shows EELS spectra from black circles in A and D.*

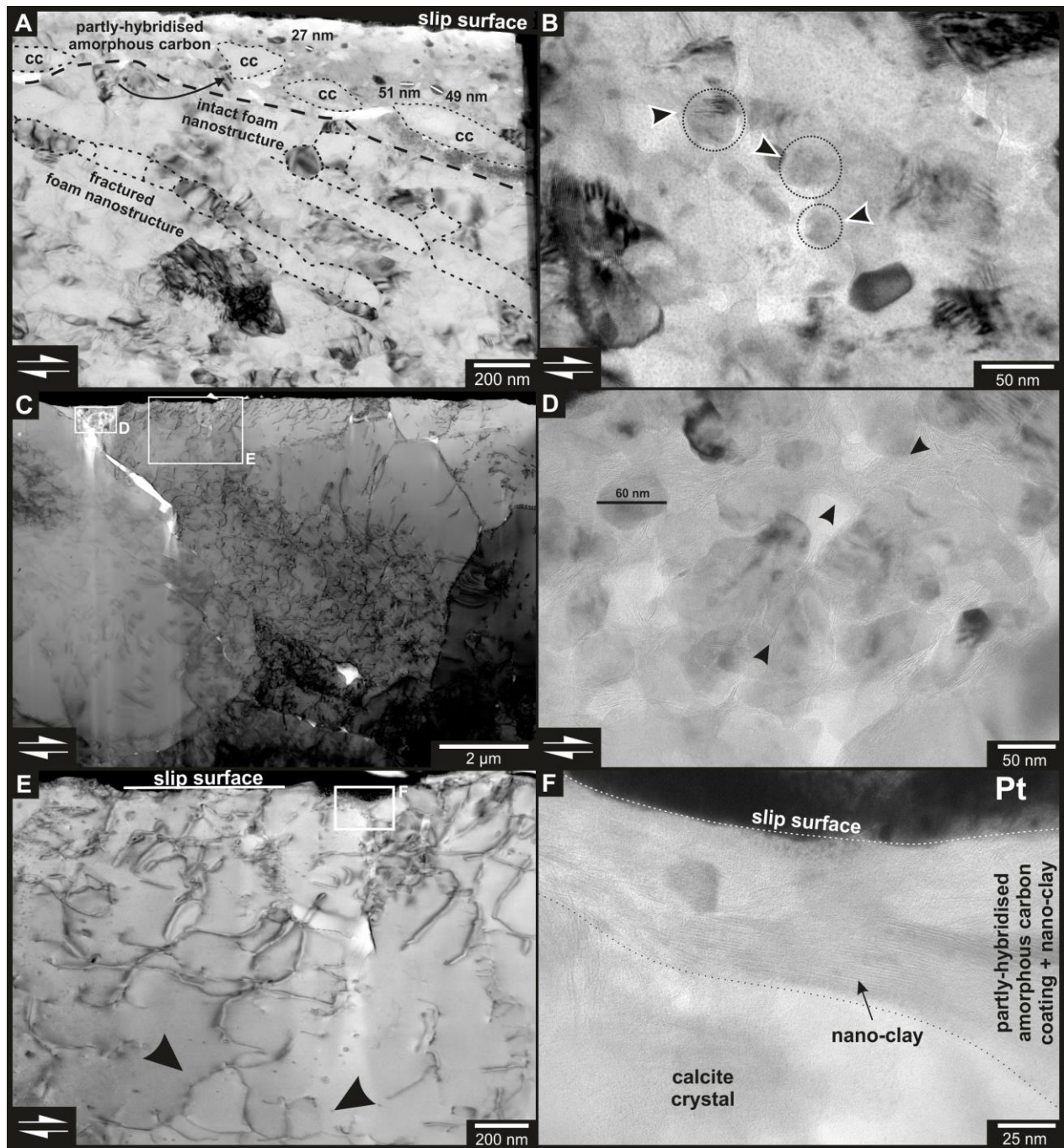


318  
 319 *Figure 7: TEM-EDX maps of calcite nanogranular layer from Fig. 5A (Arkitsa) and surface filling Fig. 8C (Schinos)*  
 320 *visualizing the element distribution. Overlap of calcium (Ca) and carbon (C) distribution confirms that the grains*  
 321 *inside are calcite. Aluminium (Al), iron (Fe) and silicon (Si) are concentrated in the interstitial region around the*  
 322 *calcite grains.*

323 Further TEM investigation of the slip surfaces reveals a variety of slip-derived deformation  
 324 nanostructures. Neither fault exposure exhibits a gradual decrease in grain size towards the slip  
 325 surface but both show an abrupt change, instead. Adjacent to the principal slip surface, we observe  
 326 twinned calcite crystals about 2–5  $\mu\text{m}$  in size (Fig. 5A and B). Dislocations are homogeneously

327 distributed in the larger crystals from the Arkitsa exposure (Fig. 5A). For the Schinos case, the  
328 dislocation distribution appears to be less ordered, with dislocations concentrated towards grain  
329 contacts in slip direction (Fig. 8C). Bent cleavage fractures dip down from the Arkitsa slip surface  
330 and exhibit minor displacements together with slight increases in dislocation density along fracture  
331 planes (Fig. 5A and 6A). Along the slip surface, the nanostructure of the Arkitsa fault varies in  
332 intensity and develop a foam nanostructure (Fig. 8A). The layer thickness of the foam nanostructure  
333 is about 1  $\mu\text{m}$  and the grain size ranges from 100–200 nm with the grain boundaries exhibiting  
334 triple junctions at  $\sim 120^\circ$  angles (Fig. 8A). This nanostructure is not observed in the samples  
335 analysed from the Schinos exposure, which instead contain dislocation arrays (Fig. 8C and E). In  
336 general, the foam nanostructure at Arkitsa appears to be less developed adjacent to the slip surface,  
337 especially where the fracture density is highest.





338

339 *Figure 8: Nanostructures of fault surface coating and nanograin abundance not associated with the principal slip*  
 340 *surface. **A:** Bright-field TEM image from the Arkitsa exposure. Amorphous slip surface coating with calcite nanograins*  
 341 *and crystal fragments from underlying foam nanostructure (black arrow). A sharp contact separates the underlying*  
 342 *nanostructure from the slip surface coating. Nanostructure with former fracture planes (dashed lines), which originate*  
 343 *from the principal slip surface, and overprint the existing nanostructure. **B:** Calcite nanograins away from the*  
 344 *principal slip surface hosted inside non-crystalline carbon. Tip direction of black-white arrows indicate lattice fringe*  
 345 *orientation of new grains. **C:** Bright-field STEM image overview. Large grains (5 μm) with a heterogeneous dislocation*  
 346 *density distribution. **D:** Calcite nanograins in non-crystalline carbon as a filling between two grain boundaries directly*  
 347 *at the principal slip surface. Clay minerals wrap around the nanograins (black arrows). **E:** Bright-field STEM close-*  
 348 *up from C showing the dislocation structure directly at the principal slip surface. Black arrows point to frozen*

349 *dislocation loops. **F:** Bright-field TEM image from the Schinos exposure. Amorphous slip surface coating mixed with*  
350 *nano-clay on top of calcite crystal. cc = calcite.*

## 351 5 Discussion

### 352 5.1 Processes revealed by nano/microstructures

#### 353 5.1.1 Decarbonation and carbon reduction

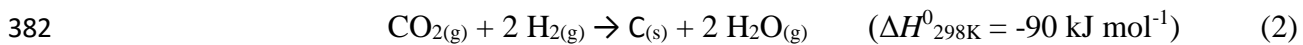
354 Nanostructural investigation of the Arkitsa and Schinos fault exposures reveal the  
355 importance of calcite decarbonation products for the formation of a smooth fault surface. TEM  
356 images reveal a coat of an amorphous material. We suggest that the amorphous material formed  
357 following calcite decarbonation, as a result of shear heating at asperity contacts or alternatively,  
358 the 'severe' introduction of dislocations during fault slip. In our natural samples, holes piercing the  
359 principal slip surface (Fig. 3A and D) indicate CO<sub>2</sub> degassing as a product of the decarbonation  
360 reaction



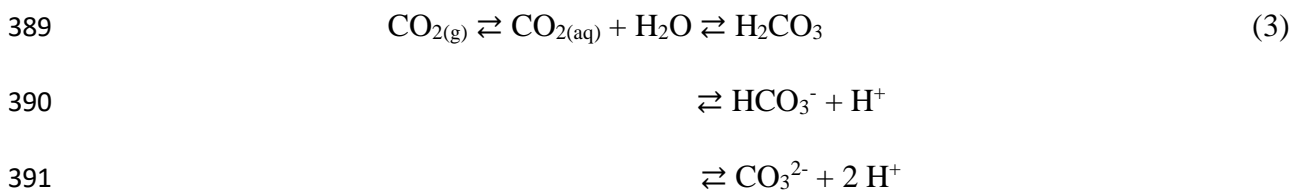
362 reported to occur due to thermal decomposition of calcite starting at 600 °C (Rodriguez-Navarro  
363 et al., 2009). However, experiments by Martinelli and Plescia (2004) recorded CO<sub>2</sub> emission during  
364 ball milling experiments of calcite without increasing the macroscopic temperature. In addition to  
365 the CO<sub>2</sub> emissions, the authors detected the presence of amorphous lime (CaO) in their milling  
366 product as a result of a process termed 'mechanical liming'. Subsequent heating of the milling  
367 product resulted in a decrease of amorphous lime and the formation of calcite. The experiments of  
368 Martinelli and Plescia (2004) demonstrate that reaction (1) can also take place just by deforming  
369 calcite without reaching macroscopic temperatures > 600 °C, although the temperature at asperity  
370 contacts could be higher. Mechanical liming may be important for wet deformation conditions  
371 where fluids are present and will buffer the temperature increase until all fluid of the system  
372 transforms into vapor. The buffering of temperature during deformation is in agreement with  
373 observations by (Demurtas et al., 2019), where temperatures of < 200 °C were recorded during

374 water-damped experiments at seismic velocities. In contrast, the same study records temperatures  
375 of > 600 °C for experiments at room humidity. We suggest that decarbonation can occur at  
376 macroscopic temperatures < 600 °C by mechanical liming, even under wet conditions, leading to  
377 the formation of lime and CO<sub>2</sub>.

378 The acquired electron energy loss and Raman spectra indicate the presence of a carbon  
379 phase. Hence, the presence of this phase implies the occurrence of carbon-reducing processes  
380 because reaction (1) does not produce elemental carbon. Precipitation of carbon in deformation  
381 experiments of Oohashi et al. (2014) under a hydrogen atmosphere suggests that the reaction



383 can reduce CO<sub>2</sub> to form carbon. Similarly, Spagnuolo et al. (2015) detected H<sub>2(g)</sub> and CO<sub>2(g)</sub>  
384 degassing during their experiments. Although reaction (2) is expressed as a chemical vapour  
385 deposition reaction, some of the hydrogen produced can go into solution to reduce dissolved CO<sub>2</sub>.  
386 This process is known to occur in hydrothermal systems where dissolved hydrogen can interact  
387 with CO<sub>2(aq)</sub> to form reduced carbon phases (e.g., Milesi et al., 2015). Under wet conditions, the  
388 CO<sub>2</sub> from reaction (1) goes into solution starting the hydrolysis reaction



392 to give carbonic acid, hydrogen carbonate, carbonate ions, and protons (Ruiz-Agudo et al., 2013).  
393 The mole fraction solubility ( $X_1$ ) of CO<sub>2</sub> in water at room temperature is, with  $X_1 = 6.15 \times 10^{-4}$ , one  
394 order of magnitude larger than H<sub>2</sub> with  $X_1 = 1.411 \times 10^{-5}$  (Gevantman, 2000). It is, therefore, more  
395 likely that a larger amount of CO<sub>2</sub> will dissolve in water and form carbonic acid, being the more  
396 probable proton donor, for reducing carbon from the residual CO<sub>2</sub>. As an alternative, CaO surfaces

397 can act as potential catalytic sites and promote the reaction with H<sub>2</sub>O at kinks, corners and/or steps.  
398 These surfaces can split H<sub>2</sub>O leading to one hydroxyl ion (OH<sup>-</sup>) and one proton (H<sup>+</sup>) which in turn  
399 may also be able to reduce carbon from CO<sub>2</sub> (Kudłacz and Rodriguez-Navarro, 2014). The main  
400 requirement for this reaction to occur is the presence of crystalline CaO. This would either require  
401 prior crystallisation of the suspected amorphous lime from reaction (1) or the lime resulting from  
402 reaction (1) was already crystalline. Calcium ions (Ca<sup>2+</sup>) decrease the pH of the water facilitating  
403 CO<sub>2</sub> solubility. Earth-alkali metal reactions with water are commonly known to produce hydrogen  
404 by reaction of the metal with water to form H<sub>2</sub> and the corresponding metal hydroxide. As the  
405 reduction of CO<sub>2</sub> by hydrogen does not require the participation of an additional reducing agent, it  
406 is likely to occur also under closed-system experimental conditions. With a variety of possible  
407 production reactions, it remains challenging to identify the main reaction responsible for the source  
408 of hydrogen for CO<sub>2</sub> reduction. However, a combination of several processes is likely, with the  
409 simplest being the hydrolysis of CO<sub>2</sub> in H<sub>2</sub>O. The CaO or Ca<sup>2+</sup> interaction with water can increase  
410 the pH up to 12.4, as shown in experiments by Ruiz-Agudo et al. (2013), which increases the  
411 solubility of CO<sub>2</sub> in water and would facilitate reaction (3). In general, the result would be the  
412 precipitation of carbon from the fluid, which would not be restricted to the principal slip surface  
413 but would also occur within existing porosity away from the slip surface.

#### 414 5.1.2 Carbon-bond sp<sup>2</sup> hybridisation

415 The obtained  $\frac{I(D)}{I(G)}$  ratios from Raman spectroscopy indicate that the slip surfaces are coated  
416 with nanocrystalline graphite according to the interpretation of the Raman modes by Ferrari and  
417 Robertson (2000). With the obtained  $\frac{I(D)}{I(G)}$  ratios we can calculate the size of the nanocrystallites  
418 according to the equation from Pimenta et al. (2007)

$$419 \quad L_a = \frac{560}{E_{laser}^4} \left( \frac{I(D)}{I(G)} \right)^{-1} \quad (4),$$

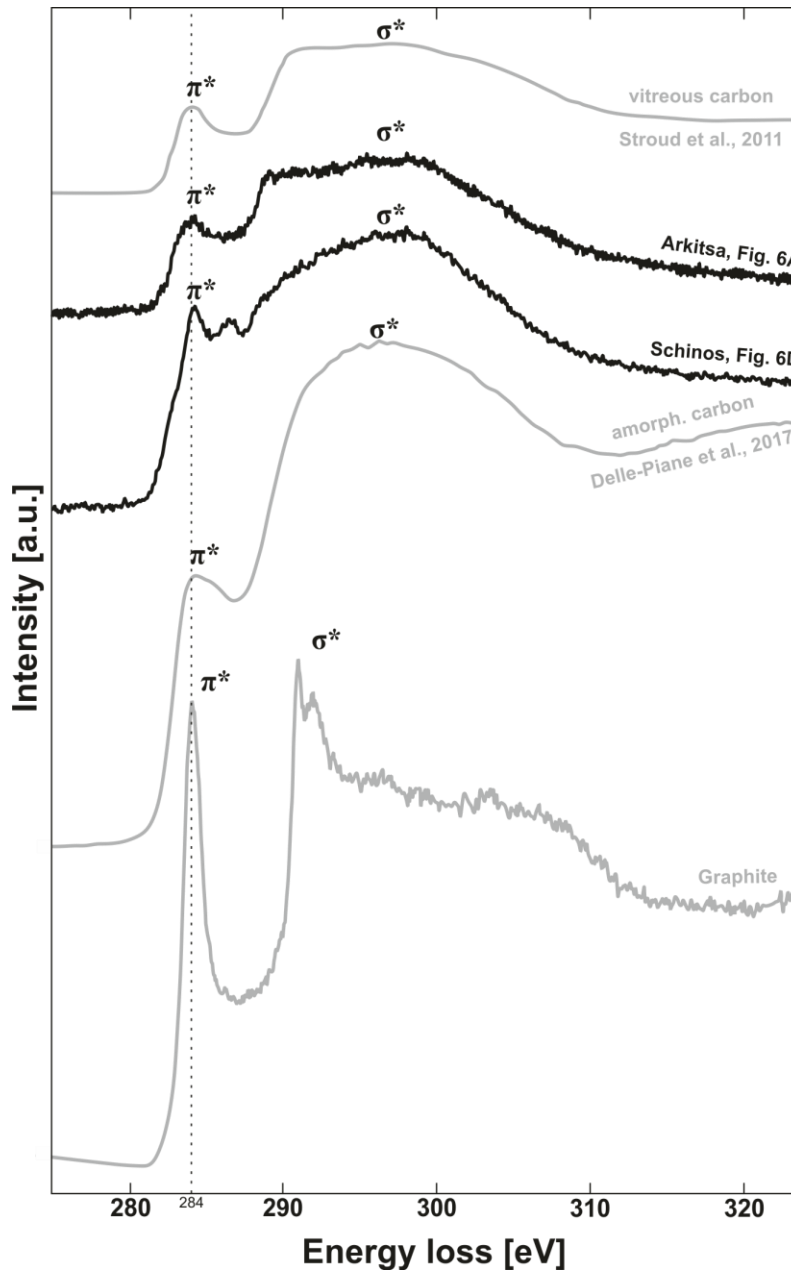
420 where  $L_a$  is the cluster/crystallite size in nm,  $E_{laser}$  is the energy (wavelength) of the laser with  
421 2.33 eV (532 nm), and  $\frac{I(D)}{I(G)}$  the intensity ratio of D and G peaks. Equation (4) is based on results  
422 from Pimenta et al. (2007), which revealed an inversely proportional relationship between the  
423 fourth power of  $E_{laser}$  and the  $\frac{I(D)}{I(G)}$  ratio. For the Arkitsa exposure we calculate a size of  $L_a = 35$   
424 nm ( $\frac{I(D)}{I(G)} = 0.543$ ) and for the Schinos exposure  $L_a = 24$  nm ( $\frac{I(D)}{I(G)} = 0.789$ ). While Pimenta et al.  
425 (2007) calculate with  $L_a$  the graphite nanocrystallite size, Ferrari and Robertson (2000) mention  
426  $L_a$  as the cluster diameter of carbon aromatic rings. The terminology used by Ferrari and Robertson  
427 (2000) for nanocrystalline graphite and amorphous carbon is based on the starting material  
428 experiencing disorder. When disordering graphite,  $\frac{I(D)}{I(G)}$  will increase with increasing disorder and  
429 development of a D peak. With amorphous carbon as the starting material, the development of a D  
430 peak and the increase in  $\frac{I(D)}{I(G)}$  indicates clustering. Because the D peak arises from aromatic rings,  
431 for small  $L_a$  the D-mode strength is proportional to the cluster area or diameter Ferrari and  
432 Robertson (2000).

433 According to our discussion of possible options for carbon reduction in Section 5.1.1, where  
434 the majority of carbon is most likely produced as the outcome of post-seismic, chemical reduction  
435 processes, the initially precipitated carbon may be amorphous. Nathan et al. (1974) reported for  
436 glass-like carbon a D-peak position at  $1340\text{ cm}^{-1}$  and a G-peak position at  $1590\text{ cm}^{-1}$ , while Wang  
437 et al. (1990) reported a D-peak position at  $1347\text{ cm}^{-1}$  and a G-peak position at  $1588\text{ cm}^{-1}$ . The band  
438 positions from Wang et al. (1990) were obtained with a 515 nm laser on glass-like carbon, which  
439 was heat treated at  $2000\text{ }^\circ\text{C}$ . Wang et al. (1990) also noted that the D-peak position is sensitive to  
440 the wavelength of the laser used, exhibiting a band-position shift towards lower wavenumbers with  
441 increasing laser wavelength. The reported values from Nathan et al. (1974) and Wang et al. (1990)



442 are in agreement with our measured band positions for Arkitsa (D peak:  $1339\text{ cm}^{-1}$ , G peak:  $1584$   
443  $\text{cm}^{-1}$ ) and Schinos (D peak:  $1344\text{ cm}^{-1}$ , G peak:  $1587\text{ cm}^{-1}$ ), suggesting that the slip surface coating  
444 resembles glass-like carbon. Therefore, instead of the development of nanocrystalline graphite, our  
445 results are consistent with the presence of a partly-ordered carbon species exhibiting an electron-  
446 bond environment akin to glass-like carbon. The different electron-bond structure may be evidence  
447 for the beginning of a clustering process of the aromatic carbon rings ( $\text{sp}^2$ -clustering), which is  
448 likely to have occurred during post-seismic annealing.

449 High-resolution TEM imaging coupled with electron diffraction suggests that the surface  
450 coating is non-crystalline. This interpretation is supported by our EELS analysis, which results in  
451 spectra resembling amorphous carbon (Fig. 9). However, although both spectra have a  $\pi^*$  peak, the  
452 phase identified here has a more pronounced  $\sigma^*$  region indicating a stronger graphitic order, yet  
453 lacking full long-range order as observed in crystalline graphite (Rosenberg et al., 1986). Our  
454 spectra resemble those of Stroud et al. (2011), suggested to be indicative of glass-like carbon with  
455  $\text{sp}^2$  hybridisation. Although the EEL spectra resemble glass-like carbon, we will refer to the carbon  
456 phase from the fault exposures as partly-hybridised amorphous carbon (PHAC). Impurities of Al,  
457 Fe, and Si (Fig. 7) in the PHAC suggest either the concomitant decomposition of silicate phases  
458 (e.g., clays) during slip or the influence of hydrothermal fluids percolating along the fault zone,  
459 precipitating clay. We suggest that the presence of PHAC demonstrates that amorphous carbon is  
460 precipitated during or immediately after slip but undergoes ordering ( $\text{sp}^2$ -hybridisation) via post-  
461 seismic annealing, as demonstrated by Raman spectroscopy and EELS.

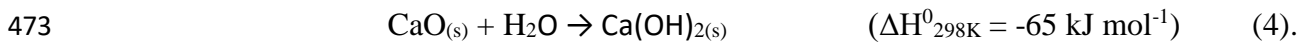


462  
 463 *Figure 9: Comparison of electron energy loss spectra. Calcite decomposition products presented here have a small*  
 464 *intensity difference of  $\pi^*$  to  $\sigma^*$  peak. The EEL spectrum from vitreous (glass-like) carbon (Stroud et al., 2011) exhibits*  
 465 *a similar  $\sigma^*$ -peak shape compared to the acquired spectra of this study. Reference spectrum of graphite has three weak*  
 466  *$\sigma^*$  peak features which are absent in all other carbon spectra.*

### 467 5.1.3 Secondary nanograin nucleation

468 Our nanostructural investigations reveal a close spatial relationship between PHAC and  
 469 newly formed calcite nanograins (e.g., Fig. 6A). Nucleation of nanograins inside the decomposition  
 470 product implies a similar chemistry of reactant and product. As discussed in Section 5.1.1, reaction

471 (1) also produces lime, which is highly reactive and can form portlandite ( $\text{Ca}(\text{OH})_2$ ) under hydrous  
472 conditions via the hydration reaction



474 In the presence of  $\text{CO}_2$ , portlandite back-reacts to calcite and water by the carbonation reaction



476 Kudłacz and Rodriguez-Navarro (2014) observed a nucleation-related crystallographic preferred  
477 orientation in two sets of portlandite crystals. The angle between the  $\{11\text{-}20\}_{\text{portlandite}}$  planes was  
478  $75^\circ$ , implying that portlandite developed with a topotactic relationship of the  $\{11\text{-}20\}_{\text{portlandite}}$  planes  
479 parallel to the  $\{110\}_{\text{CaO}}$  planes of lime. The SAED inset of Fig. 6A indicates two preferred  
480 orientations with an angle of about  $75^\circ$  (centre to centre) between the  $\{11\text{-}20\}$  planes of calcite in  
481 our natural samples, supporting our described back-reaction steps from lime, through portlandite,  
482 to calcite. In addition, we also observe dislocation-free nanograins away from the slip surface (Fig.  
483 8B). Hence, we suggest that the calcite nanograins of the nanogranular layer in figure 5A, 6A, B  
484 and 8D are not the outcome of comminution or plastic grain-size reduction during slip. We propose  
485 that these nanograins are secondary and originate from pseudomorphic growth of calcite after  
486 portlandite and portlandite after lime from decarbonation.

## 487 5.2 Fault surface evolution during the seismic cycle

488 The onset of fault slip triggers a variety of inelastic deformation processes. Cleavage  
489 fractures in Figure 3E demonstrate that slip can localise within the first  $20 \mu\text{m}$  inside the fault rock.  
490 The localisation shows that the transfer of strain from the slip surface is limited, because the  
491 cleavage fractures are concentrated close to the slip surface. Fractures cross-cut the foam  
492 nanostructure and produce a nanofabric where grains are trapped between sheet-like fracture planes  
493 (Fig 8A). Fracturing leads to grain-size reduction via comminution and produces the primary  
494 nanograins, which can fill in surface depressions (Fig. 5B). Shear heating may cause a considerable

495 temperature rise along grain and asperity contacts. The temperature increase or grain fracturing (=   
496 mechanical liming) trigger the decarbonation reaction (Eq. 1), which leads to CO<sub>2</sub> degassing and   
497 the production of lime.

498         During and immediately after fault slip, lime infiltrates cracks and cavities and reacts with   
499 the fluids to crystalline portlandite (Eq. 4). The resulting portlandite crystals back-react with the   
500 released CO<sub>2</sub> to form secondary calcite nanograins (Eq. 5). During the pseudomorphic replacement   
501 of portlandite by calcite, the secondary calcite nanograins preserve the crystallographic nucleation   
502 relationship from portlandite. Nano-clay precipitates inside the reaction medium either from the   
503 fluid or from previously decomposed clay minerals and may contribute to the fault-mirror   
504 appearance. Reduction of carbon can occur by reducing CO<sub>2</sub> with dissolved hydrogen (Eq. 2) either   
505 from the hydrolysis reaction (Eq. 3) or by water splitting with lime as a catalyst. Subsequent sp<sup>2</sup>-   
506 hybridisation of the amorphous carbon results in aromatic ring clustering and hence, in a stronger   
507 near-field bonding of the carbon. As a result, PHAC is formed.

508         Graphitisation of carbonaceous materials in active faults may be an important factor for   
509 fault reactivation because graphite is characterised by a low friction coefficient of  $\mu_{ss} = 0.08$    
510 (Oohashi et al., 2011). Glass-like carbon is a stable, non-graphitising type of carbon, which requires   
511 an activation energy of +215 kJ mol<sup>-1</sup> to initiate graphitisation (Saxena and Bragg, 1978; Hokao et   
512 al., 2000). The high activation energy would imply that on natural faults graphite can only be found   
513 after a considerable energy input to induce graphitisation of glass-like carbon species or when   
514 amorphous carbon is initially produced. Kaneki and Hirono (2019) illustrate with their deformation   
515 experiments on carbonaceous materials (CM) that CM found in natural fault zones can respond   
516 differently to imposed deformation. In essence, CMs can decrease in maturity when starting with   
517 a mature CM and increase in maturity when deforming a low-maturity CM. Saxena and Bragg   
518 (1978) identified three temperature-dependent stages for graphitisation of glass-like carbon: (1)

519 dehydrogenation at  $< 1500\text{ }^{\circ}\text{C}$  , (2) stress relief of the structure at  $1500\text{--} \leq 2300\text{ }^{\circ}\text{C}$ , and (3) the  
520 onset of graphitisation at  $> 2300\text{ }^{\circ}\text{C}$ . A high temperature treatment of synthetic glass-like carbon  
521 can cause graphitisation, suggesting a thermodynamic control of the graphitisation process.

522 Molecular dynamics simulations of  $\text{sp}^2$  carbon-bond self-assembly illustrate that  $\text{sp}^2$ -  
523 clustering from amorphous precursors can commence after 200 picoseconds at  $\sim 3200\text{ }^{\circ}\text{C}$  (Powles  
524 et al., 2009). This short time span suggests that the  $\text{sp}^2$ -clustering we observe here takes place  
525 immediately after carbon reduction. In addition, the simulations suggest that once  $\text{sp}^2$ -bonded  
526 stable clusters form, breaking these bonds to further rearrange and increase  $\text{sp}^2$ -ordering may  
527 require the need to overcome an additional energy barrier. This behaviour could explain why the  
528 PHAC observed here never reached a higher degree of graphitisation, because the energy input was  
529 not enough to break the initial stable carbon clusters. Also, synthetic glass-like carbon displays the  
530 often described, entangled, lath-like structure (e.g., Saxena and Bragg, 1978). Although the PHAC  
531 observed here lacks the aforementioned nanostructure, it does have the vibrational and electronic  
532 properties of glass-like carbon (Fig. 4 C and D, Fig. 9).

533 Because PHAC is formed by chemical precipitation and hybridisation, in contrast to  
534 pyrolysis synthesis of glass-like carbon (Powles et al., 2009), we propose that the way the carbon  
535 is formed influences the structure of the material while maintaining a similar interatomic bond  
536 structure. This suggestion is supported by Powles et al. (2009), who point out that the physical  
537 properties of glass-like carbon depend on the formation conditions, the precursor material, and on  
538 the annealing conditions. Further research is required to clarify whether the consideration of  
539 geological time scales for post-seismic annealing of PHAC at lower temperatures may yield a  
540 similar result in graphitisation or whether fault reactivation might contribute to slightly increasing  
541 the PHAC maturity by imposing deformation as suggested by Kaneki and Hirono (2019).

542 5.3 Advances in understanding fault surface evolution

543 Amorphous carbon has been observed in deformation experiments (Verberne et al., 2014;  
544 Spagnuolo et al., 2015; Delle Piane et al., 2017) and its mechanical role has been assessed by  
545 Oohashi et al. (2011) and Kaneki and Hirono (2019). Oohashi et al. (2011) measured a friction  
546 coefficient for amorphous carbon of  $\mu = 0.54$  at the initiation of slip and a low steady-state friction  
547 coefficient of  $\mu_{ss} = 0.15$  at seismic slip velocities. Di Toro et al. (2011) speculated about fault  
548 lubrication by carbonate decomposition products but the present study shows the extent of  
549 amorphous coats. Decarbonation may be an important process for producing a thin fault surface  
550 coat and mainly responsible for the low surface roughness by smoothing out surface corrugations.  
551 Because the fault surfaces show little variation in the field, our results suggest that the fault planes  
552 of both fault exposures are at present covered with the decarbonation and carbon hybridisation  
553 products.

554 Our EEL and Raman spectroscopy results indicate that PHAC has a similar carbon-bonding  
555 environment compared to glass-like carbon. Synthetic glass-like carbon has an entangled lath-like  
556 structure consisting of ordered carbon sheets while the natural PHAC has no nanostructure. The  
557 mechanical properties of glass-like carbon, heat treated at 1000 °C, show a Youngs modulus of  $E$   
558 = 29 GPa and a shear modulus of  $G = 12.5$  GPa (Robertson, 1991). Hokao et al. (2000) report a  
559 friction coefficient for glass-like carbon of  $\mu = 0.21$  and  $\mu = 0.13$  for mixtures of glass-like carbon  
560 with graphite. However, as the structure of the PHAC is different from synthetic glass-like carbon,  
561 we suggest that mechanical properties of the synthetic material may not fully reflect the mechanical  
562 behaviour of the carbon phase reported here. Because little is known about the deformation of  
563 various crystalline and non-crystalline carbon phases further experimental investigations are  
564 warranted.

565           While previous research focused mainly on mechanisms of grain size reduction (Sammis  
566 and Ben-Zion, 2008; Siman-Tov et al., 2013), our results highlight the importance of decarbonation  
567 and back-reactions as part of the overall fault deformation and healing behaviour. In contrast to  
568 Siman-Tov et al. (2013), we do not observe small grains evolving from fracturing long, twin-  
569 derived beams of calcite. However, our results indicate that, for the natural carbonate faults studied  
570 here, a nanogranular coat is not the only feature producing a fault mirror surface, similar to results  
571 of Fondriest et al. (2013). For example, the Schinos fault exposure illustrates that MMSs can  
572 contain no coat of calcite nanograins (Fig. 7C, E and F). Instead, MSS are produced by the interplay  
573 of grain-size reduction, decarbonation, back-reactions, and annealing. The present study does not  
574 exclude crystal-plastic deformation as an important grain-size reduction mechanism in fault gouges  
575 to form primary nanograins. Instead, we demonstrate that grain-size reduction alone may not be  
576 the governing mechanism to form a fault mirror surface. The formation of an amorphous phase  
577 with low viscosity during deformation could explain the low friction, stable, near-steady-state  
578 behaviour after the onset of weakening observed by Han et al. (2007), as well as the increase in  
579 friction coefficient via re-carbonation and carbon hybridisation after displacement stopped  
580 (Spagnuolo et al., 2015). The chemical reactions revealed here suggest that a succession of healing  
581 reactions take place after fault slip ceases.

## 582 6 Conclusions

583           We conclude that decarbonation of calcite and the subsequent reaction of the decarbonation  
584 products produces fault mirror surfaces. The decarbonation process itself may be a major co-  
585 seismic fault weakening factor and fault slip is facilitated on a decarbonation-product glide film.  
586 Frictional behaviour during slip may be dictated by lubrication of low viscosity (amorphous)  
587 calcium oxide and (amorphous) carbon. Occasional precipitation of clay inside the decarbonation  
588 products may contribute to decreasing surface roughness. Post-seismic hybridisation produces

589 partly-hybridised amorphous carbon and connects footwall with hanging wall. Ultimately, post-  
590 seismic carbonation of portlandite produces new, secondary nanograin calcite crystals by  
591 pseudomorphic replacement and this carbonation facilitates fault healing.

592

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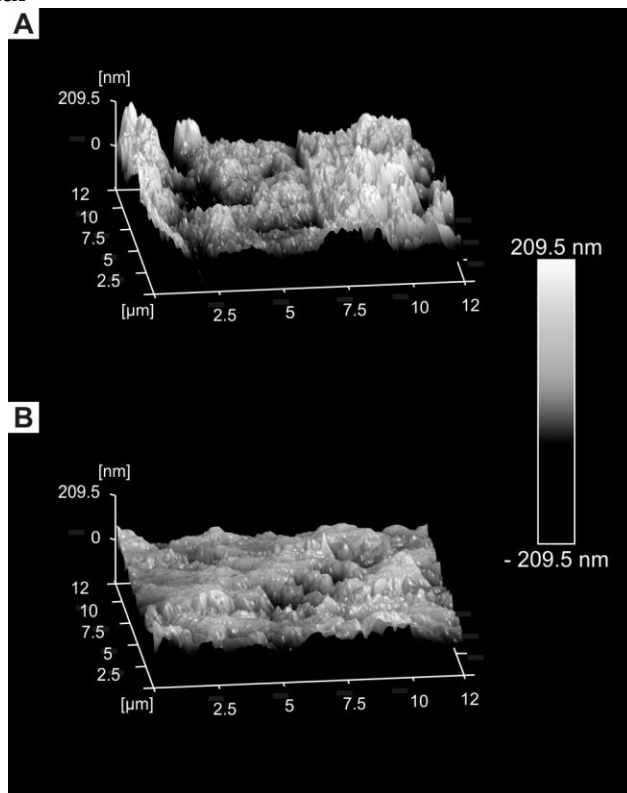
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Supplementary material



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*SF 1: Atomic force microscope measurements of the surface roughness of both fault exposures. A: Arkitsa fault surface with a calculated mean surface roughness of 63.5 nm. B: Schinos fault surface with a calculated mean surface roughness of 32.3 nm.*