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# Strong dispersion effect of cobalt spinel active phase spread over ceria for catalytic N<sub>2</sub>O decomposition: the role of the interface periphery

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Highlights

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10 wt.% of Co<sub>3</sub>O<sub>4</sub>/CeO<sub>2</sub> reproduces the activity of bare Co<sub>3</sub>O<sub>4</sub> catalyst

The highest apparent reaction rate is observed for the 1 wt.% of Co<sub>3</sub>O<sub>4</sub>

Strong dispersion effect is associated with morphology of Co<sub>3</sub>O<sub>4</sub>/CeO<sub>2</sub> system

Co<sub>3</sub>O<sub>4</sub>/CeO<sub>2</sub> interface length accounts for the observed specific activity in deN<sub>2</sub>O

Abstract

A series of Co<sub>3</sub>O<sub>4</sub>/CeO<sub>2</sub> catalysts with increasing cobalt spinel loading in the range of 1-20 wt.% was prepared by incipient wetness impregnation of CeO<sub>2</sub>. The obtained catalysts were thoroughly examined by XRD, XPS, XRF, RS, TEM/EDX/EELS, TPR and BET techniques. The catalytic tests in deN<sub>2</sub>O reaction revealed that the 10 wt.% of cobalt spinel in supported system is able to reproduce the activity of bare Co<sub>3</sub>O<sub>4</sub> catalyst. However, it was found that the catalyst with the lowest content of Co<sub>3</sub>O<sub>4</sub> equal to 1 wt.% exhibits the highest apparent reaction rate per mass of the spinel active phase. The observed activity was explained basing

on the transmission electron microscopy analysis in terms of the dispersion of spinel phase over ceria support. A simple model that accounts for the observed strong dispersion effect is proposed. It consists in a two-step mechanism, where  $\text{N}_2\text{O}$  is dissociated on the spinel nanograins and the resultant oxygen species are preferentially recombined at the  $\text{Co}_3\text{O}_4/\text{CeO}_2$  interface periphery.

## Keywords

cobalt spinel;  $\text{CeO}_2$ ;  $\text{N}_2\text{O}$  decomposition; supported catalyst; oxygen spillover

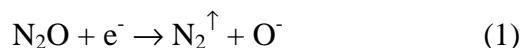
## 1. Introduction

Nitrous oxide has been recognized as one of the most important greenhouse gases with a lifetime of 150 years in the atmosphere, and the global warming potential 310 times higher than that of  $\text{CO}_2$  [1]. Moreover,  $\text{N}_2\text{O}$  contributes to destruction of the ozone layer in the stratosphere. For these reasons, decomposition of  $\text{N}_2\text{O}$  into  $\text{N}_2$  and  $\text{O}_2$  is a subject of continuous intensive investigations [2]. Taking into account the high temperature of thermal decomposition of nitrous oxide, over 650 °C, catalytic decomposition of  $\text{N}_2\text{O}$  is considered as an economically driven solution practical for practical applications.

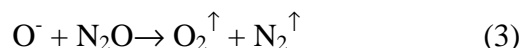
Among the anthropogenic sources of nitrous oxide, adipic and nitric acid plants dominate [3]. In the case of the former, due to high concentration of  $\text{N}_2\text{O}$  in the tail gases and the exothermal effect of its decomposition, operational abatement system has been already developed [3,4]. On the contrary, the low concentration of the pollutant and presence of inhibiting co-reactants ( $\text{H}_2\text{O}$ ,  $\text{O}_2$  and  $\text{NO}_x$ ) in the residual gases of the nitric acid plant results in still persistent searching for efficient low-cost catalyst for low temperature conditions (below 400°C).

Up to now, many catalytic systems such as noble metals [5,6,7], metal oxides [8,9,10], ion-exchanged zeolites, calcined hydrotalcites [[11,12,13], mesoporous silicas [14,15] were investigated for the low temperature de $\text{N}_2\text{O}$  reaction. One of the simple promising generic systems exhibiting high catalytic activity in this reaction is cobalt spinel [16,17,18,19,20], which can be additionally modified by structural [21,22,23,24] and surface donation [24,25,26,27,28]. The  $\text{N}_2\text{O}$  decomposition over cobalt spinel is a classic redox reaction, involving  $\text{N}_2\text{O}$  molecule activation by electron transfer from cobalt(II) cation to  $\text{N}_2\text{O}$

molecule (Equation 1), oxygen species diffusion and finally their recombination into O<sub>2</sub> molecule (Equation 2), closing the catalytic cycle:



Alternatively, dioxygen may also be produced in a non-diffusive pathway via Eley-Rideal mechanism:



Despite the promising reactivity in deN<sub>2</sub>O reaction, preparation of final catalyst based on Co<sub>3</sub>O<sub>4</sub> as a bulk oxide active phase cannot be taken into account due to the high price of cobalt. However, for practical applications this problem can be easily solved by dispersion of cobalt spinel active phase on a high surface area support such as alumina, silica, magnesia or ceria [29,30,31,32,33,34,35]. Among them CeO<sub>2</sub>, owing to its particular properties (redox and oxygen storage capacity), in many cases appeared to be superior for deN<sub>2</sub>O catalytic application [36]. Although ceria was quite frequently used as a carrier of noble metals for nitrous oxide decomposition, it is less used to support redox oxides of transition metals.

The aim of this study was to investigate a supported Co<sub>3</sub>O<sub>4</sub>/CeO<sub>2</sub> catalyst of high deN<sub>2</sub>O activity, in order to understand the mechanistic role of active phase/support periphery. To achieve this goal a series of catalysts was prepared by dispersion of nanocrystalline Co<sub>3</sub>O<sub>4</sub> on the surface of CeO<sub>2</sub> support with various Co loadings and tested in the N<sub>2</sub>O decomposition. Thorough characterization by means of XRD, XPS, XRF, RS, TEM/EDX/EELS, TPR and BET allowed to propose a simple topological model of the catalyst for interpretation of the observed reactivity trends.

## 2. Experimental

### 2.1. Catalysts preparation

The ceria support was prepared from Ce(NO<sub>3</sub>)<sub>3</sub>·6H<sub>2</sub>O (Sigma Aldrich, 99,99%) by calcination at 500°C for 4 h. Cobalt spinel was loaded on CeO<sub>2</sub> by incipient wetness impregnation of CeO<sub>2</sub> with water solution of Co(NO<sub>3</sub>)<sub>2</sub>·6H<sub>2</sub>O (Sigma Aldrich, 98%) with the appropriate concentrations to obtain a cobalt spinel content of 1, 5, 7.5, 10 and 20 wt.% for series. The samples were next dried at 100°C for 12 h, and calcined in air at 400°C for 4 h. Before deN<sub>2</sub>O activity tests the catalysts were activated *in situ* in gas flow by temperature-programed heating

till 600°C monitoring simultaneously the removal of the adsorbed impurity gases, such as H<sub>2</sub>O and CO<sub>2</sub> by mass spectrometer.

## 2.2. Catalysts characterization

Relative bulk content of cobalt and cerium oxides in the investigated catalysts (in form of pellets, 10 mm in diameter) was determined with the use of Energy-Dispersive XRF spectrometer (Thermo Scientific, ARL QUANT'X). The X-rays of 4-50 kV (1 kV step) with the beam size of 1 mm were generated with the Rh anode. The detector used was a 3.5 mm Si(Li) drifted crystal with a Peltier cooling (~ 185 K). For quantitative analysis, the calibration with a series of metallic standards and an UniQuant software were used.

X-ray diffractograms were recorded in a Bruker D8-advance diffractometer, using CuK $\alpha$  radiation ( $\lambda = 1.540598 \text{ \AA}$ ). Diffractograms were recorded for  $2\theta$  in the range of 10° - 80° with a step of 0.02°, and a time of 3 s per step. The BET surface area of the catalysts determined by N<sub>2</sub> adsorption at -196 °C in an automatic volumetric system (Autosorb-6, Quantachrome) showed no significant changes in the surface area upon cobalt spinel loading (65 – 80 m<sup>2</sup>/g).

The Raman spectra recorded at room temperature in ambient conditions were taken using a Renishaw InVia spectrometer equipped with a Leica DMLM confocal microscope and a CCD detector with an excitation wavelength of 785 nm. The Raman scattered light was collected in the spectral range of 100 - 800 cm<sup>-1</sup> with a resolution of 1 cm<sup>-1</sup>. At least five scans were accumulated to ensure a sufficient signal to noise ratio.

Temperature programmed reduction by H<sub>2</sub> of the catalysts was carried out in a Micromeritics Pulse ChemiSorb 2705 device with a thermal conductivity detector (TCD). The tests were performed in a tubular quartz reactor in the range of 20–1050°C (10 °C/min) with the use of 20 mg of fresh catalyst, and the flow rate of the feed (5 % H<sub>2</sub> in Ar) of 40 ml/min. Water was removed by a cold trap (-89 °C) placed before the TCD.

The X-ray photoelectron spectra (XPS) were measured with a Prevac spectrometer equipped with a hemispherical VG SCIENTA R3000 analyzer. The spectra were recorded using a monochromatized AlK $\alpha$  source ( $E = 1486.6 \text{ eV}$ ), and an electron flood gun (FS40A-PS) to compensate the residual charge on the surface. Before spectra acquisition the samples were outgassed at c.a. 50°C for at least 15 min, and the background pressure in the chamber during the measurements was  $5 \times 10^{-9}$  mbar. The spectra were recorded with a pass energy of 100 eV for the survey and narrow scans. All the binding energies were referenced to the C 1s

peak at 285 eV of the adventitious carbon. The Tougaard-Shirley and linear type baselines implemented in CasaXPS software were applied prior the analysis of the area of the photopeaks and estimation of the integration error. The surface composition of the samples was determined by analysis of the Co 2p, Ce 3d and O 1s photoelectron peaks areas.

The  $\text{Co}_3\text{O}_4/\text{CeO}_2$  catalysts (1, 5, 10 wt. % loading) were investigated by means of high resolution analytical transmission electron microscopy (FEI Tecnai Osiris) with X-FEG Schottky field emitter operated at accelerating voltage of 200 kV. The samples were dispersed in ethanol, ultrasonicated (20kHz, 200W, 5min), dropped into a lacey carbon-coated copper grid, and dried at room temperature. The samples were loaded into a beryllium double-tilt low-background holder in order to reduce the production of spurious X-rays. TEM images were recorded by a Gatan CCD camera in a digital format. A Fischione High Angle Annular Dark Field (HAADF) detector for Z contrast imaging was used in the scanning transmission electron microscopy (STEM) mode. Super-X windowless EDX (Energy Dispersive X-ray) detector system with silicon drift detector (SDD) was used for elemental mapping. STEM images coupled with EDX elemental mapping were acquired with applied sample drift correction (Bruker Esprit software). The EELS spectroscopy measurements were acquired on a GIF-Quantum spectrometer (with about 1.0 eV energy resolution) using spectrum imaging technique (SI). The spatial resolution of the chemical information gathered by EELS was kept below 1 nm.

### 2.3. $\text{N}_2\text{O}$ decomposition tests

The  $\text{N}_2\text{O}$  decomposition tests were performed in two modes: isothermal and with a steady temperature ramp. The isothermal tests were carried out at steady-state conditions in the range of 200–600 °C with the heating step of 25 °C, at atmospheric pressure in a 10 mm i.d. cylindrical fixed-bed reactor, with 50 mg of the catalyst and a total gas flow of 50 ml/min ( $\text{GHSV} = 30'000 \text{ h}^{-1}$ ). The catalytic bed was packed between plugs of quartz wool. The 1000 ppm  $\text{N}_2\text{O}/\text{He}$  mixture was used. The gas composition was analyzed by a HP 6890 gas chromatograph equipped with a thermal conductivity detector and two serial columns (Porapak Q, for  $\text{N}_2\text{O}$ , and Molecular Sieve 13X, for  $\text{O}_2$  and  $\text{N}_2$ ). Tests with a steady temperature ramp, temperature-programmed catalytic reaction, were performed in a quartz flow reactor in the range of 20–600°C (10 °C/min) with the use of 300 mg of the catalyst (sieve fraction of 0.2–0.3 mm), flow rate of the feed (5%  $\text{N}_2\text{O}$  in Ar) of  $7'000 \text{ h}^{-1}$ . The

reaction progress was monitored with a quadrupole mass spectrometer (RGA200, SRS, lines for  $m/z = 44, 32, 28, 30$  and  $18$ , corresponding to  $N_2O, O_2, NO, N_2$  and  $H_2O$ ).

To check the catalyst stability catalytic tests were always performed at least in two consecutive runs. In all cases the results were fully reproducible within the experimental error. It is worth mentioning that this crucial issue was already investigated at both laboratory and pilot plant scales, as describes in our previous papers [24,37]. The results of the studies confirmed that no appreciable deterioration of the catalyst performance takes place for more than 1600 h time on steam.

### 3. Results and Discussion

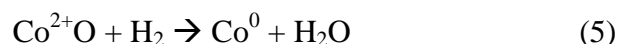
#### 3.1. Elemental and structural characterization

Elemental composition analysis of the investigated samples is summarized in Table 1. The bulk content of  $Co_3O_4$  phase, based on XRF analysis, reflects the desired cobalt phase loading. The surface composition is discussed in the XPS section. The crystalline structure of the prepared catalysts was studied by XRD and Raman spectroscopy. The X-ray diffraction patterns of the bare ceria support and supported cobalt spinel catalysts are shown in Figure 1. The most intense diffraction line at a  $2\theta$  value of  $28.6^\circ$  corresponds to the (111) plane of the cubic ceria oxide phase (ICSD – 28753). The diffraction lines at  $2\theta$  values of  $33.2^\circ, 47.5^\circ$  and  $56.4^\circ$  are assigned to (200), (220) and (311) reflections of  $CeO_2$ , respectively. All of the diffraction peaks of cerium(IV) oxide are observed in supported cobalt catalysts, as well. The presence of cobalt spinel for the samples with the loading of 10 and 20 wt.% is manifested by peaks at  $2\theta$  value of  $36.8^\circ, 44.8^\circ$  and  $65.5^\circ$  due to (311), (400) and (440) reflections of  $Co_3O_4$  phase, respectively (69378-ICSD). The absence of the diffraction peaks of the cobalt spinel samples with the content lower than 10 wt.% is caused by the small particles size of  $Co_3O_4$  dispersed on the ceria surface as reported in [38,39], but their presence was confirmed by the TEM and Raman results (see below). The XRD results were supported by the Raman spectroscopy characterization. The Raman spectrum of the ceria support, presented in Figure 2, shows a characteristic single, broad and intense band located at  $465\text{ cm}^{-1}$ . This peak is assigned to the  $F_{2g}$  phonon mode of cubic fluorite  $CeO_2$  [40,41]. This band was also detected in the Raman spectra for the ceria-supported cobalt spinel catalysts with the  $Co_3O_4$  content in the whole range of 1-20 wt.%. Additionally, in the spectra of the ceria-supported cobalt catalysts (5-20 %wt.) five active Raman modes typical of the spinel structure located at 196,

482, 522, 619 and 691  $\text{cm}^{-1}$ , were observed, confirming the presence of nanocrystalline  $\text{Co}_3\text{O}_4$  [42,43,44]. The position of the  $\text{F}_{2g}$  peak of the ceria support does not change upon deposition of the cobalt oxide phase regardless the Co loading, implying that cobalt do not form a solid solution with ceria, in contrast to previous claims [45]. Furthermore, it can be seen that introducing cobalt oxide results in a strong decrease of the  $\text{CeO}_2$  support peak intensity, which may be caused by dispersion of cobalt oxide over the support and its higher absorption coefficient.

### 3.2. Catalysts characterization by $\text{H}_2$ -TPR

The reducibility of  $\text{CeO}_2$ , 1%  $\text{Co}_3\text{O}_4/\text{CeO}_2$ , 10%  $\text{Co}_3\text{O}_4/\text{CeO}_2$  and  $\text{Co}_3\text{O}_4$ , studied by  $\text{H}_2$ -TPR is presented in Figure 3. The TPR profile of pure  $\text{CeO}_2$  shows two broad peaks centered at 550°C and 850 °C. They have been ascribed to the reduction of the surface and bulk  $\text{Ce}^{4+}$  ions to  $\text{Ce}^{3+}$ , respectively [46]. Deposition of cobalt spinel on the ceria surface leaves the bulk reduction peak of ceria intact, whereas the surface reduction is broadened and gradually disappears. The ceria carrier exhibits even stronger effect on the reducibility of the supported  $\text{Co}_3\text{O}_4$  particles. The two overlapping peaks observed for bulk spinel, due to reduction of  $\text{Co}^{3+}$  ions to  $\text{Co}^{2+}$  (below 350 °C) and  $\text{Co}^{2+}$  ions to metallic Co (above 350 °C) [27,47] become better resolved and systematically shifted toward lower temperatures (Equations 4 and 5).



This indicates that ceria is acting not only as a mere carrier but interacting strongly with the deposited spinel changing substantially the redox properties of both the active phase and the support. Since this shift increases with the cobalt spinel dispersion (see the microscopic observations section) it can be assumed that the observed redox changes are due to the interaction at the periphery of the  $\text{Co}_3\text{O}_4/\text{CeO}_2$  interface.

### 3.3. Spectroscopic XPS characterization

The ceria support together with the  $\text{Co}_3\text{O}_4/\text{CeO}_2$  catalysts with 1, 5, 10 and 20 wt. %  $\text{Co}_3\text{O}_4$  loadings were investigated by XPS spectroscopy. The XPS spectra of Co 2p, Ce 3d and O 1s regions are shown in Figure 4, Figure 5 and Figure 6, respectively. The relative amount of Co and Ce are summarized in Table 1. The XPS and XRF results differ with regard to the



cobalt content because the both methods probe different sample regions. Whereas XPS collects electrons emitted from about 2, 3 top crystal layers of the unshedded surface grains, the XRF which is based on the X-ray photons measurements penetrates the sample on a micrometric scale. In other words, a low escape depth of photoelectrons in contrast to the long escape depth of X-radiation result in different concentrations of the components obtained from the two methods, if the sample is inherently heterogeneous, which is the case of the cobalt spinel supported over ceria. Moreover, the XPS based results are more strongly dependent on the material morphology than those obtained from XRF. The larger the agglomeration of the cobalt particles the less are probed by surface sensitive XPS. These results, in fact, point out that the ceria support is non-uniformly covered by the  $\text{Co}_3\text{O}_4$  particles in the case of the higher loadings, which is in line with the TEM observations.

For the  $\text{Co}_3\text{O}_4/\text{CeO}_2$  samples the Co  $2p_{3/2}$  and  $2p_{1/2}$  lines appeared at 780 and 795 eV. The shape and the position of the observed lines are typical for cobalt spinel structure. With the increasing cobalt loading, the position of the main Co  $2p_{3/2}$  peak is shifted towards lower BE, from 780.6 eV for 1%  $\text{Co}_3\text{O}_4$  to 780.3, 780.1 and 779.7 eV for 5%, 10% and 20%, respectively. The energy shift is brought about by the changing in the relative content of  $\text{Co}^{2+}$  and  $\text{Co}^{3+}$  ions. At lower concentrations, due to a stronger interaction with ceria, the relative abundance of  $\text{Co}^{2+}$  with a 2p photopeak at higher BE is increased. Also, slightly smaller  $\text{Co}_3\text{O}_4$  crystals, observed in TEM for lower loadings, may result in a change of local surface Madelung energy of cobalt ions resulting in the observed binding energy changes. The main Co 2p photopeak is often analyzed in terms of components arising from tetrahedral  $\text{Co}^{2+}$  and octahedral  $\text{Co}^{3+}$  [48]. The band at 779.2–779.6 eV is characteristic for  $\text{Co}^{3+}$  and that at 780.3–780.7 eV for  $\text{Co}^{2+}$ . The low quality of the Co 2p spectra, despite the long data acquisition during XP analysis, renders it impossible to decompose the peak in order to quantitatively compare  $\text{Co}^{3+}$  and  $\text{Co}^{2+}$  components for the series of the catalysts.

The O 1s spectra of  $\text{CeO}_2$  reveal two main peaks, as shown in Figure 5. The peak at 529.3 eV is assigned to the lattice  $\text{O}^{2-}$  oxygen ions, while the second, less intense peak above 531 eV is usually assigned to surface hydroxyl groups, carbonates/adsorbed  $\text{CO}_2$ , etc. [49]. The position of the lattice oxygen peak does not change upon addition of different amounts of the  $\text{Co}_3\text{O}_4$  phase. This is not surprising, since the O 1s peak for  $\text{Co}_3\text{O}_4$  is expected around 529.7 eV [50].

The Ce 3d XPS spectra of  $\text{Co}_3\text{O}_4/\text{CeO}_2$  samples consist of several overlapping peaks (Figure 6). Their positions and assignment are summarized in Table 2. Those denoted by v, v'', v'''

and  $u$ ,  $u''$ ,  $u'''$  are attributed to the  $Ce^{4+}$  state, while those denoted by  $v^0$  (usually poorly resolved),  $v'$  and  $u$ ,  $u'$  are assigned to the  $Ce^{3+}$  state [49]. Since the Scofield coefficient,  $\sigma$ , escape depth,  $\lambda$ , and other instrumental parameters,  $S(E_{kin})$  are virtually the same for  $Ce^{3+}$  and  $Ce^{4+}$  states, the ratio of cumulative areas,  $I$ , due to bands assigned to  $Ce^{3+}$  and  $Ce^{4+}$  is practically equal the surface concentration ratio  $x_{Ce^{3+}}/x_{Ce^{4+}}$  as implied by Equation 6:

$$\frac{I_{Ce^{3+}}}{I_{Ce^{4+}}} = \frac{\sigma_{Ce^{3+}} S(E_{kin,Ce^{3+}}) \lambda_{Ce^{3+}} (S_{kin,Ce^{3+}})^{x_{Ce^{3+}}}}{\sigma_{Ce^{4+}} S(E_{kin,Ce^{4+}}) \lambda_{Ce^{4+}} (S_{kin,Ce^{4+}})^{x_{Ce^{4+}}}} = \frac{x_{Ce^{3+}}}{x_{Ce^{4+}}} \quad (6)$$

The presence of  $Ce^{3+}$  is observed even for the support reference sample, without the  $Co_3O_4$ , and in all cases it can be accounted for by the oxygen release upon heat treatment of the catalysts [49]. It should be noted, however, that the relative amount of  $Ce^{3+}$  increases with the content of cobalt spinel loading (Table 1), in line with the TPR results. The XPS derived atomic Co/Ce ratio for the investigated catalysts differs from that obtained from the bulk XRF analysis, indicating much lower quantities of cobalt at the surface. Apparently, this effect is of general nature as it is observed for cobalt oxide supported on ceria [51] and ceria supported on cobalt oxide as well [52].

### 3.4. Morphological characterization by TEM microscopy

HAADF STEM images coupled with EDX maps of the  $Co_3O_4/CeO_2$  catalysts presented in Figure 7a-c provide an overview of the typical morphological and chemical features of the examined samples with the  $Co_3O_4$  loading of 1, 5 and 10 wt. %. Each sample exhibited 5-30 nm individual  $CeO_2$  particles of elongated shape epitomized in the Figure 7a'-c' as nanofibers (marked in orange), most often forming dispersed and loosely-bound particle aggregates. In turn, cobalt spinel particles exhibit a round shape (marked in red). In the case of 1 wt.% load sample ultrafine  $Co_3O_4$  essentially isolated particles of the size below 10 nm can be distinguished, decorating ceria support (Figure 7a, a'). As the cobalt loading increases the individual spinel particles agglomerate forming larger patches (marked in red envelopes in Figure 7b'-c') of the averaged size of 20-50 nm and 50-100 nm for 5 and 10 wt. %  $Co_3O_4$  loading, respectively. In the corresponding HRTEM image (Figure 7d, d') it is revealed that the spinel agglomerates are formed by individual nanocrystals of rhombicuboctahedral shape described previously in detail in [53]. The appearance of lattice fringes indicates a crystalline character of both the ceria support and cobalt spinel active phase. The FFT patterns of the  $Co_3O_4/CeO_2$  catalysts were projected by formula units of the cubic crystal structure along different crystallographic directions, as exemplified in Figure 7d-insert. Several exposed

facets of the  $\text{CeO}_2$  can be identified as the (111) and (200) crystal planes with 0.31 nm and 0.27 nm lattice spacing, indexed along the [011] zone axis. For  $\text{Co}_3\text{O}_4$  the interplanar distance of HRTEM fringes were identified as 0.47 nm and 0.28 nm, which agreed well with the  $d$ -value of the (111) and (110) crystal planes of cobalt spinel. The EELS chemical composition map [54], acquired from the mixed particles area with nanometer spatial resolution (Figure 7d'-insert) clearly shows the presence of the individual elemental contrast, which in combination with HRTEM images reveals the  $\text{Co}_3\text{O}_4$ - $\text{CeO}_2$  periphery in the nanoscale. Summarizing, in the case  $\text{Co}_3\text{O}_4$  loading of 1% wt. the spinel active phase is present as isolated nanocrystal of 5-30 nm, whereas for increased loading they agglomerate into patches of 20-50 nm and 50-100 nm for 5 and 10 wt. %  $\text{Co}_3\text{O}_4$ , respectively. As a result the periphery between the active spinel phase and the ceria support is dramatically decreased calculated per cobalt loading (vide infra).

### 3.5. $\text{N}_2\text{O}$ decomposition tests

Typical  $\text{N}_2\text{O}$  decomposition results for ceria supported  $\text{Co}_3\text{O}_4$  catalysts are shown in Figure 8, where the experimental data are expressed as  $\text{N}_2\text{O}$  conversion ( $X_{\text{N}_2\text{O}}$ ) as a function of temperature. In order to analyze the effect of the cobalt spinel loading on the catalytic activity, the conversion and the apparent reaction rate ( $\mu\text{mol}_{\text{N}_2\text{O}}\cdot\text{g}_{\text{Co}_3\text{O}_4}^{-1}\cdot\text{s}^{-1}$ ) at 400°C for the series of  $\text{Co}_3\text{O}_4/\text{CeO}_2$  catalysts were calculated (Table 3). The results clearly show that the catalytic activity of the material increases substantially with the loading of the cobalt spinel (Figure 9, blue line, conversion). It may be noticed that the most active sample, 10 wt.%  $\text{Co}_3\text{O}_4/\text{CeO}_2$ , with the conversion equal to 43% at 400°C, corresponds to the activity of the reference bulk cobalt spinel. However, the apparent reaction rate normalized to the spinel content derived from XRF for the series of samples increases dramatically with at low  $\text{Co}_3\text{O}_4$  loadings (Figure 9, green line,  $r_{\text{XRF}}$ ). This effect persists even in the case when reactivity data are normalized not to total (XRF) but to the surface cobalt concentration obtained from XPS data (Figure 9, black line,  $r_{\text{XPS}}$ ).

These results imply that apart from the cobalt loading there is also a morphological factor associated with the different dispersion of the spinel active phase over the ceria, depending on the cobalt content (see Figure 7). Indeed, following the recent suggestions [35] we may assume that for the  $\text{N}_2\text{O}$  decomposition a well-developed interface between  $\text{Co}_3\text{O}_4$  and  $\text{CeO}_2$  plays a beneficial role. We may thus propose that the two-step mechanism of  $\text{deN}_2\text{O}$  operates

preferably at two topographically different sites. The first step of  $\text{N}_2\text{O}$  dissociation occurs on the spinel component, whereas the second step of recombination of the resultant oxygen species proceeds at higher turnover at the periphery of the  $\text{Co}_3\text{O}_4$  grains/agglomerates on the  $\text{CeO}_2$  surface. The involvement of the ceria support is implied by the dependence of the  $\text{N}_2\text{O}$  conversion on the ceria reduction degree (Figure 10, where  $X_{\text{N}_2\text{O}}$  at  $400^\circ\text{C}$  is plotted against  $\text{Ce}^{3+}$  fraction). The extent of interaction of the cobalt phase with the ceria support can be followed by the relative amount of the cerium in the reduced state, based on the XPS measurements (Table 1). With the increasing cobalt loading, the amount of the  $\text{Ce}^{3+}$  ions increases, indicating again the importance of the  $\text{Co}_3\text{O}_4/\text{CeO}_2$  interface area, at least in an indirect way.

In order to account for the observed trends of specific activity of the dispersed cobalt spinel phase ( $r_{\text{XRF}}$ ,  $r_{\text{XPS}}$ ) two simple geometrical models of the  $\text{Co}_3\text{O}_4/\text{CeO}_2$  interface were tested. Basing on the TEM observations, at the first approximation we may approximate the  $\text{Co}_3\text{O}_4$  spinel nanometric domains as hemispherical or cylindrical in shape. These geometrical models were then used to validate the assumption that the specific reaction rate,  $r_{\text{peri}}$ , is proportional to the extent of the  $\text{Co}_3\text{O}_4/\text{CeO}_2$  interface periphery. As revealed by the TEM analysis for 1 wt.%  $\text{Co}_3\text{O}_4$  loading, the cobalt spinel nanoparticles are in the size of about 10 nm, while for higher loadings, the  $\text{Co}_3\text{O}_4$  crystallites agglomerate into the compact domains with the average sizes of ~35 nm and ~75 nm for 5 and 10 wt.%  $\text{Co}_3\text{O}_4$ , respectively. The periphery of such domains may be then used for regularization of the observed  $r_{\text{XRF}}$  values in terms of reaction rate express as number of  $\text{N}_2\text{O}$  molecules ( $\mu\text{mol}_{\text{N}_2\text{O}}$ ) decomposed per unit periphery length ( $\text{nm}^{-1}$ ), thus the ultimate specific rate of de $\text{N}_2\text{O}$  decomposition was calculated ( $r_{\text{peri}}/\mu\text{mol}_{\text{N}_2\text{O}}\cdot\text{s}^{-1}\cdot\text{nm}^{-1}$ ). In case of the cylindrical model all of the calculated specific reaction rates were in the narrow range of 0.8 – 0.9  $\mu\text{mol}_{\text{N}_2\text{O}}\cdot\text{s}^{-1}\cdot\text{nm}^{-1}$  (Figure 9, red line,  $r_{\text{peri}}$ ). For the semi-spherical model the reaction rate varied from 0.27  $\mu\text{mol}_{\text{N}_2\text{O}}\cdot\text{s}^{-1}\cdot\text{nm}^{-1}$  for 1 wt.%  $\text{Co}_3\text{O}_4/\text{CeO}_2$  to 2.5  $\mu\text{mol}_{\text{N}_2\text{O}}\cdot\text{s}^{-1}\cdot\text{nm}^{-1}$  for 10 wt.%  $\text{Co}_3\text{O}_4/\text{CeO}_2$ . Thus, only in case of the cylindrical model, the size of the  $\text{Co}_3\text{O}_4$  domains, derived from the TEM pictures (Figure 7a'-c'), allows for the regularization of the specific reactivity rates and their fairly good invariance with the  $\text{Co}_3\text{O}_4$  loading implies that it can be regarded as circumferential turnover frequency.

The revealed regularization of the specific activity related to the interface length accounts for the strong dispersion effect of cobalt spinel active phase spread over ceria for the catalytic  $\text{N}_2\text{O}$  decomposition. In relation to the influence of the dispersion of cobalt spinel

nanoparticles on the deN<sub>2</sub>O reactivity, an important role of the average distance between active centres located on cobalt spinel and those on ceria support can be pointed out. The redox properties of the cobalt centers would be in this case responsible for the activation of N<sub>2</sub>O molecules and formation of atomic oxygen. The ceria support, in turn, would be responsible for the enhanced diffusion and recombination of oxygen, closing the catalytic cycle. It should be noted however, that in the case of large spinel domains the surface oxygen intermediates, formed in their central part, most probably recombine before reaching the Co<sub>3</sub>O<sub>4</sub>/CeO<sub>2</sub> interface region and are excluded from the beneficial two-phase mechanism. This explains why the cylindrical model is more adequate than the semi-spherical one.

## Conclusions

A series of ceria supported cobalt spinel with 1-20 wt.% loading catalysts was thoroughly characterized by XRD, XRF, XPS, TEM/EDX/EELS, RS, TPR, BET techniques and investigated in the catalytic decomposition of nitrous oxide. A strong dispersion effect on the catalyst turnover rate was observed and associated with a specific morphology of the spinel active phase due to progressive agglomeration of the Co<sub>3</sub>O<sub>4</sub> nanocrystalites into compact domains with the increasing cobalt loading. A simple topological model of the Co<sub>3</sub>O<sub>4</sub>/CeO<sub>2</sub> interface periphery was proposed for regularization of the observed specific reactivity rates. The cylindrical shape of Co<sub>3</sub>O<sub>4</sub> domains were used to normalize the deN<sub>2</sub>O reaction rate with respect to the cobalt content and the length of the Co<sub>3</sub>O<sub>4</sub>/CeO<sub>2</sub> interface periphery. A fairly good invariance of such reactivity parameter with the Co<sub>3</sub>O<sub>4</sub> loading indicates that it can be regarded as circumferential turnover frequency. It implies a two-step mechanism operating at the interface, where the redox properties of the cobalt component is responsible for the dissociation of N<sub>2</sub>O molecules and formation of oxygen intermediates, whereas the ceria periphery is responsible for the enhanced diffusion and recombination of oxygen adspecies, closing the catalytic cycle.

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## Table captions

wt.% Co <sub>3</sub> O <sub>4</sub> *	XRF		XPS		
	wt.% Co <sub>3</sub> O <sub>4</sub> #	Co/Ce*	Co/Ce#	wt.% Co <sub>3</sub> O <sub>4</sub> #	Ce <sup>3+</sup> /(Ce <sup>3+</sup> +Ce <sup>4+</sup> ) x100 % (atomic)
0	-	-	-	-	33.4
1	1.5	0.02	0.02	0.7	34.9
5	5.9	0.11	0.03	1.4	35.0
10	10.4	0.24	0.05	2.1	36.4
20	16.7	0.54	0.13	5.9	37.3

Table 2

band	Binding energy/eV				
	CeO <sub>2</sub>	1 wt.% Co <sub>3</sub> O <sub>4</sub> /CeO <sub>2</sub>	5 wt.% Co <sub>3</sub> O <sub>4</sub> /CeO <sub>2</sub>	10 wt.% Co <sub>3</sub> O <sub>4</sub> /CeO <sub>2</sub>	20 wt.% Co <sub>3</sub> O <sub>4</sub> /CeO <sub>2</sub>
v <sub>0</sub>	880.0	880.9	-	-	-
v	882.0	882.2	882.1	882.3	882.2
v'	883.9	884.1	883.4	883.7	883.6
v''	888.6	888.8	888.7	888.9	888.9
v'''	898.1	898.2	898.1	898.3	898.2
u <sub>0</sub>	900.6	900.8	900.6	900.8	900.7
u'	902.1	902.6	901.9	902.6	902.1
u''	907.3	907.3	907.3	907.3	907.4
u'''	916.5	916.6	916.5	916.7	916.6

Table 3

wt.% $\text{Co}_3\text{O}_4$	$\text{N}_2\text{O}$ conversion at 400 $^{\circ}\text{C}$	$r_{400}/\mu\text{mol}_{\text{N}_2\text{O}}\cdot\text{g}_{\text{Co}_3\text{O}_4}^{-1}\cdot\text{s}^{-1}$
0	0.04	-
1	0.22	8.18
5	0.24	1.78
10	0.43	1.60
20	0.38	0.71
100	0.44	0.16



## Figure captions

**Figure 1.** XRD patterns of  $\text{Co}_3\text{O}_4/\text{CeO}_2$  samples with different content of  $\text{Co}_3\text{O}_4$ ; A – 20 wt.%  $\text{Co}_3\text{O}_4/\text{CeO}_2$ , B – 10 wt.%  $\text{Co}_3\text{O}_4/\text{CeO}_2$ , C – 5 wt.%  $\text{Co}_3\text{O}_4/\text{CeO}_2$ , D – 1 wt.%  $\text{Co}_3\text{O}_4/\text{CeO}_2$ , E –  $\text{CeO}_2$ .

**Figure 2.** Raman spectra of the  $\text{Co}_3\text{O}_4/\text{CeO}_2$  catalysts with different content of  $\text{Co}_3\text{O}_4$ ; A – 20 wt.%  $\text{Co}_3\text{O}_4/\text{CeO}_2$ , B – 10 wt.%  $\text{Co}_3\text{O}_4/\text{CeO}_2$ , C – 5 wt.%  $\text{Co}_3\text{O}_4/\text{CeO}_2$ , D – 1 wt.%  $\text{Co}_3\text{O}_4/\text{CeO}_2$ , E –  $\text{CeO}_2$ .

**Figure 3.**  $\text{H}_2$ -TPR profiles for the  $\text{Co}_3\text{O}_4/\text{CeO}_2$  catalysts with different content of  $\text{Co}_3\text{O}_4$ ; A –  $\text{Co}_3\text{O}_4$ , B – 10 wt.%  $\text{Co}_3\text{O}_4/\text{CeO}_2$ , C – 1 wt.%  $\text{Co}_3\text{O}_4/\text{CeO}_2$ .

**Figure 4.** XPS spectra of Co2p, region of  $\text{Co}_3\text{O}_4/\text{CeO}_2$  catalysts; A – 20 wt.%  $\text{Co}_3\text{O}_4/\text{CeO}_2$ , B – 10 wt.%  $\text{Co}_3\text{O}_4/\text{CeO}_2$ , C – 5 wt.%  $\text{Co}_3\text{O}_4/\text{CeO}_2$ , D – 1 wt.%  $\text{Co}_3\text{O}_4/\text{CeO}_2$ .

**Figure 5.** XPS spectra of O1s, region of  $\text{Co}_3\text{O}_4/\text{CeO}_2$  catalysts; A) 20 wt.%  $\text{Co}_3\text{O}_4/\text{CeO}_2$ , B) 10 wt.%  $\text{Co}_3\text{O}_4/\text{CeO}_2$ , C) 5 wt.%  $\text{Co}_3\text{O}_4/\text{CeO}_2$ , D) 1 wt.%  $\text{Co}_3\text{O}_4/\text{CeO}_2$ , E)  $\text{CeO}_2$ .

**Figure 6.** XPS spectra of Ce3d, region of  $\text{Co}_3\text{O}_4/\text{CeO}_2$  catalysts; A – 20 wt.%  $\text{Co}_3\text{O}_4/\text{CeO}_2$ , B – 10 wt.%  $\text{Co}_3\text{O}_4/\text{CeO}_2$ , C – 5 wt.%  $\text{Co}_3\text{O}_4/\text{CeO}_2$ , D – 1 wt.%  $\text{Co}_3\text{O}_4/\text{CeO}_2$ , E –  $\text{CeO}_2$ .

**Figure 7.** Representative series of S/TEM images:  $\text{Co}_3\text{O}_4/\text{CeO}_2$  with different  $\text{Co}_3\text{O}_4$  loading equal to 1, 5, 10 wt.%, respectively together with HRTEM images. HAADF STEM images coupled with EDX elemental distribution of Co-K $\alpha$  (red) (a-c), graphical overlay of the  $\text{Co}_3\text{O}_4$ (red)/ $\text{CeO}_2$ (orange) epitomizing samples differentiation in particles distribution and agglomeration (a'-c'). HRTEM images together with indicated crystal shapes overlays (d-d'), indexed  $\text{Co}_3\text{O}_4/\text{CeO}_2$  FFT patterns (d-insert), EELS elemental map of the 10%  $\text{Co}_3\text{O}_4/\text{CeO}_2$  at nanometer scale: color map overlaying background corrected absorption edges of Ce-M $_{4,5}$  (green), Co-L $_{2,3}$  (red), O-K (black) (d'-insert).

**Figure 8.** Conversion curves for  $\text{N}_2\text{O}$  decomposition reaction over the  $\text{Co}_3\text{O}_4/\text{CeO}_2$  catalysts.

**Figure 9.** Catalytic activity in  $\text{N}_2\text{O}$  decomposition reaction represented by the conversion of  $\text{N}_2\text{O}$  and apparent reaction rate per gram of cobalt spinel with its concentration derived from

XRF –  $r_{\text{XRF}}$ , XPS –  $r_{\text{XPS}}$ , and apparent reaction rate per the interface  $\text{Co}_3\text{O}_4\text{-CeO}_2$  peripheral length -  $r_{\text{peri}}$ , at 400 °C.

**Figure 10.** Conversion of  $\text{N}_2\text{O}$  at 400°C plotted against relative  $\text{Ce}^{3+}$  content.

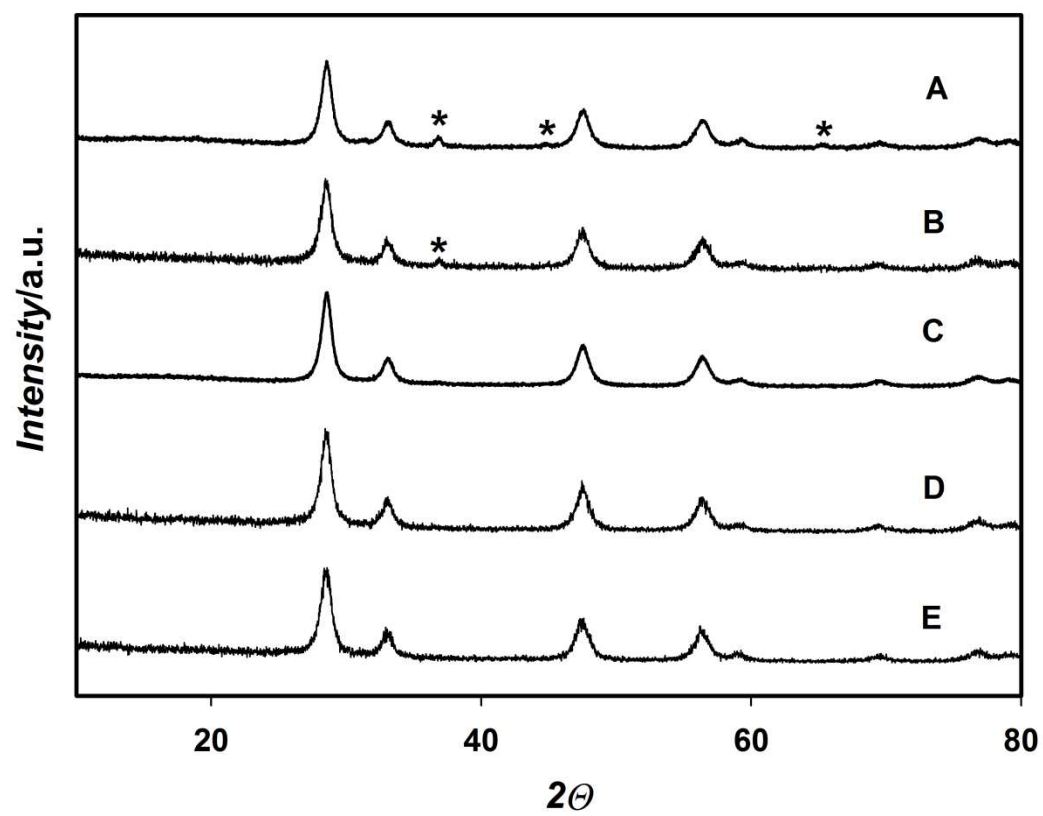


Figure 1

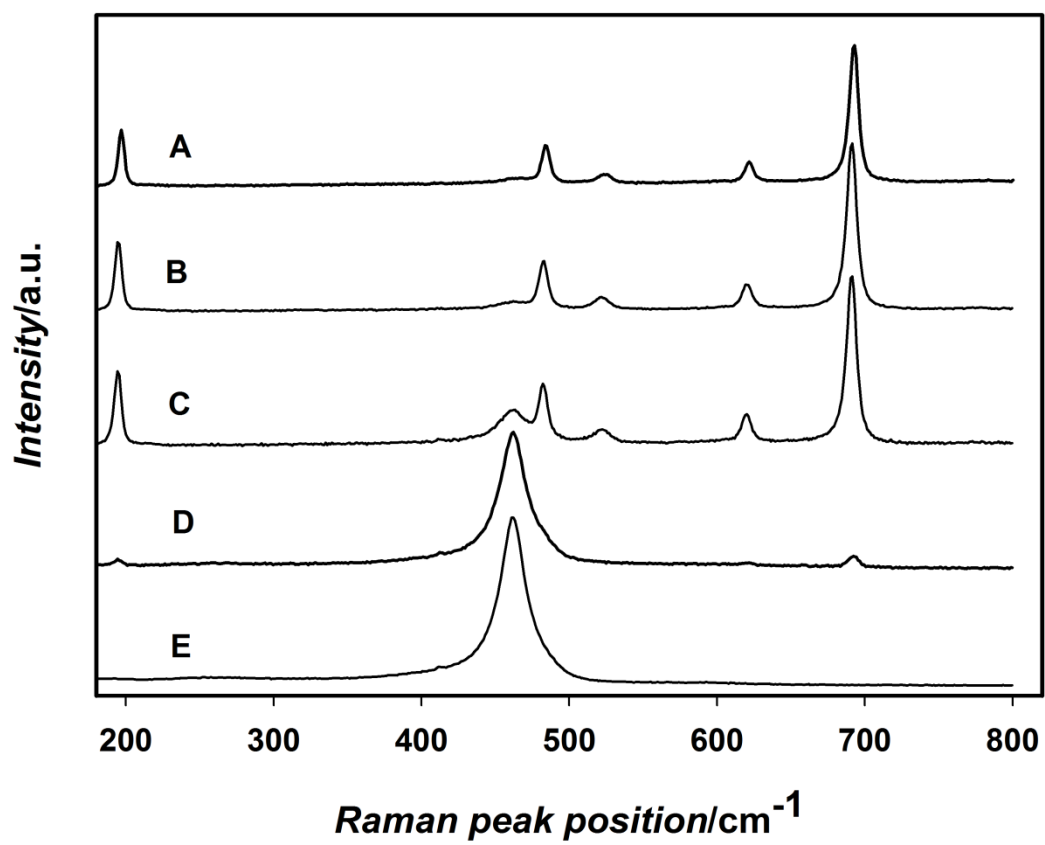


Figure 2

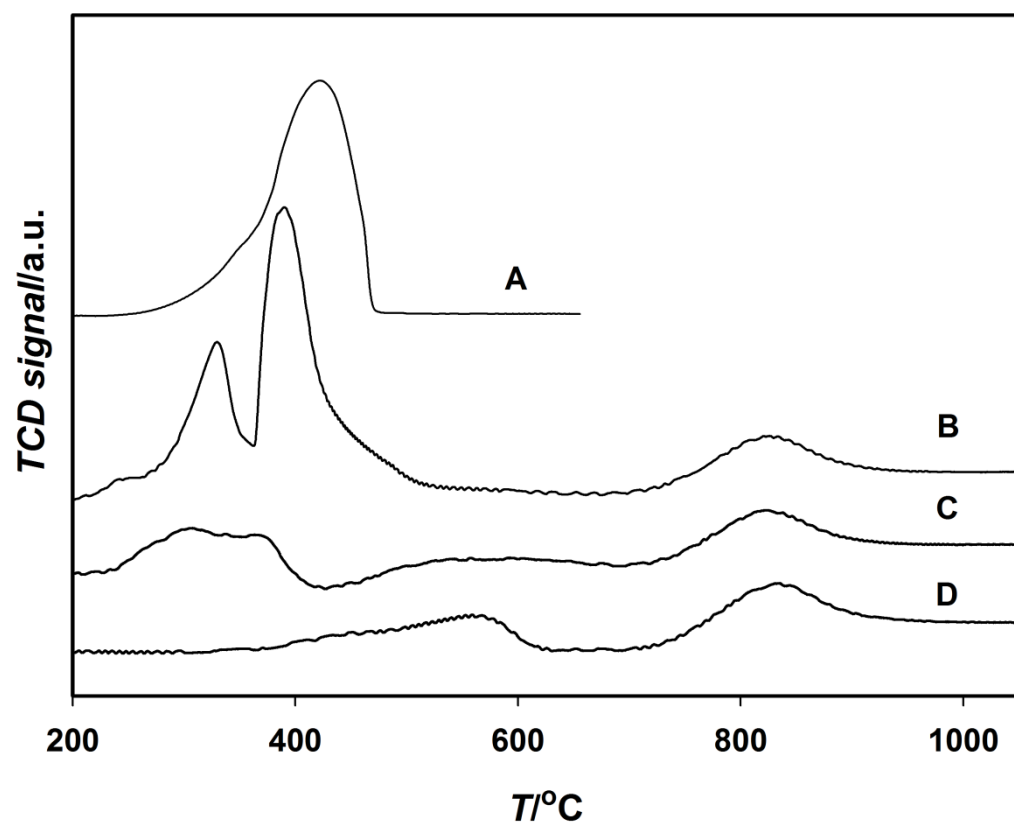


Figure 3

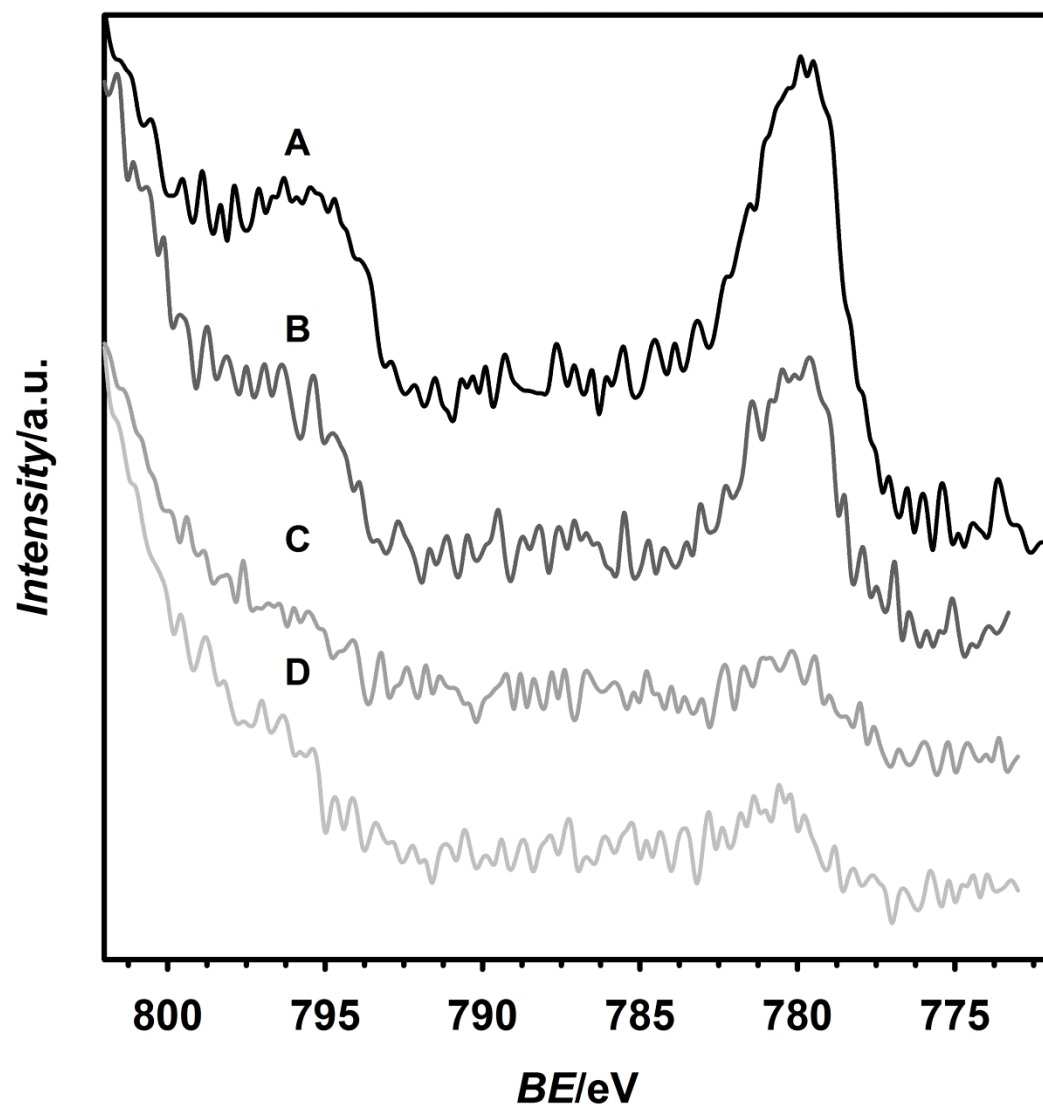


Figure 4

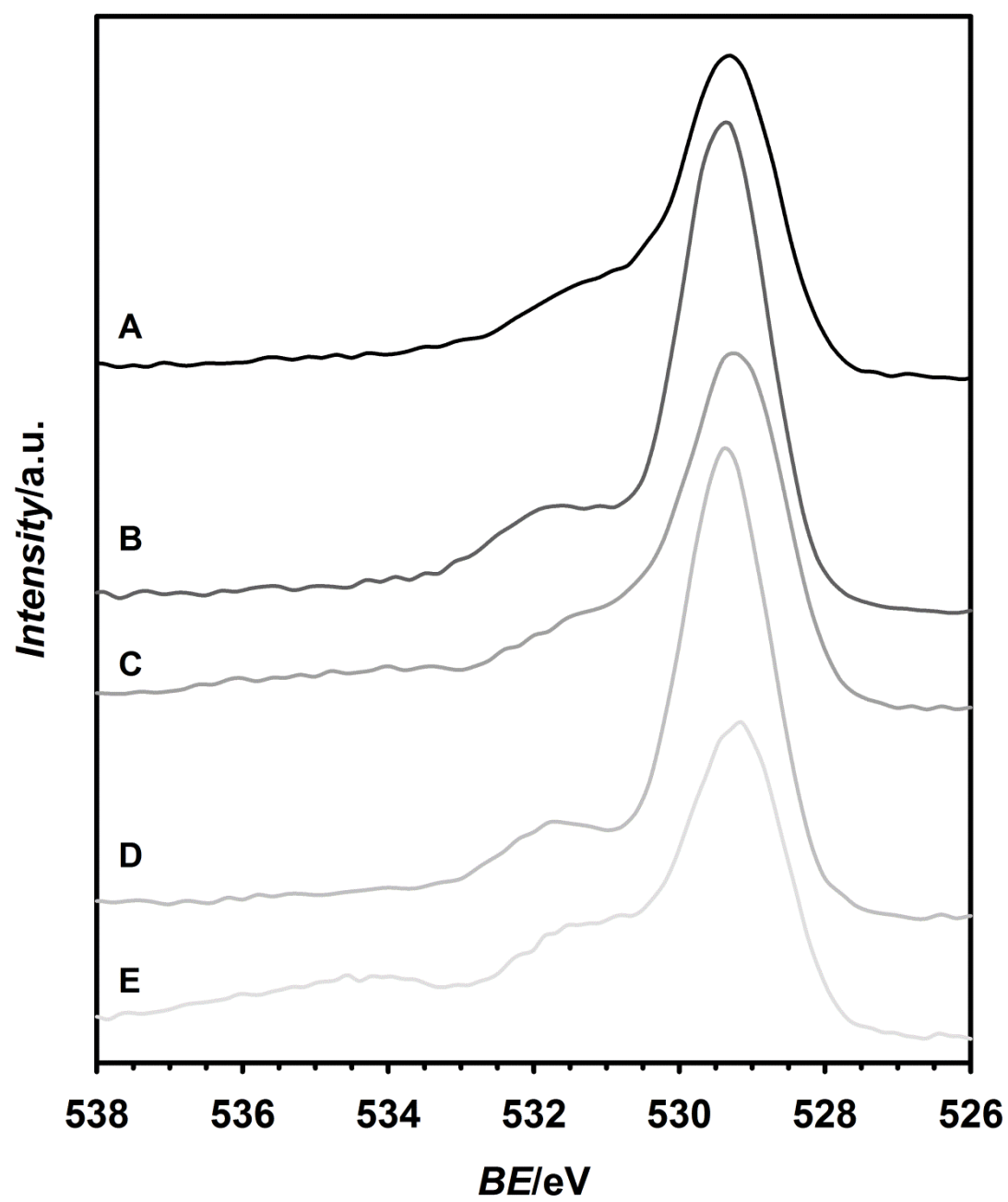


Figure 5

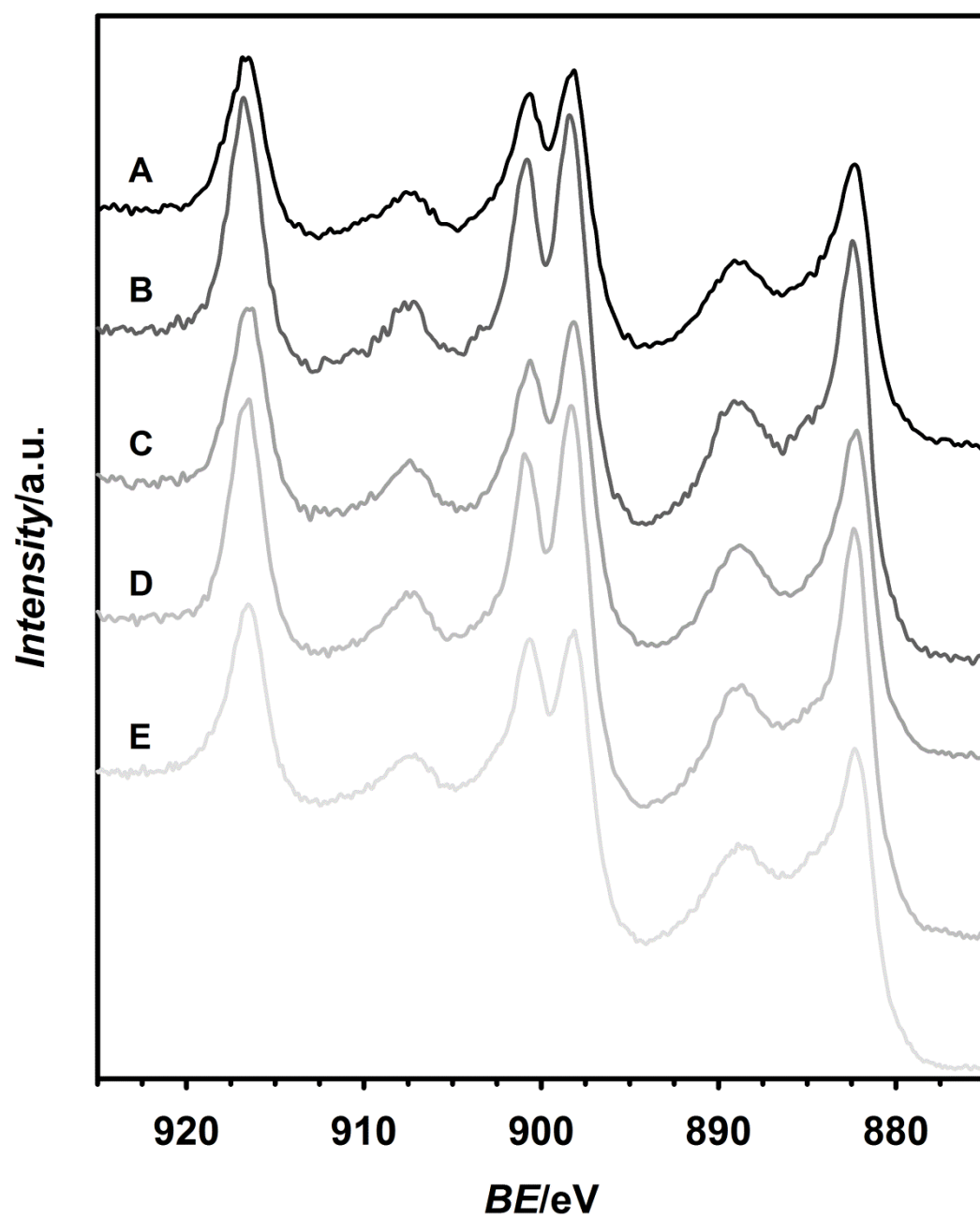


Figure 6



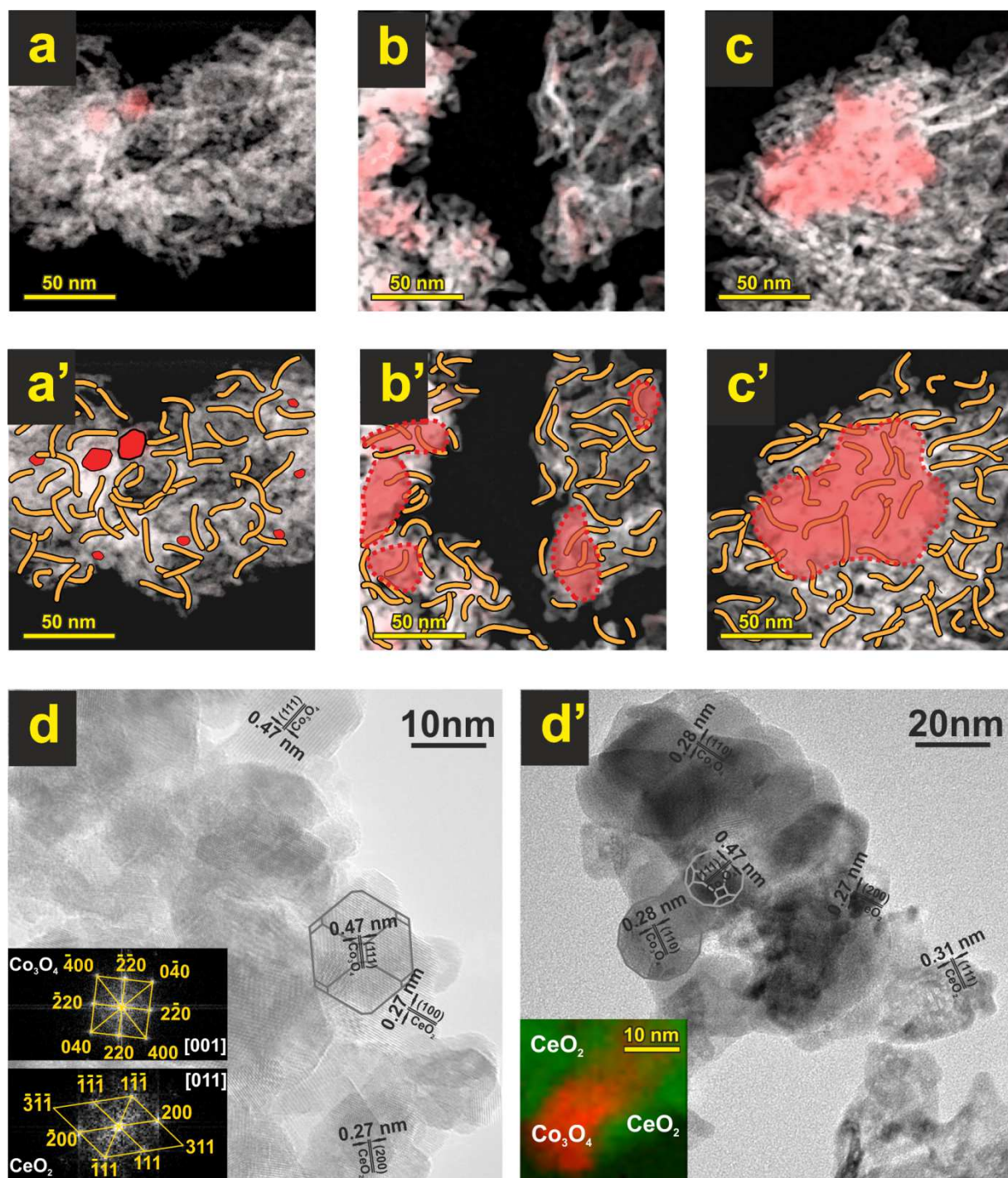


Figure 7

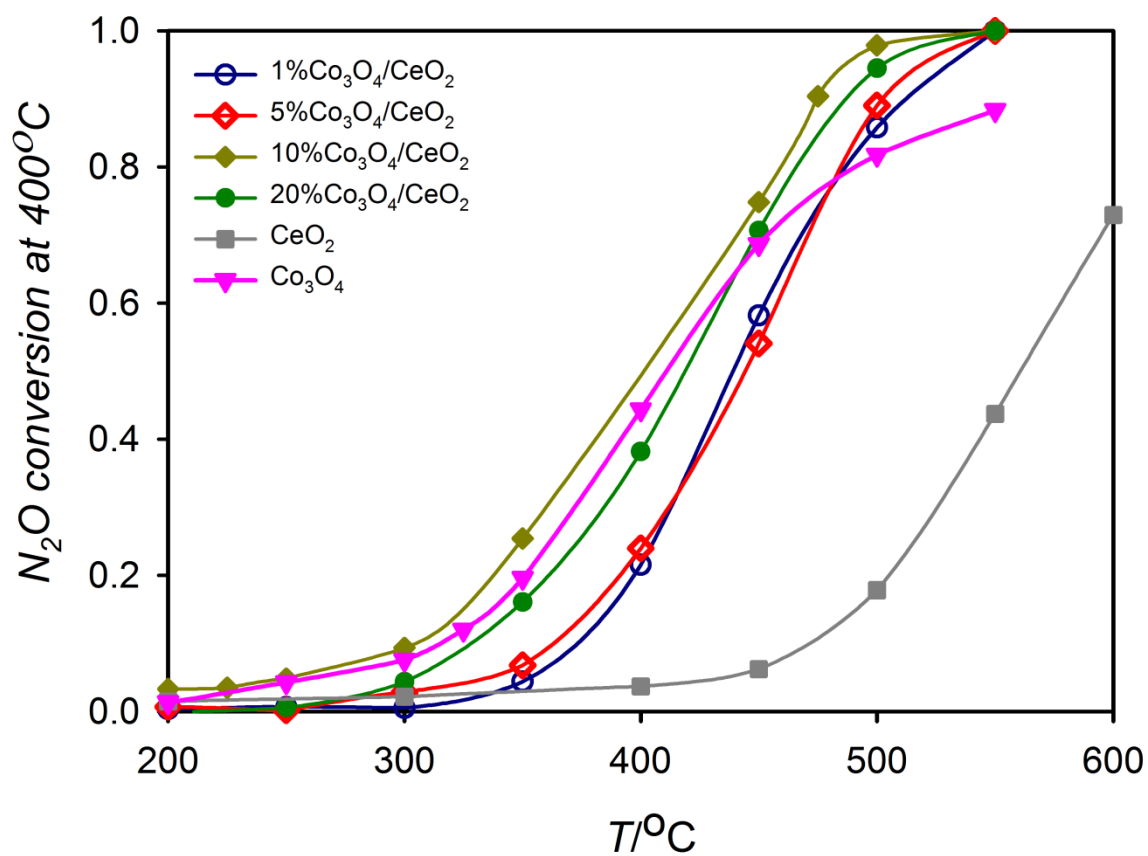


Figure 8

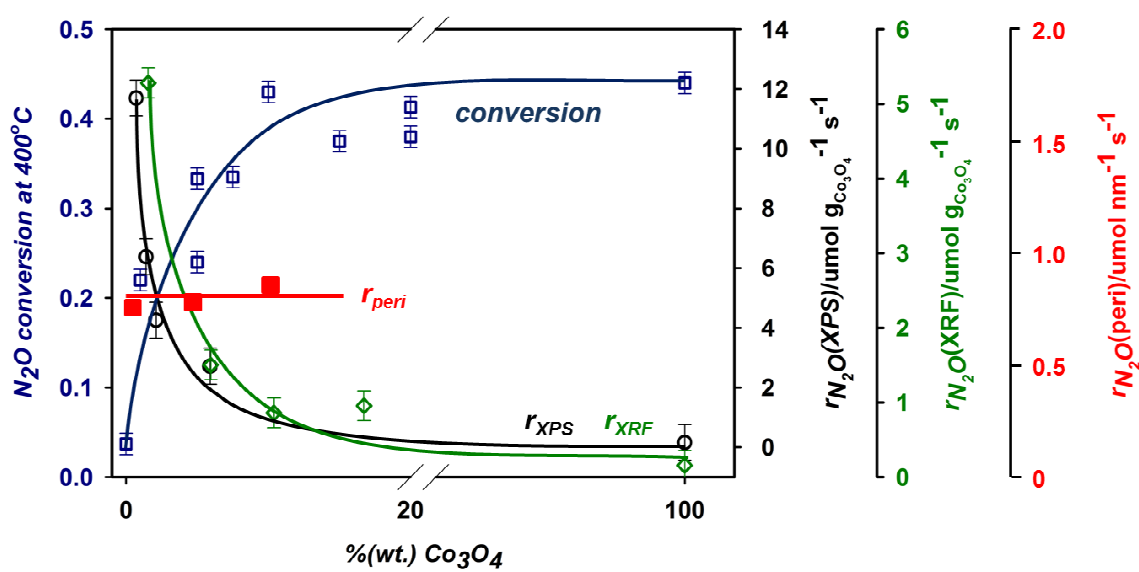


Figure 9

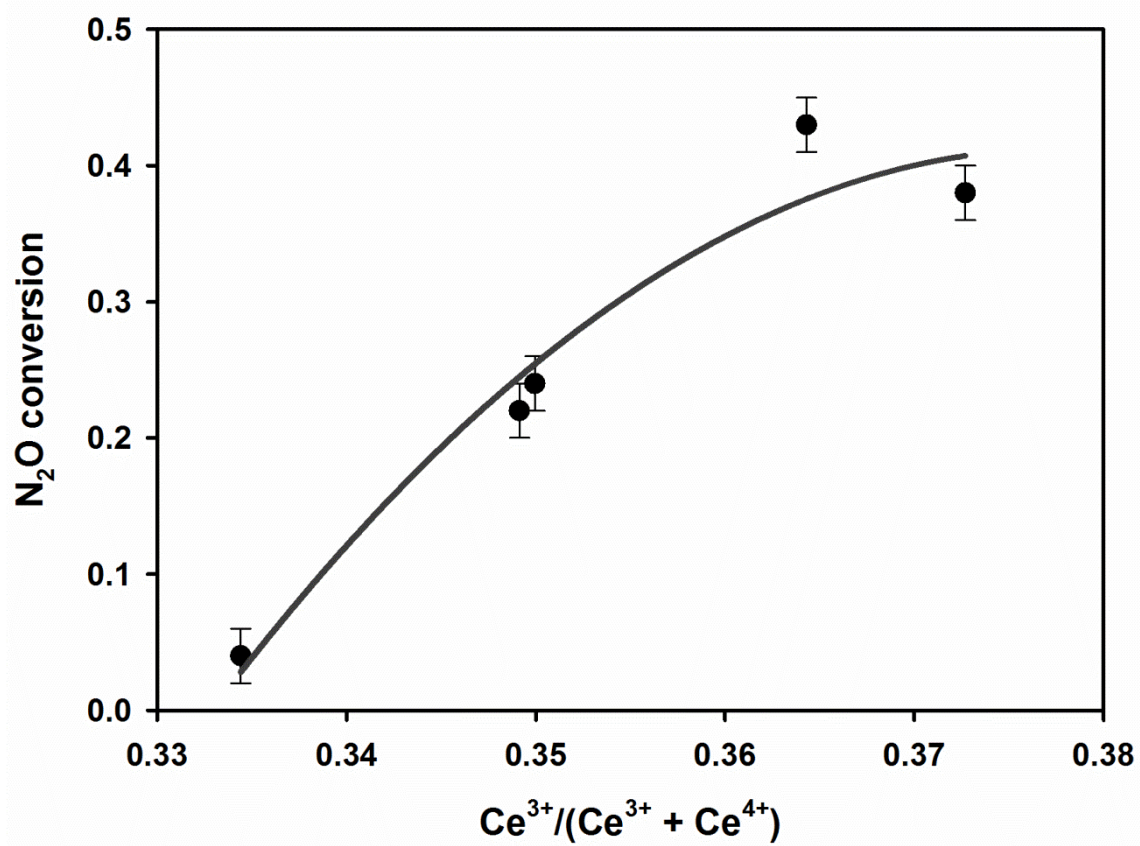


Figure 10

## References

**Table 1** Comparison of the surface (XPS) and bulk (XRF) content of cobalt oxide and the relative amount of  $\text{Ce}^{3+}$  in series of ceria supported cobalt catalysts. (\*) theoretical and (#) experimental.

**Table 2** XPS peak positions in the Ce 3d energy range obtained for the supported oxide samples and  $\text{CeO}_2$ .

**Table 3** Kinetic parameters for  $\text{N}_2\text{O}$  decomposition over  $\text{CeO}_2$  supported  $\text{Co}_3\text{O}_4$  catalysts.

## Table 1

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