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Graphene Triggered Enhancement in Visible-Light Active Photocatalysis as Well As in Energy Storage Capacity of $(\text{CFO})_{1-x}(\text{GNPs})_x$ Nanocomposites

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Abstract:

Cobalt ferrite-graphene nanoplatelets ($(\text{CFO})_{1-x}(\text{GNPs})_x$) nanocomposites are promising for efficient photocatalysis and high-performance supercapacitors. Multifunctional $(\text{CFO})_{1-x}(\text{GNPs})_x$ nanocomposites prepared via facile chemical method have been investigated for their physio-chemical characteristics like crystal structure, morphology, chemical composition, optical properties, infrared vibrational modes, photocatalytic and supercapacitor applications. Interestingly, the photocatalytic activity of CFO nanostructures has been improved significantly from 38.3 % to 98.7 % with the addition of graphene which can be attributed to control over recombination of charge carriers. It is also found that the specific capacitance of the prepared $(\text{CFO})_{1-x}(\text{GNPs})_x$ nanocomposite electrode at 0.5Ag^{-1} is three times higher than that of only CFO based electrode which could be due to the conducting nature of graphene nanoplatelets (GNPs). The enhanced photocatalytic and improved electrochemical characteristics suggest the effective use of prepared nanocomposites in water purification and supercapacitor nanodevices.

Keywords: Nanocomposites; electron-hole recombination; photocatalysis; supercapacitors

1. Introduction

The organic contaminants in wastewater pose a great threat to the human wellbeing and environmental safety, due to their highly toxic nature. Therefore, the development of efficient technologies for fast water purification is need of the day [1]. Among various technologies, the

visible light driven photocatalysis is the most promising and cost-effective technique for water purification. Over the past few years, the ferrite nanostructures have been used as photocatalyst because of their excellent visible light response and easy magnetic separation [2]. However, these magnetic nanostructures have a great challenge to be used in various applications due to fast charge recombination, poor dye adsorption, and agglomeration [3]. In order to address these issues, the addition of carbon derivatives such as graphene as supportive constituents to the ferrites could be a possible solution [4]. Apart from the high conductivity and suitable work function of graphene, the strong ability of graphene to accept electrons from ferrites plays a significant role to enhance the photocatalytic performance. Moreover, the graphene can also prevent the agglomeration of magnetic ferrites due to its nonmagnetic nature and availability of large surface area for interfacing [5]. Various ferrites composited with different forms of graphene (~~e.g. graphene oxide (GO) and reduced graphene oxide (rGO)~~), such as reduced graphene oxide, graphene oxide etc. have been ~~extensively explored~~ widely investigated and employed ~~as photocatalyst for photocatalysis~~ because of their high photocatalytic activities [6-10]. However, ~~the oxygen-containing functional groups~~ oxygen moieties present in ~~GO or rGO~~ rGO or GO act as defect states and lower the conductivity of graphene. These oxygen based defects lead towards ~~ow~~ low photocatalytic efficiency [11]. Therefore, pure graphene would be a better choice to boost the performance of the graphene based photocatalytic nanocomposites.

The need of cost effective and eco-friendly storage devices of renewable energy is one of the major challenge in the energy sector. ~~Supercapacitors are considered to be promising for energy storage; because of their~~ Supercapacitors are thought to be potential devices for the storage of electrical energy; due to their rapid charge discharge efficiency and long cyclic performance [12, 13]. These are highly desired in hybrid electric vehicles, mobile electronic devices and smart electricity grids [14]. Due to good pseudo capacitance and better thermal stability, the ferrites have been used traditionally as electrode materials in energy devices. However, their high resistivity and agglomeration leads towards poor rate performance with low cycling stability which limit their use in storage devices [15]. The carbon-based nanomaterials such as graphene has longer cyclic life and more rapid charge discharge efficiency. Therefore, the addition of conductive graphene might be the better solution to overcome the limitations of ferrites [16-19].

This work has been focused on the enhancement of visible light driven photocatalytic response and improvement of electrochemical properties of $(\text{CFO})_{1-x}(\text{GNPs})_x$ nanocomposites. The study presents the degradation of methylene blue (MB) dye found in textile wastewater using $(\text{CFO})_{1-x}(\text{GNPs})_x$ nanocomposites under visible light irradiation. The electrochemical performance of the prepared $(\text{CFO})_{1-x}(\text{GNPs})_x$ nanocomposites has also been investigated under different conditions.

2. Experimental

2.1. Preparation of CFO and $(\text{CFO})_{1-x}(\text{GNPs})_x$ nanocomposites

Optimized parameters have been used for synthesis of CFO nanostructures and $(\text{CFO})_{1-x}(\text{GNPs})_x$ nanocomposites. In typical procedure, $\text{FeCl}_3 \cdot 6\text{H}_2\text{O}$ (0.4M) and $\text{CoCl}_2 \cdot 6\text{H}_2\text{O}$ (0.2M) were dissolved in 100 mL distilled water. Then cetrimonium bromide (CTAB) (0.1 g) ~~was added to the mixture and sonicated~~ was dissolved in the mixed solution under ultra-sonication for 10 minutes. NaOH (6 M) solution was prepared in 50 mL distilled water in a separate beaker and heated at 80°C. The mixed aqueous solution of $\text{FeCl}_3 \cdot 6\text{H}_2\text{O}$ (0.4M) and $\text{CoCl}_2 \cdot 6\text{H}_2\text{O}$ (0.2M) ~~mixture solution was added dropwise~~ was added dropwise to the stock solution of NaOH and CTAB ~~stock solution. The mixture was.~~ The reaction mixture was kept at optimized 80°C for two hours under constant stirring. ~~The precipitates were collected after washing many times with ethanol and distilled water.~~ The precipitates were collected after washing with distilled water and ethanol several times and then dried at 100°C for 24 hours to get powdered sample [20]. $(\text{CFO})_{1-x}(\text{GNPs})_x$ nanocomposites with different GNPs content (25, 50, 75 wt. %) were synthesized. A typical experimental procedure for the preparation of $(\text{CFO})_{0.75}(\text{GNPs})_{0.25}$ (25 wt. % GNPs) nanocomposites is given as follows: 25wt.% of GNPs were mixed with CFO nanostructures and the mixture was dispersed in 30 mL ethanol. The suspension was then sonicated for 2 hours in ultrasonic bath ~~and dried at 60°C for 12 hours~~ and dried for 12 hours at 60°C ~~in an oven~~. The obtained sample was further annealed at 500°C for 2 hours in N_2 atmosphere to obtain $(\text{CFO})_{1-x}(\text{GNPs})_x$ nanocomposite [21]. Same procedure has been adapted for fabrication of other nanocomposites with desired compositions.

2.2. Characterizations

XRD data was obtained ~~using with the help of~~ a Malvern PANalytical X'pert PRO MPD Cu K α ($\lambda = 1.5406 \text{ \AA}$) ~~radiation source~~. Surface morphology of the samples was studied using TEM (Thermo Fisher Tecnai G2 F20). FTIR spectra were recorded using FTIR spectrometer (Tensor 27 FT-IR Spectrometer (Bruker)). Optical absorption spectroscopy of the samples was carried out by UV-visible spectrophotometer (UV-3600 spectrophotometer, Shimadzu Co., Japan). ~~Room temperature XPS measurements characterizations~~ were ~~performed~~ ~~carried out at room temperature~~ ~~with a~~ using SPECS PHOIBOS 150 analyzer (SPECS GmbH, Berlin, Germany) at 5×10^{-10} mbar base pressure, using monochromatic Al K α excitation source (1486.74 eV). The PL spectra were collected using Cary Eclipse Fluorescence Spectrophotometer (Agilent).

2.3. Measurements of photocatalytic ~~activity~~ performance

All the ~~photocatalytic experiments~~ the experiments for measuring degradation performance have been ~~carried out~~ performed under visible light ~~illumination~~ irradiation at room temperature. To evaluate the photocatalytic performance, a 100 mL of 20 mg L⁻¹ of dye aqueous solution was taken from the stock solution with addition of 25 mg photocatalyst. 1 mL of H₂O₂ was added to this solution. Adsorption-desorption equilibrium between the catalyst and dye was achieved by stirring the reaction mixture for 60 min in dark. A xenon lamp (500 W) was used as a light source to irradiate the samples. About 4 mL of the test solution was taken at regular time intervals. The absorbance of MB during the degradation process was measured using UV-visible spectrophotometer.

2.4. Electrochemical measurements

Electrochemical ~~performances~~ characteristics of the prepared ~~electrode materials~~ electrodes have been tested on Biologic VMP3 potentiostat using cyclic voltammetry (CV) and galvanostatic charge-discharge (GCD) and electrochemical impedance spectroscopy (EIS) in 0.5 M K₂SO₄ aqueous ~~electrolyte~~ solution as an electrolyte. The working electrodes were fabricated by mixing 80% active material with 10% PVDF and 10% carbon super-P in N-methyl-2-pyrrolidone to get uniform ~~slurry~~ paste. Finally, the ~~slurry~~ paste was applied to aluminum foil by ~~doctor blade method~~ and was dried in ~~vacuum~~ oven for 24 h at 80°C ~~under vacuum conditions~~. The electrochemical properties were measured with standard two-electrode

symmetric cell in a 3-way Teflon Swagelok cell using identical electrodes with glass microfiber separator.

3. Results and discussions

3.1. Structural analysis

X-ray powder diffractograms of pristine CFO and $(\text{CFO})_{1-x}(\text{GNPs})_x$ composites are shown in Fig.1. The characteristic peaks in the diffraction patterns of CFO and $(\text{CFO})_{1-x}(\text{GNPs})_x$ composites are well indexed to the (220), (311), (400), (511), (440) and (533) planes of cubic spinel phase crystal structure of CFO as reported in JCPDS No. 22-1086 [22]. The relatively high intensities of diffraction peaks clearly depict that prepared CFO sample is highly crystalline. There is no XRD peaks related to impurities or other extra phases in the diffraction pattern. These results demonstrate that the prepared CFO and $(\text{CFO})_{1-x}(\text{GNPs})_x$ composites samples are in their pure phases. Furthermore, the peak at $2\theta = 26.3^\circ$ in $(\text{CFO})_{1-x}(\text{GNPs})_x$ samples corresponds to (002) characteristic plane of GNPs which confirms the successful formation of the desired composites. It is also observed that there is slight shift in diffraction peaks with systematic change in composition of composites, which is assigned to the interfacing of both constituents and the formation of structural defects.

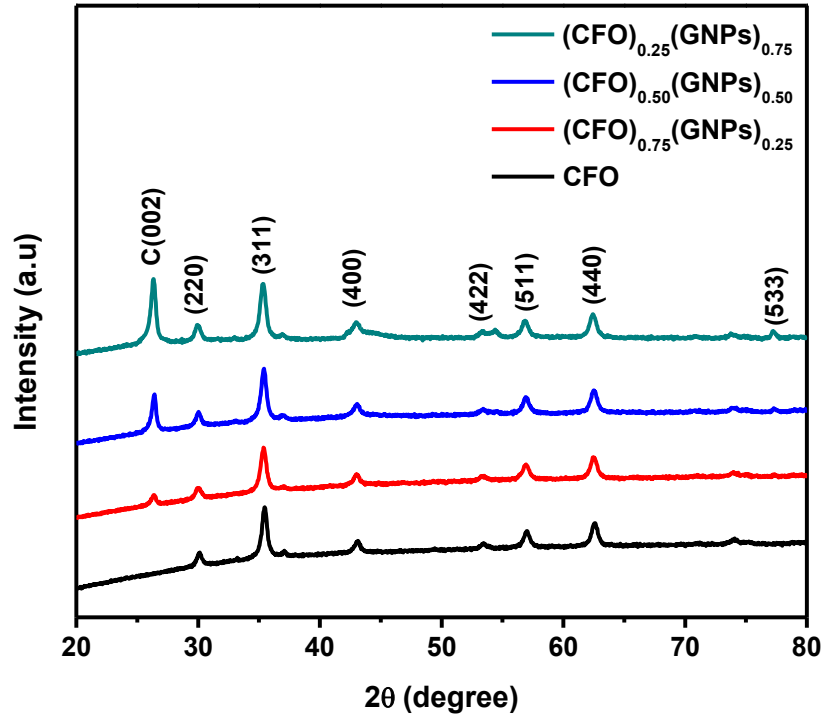


Fig.1: XRD patterns of (CFO)_{1-x}-(GNPs)_x samples

3.2.Morphological study

The morphology of prepared CFO and (CFO)_{1-x}(GNPs)_x samples has been examined using TEM. In pristine CFO sample (Fig.2 (a)), the shape of the nanostructures is quasi cubic with an average size in the range of 10 nm-31 nm. Fig. 2(b-d) depicts uniform attachment of CFO nanostructures with wrinkled GNPs which confirms the formation of the composite with well-defined interfacing. It is interesting to note that the agglomeration of CFO nanostructures has been reduced significantly by incorporation of GNPs which might be attributed to nonmagnetic nature and high surface area of GNPs. The HRTEM microstructural studies of nanostructures further confirms the high crystallinity of the CFO nanostructures as given in inset of Fig. 2(a). The fringes with lattice spacing of 0.297 nm are assigned to (220) plan of CFO [23]. The observation obtained from HRTEM is also consistent with XRD results shown in Fig.1.

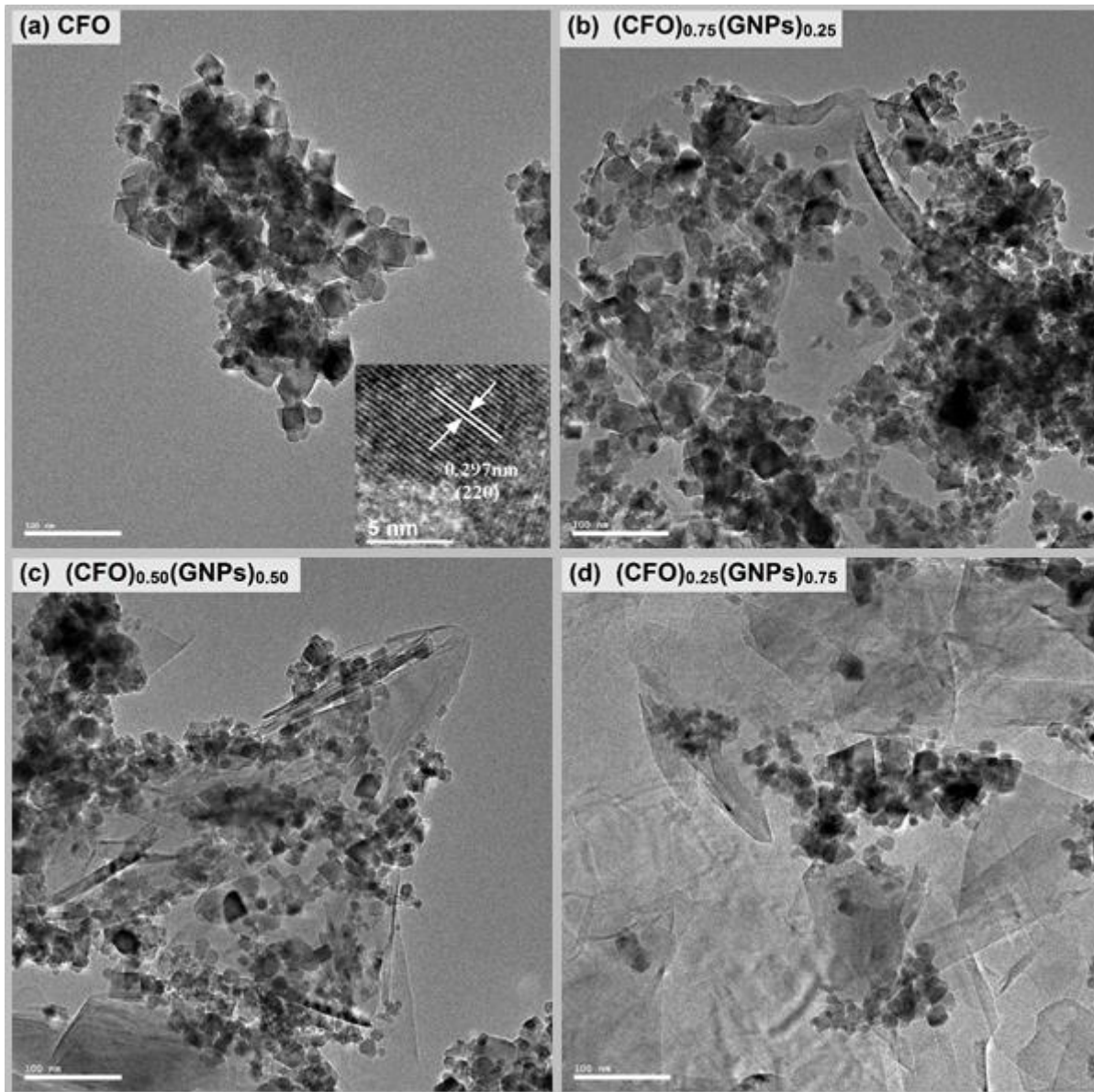


Fig.2: TEM micrographs of $(\text{CFO})_{1-x}(\text{GNPs})_x$ samples (inset: HRTEM image of CFO)

3.3.Vibrational Study

FTIR spectroscopy has been used to explore the chemical structure of pure CFO nanostructures and $(\text{CFO})_{1-x}(\text{GNPs})_x$ nanocomposites as presented in Fig.3. The bands at 431cm^{-1}

and 560cm^{-1} are attributed to Co-O and Fe-O symmetrical stretching vibrations respectively [24]. The band at 1639 cm^{-1} belongs to C=C skeletal vibrations of GNPs and the band at 3400 cm^{-1} ~~is assigned~~ originates from O-H stretching vibrations [25, 26]. There are no ~~oxygen containing functional groups~~ oxygen moieties in the FTIR spectrum of GNPs and $(\text{CFO})_{1-x}(\text{GNPs})_x$ nanocomposite samples which confirm the purity of nanocomposites. This study further verifies the structural analysis of XRD.

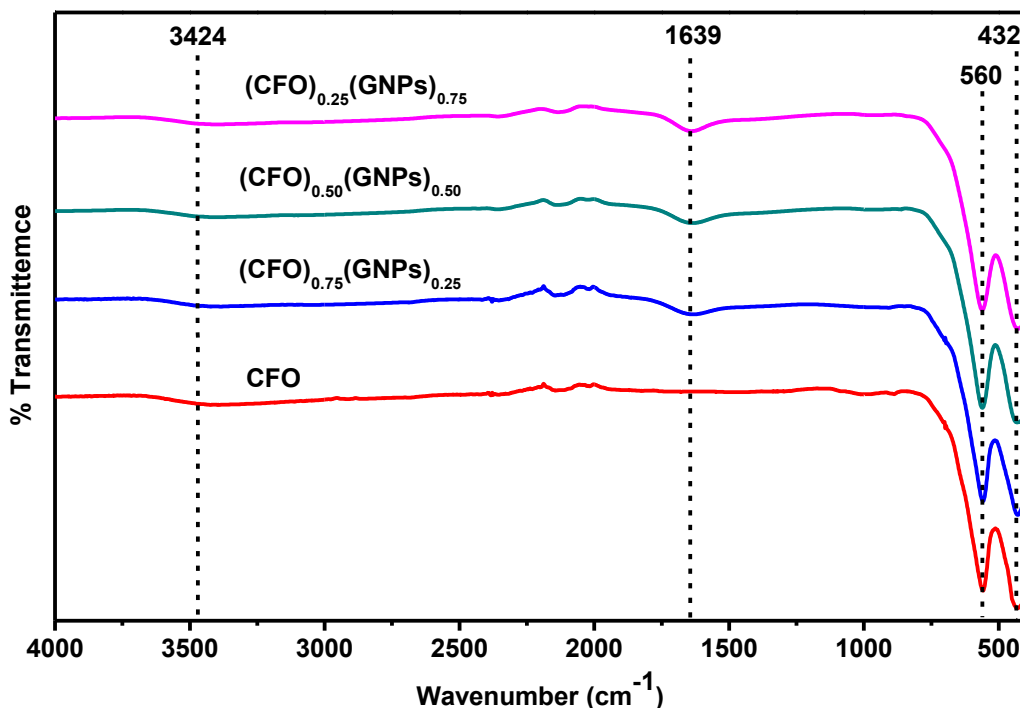


Fig.3: FTIR spectra of $(\text{CFO})_{1-x}(\text{GNPs})_x$ nanocomposites

3.4.XPS analysis

The valence oxidation state of different elements in materials plays an important role in the change of the basic science of host composites leading towards the formation of defects and impurities which may be responsible for control over electron hole recombination rates. The oxidation state and chemical compositions of elements in $(\text{CFO})_{1-x}(\text{GNPs})_x$ nanocomposites have been checked using XPS spectroscopy at room temperature. The spectra of $(\text{CFO})_{1-x}(\text{GNPs})_x$ nanocomposites have been carried out and typical results for $(\text{CFO})_{1-x}(\text{GNPs})_x$ [$x = 0, 0.25$] are given in Fig 5(a). The two distinct peaks of Co at 780.8 eV and 796.3 eV in the XPS spectrum of CFO are ~~assigned~~ attributed to Co $2p_{3/2}$ and Co $2p_{1/2}$, respectively [27]. The corresponding satellite peaks

located at binding energies 786.6 eV and 803.5 eV. The difference in binding, shape and positions of Co 2p_{3/2} and Co 2p_{1/2} in spectra with existence of satellite peaks depict that Co is 2+ oxidation state in all prepared samples [28]. However, there is slight shift in positions of both Co 2p_{3/2} and Co 2p_{1/2} peaks with GNPs addition incorporation towards the lower energy which is assigned to the formation of interfacial defect, associated with GNPs concentration.

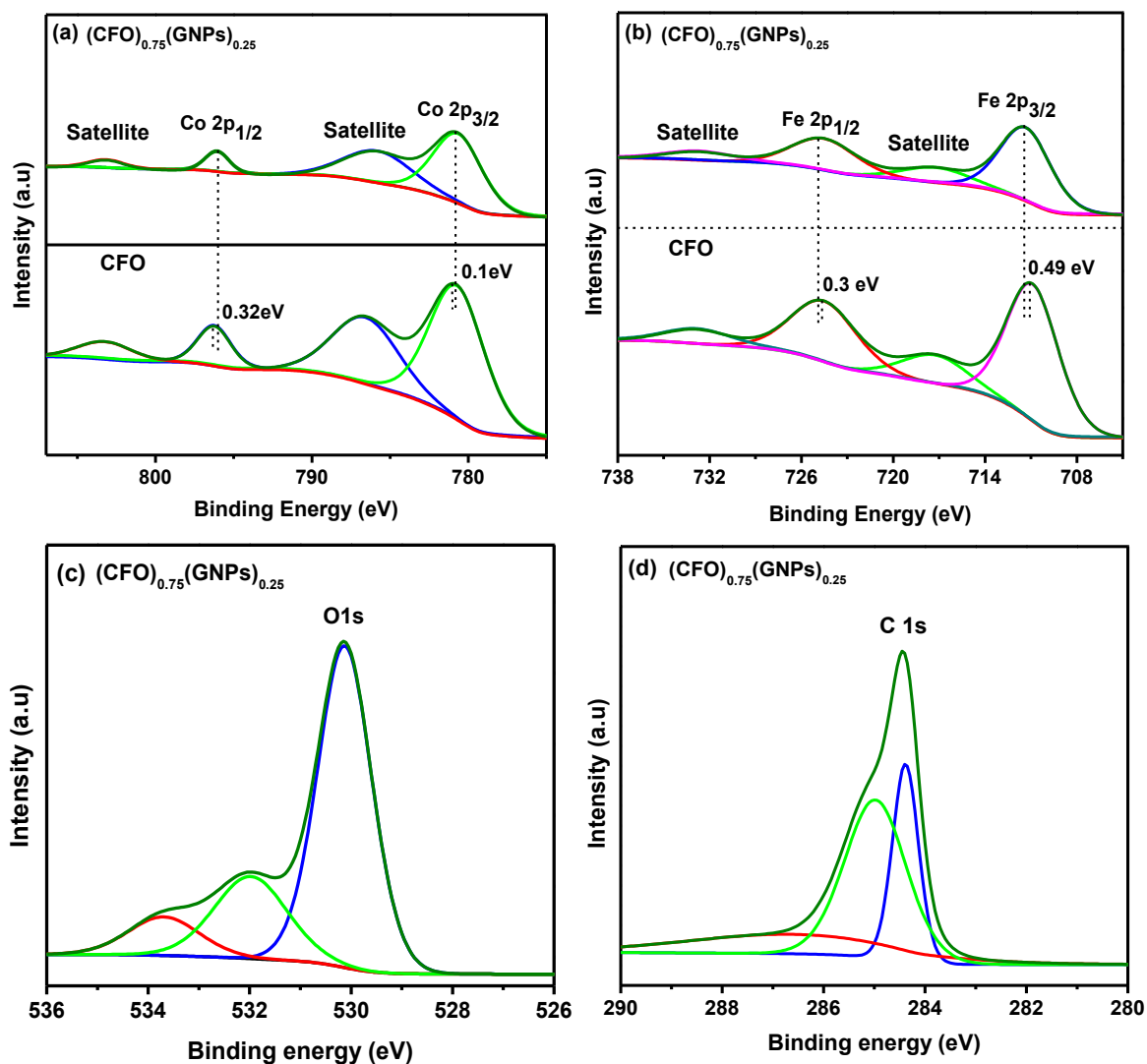


Fig. 4: XPS wide angle spectra for (CFO)_{1-x}(GNPs)_x nanocomposites.

The peaks related to Fe2p_{3/2} and Fe2p_{1/2} are centered at 710.9 eV and 724.7 eV respectively, as shown in Fig 5(b). Two shake up satellite peaks at 718.8 eV and 733.4 eV confirm Fe³⁺ oxidation state of Fe[29]. The peaks corresponding related to Fe 2p_{3/2} and Fe 2p_{1/2} in (CFO)_{1-x}

$x(\text{GNPs})_x$ nanocomposite (25 wt. %) are shifted by 0.49eV and 0.3eV respectively, towards the higher energy. The slight shift in Fe 2p and Co 2p indicates the bonding and electronic interaction between GNPs sheets and CFO nanostructures [30]. Three deconvoluted peaks in wide angle XPS of O1s in $(\text{CFO})_{1-x}(\text{GNPs})_x$ (25 wt.%), are depicted in Fig.5(c). The maxima in spectra at 530.1, 531.9 and 533.7 eV is attributed to metal-oxygen bond (M–O–M), hydroxyl groups (M–O–H) adsorbed on the surface of the sample adsorbed on the sample surface and the surface-adsorbed oxygen respectively [31]. The M–O–M peaks are slightly shifted toward lower energy in $(\text{CFO})_{0.75}(\text{GNPs})_{0.25}$ nanocomposites compared to that in CFO, which possibly is because of the formation of Fe–O–C and Co–O–C bonds between the CFO nanostructures and GNPs. This indicates attachment between CFO and the GNPs. ~~The two resolved peaks in the C1s spectrum at 284.39 eV and 285 eV given in Fig.5 (d), can be assigned attributed to C–C GNPs aromatic rings and C–O bonds respectively [32].~~ The two resolved peaks at 284.39 and 285 eV in the C1s spectrum show the presence of C=C (sp^2) and C–C (sp^3) bonds, respectively [Reference: La, D.D., et al., Fabrication of a GNP/Fe–Mg binary oxide composite for effective removal of arsenic from aqueous solution. ACS Omega, 2017. 2(1): p. 218-226.]

3.5.Optical Characteristics

The optical properties of $(\text{CFO})_{1-x}(\text{GNPs})_x$ nanocomposites have been studied by comparing their UV-vis spectra as shown in Fig.5(a). It has been observed that pristine CFO as well the nanocomposites have shown efficient absorption in broad wavelength range including both UV and visible spectrum of light. The band gap energies for CFO nanostructures and $(\text{CFO})_{1-x}(\text{GNPs})_x$ nanocomposite have been calculated using well known Tauc's relation [33].

$$(\alpha hv)^n = A(hv - E_g) \quad (1)$$

Here α , hv , n , A , and E_g are absorption coefficient, photon energy, integer, absorption constant coefficient and band gap energy respectively. The band gap is obtained using $n = 2$ for allowed direct transitions. The calculated band gap values from experimental spectra are 1.95 eV, 1.83 eV, 1.68 eV and 3.05eV for CFO, $(\text{CFO})_{0.75}(\text{GNPs})_{0.25}$, $(\text{CFO})_{0.50}(\text{GNPs})_{0.50}$ and $(\text{CFO})_{0.25}(\text{GNPs})_{0.75}$ samples as given in Fig. 4(b). It is interesting to note that the band gap is significantly tuned towards visible range with the increasing of GNPs content up to a certain extent i.e., 50 wt.% which is indication of sample's application for visible light triggered

photodegradation. This change in band gap as function of GNPs concentration is attributed to the hybridization between GNPs and CFO nanostructures [34]. The hybridization between GNPs and CFO nanostructures in the composite structures gives rise to the ~~formation of~~ strong Fe-O-C bonds. This chemical interaction introduces additional energy ~~levels~~ states between conduction and valence band of the composite material and as a result band gap energy reduces [35, 36]. However, it is also observed that there is an increase in bandgap energy for $(\text{CFO})_{0.25}(\text{GNPs})_{0.75}$ nanocomposites. This anomalies for higher concentration is due to Burstein-Moss effect arising from the high surface interactions of GNPs with CFO as compared with less concentration in other $(\text{CFO})_{1-x}(\text{GNPs})_x$ nanocomposites. This effect occurs, when electronic states at the bottom of conduction band of a semiconductor are fully populated. In this case, the electrons could flow from graphene into the conduction band of CFO, due to the difference in their Fermi levels. At higher graphene ratio, would allow more π -electrons to enter the conduction band of CFO nanostructures. As a result some states close to the conduction band of CFO are populated, and the Fermi level goes up into the conduction band, which causes blue shift in the band gap [37, 38].

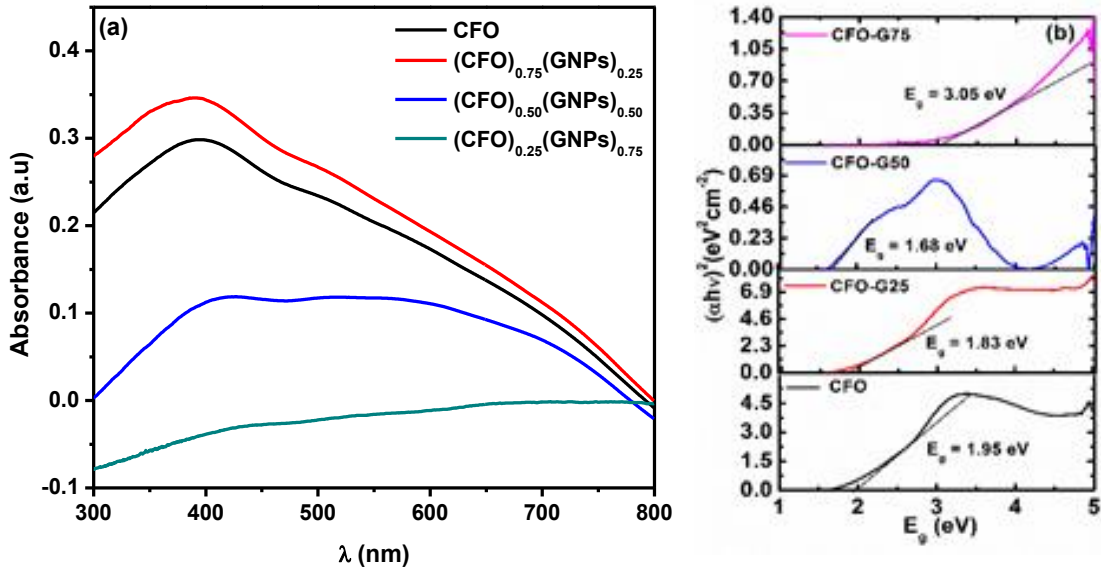


Fig.5: (a)UV-vis absorbance spectra of $(\text{CFO})_{1-x}(\text{GNPs})_x$ nanocomposites, (b) Tauc's plots of $(\text{CFO})_{1-x}(\text{GNPs})_x$ nanocomposites.

3.6. Luminescence properties

The room temperature photoluminescence (PL) spectra of $(\text{CFO})_{1-x}(\text{GNPs})_x$ nanocomposites is shown in Fig.6. It has been found that the PL intensity of nanocomposites is decreased with GNPs addition which suggests that the recombination of photo-generated e^-/h^+ is sufficiently prevented as compared to that in pristine CFO nanostructures [39]. The electrons accepting ability and conductivity of $(\text{CFO})_{1-x}(\text{GNPs})_x$ nanocomposites is higher than the CFO nanostructures, which is due to high electron affinity of graphene to delay e^-/h^+ recombination in nanocomposites. This behavior is very significant and useful for the application of CFO based nanocomposites in photocatalysis.

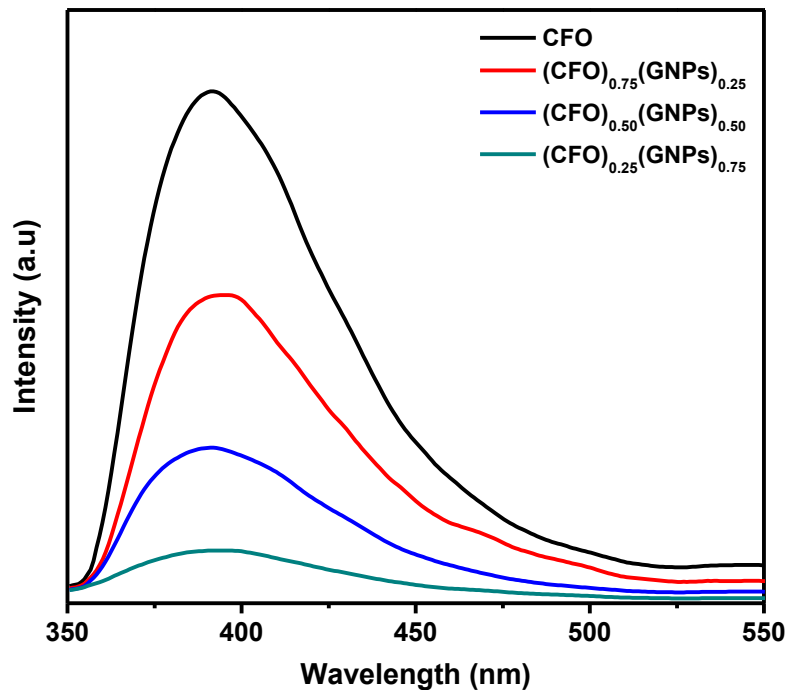


Fig. 6: PL spectra of $(\text{CFO})_{1-x}(\text{GNPs})_x$ nanocomposites

4. Photocatalysis

The degradation of organic pollutant particularly presence of methylene blue (MB) in textile waste via cost effective solar light triggered nanocomposites catalyst is one of recent demand in material science. Fig. 7(a)-(e) shows absorbance spectra of methylene blue (MB) in presence of $(\text{CFO})_{1-x}(\text{GNPs})_x$ ($x = 0, 0.25, 0.50, 0.75, 1$) as a function of time. The adsorption-

desorption equilibrium between the catalyst and the dye has been achieved first ($t = 0$ min). It is clear from the Fig.7 (a)-(e) that concentration of the dye (C/C_0) decreases with time for all samples. Fig.7(f) represents photo-degradation rates of GNPs, CFO and CoFe₂O₄-GNPs nanocomposites in the presence of visible irradiation at 25°C. It is observed that the degradation efficiencies of individual GNPs and CFO nanostructures are 23.4 % and 38.3 % respectively in 120 min under visible light, which show very slow rate with less degradation process for MB pollutant. It has been interestingly found that the prepared nanocomposites (CFO)_{0.75}(GNPs)_{0.25}, (CFO)_{0.50}(GNPs)_{0.50}, (CFO)_{0.25}(GNPs)_{0.75} have 98.7 %, 93.2% and 82.1% photocatalytic degradation under same condition respectively, which is almost three times then the individual CFO nanostructures degradation. The understanding of very high degradation rate for particular 25 wt.% GNPs composition while decrease for other 50% and 75 % GNPs compositions with CFO can be attributed to the shielding effect caused by GNPs [40]. When the content of graphene is increased beyond certain limit, the active sites in the photocatalyst are covered by graphene nanoplates. ~~Due to~~ because of the limited access of visible light to the active sites of the photocatalyst, the photocatalytic reaction slows down, which causes to degrade the photocatalytic performance. The photocatalytic degradation mechanism of MB in these prepared nanocomposites can be explained using Photo-Fenton Reaction (PFR) process [41]. In this process, Fe³⁺/Fe²⁺ ion pairs are mutually converted through redox reactions occurring at the photocatalyst surface. During the reversible redox reaction process, active sites (Fe²⁺) on the photocatalyst surface decompose H₂O₂ into reactive oxygen species (ROS) i.e. hydroxyl radicals (•OH) and hydroperoxyl radicals (•OOH). These ROS then play an important role to maximum oxidize the hazardous MB molecules to form nontoxic CO₂ and H₂O in final products [42]. Here graphene plays key role in accelerating PFR by preventing recombination and facilitating the electron transfer process [43, 44]. The main reasons for the superior visible light driven photocatalytic performance of (CFO)_{1-x}(GNPs)_x nanocomposites can be attributed to the delayed recombination of charge carriers [45], suitable band gap[46], availability of more active sites, large surface area and accelerated PFR as result of GNPs addition [47]. The recombination dynamics of the photocatalyst can be understood based on photocatalytic quantum yield ϕ , given by the relationship[48].

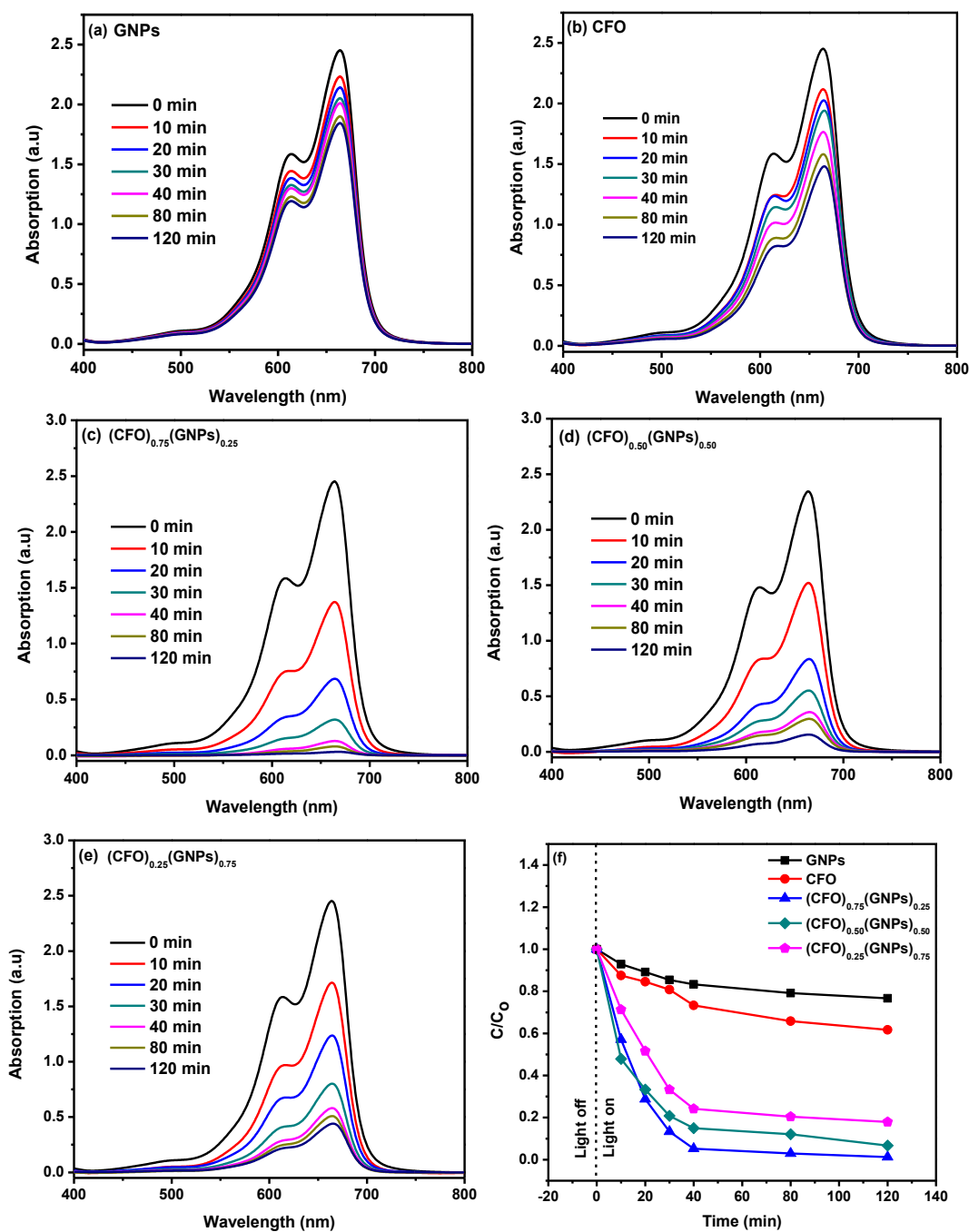


Fig. 7 (a)-(e) Absorption spectra of MB solution for $(\text{CFO})_{1-x}(\text{GNPs})_x$, and (f) photodegradation of $(\text{CFO})_{1-x}(\text{GNPs})_x$ nanocomposites

$$\varphi \propto \frac{k_{ct}}{k_{ct} + k_r} \quad (2)$$

Where, k_{ct} denotes the rate of the charge transfer and k_r is rate of recombination of the electron-hole in a material. The relation clearly indicates that preventing recombination of the electrons and holes in a material under irradiation will play a role for increasing the photocatalytic quantum yield of a material. Highly conducting and defects free GNPs support the charge transfer from CFO to GNPs which result in the decrease in the recombination rate of e^-/h^+ pair and enhance improve photocatalytic activity performance of the nanocomposite samples [49, 50]. The prepared nanocomposites with optimum ratio of 25% GNPs having 98.7% degradation for MB are potential for purifications of textile waste water.

5. Electrochemical Properties

5.1. Cyclic Voltmetry (CV)

In order to explore the electrochemical efficiency of electrode based on prepared nanocomposites, the CV curves of $(CFO)_{1-x}(GNPs)_x$ have been measured at room temperature using K_2SO_4 electrolytes with two electrodes symmetric cell at different scan rates and typical plots for $x = 0, 0.25, 1$ are shown in Fig.8 (a)-(c).

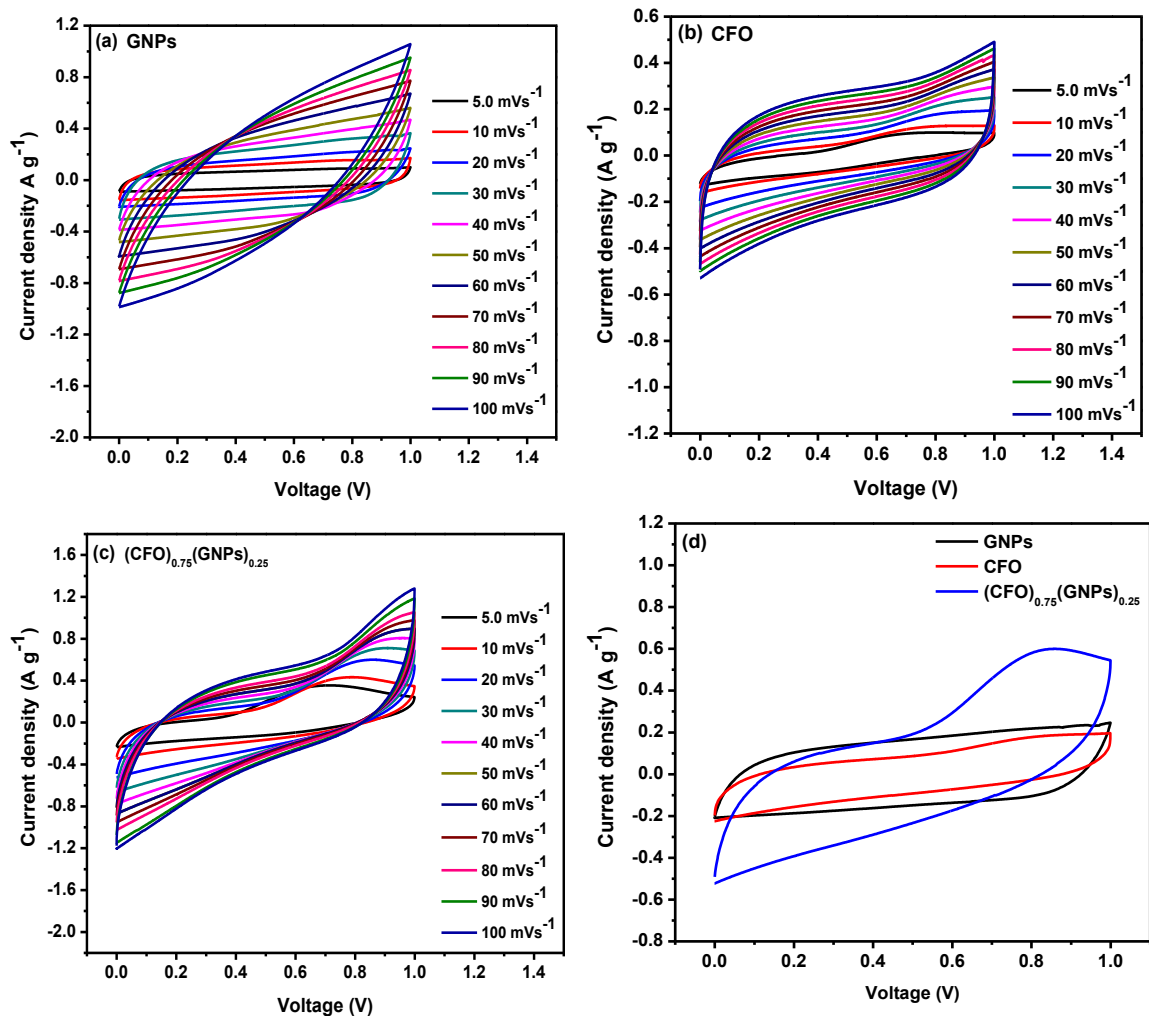


Fig. 8: (a)-(c) CV curves of the $(\text{CFO})_{1-x}(\text{GNPs})_x$ electrodes at various scan rates, (d) comparison of CV curves of $(\text{CFO})_{1-x}(\text{GNPs})_x$ electrodes at 20 mVs^{-1}

It is interestingly seen that the CV plots for both CFO and $(\text{CFO})_{0.75}(\text{GNPs})_{0.25}$ exhibit sufficiently different shape than the rectangular shape observed for GNPs which suggest that there is pseudocapacitive contribution to the specific capacitance arising from Faradic redox reactions (FRR) [51]. It is clear from the Fig.8 (a)-(c) that the current response ~~increases~~ rises with the increase in scan rate for all samples. This increasing trend in current response is because of scan rate on the ~~diffusion and migration of electrolyte ions~~ electrolyte ions migration and diffusion. At relatively small scan rate, ~~the flux of electrolyte ions~~ the electrolyte ions flux is limited by the ~~formation coating~~ coating of thick diffusion layer on ~~the surface of electrode~~ the electrode surface, due to which the current response lowers. On the other hand, at higher scan rate, the flux

of electrolyte is enhanced towards the electrodes, because the high scan rate prevents the growth of thick diffusion layer and as a result current response is seemed to be increased[52]. The typical comparison of CV plots of GNPs, CFO and (CFO)_{0.75}GNPs_{0.25}based electrodes at 20mVs⁻¹ are shown in Fig.8 (d). It is very remarkable to note that the integrated area of CV curve for (CFO)_{0.75}GNPs_{0.25}based electrode is significantly larger than individual CFO and GNPs based electrodes, indicating superior super-capacitive characteristics of the nanocomposite electrode. This excellent electrochemical performance of (CFO)_{0.75}GNPs_{0.25} is attributed to high electrical conductivity conduction, high enhanced specific surface area and efficient charge transfer at electrode/electrolyte interface as result of GNPs addition in CFO [53].

5.2. Galvanostatic charge/discharge (GCD) analysis

To know the electrochemical performance of the electrodes based on (CFO)_{1-x}(GNPs)_x nanocomposites, the GCD measurements at room temperature have been carried out. The specific capacitance of single electrode at various current densities has been calculated from the discharging portion of GCD curves using the following relation [54],

$$C_s = \frac{4I\Delta t}{m\Delta V} = 4C_T \quad (5)$$

Where, Δt , I , ΔV , m and C_T are the ~~discharge time (s)~~ time of discharge (s), current (A), applied potential, ~~total mass of the active material in both electrodes (g)~~ sum of active material mass in both of the electrodes (g) and capacitance of the cell respectively. Fig.9 (a)-(c) shows the typical GCD curves for CFO, GNPs and (CFO)_{0.75}(GNPs)_{0.25} electrodes in 0.5M K₂SO₄ electrolyte under varying current densities.

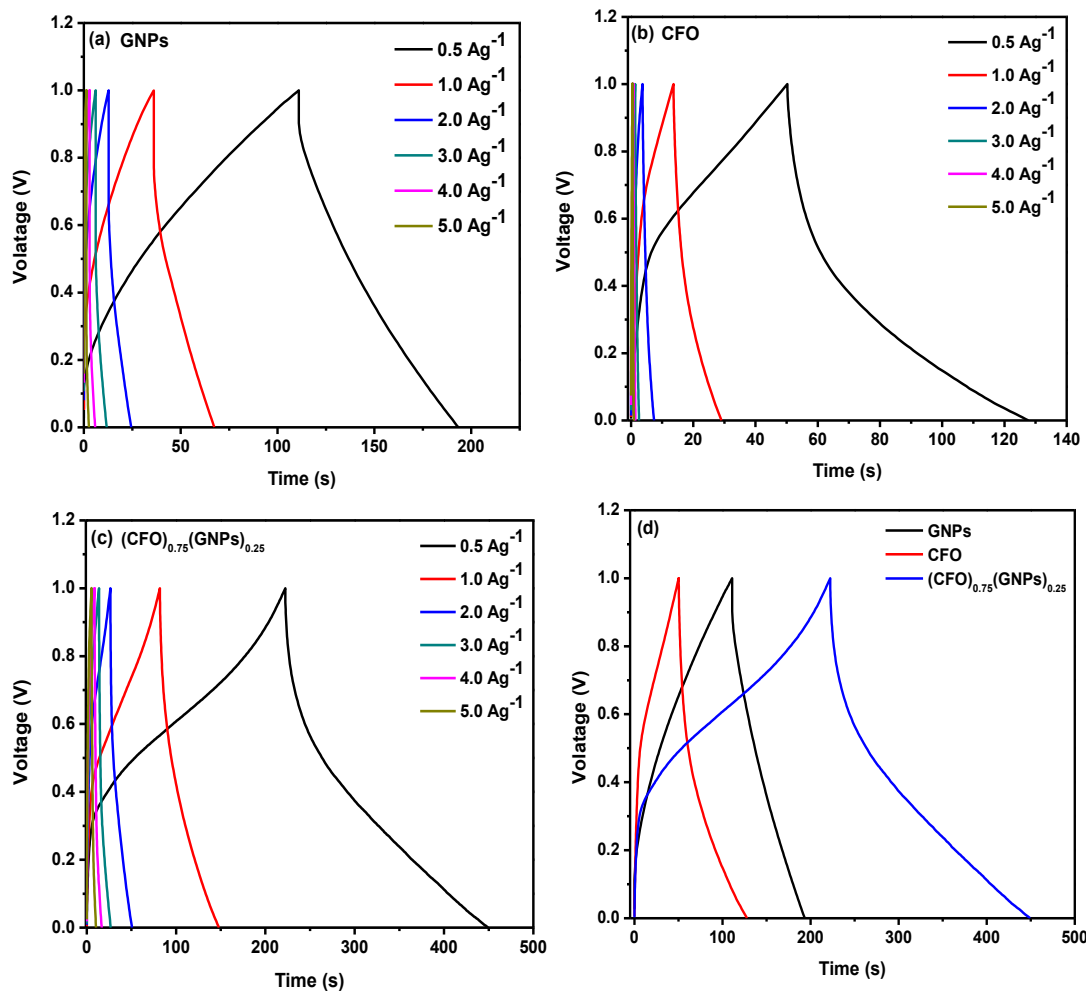


Fig. 9: (a-c) GDC curves of the $(\text{CFO})_{1-x}(\text{GNPs})_x$ electrodes at various scan rate (d) comparison of GDC curves of $(\text{CFO})_{1-x}(\text{GNPs})_x$ electrodes at 0.5 Ag^{-1} .

It can be observed that GCD curves of CFO and $\text{CFO}_{0.75}\text{GNP}_{0.25}$ electrodes depict typical battery-like pseudocapacitive character compare to GNPs electrode having double layer capacitive behavior [55]. The GCD curves having symmetric nature for $(\text{CFO})_{0.75}(\text{GNPs})_{0.25}$ clearly depict that nanocomposites have high rate performance as electrode materials compared with individual components. Typical comparison plots of GCD curves under 0.5 Ag^{-1} for GNPs, CFO and $(\text{CFO})_{0.75}(\text{GNPs})_{0.25}$ electrode are displayed in Fig.9(d). The specific capacitance was calculated to be 82 Fg^{-1} , 73 Fg^{-1} and 227 Fg^{-1} for GNPs, CFO and $(\text{CFO})_{0.75}(\text{GNPs})_{0.25}$ electrode at 0.5 Ag^{-1} respectively. It can be inferred from the obtained results that the supercapacitance of the

nanocomposite electrode is much higher than that of individual GNPs and CFO based electrodes. Fig. 10(a) shows the change in specific capacitance capacity with increasing growing current density. It can be seen from Fig. 10(a) that the nanocomposite electrode retains relatively larger higher specific capacitance even at high large current density as compared to neat GNPs and CFO based electrodes, suggesting its higher rate capability. The superior electrochemical performance efficiency of the prepared nanocomposite electrode might be due to because of the lower degree of aggregation and higher conductivity of the CFO nanostructures.

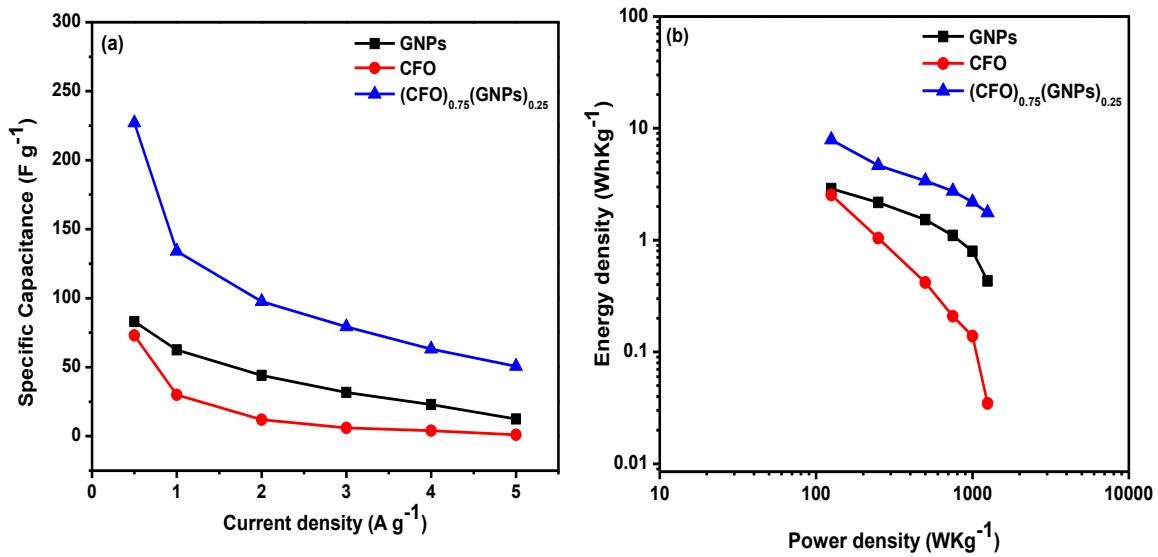


Fig. 10: (a) Specific capacitance curves of (CFO)_{1-x}(GNPs)_x electrodes at various current densities
(b) Ragone plots of (CFO)_{1-x}(GNPs)_x electrodes.

5.3. Energy density and Power Density

It is well known that materials with having both high energy density and high power density both high energy and high-power density could meet the fundamental requirements of storage devices. The energy density and power density of a supercapacitor cell energy and power density of a electrochemical supercapacitor device has been calculated using the following equations [56]:

$$E = \frac{0.5C_T(\Delta V)^2}{3.6} \quad (6)$$

$$P = \frac{E}{\Delta t} \quad (7)$$

~~Where E , C_T , ΔV , P , Δt are average energy density (Whkg^{-1}), specific capacitance of the cell, potential window of discharge (V), power density (Wkg^{-1}) and discharge time (s). Where E , ΔV , C_T , P , Δt are average energy density (Whkg^{-1}), specific capacitance of the cell, potential window of discharge (V), power density (Wkg^{-1}) and time of discharge (s). The energy density of the GNPs, CFO and $\text{CFO}_{0.75}\text{(GNPs)}_{0.25}$ is observed to be 2.9Whkg^{-1} , 2.5Whkg^{-1} and 7.9Whkg^{-1} at 125Wkg^{-1} power density respectively. The Ragone plots results in Fig. 10(b) demonstrate that the energy density of $\text{CFO}_{0.75}\text{(GNPs)}_{0.25}$ electrode is much higher than the GNPs and CFO based electrode. This high energy density of CFO–GNPs composites as compare to only CFO can be ascribed to its higher specific capacitance arising from the synergistic contribution of ECDL and pseudocapacitance [57, 58].~~

5.4. Electrochemical Impedance Spectroscopy (EIS)

The electrical conductivity and charge transfer characteristics at electrode/electrolyte interface $\text{CFO}_{1-x}\text{(GNPs)}_x$ samples are checked using EIS measurements in the frequency range 0.1Hz-100kHz. EIS plots of $\text{CFO}_{1-x}\text{(GNPs)}_x$ nanocomposites show a ~~straight-line~~ linear portion in ~~the low frequency range in the region of low frequency~~ and a ~~semicircle~~ are semicircular portion in the ~~region of high frequency~~ high frequency region as shown in Figure 11. The X-intercept (Z') of the semicircular part indicates the intrinsic resistance within the electrochemical cell, which is combination of three types of resistances: i) electrolyte's ionic resistance ii) inherent resistance of the material in electrode iii) resistance originated from the contact between current collector and electrode. The radius of the semicircle signifies the resistance to charge transfer at the electrode/electrolyte interface [59].

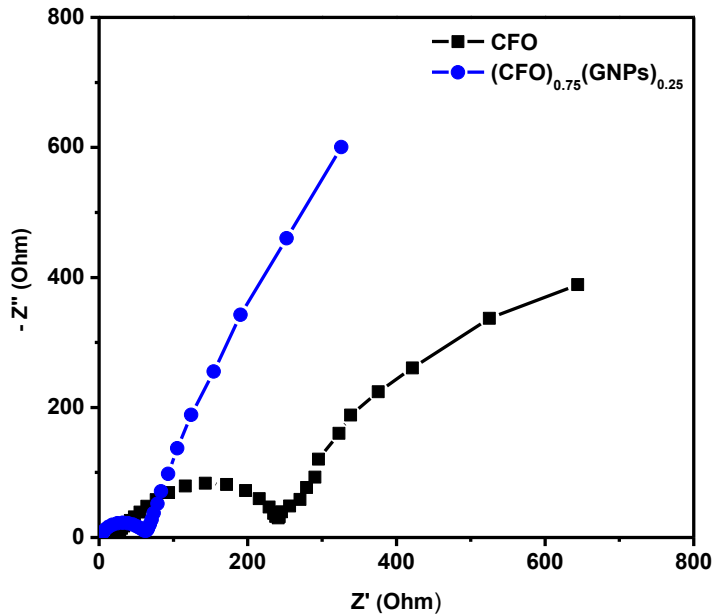


Fig. 11: EIS spectra of $(\text{CFO})_{1-x}(\text{GNPs})_x$ electrodes in frequency range 100kHz-0.1Hz

It is clear from the figure that the radius of semicircular portion of impedance spectra for $(\text{CFO})_{0.75}(\text{GNPs})_{0.25}$ is smaller than the CFO sample. This implies that the resistance to the interfacial charge transfer for $(\text{CFO})_{0.75}(\text{GNPs})_{0.25}$ is smaller than the CFO electrode. It is also noted that the slope of the line in Nyquist plots is more vertical in case of $(\text{CFO})_{0.75}(\text{GNPs})_{0.25}$ as compared to neat CFO, which is a manifestation of low diffusive resistance. This significant decrease in resistance to charge transfer of the $(\text{CFO})_{0.75}(\text{GNPs})_{0.25}$ can be attributed to the high conductivity of GNPs component present in the nanocomposites.

5.5. Cyclic performance and Columbic efficiency

The **eyelie** cycling **performance stability** and Columbic efficiency of supercapacitors are key factors for their practical use in energy storage devices[60]. The cycling stability tests have been performed by GCD over 2000 cycles. Fig. 12 shows capacitive retention and Columbic efficiency of the $(\text{CFO})_{0.75}(\text{GNPs})_{0.25}$ electrode at 5 Ag^{-1} . The results demonstrate that 72% of the initial specific capacitance is retained after 2000 continuous cycles, whereas the device has constant Columbic efficiency of 94 %. This outstanding cyclic stability achievement could be ascribed to the prevention of aggregation of CFO nanostructures during charge-discharge as

well as the formation establishment of Fe–O–C and Co–O–C chemical bonds in the (CFO)_{0.75}(GNPs)_{0.25} nanocomposites [61].

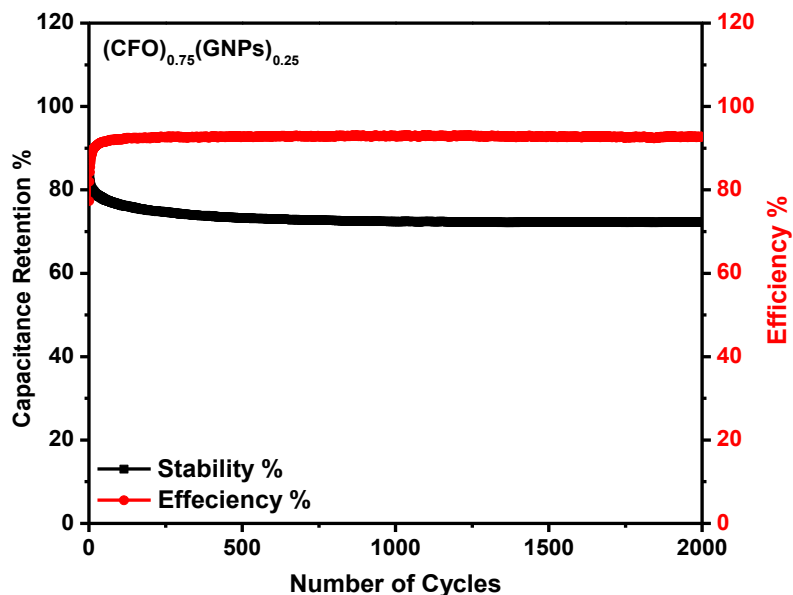


Fig.12: Cycling stability and efficiency of (CFO)_{1-x}(GNPs)_x electrodes

6. Conclusion

A series of (CFO)_{1-x}(GNPs)_x nanocomposites have been prepared through simple eco-friendly chemical method. GNPs induced enhancement in photocatalytic and electrochemical characteristics of prepared nanocomposites have been tailored and explored. All composites have shown significant photocatalytic activities, however the (CFO)_{0.75}(GNPs)_{0.25} nanocomposite has been found to be an excellent photocatalyst among them by degrading 98.51% of MB within 120 minutes via visible light driven photo Fenton reaction. This dye degradation performance is attributed towards high specific area, improved interfacial charge transfer, GNPs promoted charge carrier separation, and photo Fenton reaction. It has been also found that this particular (CFO)_{0.75}(GNPs)_{0.25} nanocomposite has better electrochemical specific capacitance (227 Fg⁻¹ at 0.5 Ag⁻¹) and outstanding stability (72% capacitance retention) after 2000 cycles. The improved electrochemical performance and cycling stability of (CFO)_{0.75}(GNPs)_{0.25} electrodes can be attributed to the prevented the agglomeration of CFO nanostructures, improved charge-transfer transfer of charge at the interface of the electrode/electrolyte interface and mechanical integrity

provided by GNPs. These findings recommend that $(\text{CFO})_{1-x}(\text{GNPs})_x$ nanocomposites can be used in cost-effective water purification technologies and electrochemical energy storage nanodevices.

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8. References

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List of Figures

Fig.1: XRD patterns of $(\text{CFO})_{1-x}(\text{GNPs})_x$ samples

Fig.2: TEM micrographs of $(\text{CFO})_{1-x}(\text{GNPs})_x$ samples (inset: HRTEM image of CFO)

Fig.3: FTIR spectra of $(\text{CFO})_{1-x}(\text{GNPs})_x$ nanocomposites

Fig.4: XPS wide angle spectra for $(\text{CFO})_{1-x}(\text{GNPs})_x$ nanocomposites.

Fig.5: a) UV-vis absorbance spectra of $(\text{CFO})_{1-x}(\text{GNPs})_x$ nanocomposites b) Tauc's plots of $(\text{CFO})_{1-x}(\text{GNPs})_x$ nanocomposites.

Fig. 6: PL spectra of $(\text{CFO})_{1-x}(\text{GNPs})_x$ nanocomposites

Fig. 7: (a)-(e) Absorption spectra of MB solution for $(\text{CFO})_{1-x}(\text{GNPs})_x$, and (f) photodegradation of $(\text{CFO})_{1-x}(\text{GNPs})_x$ nanocomposites

Fig. 8: (a)-(c) CV curves of the $(\text{CFO})_{1-x}(\text{GNPs})_x$ electrodes at various scan rate, and (d) comparison of CV curves of $(\text{CFO})_{1-x}(\text{GNPs})_x$ electrodes at 20 mVs^{-1}

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(b) Ragone plots of $(\text{CFO})_{1-x}(\text{GNPs})_x$ electrodes.

Fig. 11: EIS spectra of $(\text{CFO})_{1-x}(\text{GNPs})_x$ electrodes in frequency range $100 \text{ kHz} - 0.1 \text{ Hz}$

Fig.12: Cycling stability and efficiency of $(\text{CFO})_{1-x}(\text{GNPs})_x$ electrodes