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Nanoscale Junctions for Single Molecule Electronics Fabricated using Bilayer Nanoimprint Lithography combined with Feedback Controlled Electromigration

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Abstract

Nanoimprint lithography (NIL) is a fast, simple and high throughput technique that allows fabrication of structures with nanometre precision features at low cost. We present an advanced bilayer nanoimprint lithography approach to fabricate four terminal nanojunction devices for use in single molecule electronic studies. In the first part of this work, we demonstrate a NIL lift-off process using a bilayer resist technique that negates problems associated with metal side-wall tearing during liftoff. In addition to precise nanoscale feature replication, we show that it is possible to imprint micronsized features while still maintaining a bilayer structure enabling an undercut resist structure to be formed. This is accomplished by choosing suitable imprint parameters as well as residual layer etching depth and development time. We then use a feedback controlled electromigration procedure, to produce room-temperature stable nanogap electrodes with sizes below 2 nm. This approach facilitates the integration of molecules in stable, solid-state molecular electronic devices as demonstrated by incorporating benzenethiol as molecular bridges between the electrodes and characterizing its electronics properties through current-voltage measurements. The observation of molecular transport signatures, showing current suppression in the I-V behaviour at low voltage, which is then lifted at high voltage, signifying on- and off-resonant transport through molecular levels as a function of voltage, is confirmed in repeated I-V sweeps. The large conductance, symmetry of the I-V sweep and small value of the voltage minimum in Transition Voltage Spectroscopy indicates the bridging of the two benzenethiol molecules is by π -stacking.

1. Introduction

Since the initial ideas of using molecules to process signals, heralded by the molecular rectifier work of Aviram and Ratner[1] in 1974, scientists have steadily improved the understanding of these complex electrode-molecule systems and have developed better and more robust ways to integrate molecules into devices. In the last decade, molecular electronics has undergone a renaissance as a suite of new achievements has arisen. A single-molecule light emitting diode has been developed[2], single molecule rectification has reached levels of more than 200[3] and electrical control of the nuclear spin of a single molecule in a device has been achieved via the hyperfine Stark effect[4]. These

achievements occurring because molecular electronics provides not only device miniaturization, but a route to the introduction of functional molecules with electronic, optical and mechanical properties tailored through chemical synthesis [5–9]. The future of molecular electronics is also bright. In recent years, traditional semiconductor devices have reached their scaling limit[10], and there is an increased need to find alternate ways to not only make electronic devices[11] but also explore new types of architectures[12,13] that non-traditional materials and methods of fabrication offer. However, many challenges in the field remain, especially at the single molecule electronic level, such as device variability, stability, improved size alignment between the "leads" and the "molecule" and upscaling from single to multiple device architectures to make computer-like systems.

Nanoimprint lithography (NIL) is a versatile nanofabrication technique that allows the replication of patterns with high throughput and resolution. In recent years it has experienced rapid development and uptake as a fabrication technique in both industrial and research environments [14–17]. One particular applicable field for NIL is molecular electronics. This field requires devices with high-resolution sub-micron features and fabrication in large numbers in order to build-up statistically significant results.

In thermal NIL, a substrate is coated with resist which is heated above the glass transition temperature (T_g) and then a stamp with the desired features is pressed into the coated substrate causing the resist to flow across the surface. After cooling, the substrate is mechanically separated leaving an impression of the stamp [18]. It currently has a demonstrated resolution of less than 10 nm [19].

NIL is commonly used as an etch mask for pattern transfer but it does not lend itself well to lift-off as the imprinting process always leaves the patterned resist with sidewall angles greater than 90° as this is necessary to remove the stamp after imprinting. During metal deposition, material will adhere to the sidewalls and form a continuous film over features on the resist, making clean lift-off nearly impossible [20].

An alternative approach often used in lithography processes is the use of a lift-off resist (LOR) layer. This is a bottom resist layer that allows an undercut to be formed once the lithographic process has patterned the topmost layer. This process reduces the occurrence of an incomplete lift-off and tearing of the patterned metal films by making it very unlikely for a continuous metal film to exist between material patterned on the substrate and material deposited on the resist layer. We chose to apply this technique in NIL. Previously, it has been demonstrated that a bilayer resist coating can be imprinted without disturbing the bilayer structure for pillars [21]. However, this was demonstrated only for small pillars and lines with sizes between 20 nm and 600 nm. Another technique exploiting the differing plasma etch resistance of polystyrene-polydimethylsiloxane (PS-PDMS) and polymethyl methacrylate (PMMA) has been shown to produce undercut resist structures with NIL [22]. However, the line quality of features produced using this technique may suffer due to the irregular nature of the top PS-PDMS layer after extended plasma exposure.

In addition to having high-quality lift-off, there is a need as well to be able to imprint nanometre as well as micrometre scale features with NIL. This is especially relevant in the case of electronic devices where nanosized features must be present alongside larger sized pads that are needed to make electrical connections with wire-bonding or a probe-station. Using standard NIL methods, this can be very difficult since large amounts of lateral resist flow occurs from imprinting the larger features, which can degrade or even make impossible the imprinting of the smaller nanoscale features. Using the above bi-layer approach, we explore this problem and show how to resolve this issue by choosing suitable imprint parameters and residual layer etching.

Lastly, we demonstrate the success of the above technique in the fabrication of devices for molecular electronics. Molecular electronics needs reliable methods to fabricate gaps between electrodes, of order 2 nm, and methods to integrate, in a reliable manner, single molecules that bridge between source and drain contacts. Electromigration is one such method to make very small gaps between electrodes [9,23–25]. Using our double layer NIL method we demonstrate the suitable fabrication of devices for electromigration and form molecular devices by incorporating benzenethiol as molecular bridges between the electrodes.

2. Method

The Master stamp used in this work was fabricated by standard electron beam lithography (EBL) and reactive ion etching (RIE) on a 100 mm diameter Si wafer and was cleaned prior to use with sonication in acetone, 2-propanol and water before being dried in nitrogen and baked prior to spin coating of the e-beam resist (AR-N 7520, Allresist). After the patterning process the stamp was coated with an antistick coating, perfluorodecyltrichlorosilane (FDTS), that aids stamp separation after imprinting. The Si wafer, shown in Figure 1 (a), was patterned in a square-like grid with 10 x 12 mm separating each patterned area. Each patterned area contains a group of 15 imprintable devices arranged in a horizontal line, as shown in Figure 1 (b). The design of a single device consists of a cross with 4 arms that are used for 4-terminal electrical connections. In the centre of the cross the four arms meet and form a single, nanoscale, vertical line with dimensions of 100 x 200 nm, as shown in Figure 1 (c), and having two arms connected to either side. The nanoscale line is the main part of the device and is used to make the conducting channel for the electromigration process. The electromigration process occurs after the NIL process and is used to make the nanoscale junction, which consists of a very small airgap (<2 nm) between two metal electrodes.

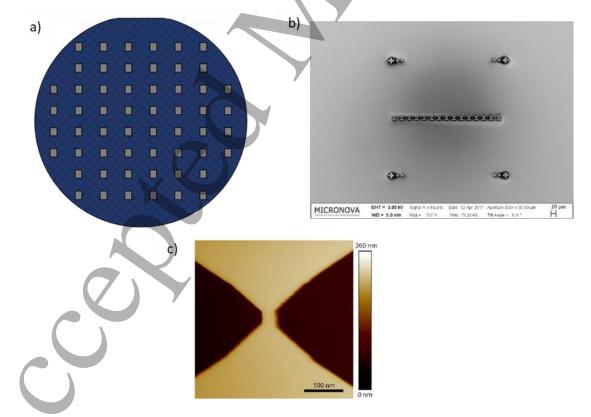


Figure 1. a) Stamp design consisting of 56 groups of 15 nanojunction (crosses) on a 100 mm diameter Si wafer fabricated using EBL and RIE. b) SEM image of one group of 15 nanojunctions showing marks for

photolithography alignment. c) AFM height map of the stamp and showing the channel region (yellow) that is at the centre of a single cross and used to make a nanojunction by electromigration.

The NIL method used for copying the master pattern is shown in Figure 2 and consists of the following steps. Firstly, the bilayer resist was deposited onto a cleaned Si wafer with 1 µm of thermally grown oxide. The bilayer consisted of polymethylglutarimide (PMGI) SF5 resist (MicroChem Corp), spin-coated to give a thickness of 150 nm and baked at 190°C. This was followed by a layer of 15k PMMA that was dissolved in chlorobenzene (8% wt.), spin coated to a thickness of 200 nm and baked at 160°C. Imprinting was carried out by applying 2 bar pressure (used also for all test studies described in this work) once the set-point temperature was reached. Once the desired imprinting time was reached, the stamp was cooled whilst still maintaining pressure until the temperature had reached 90°C. The stamp is separated by inserting a scalpel blade between the stamp and wafer and the residual PMMA is removed using low power oxygen plasma.

Spin coated films and residual layer thicknesses were characterised by AFM (Bruker Dimension Edge) and/or surface profilometry (Bruker DektakXT). A CNI v2.1 machine (NILT) was used to perform the imprinting and the process of coating the stamps and NIL was carried out in a class 1000 cleanroom. Metal deposition was carried out using e-beam evaporation in a system with base pressure of 1×10^{-7} mbar (HHV Auto500). The deposition rate was maintained at 0.1 Å/s and the substrate temperature did not rise above 40°C to not influence the imprinted resist structure.

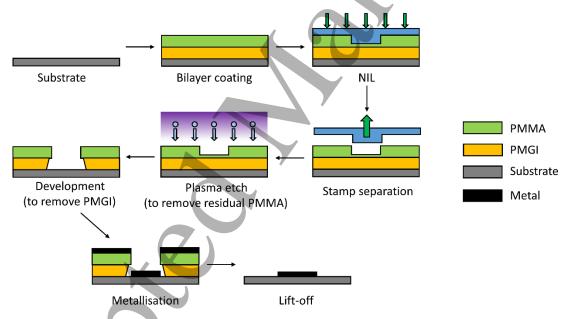


Figure 2. The Bilayer NIL process consists of spin-coating a substrate with a bilayer of PMGI/PMMA, imprinting and removal of the stamp. This is followed by plasma etching to remove residual PMMA resist, which then permits an undercut to be formed by selective removal of PMGI, allowing metallisation and a clean lift-off.

3. Results and Discussion

The results and discussion section is divided into three sections, covering **i**. Optimization of the bilayer nanoimprint lithography process for devices containing both nano and micron feature sizes, **ii**. The electromigration process and **iii**. Single molecule electronic devices.

i. Bilayer Nanoimprint Lithography for Devices containing Both Nano and Micron Feature Sizes

In order to determine the correct imprint parameters several trial imprints were made across a range of temperatures (140°C to 190°C) whilst keeping the resist material, film thickness and imprint pressure constant. As expected, higher temperatures resulted in faster imprints, as shown in Figure 3 (b). For process temperatures greater than 150°C, very little change in imprinted depth was observed after 120 seconds, which signifies that the imprinting is complete in this time.

At the highest temperature, a reduction in the quality of the imprinted features was observed. Voids also appeared in the films when heated above 190°C. It is possible that intermixing of the polymer bilayer may occur above this temperature resulting in localised disturbances of the film surface giving the appearance of voids. At 140°C and 150°C, very slow imprinting occurred and the displaced resist was observed to flow upwards rather than laterally, producing raised regions. This can be seen in Figure 3 (a) as rings of defect free resist surrounding the imprinted regions[26]. (Note: when this occurred, the imprinted depth was measured from the undisturbed resist level away from the imprint.)

For imprints carried out at temperatures between 140°C and 170°C, a reoccurrence of void defects in the resist film was also observed. Unlike those observed at temperatures above 190°C, these defects likely occur as a result of the stamp bending around the stamp protrusions which are in the process of imprinting. The resulting small separation between the low viscosity resist surface and stamp as a result of this bending has been shown to result in a polymer thinning effect as a result of electrostatic interactions between the stamp and resist surface [27]. At 180°C, only a very small number of these void defects occurred, usually located in proximity to dust particles present on the resist but not around the desired features. This is most likely because of reduced stamp bending as the polymer has a reduced viscosity at this higher temperature. Reducing foreign particles should totally eliminate the occurrence of this type of defect at this temperature and imprinting pressure.

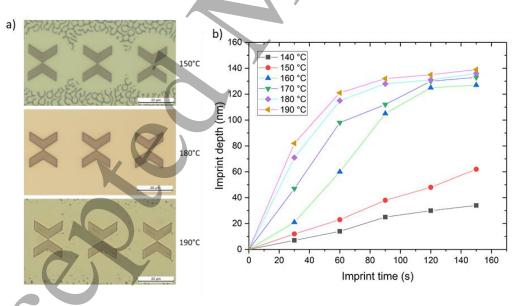


Figure 3. a) Imprinting of the nanojunction (Master) stamp into bilayer resist at increasing temperatures leads to the appearance of flow defects. Areas immediately surrounding the imprinted features appear well imprinted due to a squeeze flow of resist at 150°C while at 190°C voids appear. 180°C gives optimum imprinting behaviour. b) Imprinting depth measured at increasing temperature into 150 nm thick, 15k PMMA resist with constant applied pressure. The CNI imprinter was rapidly cooled to a temperature below 110°C in order to halt the imprinting process, however it is likely that some small amount of imprinting depth will occur during cooling. The bilayer NIL process produces an unwanted residual layer of PMMA (20 nm) after stamping that must be removed before the development and metallization processes. An isotropic oxygen plasma etch (Diener Zepto, 14 minutes) was used to remove this layer. As well as removing the residual polymer, the process can also broaden the imprinted features and for this reason, the duration of etching should be limited to preserve the size of the original design. The correct plasma etch duration was determined by an iterative process. For under-etching, either no development of the PMGI occurred as the residual PMMA layer completely blocks access of the developer solution, or islands of PMMA were observed, which can only be removed after excessive development time which then risks collapse of the bilayer structure.

The correct development time is also crucial to the process. If the development time is too short, remnant PMGI is present as shown by the yellow shadowing of the edges in Figure 4 (a), resulting in poor metallization and lift-off. If the development time is too long, the quality of the replication is poor as there is a partial collapse of the edges, as shown in Figure 4 (c). A development time of 20 s was found to be ideal for the thickness of the PMGI layer used, resulting in clean and crisp features, as shown in Figure 4 (b).

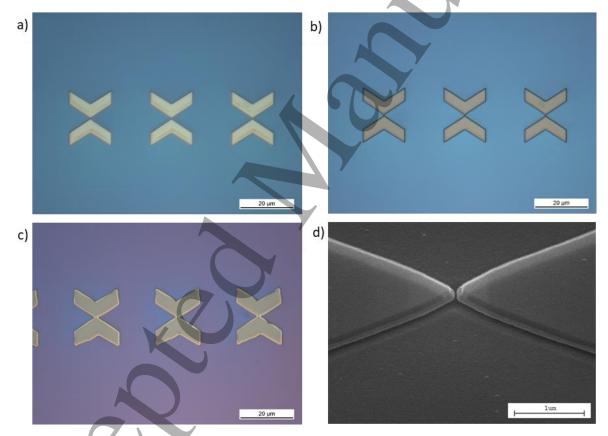


Figure 4. Increasing the length of development from a) 10s, b) 20s and c) 30s of the features in PMMA/PMGI bilayer resist after removal of the residual layer of resist through plasma etching. Imprint time d) SEM image of the nanojunction region after 20s development.

After patterning of the resist by the NIL process, 20nm of gold was deposited onto the wafer, followed by lift-off using acetone and then n-methyl-2-pyrrolidone. Substrates were also gently cleaned using a low power (25W) oxygen plasma exposure for 10 minutes. In the case of the devices used for molecular electronics studies, a two-step lithography approach was used. The second lithography step involved a standard UV lithography process and was used to pattern large area and thick Al contact

pads onto the ends of the four arms of the crosses in order to facilitate electrical connections. The completed devices, in this case, are shown in Figure 5.

Using this approach, and with periodic cleaning of the Master stamp using sonication in acetone and 2-propanol, we have been able to carry out the NIL process over 50 times with no degradation in the imprinted features making this process very attractive for producing large numbers of devices. We see no reason why this process cannot be continued many hundreds of times.

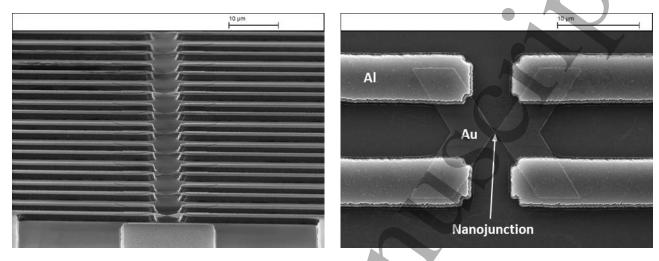


Figure 5. SEM images of a completed device consisting of 15 Au nanojunctions connected to thick Al contact pads.

ii. <u>Electromigration</u>

In this work, electromigration was used to make very small, nanosized gaps (< 2 nm) for molecular electronics studies. Electromigration is a process in which a large current density is driven into a narrow and confined conducting channel [28,29]. The related electron wind of the current displaces atoms and defects, causing them to migrate. Once the current density flowing through the constriction surpasses the electromigration threshold (typically > 10⁸ Am⁻² for gold[30]) the resistance of the nanojunction increases as its cross-sectional area narrows due to material transport. This results in a thinning (or pinching) of the conduction channel in the lateral direction, and usually occurring at the midpoint of the channel where there is maximum heating or at desired position along the length of the channel, that can be determined by placing a notch. With continued application of a current density, usually imparted via voltage pulses, the pinching effect occurs until at some point, a single atom is present in the gap, as typically confirmed by the observation of quantized steps in the conductance[31]. With time, and or more application of pulses, a nanojunction typically forms, which generally consists of the formation of a sub-2 nm gap between the two electrodes [23].

Thermal runaway is a problem inherent with electromigrated break junctions [24,32]. This occurs because the nanojunction resistance increases as electromigration begins and the power being dissipated by the junction, in the form of heat, also increases. The rate of electromigration is increased with elevated temperature and a thermal runaway will most likely occur in the absence of any form of limiting mechanism [33,34]. Nanojunctions broken as a result of thermal runaway tend to have large gaps [35]. Previously reported nanoimprinted devices have been shown to produce large and uncontrollable gap sizes and in addition require heating of the device in order to increase the nanojunction resistance and increase material diffusion [36]. When used for molecular electronics studies this requirement poses the potential risk of thermal decomposition of any molecules present

prior to breaking as well as also excluding the possibility of carrying out electromigration in experiments that study low temperature molecular phenomenon. In our system, this process is avoided by using a feedback voltage source (FVS). This circuit maintains a constant voltage across the nanojunction with the result that the power dissipated by the junction is reduced as electromigration progresses. Using a four terminal device allows us to compensate from resistance that arises as a result of the thin film interconnects on the device as well as contact resistance from the probes used to connect the device to equipment.

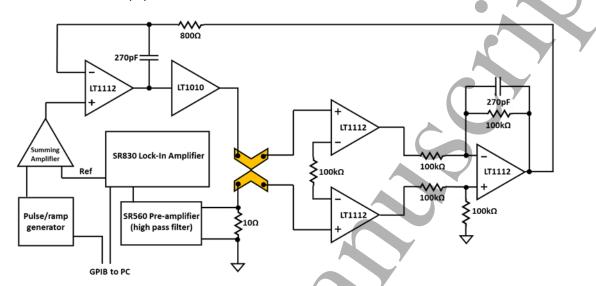


Figure 6. Schematic of the experimental setup including feedback voltage source used to carry out electromigration of the nanojunctions. The gold cross represents the nanojunction device

Figure 6 shows the schematic of the FVS. The voltage drop across the junction is measured by an additional pair of leads that connect to an instrumentation amplifier, the output of which is fed into the non-inverting input of the feedback op-amp. The circuit is stabilised by tuning the capacitance and resistance in the feedback stage such that an adequately fast response is obtained whilst preventing overshoot and oscillation when applying step changes in voltage, similar to the pulses used to carry out electromigration. It was found that adding capacitance across the feedback path of the instrumentation amplifier was also needed in order to provide a stable feedback mechanism without excessive overshoots and oscillation. In our setup, the resistance of the junction is monitored by a lock-in technique by measuring the voltage drop across a 10 Ω resistor between the junction and ground. The lock-in excitation (4 mV, 1 kHz) and is added to the voltage pulses which drive the electromigration process (17% duty cycle, 17 Hz). The setup is computer controlled to increase the voltage applied to the device incrementally per pulse (approximately 150 μ Vs⁻¹) while continually monitoring the nanojunction resistance. As electromigration progresses we see transitions from normal diffusive conduction, to a quasi-ballistic regime as the nanojunction size reduces below the electron scattering length to a size of just a few atoms. At this point only a few transport channels contribute to the overall conduction of the device and in some runs we observe quantised conductance values just before the wire breaks to form nano-sized gap. A sudden jump in resistance indicates that a tunnelling gap has developed, after which the electromigration procedure is automatically halted by the program.

Figure 7 (a) shows the evolution of the electromigration process in the formation of a typical nanojunction. The resistance vs pulse voltage shows the transition from the conductive diffusive

regime (high conductance) to the quasi-ballistic regime, with steps in the conductance in the vicinity of G₀ denoting a limited number of quantized conduction channels. Figure 7 (b) shows a typical AFM image of one junction after undergoing electromigration to form a nanogap. It is clear that a change in the junction dimensions has occurred. However, the actual gap size is well below the resolution of the instrument. We therefore rely on electrical measurements to determine the typical gap size. Figure 7 (c) shows a typical nonlinear current-voltage (I-V) characteristics fitted to the Simmons tunnelling model (equation 1) derived from dI/dV measurements. Although we are not able to measure directly the barrier height or the junction area, these two parameters are quite robust and the fit is determined mainly by the gap size. Fitting over 10 such devices with $\varphi = 1$ eV, we find that the gap size is less than 2 nm confirming that electromigration can be carried out smoothly in devices fabricated using this bilayer technique.

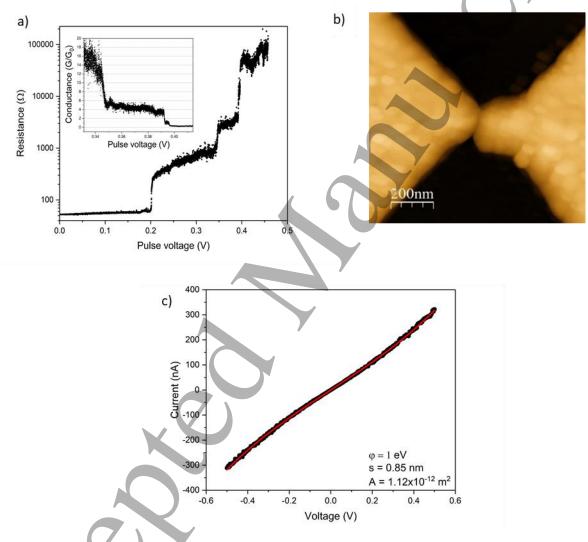


Figure 7. a) Evolution of the device resistance during the electromigration procedure. Resistance is measured between voltage pulses. Inset shows the conductance plotted in terms of the quantized conductance, G_0 , in the region from $20G_0$ to before the transition into a tunnelling regime (< $1G_0$). b) AFM image of a nanojunction after electromigration has formed a nanogap. c) Representative I-V curve measured at room temperature after a nanogap has formed in two devices and showing a fit to the Simmons tunnelling model.

The Simmons tunnelling model:

$$I(V) = \frac{Ae}{2\pi h(\beta \Delta s)^2} \left\{ \varphi \exp\left(\frac{4\pi \beta \Delta s}{h} \sqrt{2m} \sqrt{\varphi}\right) - (\varphi + eV) \exp\left(\frac{4\pi \beta \Delta s}{h} \sqrt{2m} \sqrt{\varphi + eV}\right) \right\}$$
[1]

where *I* is the current, *A* is the junction area, Δs is the gap size, φ is the barrier height and *V* is the applied voltage [37]. β is a constant which can be approximated to unity in the voltage regime over which we make measurements.

iii. Single Molecule Electronic Device

As a proof of concept of our fabrication process, single molecule electronic devices were made containing diphenyl disulphide (DPDS). This compound consists of two phenyl rings connected via two thiol groups. It has been well documented that upon exposure to a gold surface the S-S bond of the DPDS molecule oxidizes to form benzenethiol (BT), which self-assemble onto the electrode surfaces [38–40]. The resulting BT monolayers present an ideal testbed for measuring signatures of molecular electronic transport due to the high conductivity of the delocalised π -orbitals on the phenyl rings. In addition, the versatility of this molecule allows for the construction of many types of molecular junction with differing functionalities though step-wise chemical synthesis with BT forming the initial anchoring unit of a molecular wire [41].

In a typical molecular transport experiment, the molecules are allowed to self-assemble onto the nanojunction and electromigration is used to form a nanogap. The increased temperature during the electromigration process is expected to increase the mobility of molecular species present in the vicinity of the junction area and increase the chance of a molecule moving into the nanogap once formed. An advantage is that the experiment can be performed in situ without the need to remove the device from the measurement setup, which would otherwise expose the clean nanogap to ambient contamination. We first oxygen plasma clean the unbroken nanojunction (15 minutes, 50 W, 0.35 mbar) before removing them from the etching chamber and immediately immersing the device into a 0.04 mM solution of DPDS in ethanol for 30 minutes. We then carry out the electromigration procedure and measure the I-V characteristics of the device. Out of 10 electromigrated nanogaps exposed to DPDS, two measured devices displayed behaviour that was indicative of molecular transport i.e. current suppression at low voltage, due to off-resonant transport, and a lifting of the current suppression at higher voltage. In contrast, devices that contained no DPDS molecules exhibited the typical Simmons I-V curve that is shown in Figure 7 (c). Furthermore, at a voltage of 0.5 V, empty junctions typically had a current of 300 nA compared to 1 µA for a junction containing exposed to DPDS, as shown in Figure 8 (a).

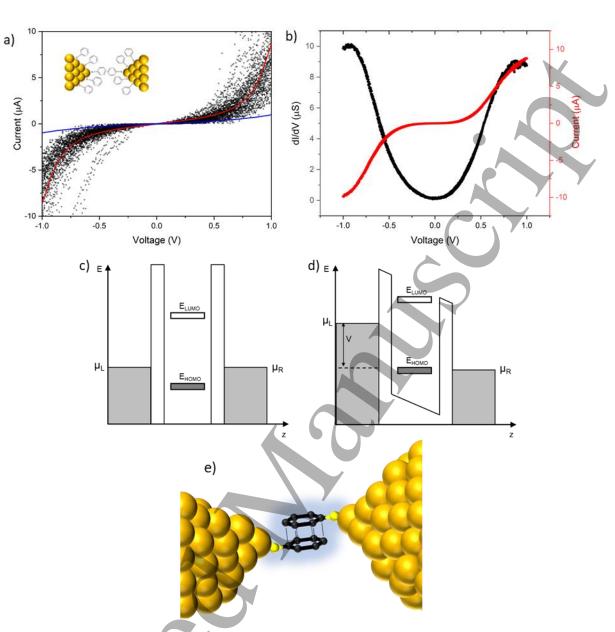


Figure 8. a) Repeated I-V measurements made on a DPDS electromigrated nanogap made over a 12 hrs, (lockin technique 4 mV at 1447 Hz). Current is extracted from dI/dV measurement. Red curve is a fit to the resonant tunnelling model (equation 2) with $\Gamma_{S,D}$ = 34 meV, ε_0 = -0.49 V. The blue curve, showing the lack of a good fit using the Simmons model (φ = 1 eV, s = 0.85 nm, A = 1.12x10⁻¹² m²) indicates that the junction is not an empty tunnel gap. Inset – a schematic of a possible DT configurations within the nanogap. b) Selected single I-V curve (red) and dI/dV (black) from another junction showing similar current suppression c) Schematic of molecular conduction through a gold-BT-BT-gold junction at low voltage where the current is suppressed due to offresonant transport. d) With application of a sufficiently high voltage, the levels shift bringing a molecular level into resonance, lifting the current suppression. e) Expected molecular configuration of the nanojunction consisting of overlapping π orbitals from stacked benzene rings that are thiol bonded to opposite electrodes.

Figure 8 (a) shows the I-V behaviour of a device exposed to DPDS, electromigrated and then measured continuously over a 12 hr period. The I-V curves show some variation, mostly liked due to minor movement of the molecules due to thermal effects, but the current suppression expected at low

voltage is clearly visible. We fit the data with again with the Simmons model as well as a single level resonant tunnelling model (equation 2):

$$I(V) = \frac{4G_0}{e} \frac{\Gamma_S \Gamma_D}{\Gamma_S + \Gamma_D} \left[\arctan\left[\frac{\varepsilon_0 + eV/2}{\Gamma_S + \Gamma_D}\right] - \arctan\left[\frac{\varepsilon_0 - eV/2}{\Gamma_S + \Gamma_D}\right] \right]$$
[2]

where $G_0 = \frac{2e^2}{h}$, ε_0 represents the level position relative to the fermi level of the electrodes, $\Gamma_{S,D}$ are the coupling parameters of the molecule to the electrodes and are set so that $\Gamma_S = \Gamma_D$ [42].

To the best of our knowledge, these are the first reported results of BT measured in Au breakjunction devices. The shape of the I-V curves are very consistent with measurements on similar phenyl compounds in mechanically controlled break junctions[43] and first principles calculations[44]. However, there are a number of differences due to the different molecules used in this study. In contrast to previous work that had molecules with two thiols that could bridge on opposite sides, the molecules used in this study can only bind to Au on one side. The large current (10 μ A at 1.0 V) exhibited in conjunction with the on/off resonant transport indicates strong conduction between the source and drain electrodes. This current is much larger than that observed for a linear bridged biphenyl dithiol[45], having an experimentally measured conductance value of 1.7×10^{-4} G₀, which further indicates the molecules are in a different geometry. The strong symmetry in the current-voltage curves also provides further evidence of a transmission function that is independent of the direction of current flow, as expected in a symmetric system having similar thiol-gold barriers on the drain and source electrodes. We speculate that the high conduction (≈ 0.13 G₀ at 1.0V) is due to overlapping π orbitals from the stacked benzene rings that are bound from opposing electrodes, as shown in Figure 8 (e). This has similarly been observed in π -folded molecular junctions[46].

Further information can be gained by examining the data using Transition Voltage Spectroscopy (TVS), which is now a commonly used tool for probing the frontier molecular orbitals (HOMO or LUMO) in molecular devices. First discussed[47] in terms of tunnelling over an energy barrier but now understood to be better interpreted in terms of coherent Landauer[48] transport, TVS studies have been used both experimentally and theoretically to investigate a variety of partially bridged and fully bridged electrode-molecule-electrode systems. Studies are typically based on traditional gold electrodes[49] but also novel approaches such as contacts made with graphene and carbon nanotubes[50–52].

In Figure 9 the I-V graphs are analysed by plotting $ln(I/V^2)$ vs 1/V, the so-called Fowler-Nordheim (F-N) plot that is used to study the tunnelling to field-emission transition in solid-state junctions. A key feature of the TVS plot is the transition voltage, V_{trans} , which is the position of the first minimum and is proportional to the relative difference between the energy of the highest occupied molecular orbital (HOMO) and the Fermi energy. Figure 9 shows a TVS plot for the data of several I-V sweeps shown in Figure 8 (b). From the graph there is a clear minimum at a voltage 9.51 V⁻¹, which gives a transition voltage of $V_{trans} = 0.11$ V.

Although no previous data exists for the system investigated in this article, a comparison of experimental and theoretical values of V_{trans} with molecules of similar type can be made. In the case of a single phenyl-SH molecule, bonded only to a single gold electrode, a range of theoretical values based on *ab initio* calculations exist, including 0.32 V [53], 0.9 V [54] and 1.69 V [55]. In terms of experimental measurements, a value of 0.95±0.11 V [47] has been found, which would indicate a

midpoint value of the theoretical range is appropriate. In bridged systems, whereby the phenyl molecules have thiols attached to both drain and source electrodes, experimentally determined TVS values of 0.16 V [56] and 0.79±0.2 V [49] have been found for a single phenyl molecule (benzene-1,4-dithiol) and for two phenyl molecules (biphenyl-4,4'-dithiol) respectively. Thus from these literature values, our TVS value of 0.11 V is of closest match to the bridged single phenyl molecule (benzene-1,4-dithiol), having a TVS minimum at 0.16 V. This comparison suggests the TVS value is largely unaffected when either two phenyl molecules are directly bridged via σ bonds or indirectly bridged via vertical stacking. We expect this is because of the nature of the sp² hybridized carbons in the benzene molecule, which forms a ring of delocalized π electrons above the plane of the benzene molecule. In the case of two or more stacked aromatics, such a π –stacked configuration supports through-space transfer which allows the exchange of electrons[57] over sometimes long-range distances[58]. The importance of π - π overlap has long been recognized in thin-film organic electronics (small molecule and polymeric systems), tailored graphene devices[59], supramolecular chemistry and electron transport in biological systems. It has also been used to guide the formation of molecular bridges[60] in single molecule electronics.

In both cases, we expect the low value of the TVS is due to the close alignment between the HOMO level (transport through the HOMO level is known to dominate in this system[61]) of the molecule and the electrodes which is hybridized because of the strong coupling between the thiol and electrode. This is particularly applicable in the case of small molecules[42].

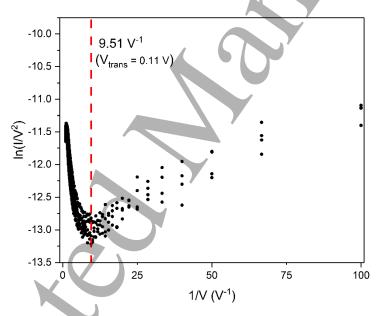


Figure 9. a) Transition Voltage Spectroscopy plot showing a voltage minimum at 9.51 V⁻¹ corresponding to a transition voltage of V_{trans} =0.11 V. The transition voltage relates to the difference between the energy of the highest occupied molecular orbital (HOMO) and the Fermi energy of the electrode.

4. Conclusion

The fabrication of low cost, high-resolution nanoscale devices has been demonstrated using a fast and high throughput, bilayer nanoimprint lithography approach. The approach, allowing the replication of both nano and micron scale features is applicable to a wide range of device types in nano and microelectronics. Optimum parameters were obtained to achieve highly quality imprinting, speed of imprinting and highly reliable lift-off process after metallization. The flow behaviour of the bilayer resist was discussed in relation to the control of flow defects and preserving the bilayer structure to

enable successful tear-free lift-off after metal deposition. The optimum imprinting speed and replication of the features was obtained at a temperature of 180°C and pressure of 2 bar. The bilayer resist structure is maintained after imprinting allowing the development of an undercut and successful lift-off of a thermally deposited 20 nm gold thin film without the need for ultra-sonication. Following fabrication of the nanoscale devices by the bilayer nanoimprint approach, feedback controlled electromigration was used to make nanogaps with separation between the electrodes of less than 2 nm, as determined by fitting of the I-V tunnelling curves to the Simmons tunnel model.

This high throughput method of device fabrication has promising applications in the field of molecular electronics as it allows a large number of devices to be fabricated in one-step, at low cost, and with high speed. To demonstrate the applicability of this approach, devices were made having electrodes functionalized with benzenethiol molecules so that the molecules bridged across the gap to form a coupled junction, as confirmed by the observation of molecular transport signatures in repeated I-V sweeps. In this case, transport through molecular levels is indicated by a current suppression present in the I-V behaviour at low voltage, which is then lifted at high voltage, signifying on- and off-resonant transport as a function of voltage. The large conductance, symmetry of the I-V sweep and small value of the voltage minimum in Transition Voltage Spectroscopy, V_{trans} = 0.11 V, indicates the bridging of the two benzenethiol molecules is by π -stacking.

Formation of the electrodes in this way, having bridged benzenethiol molecules, could have important application for the development of new types of electronic switches/memory devices. Here the device resistance is dependent upon the electronic coupling between the two benzenethiol molecules. This coupling would be dependent upon both the conformation and or electronic spin state of a small molecule present within the transport pathway, which has the ability to be modulated by an external field.

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