

Advanced Functional Materials

Cell Encapsulation Systems Towards Modular Tissue Regeneration: From Immunoisolation to Multifunctional Devices

--Manuscript Draft--

Manuscript Number:	adfm.201908061R2
Article Type:	Invited Review
Corresponding Author:	João F. Mano, Prof. University of Aveiro, Portugal Aveiro, Aveiro PORTUGAL
Corresponding Author E-Mail:	jmano@ua.pt
Order of Authors:	Clara Correia Sara Nadine João F. Mano, Prof.
Keywords:	cell encapsulation, hydrogels, building blocks, modular tissue engineering, tissue regeneration
Manuscript Classifications:	BIOMEDICAL ENGINEERING - biomaterials, biointerfaces, tissue engineering, regenerative medicine, bioelectronic devices, biosensors, biofabrication, biotechnology
Section/Category:	By Invitation Only: Assembly of Materials Building Blocks into Integrated Complex Functional Systems
Abstract:	<p>In the primordial cell encapsulation systems, the main goal was to treat endocrine diseases avoiding the action of the immune system. Although lessons afforded by such systems were of outmost importance for the demands of Tissue Engineering and Regenerative Medicine, the paradigm has recently completely changed. If before the most important feature was to mask the encapsulated cells from the immune system, now it is known that the synergetic interplay between immune cells and the engineered niche is responsible by an adequate regenerative process. Combined with such immuno-awareness, novel or non-conventional emerging techniques are being proposed developed the new generation of cell encapsulation systems, namely layer-by-layer, microfluidics, superhydrophobic surfaces, and bioprinting technologies. Alongside with the desire to create more realistic cell encapsulation systems, cell-laden hydrogels are being explored as building blocks for bottom-up strategies, within the concept of modular tissue engineering. The idea is to use the well-established cell friendly environment provided by hydrogels, and create more close-to-native systems owning high heterogeneity, while providing multifunctional and adaptive inputs.</p>
Author Comments:	
Additional Information:	
Question	Response
Please submit a plain text version of your cover letter here.	<p>Dear editor Dr. Jos Lenders,</p> <p>Attached you can find your revised version of the manuscript entitled "Cell Encapsulation Systems Towards Modular Tissue Regeneration: From Immunoisolation to Multifunctional Devices". All the minor changes required by the reviewer were performed, which are highlighted in the manuscript in yellow.</p> <p>I am fully available to provide any further information you may need.</p> <p>With my very best regards,</p> <p>João F. Mano Full Professor at University of Aveiro, Portugal</p>

Do you or any of your co-authors have a conflict of interest to declare?	No. The authors declare no conflict of interest.
Response to Reviewers:	<p>Response to the reviewer</p> <p>Dear reviewer,</p> <p>Please find bellow our responses to the suggested minor modifications.</p> <p>Kind regards,</p> <p>The authors</p> <p>Reviewer #1: In general, this article highlights the significance of immunomodulatory cell encapsulation approaches for tissue engineering and discusses the challenges and limitations of these cell encapsulation techniques. In this revised manuscript, the authors have addressed all the comments raised by the previous reviewers and included the additional texts to make it clearer. Thus, I would like to recommend the publication of this work. Some minor comments are as follow.</p> <ol style="list-style-type: none"> 1. In page 9, line 30, please omit 'the' after customized. Authors response: Misspelling corrected. 2. In page 21, line 52 'the prolonged presence of M1 macrophages can induce the release of fibrosis-enhancing cytokine pattern by M2 macrophages' is confusing. Please elaborate a bit to make it clear. Authors response: As suggested, new sentences were added to clarify this topic. 3. In page 28, line 27, when you said that, 'it is noteworthy that most of monocytes and macrophages used for biocompatibility tests are derived from leukemia or lymphomas tumours', it would be clearer if you also mention cell types and sources. Authors response: Examples of immortalized cell lines were given, as suggested.

1 **Cell Encapsulation Systems Towards Modular Tissue Regeneration: From**
2 **Immunoisolation to Multifunctional Devices**

3
4 *Clara R. Correia**, *Sara Nadine*, and *João F. Mano**

5
6 Dr. C. R. Correia, Sara Nadine, Prof. João F. Mano
7 CICECO-Aveiro Institute of Materials, Department of Chemistry, Campus Universitário de
8 Santiago, 3810-193 Aveiro, Portugal
9 E-mails: claracorreia@ua.pt and jmano@ua.pt

10
11
12
13
14 **Keywords:** cell encapsulation, hydrogels, immunomodulation, tissue regeneration

15
16
17 In the primordial cell encapsulation systems, the main goal was to treat endocrine diseases
18 avoiding the action of the immune system. Although lessons afforded by such systems were of
19 utmost importance for the demands of Tissue Engineering and Regenerative Medicine, the
20 paradigm has recently completely changed. If before the most important feature was to mask
21 the encapsulated cells from the immune system, now it is known that the synergetic interplay
22 between immune cells and the engineered niche is responsible by an adequate regenerative
23 process. Combined with such immuno-awareness, novel or non-conventional emerging
24 techniques are being proposed developed the new generation of cell encapsulation systems,
25 namely layer-by-layer, microfluidics, superhydrophobic surfaces, and bioprinting technologies.
26
27 Alongside with the desire to create more realistic cell encapsulation systems, cell-laden
28 hydrogels are being explored as building blocks for bottom-up strategies, within the concept of
29 modular tissue engineering. The idea is to use the well-established cell friendly environment
30 provided by hydrogels, and create more close-to-native systems owning high heterogeneity,
31 while providing multifunctional and adaptive inputs.
32
33
34
35
36
37
38
39
40
41
42
43
44
45
46
47
48
49
50
51
52
53
54
55
56
57
58
59
60
61
62
63
64
65

1. Introduction

The primordial cell encapsulation systems were designed to treat endocrine diseases, while protecting and masking the encapsulated cells from the immune system.^[1,2] Aimed to provide long-lasting solutions for multiple cell dysfunctions,^[3-5] such immuneprivileged systems prompted the beginning of the bioartificial organs era with the creation of the first fully artificial pancreas.^[6] Another remarkable milestone, is the work of Lim and Sun ^[7] with the microencapsulation of pancreatic islet cells in alginate spherical hydrogels for transplantation in diabetic rats. Analyzing the historical root of hydrogels encapsulating cells, it is easy to understand why most of the studies reported in the literature are focused in the treatment of endocrine diseases, mainly diabetes mellitus through the encapsulation of pancreatic islets. This is driven by the high number of diabetic patients and the recognized benefits of cell encapsulation strategies.^[8] Although these studies did not foresee the applicability of each cell-containing microcapsule as building blocks for modular tissue engineering (TE), they belong to its historical roots. Put simply, the primordial cell encapsulation systems were cell-containing blocks packed to create an artificial structure with biological activity. The bioactive molecules released to target an endocrine disease were dependent on the type of cells encapsulated. However, such artificial biologically active structures lack heterogeneity, and thus do not reassemble tissue-like structures. Native tissues are composed by heterogeneous mixtures of cell phenotypes or morphologies (e.g. the particular case of cartilage tissue with only one cell type, chondrocytes, but with depth-dependent morphology). As such, although current strategies for creating modular tissues draw from the primordial cell encapsulation systems, the paradigm has significantly change with the evolution of the tissue engineering and regenerative medicine (TERM) field. One of the strategies of the TERM field is to create engineered tissues through the bottom-up assemble of microstructural functional units.^[9] This strategy, termed as

1 modular TE, is based on the concept that the replication of functional units may aid in the
2 reconstruction of the heterogeneity of native tissues, and thus accelerate tissue healing. As such,
3 when aiming tissue regeneration, cell encapsulation systems must be capable of facilitating the
4 reconstruction of this heterogeneity upon implantation into a lesion site. These modules can be
5 created in multiple ways,^[10] but the present review will focus on the use of hydrogels produced
6 by novel or non-conventional techniques, namely layer-by-layer (LbL), microfluidics,
7 superhydrophobic surfaces, and bioprinting. Hydrogels are the gold standard material of most
8 cell encapsulation systems proposed for TERM, mainly due to their highly hydrated 3D
9 environment, which resembles the tissue-like elasticity of the native extracellular matrix
10 (ECM), and maximizes the diffusion of essential molecules for cell survival. Moreover,
11 hydrogels allow mild processing conditions and can be easily functionalized to enhance cell-
12 material interactions.^[11-14] Despite the established applicability of hydrogels as cell
13 encapsulation systems for TE, it is only recently that researchers have begun to explore its
14 potential to modulate the immune response, with significant clinical impulse observed in the
15 past decade itself.^[15,16] Of note, such immunomodulation is completely distinct from the
16 classical long-term immunoprotective feature of the primordial cell encapsulation systems. It is
17 true that the encapsulation matrix mediates the interaction with the host environment, and it
18 may find great applicability during the acute inflammatory phase following any tissue damage.
19 But then, it is desirable to occur the degradation of the encapsulation matrix balanced with the
20 newly deposited ECM in order to promote a proper vascularization and tissue integration that
21 dictate the success of the biomaterial implanted. Therefore, immunoprotection is no longer a
22 requisite if tissue regeneration is aimed, but hydrogels with immunomodulatory properties
23 allow to control the immune response upon implantation while promoting tissue regeneration.
24 In the present review, we intend to bridge the path of cell encapsulation systems from
25 immunoisolation devices to treat endocrine diseases to multifunctional-engineered systems that

1 aim to stimulate the regeneration of damaged tissues. A brief discussion of this path will be
2
3 given to highlight the new immunomodulatory biomaterials being proposed, while emphasizing
4
5 the use of novel or non-conventional emerging techniques that are contributing to the
6
7 development of the next generation of such systems. Subsequently, we will highlight the
8
9 applicability of cell encapsulation systems for creating modular engineered tissues that can
10
11 effectively direct the formation of larger and clinically-relevant tissues using bottom-up TE
12
13 principles.
14
15
16
17
18
19

20 **2. “Open” vs. “closed” scaffolds: limitations, advantages and practical considerations**

21
22

23 Independently of the type of strategy used, all TERM strategies aim to regenerate living,
24
25 healthy, and functional tissues, either partially by tissue grafts or even a total replacement of a
26
27 severely damaged organ. In the perspective of using scaffolds for TERM, there are two main
28
29 strategies: (i) scaffolds with adhered cells that contact directly with the host environment, here
30
31 termed as “open” scaffolds, or (ii) scaffolds at which core cells are encapsulated, here termed
32
33 as “closed” scaffolds that comprise cell encapsulation strategies (**Figure 1**). The designation of
34
35 “open” or “closed” scaffolds is thus respectively related to the direct contact or isolation
36
37 between cells and the surrounding environment. In “open” scaffolds the production methods
38
39 are not limited by the presence of cells, thus can include the use of precursors and harsh solvents
40
41 and/or reactants, as long as the cytocompatibility of the obtained scaffold is assured, including
42
43 its degradation products. On the contrary, the production methods of “closed” scaffolds must
44
45 ensure mild conditions since cells are present before the processing of the biomaterial. This
46
47 significantly impairs the available spectrum of technologies and biomaterials that can be applied
48
49 to produce closed scaffolds. Nevertheless, “closed” scaffolds can offer several advantages
50
51 compared to “open” scaffolds. “Closed” scaffolds (i) enable the creation of privileged and
52
53
54
55
56
57
58
59
60
61
62
63
64
65

1 controlled microenvironment for cells; (ii) allow minimal invasive implantation by *in situ*
2 injection, and thus the hydrogel can be easily fixed into injured sites with variable geometries
3 without requiring glue or sutures;^[17] and (iii) facilitate the incorporation within a single
4 structure of multiple compartments or elements with distinct functions, which confers
5 multifunctionality to the engineered system. Additionally, “closed” scaffolds can also
6 maximize cell interactions (Figure 1-red dots) by allowing direct cell-cell contact in the
7 particular case of liquefied systems.
8
9

20 **3. Critical properties of hydrogels for cell encapsulation**

21
22
23 The application of cell encapsulation principles to be used in the regeneration of tissues brought
24 several advantages to the TERM field as compared to conventional strategies that mainly use
25 porous scaffolds. Knowing that the key element in TERM is to combine cells with instructive
26 biomaterials to ultimately regenerate damaged tissues, it is easy to understand that the function
27 of the engineered device is not limited to the protection of cells, as it is its main function when
28 cell encapsulation is applied to treat endogenous diseases. Therefore, the development of cell
29 encapsulation devices that aim to promote tissue regeneration should be carefully pondered
30 before its conception, in order to meet all the complex requirements of the field.
31
32
33
34
35
36
37
38
39
40
41
42
43
44

45 **3.1 Mild and sterile conditions**

46
47
48 The selection of the type of methodology to produce cell encapsulation systems,^[18] as well as
49 the type the biomaterials that compose the encapsulation matrix is restrained by the imperative
50 preservation of the viability of the encapsulated cells. Allied to a number of appealing features,
51 such as their structural resemblance to many natural biological tissues, viscoelasticity, and high-
52 water content, hydrogels have become the most favourable material used for cell encapsulation.
53
54
55
56
57
58
59

1 Since the first hydrogel application in cell encapsulation when Lim and Sun developed calcium
2 alginate microcapsules for islet encapsulation,^[19] both synthetic and naturally derived cell
3 encapsulation matrixes have been developed. Most of the previous research has been focused
4 on natural-derived hydrogels, such as alginate, chitosan, pectin, agarose, gelatin, collagen,
5 fibrin, hyaluronic acid, and gellan gum.^[20] Their main advantages are: (i) their resemblance
6 with the ECM of native tissues, originating in most cases native-like responses under
7 physiological conditions (e.g. biodegradation), (ii) abundance, including the vast low-cost
8 resources provided by the marine environment,^[21] and (iii) their ability to produce hydrogels at
9 mild condition. In particular, some natural polymers have an intrinsic cell adhesion ability due
10 to the presence of cell-binding domains, and thus the functionalization of the encapsulation
11 matrix to allow the adhesion of cells is not required;^[22] although this is not applicable for the
12 most widely used natural polymer in cell encapsulation, namely alginate. On the other hand,
13 the main limitations of natural-derived hydrogels are the batch-to-batch variations, and the
14 presence of impurities and contaminants. As an alternative, synthetic polymers have been
15 employed, such as poly(ethylene glycol) (PEG), and poly(vinyl alcohol), which are commonly
16 used in combination with natural polymers to produce hydrogels for cell encapsulation.^{[18,20,23–}
17
18
19
20
21
22
23
24
25
26
27
28
29
30
31
32
33
34
35
36
37
38
39
40
41
42
43
44
45
46
47
48
49
50
51
52
53
54
55
56
57
58
59
60
61
62
63
64
65

1 carrageenan,^[27] and calcium carbonate/D-glucono- δ -lactone for pectin hydrogels.^[28]

3.2 Permeability and mass transfer

2
3
4
5
6
7
8
9 Once processed and formed, the cell encapsulation matrix will exhibit different transport
10 properties depending on its structure, chemical composition and, in the particular case of
11 hydrogels, the degree and type of crosslinking. Only an adequate mass transport will allow an
12 efficient permeability of essential molecules through the entire matrix, thus ensuring the
13 viability of encapsulated cells. Permeability and mass transfer can thus dictate the successful
14 of cell encapsulation strategies. If desired, either by the presence of a protective enveloping
15 membrane or by tuning the permeability properties of the matrix itself, the entrance of immune
16 cells or resultant redox molecules can be blocked. In fact, even in the absence of a membrane,
17 the hydrogel matrix can act as a mechanical and/or chemical barrier towards in- and out-flowing
18 molecules. The efficiency of the barrier is application dependent, as well as being intrinsically
19 connected with other parameters such as the stability of the cell encapsulation matrix, as further
20 discussed. Nevertheless, in all the encapsulation strategies proposed for tissue regeneration,
21 while the interaction with immune cells can be controlled (see section **4. Immunomodulation**
22 **in cell encapsulation systems**), they all must ensure an appropriate exchange of essential
23 molecules for cell survival, such as nutrients, oxygen, metabolites, and waste products.
24 Additionally, also the exchange of important signalling biomolecules must be ensured, either
25 between the encapsulated cells (inward diffusion) or between encapsulated and neighbouring
26 cells (outward diffusion). Additionally, in order to promote the integration of the implanted cell
27 encapsulation device with the host tissue, it is desirable that the device should be permeable to
28 the in-growth of blood vessels, except when regenerating avascular tissues such as cartilage.
29 Therefore, either existing or not the presence of an external membrane, the permeability of
30
31
32
33
34
35
36
37
38
39
40
41
42
43
44
45
46
47
48
49
50
51
52
53
54
55
56
57
58
59
60
61
62
63
64
65

1 essential molecules ensuring cell survival across the entire 3D structure is of great importance
2
3 in the design of the encapsulation matrix. For the particular case of modular TE, the interstitial
4
5 void spaces between the hydrogel building blocks can facilitate the diffusion exchange by
6
7 facilitating the ingrowth of the recruited blood vessels. It is important to highlight that the
8
9 diffusion of essential molecules for cell survival will have a delay when comparing
10
11 encapsulated cells localized in the core with the ones in the border of the matrix. The same is
12
13 applied for hydrogel blocks localized in inner regions of the 3D construct. This directly
14
15 influences the size and/or geometry of the building blocks. Consequently, the majority of cell
16
17 encapsulation building blocks are typically limited to a diameter of 400 μm , since 200 μm is
18
19 reported as the maximum diffusion distance of oxygen and nutrients from blood vessels to
20
21 cells.^[29] With the increasing techniques to fabricate innovative and complex hydrogel-based
22
23 systems for modular TE, other geometries have been progressively proposed (see section 6.
24
25 **Cell encapsulated building blocks to generate complex functional systems**).

26
27
28
29
30
31
32 The diffusion and permeability of hydrogels depends on at least three factors, namely *(i)* the
33
34 obstruction effect caused by the presence of impenetrable slowly moving polymer chains that
35
36 increase the path length for diffusion, *(ii)* the hydrodynamic drag at the polymer interface due
37
38 to polymer-solvent and polymer-solute bonds during the solute diffusion, and *(iii)* the residual
39
40 charges of the matrix, presence of counter ions, hydrogen bonds, polar and hydrophobic
41
42 interactions, which will affect the transportation of solutes exhibiting similar interactive groups
43
44 (especially essential in the transport of biological molecules).^[30] The mass transportation is
45
46 driven by two main forces, namely *(i)* by pressure gradient (convective), and *(ii)* by
47
48 concentration gradient (diffusive). In most hydrogels, the transport of solutes occurs by
49
50 diffusion. In terms of solute diffusion depending on their pore size, hydrogels can be divided
51
52 into three different classes, namely *(i)* macroporous, with pores $>0.1 \mu\text{m}$, which the transport
53
54 occurs mainly by convection *(ii)* microporous, with pores ranging from 0.005 to 0.02 μm , which
55
56
57
58
59
60
61
62
63
64
65

1 are in most cases smaller than the solute resulting in hindered diffusion, and (iii) nonporous
2 hydrogels, the most commonly used in cell encapsulation, in which solute transport occurs only
3 by diffusion through spaces between macromolecular chains. The different parameters affecting
4 the permeability of cell encapsulation systems using hydrogels have been detailed discussed.^[31]
5
6 Additionally, some cell encapsulation strategies are composed by membranes that surround the
7 encapsulation matrix or directly the encapsulated cells (conformal coating). Although
8 multilayered cell encapsulation systems have been commonly used to allow
9 immunoisolation,^[32–34] they have been poorly explored in tissue engineering aiming the
10 regeneration of tissues. This is because in tissue regeneration the immunoisolation feature is
11 not required, although the presence of multilayers can confer other functionalities and
12 advantages to the hydrogels. For example, the properties of each membrane can be
13 independently controlled, and the stability and permeability of the system can be easily
14 **customized by** varying the number of the multilayers. Multilayered cell encapsulation systems
15 can be obtained by the sequential adsorption of oppositely charged polyelectrolytes using the
16 mild conditions of the LbL technology,^[35,36] as further discussed in section **5. Novel and**
17 **nonconventional technologies to produce cell encapsulation systems.** Of note, in hydrogels
18 assembled for modular TE, the influence of the presence of a membrane surrounding each
19 building block on the permeability and diffusion properties must be carefully evaluated, since
20 the different extent of material heterogeneity between the encapsulation matrix and the
21 membrane might create fluctuations of transport properties across the different 3D structure.
22 An interesting approach is the assembled of multilayered and liquefied capsules by the action
23 of cells cultured on the outside environment. Simultaneously, at the core, 3D microaggregates
24 of encapsulated cells and surface-modified microparticles were developed, in a concept termed
25 has 3D+3D bottom-up TE.^[37]

3.3 Stability

The stability of the cell encapsulation matrix is related with its physical features, namely the mechanical resistance that it can support without disruption of its integrity or chemical features related with their ability to maintain its structure without being dissolved by physiological chelator agents or degraded. Either physical or chemical instability would lead to the premature release of the encapsulated cells and other materials of interest, releasing its content to other regions of the body rather than the implantation site. Increasing the stability of the encapsulation matrix can be performed by different approaches that could, however, influence other important parameters for cell survival. Developing systems with both adequate mass transportation and mechanical stability remains a key challenge in cell encapsulation technology, as they are often inversely related.^[31] Since alginate is the most widely applied hydrogel in cell encapsulation systems, efforts have been made to increase its stability. The first approach was to coat alginate beads with the oppositely charged polyelectrolyte PLL,^[19] but other polymers are being employed to construct improved biocompatible membranes, mainly using the LbL technology section (see section **5. Novel and nonconventional technologies to produce cell encapsulation systems**).

Another possibility is to increase the crosslinking density of the cell encapsulation matrix. The simplest example for the case of alginate systems is the use of barium instead of calcium as the gelling divalent agent, or alginates richer in guluronic acid units. However, it may lead to encapsulation matrixes with lower swelling capabilities and decreased mesh sizes, thus influencing the permeability of the construct. Crosslinking control will also result in matrices with distinct stiffness that will play an important role in cell behaviour, namely on the ability of stem cells to differentiate into specific lineages.^[38-41] For example, different osteogenesis levels could be obtained by controlling the viscoelastic properties^[42] or the stress-stiffening

1 effect of hydrogels.^[43] Different groups are increasingly exploring the use of precise chemical
2 routes to produce intrinsically robust hydrogels. A successful strategy was proposed by
3 incorporating in alginate hydrogels 2,6,6-tetramethylpiperidine-1-oxyl (TEMPO)-mediated
4 oxidized bacterial cellulose (TOBC). TOBC and alginate participate in the ionotropic gelation
5 with calcium ions.^[44] The carboxyl groups available on the surface of TOBC provided the
6 possibility of participating in the construction of an alginate-based composite and played crucial
7 roles in the structural, mechanical and chemical stability of the formed hydrogel. The
8 encapsulated NIH 3T3 cells remained viable and proliferative. Others, proposed robust
9 hydrogels with adequate permeability inspired by the role of glycosaminoglycans in providing
10 rigidity to the ECM due to their rigid sugar units and hydrophilic groups.^[45] Inspired by that, a
11 polysaccharide containing multivalent methacrylate groups and hydrophilic groups was
12 incorporated into a hydrogel to control its stiffness over a broad range, while controlling the
13 swelling ratio. This was achieved by the chemical crosslinking between methacrylic alginate
14 and PEG-dimethacrylate (PEG-DMA). The increase of gel stiffness resulting from the
15 incorporation of methacrylic alginate into a PEG-DMA hydrogel was related to the high chain
16 rigidity of alginate as well as the multivalent methacrylate groups. In parallel, multiple hydroxyl
17 groups of methacrylic alginate thermodynamically counterbalanced kinetic limits of osmotic
18 water entry. Results showed that the chemical crosslinking of PEG-DMA allowed controlling
19 the hydrogel stiffness without compromising its permeability, as demonstrated by the suitable
20 viability of encapsulated neural cells (PC12 cells). In another study, alginate and PEG were also
21 incorporated into interpenetrating network structures, improving significantly the toughness
22 and elasticity of the final hybrid construct, without compromising the viability of encapsulated
23 mesenchymal stem cells (MSCs).^[46] The resultant toughness relied on two mechanisms, namely
24 the reversible calcium ions crosslinking of alginate that dissipated mechanical energy, while
25 the covalent crosslinking of PEG maintained elasticity under large deformations. Besides
26
27
28
29
30
31
32
33
34
35
36
37
38
39
40
41
42
43
44
45
46
47
48
49
50
51
52
53
54
55
56
57
58
59
60
61
62
63
64
65

1 alginate, other polymers have also been employed to produce hydrogels with enhanced
2 mechanical properties and encapsulated cells by the use of precise chemical routes.^[47] For
3 example, two chitosan derivatives, namely low molecular weight methacrylamide and medium
4 molecular weight, were mixed with a photoinitiator (I2959) and the weak base β -
5 glycerophosphate.^[48] To produce double-network ultra-tough hydrogels a sequential dual-
6 crosslinking was performed, first by using UV-light exposure for the methacrylamide chitosan,
7 and then by immersion in a solution containing negatively-charged tripolyphosphate (TPP) for
8 the ionic crosslinking of medium molecular weight chitosan through their positively charged
9 amine groups. This strategy allowed to create tough hydrogels that were able to withstand an
10 impressive compressive stress in the same order of magnitude as the ones found in native load-
11 bearing soft tissues, with fast recover ability of their mechanical properties upon unloading,
12 while allowing cell encapsulation.
13
14
15
16
17
18
19
20
21
22
23
24
25
26
27
28
29
30
31

3.4 Degradation

32 There are a number of factors influencing the degradation of hydrogels, which can be mediated
33 by (i) the chemistry of its matrix and the density of degradable groups, (ii) the presence and
34 biological activity of cells, including the deposition of ECM and metabolite products, and (iii)
35 environmental triggers in the *in vivo* physiological environment of the host. Considering the
36 first referred circumstance, in physically crosslinked hydrogels the gelation is reversible. Using
37 the example of the ionotropic gelation of alginate with calcium chloride, events such as the
38 presence of electrolytes or the deposition of a newly formed ECM, can lead to the exchange of
39 divalent calcium ions for monovalent cations, leading to the dissolution of the polymer chains
40 or to the disruption of hydrogen bonds. In many cases, hydrogels are engineered to degrade by
41 hydrolysis and/or enzymatically by adding specific degradable sequences within its chemical
42
43
44
45
46
47
48
49
50
51
52
53
54
55
56
57
58
59
60
61
62
63
64
65

1 structure. This allows to control both the degradation rate and profile. Different to the
2 commonly used “*open*” scaffolds composed of pores in which cells can deposited ECM prior
3 to the scaffold degradation (see section 2. “*Open*” vs. “*closed*” scaffolds: limitations,
4 **advantages and practical considerations**), in cell encapsulation systems the degradation of
5 the matrix has to occur to provide space for the deposition of newly ECM by the encapsulated
6 cells. Therefore, the formation of the new tissue and the degradation of the hydrogel matrix are
7 intrinsically linked in cell encapsulation strategies. If the degradation of the hydrogel occurs
8 too quickly, i.e. prior to an appropriate deposition of ECM, the encapsulate cells will be
9 deprived from the physical support required for different cell anchorage processes. On the other
10 hand, if the degradation occurs too slowly, the deposition of ECM will occur in the pericellular
11 regions, since those are the regions at which the diffusion of essential molecules for cell survival
12 is higher, leading to a heterogeneous distribution of ECM in the hydrogel as well as cell necrosis
13 at the core of its structure, a very common drawback in 3D systems, and particularly in cell
14 encapsulation systems. It is thus highly desirable that degradation of the encapsulation matrix
15 must match the formation of the new tissue. For that, one of the options is to control the
16 chemical degradation of hydrogen bonds through the chemistry of the degradable linker. An
17 example is the controlled degradation of PEG by combining slowly and rapidly degrading
18 linkers such as polycaprolactone (PCL) and poly(L-lactic acid) (PLLA), respectively.^[49] While
19 these methods are frequently used nowadays, alternative attractive possibilities involve the use
20 of cell-mediated degradation.^[50–53] Through enzymatic cleavage or degradation, the cells can
21 direct the time line of degradation and adjust their surrounding environment as needed for
22 cellular growth, matrix deposition, and matrix re-organization. Techniques to achieve cellular-
23 mediated matrix degradation can be performed either by using hydrogels from natural
24 biopolymers, such as hyaluronic acid which degrades by hyaluronidases activity^[54,55] or thiol-
25 ene pectin hydrogels by collagenase type II^[50] or to program short amino acid sequences into

1 the hydrogel network, which are susceptible to enzymatic cleavage.^[52] The incorporation of
2
3 short peptide sequences in hydrogels present several advantages for TE applications, including
4
5 specificity for cell binding, and the development of cell instructive 3D systems on a large
6
7 scale.^[56,57] Examples of enzymatically degradable segments used in cell encapsulation
8
9 strategies are polysaccharide-based systems composed of ECM proteins, such as collagen,
10
11 fibrin, fibronectin and laminin proteins, and peptide-based linkages that have specific cleavage
12
13 sites for degradation by enzymes, such as elastase, plasmin or matrix metalloproteinases
14
15 (MMPs).^[57,58] Longer enzyme-cleavable chains can also be used as connecting points for
16
17 hydrogel formation, such as fibrinogen.^[59,60] Although enzymatically degradable single-phase
18
19 hydrogel materials offer elegant control over the cellular invasion, cell confinement within
20
21 these systems remains strongly coupled to matrix elasticity, and enzyme-mediated changes to
22
23 local mechanical properties may be difficult to control in a pre-determined manner. Taking
24
25 advantage of controlling the degradation of hydrogels, an interesting work was proposed using
26
27 void-hydrogels.^[61] Void-forming hydrogels were obtained by encapsulating sacrificial gel
28
29 porogens composed by oxidized hydrolytically labile alginate within a high molecular weight
30
31 alginate hydrogel, which has thus a slow degradation rate. The rate of pore formation was
32
33 controlled by the rate of porogen degradation and cell migration and proliferation within pores.
34
35 Remarkably, this strategy allowed decoupling the pore formation from the elasticity of
36
37 hydrogels, while controlling MSCs osteogenesis *in vitro*.
38
39
40
41
42
43
44
45
46
47
48

49 **3.5 Biotolerability**

50
51 While some groups claim that biocompatible polymers for cell encapsulation are available,
52
53 others doubt whether such materials can ever be designed. This paradox is explained by the
54
55 interpretation of the expression “*appropriate host response*” from the original definition of
56
57
58
59
60
61
62
63
64
65

1 biocompatibility. During the emerging of artificial organs field, the definition of
2
3 biocompatibility emerged as “*the ability of a biomaterial to perform with an appropriate host*
4
5 *response in a specific application*”.^[62] An “appropriate host response” included the fibrotic
6
7 capsules formation surrounding the implant. However, for cell encapsulation systems, defining
8
9 an appropriate host response is more complex because any inflammatory response against the
10
11 implanted system is potentially harmful to the encapsulated cells. This is because if immune
12
13 cells led to the formation of a fibrotic capsules surrounding the encapsulation system, the
14
15 diffusion of essential molecules for cells survival and the exchange of therapeutic molecules
16
17 will be jeopardized (we also recommend the reading of^[63] for a deep understand of the
18
19 interaction between the physicochemical properties and the biological responses in cell
20
21 encapsulation systems). Encapsulation systems developed for tissue regeneration applications
22
23 are not meant to prevent immune responses as cell encapsulation strategies designed to treat
24
25 endocrine diseases were. When implanting encapsulation systems aimed for tissue
26
27 regeneration, the immune system will be inevitably activated due to leakage of antigens,
28
29 protrusion of cells due to their proliferative ability, and to native responses associated with the
30
31 surgery. Therefore, the current leading opinion is that cell encapsulation systems for tissue
32
33 regeneration should preferably elicit a minimal immune response to avoid cellular overgrowth
34
35 surrounding the capsules. With that, emerges the term biotolerability as “*the ability of a*
36
37 *material to reside in the body for long periods of time with only low degrees of inflammatory*
38
39 *reactions*”.^[64] The current leading opinion is that, rather than “*an appropriate host response*”
40
41 of the term biocompatibility, the biotolerability concept of “low degrees of inflammatory
42
43 reactions” is more appropriate in the cell encapsulation field. Within the broad spectrum of
44
45 biomaterials available that induce different appropriate foreign body responses, other factors
46
47 such as chemical modifications,^[65] and the dimension of the encapsulation matrix^[66] also
48
49 influence immune-mediated reactions. Combinatorial methods were proposed to study the
50
51
52
53
54
55
56
57
58
59
60
61
62
63
64
65

1 immunological *in vivo* outcome of different biomaterials,^[67] or with a large library of the same
2
3 biomaterial, namely alginate hydrogels, but with variable dimensions and chemical
4
5 modifications.^[65] The interaction of hydrogels with the immune system is detailed discussed in
6
7 the following section.
8
9

10
11
12 Ultimately, for a cell encapsulation strategy to be successfully applied in TERM, a number of
13
14 practical factors must be also considered in addition to the critical properties discussed above.
15

16
17 The process must allow low-cost production, scale up manufacturing (e.g. encapsulation
18
19 process in the order of seconds to minutes), and easy handling and implementation. Such
20
21 characteristics are fundamental to facilitate the market acceptance by the target community,
22
23 including the appliers, surgeons; those who will receive the treatment, patients; those who will
24
25 commercialize it, the healthcare providers; and ultimately those who will allow such flow of
26
27 events, namely the ethical authorities, such as the FDA and equivalents. The incorporation of
28
29 all these considerations into the early stages of the design and conception of the cell
30
31 encapsulation system will significantly accelerate the translation of such hydrogels from the
32
33 bench to the clinics.
34
35
36
37
38
39
40
41

42 **4. Immunomodulation in cell encapsulation systems**

43
44

45 For a long time, the majority of biomaterials were designed to be biologically inert, in order to
46
47 avoid an acute inflammatory response that would end with the formation of a collagenous
48
49 fibrotic capsule surrounding and isolating the implant.^[62] Nowadays, it is well-known that a
50
51 proper tissue healing involves a well-regulated set of immune responses. Either by trauma or
52
53 simply by the implantation of a biomaterial, such inflammatory response starts with the rapid
54
55 arrival of cells from the innate immune system to the injury scene. Among all immune cells,
56
57
58
59

1 macrophages tend to be fundamental during all stages of the tissue repair process. In response
2
3 to factors present in the local tissue environment, recruited and resident macrophages mediate
4
5 multiple cellular events, namely proliferation, angiogenesis, and the deposition of ECM. Only
6
7 an efficient and precise timely switch from proinflammatory (“M1”) to regenerative (“M2”)
8
9 macrophage phenotype results in a tissue remodeling cytokine release, which appears
10
11 mandatory to tissue healing.^[68] Therefore, the paradigm has completely changed, and nowadays
12
13 the interplay between the two fields of immunology and TERM is a hot topic among the
14
15 scientific community. **Figure 2** schematically represents the different phases of the actuation
16
17 of the immune system following the implantation of a biomaterial into a lesion site. Knowing
18
19 that immune system plays a central role in tissue regeneration has contributed to the design of
20
21 a new generation of smart biomaterials able to modulate the action of the immune cells towards
22
23 tissue remodeling and regeneration. Furthermore, the availability of methods for specific blood-
24
25 derived monocytes isolation (CD14⁺) and their differentiation into desired phenotypes, such as
26
27 macrophages or dendritic cells, has led to an emergent development of engineered strategies
28
29 with autologous cells.^[69] Such strategies are mainly focused in co-culture studies of immune
30
31 and stromal cells encapsulated within hydrogels, due to a number of appealing features, namely
32
33 the resemblance of the native scenario of tissue regeneration.^[70] Particularly, regulatory
34
35 macrophages are essential in TERM systems, since they are involved in neovascularization,
36
37 granulation tissue removal, and new ECM components synthesis. Therefore, over the last few
38
39 years, novel stimuli-responsive encapsulation strategies aimed to dictate the kinetics of
40
41 macrophage polarization and hence, a more realistic tissue regeneration process, have been
42
43 validated in numerous 3D cell encapsulation systems.
44
45
46
47
48
49
50
51
52
53
54
55
56

57 **4.1 Interaction of hydrogels with the immune system**

58
59
60
61
62
63
64
65

1 A healthy immune system is able to protect the host by recognizing and eradicating pathogens
2 and other foreign molecules. In the classical design of biomaterials, the main impetus was to
3 avoid an immunological response. In fact, following implantation of a biomaterial, a set of
4 adverse immune reactions can occur. After material-immune system interaction, a provisional
5 matrix on material surface is formed, resultant from the blood and interstitial fluid proteins
6 precipitation and adsorption.^[71,72] The adsorption of proteins, such as albumin, vitronectin, and
7 fibronectin, have an influence on desired cell migration and attachment and subsequent
8 interplay between them and the material. Moreover, the blood-based transient matrix sustains
9 the release of bioactive compounds, which are crucial for the subsequent inflammatory
10 response. The acute inflammatory reaction is initiated in a sequential fashion, mainly driven by
11 neutrophils (polymorphonuclear leukocytes, PMNs) and mast cells. While PMNs release
12 proteolytic enzymes and reactive oxygen species (ROS) in an effort to degrade the biomaterial,
13 the degranulation of mast cells leads to the secretion of histamine, growth factors, and
14 inflammatory cytokines and chemokines, increasing the intensity of the immune response.^{[73–}
15 ^{75]} Additionally, the released chemotactic agents induce the recruitment and further
16 differentiation of monocytes into M1 macrophages, in an attempt to increase antimicrobial and
17 phagocytic responses.^[76,77] The migration and activation of lymphocytes are also involved in
18 the cascade of the immune response, resulting on the production of pro-fibrotic factors,
19 including interleukin (IL)-4, IL-13, and transforming growth factor (TGF) β . The prolonged
20 presence and stimulation of mononuclear cells, i.e. monocytes and lymphocytes, surrounding
21 the implanted biomaterial, give rise to a chronic inflammatory phase.^[78,79] If the material is
22 biotolerable, this chronic phase is typically of short duration, and subsequent remodeling and
23 regenerative responses are identified with fibroblasts recruitment and new healing tissue
24 formation.^[80] However, the exposure of biomaterials to host cells can trigger a foreign-body
25 reaction (FBR), which if not controlled, may lead to end-stage tissue fibrosis and scarring. An

1 indicative of FBR is the large presence of macrophages and foreign body giant cells (FBGCs).
2
3 FBGCs arise from macrophages that adhere to the transient matrix on material surface and fuse
4
5 to form these multinucleated cells, since the biomaterial is too large to be internalized by cells.
6
7
8 In later stages of FBR, if the excessive inflammatory response to biomaterials continues, it may
9
10 originate a fibrotic response.^[75,81] Fibrosis occurs mainly if the host environment fails to
11
12 naturally and sequentially stimulate the polarization of macrophages into a M2 pro-healing
13
14 phenotype. An imbalanced M1/M2 ratio, which can occur due to the prolonged presence of M1
15
16 macrophages leading to a delayed switch towards the M2 phenotype, can induce the release of
17
18 fibrosis-enhancing cytokine pattern by M2 macrophages.^[82,83] Briefly, M1 macrophages are
19
20 able to metabolize arginine into (i) nitric oxide (NO), which can be further metabolized to
21
22 downstream reactive nitrogen species, and (ii) citrulline, which can be reused for efficient NO
23
24 synthesis via the citrulline-NO cycle. On the other hand, M2 macrophages are able to hydrolyze
25
26 arginine to ornithine and urea through the expression of the enzyme arginase. Therefore, the
27
28 imbalance M1/M2 ratio is directly correlated with the arginase pathway, which limits the
29
30 availability of arginine for the NO synthesis, and ornithine can thus downstream the pathways
31
32 of polyamine and proline syntheses. Polyamine and proline are key mediators for cellular
33
34 proliferation and tissue repair. Since both arginine metabolic pathways cross-inhibit each other,
35
36 the M1/M2 polarization can thus lead to the formation of a fibrotic capsule around the implanted
37
38 biomaterial, impairing a proper interaction with the host. Consequently, the tissue integration
39
40 of the biomaterial, and the subsequent tissue regeneration, key indicators of the success of the
41
42 implantation of the biomaterial, is inversely related with the extent of FBR.^[75]

43
44
45 One important variable for the interaction of the immune system with the implanted biomaterial
46
47 is its physicochemical composition. The main challenge is to design a biomaterial that allows
48
49 not only the regeneration of the target tissue, while simultaneously controlling intra- and
50
51 intercellular mechanisms of the recruited immune cells. To achieve such goal is imperative to
52
53
54
55
56
57
58
59
60

1 fully understand the immunological profile of the biomaterials.^[84] Hydrogels are the most
2
3 widely explored systems for cell encapsulation strategies aiming the regeneration of tissues,
4
5 due to their high-water content, good biotolerability, and similarity with the ECM of native
6
7 tissues.^[85,86] Hydrogels can be produced from natural-derived polymers, such as alginate,
8
9 chitosan, hyaluronic acid, and collagen,^[85] or in combination with synthetic polymers, such as
10
11 PEG, PCL, polyacrylamide, and poly(vinyl alcohol), among others.^[23,24] Overall, natural-
12
13 derived polymers are known to improve the biotolerability of the implant, since they are
14
15 composed by low immunogenicity structures and induce a type-2-like immune response with
16
17 the upregulation of genes involved with damage-associated molecular pathways. On the other
18
19 hand, synthetic materials can display a more exacerbated inflammatory reaction by the
20
21 recruitment of a high portion of neutrophils, despite efforts to mitigate this response by tuning
22
23 their chemical and topographical surface characteristics.^[87-89] Moreover, natural-derived
24
25 polymers induce a positive innate immune response with a constructive remodeling phenotype,
26
27 a crucial gateway for tissue repair and regeneration. Alginate has been the material of choice
28
29 for encapsulation, but batches of this natural polymer need to be standardized to present
30
31 minimum endotoxin and proteins contents. In this regard, ultrapure alginate and other polymers
32
33 are being commercialized. A classic example of a natural-derived highly biotolerable hydrogel
34
35 is collagen. Collagen hydrogels are reported to induce a mild inflammatory response, mostly
36
37 guided by macrophages, while prompting the deposition of newly generated host ECM.^[90,91] In
38
39 an attempt to improve one of the main disadvantages of natural-derived hydrogels, namely their
40
41 poor mechanical properties, collagen hydrogels were modified with glutaraldehyde and an
42
43 aminosilane. *In vivo* results showed that such silicified collagen hydrogels recruited more new
44
45 blood-vessels from the host compared to collagen hydrogels modified only with glutaraldehyde.
46
47 Importantly, only collagen hydrogels lacking the aminosilane induced a FBR, and the
48
49 consequent fibrous capsule.^[92] Chitosan has also been shown to generate a pro-inflammatory
50
51
52
53
54
55
56
57
58
59
60
61
62
63
64
65

1 response by dendritic cells, while avoiding the proliferation of T lymphocytes and timely
2 polarizing the phenotype of macrophages into a remodeling state.^[93] The immunomodulatory
3 properties of hydrogels could be improved with the encapsulation of MSCs. Besides low
4 immunogenicity, MSCs are able to change the phenotype of native immune cells that infiltrate
5 the biomaterial. Due to their ability to play both enhancing and inhibiting roles on immune
6 cells, MSCs are very attractive for TE approaches.^[94–100] MSCs encapsulated in PEG hydrogels
7 were shown to down-regulate the response of M1 macrophages, and hence decreased the
8 fibrotic response of the FBR upon subcutaneously implantation in C57BL/6 mice for 28
9 days.^[96] Additionally, the encapsulation of MSCs is linked with the suppression of the
10 allogeneic lymphocyte activity, as well as with M2 macrophages recruitment and
11 polarization.^[101,102] Of note, such studies used autologous MSCs. To date, any study showed
12 the immune privileged feature of allogeneic MSCs, and their advantage compared to the use of
13 autologous MSCs.^[95,103] In fact, there are studies reporting adverse side effects following intra-
14 articular injection of allogeneic MSCs in equine models, such as synovial cellularity and the
15 formation of FBGCs.^[104,105] The immunomodulatory feature of MSCs can be enhanced using
16 TE strategies for cell delivery, such as cell encapsulation systems using biomaterials. Such
17 biomaterials can also contain within their polymeric matrix specific bioactive growth factors
18 enabling a local delivery, and thus directly modulating the cellular infiltration around the
19 implanted scaffold. For example, the conjugation and sequential release of immunomodulatory
20 cytokines were shown to control macrophage phenotype with resulting effects on scaffold
21 vascularization.^[106] Firstly, M1 macrophage response was enhanced by the presence of
22 interferon γ , an inflammatory cytokine, physically adsorbed onto the scaffold. Then, IL-4
23 attached via biotin and streptavidin binding, was continuously released to polarize macrophages
24 into the M2 remodeling phenotype. This strategy allowed the host macrophages to achieve a
25 greater vascularization and healing, following murine subcutaneous implantation.

1 Indeed, it is clear that the immune system plays a key role in tissue repair and regeneration
2 process. Therefore, the integration of a functional biomaterial is facilitated when the crosstalk
3 with the host immune cells is well-established. Furthermore, using biocompatible biomaterials
4 in combination with autologous MSCs, which besides their well-described advantages for TE
5 applications, namely ease of isolation, manipulability and multilineage differentiation potential,
6 would also avoid the use of immunosuppressant drugs.^[107] The administration of such drugs
7 are related with several adverse side effects known to down-regulate the immune response.
8 Moreover, numerous studies have been shown that glucocorticoid-based anti-inflammatory
9 treatments are related with the reduction of inflammatory cytokines, and hence jeopardize the
10 healing of the injured tissue by delaying the clearance process, and decreasing the angiogenesis
11 capability and cell proliferation.^[108–110] In summary, immunomodulatory biomaterials should
12 be able to control the immune environment surrounding the implantation site by eliciting a pro-
13 regenerative immune response rather than avoiding the initial inflammatory reaction.
14
15
16
17
18
19
20
21
22
23
24
25
26
27
28
29
30
31

32 **4.2 Immunomodulatory encapsulation strategies for tissue regeneration**

33
34
35
36
37
38 Immune cells have been identified as potential targets to integrate tissue engineered constructs
39 and supplement or ameliorate a desired event, such as regeneration and vascularization. Several
40 studies have attempted to prove that the incorporation of macrophages is feasible and can
41 actually improve the proposed TERM strategies. Some of these approaches are discussed
42 hereafter.
43
44
45
46
47
48

49
50 A hyaluronic acid hydrogel encapsulating MSCs led to an anti-inflammatory polarization of
51 monocyte-derived macrophages (MDM) cultured in the external environment. Therefore, not
52 only the inert feature of most biomaterials is a utopic scenario, as in fact, it can be beneficial to
53 the healing process.^[111] Such beneficial biological outcome was enhanced when MSCs and
54
55
56
57
58
59

1 MDM were co-cultured within the same hydrogel, leading to the production of bioactive
2
3 molecules involved in collagen homeostasis, cell adhesion, angiogenesis, immunosuppression
4
5 and tissue repair.^[112] Alternatively, a photoresponsive hyaluronic hydrogel combined with Arg–
6
7 Gly–Asp (RGD) adhesive peptide was engineered to control the immunomodulatory crosstalk
8
9 of encapsulated macrophages. Here, results shown that the RGD peptide can activate
10
11 macrophage $\alpha\beta3$ integrin, and hence, enhance an anti-inflammatory “M2” macrophage
12
13 polarization.^[113] Otherwise, macrophages encapsulated in two different hydrogels in the
14
15 presence of IL-4, which chemically polarizes macrophages into a regenerative “M2” phenotype,
16
17 had the most distinguishing reactions. Whereas macrophages encapsulated in gelatin
18
19 methacryloyl (GelMA) hydrogels, were driven into a regenerative profile, macrophages that
20
21 were encapsulated in poly(ethylene glycol) diacrylate (PEGDA) hydrogels, expressed a more
22
23 proinflammatory “M1” phenotype.^[114] Additionally, a reduced availability of soluble tumour
24
25 necrosis factor (TNF)- α following a pro-inflammatory stimulation was observed in monocytes
26
27 entrapped in GelMA hydrogels. These results indicate that under pro-inflammatory conditions,
28
29 GelMA can be potentially characterized with anti-inflammatory properties.^[115] When cultured
30
31 on polyethylene terephthalate coated with collagen (PET/Col), macrophages expressed a pro-
32
33 inflammatory “M1” profile, while on polypropylene (PP), these immune cells expressed a
34
35 regenerative “M2” phenotype.^[116] In order to create a wound healing model, a co-culture of
36
37 macrophages and adipose-derived stromal cells (ASCs) were encapsulated in PET/Col and PP
38
39 hydrogels. Results suggested that such hydrogels influenced the process of tissue regeneration
40
41 by guiding the polarization of macrophages. In fact, genes involved in proliferation,
42
43 vasodilation and collagen deposition, such as COX2 and PTGS2, were differentially expressed
44
45 by ASCs when co-cultured with macrophages.^[117]

56 An *in vitro* model was created using gelatin hydrogels to assess the relevance of resident
57
58 macrophages in engineered tissues aiming regeneration.^[118] For that, non-polarized monocytes,
59
60

1 polarized macrophages with “M1” or “M2” stimuli, or incoming cells were combined. The
2
3 incoming cells formulation encompassed the co-encapsulation of non-polarized THP-1 (human
4
5 monocyte cell line), fibroblasts, and endothelial cells (ECs). Results show that hydrogels
6
7 encapsulating macrophages were able to recruit more ECs and fibroblasts, which are key
8
9 elements in the wound healing process, compared to non-polarized monocytes. Furthermore,
10
11 after an initial characteristic proinflammatory phenotype of the wound healing, the
12
13 microenvironment has become more pro-regenerative through the release of IL-1RA, CCL-18
14
15 and IL-4 cytokines. To recreate an artificial homeostasis of wound regeneration in the gelatin
16
17 hydrogels via paracrine and cell-cell contact effect, macrophages were co-encapsulated with
18
19 ECs or/and fibroblasts. Firstly, macrophages in co-culture with fibroblasts significantly
20
21 enhanced cell proliferation and cytokine secretion, creating a more stimulating
22
23 microenvironment in the encapsulation system. When macrophages were co-cultured with ECs,
24
25 a favourable microenvironment for angiogenesis was created with the up-regulation of IL-6 and
26
27 IL-1RA. Finally, to recreate the actual *in vivo* microenvironment, a tri-culture of macrophages,
28
29 fibroblasts and ECs encapsulated in the gelatin hydrogels resulted in a denser like-tissue
30
31 structure. Moreover, macrophages affected the angiogenic and proliferation secretory
32
33 environment by significantly boosting the release of activin, IL-6 and IL-8.^[119] Likewise, after
34
35 the encapsulation of macrophages and ECs on a 3D PEG-based system, macrophages were
36
37 capable to influence vessel formation inside of the hydrogel. In particular, macrophages were
38
39 shown to associate with ECs in a pericyte-like manner, as well as, bridging between endothelial
40
41 structures in a cell-chaperoning fashion.^[120]

42
43 In particular, the complex role of immune cells in musculoskeletal diseases has motivated the
44
45 development of the osteoimmunology field. The study of the interactions between MSCs and
46
47 macrophages in gelatin/PEG matrices demonstrated that although the co-culture attenuates
48
49 chondrogenic differentiation, it actually enhances osteogenic and adipogenic
50
51
52
53
54
55
56
57
58
59
60
61
62
63
64
65

1 differentiation.^[121] Otherwise, transglutaminase cross-linked gelatin (TG-gel) for 3D culture
2 was used to study how stiffness-tuneable matrices can affect macrophage induced osteogenesis.
3 Here, despite the high-stiffness harnessed MSC osteogenic differentiation, macrophages
4 presented a pro-inflammatory phenotype. However, when macrophages and MSCs were
5 encapsulated in the same type of TG-gel, the gap of osteogenesis levels between low and high
6 stiffness matrices was narrowed. In fact, both stiffness systems showed mineralized nodules
7 development and enhanced alkaline phosphatase activity.^[122] Overall, accumulating evidence
8 indicates that dimensional and mechanical parameters of cell encapsulation systems facilitate
9 the polarization of macrophages towards a target phenotype commitment, and thus, it should
10 be considered when designing hydrogels as cell encapsulation systems aiming the regeneration
11 of tissues.
12
13
14
15
16
17
18
19
20
21
22
23
24
25
26
27
28
29

30 **4.3 Challenges of immunomodulatory hydrogels aiming tissue regeneration**

31
32
33 One of the most critical parameters in TE strategies is the biotolerability character of the
34 biomaterials. Biotolerability is a controlled low degree inflammatory reaction, in which the host
35 immune cells tolerate the implanted biomaterial for long periods of time.^[64] As aforementioned,
36 it is essential that a biomaterial stimulates tissue repair and regeneration without eliciting a
37 FBR. However, in strategies such as cell encapsulation, the definition of an appropriate host
38 response is more complex. Here, any inflammatory reaction against the implanted biomaterial
39 can be destructive for the encapsulated cells, because this response is associated with the
40 diffusion of harmful cytokines, leading to cell death and further failure of the cell encapsulation
41 system.^[64,123] Additionally, the presence of cells within the biomaterial also can elicit a more
42 intense immune reaction. Not only the biomaterial composition and processability should be
43
44
45
46
47
48
49
50
51
52
53
54
55
56
57
58
59
60
61
62
63
64
65

1 compatible with the encapsulated cells, as such cells should be tolerated by the immune system
2
3 of the host.
4

5 A major challenge dictating the hydrogels biotolerability is the presence of endotoxins. Prior to
6
7 implantation, a highly pure and sterile polymer is required. Low levels of endotoxins can induce
8
9 severe inflammatory responses, leading to inadequate integration of the cell encapsulation
10
11 system.^[124,125] Occasionally, other adverse effects are observed on natural-derived hydrogels.
12
13

14 Due to allogeneic/xenogeneic properties of the natural materials, enzymatic and hydrolytic
15
16 degradation can occur *in vivo*.^[126] Additionally, it is noteworthy that most of monocytes and
17
18 macrophages used for biocompatibility tests are derived from leukemia, such as THP-1 and
19
20 HL-60 cell lines isolated from acute monocytic leukemia, or lymphomas, such as the U-937
21
22 cell line obtained from the pleural effusion of a 37-years old patient with histiocytic lymphoma.
23
24

25 The differences between the human blood-derived monocyte and the cell lines should be better
26
27 understood, and the tumour-derived cells used for the *in vitro* assays must be chosen
28
29 accordingly to the intended use of the medical device.^[127]
30
31
32
33
34
35
36

37 **5. Novel and nonconventional technologies to produce cell encapsulation systems**

38
39

40 To face the demanding requirements of cell encapsulation systems aiming the regeneration of
41
42 tissues, different technologies are being proposed. Herein, we highlight the contribution of LbL,
43
44 microfluidics, superhydrophobic surfaces, and 3D bioprinting technologies to produce the next
45
46 generation of cell encapsulation systems (**Figure 3**). LbL is proposed for the build-up of a
47
48 multilayered membrane surrounding the cell encapsulation matrix. The membrane is formed
49
50 due to the electrostatic interaction of oppositely charged polyelectrolytes. Microfluidics allows
51
52 the homogenous production of microgels encapsulating multiple or single cells. Due to the
53
54 repellence properties of superhydrophobic surfaces, spherical hydrogels can be produced using
55
56
57
58
59

1 different bath-free crosslinking methodologies, such as by placing a drop on top of the
2 previously formed droplet (drop-on-drop) or by photopolymerization with UV light. 3D
3 bioprinting allows the production of clinically relevant structures using bioinks encapsulating
4 individualized cells, cellular aggregates or combining cells with microcarriers. The different
5 cell encapsulation systems described in the literature using such techniques are highlighted in
6 the following subsections.
7
8
9
10
11
12
13
14
15
16
17

18 **5.1 Layer-by-layer**

19
20
21 Since its introduction,^[128] LbL has become one of the mostly used techniques to coat with
22 multilayers the surface of biomaterials. The main advantages of the LbL technique are the
23 ability to provide a reliable, easy, versatile, environment friendly, and cost-effective way of
24 coating and consequently modifying surfaces.^[36] The principle of the technique is based on the
25 sequential adsorption of a wide range of polyelectrolytes.^[35]
26
27
28
29
30
31
32

33 In particular, LbL has been widely applied to produce polymeric multilayered capsules (PMCs).
34 PMCs are fabricated through sequential deposition of polymers in the surface of a sacrificial
35 core, which is subsequently eliminated.^[129,130] The obtained nanometer thin membrane,
36 composed by a few or several multilayers, is “permselective” allowing the diffusion of water,
37 ions or other relevant bioproducts (e.g. nutrients, oxygen, metabolites, and waste products),
38 while excluding larger components (e.g. high immune components and cells). Additionally, the
39 sequential fabrication procedure combined with multiple post-processing modifications (e.g.
40 incorporation of molecules or micro/nanoparticles, elimination or solubilization of the core)
41 allows a precise fine-tuning of the system properties.^[131] PMCs have been used in a wide range
42 of biomedical applications, such as in imaging,^[132] drug delivery,^[133–135] biosensors,^[136,137],
43 synthetic vaccines,^[138] nanoreactors,^[139] catalysts,^[140] cell coating,^[141] and many others. More
44
45
46
47
48
49
50
51
52
53
54
55
56
57
58
59
60
61
62
63
64
65

1 recently, these properties have put PMCs under attention in the field of cell encapsulation.

2
3 The most appealing features that the LbL technique could offer to the cell encapsulation field
4
5 are (i) the possibility to be performed at room temperature or at 37°C and in mild conditions,
6
7 assuring the ideal conditions for cell viability, (ii) it is a aqueous-based procedure compatible
8
9 with a broad range of natural and synthetic polyelectrolytes as well as with biomolecules, (iii)
10
11 3D structures, including those with complex shapes and irregular topographies, can be easily
12
13 coated with multilayers, (iv) it offers precision control over the composition and thickness of
14
15 composite membranes through control over the number and nature of layers deposited, allowing
16
17 to tune the semipermeability of the membrane, and (v) it allows increasing the complexity of
18
19 the encapsulation system by adding new functionalities and capabilities - the multilayered
20
21 membrane can act as drug reservoirs or include biological functional components, such as
22
23 proteins, enzymes, antibodies, and peptide sequences that elicit specific biological
24
25 responses.^[142-144] Therefore, LbL can be performed in cell encapsulation strategies by using the
26
27 jellified hydrogel matrices as templates to build an engineered membrane over its surface,
28
29 which in tissue regeneration its presence is not compromised to the immunobarrier role. While
30
31 being a very promising technique, with the possibility of open new prospects to the function of
32
33 semipermeable membranes, LbL has been poorly explored in 3D cell encapsulation systems for
34
35 tissue regeneration. The main exploited field of the LbL technique in cell encapsulation is
36
37 within the single-cell encapsulation, which although being a promising instrument for
38
39 engineering cells with enhanced properties is not on the scope of the present review (we
40
41 recommend the reading elsewhere^[145,146]). We believe that this is mainly related to its inherent
42
43 time-consuming aspect of LbL. Therefore, to the effort required to obtain an engineered cell
44
45 encapsulation matrix, then adding the time-consuming task of the construction of a multilayered
46
47 membrane has to be carefully pondered. This aspect is correlated to the viability of the
48
49 encapsulated cells, limiting the number of layers composing the membrane. Clearly more
50
51
52
53
54
55
56
57
58
59
60
61
62
63
64
65

1 efforts should be made to decrease the processing time by, for example, reduce the adsorption
2 time of the layers (avoiding the adsorption until equilibrium or increasing the polyelectrolyte
3 concentrations) or by using other assembly mechanisms between the layers, such as fast
4 chemical reaction towards robust covalent bonds formation.
5
6
7
8
9

10 LbL technology was combined with the ionotropic gelation of alginate to produce a liquified
11 cell encapsulation strategy.^[147] Alginate hydrogel spheres were used as cell encapsulation
12 templates for the construction of the LbL membrane. Once the LbL membrane was built, the
13 alginate core was liquified by chelation of the calcium ions through ethylenediaminetetraacetic
14 acid (EDTA) treatment. The multilayered membrane was built due to the electrostatic bonds
15 between chitosan and alginate polyelectrolytes. Results showed that mechanical strength of the
16 capsules and the viability of the encapsulated cells were affected by the different number of
17 layers employed. More recently, a liquified cell encapsulation strategy was developed
18 combining the ionotropic gelation of alginate, LbL assembly, and the co-encapsulation of
19 stromal cells with surface modified PLLA microparticles.^[148–151] In this concept, by providing
20 to the encapsulated cells surface functionalized microparticles as solid cell adhesion sites, the
21 viability of the encapsulated cells could be enhanced. Additionally, the number of layers was
22 increased to 12 layers without compromising cell viability, and the mechanical strength of the
23 capsules was improved by using a three-component polyelectrolytes assembly, namely poly(L-
24 lysine), chitosan, and alginate. Importantly, the LbL technique allowed creating an
25 encapsulation strategy in which cells are encapsulated in a liquid environment. This allowed to
26 confer freedom for the encapsulated cells to freely self-construct their 3D organization, while
27 providing an appropriate diffusion of essential molecules for cell survival, a major concern in
28 cell encapsulation strategies. However, maximizing the core dissolution to achieve an excellent
29 diffusion required the introduction of solid cell adhesion spots, provided by the PLLA
30 microparticles. Consequently, this innovative cell encapsulation strategy allowed capsules to
31
32
33
34
35
36
37
38
39
40
41
42
43
44
45
46
47
48
49
50
51
52
53
54
55
56
57
58
59
60
61
62
63
64
65

1 have a much higher diameter (of ca. 2 mm) rather than the established 400 μm for cell
2 encapsulation matrices, without observation of a necrotic core. The proposed capsules were
3 proposed as an alternative methodology to the commonly used “*open*” scaffolds, since here
4 cells are also adhered at the surface of a substrate but then the system is wrapped by a membrane,
5 being physically isolated from the environment but without being embedded in an elastic matrix
6 as usually observed in “*closed*” scaffolds. The successful of this strategy boosted its application
7 to the encapsulation of stromal cells for bone^[149,150] and cartilage^[148] TERM applications. Such
8 system was already validated *in vivo* for bone TE.^[152] We recently proposed to combine the
9 technology to produce liquefied and multilayered capsules with the bioelectrospraying
10 technology for the production of microcapsules at high rates.^[151,153] Additionally, we also
11 proposed the development of multilayers surrounding liquified capsules encapsulating cells as
12 an assembly methodology to construct 3D macrostructures from a bottom-up approach.^[154]

13 Alternatively to the diffusion-driven kinetics of classical LbL assembly, which consists of
14 dipping the substrate in the polymer solution, recent advances in LbL assembly technologies
15 have explored other driving forces (e.g. dewetting, centrifugation, immobilization, spinning,
16 spraying, atomization, electrodeposition, magnetic assembly, microfluidics, among others^[146]).

17 Besides many other advantages, some of those different assembly technologies are also able to
18 solve the time-consuming feature of the process. Additionally, there is now a growing
19 realization that the assembly method not only determines the process inherent properties, but
20 also directly affects the physicochemical properties of the membrane built. However, the
21 applicability of such LbL assemblies has not been tested yet in cell encapsulation devices.

22 Nonetheless, we anticipate that its extrapolation and application will have a tremendous impact
23 on the field. Besides the mechanical protection and mass transfer control, multilayers could also
24 confer new features and elicit specific functions. For example, they could integrate inorganic
25 elements to enhance bioactivity,^[155] magnetic-responsive nanoparticles to control and

1 manipulate its movement,^[148] and light-responsive multilayers^[156] or containing gold
2 nanoparticles for light-activated disruption.^[157] Moreover, they could include or expose
3 biochemical elements to exhibit specific bioinstructive characteristics,^[158] such as growth
4 factors to stimulate cell differentiation.^[144,159,160]

5
6
7
8
9
10 Besides using the LbL methodology to surround cell encapsulation matrices, conformal coating,
11 in which cells are directly coated with ultrathin (2-100 nm) protective soft shells, are also an
12 application example of LbL in cell encapsulation.^[161,162] The great advantage that conformal
13 coating brought to the cell encapsulation field was the possibility to improve cell functionality
14 and viability using a very simple technique rather than adding soluble factors to the culture
15 medium, as commonly used in *in vitro* culture, or other complex techniques such as genetic
16 manipulation. For example, the multilayers can improve their mechanical stability, to protect
17 them against phagocytosis by masking the cells surface from immunological agents, provide
18 chemical resistance to aggressive environments, and to supply cells with additional
19 instrumentation for their functionality, as similarly above discussed for cell encapsulation
20 matrices. The production of ultrathin protective shells directly surrounding cells by LbL
21 technology can be performed through (i) synthetic polyelectrolyte shells, in which due to the
22 cell surface negatively charge at physiological pH, the shell assembly begins with the deposition
23 of a polycation, then a polyanion is deposited, and so on, until the planned shell architecture is
24 realized, (ii) synthetic hydrogen-bonded shells, in which the assembly occurs via non-covalent
25 hydrogen-bonding interactions and their micromechanical properties can be controlled by
26 changing pH, ionic strength, light conditions, salt concentration, or temperature; additionally,
27 the degradability of the shells can be tuned by adjusting conditions that result in a controllably
28 disassembled, and (iii) natural proteins, including hemoglobin, bovine serum albumin, and
29 human serum albumin (the different LbL protective shells have been detailed reviewed
30 elsewhere^[161]). The presence of protective shells has been poorly explored in the field of cell
31
32
33
34
35
36
37
38
39
40
41
42
43
44
45
46
47
48
49
50
51
52
53
54
55
56
57
58
59
60
61
62
63
64
65

1 encapsulation towards tissue regeneration, since usually the living organisms coated are mainly
2 bacterial and yeast cells. However, the application of conformal coatings on such living
3 organisms has allowed improving one of the main drawbacks of the LbL technique in cell
4 encapsulation, which is related with its extensive time-consuming. In an innovative study, it
5 was eliminated the adsorption step required between polyelectrolytes,^[163] thus significantly
6 decreasing the required time to produce a thin membrane. For that, during the polycation
7 adsorption, the surface potential of each layer was permanent monitoring. When the surface
8 recharging process to positive is completed, the polycation solution is immediately replaced by
9 the polyanion solution until negative surface charge saturation is reached, and so on.

10 **Table 1** summarizes the polyelectrolytes, encapsulation matrix, type of encapsulated cells, and
11 the TERM application of examples of cell encapsulation systems aiming tissue repair using the
12 LbL technique.

33 5.2 Microfluidics

34 Micro-technologies in cell encapsulation allow a high degree of control over the morphological
35 and dimensional (size and shape) properties of the encapsulation matrices. It is also possible to
36 encapsulate cells in different geometries, such as spherical hydrogels and in fibers, in a short
37 time. The microspheres and spheres can also be used to construct complex geometries through
38 assembly into larger architectures mimicking the structure of tissues and organs, as discussed
39 in section 5. *Cell encapsulated building blocks to generate complex functional systems*. In the
40 case of microspheres, microfluidics is mainly performed by flow-focusing or T-junction. In
41 flow-focusing, microspheres are formed by intercepting the core solution with a sheath stream
42 flowing, while in T-junction microspheres are formed by permitting the core fluid to be swept
43 away by one sheath stream in one direction.^[164] Typically, an aqueous alginate solution is

1 emulsified in an oil phase and crosslinked ionically with divalent ions, immediately upon
2 contact of the two solutions.^[165] However, the gelation process is poorly controlled and,
3 consequently, clogging and polydispersion are often observed.^[166,167] To overcome these
4 problems, different studies had proposed the use of calcium carbonate (CaCO₃)
5 nanoparticles,^[168,169] which allow to deliver calcium ions to the alginate solution without
6 inducing unintended gelation prior to drop formation. CaCO₃ nanoparticles are dispersed in the
7 alginate matrix to avoid premature gelation. After drop formation, nanoparticles are dissolved
8 under acidic conditions after drop formation. The main drawback is the nonhomogeneous
9 microspheres due to the heterogeneous distribution of calcium ions. Other similar techniques
10 use calcium chloride or acetate particles dispersed in the oil phase to initiate the crosslinking
11 process, which are subsequently dissolved in the emulsion droplet.^[170,171] However, the same
12 drawbacks of inhomogeneous calcium distribution and clogging are observed. Therefore, the
13 new generation of microfluidics system to develop microspheres as cell encapsulation systems
14 are focused in controlling the crosslinking process to produce homogenous microspheres with
15 reliable and precisely tunable properties, which is of great importance in TERM, stromal cell
16 research, and disease treatments.^[172–174] Alternatively, the generation of alginate microspheres
17 via coalescence of separate droplets containing alginate and calcium chloride has been
18 proposed.^[175] However, mixing inside the coalesced droplets still results in heterogeneous
19 microspheres since crosslinking occurs before a homogeneous distribution of calcium ions can
20 be achieved. The fabrication of monodisperse alginate microspheres with structural
21 homogeneity via droplet-based flow-focusing microfluidics was successfully developed.^[176]
22 The solution to overcome the above-mentioned drawbacks observed was to deliver calcium
23 ions by a solution containing water-soluble calcium mixed with the chelator EDTA. By
24 chelating the calcium ions with the EDTA, the ions remained in solution while being
25 inaccessible to the alginate chains. After drop formation, acetic acid is added to the continuous

1 phase to dissociate the calcium-EDTA complex, which results in the release of calcium ions.

2
3 The free calcium ions react with the alginate chains in a highly controlled fashion, reticulating
4
5 the alginate microspheres. Results demonstrated that the proposed gelation process was suitable
6
7 for the encapsulation of living MSCs.
8
9

10 Besides microspheres, microfluidics technique has been also used to fabricate long hydrogel
11
12 microfibers. These microfibers were generally prepared by embedding dispersed cells directly
13
14 within a hydrogel precursor, such as alginate,^[177-182] chemically modified gelatin,^[183,184] and
15
16 supramolecular hydrogels.^[65] Ideal platforms to mimic the complexity of biological systems
17
18 are fiber-based systems.^[164,177,185] Of note, the fabrication process of such systems must
19
20 withstand the use of proteins and other soft materials, incompatible with the processing
21
22 conditions of the conventional spinning techniques to produce fibers. As already mentioned,
23
24 the processing of such biological materials in cell encapsulation systems requires aqueous
25
26 conditions with precisely tuned temperature and pH in order to do not jeopardize the viability
27
28 of the encapsulated cells and the incorporated bioactive molecules. These processing limitations
29
30 have impaired the successful outcome of the traditional fiber spinning processes. Consequently,
31
32 a new fiber spinning methodology by microfluidics technique was proposed. Microfluidic-
33
34 spinning methodology has been employed to produce fibers in a microchannel using the coaxial
35
36 flow of a pre-polymer and the crosslinking agent.^[164] It is similar to the wet spinning,^[186] but
37
38 the bath is substituted by a direct supply of the crosslinking agent through the coaxial flow.
39
40 Microfluidics spinning is thus the most suitable fiber formation technique for cell encapsulation
41
42 because it does not require high voltage or temperatures, fibers can be fabricated continuously,
43
44 and allows a precise control over the diameter of the fibers only by regulating the flow rate,
45
46 which can be tuned from a few microns to a few hundred microns, and a wide diversity of cells
47
48 can be encapsulated without incurrent significant damage to cells.
49
50
51
52
53
54
55
56
57
58
59
60
61
62
63
64
65

1 **Table 2** summarizes the type of chip, encapsulation matrix, type of encapsulated cells, and the
2
3 TERM application of examples of cell encapsulation systems using microfluidics.
4
5
6
7

8 **5.3 Superhydrophobic surfaces**

9

10
11 Superhydrophobic (SH) surfaces have a unique chemistry and nano/microstructure
12 organization. They can be universally found in nature, such as in the classical example of lotus
13 leaf, but also in many others.^[187,188] SH have inspired biomimetic designs for controlling surface
14 wettability for TERM, namely in microfluidics, drugs and/or cell encapsulation, and cell
15 spheroids formation as discussed in different studies.^[189–194] Inspired by the rolling of water
16 drops on the lotus leaf, SH surfaces with water contact angles higher than 150° have triggered
17 increasing interest in the scientific community for their application in the biomedical
18 field,^[187,192,195] such as to produce cell spheroids in an innovative hanging drop
19 methodology.^[196,197] Of particular interest to the present review, SH surfaces have been also
20 used as an alternative methodology to produce spherical objects for cell encapsulation. Usually,
21 cell encapsulation systems imply the use of two solutions: one loaded with the encapsulation
22 materials and the other comprising the precipitation/crosslinking bath. It is precisely here that
23 SH bring a great advantage for encapsulation systems: its major advantage is the high
24 encapsulation efficiency of cells and bioactive molecules, which is of almost 100%, by
25 eliminating the need of a crosslinking bath. As first reported,^[191] the process involves the
26 dispensing of a polymeric solution loaded with cells on its surface, which leads to spherically
27 shaped droplets due to the repellence properties of the surface, and, subsequently, the liquid
28 droplets are crosslinked under mild conditions, originating cell encapsulated hydrogel spheres.
29
30 The crosslinking process is often performed by dispensing another drop on top of the previously
31 formed droplet (Figure 3-drop-on-drop) or by photopolymerization (Figure 3-UV light). SH
32
33
34
35
36
37
38
39
40
41
42
43
44
45
46
47
48
49
50
51
52
53
54
55
56
57
58
59
60
61
62
63
64
65

1 polystyrene surfaces were further explored to produce alginate spheres encapsulating MSCs
2 and fibronectin.^[198] The alginate drops on the top of the SH polystyrene surfaces were
3 crosslinked by adding a small amount of calcium chloride at the top of each droplet. Similarly,
4 the same authors developed a thermoresponsive chitosan-based cell encapsulation system.^[199]
5 Briefly, the acidic chitosan solution was first neutralized with β -glycerophosphate (β GP), and
6 to crosslink the matrix sodium tripolyphosphate was added at the top of each droplet. After
7 incubation at 37°C, a second gelation step occurred due to the thermoresponsive ability
8 conferred by adding β GP to chitosan, while the pH-responsive behavior of chitosan was
9 maintained. Using the same type of SH surfaces, the authors also proposed a
10 multicompartimentalized “onion-like” hydrogel^[200]. Methacrylated dextran (DEX-MA)
11 solution containing a photoinitiator and calcium chloride was dispensed on SH surfaces and
12 then crosslinked under UV light. Then, a solution of sodium alginate containing mouse
13 fibroblast cell line L929 cells was dispensed on the top of the spherical DEX-MA hydrogels.
14 Subsequently, the alginate outer layer was crosslinked by the release of calcium ions previously
15 immobilized in the core of the DEX-MA hydrogels.
16

17 Other types of SH surfaces were also explored to produce cell encapsulation systems. For
18 example, glass SH surfaces produced by chemical vapor deposition were used to develop a
19 hierarchical system encapsulating cells and/or drugs.^[201] For that, microdroplets of DEX-MA
20 containing the cytocompatible photoinitiator I2959 and L929 cells were dispensed through a
21 350 μ m diameter nozzle on the top of the developed glass SH surfaces. After UV crosslinking,
22 the obtained microspheres were encapsulated in alginate spheres using again the same SH
23 surfaces. With this simple and low-cost technique, authors were able to produce hierarchical
24 (micro-in-macro) encapsulation systems.
25

26 In order to propose encapsulation systems using SH surfaces but that would be not limited to
27 the spherical shape, SH surfaces with wettable spots were developed. The idea is to vary the
28

1 hydrogel geometries by varying the geometry of the wettable regions. Those SH surfaces were
2
3 combined with ultra-rapid production of multi-shaped hydrogels.^[202] Based on the co-existence
4
5 of superhydrophobic-superhydrophilic patterns, also called “discontinuous dewetting”, the
6
7 authors were able to produce arrays of droplets containing maleimide-polyvinyl alcohol
8
9 encapsulating HeLa cells. Using this technique, the droplets of future hydrogels are instantly
10
11 formed as the liquid moves along the superhydrophilic-superhydrophobic patterned surface.
12
13 Remarkably, any HeLa cells were detected on the superhydrophobic regions, thus all cells were
14
15 encapsulated within the microgels.
16
17
18

19
20 **Table 3** summarizes the substrate and respective treatment to produce superhydrophobic
21
22 surfaces, the encapsulation matrix, the geometry, the type of encapsulated cells, and the TERM
23
24 application of examples of cell encapsulation systems using superhydrophobic surfaces.
25
26
27
28
29
30

31 **5.4 3D Bioprinting**

32
33

34 One of the main disadvantages of processing hydrogels for cell encapsulation systems is the
35
36 difficulty to shape them in predesigned geometries to mimic the complex microenvironments
37
38 of natural tissues. To solve this specific drawback, different rapid prototyping (RP) techniques
39
40 have emerged to produce cell encapsulation systems with complex 3D structures.^[203] 3D
41
42 computer models shape the external design that will dictate the final structure, and such models
43
44 can either be designed by Computer-Aided Design, known as CAD software, or by modelling
45
46 imaging data (e.g. computer tomography and magnetic resonance imaging). In fact, this is one
47
48 of the greatest advantages of RP, namely the direct fabrication of patient-specific structures
49
50 independently how complex is the geometry of the defect.^[204,205] In the context of TERM, RP
51
52 techniques can be divided in two main strategies, namely (*i*) scaffold-based TERM systems, in
53
54 which is assumed that cells require a 3D structure acting as a cell guide and supporting template
55
56
57
58
59
60
61
62
63
64
65

1 that mimics the natural environment of the tissue to regenerate^[206,207] or (ii) scaffold-free TERM
2 systems, in which cell-cell interactions and self-organization are the main key points of the
3 system to regenerate the damaged tissue.^[208] While scaffold-based TERM strategies emphasize
4 the role of biomaterials as a supporting structure to guide cell function, minimizing the self-
5 assembly and self-organization capability of the encapsulated cells, scaffold-free TERM
6 reverses the importance of both contributions. A primary classification of the scaffold-based
7 RP techniques supporting biomedical applications can be made hinged on the working
8 principle, namely laser-based (photopolymerized hydrogels), nozzle-based (pre-polymers by
9 dint of extrusion/deposition), and printer-based (powder beds and deposition of a binder that
10 fuses the particles or directly depositing material using inkjet technology) systems. Although
11 the wide diversity of scaffold-based TERM systems produced by RP technologies for
12 biomedical applications, only some of them are compatible with the processing of hydrogels.
13 Additionally, among of them, the number of cell compatible RP technologies allowing mild
14 processing is still reduced, thus are unable to produce cell encapsulation systems. Different
15 studies using the referred scaffold-based RP techniques have been proposed, as reviewed
16 elsewhere,^[203] including a detailed discussion of the advantages and limitations of each
17 technique. On the other hand, scaffold-free RP techniques are based on the principle that cells
18 and tissues do not require an engineered biomaterial, due to their ability of self-assembly (the
19 autonomous organization of components without externally manipulation) and self-
20 organization. Based on the implementation of RP technology, a fascinated perspective on
21 scaffold-free TERM systems emerged, termed as bioprinting.^[209,210] Bioprinting emerged as the
22 process of creating 3D structures using a “bioink”, which was basically individual cells or
23 spheroids dispersed in a “biopaper”, i.e. hydrogels. Currently, bioprinting is a hot topic in the
24 TERM field, and also encompasses the use of other supporting materials by combining
25 bioprinting with other techniques, such as microcarrier^[211] or melt-electrowriting

1 technologies.^[212] The main advantage that bioprinting brought to the TERM field was the ability
2
3 to produce custom-made cell encapsulation systems for personalized treatment, while allowing
4
5 the precise positioning of cells and biologics in an automated fashion. With this technology
6
7 clinically-relevant 3D structures can be developed in a spatially controlled manner with high
8
9 precision over the shape, size, and cell location across the entire hydrogel 3D structure. The
10
11 main techniques currently used in bioprinting are (i) laser-based, (ii) droplet-based, including
12
13 inkjet, electro-hydrodynamic jet, acoustic-droplet-ejection, and micro-valve, and (iii)
14
15 extrusion.^[213] Using droplet-based or extrusion-based technologies, the bioink objects, typically
16
17 with spherical or cylindrical shape and composed of single or multiple cell types, are deposited
18
19 in well-defined topological patterns into biopaper sheets. Then, the obtained construct is
20
21 transferred to a bioreactor and the assembled bioink objects are fused. After that, the biopaper
22
23 can be removed, if required. Laser-based bioprinting utilizes a laser pulse directed via mirrors
24
25 onto a bioink layer above the substrate. Bioprinters can be classified in (i) nozzle-based, which
26
27 can be further divided into intermittent drop-wise printers, such as inkjet printers (both thermal
28
29 and piezoelectric), and continuous robotic dispensing printers, and (ii) dropwise nozzle-free,
30
31 which is based on laser-induced forward transfer printing techniques.^[214] In the case of inkjet
32
33 technology, individual or small cell clusters are printed. Despite the advantageous speed,
34
35 versatility and cost, high cell densities are difficult to obtain and considerable cell damage is
36
37 induced.^[215,216] On the other hand, extrusion-based bio-printers are more expensive but offer a
38
39 more “mild approach” towards cells.^[203] Common hydrogels proposed for bioprinting obtained
40
41 from natural polymers, such as collagen,^[217] hyaluronic acid,^[218] chitosan,^[219] gelatin,^[220] and
42
43 alginate.^[221] To be suitable for bioprinting, a hydrogel must be viscous enough to keep its shape
44
45 during printing and must have crosslinking abilities allowing the maintenance of the 3D
46
47 structure after printing. Crosslinking can occur by temperature change,^[217]
48
49 photopolymerization,^[222–225] and ionic crosslinking.^[221] A common challenge when bioprinting
50
51
52
53
54
55
56
57
58
59
60
61
62
63
64
65

1 hydrogels is that the printed shapes tend to collapse due to low viscosity. The viscosity of
2
3 alginate, the most widely used polymer in cell encapsulation, can be increased by varying the
4
5 concentration and molecular weight.^[226] However, it has not been sufficient for achieving shape
6
7 fidelity while printing.^[221] To increase the structural accuracy hydrogels are often printed in
8
9 combination with other materials. In such cases the printability of alginate has been improved
10
11 by the addition of gelatin,^[227] or by printing with a supporting sacrificial polymer.^[180] When
12
13 combined with nanofibrillated cellulose, alginate was successfully proposed to produce
14
15 anatomically shaped cartilage structures.^[221] Human ears and sheep meniscus were bioprinted,
16
17 encapsulating human chondrocytes. The complex cell encapsulation devices retained its shape
18
19 during the 7 days of *in vitro* culture, while assuring the viability of the encapsulated cells.
20
21 Composite bioinks have also been proposed to combine the above-referred advantages of
22
23 bioprinting in constructing clinically-relevant structures with biological cues.^[211] Aggregates
24
25 of MSCs and PLLA microparticles were produced via static culture or spinner flask expansion,
26
27 and further encapsulated in gelatin methacrylamide-gellan gum bioinks. Such hybrid bioink
28
29 was successfully proposed to construct bilayered osteochondral models.
30
31
32
33
34
35

36
37 **Table 4** summarizes the encapsulation matrix, geometry, type of encapsulated cells, and the
38
39 TERM application of examples of cell encapsulation systems using bioprinting.
40
41
42
43
44

45 **6. Cell encapsulated building blocks to generate complex functional systems**

46
47

48 The assembly of 3D scaffolds into building blocks is a strategic idea to overcome the main
49
50 drawbacks of TE approaches. Issues related with nutrients and waste diffusion, limited to a size
51
52 of ca. 200 μm , inhomogeneous cell distribution, as well as, the manipulation of biomaterials
53
54 microenvironment in space and time, can be solved with a modular approach. Larger number
55
56 of identical 3D engineered structures with smaller volumes can be assembled to create complex
57
58
59

1 functional tissues, while being structured across multiple length scales. In fact, the native tissues
2 are characterized by repetitive functional units, which include heterogeneous types of cells and
3 ECM, organized in a multiscale fashion.^[228] Furthermore, the development, maintenance, and
4 function of the tissue is regulated by the tissue form and architecture.^[229,230] Inspired by that,
5 different tissue building blocks are envisioned to recreate larger tissues with specific
6 microarchitectural features, while focus on an adequate multicellular geometry to promote a
7 proper remodeling. Such modular systems can be created by cell-sheets generation,^[231] cell
8 aggregation assembly,^[232] and by encapsulation of cells in microgels, subject that we will
9 deepen bellow. The assembly of these cell encapsulation systems as building blocks by bottom-
10 up approaches are discussed in terms of their variable 3D modular structures, comprising
11 spherical, fiber-shaped, and multi-shaped complex structures. **Figure 4** shows examples of
12 multi-shaped cell-laden hydrogels, which can be further assembled into clinically-relevant 3D
13 structures, based on the concept of modular TE.
14
15
16
17
18
19
20
21
22
23
24
25
26
27
28
29
30
31

32 **6.1 Spherical systems**

33
34
35
36
37
38
39 Spherical microgels are the most widely used geometry for modular TE. This type of shape
40 offers an optimal surface-to-volume ratio, providing an efficient mass exchange, thus favoring
41 a long-term cell viability. The micrometric size of such microgels allows minimal invasive
42 implantation procedures, while protecting cells from shear force damage during injection.^[233]
43
44 However, the main limitation is that dispersed spherical microgels cannot mimic the higher
45 order structure of native tissues. For that, several research groups have been using spherical cell
46 encapsulation systems as building blocks for bottom-up TE strategies. For example, the
47 assembly of MSCs-encapsulated microgels assembled by covalent crosslinking was proposed
48 for cartilage repair.^[234] Here, the 4-arm poly(ethylene glycol)-N-hydroxysuccinimide
49
50
51
52
53
54
55
56
57
58
59
60
61
62
63
64
65

1 crosslinker not only allowed the assembly of MSCs-encapsulated microgels, but also induced
2 spontaneous adhesion between the assembled construct and an *in vitro* tissue mimetic model.
3
4 Furthermore, such tissue mimetic model provided physical and biological cues for MSCs
5
6 chondrogenesis, leading to the production of a mature hyaline cartilage structure. Similarly,
7
8 spherical microgels encapsulating L929 cells were assembled in 3D macrostructures but using
9
10 another assembly technique, namely the LbL technology.^[235] After the nanometer multilayered
11
12 coating, the cell-laden template was liquefied by chelation, using EDTA treatment. Such
13
14 technology provided a bottom-up assembly in a scalable manner of individual compartments
15
16 for cells, while the liquefied core improved the ability of the construct for long-term cell
17
18 survival. Using the self-healing-driven assembly (SHDA) strategy, microgels were combined
19
20 with macrogels to facilitate the fabrication of various programmed materials toward biological
21
22 tissues.^[236] Smart macro- and microgels, fabricated by a controllable and continuous
23
24 microfluidic technique, were used as building blocks. Then, driven by the inherent hydrogen
25
26 bonds or supramolecular interactions between the gels, linear, planar, and 3D structures were
27
28 assembled. To enhance cell spreading and proliferation, specific cellular adhesion recognition
29
30 sites using the well-established RGD peptide sequence were added into the spherical hydrogels.
31
32 After the encapsulation of co-cultures of 3T3 and L929 cell lines, cell proliferation and
33
34 migration across the hydrogel boundaries were detected to interact with neighboring cells.
35
36 Single-cell-laden microgels find also great applicability as interesting building blocks for
37
38 modular TE strategies. The encapsulation of single cells not only allows a precise microscale
39
40 control of tissue assembly, but also enhances the ability of cells to respond to exogenous
41
42 stimuli.^[236] Single-cell microgels incorporated in injectable hydrogels were projected to
43
44 engineer multifunctional tissues via a modular approach.^[237] This strategy was leveraged to
45
46 incorporate immunoprotective single-cell-laden microgels within a proangiogenic macrogel.
47
48 Basically, the uncoupled micro- and macroenvironments, which are independently tunable,
49
50
51
52
53
54
55
56
57
58
59
60
61
62
63
64
65

1 were designed to create biomaterials with the multifunctionality typically found in native
2 tissues.
3
4

5 6 7 8 9 **6.2 Fiber-shaped system**

10
11 Fiber-shaped systems have attracted attention due to their unique and useful advantages. This
12 long, thin and flexible structures are being used as building blocks to facilitate higher-order
13 assemblies such as nanoscale materials,^[238,239] and textiles.^[240] Cell-laden fibers are also being
14 proposed to resemble hierarchical structures of the human body, such as blood vessels,^[180]
15 muscle fibers,^[241] and osteons from cortical bone.^[242] These microfibers can be composed
16 exclusively by cells,^[243] or in cell-embedding hydrogels mostly generated by microfluidics
17 technique. However, most of the hydrogels used are not composed by natural ECM proteins,
18 and thus, are insufficient to reconstruct the tissue microenvironment. Therefore, an ECM-based
19 encapsulation system, named meter-long core shell hydrogel microfibers, was proposed with
20 natural ECM proteins and cells to reconstitute the intrinsic cellular morphologies and functions
21 of native tissues.^[177] Additionally, the proposed microfibers were assembled into macroscopic
22 structures, by weaving and reeling, demonstrating higher-order assembly constructs with
23 various spatial patterns. The generation of 3D vascular networks is one of the major challenges
24 of TE. Inspired by that, the assembling of cell-laden fibers in a spatially defined manner was
25 proposed to form pre-vascularized adipose and hepatic tissues.^[244] Chitin- and alginate-based
26 fibers were assembled through interfacial polyelectrolyte complexation. Then, the
27 micropatterned niche was created by assembling a central ECs-laden fiber, surrounded by
28 parenchymal cells-laden fibers. Finally, the tertiary structure construct was obtained by
29 spooling and layering the repeat unit (secondary structure). Interestingly, *in vivo* studies show
30 the ability of the patterned constructs to anastomose with the host, leading to vascularized
31
32
33
34
35
36
37
38
39
40
41
42
43
44
45
46
47
48
49
50
51
52
53
54
55
56
57
58
59
60
61
62
63
64
65

1 tissues. Microfibers produced by microfluidics were also proposed as engineered osteons from
2 cortical bone.^[242] A biomimetic osteon-like structure was obtained by the encapsulation of ECs
3 in a middle layer, surrounded by an outer layer encapsulating human osteoblasts. Cell-laden
4 microfibers were assembled into braided strand, helical tube, knot, and woven structures. The
5 achieved double-layer hollow microfibers, exhibited not only a robust cell growth, but also up-
6 regulated gene expression.
7
8
9
10
11
12
13

14 **6.3 Multi-shaped complex structures**

15
16
17
18
19 Frequently, to promote a proper tissue remodeling as well as to simulate a certain tissue
20 functionality, the fabrication of multi-shaped complex 3D structures is proposed. Bottom-up
21 approaches to build vascular-like microchannels using cell-laden microgels have been widely
22 used.^[245] Generally, most of vascularized TE systems are designed as cell-free or cell-laden
23 bulk hydrogels via microfluidics. An array of microgels, encapsulating ECs and smooth muscle
24 cells, with a particular architectural design, were sequentially assembled in a controlled manner.
25
26
27
28
29
30
31
32
33
34
35
36
37
38
39
40
41
42
43
44
45
46
47
48
49
50
51
52
53
54
55
56
57
58
59
60
61
62
63
64
65

The microchannels of each microgel were assembled, to create an interconnected network mimicking the bifurcating structure of the native vasculature. Others have engineered the heart tissue using ring shaped molds.^[246] Native rat cardiomyocytes were mixed with collagen or Matrigel and then, assembled with other engineered cardiac bands on a cycling stretching device. Results show that the obtained cardiac grafts significantly improved cardiac function in rat myocardial infarct models, improving cell alignment and supporting contractile function of infarcted hearts.^[247] A different microgel-shaped structure was proposed for the fabrication of 3D multilayer hepatic lobule-like tissues.^[248] The drive was based on the fact that only an appropriate architectural organization of hepatocytes allows a proper functioning of the liver. Therefore, a new method for forming hepatic lobule-shaped microtissue was proposed using

1 poly(L-lysine)/alginate spherical microgels encapsulating rat liver cells. Using a repetitive one-
2
3 step micromanipulator system, four-layered hepatic lobule models were feasibly constructed,
4
5 demonstrating the applicability for *in vitro* artificial liver fabrication.
6

7
8 To precisely control the assembly process, the microgels can be functionalized in more complex
9
10 approaches. A molecular recognition-assisted self-assembly strategy was proposed through the
11
12 surface functionalization of cube-shaped gels with single-stranded segments of DNA.^[249]
13
14 Acting like sequence-specific glue, this strategy can control the assembly of microgels to create
15
16 complex microarchitectures. Another strategy is to use “magnetoceptive” hydrogel subunits
17
18 that self-assemble into 3D structures. The assembly of cell-laden microgels was achieved
19
20 through the paramagnetism of free radicals as a driving mechanism. Under a permanent
21
22 generated magnetic field, complex heterogeneous structures could be built.
23
24
25
26
27
28
29
30

31 **7. Conclusion**

32
33
34 Lessons afforded by the primordial cell encapsulation systems were of outmost importance for
35
36 the evolution path that culminated in the current hydrogels encapsulating cells for tissue
37
38 regeneration. The complexity of the current cell-laden hydrogels is being enriched, due to their
39
40 recognize potential as modules for bottom-up TE. Consequently, the new generation of
41
42 hydrogels for cell encapsulation possess a hierarchical and highly complex organization,
43
44 including smart and adaptive matrices with adequate environmental signals able to mimic the
45
46 regeneration process of native tissues. Given the importance of physicochemical cues of the
47
48 hydrogel matrices on cell behavior, we believe that a fine control over chemical and mechanical
49
50 cues might boost their application towards the clinics. For example, by designing hydrogels
51
52 with controlled stiffness, fabricating proteolytically degradable hydrogels, or by enriching the
53
54 cell encapsulation matrices with bioactive molecules, such as soluble particles derived from
55
56
57
58
59
60
61
62
63
64
65

1 decellularized tissues, will boost cell encapsulation systems to the next level. Importantly, the
2
3 process must allow low-cost production, scale up manufacturing, and easy handling and
4
5 implementation. Such characteristics are fundamental to facilitate the market acceptance by the
6
7 target community, including the applicers, surgeons; those who will receive the treatment,
8
9 patients; those who will commercialize it, the healthcare providers; and ultimately those who
10
11 will allow such flow of events, namely the ethical authorities, such as the FDA and equivalents.
12
13 Additionally, to ensure a successful translation into the clinics, issues related to long-term cell
14
15 viability, tissue integration, and risk of FBR due to an acute immune response, should be
16
17 carefully evaluated. In fact, using the well-established 3D and highly-hydrated environment of
18
19 hydrogels for cell encapsulation, new technologies are being increasingly proposed for the
20
21 development of systems with tissue-like complexity. But, technologies how to assemble such
22
23 units are mandatory One of the major challenges on using cell encapsulation systems as building
24
25 blocks for modular TE is the spatial and time resolution to create 4D realistic modules that
26
27 precisely tune the heterogeneity of native tissues. Combining bioprinting with technologies to
28
29 enrich the multifunctionality of the encapsulation matrix, and with more sophisticated
30
31 assembly-techniques able to generate 3D structures with tissue-like complexity are mandatory.
32
33 Such automated techniques might rely on biomaterials with self-assembly or stimuli-responsive
34
35 (e.g. acoustic, light, pH, temperature) capability. Additionally, the development of highly
36
37 ambitious bioinks possessing print-fidelity and biological cues to guide cell behavior are also
38
39 required. While this more engineer atmosphere is under the spotlight of the scientific
40
41 community, also the importance of the immune system contribution into the regenerative
42
43 process cannot be depreciated, and thus must be considered when designing novel TE systems.
44
45 To achieve such ambitious goal the *cliché* of a “multidisciplinary team requisite” was never so
46
47 demanding in the TERM field as today.
48
49
50
51
52
53
54
55
56
57
58
59
60
61
62
63
64
65

Acknowledgements

The authors are grateful to the Portuguese Foundation for Science and Technology (FCT) for the Doctoral grant of Sara Nadine with the reference number SFRH/BD/130194/2017, and funding for project CIRCUS (PTDC/BTM-MAT/31064/2017). The authors also acknowledge funding from the European Research Council for project ATLAS (grant agreement ERC-2014-ADG-669858). This work was developed within the scope of the project CICECO-Aveiro Institute of Materials, FCT Ref. UID/CTM/50011/2019, financed by national funds through the FCT/MCTES.

Received: ((will be filled in by the editorial staff))

Revised: ((will be filled in by the editorial staff))

Published online: ((will be filled in by the editorial staff))

References

- [1] V. Bisceglie, *Z. Krebsforsch.* **1934**, *40*, 122.
- [2] G. H. Algire, J. M. Weaver, R. T. Prehn, *JNCI J. Natl. Cancer Inst.* **1954**, *15*, 493.
- [3] D. Ferreira, E. Westman, H. Eyjolfsdottir, P. Almqvist, G. Lind, B. Linderöth, Å. Seiger, K. Blennow, A. Karami, T. Darreh-Shori, et al., *J. Alzheimer's Dis.* **2014**, *43*, 1059.
- [4] R. Krishnan, M. Alexander, L. Robles, C. E. Foster, J. R. T. Lakey, *Rev. Diabet. Stud.* **2014**, *11*, 84.
- [5] O. Lindvall, L. U. Wahlberg, *Exp. Neurol.* **2008**, *209*, 82.

- 1 [6] W. Chick, A. Like, V. Lauris, *Science* (80-.). **1975**, 187, 847.
2
3
4 [7] F. Lim, A. Sun, *Science* (80-.). **1980**, 210, 908.
5
6
7 [8] F. B. Barton, M. R. Rickels, R. Alejandro, B. J. Hering, S. Wease, B. Naziruddin, J.
8 Oberholzer, J. S. Odorico, M. R. Garfinkel, M. Levy, et al., *Diabetes Care* **2012**, 35,
9 1436.
10
11
12
13
14
15 [9] V. M. Gaspar, P. Lavrador, J. Borges, M. B. Oliveira, J. F. Mano, *Adv. Mater.* **2020**.
16
17
18 [10] S. Khetan, J. A. Burdick, *Soft Matter* **2011**, 7, 830.
19
20
21
22 [11] A. K. A. S. Brun-Graepi, C. Richard, M. Bessodes, D. Scherman, O.-W. Merten, *J.*
23 *Control. Release* **2011**, 149, 209.
24
25
26
27 [12] S. Mazzitelli, L. Capretto, F. Quinci, R. Piva, C. Nastruzzi, *Adv. Drug Deliv. Rev.* **2013**,
28 65, 1533.
29
30
31
32
33 [13] H. Tian, Z. Tang, X. Zhuang, X. Chen, X. Jing, *Prog. Polym. Sci.* **2012**, 37, 237.
34
35
36
37 [14] J. L. Wilson, T. C. McDevitt, *Biotechnol. Bioeng.* **2013**, 110, 667.
38
39
40 [15] A. Singh, N. A. Peppas, *Adv. Mater.* **2014**, 26, 6530.
41
42
43 [16] A. Vishwakarma, N. S. Bhise, M. B. Evangelista, J. Rouwkema, M. R. Dokmeci, A. M.
44 Ghaemmaghami, N. E. Vrana, A. Khademhosseini, *Trends Biotechnol.* **2016**, 34, 470.
45
46
47
48 [17] G. M. Peretti, V. Campo-Ruiz, S. Gonzalez, M. A. Randolph, J. Wei Xu, K. R. Morse,
49 R. E. Roses, M. J. Yaremchuk, *Connect. Tissue Res.* **2006**, 47, 190.
50
51
52
53 [18] A. K. A. S. Brun-Graepi, C. Richard, M. Bessodes, D. Scherman, O.-W. Merten, *J.*
54 *Control. Release* **2011**, 149, 209.
55
56
57
58
59
60
61
62
63
64
65

- 1 [19] F. Lim, A. M. Sun, *Science* **1980**, *210*, 908.
2
3
4 [20] L. Gasperini, J. F. Mano, R. L. Reis, *J. R. Soc. Interface* **2014**, *11*, DOI
5 10.1098/rsif.2014.0817.
6
7
8
9 [21] T. H. Silva, A. Alves, B. M. Ferreira, J. M. Oliveira, L. L. Reys, R. J. F. Ferreira, R. A.
10 Sousa, S. S. Silva, J. F. Mano, R. L. Reis, *Int. Mater. Rev.* **2012**, *57*, 276.
11
12
13
14 [22] J. J. Schmidt, J. Rowley, H. J. Kong, *J. Biomed. Mater. Res. Part A* **2008**, *87A*, 1113.
15
16
17
18 [23] P. De Vos, H. A. Lazarjani, D. Poncelet, M. M. Faas, *Adv. Drug Deliv. Rev.* **2014**, *67–*
19 *68*, 15.
20
21
22
23 [24] J. F. Mano, G. A. Silva, H. S. Azevedo, P. B. Malafaya, R. A. Sousa, S. S. Silva, L. F.
24 Boesel, J. M. Oliveira, T. C. Santos, A. P. Marques, et al., *J. R. Soc. Interface* **2007**, *4*,
25 999.
26
27
28
29 [25] S. Mazzitelli, L. Capretto, F. Quinci, R. Piva, C. Nastruzzi, *Adv. Drug Deliv. Rev.* **2013**,
30 *65*, 1533.
31
32
33 [26] R. Silva, R. Singh, B. Sarker, D. G. Papageorgiou, J. A. Juhasz, J. A. Roether, I. Cicha,
34 J. Kaschta, D. W. Schubert, K. Chrissafis, et al., *J. Mater. Chem. B* **2014**, *2*, 5441.
35
36
37
38 [27] E. G. Popa, S. G. Caridade, J. F. Mano, R. L. Reis, M. E. Gomes, *J. Tissue Eng. Regen.*
39 *Med.* **2015**, *9*, 550.
40
41
42
43 [28] S. C. Neves, D. B. Gomes, A. Sousa, S. J. Bidarra, P. Petrini, L. Moroni, C. C. Barrias,
44 P. L. Granja, *J. Mater. Chem. B* **2015**, *3*, 2096.
45
46
47
48 [29] P. de Vos, M. M. Faas, B. Strand, R. Calafiore, *Biomaterials* **2006**, *27*, 5603.
49
50
51
52 [30] P. de Vos, M. Bučko, P. Gemeiner, M. Navrátil, J. Švitel, M. Faas, B. L. Strand, G.
53
54
55
56
57
58
59
60
61
62
63
64
65

- 1 Skjak-Braek, Y. A. Morch, A. Vikartovská, et al., *Biomaterials* **2009**, *30*, 2559.
- 2
- 3
- 4 [31] E. H. Nafea, A. M. L. A. Poole-Warren, P. J. Martens, *J. Control. Release* **2011**, *154*,
- 5
- 6 110.
- 7
- 8
- 9
- 10 [32] T. T. Pham, T. T. Nguyen, S. Pathak, S. Regmi, H. T. Nguyen, T. H. Tran, C. S. Yong,
- 11
- 12 J. O. Kim, P. H. Park, M. H. Park, et al., *Biomaterials* **2018**, *154*, 182.
- 13
- 14
- 15 [33] S. Pathak, T. T. Pham, J. H. Jeong, Y. Byun, *J. Control. Release* **2019**, *305*, 176.
- 16
- 17
- 18 [34] J. R. Day, A. David, A. L. Cichon, T. Kulkarni, M. Cascalho, A. Shikanov, *J. Biomed.*
- 19
- 20 *Mater. Res. Part A* **2018**, *106*, 1381.
- 21
- 22
- 23
- 24 [35] J. Borges, J. F. Mano, *Chem. Rev.* **2014**, *114*, 8883.
- 25
- 26
- 27 [36] R. R. Costa, J. F. Mano, *Chem. Soc. Rev.* **2014**, *43*, 3453.
- 28
- 29
- 30
- 31 [37] C. R. Correia, I. M. Bjørge, J. Zeng, M. Matsusaki, J. F. Mano, *Adv. Healthc. Mater.*
- 32
- 33 **2019**, *8*, 1901221.
- 34
- 35
- 36 [38] S. Zigon-Branc, M. Markovic, J. Van Hoorick, S. Van Vlierberghe, P. Dubruel, E.
- 37
- 38 Zerobin, S. Baudis, A. Ovsianikov, *Tissue Eng. - Part A* **2019**, *25*, 1369.
- 39
- 40
- 41
- 42 [39] M. Bao, J. Xie, W. T. S. Huck, *Adv. Sci.* **2018**, *5*, 1800448.
- 43
- 44
- 45 [40] C. Liu, J. W. Luo, T. Liang, L. X. Lin, Z. P. Luo, Y. Q. Zhuang, Y. L. Sun, *Exp. Cell*
- 46
- 47 *Res.* **2018**, *373*, 62.
- 48
- 49
- 50
- 51 [41] D. Richards, J. Swift, L. S. Wong, S. M. Richardson, in *Adv. Exp. Med. Biol.*, Springer
- 52
- 53 New York LLC, **2019**, pp. 53–69.
- 54
- 55
- 56 [42] O. Chaudhuri, L. Gu, D. Klumpers, M. Darnell, S. A. Bencherif, J. C. Weaver, N.
- 57
- 58
- 59
- 60
- 61
- 62
- 63
- 64
- 65

- 1 Huebsch, H. Lee, E. Lippens, G. N. Duda, et al., *Nat. Mater.* **2016**, *15*, 326.
2
3
4 [43] R. K. Das, V. Gocheva, R. Hammink, O. F. Zouani, A. E. Rowan, *Nat. Mater.* **2016**, *15*,
5
6 318.
7
8
9 [44] M. Park, D. Lee, J. Hyun, *Carbohydr. Polym.* **2015**, *116*, 223.
10
11
12 [45] C. Cha, S. Y. Kim, L. Cao, H. Kong, *Biomaterials* **2010**, *31*, 4864.
13
14
15
16 [46] S. Hong, D. Sycks, H. F. Chan, S. Lin, G. P. Lopez, F. Guilak, K. W. Leong, X. Zhao,
17
18 *Adv. Mater.* **2015**, *27*, 4035.
19
20
21 [47] A. M. S. Costa, J. F. Mano, *Eur. Polym. J.* **2015**, *72*, 344.
22
23
24 [48] A. M. S. Costa, J. F. Mano, *Chem. Commun.* **2015**, *51*, 15673.
25
26
27 [49] † Mark A. Rice, † and Johannah Sanchez-Adams, †,‡ Kristi S. Anseth*, **2006**, DOI
28
29 10.1021/BM060086+.
30
31
32
33 [50] R. F. Pereira, C. C. Barrias, P. J. Bártolo, P. L. Granja, *Acta Biomater.* **2018**, *66*, 282.
34
35
36
37 [51] A. H. Aziz, S. J. Bryant, *Biotechnol. Bioeng.* **2019**, *116*, 1523.
38
39
40 [52] M. P. Lutolf, J. L. Lauer-Fields, H. G. Schmoekel, A. T. Metters, F. E. Weber, G. B.
41
42 Fields, J. A. Hubbell, *Proc. Natl. Acad. Sci. U. S. A.* **2003**, *100*, 5413.
43
44
45 [53] M. P. Lutolf, G. P. Raeber, A. H. Zisch, N. Tirelli, J. A. Hubbell, *Adv. Mater.* **2003**, *15*,
46
47 888.
48
49
50 [54] N. Cui, J. Qian, T. Liu, N. Zhao, H. Wang, *Carbohydr. Polym.* **2015**, *126*, 192.
51
52
53 [55] K. Xu, F. Lee, S. Gao, M.-H. Tan, M. Kurisawa, *J. Control. Release* **2015**, *216*, 47.
54
55
56 [56] S. C. Neves, R. F. Pereira, M. Araújo, C. C. Barrias, in *Pept. Proteins as Biomater. Tissue*
57
58
59

1 *Regen. Repair*, Elsevier Inc., **2018**, pp. 101–125.

- 2
3
4 [57] Y. Aizawa, S. C. Owen, M. S. Shoichet, *Prog. Polym. Sci.* **2012**, *37*, 645.
5
6
7 [58] Y. Park, M. P. Lutolf, J. A. Hubbell, E. B. Hunziker, M. Wong, *Tissue Eng.* **2004**, *10*,
8
9 515.
10
11
12 [59] L. Almany, D. Seliktar, *Biomaterials* **2005**, *26*, 2467.
13
14
15
16 [60] M. B. Oliveira, O. Kossover, J. F. Mano, D. Seliktar, *Acta Biomater.* **2015**, *13*, 78.
17
18
19 [61] N. Huebsch, E. Lippens, K. Lee, M. Mehta, S. T. Koshy, M. C. Darnell, R. M. Desai, C.
20
21 M. Madl, M. Xu, X. Zhao, et al., *Nat. Mater.* **2015**, *14*, 1269.
22
23
24
25 [62] D. F. Williams, *Biomaterials* **2008**, *29*, 2941.
26
27
28 [63] A. M. A. Rokstad, I. Lacić, B. L. Strand, *Adv. Drug Deliv. Rev.* **2014**, *67–68*, 111.
29
30
31
32 [64] B. D. Ratner, *J. Cardiovasc. Transl. Res.* **2011**, *4*, 523.
33
34
35 [65] A. J. Vegas, O. Veiseh, J. C. Doloff, M. Ma, H. H. Tam, K. Bratlie, J. Li, A. R. Bader,
36
37 E. Langan, K. Olejnik, et al., *Nat. Biotechnol.* **2016**, *34*, 345.
38
39
40
41 [66] O. Veiseh, J. C. Doloff, M. Ma, A. J. Vegas, H. H. Tam, A. R. Bader, J. Li, E. Langan,
42
43 J. Wyckoff, W. S. Loo, et al., *Nat. Mater.* **2015**, *14*, 643.
44
45
46 [67] M. B. Oliveira, M. P. Ribeiro, S. P. Miguel, A. I. Neto, P. Coutinho, I. J. Correia, J. F.
47
48 Mano, *Tissue Eng. Part C Methods* **2014**, *20*, 851.
49
50
51
52 [68] T. A. Wynn, K. M. Vannella, *Immunity* **2016**, *44*, 450.
53
54
55
56 [69] L. M. Wahl, S. M. Wahl, L. E. Smythies, P. D. Smith, in *Curr. Protoc. Immunol.*, John
57
58 Wiley & Sons, Inc., Hoboken, NJ, USA, **2006**, pp. 7.6A.1-7.6A.10.
59
60

- 1 [70] M. Saclier, H. Yacoub-Youssef, A. L. Mackey, L. Arnold, H. Ardjoune, M. Magnan, F.
2
3 Sailhan, J. Chelly, G. K. Pavlath, R. Mounier, et al., *Stem Cells* **2013**, *31*, 384.
4
5
6 [71] K. N. Ekdahl, J. D. Lambris, H. Elwing, D. Ricklin, P. H. Nilsson, Y. Teramura, I. A.
7
8 Nicholls, B. Nilsson, *Adv. Drug Deliv. Rev.* **2011**, *63*, 1042.
9
10
11 [72] M. B. Gorbet, M. V. Sefton, in *Biomater. Silver Jubil. Compend.*, Elsevier Ltd, **2006**, pp.
12
13 219–241.
14
15
16 [73] A. D. Kennedy, F. R. Deleo, *Immunol. Res.* **2009**, *43*, 25.
17
18
19 [74] L. Tang, T. A. Jennings, J. W. Eaton, *Proc. Natl. Acad. Sci. U. S. A.* **1998**, *95*, 8841.
20
21
22 [75] J. M. Anderson, A. Rodriguez, D. T. Chang, *Semin. Immunol.* **2008**, *20*, 86.
23
24
25 [76] D. M. Mosser, J. P. Edwards, *Nat. Rev. Immunol.* **2008**, *8*, 958.
26
27
28 [77] J. M. Anderson, *Annu. Rev. Mater. Res.* **2001**, *31*, 81.
29
30
31 [78] M. E. Scarritt, R. Londono, S. F. Badylak, in *Immune Response to Implant. Mater.*
32
33 *Devices Impact Immune Syst. Success an Implant*, Springer International Publishing,
34
35 **2016**, pp. 1–14.
36
37
38 [79] J. I. Andorko, C. M. Jewell, *Bioeng. Transl. Med.* **2017**, *2*, 139.
39
40
41 [80] G. Broughton, J. E. Janis, C. E. Attinger, *Plast. Reconstr. Surg.* **2006**, *117*, 12S.
42
43
44 [81] A. Vishwakarma, N. S. Bhise, M. B. Evangelista, J. Rouwkema, M. R. Dokmeci, A. M.
45
46
47
48
49
50
51
52
53
54
55
56
57
58
59
60
61
62
63
64
65

- 1 [83] T. A. Wynn, A. Chawla, J. W. Pollard, *Nature* **2013**, 496, 445.
2
3
4 [84] C. R. Correia, J. Gaifem, M. B. Oliveira, R. Silvestre, J. F. Mano, *Biomater. Sci.* **2017**,
5
6 5, 551.
7
8
9
10 [85] L. Gasperini, J. F. Mano, R. L. Reis, *J. R. Soc. Interface* **2014**, 11, 20140817.
11
12
13 [86] Y. S. Zhang, A. Khademhosseini, *Science* (80-.). **2017**, 356, DOI
14
15 10.1126/science.aaf3627.
16
17
18
19 [87] E. Mariani, G. Lisignoli, R. M. Borzì, L. Pulsatelli, *Int. J. Mol. Sci.* **2019**, 20, 636.
20
21
22 [88] A. H. Morris, D. K. Stamer, T. R. Kyriakides, *Semin. Immunol.* **2017**, 29, 72.
23
24
25 [89] K. Sadtler, M. T. Wolf, S. Ganguly, C. A. Moad, L. Chung, S. Majumdar, F. Housseau,
26
27 D. M. Pardoll, J. H. Elisseeff, *Biomaterials* **2019**, 192, 405.
28
29
30
31 [90] T. Yuan, L. Zhang, K. Li, H. Fan, Y. Fan, J. Liang, X. Zhang, *J. Biomed. Mater. Res.*
32
33 *Part B Appl. Biomater.* **2014**, 102, 337.
34
35
36
37 [91] C. Helary, A. Abed, G. Mosser, L. Louedec, A. Meddahi-Pellé, M. M. Giraud-Guille, *J.*
38
39 *Tissue Eng. Regen. Med.* **2011**, 5, 248.
40
41
42
43 [92] M. L. Foglia, R. Mitarotonda, M. C. De Marzi, M. F. Desimone, *Mater. Sci. Eng. C* **2019**,
44
45 99, 47.
46
47
48
49 [93] M. I. Oliveira, S. G. Santos, M. J. Oliveira, A. L. Torres, M. A. Barbosa, **2012**, 24, 136.
50
51
52 [94] M. Mounayar, E. Kefaloyianni, B. Smith, Z. Solhjoui, O. H. Maarouf, J. Azzi, L.
53
54 Chabtini, P. Fiorina, M. Kraus, R. Briddell, et al., *Stem Cells* **2015**, 33, 1892.
55
56
57
58 [95] J. A. Ankrum, J. F. Ong, J. M. Karp, *Nat. Biotechnol.* **2014**, 32, 252.
59
60
61
62
63
64
65

- 1 [96] M. D. Swartzlander, A. K. Blakney, L. D. Amer, K. D. Hankenson, T. R. Kyriakides, S.
2
3 J. Bryant, *Biomaterials* **2015**, *41*, 79.
4
5
6 [97] A. Uccelli, L. Moretta, V. Pistoia, *Nat. Rev. Immunol.* **2008**, *8*, 726.
7
8
9 [98] G. Chamberlain, J. Fox, B. Ashton, J. Middleton, *Stem Cells* **2007**, *25*, 2739.
10
11
12 [99] A. Papalamprou, C. W. Chang, N. Vapniarsky, A. Clark, N. Walker, L. G. Griffiths, *Acta*
13
14 *Biomater.* **2016**, *45*, 155.
15
16
17 [100] E. R. Molina, B. T. Smith, S. R. Shah, H. Shin, A. G. Mikos, *J. Control. Release* **2015**,
18
19 *219*, 107.
20
21
22 [101] J. Yang, X. Chen, T. Yuan, X. Yang, Y. Fan, X. Zhang, *Mater. Sci. Eng. C* **2017**, *70*,
23
24 983.
25
26
27 [102] J. Ding, B. Chen, T. Lv, X. Liu, X. Fu, Q. Wang, L. Yan, N. Kang, Y. Cao, R. Xiao,
28
29 *Stem Cells Transl. Med.* **2016**, *5*, 1079.
30
31
32 [103] A. K. Berglund, L. A. Fortier, D. F. Antczak, L. V. Schnabel, *Stem Cell Res. Ther.* **2017**,
33
34 8, DOI 10.1186/s13287-017-0742-8.
35
36
37 [104] J. H. Pig, A. Ishihara, M. L. Wellman, D. S. Russell, A. L. Bertone, *Vet. Comp. Orthop.*
38
39 *Traumatol.* **2013**, *26*, 453.
40
41
42 [105] A. J. Joswig, A. Mitchell, K. J. Cummings, G. J. Levine, C. A. Gregory, R. Smith, A. E.
43
44 Watts, *Stem Cell Res. Ther.* **2017**, *8*, 42.
45
46
47 [106] K. L. Spiller, S. Nassiri, C. E. Witherel, R. R. Anfang, J. Ng, K. R. Nakazawa, T. Yu, G.
48
49 Vunjak-Novakovic, *Biomaterials* **2015**, *37*, 194.
50
51
52 [107] N. E. Vrana, *Futur. Sci. OA* **2016**, *2*, DOI 10.4155/fsoa-2016-0060.
53
54
55
56
57
58
59
60
61
62
63
64
65

- 1 [108] G. Hübner, M. Brauchle, H. Smola, M. Madlener, R. Fässler, S. Werner, *Cytokine* **1996**,
2 8, 548.
3
4
5
6 [109] J. Sun, S.-H. Li, S.-M. Liu, J. Wu, R. D. Weisel, Y.-F. Zhuo, T. M. Yau, R.-K. Li, S. S.
7 Fazel, *Am. J. Physiol. Circ. Physiol.* **2009**, 296, H43.
8
9
10
11 [110] L. K. Mathew, S. Sengupta, A. Kawakami, E. A. Andreasen, C. V. Löhr, C. A. Loynes,
12 S. A. Renshaw, R. T. Peterson, R. L. Tanguay, *J. Biol. Chem.* **2007**, 282, 35202.
13
14
15
16 [111] S. E. Hanson, S. N. King, J. Kim, X. Chen, S. L. Thibeault, P. Hematti, *Tissue Eng. Part*
17 *A* **2011**, 17, 2463.
18
19
20
21 [112] S. N. King, S. E. Hanson, X. Chen, J. Kim, P. Hematti, S. L. Thibeault, *J. Biomed. Mater.*
22 *Res. Part A* **2014**, 102, 890.
23
24
25
26 [113] H. Wang, R. T. Morales, X. Cui, J. Huang, W. Qian, J. Tong, W. Chen, *Adv. Healthc.*
27 *Mater.* **2018**, 8, 1801234.
28
29
30
31 [114] B.-H. Cha, S. R. Shin, J. Leijten, Y.-C. Li, S. Singh, J. C. Liu, N. Annabi, R. Abdi, M.
32 R. Dokmeci, N. E. Vrana, et al., *Adv. Healthc. Mater.* **2017**, 6, 1700289.
33
34
35 [115] A. R. Donaldson, C. E. Tanase, D. Awuah, P. Vasanthi Bathrinarayanan, L. Hall, M.
36 Nikkhah, A. Khademhosseini, F. Rose, C. Alexander, A. M. Ghaemmaghami, *Front.*
37 *Bioeng. Biotechnol.* **2018**, 6, 116.
38
39
40
41 [116] N. Grotenhuis, Y. Bayon, J. F. Lange, G. J. V. M. Van Osch, Y. M. Bastiaansen-
42 Jenniskens, *Biochem. Biophys. Res. Commun.* **2013**, 433, 115.
43
44
45
46 [117] N. Grotenhuis, S. F. H. De Witte, G. J. V. M. van Osch, Y. Bayon, J. F. Lange, Y. M.
47 Bastiaansen-Jenniskens, *Tissue Eng. Part A* **2016**, 22, 1098.
48
49
50
51
52
53
54
55
56
57
58
59
60
61
62
63
64
65

- 1 [118] C. Dollinger, S. Ciftci, H. Knopf-Marques, R. Guner, A. M. Ghaemmaghami, C. Debry,
2
3 J. Barthes, N. E. Vrana, *J. Tissue Eng. Regen. Med.* **2018**, *12*, 330.
4
5
- 6 [119] J. Barthes, C. Dollinger, C. B. Muller, U. Liivas, A. Dupret-Bories, H. Knopf-Marques,
7
8 N. E. Vrana, *Front. Bioeng. Biotechnol.* **2018**, *6*, 108.
9
- 10
11
12 [120] E. M. Moore, G. Ying, J. L. West, *Adv. Biosyst.* **2017**, *1*, 1600021.
13
14
- 15 [121] D. A. Cantu, P. Hematti, W. J. Kao, *Stem Cells Transl. Med.* **2012**, *1*, 740.
16
17
- 18 [122] X.-T. He, R.-X. Wu, X.-Y. Xu, J. Wang, Y. Yin, F.-M. Chen, *Acta Biomater.* **2018**, *71*,
19
20 132.
21
22
- 23 [123] S. V. Bhujbal, B. De Haan, S. P. Niclou, P. De Vos, *Sci. Rep.* **2014**, *4*, DOI
24
25 10.1038/srep06856.
26
27
- 28 [124] S. Ponce, G. Orive, R. Hernández, A. R. Gascóon, J. L. Pedraz, B. J. de Haan, M. M.
29
30 Faas, M. M. Faas, H. J. Mathieu, P. de Vos, *Biomaterials* **2006**, *27*, 4831.
31
32
- 33 [125] U. Zimmermann, F. Thürmer, A. Jork, M. Weber, S. Mimietz, M. Hillgärtner, F.
34
35 Brunnenmeier, H. Zimmermann, I. Westphal, G. Fuhr, et al., *Ann. N. Y. Acad. Sci.* **2006**,
36
37 *944*, 199.
38
39
- 40 [126] S. Hanson, R. N. D'Souza, P. Hematti, *Tissue Eng. Part A* **2014**, *20*, 2162.
41
42
- 43 [127] J. M. Anderson, *Regen. Biomater.* **2016**, *3*, 73.
44
45
- 46 [128] G. Decher, J. D. Hong, J. Schmitt, *Thin Solid Films* **1992**, *210–211*, 831.
47
48
- 49 [129] K. Ariga, Y. M. Lvov, K. Kawakami, Q. Ji, J. P. Hill, *Adv. Drug Deliv. Rev.* **2011**, *63*,
50
51 762.
52
53
54
55
56
57
58
59
60
61
62
63
64
65

- 1 [130] S. De Koker, L. J. De Cock, P. Rivera-Gil, W. J. Parak, R. Auzély Velty, C. Vervaet, J.
2
3 P. Remon, J. Grooten, B. G. De Geest, *Adv. Drug Deliv. Rev.* **2011**, *63*, 748.
4
5
6 [131] R. R. Costa, E. Castro, F. J. Arias, J. C. Rodríguez-Cabello, J. F. Mano,
7
8 *Biomacromolecules* **2013**, *14*, 2403.
9
10
11 [132] J. Kim, D. R. Arifin, N. Muja, T. Kim, A. A. Gilad, H. Kim, A. Arepally, T. Hyeon, J.
12
13 W. M. Bulte, *Angew. Chemie Int. Ed.* **2011**, *50*, 2317.
14
15
16 [133] R. Cheng, F. Meng, C. Deng, H.-A. Klok, Z. Zhong, *Biomaterials* **2013**, *34*, 3647.
17
18
19 [134] R. R. Costa, C. A. Custódio, F. J. Arias, J. C. Rodríguez-Cabello, J. F. Mano,
20
21
22 *Nanomedicine Nanotechnology, Biol. Med.* **2013**, *9*, 895.
23
24
25 [135] J. M. Silva, A. R. C. Duarte, S. G. Caridade, C. Picart, R. L. Reis, J. F. Mano,
26
27
28 *Biomacromolecules* **2014**, *15*, 3817.
29
30
31 [136] H.-Y. Lee, K. R. Tiwari, S. R. Raghavan, *Soft Matter* **2011**, *7*, 3273.
32
33
34 [137] C. Lin, W. Zhu, H. Yang, Q. An, C. Tao, W. Li, J. Cui, Z. Li, G. Li, *Angew. Chemie*
35
36
37 **2011**, *50*, 4947.
38
39
40 [138] B. G. De Geest, M. A. Willart, B. N. Lambrecht, C. Pollard, C. Vervaet, J. P. Remon, J.
41
42
43 Grooten, S. De Koker, *Angew. Chemie Int. Ed.* **2012**, *51*, 3862.
44
45
46 [139] M. Sanlés-Sobrido, M. Pérez-Lorenzo, B. Rodríguez-González, V. Salgueiriño, M. A.
47
48
49 Correa-Duarte, *Angew. Chemie* **2012**, *124*, 3943.
50
51
52 [140] Q. Jin, J. Bao, H. Sakiyama, N. Tsubaki, *Res. Chem. Intermed.* **2011**, *37*, 177.
53
54
55 [141] M. B. Oliveira, J. Hatami, J. F. Mano, *Chem. - An Asian J.* **2016**, *11*, 1753.
56
57
58
59
60
61
62
63
64
65

- 1 [142] R. R. Costa, M. Alatorre-Meda, J. F. Mano, *Biotechnol. Adv.* **2015**, *33*, 1310.
2
3
- 4 [143] D. M. Lynn, *Soft Matter* **2006**, *2*, 269.
5
6
- 7 [144] S. M. Oliveira, V. E. Santo, M. E. Gomes, R. L. Reis, J. F. Mano, *Biomaterials* **2015**,
8
9 48, 56.
10
11
- 12 [145] K. Ariga, Y. Yamauchi, G. Rydzek, Q. Ji, Y. Yonamine, K. C.-W. Wu, J. P. Hill, *Chem.*
13
14 *Lett.* **2014**, *43*, 36.
15
16
- 17 [146] J. J. Richardson, M. Björnmalm, F. Caruso, *Science (80-.)*. **2015**, *348*, DOI
18
19 10.1126/science.aaa2491.
20
21
- 22 [147] N. L. Costa, P. Sher, J. F. Mano, *Adv. Eng. Mater.* **2011**, *13*, B218.
23
24
- 25 [148] C. R. Correia, S. Gil, R. L. Reis, J. F. Mano, *Adv. Healthc. Mater.* **2016**, *5*, 1346.
26
27
- 28 [149] C. R. Correia, R. P. Pirraco, M. T. Cerqueira, A. P. Marques, R. L. Reis, J. F. Mano, *Sci.*
29
30 *Rep.* **2016**, *6*, 21883.
31
32
- 33 [150] C. R. Correia, T. C. Santos, R. P. Pirraco, M. T. Cerqueira, A. P. Marques, R. L. Reis, J.
34
35 F. Mano, *Acta Biomater.* **2017**, *53*, 483.
36
37
- 38 [151] S. Nadine, S. G. Patrício, C. R. Correia, J. Mano, *Biofabrication* **2019**, DOI
39
40 10.1088/1758-5090/ab3e16.
41
42
- 43 [152] C. R. Correia, T. C. Santos, R. P. Pirraco, M. T. Cerqueira, A. P. Marques, R. L. Reis, J.
44
45 F. Mano, *Acta Biomater.* **2017**, *53*, 483.
46
47
- 48 [153] C. R. Correia, M. Ghasemzadeh-Hasankolaei, J. F. Mano, *PLoS One* **2019**, *14*,
49
50 e0218045.
51
52
53
54
55
56
57
58
59
60
61
62
63
64
65

- 1 [154] P. Sher, C. R. Correia, R. R. Costa, J. F. Mano, *RSC Adv.* **2015**, *5*, 2511.
2
3
4 [155] D. S. Couto, N. M. Alves, J. F. Mano, *J. Nanosci. Nanotechnol.* **2009**, *9*, 1741.
5
6
7 [156] L. C. Rodrigues, C. A. Custódio, R. L. Reis, J. F. Mano, *J. Mater. Chem. B* **2016**, *4*,
8
9 1398.
10
11
12 [157] M. F. Bédard, B. G. De Geest, A. G. Skirtach, H. Möhwald, G. B. Sukhorukov, *Adv.*
13
14 *Colloid Interface Sci.* **2010**, *158*, 2.
15
16
17 [158] C. A. Custódio, R. L. Reis, J. F. Mano, *Adv. Healthc. Mater.* **2014**, *3*, 797.
18
19
20 [159] S. G. Caridade, C. Monge, J. Almodóvar, R. Guillot, J. Lavaud, V. Jossierand, J.-L. Coll,
21
22 J. F. Mano, C. Picart, *Acta Biomater.* **2015**, *15*, 139.
23
24
25 [160] N. J. Shah, J. Hong, M. N. Hyder, P. T. Hammond, *Adv. Mater.* **2012**, *24*, 1445.
26
27
28 [161] I. Drachuk, M. K. Gupta, V. V. Tsukruk, *Adv. Funct. Mater.* **2013**, *23*, 4437.
29
30
31 [162] R. F. Fakhrullin, Y. M. Lvov, *ACS Nano* **2012**, *6*, 4557.
32
33
34 [163] G. Bantchev, Z. Lu, Y. Lvov, *J. Nanosci. Nanotechnol.* **2009**, *9*, 396.
35
36
37 [164] A. Kang, J. Park, J. Ju, G. S. Jeong, S.-H. Lee, *Biomaterials* **2014**, *35*, 2651.
38
39
40 [165] C. J. Martinez, J. W. Kim, C. Ye, I. Ortiz, A. C. Rowat, M. Marquez, D. Weitz,
41
42 *Macromol. Biosci.* **2012**, *12*, 946.
43
44
45 [166] G. Orive, R. M. Hernández, A. R. Gascón, R. Calafiore, T. M. S. Chang, P. De Vos, G.
46
47 Hortelano, D. Hunkeler, I. Lacić, A. M. J. Shapiro, et al., *Nat. Med.* **2003**, *9*, 104.
48
49
50 [167] D. Velasco, E. Tumarkin, E. Kumacheva, *Small* **2012**, *8*, 1633.
51
52
53 [168] W.-H. Tan, S. Takeuchi, *Adv. Mater.* **2007**, *19*, 2696.
54
55
56
57
58
59
60

- 1 [169] H. Zhang, E. Tumarkin, R. M. A. Sullan, G. C. Walker, E. Kumacheva, *Macromol. Rapid*
2
3 *Commun.* **2007**, 28, 527.
4
5
- 6 [170] M. Lian, C. P. Collier, M. J. Doktycz, S. T. Retterer, *Biomicrofluidics* **2012**, 6, 044108.
7
8
- 9 [171] † Hong Zhang, † Ethan Tumarkin, †,‡ Raheem Peerani, † Zhihong Nie, † Ruby May
10
11 A. Sullan, † and Gilbert C. Walker, †,‡,§ Eugenia Kumacheva*, **2006**, DOI
12
13 10.1021/JA0635682.
14
15
- 16 [172] D. E. Discher, D. J. Mooney, P. W. Zandstra, *Science* **2009**, 324, 1673.
17
18
- 19 [173] B. Geiger, A. Bershadsky, *Cell* **2002**, 110, 139.
20
21
- 22 [174] R. O. Hynes, *Science* **2009**, 326, 1216.
23
24
- 25 [175] C.-H. Choi, J.-H. Jung, Y. W. Rhee, D.-P. Kim, S.-E. Shim, C.-S. Lee, *Biomed.*
26
27 *Microdevices* **2007**, 9, 855.
28
29
- 30 [176] S. Utech, R. Prodanovic, A. S. Mao, R. Ostafe, D. J. Mooney, D. A. Weitz, *Adv. Healthc.*
31
32 *Mater.* **2015**, 4, 1628.
33
34
- 35 [177] H. Onoe, T. Okitsu, A. Itou, M. Kato-Negishi, R. Gojo, D. Kiriya, K. Sato, S. Miura, S.
36
37 Iwanaga, K. Kuribayashi-Shigetomi, et al., *Nat. Mater.* **2013**, 12, 584.
38
39
- 40 [178] S. Sugiura, T. Oda, Y. Aoyagi, M. Satake, N. Ohkohchi, M. Nakajima, *Lab Chip* **2008**,
41
42 8, 1255.
43
44
- 45 [179] E. Kang, G. S. Jeong, Y. Y. Choi, K. H. Lee, A. Khademhosseini, S.-H. Lee, *Nat. Mater.*
46
47 **2011**, 10, 877.
48
49
- 50 [180] K. H. Lee, S. J. Shin, Y. Park, S.-H. Lee, *Small* **2009**, 5, 1264.
51
52
53
54
55
56
57
58
59
60
61
62
63
64
65

- 1 [181] † Su-Jung Shin, † Ji-Young Park, Jin-Young Lee, Ho Park, Yong-Doo Park, Kyu-
2 Back Lee, and Chang-Mo Whang, S.-H. Lee*, **2007**, DOI 10.1021/LA700818Q.
3
4
5
6 [182] M. Yamada, S. Sugaya, Y. Naganuma, M. Seki, *Soft Matter* **2012**, 8, 3122.
7
8
9 [183] M. Hu, R. Deng, K. M. Schumacher, M. Kurisawa, H. Ye, K. Purnamawati, J. Y. Ying,
10 *Biomaterials* **2010**, 31, 863.
11
12
13
14
15 [184] N. A. Raof, M. R. Padgen, A. R. Gracias, M. Bergkvist, Y. Xie, *Biomaterials* **2011**, 32,
16 4498.
17
18
19
20
21 [185] S. Zhang, M. A. Greenfield, A. Mata, L. C. Palmer, R. Bitton, J. R. Mantei, C. Aparicio,
22 M. O. de la Cruz, S. I. Stupp, *Nat. Mater.* **2010**, 9, 594.
23
24
25
26
27 [186] D. Puppi, D. Dinucci, C. Bartoli, C. Mota, C. Migone, F. Dini, G. Barsotti, F. Carlucci,
28 F. Chiellini, *J. Bioact. Compat. Polym.* **2011**, 26, 478.
29
30
31
32 [187] A. C. Lima, J. F. Mano, *Nanomedicine* **2015**, 10, 103.
33
34
35
36 [188] † Taolei Sun, ‡ Lin Feng, § and Xuefeng Gao, †,§ Lei Jiang*, **2005**, DOI
37 10.1021/AR040224C.
38
39
40
41
42 [189] M. J. Hancock, K. Sekeroglu, M. C. Demirel, *Adv. Funct. Mater.* **2012**, 22, 2223.
43
44
45 [190] K. Koch, W. Barthlott, *Philos. Trans. R. Soc. A Math. Phys. Eng. Sci.* **2009**, 367, 1487.
46
47
48 [191] W. Song, A. C. Lima, J. F. Mano, *Soft Matter* **2010**, 6, 5868.
49
50
51
52 [192] E. Ueda, P. A. Levkin, *Adv. Mater.* **2013**, 25, 1234.
53
54
55 [193] Y. Y. Yan, N. Gao, W. Barthlott, *Adv. Colloid Interface Sci.* **2011**, 169, 80.
56
57
58 [194] X. Yao, Y. Song, L. Jiang, *Adv. Mater.* **2011**, 23, 719.
59

- 1 [195] A. C. Lima, J. F. Mano, *Nanomedicine* **2015**, *10*, 271.
2
3
4 [196] A. I. Neto, C. R. Correia, C. A. Custódio, J. F. Mano, *Adv. Funct. Mater.* **2014**, *24*, 5096.
5
6
7 [197] M. B. Oliveira, A. I. Neto, C. R. Correia, M. I. Rial-Hermida, C. Alvarez-Lorenzo, J. F.
8
9 Mano, *ACS Appl. Mater. Interfaces* **2014**, *6*, 9488.
10
11
12 [198] A. C. Lima, P. Batista, T. A. M. Valente, A. S. Silva, I. J. Correia, J. F. Mano, *Tissue*
13
14 *Eng. Part A* **2013**, *19*, 1175.
15
16
17 [199] A. C. Lima, C. R. Correia, M. B. Oliveira, J. F. Mano, *J. Bioact. Compat. Polym.* **2014**,
18
19
20
21
22 29, 50.
23
24 [200] A. C. Lima, C. A. Custódio, C. Alvarez-Lorenzo, J. F. Mano, *Small* **2013**, *9*, 2487.
25
26
27 [201] A. M. S. Costa, M. Alatorre-Meda, C. Alvarez-Lorenzo, J. F. Mano, *Small* **2015**, *11*,
28
29
30
31 3648.
32
33 [202] A. I. Neto, K. Demir, A. A. Popova, M. B. Oliveira, J. F. Mano, P. A. Levkin, *Adv.*
34
35
36
37 *Mater.* **2016**, *28*, 7613.
38
39 [203] T. Billiet, M. Vandenhaute, J. Schelfhout, S. Van Vlierberghe, P. Dubruel, *Biomaterials*
40
41
42
43
44 **2012**, *33*, 6020.
45 [204] S. M. Peltola, F. P. W. Melchels, D. W. Grijpma, M. Kellomäki, *Ann. Med.* **2008**, *40*,
46
47
48
49 268.
50 [205] E. Sachlos, J. T. Czernuszka, *Eur. Cell. Mater.* **2003**, *5*, 29.
51
52
53 [206] R. C. Dutta, A. K. Dutta, *Biotechnol. Adv.* **2009**, *27*, 334.
54
55
56 [207] D. W. Hutmacher, S. Cool, *J. Cell. Mol. Med.* **2007**, *11*, 654.
57
58
59
60
61
62
63
64
65

- 1 [208] V. Mironov, T. Boland, T. Trusk, G. Forgacs, R. R. Markwald, *Trends Biotechnol.* **2003**,
2
3 21, 157.
4
5
6 [209] P. Calvert, *Mater. Sci.* **2007**, 318, 208.
7
8
9 [210] S. R. Pajoum Shariati, S. Moeinzadeh, E. Jabbari, Springer, Cham, **2015**, pp. 89–108.
10
11
12 [211] R. Levato, J. Visser, J. A. Planell, E. Engel, J. Malda, M. A. Mateos-Timoneda,
13
14 *Biofabrication* **2014**, 6, 035020.
15
16
17 [212] M. de Ruijter, A. Ribeiro, I. Dokter, M. Castilho, J. Malda, *Adv. Healthc. Mater.* **2019**,
18
19 8, DOI 10.1002/adhm.201800418.
20
21
22 [213] A. N. Leberfinger, S. Dinda, Y. Wu, S. V. Koduru, V. Ozbolat, D. J. Ravnice, I. T.
23
24 Ozbolat, *Acta Biomater.* **2019**, 95, 32.
25
26
27 [214] J. Malda, J. Visser, F. P. Melchels, T. Jüngst, W. E. Hennink, W. J. A. Dhert, J. Groll,
28
29 D. W. Hutmacher, *Adv. Mater.* **2013**, 25, 5011.
30
31
32 [215] K. Jakab, C. Norotte, F. Marga, K. Murphy, G. Vunjak-Novakovic, G. Forgacs,
33
34 *Biofabrication* **2010**, 2, 022001.
35
36
37 [216] V. Mironov, G. Prestwich, G. Forgacs, *J. Mater. Chem.* **2007**, 17, 2054.
38
39
40 [217] V. Lee, G. Singh, J. P. Trasatti, C. Bjornsson, X. Xu, T. N. Tran, S.-S. Yoo, G. Dai, P.
41
42 Karande, *Tissue Eng. Part C Methods* **2014**, 20, 473.
43
44
45 [218] S.-J. Song, J. Choi, Y.-D. Park, J.-J. Lee, S. Y. Hong, K. Sun, *Artif. Organs* **2010**, 34,
46
47 1044.
48
49
50 [219] C. R. Almeida, T. Serra, M. I. Oliveira, J. A. Planell, M. A. Barbosa, M. Navarro, *Acta*
51
52 *Biomater.* **2014**, 10, 613.
53
54
55
56
57
58
59
60
61
62
63
64
65

- 1 [220] M. Du, B. Chen, Q. Meng, S. Liu, X. Zheng, C. Zhang, H. Wang, H. Li, N. Wang, J. Dai,
2
3 *Biofabrication* **2015**, 7, 044104.
4
5
- 6 [221] K. Markstedt, A. Mantas, I. Tournier, H. Martínez Ávila, D. Hägg, P. Gatenholm,
7
8 *Biomacromolecules* **2015**, 16, 1489.
9
- 10 [222] L. E. Bertassoni, J. C. Cardoso, V. Manoharan, A. L. Cristino, N. S. Bhise, W. A. Araujo,
11
12 P. Zorlutuna, N. E. Vrana, A. M. Ghaemmaghami, M. R. Dokmeci, et al., *Biofabrication*
13
14 **2014**, 6, 024105.
15
16
17
18
19
- 20 [223] D. B. Kolesky, R. L. Truby, A. S. Gladman, T. A. Busbee, K. A. Homan, J. A. Lewis,
21
22 *Adv. Mater.* **2014**, 26, 3124.
23
24
25
- 26 [224] L. Pescosolido, W. Schuurman, J. Malda, P. Matricardi, F. Alhaique, T. Coviello, P. R.
27
28 van Weeren, W. J. A. Dhert, W. E. Hennink, T. Vermonden, *Biomacromolecules* **2011**,
29
30 *12*, 1831.
31
32
33
- 34 [225] L. Li, J. M. Scheiger, P. A. Levkin, *Adv. Mater.* **2019**, 31, 1807333.
35
36
- 37 [226] H.-J. Kong, K. Y. Lee, D. J. Mooney, *Polymer (Guildf)*. **2002**, 43, 6239.
38
39
- 40 [227] J. H. Y. Chung, Sina Naficy, Zhilian Yue, Robert Kapsa, Anita Quigley, S. E. Moulton,
41
42 G. G. Wallace, *Biomater. Sci.* **2013**, 1, 763.
43
44
45
- 46 [228] P. Lenas, F. P. Luyten, M. Doblare, E. Nicodemou-Lena, A. E. Lanzara, *Artif. Organs*
47
48 **2011**, 35, 656.
49
50
51
- 52 [229] A. J. Engler, P. O. Humbert, B. Wehrle-Haller, V. M. Weaver, *Science* **2009**, 324, 208.
53
54
- 55 [230] D. E. Ingber, *Proc. Natl. Acad. Sci. U. S. A.* **2005**, 102, 11571.
56
57
58
- 59 [231] N. G. Rim, A. Yih, P. Hsi, Y. Wang, Y. Zhang, J. Y. Wong, *Biomaterials* **2018**, 181,
60
61
62
63
64
65

126.

- 1
2
3
4 [232] H. Lin, Q. Li, Y. Lei, *Mater. Eng. Sci. Div. 2017 - Core Program. Area 2017 AIChE*
5
6 *Annu. Meet.* **2017**, 2, 863.
7
8
9
10 [233] P. J. Hernández RM, Orive G, Murua A, *Adv Drug Deliv Rev* **2010**, 62, 711.
11
12
13 [234] F. Li, V. X. Truong, P. Fisch, C. Levinson, V. Glattauer, M. Zenobi-Wong, H. Thissen,
14
15 J. S. Forsythe, J. E. Frith, *Acta Biomater.* **2018**, 77, 48.
16
17
18 [235] P. Sher, C. R. Correia, R. R. Costa, J. F. Mano, *RSC Adv.* **2015**, 5, 2511.
19
20
21
22 [236] Q. Li, Y.-W. Zhang, C.-F. Wang, D. A. Weitz, S. Chen, *Adv. Mater.* **2018**, 30, 1803475.
23
24
25 [237] A. S. Mao, J.-W. Shin, S. Utech, H. Wang, O. Uzun, W. Li, M. Cooper, Y. Hu, L. Zhang,
26
27 D. A. Weitz, et al., *Nat. Mater.* **2017**, 16, 236.
28
29
30
31 [238] Y. Dzenis, *Science (80-.)*. **2004**, 304, 1917.
32
33
34 [239] P. W. K. Rothmund, *Nature* **2006**, 440, 297.
35
36
37
38 [240] A. P. Mouritz, M. K. Bannister, P. J. Falzon, K. H. Leong, *Compos. Part A Appl. Sci.*
39
40 *Manuf.* **1999**, 30, 1445.
41
42
43 [241] Y. Li, C. T. Poon, M. Li, T. J. Lu, B. Pingguan-Murphy, F. Xu, *Adv. Funct. Mater.* **2015**,
44
45 25, 5999.
46
47
48
49 [242] Y. Zuo, X. He, Y. Yang, D. Wei, J. Sun, M. Zhong, R. Xie, H. Fan, X. Zhang, *Acta*
50
51 *Biomater.* **2016**, 38, 153.
52
53
54
55 [243] A. R. Sousa, C. Martins-Cruz, M. B. Oliveira, J. F. Mano, *Adv. Mater.* **2020**.
56
57
58
59 [244] M. F. Leong, J. K. C. Toh, C. Du, K. Narayanan, H. F. Lu, T. C. Lim, A. C. A. Wan, J.

1 Y. Ying, *Nat. Commun.* **2013**, *4*, 1.

2
3
4 [245] Y. Du, M. Ghodousi, H. Qi, N. Haas, W. Xiao, A. Khademhosseini, *Biotechnol. Bioeng.*
5
6 **2011**, *108*, 1693.

7
8
9 [246] H. Naito, *Circulation* **2006**, *114*, I.

10
11
12 [247] W.-H. Zimmermann, I. Melnychenko, G. Wasmeier, M. Didié, H. Naito, U. Nixdorff, A.
13
14 Hess, L. Budinsky, K. Brune, B. Michaelis, et al., *Nat. Med.* **2006**, *12*, 452.

15
16
17 [248] Z. Liu, M. Takeuchi, M. Nakajima, C. Hu, Y. Hasegawa, Q. Huang, T. Fukuda, *Acta*
18
19 *Biomater.* **2017**, *50*, 178.

20
21
22 [249] H. Qi, M. Ghodousi, Y. Du, C. Grun, H. Bae, P. Yin, A. Khademhosseini, *Nat. Commun.*
23
24 **2013**, *4*, 2275.

25
26
27 [250] P. Sher, S. M. Oliveira, J. Borges, J. F. Mano, *Biofabrication* **2015**, *7*, 011001.

28
29
30 [251] C. R. Correia, P. Sher, R. L. Reis, J. F. Mano, *Soft Matter* **2013**, *9*, 2125.

31
32
33 [252] C. R. Correia, R. L. Reis, J. F. Mano, *Biomacromolecules* **2013**, *14*, 743.

34
35
36 [253] C. R. Correia, S. Gil, R. L. Reis, J. F. Mano, *Adv. Healthc. Mater.* **2016**, *5*, 1346.

37
38
39 [254] J.-S. Lee, J. M. Hong, J. W. Jung, J.-H. Shim, J.-H. Oh, D.-W. Cho, *Biofabrication* **2014**,
40
41 *6*, 024103.

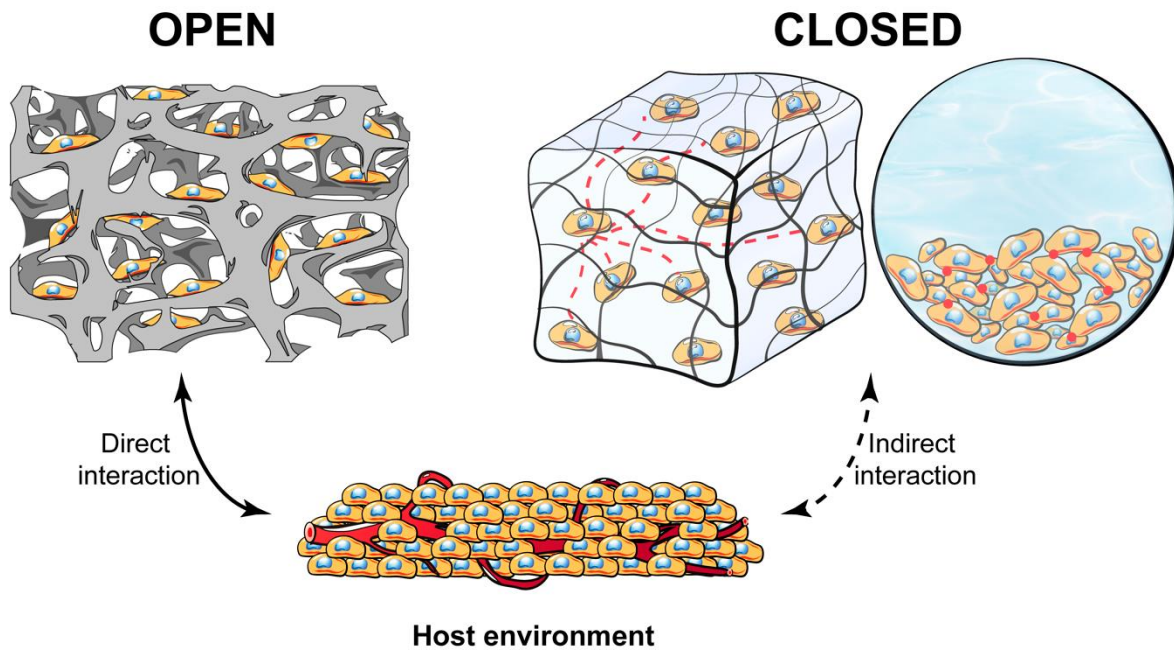
42
43
44 [255] F. Guillemot, A. Souquet, S. Catros, B. Guillotin, J. Lopez, M. Faucon, B. Pippenger, R.
45
46 Bareille, M. Rémy, S. Bellance, et al., *Acta Biomater.* **2010**, *6*, 2494.

47
48
49 [256] B. Guillotin, A. Souquet, S. Catros, M. Duocastella, B. Pippenger, S. Bellance, R.
50
51 Bareille, M. Rémy, L. Bordenave, J. Amédée, et al., *Biomaterials* **2010**, *31*, 7250.

1 [257] X. Cui, T. Boland, *Biomaterials* **2009**, *30*, 6221.

2
3
4 [258] A. Skardal, J. Zhang, L. McCoard, X. Xu, S. Oottamasathien, G. D. Prestwich, *Tissue*
5
6 *Eng. Part A* **2010**, *16*, 2675.

7
8
9
10 [259] A. Skardal, J. Zhang, G. D. Prestwich, *Biomaterials* **2010**, *31*, 6173.



46
47 **Figure 1.** Interaction of tissue engineered scaffolds with the host environment. In “open”
48 scaffolds, there is a direct interaction between the cells seeded at the surface of the biomaterial
49 and the external environment. In “closed” systems the encapsulation matrix mediates this
50 interaction, and provides an indirect interaction with the external environment. If the matrix is
51 liquefied, cell-cell interactions are maximized due to the direct contact between the
52
53
54
55
56
57
58
59
60
61
62
63
64
65

encapsulated cells (red dots), contrary to the indirect cell signaling in crosslinked matrices (dotted red lines).

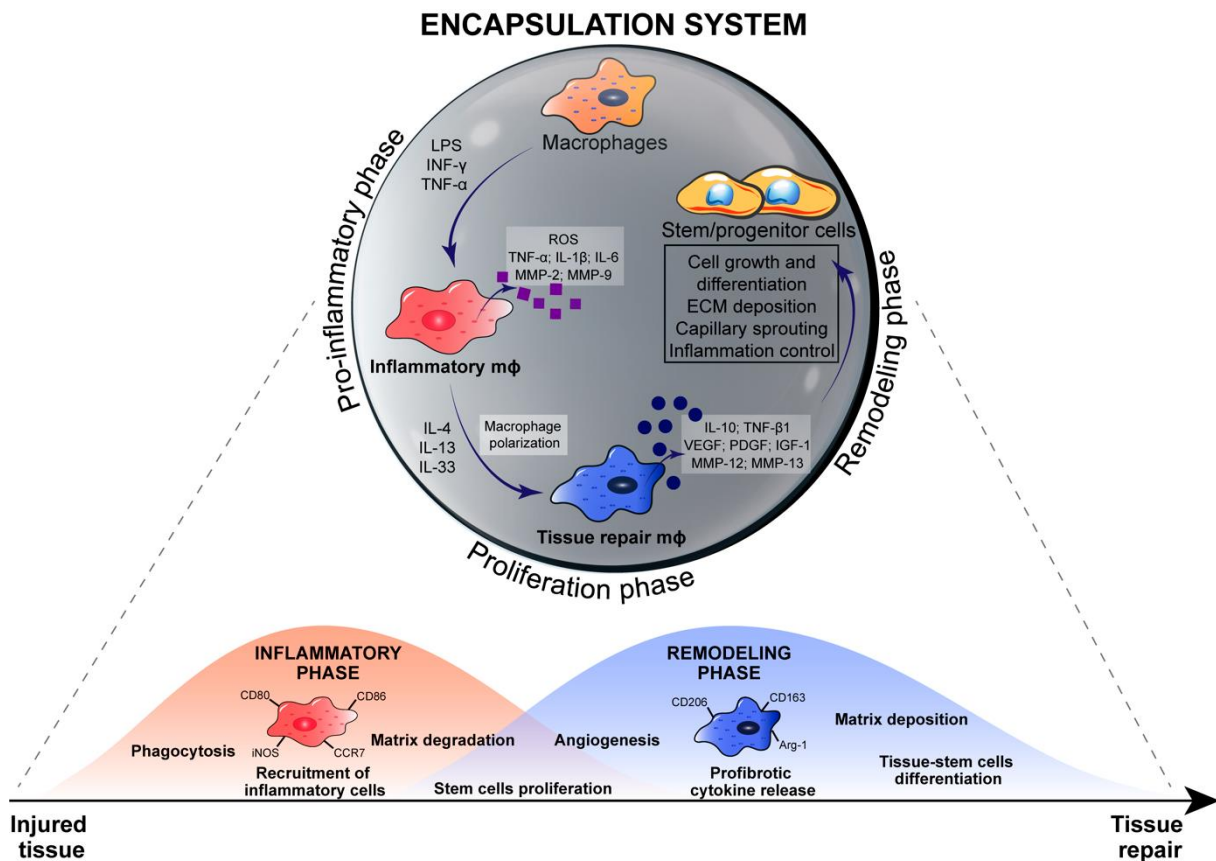


Figure 2. Upon the implantation of a biomaterial into a lesion site, a cascade of immune-related processes towards tissue repair occurs. As example, a stem-cell encapsulation system is represented.

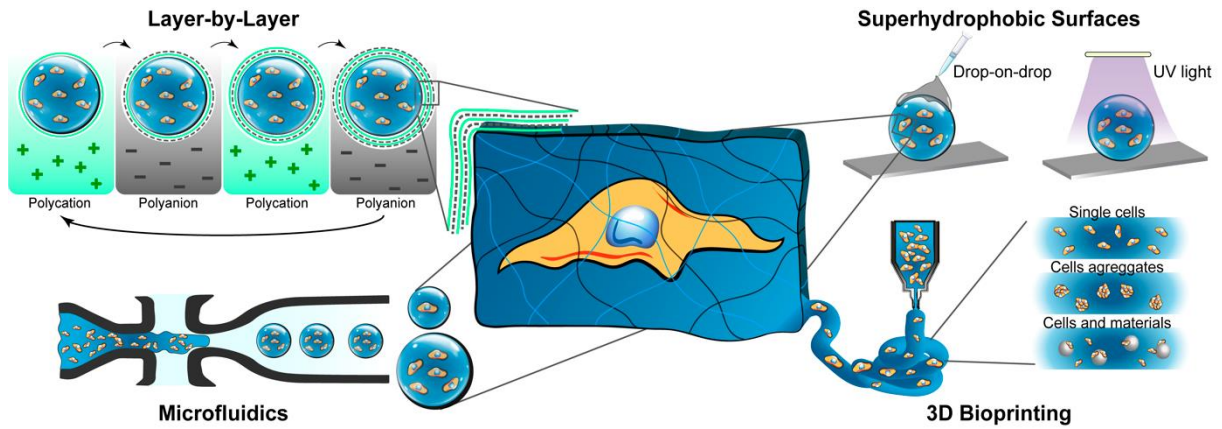
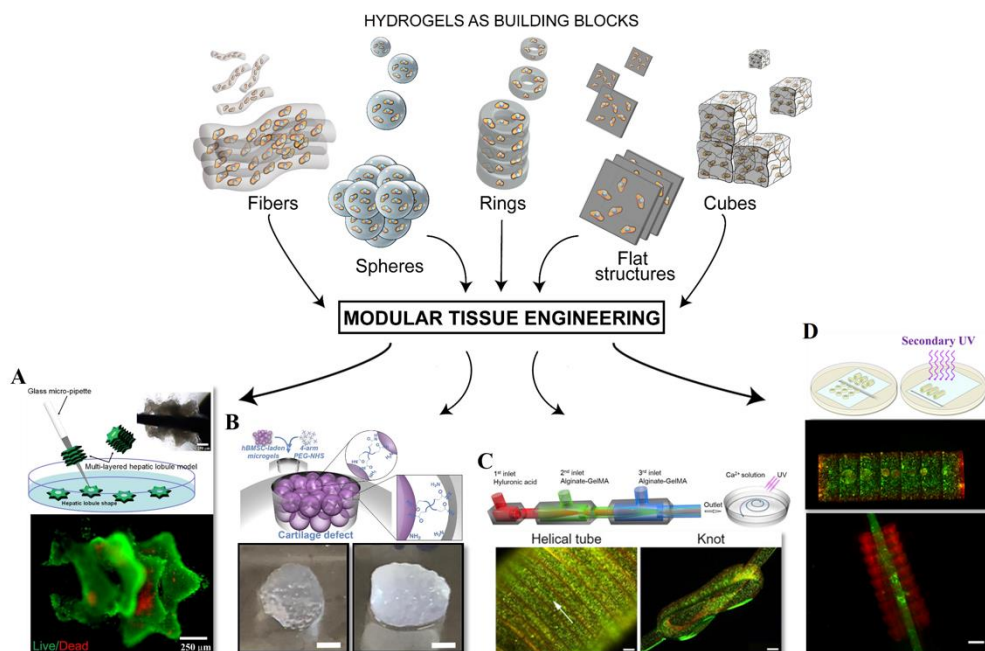


Figure 3. Layer-by-layer, microfluidics, superhydrophobic surfaces, and 3D bioprinting as novel or unconventional technologies to produce innovative cell encapsulation systems.



1 **Figure 4.** Multiple-shaped hydrogels, including fibers, spheres, rings, flat structures and cubes,
2
3 as cell encapsulation building blocks for modular tissue engineering. **(A-D)** Such hydrogels can
4
5 be further assembled by modular tissue engineering to produce clinically-relevant 3D structures
6
7 with close-to-native heterogeneity. **(A)** Flatted poly(L-lysine)/alginate spherical microgels
8
9 encapsulating rat liver cells. Flatted structures were assembled into 4-layers to construct 3D
10
11 hepatic lobule-like tissues. Reproduced with permission.^[248] Copyright 2017, Elsevier. Living
12
13 cells are stained in green and dead cells in red by LiveDead fluorescence assay. Scale bar is 250
14
15 μm . **(B)** Spherical microgels of 4-arm poly(ethylene glycol)-N-hydroxysuccinimide
16
17 encapsulating mesenchymal stem cells. After crosslinking, microgels spontaneously adhered,
18
19 and originated a 3D model for cartilage repair. Scale bars are 1 mm. Reproduced with
20
21 permission.^[234] Copyright 2019, Elsevier. **(C)** Double-layer hollow microfibers production
22
23 after calcium-alginate reaction and UV exposure for gelMA crosslinking. Fluorescence images
24
25 (microballoon) of the produced woven structures, namely in helical tube and knot
26
27 conformations. Scale bars are 200 μm . Reproduced with permission.^[242] Copyright 2016,
28
29 Elsevier. **(D)** Chitin- and alginate-based microfibers assembled by polyelectrolyte
30
31 complexation to create pre-vascularized adipose and hepatic tissues. Central fibers
32
33 encapsulating endothelial cells were surrounded by parenchymal cell-laden fibers. The tertiary
34
35 structure construct was obtained by spooling and layering the repeat unit (secondary structure).
36
37 Reproduced with permission.^[245] Copyright 2011, John Wiley and Sons. Scale bar is 500 μm .
38
39
40
41
42
43
44
45
46
47
48
49
50

51 **Table 1.** Examples of cell encapsulation systems for Tissue Engineering and Regenerative
52
53 Medicine (TERM) using the layer-by-layer technique. The examples cover the type of
54
55 polyelectrolytes used to produce the different cell encapsulation systems, the biomaterials
56
57
58
59
60
61
62
63
64
65

employed to produce the cell encapsulation matrix, the type of encapsulated cells, and the TERM application.

MEMBRANE COMPOSITION	ENCAPSULATION MATRIX	GEOMETRY	ENCAPSULATED CELLS	TERM APPLICATION	REF.
Alginate and chitosan	Alginate	Fibers	Human kidney 293	Vascular	[178]
					[250]
Spheres		L929	General	[147,154,251]	
				[252]	
		hASCs and hAMECs	Bone	[149,150]	
Alginate, chitosan and PLL			hASCs	Cartilage	[253]
and MNPs					

CaCl₂ – Calcium chloride; PLL – poly(L-lysine); hASCs – Human adipose-derived stem cells; hAMECs – Human adipose-derived microvascular endothelial cells; MNPs – Magnetic nanoparticles.

Table 2. Examples of cell encapsulation systems for Tissue Engineering and Regenerative Medicine (TERM) using microfluidics. The examples cover the type of chip used to produce the different cell encapsulation systems, the biomaterials employed to produce the cell encapsulation matrix, the type of encapsulated cells, and the TERM application.

TYPE OF CHIP	ENCAPSULATION MATRIX	CONTINUOUS PHASE	GEOMETRY	ENCAPSULATED CELLS	TERM APPLICATION	REF.
T-junction with a downstream entrance	Alginate containing CaCO ₃ nanoparticles in RPMI medium	Corn oil and lecithin Acetic acid (downstream entrance)	Spheres	Jurkat cells	General	[168]
Flow-focusing	Alginate and calcium-EDTA complex	Fluorinated carbon oil (HFE7500) and a biocompatible		MSCs		[176]

		surfactant (PFPE/PEG)					
Spinning chip (multiple inlets)	Alginate and chitosan	CaCl ₂	Fibers	Primary rat hepatocytes and L929 fibroblasts (alone or co-cultured)	[179]		
Capillaries	Alginate			Human fibroblasts	Nerve or muscular	[181]	
Y-junction (3 inlets)				Mouse ESCs	General	[184]	
Micro nozzle array				HIVE-78	Vascular	[180]	
Axisymmetric with uniform depth				Alginate with or without propylene glycol alginate		HEK 293	[178]
Three-phase coaxial flow (5 inlets)	Gtn-HPA/HRP			Gtn-HPA diluted in H ₂ O ₂ (middle) PBS (outer)	MDCKs	Hepatic, vascular and musculoskeletal	[182]
				PBS and H ₂ O ₂ (inner)	HPTCs		
		Polysulfone dissolved in NMP (outer)					

CaCl₂ – Calcium chloride; CaCO₃ – calcium carbonate; DI – deionized water; EDTA – ethylenediaminetetraacetic acid; ESCs – embryonic stem cells; Gtn-HPA – gelatin-hydroxyphenylpropionic acid; HeLa – cell line isolated from human cervix epitheloid carcinoma; HEK 293 – human kidney (embryonic) cell line; HIVE-78 – vascular cell line; HPTCs – human proximal tubule cells; L929 – fibroblast cell line isolated from mouse adipose tissue; MDCKs – Madin-Darby canine kidney cells; MSCs – mesenchymal stem cells; NIH 3T3 – embryonic fibroblast cell line; NMP – polysulfone and N-methyl-2-pyrrolidone; PBS –

phosphate buffer saline; PFPE – oligomeric perfluorinated polyethers; PEG – poly(ethylene glycol); polyNIPAAm – Poly-(N-isopropyl acrylamide); HRP – horseradish peroxidase; H₂O₂ – hydrogen peroxide.

Table 3 – Examples of cell encapsulation systems for Tissue Engineering and Regenerative Medicine (TERM) using superhydrophobic surfaces. The examples cover the substrate used to produce the superhydrophobic, the biomaterials employed to produce the cell encapsulation matrix, the geometry, the type of encapsulated cells, and the TERM application.

SUBSTRATE/TREATMENT	ENCAPSULATION MATRIX	GEOMETRY	ENCAPSULATED CELLS	TERM APPLICATION	REF.
Polystyrene/THF, ethanol, Ar-plasma and PFDTS	Alginate and fibronectin	Spheres	rMSCs	Bone	[198]
	Alginate		L929	General	[191]
	Chitosan/ β -glycerophosphate				[199]
Copper/NH ₄ OH, PFDTS and ethanol	Alginate and DEX-MA	Spheres	L929	General	[200]
Glass/soot coated with paraffin candle, TEOS, and ammonia; calcination, and silane by CVD					[201]
Glass/HEMA-EDMA photographed with PFPMA (quartz chromium photomask)	MI-PVA	Crosssections with variable geometries: circles, triangles, hexagons, and squares	HeLa		[202]

Ar – argon; CVD – chemical vapor deposition; DEX-MA – Methacrylated dextran; HeLa – cell line isolated from human cervix epitheloid carcinoma; HEMA-EDMA – poly(hydroxyethyl methacrylate-co-ethylene dimethacrylate); L929 – fibroblast cell line isolated from mouse

adipose tissue; NH₄OH – ammonium hydroxide; MI-PVA – maleimide-functionalized polyvinyl alcohol; PFDTs – 1H,1H,2H,2H-perfluorodecyltrimethoxysilane; PFPMA – 2,2,3,3,3-pentafluoropropyl methacrylate; rMSCs – mesenchymal stem cells isolated from the bone marrow of Wistar rats; THF – tetrahydrofuran.

Table 4 – Examples of cell encapsulation systems for Tissue Engineering and Regenerative Medicine (TERM) using bioprinting. The examples cover the type of biomaterials employed to produce the cell encapsulation matrix, the geometry, the type of encapsulated cells, and the TERM application.

TECHNOLOGY	ENCAPSULATION MATRIX	GEOMETRY	ENCAPSULATED CELLS	TERM APPLICATION	REF.
Electromagnetic jet	Alginate and nanofibrillated cellulose	Human ear and sheep meniscus	hNCs	Cartilage	[221]
Additive manufacturing	Alginate	Human ear	ASCs		[254]
	HA/Dex-HEMA	Free form	Chondrocytes	[224]	
	Collagen type I		HFF-1 and HaCaT	[217]	
LIFT	Alginate	Free form	Eahy926	General	[255]
	Alginate/glycerol		Eahy926		[256]
	Matrigel™ Fibrin			Myoblasts	Myocard
Microextrusion	Alginate/gelatin	Free form	BMMSCs	Bone	[220]
	Methacrylamide gelatin/CBD-BMP-2				
	gelMA		HNDFs and C3H/10T1/2	Vascular	[223]
Inkjet printing	Fibrin	Hollow fibers	hMVEC		[257]
	HA-MA;GE-MA/PEGDA		HepG2/C3A	[258]	
Microcapillary	CMHA-S:Gtn-	Hollow fibers	Int-407	Vascular	[259]
	DTPH/TetraPac8/TetraPac13		NIH 3T3		

	gelMA	HepG2	[222]
1			
2	ASCs – adipose-derived stem cells; BMMSCs – bone marrow mesenchymal stem cells; CBD-		
3			
4	BMP-2 – collagen binding domain-bone morphogenic protein 2; CMHA – thiolated hyaluronic		
5	acid derivative; C3H/10T1/2 – cloned murine embryo fibroblast cell line; Dex-HEMA –		
6			
7	hydroxyethyl-methacrylate-derivatized dextran; Eahy926 – Endothelial cell line; GE-MA -		
8			
9	methacrylated ethanolamide derivative of gelatin; gelMA – methacrylated gelatin; Gtn-		
10			
11	DTPH – thiolated gelatin (available as Gelin-S, Glycosan Byosistemas); HA – hyaluronic acid;		
12			
13	HaCaT – human keratinocytes cell line; HA-MA – methacrylated hyaluronic acid; hECs –		
14			
15	human endothelial cells; HepG2 – human liver cells isolated from a hepatocellular carcinoma;		
16			
17	HepG2/C3A – clonal derivative of HepG2; HFF-1 – fibroblasts; hMVEC – human		
18			
19	microvascular endothelial cells; Int-407 – HeLa derivative cell line isolated from human cervix;		
20			
21	hNCs – human nanoseptal chondrocytes; HNDFs – human neonatal dermal fibroblasts; dermal		
22			
23	IR – infrared; LIFT – Laser Induced Forward Transfer; NIH 3T3 – embryonic fibroblast cell		
24			
25	line; PEG – poly(ethylene glycol); PEGDA - poly(ethylene glycol) diacrylate; TetraPac8 -		
26			
27	tetra-acrylate derivatives of four armed PEG 2000 chains; TetraPac13 - tetra-acrylate		
28			
29	derivatives of four armed PEG 3400 chains.		
30			
31			
32			
33			
34			
35			
36			
37			
38			
39			
40			
41			
42			
43			
44			
45			
46			
47			
48			
49			
50			
51			
52			
53			
54			
55			
56			
57			
58			
59			
60			
61			
62			
63			
64			
65			



1
2
3
4
5
6
7
8
9
10
11
12
13 **Clara R. Correia** is a Biomedical Engineer holding a Ph.D in Tissue Engineering,
14 Regenerative Medicine and Stem Cells (TERM&SC) from University of Minho (Braga,
15 Portugal). Currently, she is a Junior Researcher at COMPASS Research Group from CICECO
16 – Aveiro Institute of Materials, and PI of the project *CIRCUS* (PTDC/BTM-MAT/31064/2017)
17 from the Portuguese Foundation for Science and Technology (FCT). Her current research
18 interests are focused in the development of smart, autonomous, and self-regulated cell
19 encapsulation devices aiming a close-to-native engineered regeneration of human tissues.
20
21
22
23
24
25
26
27
28
29
30
31
32
33
34



35
36
37
38
39
40
41
42
43
44
45
46
47
48
49 **Sara Nadine** has a MSc in Molecular Oncology from the Abel Salazar Institute of Biomedical
50 Sciences of University of Porto (Porto, Portugal). Currently, she is a Ph.D student funded by
51 the Portuguese Foundation for Science and Technology (FCT) in Biochemistry at the
52 COMPASS Research Group from CICECO – Aveiro Institute of Materials. Her main research
53
54
55
56
57
58
59
60
61
62
63
64
65

1 activities are focused in the interaction of immune and osteoprogenitor cells to develop dynamic
2
3 cell encapsulation systems for tissue engineering and regenerative medicine.
4
5
6
7
8
9



24 **João F. Mano** is a full professor at the Chemistry Department of University of Aveiro,
25
26 Portugal, and director of the COMPASS Research Group from CICECO – Aveiro Institute of
27
28 Materials. His research interests include the use of biomaterials and cells toward the progress
29
30 of transdisciplinary concepts to be employed in regenerative and personalized medicine. He has
31
32 been applying biomimetic and nano/micro-technology approaches to polymer-based
33
34 biomaterials and surfaces to develop biomedical devices with improved structural and (multi-)
35
36 functional properties, or in the engineering of microenvironments to control cell behavior and
37
38 organization, to be exploited clinically in advanced therapies or in drug screening.
39
40
41
42
43
44
45
46
47
48
49
50
51
52
53
54
55
56
57
58
59
60
61
62
63
64
65



Click here to access/download
Production Data
Figure1.tif



Click here to access/download
Production Data
Figure2.tif



Click here to access/download
Production Data
Figure3.tif



Click here to access/download
Production Data
Figure4.tif



Click here to access/download
Production Data
ToC.tif