Ablation of amorphous Polyethy(ethyl)letone (PEEK) by a femtosecond Ti:sapphire laser

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ABSTRACT

Laser micro-machining of amorphous PEEK has been demonstrated with 180 fs/1 kHz NIR (775 nm) and NUV (387 nm) laser pulses. The single pulse ablation threshold was found to be 2.01 ± 0.05 J/cm² and 0.23 ± 0.02 J/cm² at 775 nm and 387 nm respectively. The significant difference in ablation threshold is due to the requirement for 3-photon absorption at 775 nm, where PEEK is transparent while significant linear absorption within the material bandgap occurs at 387 nm, enhancing 2-photon absorption. A high 2-photon absorption coefficient, measured to be $\beta_{387} \sim 38$ cm GW¹ supports this view while at the bandgap edge, 400 nm, $\beta_{400} \sim 0.7$ cm GW¹. Multi-pulse exposure yields incubation coefficients of $S_{775} = 0.72 \pm 0.01$ and $S_{387} = 0.85 \pm 0.02$ hence incubation is significantly reduced in the NUV. Ablation of PEEK with NUV fs pulses demonstrates much reduced melting and re-deposition, thus precision NUV polymer micromachining is accomplished while laser induced periodic surface structures (LIPSS) with pitch $\Lambda < 0.4$ μ m are observed at the base of ablated regions. Scanned areas exhibit white light diffraction due to this sub-micron periodic surface modulation. With the aid of a phase only spatial light modulator, multi-beam NUV micro-structuring is achieved, speeding micro-processing while reaching a line width $< 4 \mu$ m with NA = 0.4 objective.

Keywords: PEEK, Femtosecond pulsed laser, Micro-machining, Ablation threshold, Incubation effect, Nonlinear absorption.

1. INTRODUCTION

Polyether ether ketone (PEEK) is a high performance thermoplastic polymer with outstanding mechanical and chemical properties, high temperature resistance and good biocompatibility for medical applications^{1–4}. Therefore, PEEK is widely used in electronics, medicine and surgery, aerospace and automotive^{1,2,5}. Tribological property modification and periodic structuring of PEEK surfaces (amorphous and semi-crystalline) using femtosecond infrared pulses were recently investigated^{6,7} with the window for creating LIPSS very narrow.

Topography modification of PEEK surfaces can achieve desired surface properties, such as improved wettability⁴. Processing methods include chemical etching and plasma treatments, however, aromaticity within the polymer backbone may offset the etching in plasma treatments⁸. Chemical etching may also damage the surface layer⁹. Therefore, laser processing offers a promising tool for highly specific, high resolution treatments¹⁰. Femtosecond laser pulses provide negligible heat diffusion zones and minimised energy loss for machining polymer surfaces¹¹. High intensity fs lasers can generate multiphoton induced ablation and ionization^{12,13} in transparent substrates.

The purpose of the study presented here is to investigate and compare the ablation threshold, incubation and optical properties (linear and nonlinear absorption) of amorphous PEEK films at both 775 nm and 387 nm. In the NIR, PEEK is transparent so that excitation proceeds through true multi-photon absorption - while at 387 nm within the band edge, linear absorption also occurs so that at ultrahigh intensity, reverse saturated absorption (RSA) is the likely NL mechanism in the NUV.

2. EXPERIMENTAL SETUP

2.1 Material

The material used was amorphous PEEK (25, 50, 100 and 200 μ m thick, Goodfellow). While the 25 μ m and 50 μ m films had optically flat surfaces, thicker films required optical polishing. Prior to experiments, the PEEK samples were cleaned in methanol with an ultrasonic treatment for 5 minutes.

2.2 Laser processing

Figure 1 shows the scheme of the femtosecond laser system and optical setup. A Titanium-sapphire laser system (Clark-MXR CPA 2010) provides a pulse energy of 1 mJ at 1 kHz with pulse duration of 180 fs at 775 nm ($M^2 \sim 1.3$). For NIR ablation, the sample was set perpendicular to the laser beam on an Aerotech micro position translation stage (x, y, z and θ) and the attenuated beam, expanded by \times 3 then directed to a Nutfield XLR8 scan head with 100 mm f theta lens. After a beams splitter, the transmitted component at 775 nm NIR light was attenuated (Diffractive Optic Attenuator, DOA, TOPAG), reflected from a phase only SLM (Hamamatsu X10468-02) and then frequency-doubled to generate 387 nm NUV wavelength in a thin, 0.6 mm thick Beta Barium Borate (BBO) crystal. All treatment was conducted in air at atmospheric pressure. The pulse energy was measured by a Molectron pyroelectric detector (3Sigma) over an average of 5000 pulses.

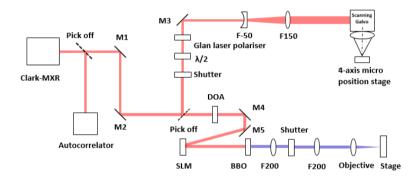


Figure 1. Scheme of laser system and experimental set-up. There are two beam lines for 775 nm and 387 nm processing.

2.3 Surface characterization

Laser-processed samples were examined by optical microscopy (Nikon), white light interferometry (WYKO NT1100), SEM and XRD.

3. RESULTS AND DISCUSSION

3.1 Ablation threshold

The experiment was performed with a different number of laser shots (1, 20, 50 and 100 pulses) over a range of pulse energies (figure 2). The peak laser fluence was calculated by the equation:

$$F = \frac{2E}{\pi\omega_0^2} \tag{1}$$

where E is the incident energy per pulse, and ω_0 is the waist radius of the focal spot. The squared diameters of the ablated area are related to the laser fluence by the following equation¹⁴:

$$D^2 = 2\omega_0^2 \ln \frac{F}{F_{th}} \tag{2}$$

where D is the diameter of the spot ablated on PEEK, and Fth is the ablation threshold of the material.

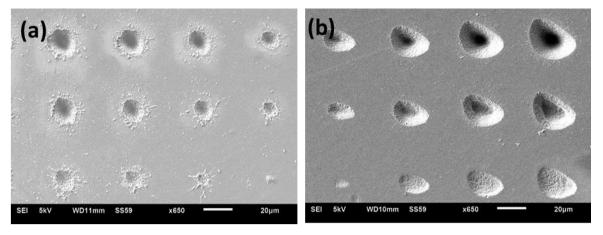


Figure 2. (a) SEM micrograph image of ablation spots on PEEK at 775 nm with 100 mm f-theta lens. From right to left, Fluence F= 0.94, 1.21, 1.48, 1.75 J cm⁻², from top to bottom: 100, 50, 20 pulses. Note the significant re-deposition of melted debris around the craters (b) SEM micrograph image of ablation spots at 387 nm with objective focal length f = 100 mm. From left to right: Fluence F=0.18, 0.26, 0.36, 0.44 J cm⁻², from top to bottom: 100, 50, 20 pulses. Ablation is much superior with almost no re-deposition.

Figure 3 shows a plot of squared diameter against lnE yielding the ablation thresholds for both 775 nm and 387 nm. The fits to equation (2) are satisfactory. The single pulse ablation threshold was found to be 2.01 ± 0.05 J cm⁻² at 775 nm compared to 0.23 ± 0.02 J cm⁻² at 387 nm, a significant reduction in the NUV. The ablation thresholds reduce with pulse number, indicating that incubation occurs.

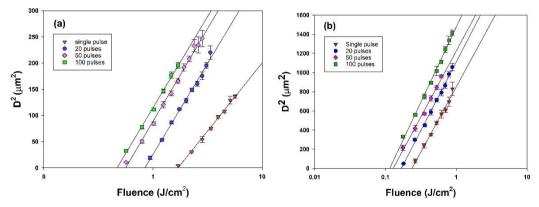


Figure 3. D² versus Fluence for 1, 20, 50 and 100 pulses. (a) 775 nm, (b) 387 nm.

3.2 Incubation coefficient

Figure 4 (a) plots the threshold fluence versus at both wavelengths with the number of applied pulses, highlighting incubation or the variation of threshold with exposure. According to the incubation model proposed by Jee^{15} , $F_{th}(N)$ can be described by,

$$F_{th}(N) = F_{th}(1) \times N^{S-1}$$
 (3)

where S is the so-called incubation coefficient and $F_{th}(1)$ is the ablation threshold for N=1.

The equation can be expressed in the following form:

$$ln[NF_{th}(N)] = S ln(N) + ln[F_{th}(1)]$$
(4)

Figure 4 (b) shows the data plotted in this manner from which the incubation coefficients are $S_{775} = 0.72$ while $S_{387} = 0.85$ hence incubation is much reduced in the NUV. The value at 387 nm is close to that of a number of metals¹⁶. It can also be inferred that incubation effectively saturates when N > 100. The solid line represents a least square fit to the equation (4).

Incubation therefore plays an important role in multi-pulse micro machining on PEEK using a femtosecond pulsed laser at both 775 nm and 387 nm but significantly reduced in the NUV. Incubation in PEEK is likely due to the chemical composition change, especially at 775 nm - and developing surface roughness.

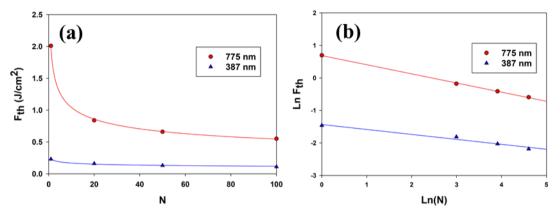


Figure 4. (a) Laser damage (ablation) threshold fluence F_{th} as a function of the number N of pulses for amorphous PEEK. (b) Plot of logarithm functions of threshold fluence F_{th} and the number N of pulses for amorphous PEEK.

3.3 Absorption coefficient

Figure 5 shows the transmission of PEEK as a function of film thickness measured with a low energy collimated beams at 775 nm, 387 nm and for comparison, 400 nm just on the band edge. Light will be attenuated when passing through the material according to the Beer-Lambert Law, which, allowing for reflection losses, gives

$$I_{(z)} = I_0 (1-R)^2 e^{-\alpha z}$$
 (5)

where I(z) is the transmitted intensity, I_0 is the incident intensity, α is the absorption coefficient (cm⁻¹), and z is the material depth (cm).

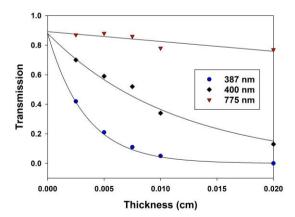


Figure 5. Linear absorption of amorphous PEEK at 387, 400 and 775 nm with film thickness. Exponential fits at 387 and 400nm yield linear absorption coefficients of 300 cm⁻¹ and 80 cm⁻¹ respectively. At 775 nm, $\alpha \sim 6.4$ cm⁻¹, essentially transparent.

Assuming for the moment that absorption is negligible, the theoretical transmission $T_{th} = (1-R)^2 = 0.88$, where the reflection per face $R = (n-1)^2/(n+1)^2 = 0.06$ from the Fresnel equation¹⁷, $n_{(PEEK)} = 1.65^{18}$, in agreement with that observed at z = 0.

Non-linear absorption was measured simply by varying incident intensity on the films and measuring transmission, figure 6, which highlights NL absorption on the 25 μ m film at 387 nm - even at a few GW cm⁻². When both linear and NL absorption occur in a material, the transmission is given by¹⁹,

$$T = (1 - R)^{2} \frac{\alpha_{0} \exp(-\alpha_{0} z)}{\alpha_{0} + \beta I_{0} [1 - \exp(-\alpha_{0} z)]}$$
(6)

where β is the 2-photon absorption coefficient. By plotting this function, setting $\alpha_0 = 300$ cm⁻¹ and $z = 2.5 \times 10^{-3}$ cm, the best fit yields $\beta_{(387)} = 38$ cm GW⁻¹, a high value. Note that in this intensity range, the function is nearly linear. With this β , the effective coupling depth d $\sim 1/\beta I$ with a 1 μJ , 200 fs pulse focused to a 30 μ m spot gives d ~ 100 nm, consistent with observations of ablation rate per pulse.

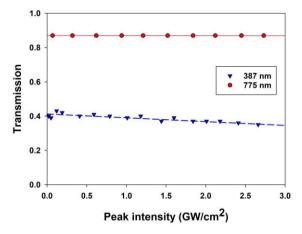


Figure 6. Transmission as a function of incident intensity of 25 µm PEEK film.

The photon energy at 775 nm is hv = 1.6 eV. As the band gap of PEEK is ~ 3.2 eV, NL two photon absorption is required to excite electrons to the lowest levels of the conduction band. Figure 7 shows the calculated band structure (density of states) in PEEK based on a QM model of the molecular polymer structure²⁰. Once excited to the conduction band via NL absorption, further sequential photon absorption will occur due to the much higher density of states at 4.8 eV (3-photon) at 775 nm. At 387 nm, (hv = 3.2 eV) single and sequential two photon absorption will occur as a high density of states is available for the second photon at 387 nm absorption. As intensity increases, the much higher absorption cross section for excited state absorption will rapidly populate states at 6.4 eV. Hence, instead of normal saturation, absorption increases with intensity, or Reverse Saturation Absorption (RSA)²¹. This excitation energy is able to break almost all the bonds in the polymer directly and consistent with the clean ablation observed with little thermal effect. Further work using the Z-scan technique (not presented here) confirmed that β is intensity dependent – indicating that RSA is indeed relevant. In the DOS calculations, Carbon atoms contribute mostly to the density of states. The transitions from valence band to conduction band near 3.2 eV is due to the degenerate e_{1g} electronic levels of benzene²².

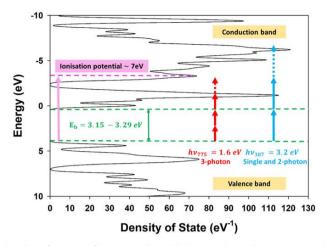


Figure 7. Band energy diagram (density of states) of PEEK and multi-photon absorption scheme. The density of states was calculated theoretically by J Ho et al²⁰.

3.4 Laser Induced Periodic Surface Structuring (LIPSS)

Near threshold laser ablation can generate laser induced periodic surface structures due to the interference of surface scattered waves with the incoming radiation²³. The resulting modulation of electron density creates periodic surface profile modulation with a pitch close to the incident wavelength. Figure 8 (a) shows an SEM image of these low frequency LIPSS on 50 μ m thick PEEK (F = 0.52 J cm⁻², scan speed = 3 mm s⁻¹). The direction of these low frequency LIPSS was parallel to the incident electric field. This ripple orientation is typical for dielectrics and polymers, while metals often show perpendicular direction LIPSS. White light diffraction of larger area LIPSS is displayed in figure 8 (b) (c).

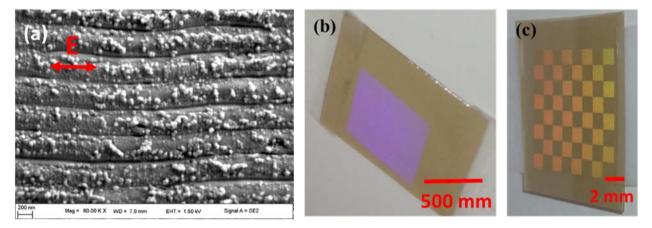


Figure 8. (a) SEM image of LIPSS on amorphous PEEK at 387 nm. Note the presence of nano-particles with sizes < 50 nm (b) White light diffraction from LIPSS on 200 μ m thick PEEK (F = 0.52 J cm⁻², 1 overscan). (c) High angle WL diffraction from chessboard pattern (F = 0.52 J cm⁻², f = 100 mm) by rotating linear polarization between squares.

3.5 NUV Parallel processing

Parallel processing can speed fabrication during micro-machining. With appropriate computer generated holograms CGHs applied to an SLM (Hamamatsu X10468-02), parallel NIR beams at 775 nm were frequency doubled to NUV beams in a thin BBO crystal. With a higher NA = 0.4 objective, fine lines width \sim 4-6 μ m on 10 μ m pitch were machined in parallel on PEEK is shown in figure 9. Scan speed here was 1 mm s⁻¹ with fluence F = 1.16 J cm⁻². Uniformity can be improved with the addition of a feedback system.

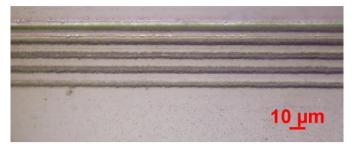


Figure 9. Parallel micro-structures machined on amorphous PEEK at 387 nm with line width of \sim 4-6 μ m, NA = 0.4. Scan speed s = 1 mm s⁻¹ with fluence F = 1.16 J cm⁻².

3.6 XRD spectrum

Bulk PEEK can have a significant semi-crystalline structure and is often opaque. The thin films used here were supplied as "amorphous" hence we checked the micro-structure using XRD (Rigaku MiniFlex). Figure 10 shows the XRD spectrum of the $100~\mu m$ amorphous PEEK film and typical of the XRD spectra obtained on various film thicknesses. There are no sharp peaks, characteristic of a semi-crystalline phase but only a broad feature which is due to X-ray fluorescence. The films used are indeed mainly amorphous. If a semi-crystalline component exists, then it is at the few % levele17.

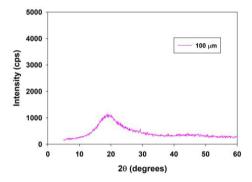


Figure 10. XRD spectrum of 100 μm amorphous PEEK.

4. CONCLUSION

Amorphous PEEK films can be ablated with ultrahigh intensity fs laser pulses at 775 nm, where the material is transparent and with much better precision at 387 nm just within the band edge. For single pulse, the ablation threshold is 2.01 J cm^{-2} and 0.23 J cm^{-2} respectively, the much lower threshold at 387 nm evidence of a much more efficient NL absorption mechanism. As linear absorption is negligible at 775 nm, ($\alpha = 6.4 \text{ cm}^{-1}$), ablation takes place through nonlinear two photon absorption followed by sequential photon absorption. Incubation is significant and results in a significant thermal effect, causing sub-surface carbonization of the material below ablation threshold while above ablation threshold, significant re-deposition was observed with melted debris around the craters. On the contrary, at 387 nm, ultrahigh intensity ablation was much cleaner, with the single pulse ablation threshold reduced by almost one order of magnitude. With a photon energy just inside the material bandgap, single and sequential photon absorption occur within the temporal pulse length and as intensity is increased, the high cross section for excited state absorption causing reverse saturation (RSA) dominates the NL absorption process. A high level of electron excitation in the conduction band with at least 6.4 eV is able to break almost all the polymer bonds directly, reminiscent of UV excimer ablation. The ability to micro-structure LIPSS becomes relatively easy with ultrahigh intensity 387 nm pulses since the absorption depth reduces to ~100 nm. These LIPSS may yet prove useful in medical applications, such as cell culture by improving cell attachment in spinal implants¹.

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