

## Communication

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# Photo-Rechargeable Zinc-Ion Capacitor using 2D Graphitic Carbon Nitride

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## Abstract

Off-grid energy storage devices are becoming increasingly important to power distributed applications such as internet of things and smart city ubiquitous sensor systems. To date, this has been achieved by combining an energy storage device e.g. battery or capacitor with an energy harvester e.g. solar cell. However, this approach inherently increases device footprint, and the output voltages of energy harvesters often do not match those required by energy storage device. We introduce the first photo-rechargeable zinc-ion capacitors, where graphitic carbon nitride is used as at the same time as the capacitor electrode and light harvesting material. This approach allows for light to recharge the capacitor directly, and can be operated in a continuous light powered mode. These capacitors show a photo-rechargeable specific capacitance of  $\sim 11377$  mF g<sup>-1</sup>, a photo-charging voltage response of  $\sim 850$  mV and a cyclability with  $\sim 90\%$  capacitance retention over 1000 cycles.

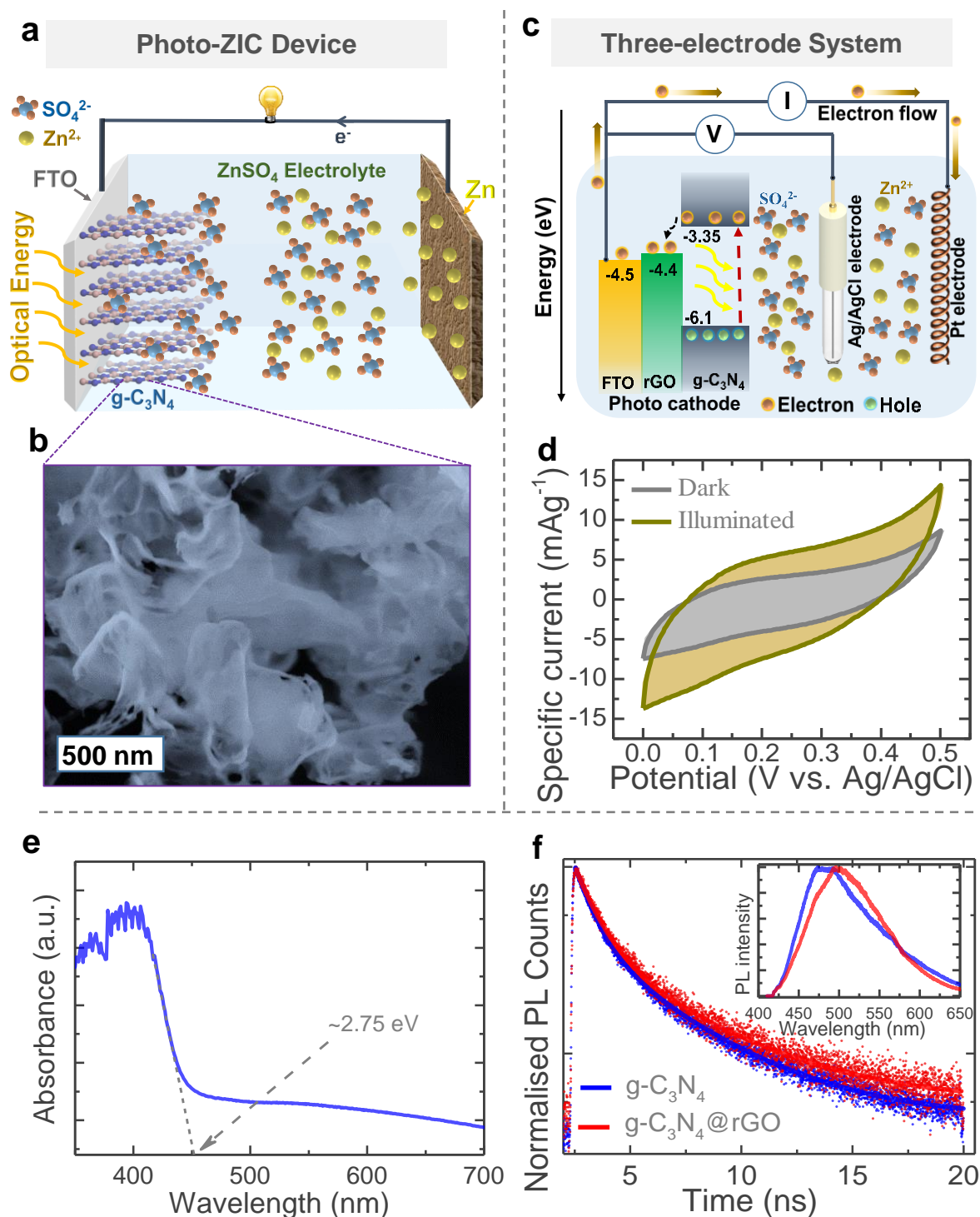
**KEYWORDS:** Hybrid zinc-ion capacitor, photo-rechargeable, off-grid energy, graphitic carbon nitride.

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3 The development of off-grid energy storage devices is becoming increasingly important to  
4 fulfil the requirements of continuous data processing and transmission for Internet of things  
5 (IoT), smart cities and autonomous integrated devices.<sup>1-3</sup> Previously, the design of energy  
6 supplies for standalone devices involved the extrinsic combination of a separate energy  
7 harvesting component (e.g. photovoltaic solar cells) and a rechargeable energy storage  
8 component (e.g. batteries, supercapacitors, hybrid capacitors). For instance, Um *et al.* reported  
9 a photo-rechargeable solid-state lithium-ion battery (LIB) integrated with Si solar cells,<sup>4</sup> Du *et*  
10 *al.* fabricated a flexible solid-state graphene-based supercapacitor and perovskite solar cell  
11 integrated system,<sup>5</sup> Li *et al.* also recently reported a photo-rechargeable lithium-ion capacitor  
12 driven by a perovskite solar cell.<sup>6</sup> However, the external integration of solar cells with  
13 rechargeable devices results in an increase of the overall device footprint, increases ohmic  
14 transport losses, and requires additional external electronics to match the output of the solar  
15 cell to the energy storage device.<sup>7,8</sup> Direct printing of both solar cells and rechargeable devices  
16 in the same housing by layer-by-layer device fabrication approach have been applied to  
17 overcome some issues of reducing packaging volume to increase volumetric energy storage  
18 performance and reduce in ohmic transport losses.<sup>9-11</sup> For instance, Liu's group reported a fiber-  
19 shaped supercapacitor integrated with a dye-sensitized solar cell sharing an electrode.<sup>12</sup> Liang  
20 *et al.* also reported shared electrode based supercapacitor and perovskite solar cell systems.<sup>13,14</sup>  
21 However, this approach usually increases the overall device complexity and fabrication cost.  
22 Photoelectrodes based on the physical combination of photovoltaic and hybrid energy storage  
23 materials can be realized in a single device architectures.<sup>15-17</sup> However, such devices offer  
24 limited photo-charging performance due to issues such as inefficient transportation of  
25 photogenerated charge carriers from photovoltaic materials to energy storage materials because  
26 of energy levels mismatch at the interface between the two materials, inefficient photon  
27 absorption in the photovoltaic active material due to the blocking of light by the energy storage  
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3 material, phase separation of the photovoltaic and energy storage materials during fabrication  
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5 and an increase in resistance due to interfacial effects between the two components. Therefore,  
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7 it is desirable to explore new materials that combine photovoltaic and energy storage  
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9 capabilities simultaneously, as proposed recently for LIBs using perovskite based  
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11 electrodes.<sup>8,18-20</sup>  
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15 Recently, zinc-ion capacitor (ZIC) has been proposed as a promising energy storage  
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17 technology, capable of balancing energy and power density favourably along with excellent  
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19 safety and cycling life.<sup>21-24</sup> Here, we present the first photo-rechargeable ZIC (Photo-ZIC)  
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21 using 2D graphitic carbon nitride (g-C<sub>3</sub>N<sub>4</sub>) photo-cathode for the simultaneous harvesting of  
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23 solar energy and charge storage. **Figure 1a** depicts the proposed photo-ZIC device architecture  
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25 schematically. The 2D g-C<sub>3</sub>N<sub>4</sub> shown in **Figure 1b** is used as a photo-active cathode material  
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27 because of its unique combination of a band gap in the visible spectrum (~ 2.7 eV see further)  
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29 and high surface area desirable for ZIC cathodes (ion adsorption). Our 2D g-C<sub>3</sub>N<sub>4</sub> was  
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31 characterised by X-ray diffraction (XRD), Fourier-transform infrared spectroscopy (FTIR),  
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33 Brunauer-Emmett-Teller (BET) and thermogravimetric analysis (TGA) which are provided in  
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35 **Figure S1-S4**. The light induced changes in electrochemical properties of g-C<sub>3</sub>N<sub>4</sub> electrodes  
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37 was first characterized in a beaker cell using three-electrode cyclic voltammetry (CV)  
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39 measurements both in the dark and under illumination ( $\lambda \sim 420$  nm, intensity  $\sim 50$  mWcm<sup>-2</sup>)  
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41 using the setup shown in **Figure 1c**. In order to facilitate the transport of photo-excited  
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43 electrons, a favourable energy band alignment is achieved by mixing g-C<sub>3</sub>N<sub>4</sub> (93 wt%) with  
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45 reduced graphene oxide (rGO, 2wt%) as described in the experimental section. The rGO acts  
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47 as a conductive additive and selective electron transport layer in the photo-cathode – allowing  
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49 the flow of photo-excited electrons to the anode through the external circuit. **Figure S5** shows  
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51 SEM, XRD and Raman characterisations of the rGO used in our devices, and top-view and  
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cross-section SEM images of the photo-cathodes are provided in (Figure S6a and b respectively).



**Figure 1.** (a,b) Schematic illustration of 2D  $\text{g-C}_3\text{N}_4$  cathode and Zn anode based photo-ZIC and SEM image of 2D  $\text{g-C}_3\text{N}_4$ . (c) Schematic representation of a three-electrode experiment of

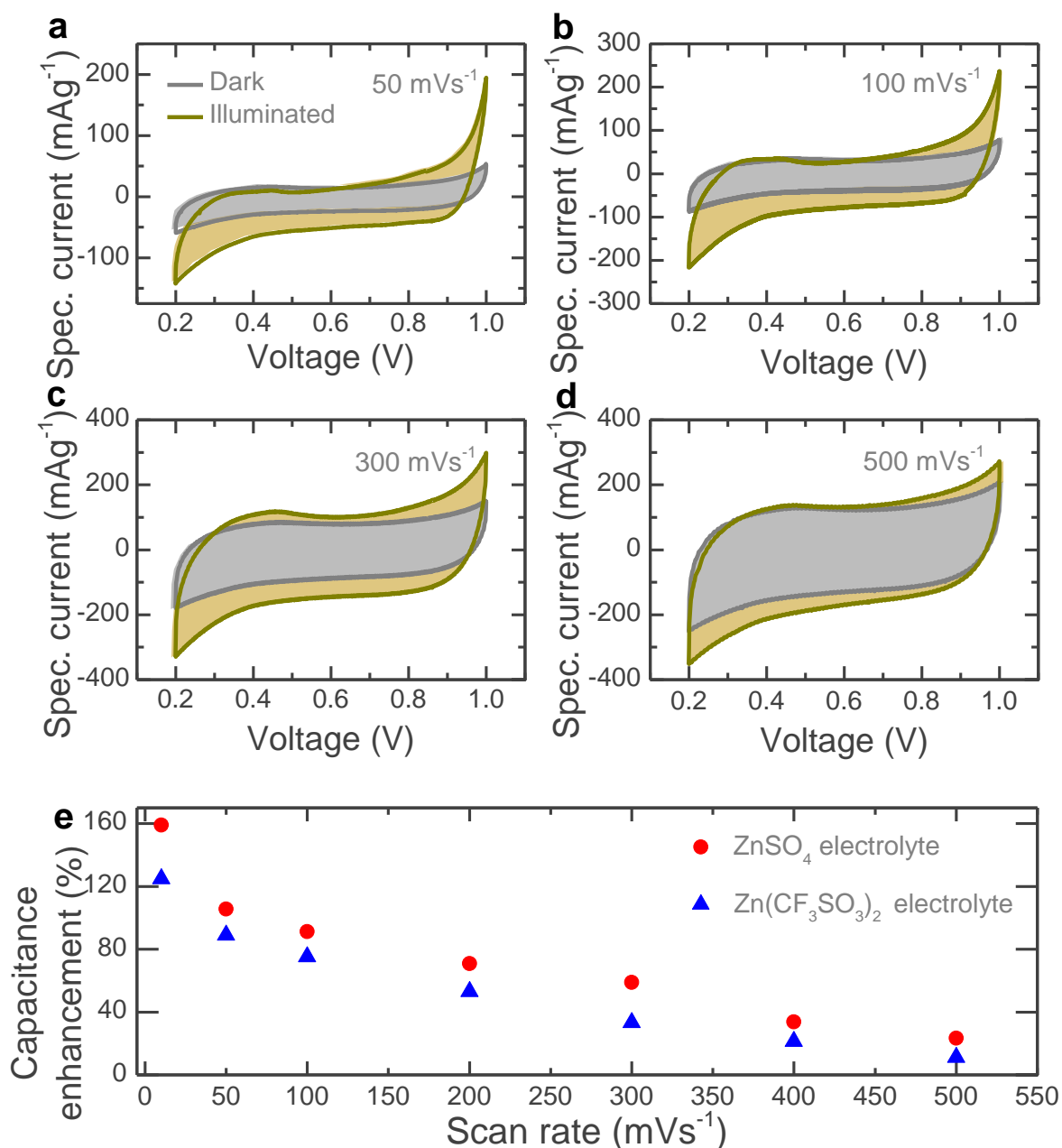
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3 *g-C<sub>3</sub>N<sub>4</sub>@rGO/FTO electrodes. (d) CV profiles (10 mVs<sup>-1</sup>) in dark and illuminated conditions*  
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5 *(λ ~ 420 nm, intensity ~ 50 mWcm<sup>-2</sup>). (e) Optical absorption spectrum of g-C<sub>3</sub>N<sub>4</sub>. (f) Time*  
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7 *resolved PL signal of g-C<sub>3</sub>N<sub>4</sub> (blue) and g-C<sub>3</sub>N<sub>4</sub>@rGO (red) samples (log scale on y-axis).*  
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9 *Lines are exponential fits to the data. (Inset) normalised steady-state PL emission spectra of*  
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11 *g-C<sub>3</sub>N<sub>4</sub> (blue) and g-C<sub>3</sub>N<sub>4</sub>@rGO (red).*

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15 For the three-electrode experiments, the electrodes are drop-casted on fluorine-doped tin oxide  
16 (FTO)-coated glass substrates (see **Figure 1c**). **Figure 1d** shows the CV responses (at a scan  
17 rate of 10 mVs<sup>-1</sup>) of the g-C<sub>3</sub>N<sub>4</sub>@rGO/FTO in the dark and under illumination. The area  
18 enclosed by the CV curve increases under illumination relative to the dark scan from which we  
19 calculate a capacitance enhancement of ~ 82% ( $= \frac{C_{light} - C_{dark}}{C_{dark}} \times 100\%$ ; with  $C_{light}$  and  $C_{dark}$   
20 the specific capacitances in illuminated and dark conditions at same scan rate). Similarly, CV  
21 scans at 50 mVs<sup>-1</sup>, 70 mVs<sup>-1</sup> and 100 mVs<sup>-1</sup> in dark and illuminated conditions show an  
22 enhancement of ~ 58%, ~ 47% and ~ 43% (**Figure S7a-c**). The increase in specific capacitance  
23 of g-C<sub>3</sub>N<sub>4</sub>@rGO/FTO under illumination is attributed to the increased charge carrier density  
24 from the photo-excited electrons and holes when illuminated with a wavelength λ ~ 420 nm (~  
25 2.95 eV), which is higher energy than the band gap of g-C<sub>3</sub>N<sub>4</sub> (~ 2.75 eV from the UV-Vis  
26 spectrum in **Figure 1e**). We suggest that the photogenerated electrons transfer from g-C<sub>3</sub>N<sub>4</sub> to  
27 FTO through the rGO, whereas photogenerated holes participate in anions adsorption on the  
28 electrode surface to enhance the overall specific capacitance under illumination (see further).  
29 Control experiments using only rGO or activated carbon electrodes without g-C<sub>3</sub>N<sub>4</sub> do not show  
30 a photo response as shown in **Figures S8**.

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Photoluminescence (PL) spectra of pristine g-C<sub>3</sub>N<sub>4</sub> and g-C<sub>3</sub>N<sub>4</sub>@rGO suggest the transport of  
photogenerated electrons from g-C<sub>3</sub>N<sub>4</sub> to rGO. **Figure 1f** (inset) compares the PL emission of  
pristine g-C<sub>3</sub>N<sub>4</sub> (blue) and g-C<sub>3</sub>N<sub>4</sub>@rGO (red), with both samples demonstrating the expected

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3 broad PL emissions from the multiple photoactive energy levels of the material.<sup>25</sup> An apparent  
4 increase on the red side of the spectrum is observed upon the addition of rGO, indicating the  
5 introduction of new photoactive sub-gap energy levels from the rGO and hence, resulting in  
6 pathways for lower-energy recombination and PL emission. **Figure 1f** shows the photo-excited  
7 charge carrier lifetimes of the pristine g-C<sub>3</sub>N<sub>4</sub> (blue) and g-C<sub>3</sub>N<sub>4</sub>@rGO (red). Average lifetimes  
8 are extracted from bi-exponential fitting of the PL decay and values of 1.76 ns ± 0.02 and 1.95  
9 ns ± 0.02 are calculated for the pristine g-C<sub>3</sub>N<sub>4</sub> and g-C<sub>3</sub>N<sub>4</sub>@rGO samples respectively,  
10 corresponding to a ~11% increase in charge carrier lifetime after adding rGO. Due to the  
11 favourable energy pathway (as discussed earlier in **Figure 1c**), the photo-excited electrons  
12 preferentially transfer from g-C<sub>3</sub>N<sub>4</sub> to rGO thus increasing the average time between excitation  
13 and recombination/emission.  
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29 The results presented this far were obtained in beaker cells using platinum wire (Pt) counter  
30 electrodes, in what follows, g-C<sub>3</sub>N<sub>4</sub>@rGO/FTO cathodes are paired with a metallic zinc (Zn)  
31 anodes to form Photo-ZICs. Metal anodes and in particular Li metal anodes in LIBs, often  
32 suffer from dendrite formation, which can cause short-circuits and subsequent ignition of the  
33 electrolyte which is a major safety risk.<sup>26</sup> Here however, we are using a non-flammable aqueous  
34 electrolyte, and some publications seem to indicate that Zn anodes are fairly stable over  
35 extended cycle show superior capacitance stability over more than 10000 cycles.<sup>21-24</sup> Post-  
36 mortem SEM imaging of our Zn anodes after 1000 cycles did not show any signs of dendrite  
37 formation (see further). The electrodes are mounted in CR2450 coin cells in which we  
38 machined holes and mounted ~ 8 mm diameter optical glass to allow for illumination (see  
39 experimental section and **Figure S9a**). CV scans of the photo-ZICs were evaluated in dark and  
40 illuminated conditions ( $\lambda \sim 420$  nm, intensity  $\sim 50$  mWcm<sup>-2</sup>) in 2M zinc sulfate (ZnSO<sub>4</sub>)  
41 aqueous electrolyte.  
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**Figure 2.** CV comparison of the Photo-ZIC at different scan rates of (a) 50 mVs $^{-1}$ , (b) 100 mVs $^{-1}$ , (c) 300 mVs $^{-1}$  and (d) 500 mVs $^{-1}$  in dark and illuminated ( $\lambda \sim 420$  nm, intensity  $\sim 50$  mWcm $^{-2}$ ). (e) Capacitance enhancement with respect to scan rate of the Photo-ZIC in ZnSO $_4$  and Zn(CF $_3$ SO $_3$ ) $_2$  aqueous electrolytes under light illumination ( $\lambda \sim 420$  nm, intensity  $\sim 50$  mWcm $^{-2}$ ).

**Figure 2a-d** show the respective CV profiles at different scan rates of 50 mVs $^{-1}$ , 100 mVs $^{-1}$ , 300 mVs $^{-1}$  and 500 mVs $^{-1}$  (**Figure S9b**, CV responses in dark). Enhancements of up to 160%



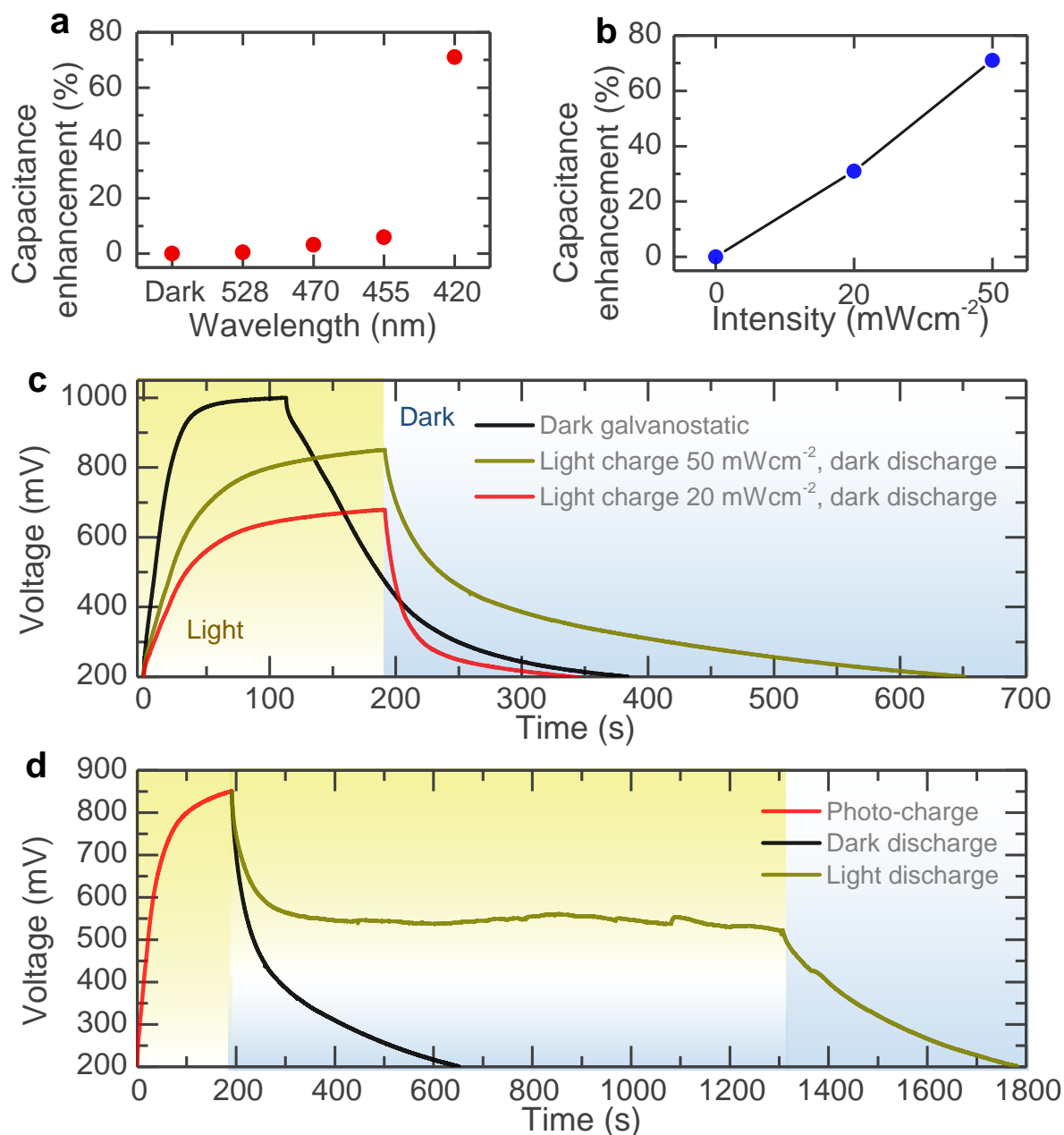
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3 are achieved at slow scan rates ( $10 \text{ mVs}^{-1}$ ), and even at fast scan rates ( $500 \text{ mVs}^{-1}$ )  
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6 enhancements of 23.43% are observed despite the relative limited time for light charging at  
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8 fast scan rates. The CV response under illumination is slightly distorted under illumination,  
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10 especially around 0.2 and 1 V (see **Figure 2a-d**), which may be due to photocatalytic  
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12 decomposition of the aqueous electrolyte. This effect is strongly reduced when using 1M zinc  
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14 trifluoromethanesulfonate ( $\text{Zn}(\text{CF}_3\text{SO}_3)_2$ ) as an electrolyte (see **Figure S10**). **Figure 2e** shows  
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16 that both electrolytes offer a comparable performance, which again suggests that  $\text{g-C}_3\text{N}_4$  is  
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18 driving the light charging rather than side reactions.  
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22 The photo-charging principle of the Photo-ZICs can be explained from the energy level  
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24 diagram (**Figure S11a**). As outlined schematically in **Figure 1c**, the photogenerated electrons  
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26 transport from  $\text{g-C}_3\text{N}_4$  to FTO through rGO and finally accumulate on the Zn anode through  
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28 the external circuit. The photogenerated free holes attract anions (e.g.  $\text{SO}_4^{2-}$ ) towards the  $\text{g-}$   
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30  $\text{C}_3\text{N}_4$ @rGO cathode for adsorption on the surface, while the accumulated photogenerated  
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32 electrons on Zn anode may help the deposition of cations ( $\text{Zn}^{2+}$ ) from the electrolyte (e.g. 2 M  
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34  $\text{ZnSO}_4$ ). Overall, this process results in the photo-charging of Photo-ZICs under illumination.  
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38 The photo-charging process of the Photo-ZICs can be further confirmed from the light-  
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40 dependent chronoamperometry measurement *i.e.* current versus time plot at zero applied  
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42 voltage as shown in **Figure S11b**. It is observed that the current response ( $\Delta I = I_{light} - I_{dark}$ )  
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44 of the Photo-ZIC increases from  $0 \mu\text{A}$  to  $\sim 68 \mu\text{A}$  ( $\sim 136 \mu\text{Acm}^{-2}$ ) under light illumination ( $\lambda$   
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46  $\sim 420 \text{ nm}$ , intensity  $\sim 50 \text{ mWcm}^{-2}$ ) and returns to  $0 \mu\text{A}$  in dark conditions. The increase in the  
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48 current response of the Photo-ZIC under light illumination is due to the transportation of the  
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50 photogenerated electrons from cathode to anode through the external circuit for photo-charging  
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52 the Photo-ZIC at zero applied voltage. Moreover, to elucidate the contribution of the different  
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54 materials, we measured CV responses of rGO/FTO//Zn and activated carbon/FTO//Zn based  
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56 ZICs in dark and illuminated conditions (**Figure S12a,b**). As anticipated from the energy band  
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3 diagram (**Figure 1c**), these do not show any photo-charging responses. In addition, we have  
4 replaced rGO by a standard SuperP conductive additive (same wt%), however, these electrodes  
5 showed very poor photo-charging as shown in **Figure S12c**. We think this is possibly due to a  
6 combination of higher light absorption and poor interface between the g-C<sub>3</sub>N<sub>4</sub> and SuperP  
7 particles, and energy levels mismatch at the interface between the two materials.  
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11 The influence of the light wavelength on the photo-charging process is shown in **Figure 3a**  
12 (CV responses in **Figure S13**). As expected, negligible capacitance enhancements is observed  
13 at 528 nm, 470 nm and 455 nm excitation because the corresponding excitation energies of ~  
14 2.34 eV, ~ 2.63 eV and ~ 2.72 eV, respectively are lower than the optical band gap of g-C<sub>3</sub>N<sub>4</sub>  
15 (~2.75 eV). However, a significant capacitance enhancement of ~ 71% is recorded when  
16 exposed to 420 nm (~2.95 eV) due to the photogeneration of electrons and holes initiating the  
17 aforementioned photo-charge mechanism. We also confirmed that the absorption spectra of the  
18 Photo-ZICs does not change with the state of charge by mounting an optical ZIC cell in a UV-  
19 Vis with integrating sphere and analysing the reflection spectra in operando while cycling  
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**(Figure S14)**.



**Figure 3.** (a) Wavelength dependence capacitance enhancement of the Photo-ZIC device. Scan rate =  $200 \text{ mVs}^{-1}$ . (b) Intensity dependence capacitance enhancement of the Photo-ZIC at  $\lambda \sim 420 \text{ nm}$ . (c) Conventional charge-discharge at  $10 \text{ mA}g^{-1}$  and photo-charged ( $\lambda \sim 420 \text{ nm}$  at intensity  $\sim 20 \text{ mWcm}^{-2}$  and intensity  $\sim 50 \text{ mWcm}^{-2}$ ) and discharged at  $10 \text{ mA}g^{-1}$  in dark of the Photo-ZIC. (d) Photo-charged ( $\lambda \sim 420 \text{ nm}$ , intensity  $\sim 50 \text{ mWcm}^{-2}$ , red) and discharged at  $10 \text{ mA}g^{-1}$  in dark (black) and illuminated (dark yellow) conditions.

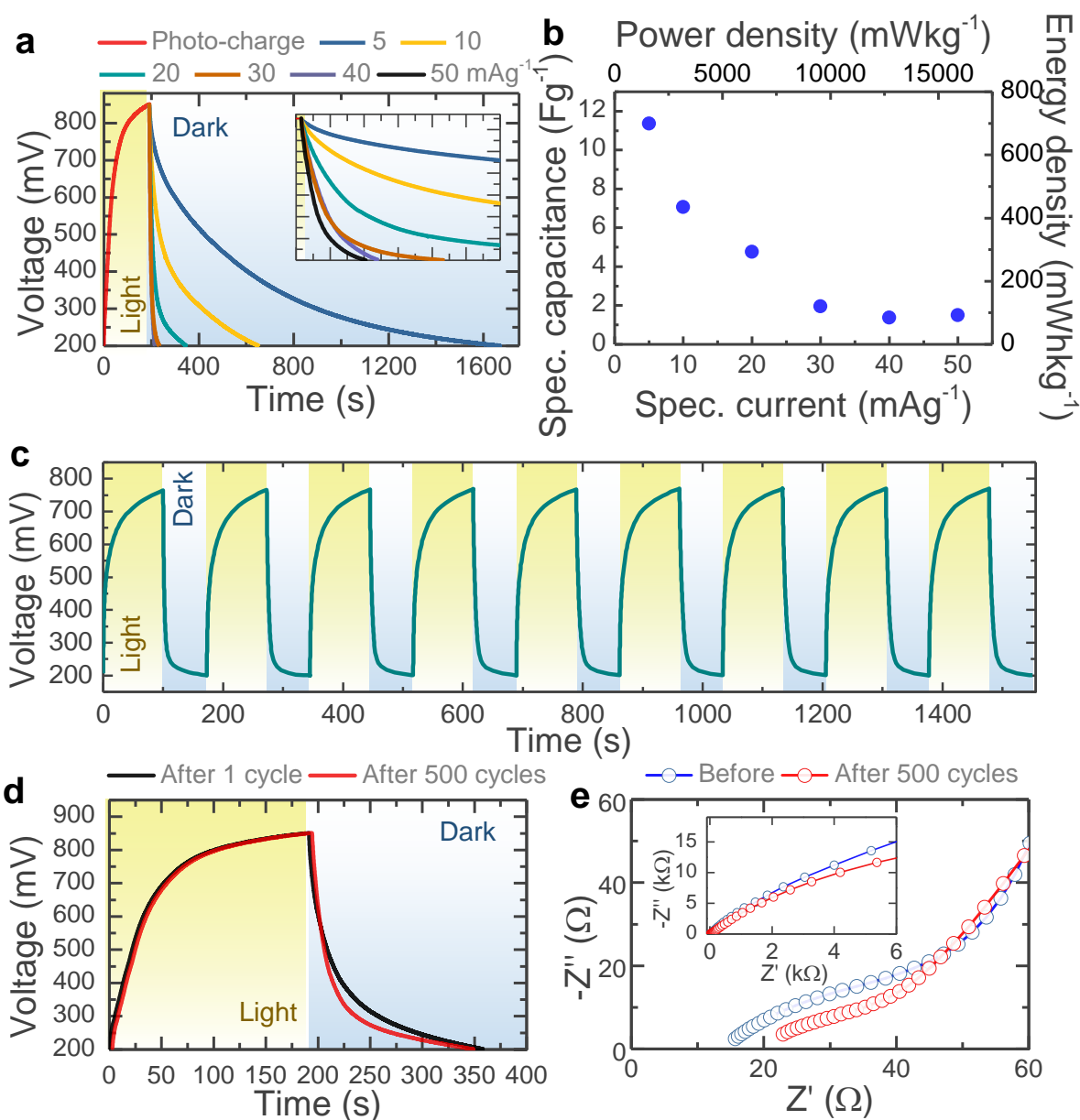
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3 **Figure 3b** shows the capacitance increases from ~ 31% to ~ 71% when increasing the light  
4 intensity from 20 mWcm<sup>-2</sup> to 50 mWcm<sup>-2</sup> (**Figure S15**, CV profiles). The increased intensity  
5 results in an increased rate of photocharge carrier generation thereby increasing the number of  
6 charge carriers available to contribute to the photo-charging mechanism and thus the observed  
7 capacitance enhancement. Next, we compare charging Photo-ZICs galvanostatically (current  
8 of 10 mA g<sup>-1</sup> and cut off voltage 1000 mV) with light charging using intensities of 20 mWcm<sup>-2</sup>  
9 and 50 mWcm<sup>-2</sup>. The latter converged to output voltages of approximately ~ 680 mV and ~  
10 850 mV, respectively after 200 s (see **Figure 3c**). The differences between the output voltage  
11 using galvanostatic and light charging is in part due to limitations in the penetration depth of  
12 light in the electrode. In addition, we expect that increasing the light intensity enhance the  
13 photocharge carrier generation efficiency, resulting in higher charging rates at higher light  
14 intensities. Finally, we observed that prolonged light charging time at 50 mWcm<sup>-2</sup> resulted in  
15 output voltages up to ~ 930 mV (see further). However, by comparison, the photo-charging  
16 voltage response (~ 850 mV) of our Photo-ZICs is comparable with the best reported an  
17 integrated photo-rechargeable supercapacitors with solar cells as summarized in **Table S1**. If  
18 the electrodes are continuously exposed to light while discharging at 10 mA g<sup>-1</sup>, the output  
19 voltage drops from ~ 850 mV to ~ 550 mV, but then remains constant as a balance is achieved  
20 between the photo-charging and discharge currents (see **Figure 3d**). When the light is  
21 subsequently turned off, the photo-ZIC output voltage drops as expected. Overall, this  
22 demonstrates the ability of the g-C<sub>3</sub>N<sub>4</sub> Photo-ZIC to operate simultaneously as an energy  
23 storage and photovoltaic device.

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52 **Figure 4a** shows photo-charge ( $\lambda \sim 420$  nm, intensity ~ 50 mWcm<sup>-2</sup>) followed by galvanostatic  
53 discharge cycles at specific currents of 5 mA g<sup>-1</sup> to 50 mA g<sup>-1</sup>, which are used to calculate the  
54 specific capacitances, energy densities and power densities shown in **Figure 4b**. Our photo-  
55 charged capacitances of up to 11377 mFg<sup>-1</sup> are comparable with values reported for  
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3 conventional solid-state supercapacitors.<sup>27,28</sup> Based on the photo-charging responses of the  
4 Photo-ZIC, we calculate the photo-charged energy density and power density of the Photo-ZIC  
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6 at different discharge specific currents. Photo-charged energy densities and power densities of  
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8 668 mWhkg<sup>-1</sup> at 1625 mWkg<sup>-1</sup> and 87.93 mWhkg<sup>-1</sup> at 16250 mWkg<sup>-1</sup> are found at discharge  
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10 specific currents of 5 mA g<sup>-1</sup> and 50 mA g<sup>-1</sup>. The photo-charging conversion efficiency ( $\eta$ ) of  
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12 the Photo-ZIC is  $\sim 0.01\%$  (see calculation in supporting information). This value is relatively  
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14 low compared to systems using energy storage devices (battery or supercapacitor) integrated  
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16 with solar cells (see **Table S1**).<sup>10,11</sup> Also the energy efficiency in charging and discharging is  
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18 still low at this stage, but we anticipate that this can be increased by tuning the electrode  
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20 composition and morphology.  
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27 The stability of the Photo-ZICs is investigated by repeated photo-charging ( $\lambda \sim 420$  nm, 50  
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29 mWcm<sup>-2</sup> and 100 s illumination) followed by galvanostatic discharging (to 200 mV) as shown  
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31 in **Figure 4c**. Repeatable photo-charging (negligible capacity loss  $\sim 1.4\%$ ) is observed during  
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33 the cycling test for nine cycles. We measured the self-discharge rate of a photo-charged Photo-  
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35 ZIC device (see **Figure S16a**), which shows a  $\sim 17\%$  voltage loss after 20 hours, this value is  
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37 comparable or even lower than other supercapacitors reported in literature.<sup>29-33</sup> Also, the  
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39 floating voltage of the devices was measured under prolonged illumination (3000 s) as shown  
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41 in **Figure S16b**. To further test the long-term cycle stability of the Photo-ZIC, the cell was  
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43 cycled for 500 cycles in dark conditions (**Figure S17a**), after which a low capacitance  
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45 degradation of  $\sim 4.7\%$  over 500 cycles is observed in the Photo-ZIC as shown in **Figure S17b**.  
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47 After 500 galvanostatic cycles, we photo-charged ( $\lambda \sim 420$  nm, intensity  $\sim 50$  mWcm<sup>-2</sup>) the  
48  
49 Photo-ZIC and discharged at 20 mA g<sup>-1</sup>, which shows a negligible loss of capacity as shown in  
50  
51 **Figure 4d**. Similarly, we cycled a cell for 1000 CV scans at 500 mVs<sup>-1</sup> under illumination ( $\lambda$   
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53  $\sim 420$  nm, intensity  $\sim 50$  mWcm<sup>-2</sup>) and found a capacitance loss of 6.2% after 500 cycles and  
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55 10.8% after 1000 cycles (see **Figure S18**). Likewise, **Figure 4e** shows the AC impedance  
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3 spectra of the Photo-ZIC before and after 500 charge-discharge cycles, where the equivalent  
4 series resistance increases from 16  $\Omega$  to 21.8  $\Omega$ . In addition, no notable changes in the photo-  
5 cathode morphology are observed using post-mortem SEM analysis of the cathode after 1000  
6 CV cycles (see **Figure S19a,b**). Similarly post-mortem SEM imaging of the Zn anode shows  
7 a slight increase in surface roughness, which is probably due to cations deposition and stripping  
8 during charge and discharge (**Figure S19c,d**). Moreover, PL spectra of g-C<sub>3</sub>N<sub>4</sub>@rGO/FTO  
9 cathodes before and after cycling show the same emission spectra positions centred at  $\lambda \sim 470$   
10 nm, 492 nm and 524 nm with only a subtle increase from the emission at 524 nm (**Figure**  
11 **S19e**). Overall, this confirms that the optical properties of the electrode are not altered  
12 significantly by long-term cycling ( $\sim 1000$  illuminated cycles) and that the optical properties  
13 are maintained.  
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**Figure 4.** (a) Photo-charge ( $\lambda \sim 420$  nm, intensity  $\sim 50$  mWcm $^{-2}$ ) and discharge cycles at different specific currents of 5 mA $g^{-1}$ , 10 mA $g^{-1}$ , 20 mA $g^{-1}$ , 30 mA $g^{-1}$ , 40 mA $g^{-1}$  and 50 mA $g^{-1}$ . (b) Photo-charged specific capacitance, energy density and power density plots as a function of discharge specific currents of the Photo-ZIC. (c) Cyclic photo-charge ( $\lambda \sim 420$  nm, intensity  $\sim 50$  mWcm $^{-2}$ ) and discharge cycles at a constant specific current. (d) Comparison plot of 1<sup>st</sup> photo-charge and after 500 charge-discharge cycles. (e) Nyquist plot of the Photo-ZIC before and after 500 charge-discharge cycles.

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3 This study demonstrates the first photo-rechargeable ZIC, which uses g-C<sub>3</sub>N<sub>4</sub> as active material  
4 to achieve direct charging by light without using any external photovoltaic devices. These ZICs  
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6 achieve photo-charged specific capacitances of up to 11377 mFg<sup>-1</sup> and retain ~ 90% of their  
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8 capacity over 1000 cycles. Because of the simple device architecture compared to alternative  
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10 solutions using a combination of photovoltaics and energy storage systems, we anticipate that  
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12 this technology could be of interest for powering ultra-low power compact off-grid devices, for  
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14 instance in IoT applications.  
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## 20 **Experimental Section**

21 **Chemicals.** Urea, Zn(CF<sub>3</sub>SO<sub>3</sub>)<sub>2</sub>, N-Methyl-2-pyrrolidone (NMP) and 2 M ZnSO<sub>4</sub> solution are  
22 received from Sigma-Aldrich. Graphene oxide (GO, 0.4 wt% in water) solution and activated  
23 carbon were purchased from Graphenea and Alfa Aesar. The epoxy (EVO-STIK) and  
24 polyvinylidene fluoride (PVDF, Solef 6020) binder are received from Arco Ltd and Solvay.  
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31 **Synthesis of g-C<sub>3</sub>N<sub>4</sub>.** Commercially available urea was used as a precursor of g-C<sub>3</sub>N<sub>4</sub>. First, 5  
32 g urea was dried at 70 °C for 5 h in air, then transferred into a crucible and covered with a lid.  
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34 The crucible was placed into a atmospheric pressure muffle furnace, then the furnace  
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36 temperature was increased to 530 °C and maintained for another 3 h. During this step, stable  
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38 g-C<sub>3</sub>N<sub>4</sub> is formed. Thereafter the furnace was cooled down to room temperature and the yellow  
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40 product was then collected and washed with deionised water and ethanol followed by vacuum  
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42 filtration process. Final product was dried at 120 °C for 12 h in a vacuum oven.  
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48 **Synthesis of rGO.** GO was collected from graphene oxide solution by drying at 70 °C for 24  
49 h. As obtained GO was reduced at 350 °C in helium gas (100 sccm, used as a carrier gas) and  
50 hydrogen gas (100 sccm) environment for 3 h using tubular CVD furnace.  
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56 **Materials Characterizations.** SEM images are taken using a FEI Magellan 400L with an  
57 acceleration voltage of 5 kV. XRD patterns are collected using a Bruker D8 Advance (Cu Kα  
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radiation). BET surface area characterization is measured using a Micromeritics 3Flex under nitrogen environment and TGA is performed using a PerkinElmer Pyris1 instrument machine in air. FTIR measurement is recorded using a PerkinElmer Frontier FT-IR Spectrometer.

**Preparation of Electrodes.** First, 2 mg rGO was dissolved into 2 ml NMP followed by ultrasonication process for 5 h. Thereafter, 93 mg g-C<sub>3</sub>N<sub>4</sub> was added in the same rGO solution and stirred for 12 h. Finally, the electrode solution was obtained by adding 5 mg PVDF binder followed by stirring another 5 h. 10  $\mu$ l electrode solution was drop casted on FTO coated glass substrate (received from Sigma-Aldrich, surface resistivity  $\sim 7 \Omega\text{sq}^{-1}$ ) and dried overnight at 70  $^{\circ}\text{C}$ .

**Three-Electrode Measurement.** CV measurements of the rGO/FTO, activated carbon/FTO and g-C<sub>3</sub>N<sub>4</sub>@rGO/FTO electrodes are recorded using Ag/AgCl as a reference electrode and Pt wire as a reference electrode both in dark and illuminated conditions at different scan rates to confirm the photosensitivity of the electrodes.

**Fabrication of Photo-ZIC.** Photo-ZICs are assembled in air. First, g-C<sub>3</sub>N<sub>4</sub>@rGO/FTO photo-cathode with a double sided copper tape for electrical connectivity was fixed on the side of coin cell can having optical window ( $\sim 8$  mm) using EPOXY. The Whatman glass microfiber filters paper separator is placed on the photo-cathode, then 100  $\mu$ l of electrolyte (ZnSO<sub>4</sub> or Zn(CF<sub>3</sub>SO<sub>3</sub>)<sub>2</sub>) is added on the separator. On the top side of the separator, a Zn metal foil (Alfa Aesar, 0.25 mm thick) with a stainless steel disk spacer are placed. Finally, the optical coin cells are assembled followed by standard procedure.

**Characterization of Photo-ZIC.** All Photo-ZICs capacitive measurements were employed by using a Biologic VMP-3 galvanostat in air. The photo-charging performances are measured by illuminating light of wavelength of 420 nm (intensities  $\sim 50 \text{ mWcm}^{-2}$  and  $\sim 20 \text{ mWcm}^{-2}$ ). Open circuit voltages of the Photo-ZICs are recorded to obtain the photo-charging voltage response,

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3 where photo-charged ZICs are discharged by applying different specific currents. Moreover,  
4 the CV responses of the Photo-ZICs are tested in dark and illuminated of different light  
5 wavelengths of 455 nm (intensity  $\sim 13.75 \text{ mWcm}^{-2}$ ), 470 nm (intensity  $\sim 43.75 \text{ mWcm}^{-2}$ ), 528  
6 nm (intensity  $\sim 7.5 \text{ mWcm}^{-2}$ ) and 420 nm (intensity  $\sim 50 \text{ mWcm}^{-2}$ ), respectively over the  
7 working voltage range of 0.2 to 1 V. This working voltage of the Photo-ZIC is optimized to  
8 obtain a stable surface related charge storage capacitance of the g-C<sub>3</sub>N<sub>4</sub> and avoid any gas  
9 evolution reaction.  
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20 **Photoluminescence Spectroscopy.** PL spectra were obtained using a confocal microscope  
21 setup in reflection (WITec alpha 300s). A 405 nm continuous wave (CW) laser was used as the  
22 excitation source with an incident power of 9.2  $\mu\text{W}$ . The PL spectra shown represent the  
23 average of 3400 individual spectra obtained from a scanned area of 50  $\mu\text{m} \times 50 \mu\text{m}$ .  
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30 **Time-Correlated Single Photon Counting (TCSPC).** Charge carrier lifetimes were collected  
31 using TCSPC spectroscopy. A 50  $\mu\text{W}$  PicoQuant LDH405 pulsed laser with  $\lambda = 407 \text{ nm}$  and  
32 pulse length of 100 ps was used as an excitation source. The PL emission was collected at 90°  
33 and passed through a monochromator to select the collection wavelength (500 nm).  
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40 **In-situ Optical Characterisation.** The in-situ reflectance spectra of the Photo-ZIC cell were  
41 recorded at different charge and discharge states using PerkinElmer UV/Vis/NIR Spectrometer  
42 (Lamda 750) in reflection mode.  
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## 47 **Supporting Information**

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49 The supporting information is available

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53 XRD, FTIR, BET and TGA of g-C<sub>3</sub>N<sub>4</sub> (Figure S1-S4); SEM, XRD and Raman of GO and rGO  
54 (Figure S5); SEM of photo-cathode (Figure S6); CV responses of g-C<sub>3</sub>N<sub>4</sub>@rGO/FTO,  
55 rGO/FTO and activated carbon/FTO electrodes (Figure S7,S8); digital photograph and CV  
56 profiles of Photo-ZIC (Figure S9); CV profiles in 1M Zn(CF<sub>3</sub>SO<sub>3</sub>)<sub>2</sub> electrolyte (Figure S10);  
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3 photo-charging mechanism and cyclic current response (Figure S11); CV curves of  
4 rGO/FTO//Zn, activated carbon/FTO//Zn and g-C<sub>3</sub>N<sub>4</sub>@SuperP/FTO//Zn based ZICs (Figure  
5 S12); CV responses at different light illuminated wavelengths (Figure S13); in operando  
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7 S12); CV responses at different light illuminated wavelengths (Figure S13); in operando  
8 optical characterisation (Figure S14); intensity dependence CV responses (Figure S15);  
9  
10 performance comparative table (Table S1); self-discharge and voltage floating test (Figure  
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12 S16); long-term cycling test (Figure S17); comparative CVs at different cycles (Figure S18),  
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14 post-mortem SEM and PL analysis before and after cycling (Figure S19); photo-charge  
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16 conversion efficiency calculation.  
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33 MSCA-IF 796648.  
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## 37 **References**

- 38  
39  
40 1. Lethien, C.; Bideaum, J. L.; Brouss, T. Challenges and prospects of 3D micro-  
41  
42 supercapacitors for powering the internet of things. *Energy Environ. Sci.* **2019**, *12*, 96-  
43  
44 115.  
45  
46 2. Raj, A.; Steingart, D. Review—Power Sources for the Internet of Things. *J.*  
47  
48 *Electrochem. Soc.* **2018**, *165*, B3130.  
49  
50 3. Zhan, Y.; Mei, Y.; Zheng, L. Materials capability and device performance in flexible  
51  
52 electronics for the Internet of Things. *J. Mater. Chem. C* **2014**, *2*, 1220-1232.  
53  
54 4. Um, H. -D.; Choi, K. -H.; Hwang, I.; Kim, S. -H.; Seo, K.; Lee, S. -Y. Monolithically  
55  
56 integrated, photo-rechargeable portable power sources based on miniaturized Si solar  
57  
58  
59  
60

1  
2  
3 cells and printed solid-state lithium-ion batteries. *Energy Environ. Sci.* **2017**, *10*, 931-  
4  
5 940.

6  
7  
8 5. Du, P.; Hu, X.; Yi, C.; Liu, H. C.; Liu, P. Self-Powered Electronics by Integration of  
9  
10 Flexible Solid-State Graphene-Based Supercapacitors with High Performance  
11  
12 Perovskite Hybrid Solar Cells. *Adv. Funct. Mater.* **2015**, *25*, 2420-2427.

13  
14  
15 6. Li, C.; Cong, S.; Tian, Z.; Song, Y.; Yu, L.; Lu, C.; Shao, Y.; Li, J.; Zou, G.;  
16  
17 Rümmele, M. H.; Dou, S. Flexible perovskite solar cell-driven photo-rechargeable  
18  
19 lithium-ion capacitor for self-powered wearable strain sensors. *Nano Energy* **2019**, *60*,  
20  
21 247-256.

22  
23  
24 7. Vlad, A.; Singh, N.; Galande, C.; Ajayan, P. M. Design Considerations for  
25  
26 Unconventional Electrochemical Energy Storage Architectures. *Adv. Energy Mater.*  
27  
28 **2015**, *5*, 1402115.

29  
30  
31 8. Ahmad, S.; George, C.; Beesley, D. J.; Baumberg, J. J.; Volder, M. D. Photo-  
32  
33 Rechargeable Organo-Halide Perovskite Batteries. *Nano Lett.* **2018**, *18*, 1856-1862.

34  
35  
36 9. Zhou, F.; Ren, Z.; Zhao, Y.; Shen, X.; Wang, A.; Li, Y. Y.; Surya, C.; Chai, Y.  
37  
38 Perovskite photovoltachromic supercapacitor with all-transparent electrodes. *ACS*  
39  
40 *Nano* **2016**, *10*, 5900-5908.

41  
42  
43 10. Liu, R.; Wang, J.; Sun, T.; Wang, M.; Wu, C.; Zou, H.; Song, T.; Zhang, X.; Lee,  
44  
45 S. T.; Wang, Z. L.; Sun, B. Silicon nanowire/polymer hybrid solar cell-supercapacitor:  
46  
47 a self-charging power unit with a total efficiency of 10.5%. *Nano Lett.* **2017**, *17*, 4240-  
48  
49 4247.

50  
51  
52 11. Xu, J.; Chen, Y.; Dai, L. Efficiently photo-charging lithium-ion battery by  
53  
54 perovskite solar cell. *Nat. Commun.* **2015**, *6*, 8103.

- 1  
2  
3 12. Liang, J.; Zhu, G.; Wang, C.; Wang, Y.; Zhu, H.; Hu, Y.; Lv, H.; Chen, R.; Ma, L.;  
4  
5 Chen, T.; Jin, Z. MoS<sub>2</sub>-based all-purpose fibrous electrode and self-powering energy  
6  
7 fiber for efficient energy harvesting and storage. *Adv. Energy Mater.* **2017**, *7*, 1601208.  
8  
9  
10 13. Liang, J.; Zhu, G.; Lu, Z.; Zhao, P.; Wang, C.; Ma, Y.; Xu, Z.; Wang, Y.; Hu, Y.;  
11  
12 Ma, L.; Chen, T. Integrated perovskite solar capacitors with high energy conversion  
13  
14 efficiency and fast photo-charging rate. *J. Mater. Chem. A* **2018**, *6*, 2047.  
15  
16  
17 14. Liang, J.; Zhu, G.; Wang, C.; Zhao, P.; Wang, Y.; Hu, Y.; Ma, L.; Tie, Z.; Liu, J.;  
18  
19 Jin, Z. An all-inorganic perovskite solar capacitor for efficient and stable spontaneous  
20  
21 photocharging. *Nano Energy* **2018**, *52*, 239.  
22  
23  
24 15. Paoletta, A.; Faure, C.; Bertoni, G.; Marras, S.; Guerfi, A.; Darwiche, A.;  
25  
26 Hovington, P.; Commariou, B.; Wang, Z.; Prato, M.; Colombo, M.; Monaco, S.; Zhu,  
27  
28 W.; Feng, Z.; Vijn, A.; George, C.; Demopoulos, G. P.; Armand, M.; Zaghbi, K. Light-  
29  
30 assisted delithiation of lithium iron phosphate nanocrystals towards photo-rechargeable  
31  
32 lithium ion batteries. *Nat. Commun.* **2017**, *8*, 14643.  
33  
34  
35 16. Nakhanivej, P.; Yu, X.; Park, S. K.; Kim, S.; Hong, J. Y.; Kim, H. J.; Lee, W.;  
36  
37 Hwang, J. Y.; Yang, J. E.; Wolverton, C.; Kong, J. Revealing molecular-level surface  
38  
39 redox sites of controllably oxidized black phosphorus nanosheets. *Nature Mater.* **2019**,  
40  
41 *18*, 156.  
42  
43  
44 17. Hou, R.; Gund, G. S.; Qi, K.; Nakhanivej, P.; Liu, H.; Li, F.; Xia, B. Y.; Park, H. S.  
45  
46 Hybridization design of materials and devices for flexible electrochemical energy  
47  
48 storage. *Energy Storage Mater.* **2019**, *19*, 212.  
49  
50  
51 18. Lee, A.; Vörös, M.; Dose, W. M.; Niklas, J.; Poluektov, O.; Schaller, R. D.; Iddir,  
52  
53 H.; Maroni, V. A.; Lee, E.; Ingram, B.; Curtiss, L. A. Photo-accelerated fast charging  
54  
55 of lithium-ion batteries. *Nat. Commun.* **2019**, *10*, 4946.  
56  
57  
58  
59  
60

- 1  
2  
3 19. Boruah, B. D.; Misra, A. Voltage Generation in Optically Sensitive Supercapacitor  
4 for Enhanced Performance. *ACS Appl. Energy Mater.* **2019**, *2*, 278-286.  
5  
6  
7  
8 20. Nguyen, O.; Courtin, E.; Sauvage, F.; Krins, N.; Sanchez, C.; Laberty-Robert, C.  
9 Shedding light on the light-driven lithium ion de-insertion reaction: towards the design  
10 of a photo-rechargeable battery. *J. Mater. Chem. A* **2017**, *5*, 5927-5933.  
11  
12  
13 21. Zhang, P.; Li, Y.; Wang, G.; Wang, F.; Yang, S.; Zhu, F.; Zhuang, X.; Schmidt, O.  
14 G.; Feng, X. Zn-ion hybrid micro-supercapacitors with ultrahigh areal energy density  
15 and long-term durability. *Adv. Mater.* **2019**, *31*, 1806005.  
16  
17  
18 22. Sun, G.; Yang, H.; Zhang, G.; Gao, J.; Jin, X.; Zhao, Y.; Jiang, L.; Qu, L. A capacity  
19 recoverable zinc-ion micro-supercapacitor. *Energy Environ. Sci.* **2018**, *11*, 3367-3374.  
20  
21  
22 23. Zhang, X.; Pei, Z.; Wang, C.; Yuan, Z.; Wei, L.; Pan, Y.; Mahmood, A.; Shao, Q.;  
23 Chen, Y. Flexible Zinc-Ion Hybrid Fiber Capacitors with Ultrahigh Energy Density and  
24 Long Cycling Life for Wearable Electronics. *Small* **2019**, *15*, 1903817.  
25  
26  
27 24. Wang, H.; Wang, M.; Tang, Y. A novel zinc-ion hybrid supercapacitor for long-life  
28 and low-cost energy storage applications. *Energy Storage Mater.* **2018**, *13*, 1-7.  
29  
30  
31 25. Yuan, Y.; Zhang, L.; Xing, J.; Utama, M. I. B.; Lu, X.; Du, K.; Li, Y.; Hu, X.;  
32 Wang, S.; Genç, A.; Dunin-Borkowski, R. High-yield synthesis and optical properties  
33 of g-C<sub>3</sub>N<sub>4</sub>. *Nanoscale* **2015**, *7*, 12343-12350.  
34  
35  
36 26. Wang, J.; Yamada, Y.; Sodeyama, K.; Watanabe, E.; Takada, K.; Tateyama, Y.;  
37 Yamada, A. Fire-extinguishing organic electrolytes for safe batteries. *Nat. Energy*  
38 **2018**, *3*, 22.  
39  
40  
41 27. Boruah, B. D.; Misra, A. A flexible ternary oxide based solid-state supercapacitor  
42 with excellent rate capability. *J. Mater. Chem. A* **2016**, *4*, 17552-17559.  
43  
44  
45  
46  
47  
48  
49  
50  
51  
52  
53  
54  
55  
56  
57  
58  
59  
60

- 1  
2  
3 28. Boruah, B. D.; Misra, A. Internal Asymmetric Tandem Supercapacitor for High  
4 Working Voltage along with Superior Rate Performance. *ACS Energy Lett.* **2017**, *2*,  
5 1720-1728.  
6  
7  
8  
9  
10 29. Chen, L.; Bai, H.; Huang, Z.; Li, L. Mechanism investigation and suppression of  
11 self-discharge in active electrolyte enhanced supercapacitors. *Energy Environ. Sci.*  
12 **2014**, *7*, 1750-1759.  
13  
14  
15  
16 30. Zhang, Q.; Cai, C.; Qin, J.; Wei, B. Tunable self-discharge process of carbon  
17 nanotube based supercapacitors. *Nano Energy* **2014**, *4*, 14-22.  
18  
19  
20 31. Xiong, T.; Yu, Z. G.; Lee, W. S. V.; Xue, J. o-Benzenediol-Functionalized Carbon  
21 Nanosheets as Low Self-Discharge Aqueous Supercapacitors. *ChemSusChem* **2018**, *11*,  
22 3307-3314.  
23  
24  
25  
26 32. Lahe<sup>ˆ</sup>a<sup>ˆ</sup>ar, A.; Arenillas, A.; B<sup>ˆ</sup>egu<sup>ˆ</sup>in, F. Change of self-discharge mechanism as a  
27 fast tool for estimating long-term stability of ionic liquid based supercapacitors. *J.*  
28 *Power Sources* **2018**, *396*, 220-229.  
29  
30  
31  
32 33. Xia, M.; Nie, J.; Zhang, Z.; Lu, X.; Wang, Z. L. Suppressing self-discharge of  
33 supercapacitors via electrorheological effect of liquid crystals. *Nano Energy* **2018**, *47*,  
34 43-50.  
35  
36  
37  
38  
39  
40  
41  
42  
43  
44  
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