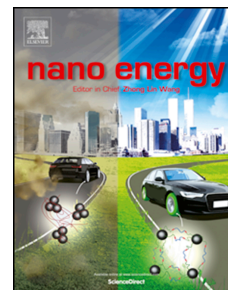


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Photochemically deposited Ir-doped NiCo oxyhydroxide nanosheets provide highly efficient stable electrocatalysts for the oxygen evolution reaction

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The contributions of the authors are as follows.

Liang-ai Huang: Experimental Investigations, experimental Methodology, Writing- paper

Hyeyoung Shin: computations, writing of computational parts and of discussion

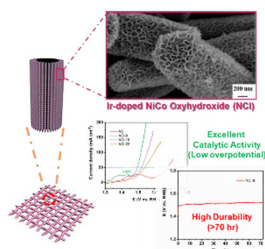
William A. Goddard III: Conceptualization of computations, writing of computational parts and of discussion

Jianming Wang: Conceptualization of experiments, Writing- Reviewing and Editing of experimental section,

Supervision of experiments

Journal Pre-proof

Photochemically deposited Ir-doped NiCo Oxyhydroxide Nanosheets provide Highly Efficient and Stable Electrocatalysts for the Oxygen Evolution Reaction



Journal Pre-proof

1 **Photochemically deposited Ir-doped NiCo Oxyhydroxide** 2 **Nanosheets provide Highly Efficient Stable Electrocatalysts for** 3 **the Oxygen Evolution Reaction**

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14 **Abstract**

15 To achieve practical production of fuel from water, it is essential to develop efficient and
16 durable electrocatalysts for the oxygen evolution reaction (OER). We report here that doping
17 NiCo-OOH nanosheets with 8% Ir leads to a low overpotential of only 260 mV for 50
18 mA/cm², far better than previous OER catalysts. We synthesized this catalyst using a novel
19 photochemical deposition method that leads to a uniform distribution of dopant, large
20 catalytic active area, high interfacial charge transfer efficiency, good adhesion between
21 catalyst and matrix, and long lifetime. Moreover, these nanosheets show significant stable
22 performance for 70 hours in alkaline media. Our density functional theory calculations, show
23 that Ir and Co both play essential bifunctional roles in stabilizing the key O radical
24 intermediate on Ir and promoting the O-O bond coupling on Co, which is optimum for the 8%
25 Ir.

26 **Keywords:** Oxygen Evolution Reaction; NiOOH Oxyhydroxide; Density Functional Theory;
27 Photodeposition; Reaction Mechanism;

28

29 1. Introduction

30 The oxygen evolution reaction (OER) plays an important role in such energy conversion
31 systems as water splitting, CO₂ reduction, and metal-air batteries [1-3]. Recently, great efforts
32 have been devoted to developing OER catalysts with improved performance [4]. However,
33 sluggish kinetics due to a high overpotential for the complex four-electron coupled process
34 have remained as obstacles to making the electrochemical system more efficient. This makes
35 identification of active and efficient electrocatalysts for high OER performance a priority.
36 IrO₂ has been regarded as promising OER catalyst in acid media due to its low overpotential
37 and high catalytic performance. However, its high cost and instability in base hinder its
38 commercial applications. Thus, it is imperative to identify and characterize new low cost
39 electrocatalysts that provide more efficient and durable electrocatalytic properties.

40 Among various transition metal oxide based OER electrocatalysts, Nickel oxyhydroxides
41 (NiOOH) have shown promising electrocatalysis toward OER in alkaline medium with
42 enhanced electrochemical activity, earth abundance nature, and open structure [5-7]. Thus γ -
43 NiOOH is known to be the active phase for OER, with a formal Ni valence of about +3.6 [8].
44 Several studies have shown that incorporation of other metals in NiOOH can further improve
45 OER catalytic activity [9]. In particular, Fe-doped γ -NiOOH oxyhydroxides (Fe- γ -NiOOH)
46 have been recognized as most active non-precious electrocatalyst for OER under alkaline
47 condition, leading to excellent catalytic performance and prominent stability [10]. However,
48 Fe- γ -NiOOH is far from adequate. To design better electrocatalyst, Boettcher showed
49 experimentally that NiCo-based oxyhydroxides (NiCoOOH) enhance catalytic performance
50 over NiFe-based oxyhydroxides because the electronic conductivity of CoOOH is far
51 superior to that of FeOOH [11]. Previous studies also found improvements for Mo- and Fe-
52 doped NiOOH [5]. However, the OER performance of bulk NiCoOOH remains
53 unsatisfactory due to its low specific surface area and sluggish interface charge transfer.
54 Moreover, the semiconductor properties of pristine NiCoOOH greatly impede charge transfer
55 from the catalyst to the supporting substrate, limiting catalytic activity and overall efficiency
56 for the OER [12-14]. Recently, *in-silico* screening by Shin and Goddard predicted that Co-,
57 Rh-, and Ir-doping of NiOOH would improve OER electrocatalysis, with particularly
58 dramatic improvement for Ir [15]. Therefore, we developed new materials synthesis strategies
59 for Ir-doped NiCoOOH catalyst aiming at boosting OER catalytic activity.

60 In addition to enhancing the catalytic activity, it is also important to design the catalyst to
61 obtain a large active surface area and good structural stability. Constructing free-standing
62 nanostructures can provide large electrochemical active area while strengthening the adhesion
63 interaction between catalyst and substrate can improve the structural stability of the catalyst.
64 To accomplish this, we developed a **novel photochemical deposition method** that has
65 prominent advantages of **uniform distribution, high interfacial charge transfer efficiency,**
66 **good adhesion between catalyst and matrix** while saving energy in fabricating the Ir-doped
67 NiCoOOH catalyst. Our photochemical deposition method utilizes the strong oxidizing
68 properties of photo-generated holes. We show that under stimulation by UV-vis light,
69 photogenerated holes are transferred to the semiconductor surface leading to oxidation of
70 metal ions and in-situ deposition on its surface, thus greatly strengthening the combination
71 between the catalyst and the substrate. This contrasts with the photochemical preparation
72 method of the Berlinguette group, who uses photochemical decomposition to fabricate OER
73 catalysts [16-18].

74 Based on these considerations, we first designed a hierarchical layered Ir-doped NiCoOOH
75 catalyst supported on ZnO/nitrogen-doped carbon cloth substrate, which is fabricated
76 successfully through our novel photochemical deposition method. Under light stimulation the
77 Ni^{2+} , Co^{2+} and Ir^{3+} were all oxidized by the strong oxidizing property of the photo-induced
78 holes so that the electrocatalyst is uniformly deposited in-situ on the outer surface of the *n*-
79 type semiconductor ZnO. Our electrochemical tests show that the Ir-doped NiCoOOH
80 electrode with 8% Ir exhibits excellent catalytic performance, requiring an **overpotential of**
81 **only 260 mV to achieve a current of 50 mA cm⁻²**, which is superior to previous
82 electrocatalysts for OER. Moreover, our photochemical based synthetic method leads to large
83 surface area and increased stability (70 hours with only 3% change in performance).

84 We find that the excellent catalytic performance of the Ir-doped NiCoOOH/ZnO composite
85 arises mainly from three aspects:

- 86 i) photochemical deposition ensures strong bonding between the catalyst and the
87 substrate, which facilitates the construction of free-standing nanostructures having large
88 electrochemical active area to improve activity and structure stability;
- 89 ii) the hierarchical porous structures expose more surface area of the catalyst, providing
90 increased active sites while promoting full contact with the electrolyte to facilitate charge
91 transfer;
- 92 iii) the Ir dopant on oxyhydroxides / oxides plays a pivotal role in promoting OER active
93 sites while modifying the electronic structure to greatly reduce the catalyst overpotential to
94 accelerate charge transfer, boosting catalytic performance.

95 To understand the origin of dramatically improved performance due to Ir doping on the
96 NiCoOOH surface, we applied density functional theory (DFT). This confirmed that the
97 catalyst with 8% Ir exhibits the best performance with the Ir facilitating formation of IrO oxo
98 bond with radical character on the O and the Co facilitating the O-O formation step.

99

100 2. Experimental section

101 2.1. Synthesis of ZnO NRAs@NCC

102 The chemical reagents used were all analytical grade, without further purification. To im-
103 prove the hydrophilicity of carbon cloth, it was surface-functionalized using concentrated HNO₃
104 for 24 h at 90 °C, then rinsed several times with deionized water, and dried at 60 °C under
105 vacuum. This N-doping treatment increases the roughness and the electronic conductivity of
106 the carbon cloth. The fabrication of nitrogen-doped carbon cloth (NCC) involves three steps:
107 followed by hydrothermal reaction, thermal reduction, and concentrated acid treatment, as
108 described in the related investigations of Tong and coworkers [19]. The growth of ZnO
109 nanorod arrays on NCC was carried out with the following steps. First, a piece of NCC (1 cm
110 × 1.5 cm) was immersed in the seeding solution containing 0.01 M $\text{Zn}(\text{CH}_3\text{COO})_2 \cdot 2\text{H}_2\text{O}$ and
111 0.01 M NaOH, followed by hydrothermal treatment at 150 °C for 15 min. The above seeding
112 process was repeated four times. After that, the growth solution of 0.10 M
113 hexamethylenetetramine (HMTA) and 0.10 M $\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ were dissolved in deionized
114 water separately at room temperature, then mixed and stirred for 30 min. Then the NCC
115 covered with the ZnO seed layer and the growth solution were transferred together to a
116 Teflon-lined stainless steel autoclave and heated at 100 °C for 12 h. The obtained greyish-
117 white film was rinsed extensively with deionized water and finally annealed in a pipe furnace

118 at 400 °C for 1 h under the flow of Ar to obtain the ZnO NRAs@NCC substrate.

119 2.2. Preparation of the Ir-doped NiCoOOH/ZnO@NCC composite film

120 The Ir-doped NiCoOOH/ZnO@NCC composite films were fabricated through a novel
121 photochemical deposition method. The photodeposition solution consisted of 0.13 M
122 NiSO₄·6H₂O, 0.10 M CoCl₂·6H₂O, 0.13 M CH₃COONa with a series of molar concentrations
123 of IrCl₃·xH₂O (Ir>52%). The ZnO NRAs@NCC substrate and the Pt electrode were placed
124 into the photodeposition reactor and connected by a wire for short-circuit. The ZnO
125 NRAs@NCC substrate was placed opposite the UV-vis light irradiation (500 W Xe lamp, 2.0
126 mW cm⁻²) through the quartz window, and its exposed area was 1 cm², the photodeposition
127 process is displayed in Figure S1. In the process of photodeposition, the solution was stirred
128 constantly to ensure the stability and uniformity of the deposition system. The composite film
129 obtained after 3 hours of photodeposition exhibits the best catalytic activity. After the
130 deposition, the as-fabricated composite film was rinsed with deionized water several times
131 and vacuum dried at 60 °C.

132 The preparation process of NiCoOOH/ZnO@NCC composite film was the same as for the Ir-
133 doped NiCoOOH/ZnO@NCC, except that there is no IrCl₃·xH₂O in the photodeposition
134 solution. We denote the Ir-doped NiCoOOH/ZnO@NCC composite films as NCI-x%, where
135 x is the atomic ratio of Ir content among all metals (Ni, Co, Ir), and the value of x is
136 determined by the ICP analysis. These atomic ratios are displayed in Table S1. Similarly, NI-
137 x% indicates the individual doping of Iridium into NiOOH, where x refers to the Ir content.
138 NiCoOOH/ZnO@NCC is described as NC.

139 2.3. Physical characterization

140 The crystalline structures of the fabricated film samples were characterized by X-ray
141 diffraction (XRD) on the D/MAX 2550 X ray diffractometer from Rigaku company using Cu
142 K α radiation. Scanning electron microscopy (SEM, Carl Zeiss, Ultra 55) at an acceleration
143 voltage of 5 kV and high-resolution transmission electron microscopy (TEM, Tecnai G2 F30)
144 with an accelerating voltage of 200 kV were used to observe the morphologies and
145 microstructures of the films. X-ray photoelectron spectroscopy (XPS) data were recorded
146 using the Escalab 250Xi X-ray physical electronics photoelectron spectrometer with Mg K α
147 radiation. The obtained spectra were corrected according to the adventitious C 1s peak (284.6
148 eV). The atomic ratios of the Ni, Co, Ir elements of the as-fabricated composite films were
149 analyzed using inductively coupled plasma atomic emission spectroscopy (ICP, SPECTRO).
150 The as-fabricated samples were dissolved in nitric acid solution, and the resulted solutions
151 were used for ICP analysis.

152 2.4. Electrochemical measurements

153 Electrochemical measurements of the catalysts were carried out using a three-electrode
154 system on a CHI660D electrochemical workstation at room temperature. The as-prepared
155 NCI composite film was used as the working electrode, while Ag/AgCl electrode (filled with
156 saturated KCl solution purchased from Rex company) and platinum foil were utilized as the
157 reference electrode and the counter electrode, respectively. 1 M KOH aqueous solution was
158 employed as an electrolyte. The Ag/AgCl electrode calibration was carried in a three-
159 electrode system with Pt foil as working counter electrode and Ag/AgCl as reference
160 electrode. The electrolyte was H₂ saturated 1.0 M KOH. The LSV curve was collected at 5
161 mV/s scan rate, and the potential at which the current crosses zero was taken as
162 thermodynamic potential (vs Ag/AgCl) for the hydrogen electrode [20]. The potential at

163 which current crosses zero is -1.027 V vs Ag/AgCl, as shown in Figure S2, so all potentials
164 were calibrated according to the equation: $E_{\text{RHE}} = E_{\text{Ag/AgCl}} + 1.027$. The electrocatalytic
165 activity of the NCI composite film was measured by linear sweep voltammetry (LSV) at a
166 scan rate of 5 mV s^{-1} with 95% iR drop compensation. The stability performance of the NCI
167 composite film was tested using chronopotentiometry at a current density of 20 mA cm^{-2} .
168 Cyclic voltammetry (CV) was carried out at a scan rate of 5 mV s^{-1} in the potential range of
169 0.2 to 0.9 V vs. Ag/AgCl. Electrochemical impedance spectroscopy (EIS) was conducted
170 within the frequency range of 100 kHz to 0.01 Hz to obtain the solution impedance (R_s) of the
171 electrochemical system and investigate the kinetics of the electrocatalysts under OER
172 conditions. The electrochemical active surface area (ECSA) was determined on the basis of
173 the measured double-layer capacitance (C_{dl}), which was calculated by CV curves within a
174 potential range of 0.1 V centered at open-circuit potential at different scan rates.

175 2.5. Computational details

176 We carried out spin-polarized density functional theory (DFT) calculations using the Vienna
177 Ab-initio Simulation Package (VASP) [21] for the investigation of the atomistic mechanisms
178 for OER on the NCI surfaces. We used the Perdew-Burke-Ernzerhof (PBE) flavor of DFT [22]
179 including the D3 empirical van der Waals correction with Becke-Johnson parameters. The
180 projector augmented wave (PAW) potentials [23] were used to describe the valence electron-
181 ion interactions. This level correctly describes the adsorption energies of intermediates on a
182 variety of surfaces [24-26].

183 For the geometry optimizations, we used a plane-wave energy cutoff of 400 eV. We employed
184 a gamma-centered Monkhorst k-point mesh of $(3 \times 3 \times 1)$. All surface models were built by
185 including an additional vacuum region of 15 Å. The bottom two layers of the slabs were fixed
186 at the lattice spacing of the bulk while the top two layers were allowed to optimize. Dipole
187 corrections were applied to the surface normal direction. Here, the solvation was included by
188 introducing explicit H_2O molecules on the surface.

189 For the calculation of free energy for each step at room temperature (298.15 K), we computed
190 the zero-point energies, enthalpy, and entropy based on the vibrational frequencies.

191

192 3. Results and discussion

193 3.1. Synthesis, structure and composition of materials

194 The fabrication of the hierarchical layered NCI composite film is displayed schematically in
195 Fig. 1. The ZnO nanorod arrays were initially grown on NCC through the hydrothermal
196 reaction using zinc acetate, sodium hydroxide, hexamethylenetetramine and zinc nitrate as
197 precursors. Subsequent heat treatment was carried out in an Ar atmosphere in order to
198 improve the crystallinity of ZnO nanorod arrays. After that, the strong oxidizing properties of
199 photogenerated holes, were used to oxidize the metal ions (Ni^{2+} , Co^{2+} , Ir^{3+}) to obtain
200 hierarchical layered NCI composite film. It is important to note that the uniformly
201 hierarchical nanorod structure of the ZnO can effectively disperse the catalytically active
202 materials and fully expose the catalytically active sites of the NCI electrocatalysts.

203 The XRD patterns of the as-fabricated NC, NCI-8, NCI-15 and NCI-22 composite films are
204 shown in Fig. 2 and Fig. S3. All diffraction peaks of the as-fabricated samples match well
205 with ZnO (JCPDS Card No. 01-070-8070) or carbon cloth (JCPDS Card No. 00-037-0474),
206 with no other peaks, indicating the amorphous structure of the as-synthesized electrocatalysts.

207 We expect that only a small driving force is required to transform the oxygen-evolving state
208 from an amorphous structure into an ordered structure, explaining the high catalytic activity
209 of the amorphous catalyst [27].

210 The Raman spectra of the as-prepared NC, NCI-8, NCI-15 and NCI-22 composite films are
211 displayed in Fig. S4. The peaks located at 305 and 335 cm^{-1} can be assigned to Ir-OH signals
212 [28], and the weak peak observed at 561 cm^{-1} is the characteristic Raman Ir-O stretch mode
213 [29-30], indicating a Ir-OOH phase. The sharp peak observed at 440 cm^{-1} is attributed to the
214 E2 (high) vibration modes of ZnO [31-32], which is consistent with the XRD results.
215 Furthermore, the broad Raman band located at 587 cm^{-1} can be assigned to CoOOH (Co(IV))
216 species [30,33], and the other two peaks centered at 475 and 550 cm^{-1} correspond to Ni-O
217 vibrations in γ -NiOOH [34-36]. However, it is clear that these two peaks shift to 538 cm^{-1}
218 with incorporation of Ir.

219 The microstructures of NCI-8 were characterized by using SEM, TEM, and HRTEM, as
220 shown in Fig. 3. NCI-8 is grown uniformly on regular hexagonal prism ZnO nanorods (Fig.
221 S5) with hierarchical arrays consisting of interlaced nanosheets, as shown in Fig. 3a-c. The
222 thickness of the nanosheets is measured to be about 5 nm. The SEM images of the as-
223 fabricated NC, NCI-15 and NCI-22 with different amounts of Ir dopant are displayed in Fig.
224 S6, showing nanosheet array structures similar to NCI-8. The TEM images in Fig. 3d-e
225 further demonstrate that the ZnO nanorods are evenly coated with hierarchical nanosheets,
226 and the interconnected nanosheets are stacked to form hierarchical porous structures. In
227 addition, visible dark strips are folded edges or wrinkled nanosheets, which show their
228 ultrathin nature. The lattice spacing of 2.48 Å in the HRTEM images (Fig. 3f and Fig. S7a)
229 can be indexed to the (101) crystal plane of ZnO, but no other obvious lattice space can be
230 observed. The corresponding SEAD pattern of amorphous layer is displayed in Fig. S7b,
231 which further confirms the existence of ZnO in the composite and the amorphous structure of
232 NCI-8 nanosheets. This is in accordance with the XRD results. The elemental mappings of
233 NCI-8 in Fig. 3g show that the elements Ni, Co, Zn, O and Ir are well-dispersed and the Ir
234 content is relatively low. The comparative NC composite film exhibits the similar
235 microstructures to NCI-8, as shown in Fig. S8a-b, and its HRTEM image (Fig. S8c) indicates
236 the presence of ZnO substrate and the weak crystalline catalyst layer. The original amorphous
237 layer has weak lattice stripes due to the rearrangement of the amorphous disordered structure
238 by high-energy electrons in the detection environment. The elemental mappings of NC (Fig.
239 S9) show the homogeneous distributions of elements Ni, Co, Zn and O.

240 X-ray photoelectron spectroscopy (XPS) measurements were conducted to further investigate
241 the valence state and element composition of the as-fabricated NCI-8 composite film. In the
242 high-resolution Ir 4f spectrum in Fig. 4a, two main peaks of Ir 4f_{7/2} and Ir 4f_{5/2} can be
243 deconvoluted into Ir³⁺ and Ir⁴⁺ [37], and two bands at 61.8 and 64.9 eV correspond to Ir³⁺,
244 while the Ir⁴⁺ peaks are located at 62.8 and 65.8 eV. But after OER, the spectrum shows a
245 significant shift toward higher binding energy, which indicates that some of the Ir³⁺/Ir⁴⁺ have
246 been further oxidized to Ir⁵⁺ species [38-39], which is consistent with the results of the CV
247 test in Fig. 5c. Fig. 4b shows the high-resolution XPS pattern of the Ni 2p spin-orbit splitting
248 of NCI-8. The Ni 2p_{3/2} and Ni 2p_{1/2} peaks can be deconvoluted into peaks of Ni³⁺ and Ni⁴⁺,
249 where the characteristic peaks at about 855.8 and 873.6 eV can be identified as Ni³⁺ species,
250 whereas the other two peaks centered at 857.9 and 875.6 eV can be attributed to Ni⁴⁺ species.
251 After OER, the proportion of Ni⁴⁺ has increased significantly, as some of the Ni³⁺ are
252 oxidized to Ni⁴⁺, leading to a final formal Ni valence is about 3.52, which is in line with the
253 average valence of the reported γ -NiOOH [40]. The high-resolution Co 2p XPS spectrum of

254 the as-fabricated NCI-8 is given in Fig. 4c, showing a pair of Co $2p_{3/2}$ and Co $2p_{1/2}$ doublet
 255 peaks, which can also be deconvoluted into peaks of Co³⁺ and Co⁴⁺. The as-deconvoluted
 256 peaks centered at 781.0 and 796.5 eV correspond to Co³⁺ species, while the peaks with the
 257 binding energies at 783.5 and 798.3 eV are ascribed to Co⁴⁺ species [41]. Similarly, the ratio
 258 of Co⁴⁺ has increased significantly after OER, causing Co species to exhibit a higher valence
 259 state. The wide-scan survey spectra of NC, NCI-8, NCI-15 and NCI-22 composite films are
 260 displayed in Fig. S10, revealing the successful photochemical depositions of NC and NCIs on
 261 the final composite films, respectively. The high-resolution Ni $2p$, Co $2p$ and Ir $4f$ XPS
 262 patterns of NC, NCI-8, NCI-15 and NCI-22 composite films after OER tests are shown in
 263 Figs. S11, S12 and S13, respectively. Compared to the NC sample, the Ni $2p$ XPS peaks of
 264 NCI-8, NCI-15 and NCI-22 all shift to a higher binding energy, indicating a strong electronic
 265 interaction between Ir dopant and NiCoOOH. In addition, Figs. S12 and S13 show that the
 266 Co $2p$ peaks and Ir $4f$ peaks of NCI-8 have the highest binding energy compared with the
 267 other as-fabricated electrocatalysts. Thus both the Co species and the Ir species exhibit the
 268 highest average valence state for NCI-8, the most active for OER [8-9,42].

269 3.2. Electrocatalytic activity of catalysts

270 In order to evaluate the effect of Ir doping, the OER catalytic activity of NCI electrodes with
 271 different atomic ratio of Ir content of 0, 8%, 15% and 22% (NC, NCI-8, NCI-15 and NCI-22,
 272 respectively) were measured via a standard three-electrode system by linear sweep
 273 voltammetry (LSV) in 1.0 M KOH with a scan rate of 5 mV s⁻¹. As indicated in the LSV
 274 curves of the various as-fabricated samples after *iR* correction (Fig. 5a), it is clear that NCI-8
 275 exhibits the highest catalytic activity. It requires only an overpotential of only ~260 mV at 50
 276 mA cm⁻², in contrast to NC, NCI-15, and NCI-22, which show much larger overpotentials of
 277 448, 290 and 310 mV, respectively. Since the surface metal (Ni, Co and Ir) oxidation is
 278 accompanied by the OER process, to eliminate the contribution of any transient responses,
 279 we measured the catalytic activity of the as-fabricated samples using multi-potential step
 280 measurements, as shown in Fig. S14. For a more rigorous experimental design, we also
 281 fabricated two other catalysts, NCI-4 and NCI-11, whose OER performance are shown in Fig.
 282 S15. Clearly NCI-4 and NCI-11 exhibit poorer OER performance than NCI-8, indicating that
 283 a slight change in the content of Ir has a significant impact on the OER performance of the
 284 NCI catalyst.

285 **NI-8 (Fig. S16) shows dramatically improved OER performance with overpotentials of**
 286 **290 mV at 50 mA cm⁻², which can be compared to an overpotential of 445 mV at 50 mA**
 287 **cm⁻² for NiOOH.** This shows that the Ir doping plays a key role in enhancing catalytic
 288 activity of NiOOH. This validates the DFT predictions made earlier by Shin, Xiao, and
 289 Goddard [15]. Comparing the activities of NCI-8 and NI-8, we see that Co doping is also
 290 important in promoting OER activity, which we ascribe to **synergistic effects, with Ir**
 291 **stabilizing the radical O character and Co stabilizing O-O bond formation while Ni**
 292 **stabilizes the overall structure.**

293 The kinetics of OER process of all as-fabricated samples are evaluated by Tafel plots
 294 (overpotential vs. log *j*) derived from their polarization curves, as displayed in Fig. 5b. The
 295 Tafel slope of NCI-8 is about 72 mV/dec, which is significantly lower than that of NiOOH
 296 (224 mV/dec), NC (198 mV/dec), NCI-15 (134 mV/dec) and NCI-22 (163 mV/dec),
 297 demonstrating its superior intrinsic activity and more favorable electrocatalytic kinetics
 298 toward OER. However, we also observe that the Tafel slope of NI-8 (80 mV/dec) (Fig. S16)
 299 is only a little larger than that of NCI-8, further illustrating the critical role of Ir doping.

300 In order to illustrate the intrinsic activity of the as-fabricated film catalysts, we obtained the
 301 mass activity and turnover frequency (TOF) of the film samples. The average loadings of the
 302 NC, NCI-8, NCI-15 and NCI-22 catalysts on substrates (1cm²) were about 0.32, 0.37, 0.39
 303 and 0.40 mg, respectively. The mass activity of the NCI-8 film sample was over twice as high
 304 as the other films (see Fig. S17). The TOF reflects the intrinsic activity of the catalyst, but the
 305 number of active sites is difficult to determine, especially in heterogeneous catalytic systems.
 306 The simplest method is to take every metal cation as the "active site" and use the total film
 307 mass to determine the moles of metal in the film [11,43-44]. Therefore, total-metal TOF
 308 (TOF_{tm}) is defined as:

$$309 \quad \text{TOF}_{\text{tm}} (\text{s}^{-1}) = \frac{\frac{\text{Current (A)}}{4F (\text{C/mol})}}{\frac{\text{film mass (g)}}{MW \text{ NiCoIrOOH (g/mol)}}} \quad (1)$$

310 At $\eta = 300$ mV, the TOF_{tm} of the as-fabricated NC, NCI-8, NCI-15 and NCI-22 catalysts are
 311 calculated to be 0.0171, 0.0988, 0.0462 and 0.0335 s⁻¹, as displayed in Fig. S17. In addition,
 312 we note that our fabricated NCI-8 electrocatalyst exhibits excellent OER activity, superior to
 313 that previously reported on state-of-the-art catalytic Ni-Co-Ir, Co-Ir, Ni-Ir systems and
 314 electrodes. (Table S2).

315 3.3. Density functional theory calculations

316 In order to understand the origin of high OER activity and the roles of each component in
 317 NCI-8, we carried out DFT calculations using methods we applied previously for *in silico*
 318 studies to identify dopants to replace Fe in Fe-doped NiOOH [15]. We built the NCI-8 slab
 319 model shown in Fig. S18, based on the crystal structure of NI-8 model used previously [15].

320 We first considered all possible reaction steps for OER to find the lowest energy pathways for
 321 OER, as shown in Fig. 6. This includes every state associated with each oxidation step (losing
 322 one electron) coupled with deprotonation and every possible O-O bond formation state on
 323 each different surface element (Ir, Co and Ni). The final results for the predicted pathway
 324 having the lowest free energy for OER on NCI-8 are reported in Fig. 6. Here we note that the
 325 presence of Ir on the NCI surface facilitates formation of a O radical sites on the Ir oxo bond
 326 while Co facilitates O-O coupling, both essential for OER. This synergy lowers the
 327 overpotential for OER.

328 As shown in Fig. 6, sequential oxidation-deprotonation steps starting from the NCI-8 slab
 329 model (Fig. S18, state 1) (state 1 → state 2 and state 2 → state 3) lead to formation of the O
 330 radical (O•), key intermediate for OER on the surface Ir site. Our DFT spin analysis finds that
 331 the Ir is in the 5+ oxidation state, in good agreement with the Cyclic Voltammogram data in
 332 Fig. 5c. The reaction (step 2 → 3) proceeds via deprotonation of the OH adsorbed on the Ir,
 333 an endothermic process requiring 0.49 eV. Next O-O coupling (state 3 → state 4) is generated
 334 by interaction between the O• on Co and an additional H₂O introduced at oxidation step 3,
 335 hydrogenating O on the Ir⁵⁺ site. Interestingly the O-O coupling prefers the Co site (requiring
 336 1.18 eV for O-O coupling) rather than the Ir site (requiring 1.41 eV for O-O coupling). This
 337 makes O-O bond coupling step the rate-determining step (RDS) for OER over NCI-8.

338 These results indicate that both Ir and Co play essential bifunctional roles in the OER
 339 catalysis just as we found earlier for Fe-doped NiOOH [15, 45]. We see that the free energy
 340 barrier required for both the O radical formation and O-O coupling steps (0.49 and 1.18) are
 341 significantly reduced compared to the free energies required for OER in the NiOOH (2.06 eV)
 342 and the Fe-doped NiOOH (1.68 eV) [15]. These low energy barriers explain the observed
 343 enhanced catalytic performance for OER on NCI-8.

344 Table S4 compares NC with NCI-8, showing that without Ir, the 2nd step of forming the O
345 radical character, is uphill by 1.50 eV compared to 0.49, showing the vital role of Ir in the
346 OER process.

347 Moreover, increasing the content of Ir from 8% to 15% also decreases the catalytic activity of
348 NCI, as shown experimentally in Fig. 5. To clarify why this increase in Ir content leads to
349 lower catalytic activity than that of NCI-8, we carried out DFT calculation using a slab model
350 of NCI-15 with an atomic ratio of 16.7% Ir in the NCI as shown in Fig. S19. The NCI-15 slab
351 model was built by choosing the lowest energy model (where the additional Ir is located on
352 the top surface of NCI-8) among five different cases assuming additional incorporation of Ir
353 onto the top- or sub-surface of the NCI-8 model. We then examined the oxidation steps on the
354 NCI-15 as for NCI-8, leading to the lowest OER pathway in Fig. 7.

355 We find that this increased concentration of Ir leads to a higher surface concentration of Ir in
356 NCI-15, making the O radical formation step more endothermic by ~ 0.3 and the O-O
357 coupling step more endothermic ~ 0.1 eV. Even more important, the higher concentration of Ir
358 in NCI-15 makes desorption of O₂ product from its adsorption site on Co (O-O coupling on
359 Co) more endothermic by ~ 0.2 eV (state 4 to state 5 in Fig. 7), which increases the onset
360 potential and reduces the catalytic activity of the whole OER process.

361 These results demonstrate why NCI-8 exhibits the optimum OER activity among various
362 NCI-x systems. Ir doping improves the OER activity of NC by stabilizing the O radical that
363 helps O-O bond formation, but with a higher concentration of Ir on the surface, O₂ bonds too
364 strongly decreasing overall OER activity.

365 3.4. Electrochemical characterization of catalysts

366 To further understand the superior OER performance of NCI-8, the electrochemical active
367 surface area (ECSA) was calculated in terms of the double-layer capacitance (C_{dl}), but it
368 should be noted here that the double-layer capacitance method does have some limitations in
369 estimating the surface area, it may serve as diagnostic tools, but can't be accurately used for
370 quantitative analysis. C_{dl} is measured by using CV measurements in a non-faradaic region
371 near the open circuit potential. Fig. S20 shows the CV curves of the NiCo-based catalysts at
372 different scan rates from 1 mV s⁻¹ to 10 mV s⁻¹. The corresponding C_{dl} values are obtained
373 through the equation: $C_{dl} = i / \nu$ (i is the charging current and ν is the scan rate) [11]. As
374 displayed in Fig. 8, the calculated C_{dl} value of NCI-8 is 5.59 mF cm⁻², which is much higher
375 than that of NC (2.89 mF cm⁻²), NCI-15 (3.71 mF cm⁻²) and NCI-22 (3.61 mF cm⁻²). In
376 addition, the size of pre-OER redox waves can give insight into the electrochemically active
377 surface area [11,46], we use the pre-OER Ni/Co-based redox couples as a way to estimate the
378 number of electrolyte-accessible Ni/Co sites in our system, the results are shown in Fig. S21.
379 It can be clearly observed that the increase in measured double-layer capacitance is related to
380 the increased number of accessible metal redox sites. This suggests that NCI-8 has a large
381 electrochemical surface area, which also provides plenty of active sites for electrochemical
382 redox reactions [47], revealing the excellent OER activity.

383 Electrochemical impedance spectroscopy (EIS) tests were performed to investigate the
384 kinetics of as-prepared electrocatalysts with different atomic ratios of Ir dopants under OER
385 conditions. The Nyquist plots of NC, NCI-8, NCI-15 and NCI-22 are given in Fig. 9, which
386 show two semicircles in the high and low frequency region, corresponding to the fast electron
387 transfer process (resistance from substrate/catalyst interface) and charge transfer process
388 (resistance from catalyst/electrolyte), respectively. The inset in Fig. 9a is the equivalent
389 circuit of the Nyquist plots mentioned above, and the corresponding fitting results are shown

390 in Table S3. The charge transfer resistance (R_{ct}) of NCI-8 is obviously smaller than that of
391 other as-prepared catalysts. It is measured to be only 0.42 Ω , indicating a faster charge
392 transfer process for NCI-8 during the OER process. The extremely low value of R_{ct} is mainly
393 due to the electronic structure modification derived from Ir doping, as well as the hierarchical
394 porous structure of the catalyst and the good binding force between the ZnO/NCC substrate,
395 all these factors contribute to the excellent OER performance of NCI-8.

396 3.5. Durability test

397 High durability of electrocatalyst is a key factor in the practical application. The galvanostatic
398 method was utilized to evaluate the electrocatalytic durability of the as-fabricated NC, NCI-8,
399 NCI-15 and NCI-22 by at 20 mA cm⁻² (Fig. 10 and Fig. S22). We show that all as-fabricated
400 electrocatalysts exhibit superior durability and well-maintained structure (Fig. S23),
401 especially the overpotential of NCI-8 electrode increased only by 3% in the long-time 70
402 hour durability test, indicating that our fabricated NCI-8 electrode possess excellent
403 durability in alkaline media, and is superior to previously reported other state-of-the-art
404 catalytic Ni-Co-Ir, Co-Ir, Ni-Ir systems and electrodes (Table S2).

405

406 4. Conclusions

407 In order to develop efficient and durable electrocatalysts for the oxygen evolution reaction
408 (OER), we designed a catalyst using mostly non-noble metal based materials to obtain
409 excellent OER catalytic activity and durability. We synthesized Ir-doped NiCo oxyhydroxide
410 (NCI) nanosheets with various concentrations of Ir using our novel photochemical deposition
411 method. This leads to the deposition of catalyst layers with uniform and large catalytic active
412 area with enhanced catalytic activity. We expect that the novel photochemical synthesis
413 method developed here may provide improved catalysts for other applications.

414 Based on structural and electrochemical analysis, we found that the most efficient OER
415 activity is for the NCI-8 nanosheets with 8% Ir and 46% Co. The net result is a current of 50
416 mA/cm² at a 260 mV overpotential, one of the best OER performance reported so far.
417 Moreover, NCI-8 leads to stable performance at a current of 20 mA cm⁻² for 70 hours in
418 alkaline media.

419 Our DFT calculations support the experimental results showing that OER on NCI-8 is faster
420 than on NC, because the barrier for forming O radical character on the Ir is much lower while
421 for NCI-15 the product O₂ is too stable to desorb. These results show that Ir and Co play
422 synergetic roles in this catalyst.

423

424 Declaration of competing interest

425 The authors declare that they have no known competing financial interests or personal
426 relationships that could have appeared to influence the work reported in this paper.

427

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435

436 **Appendix A. Supplementary data.**

437 Supplementary data associated with this article can be found online at doi:xxx.

438

Journal Pre-proof

439 **References**

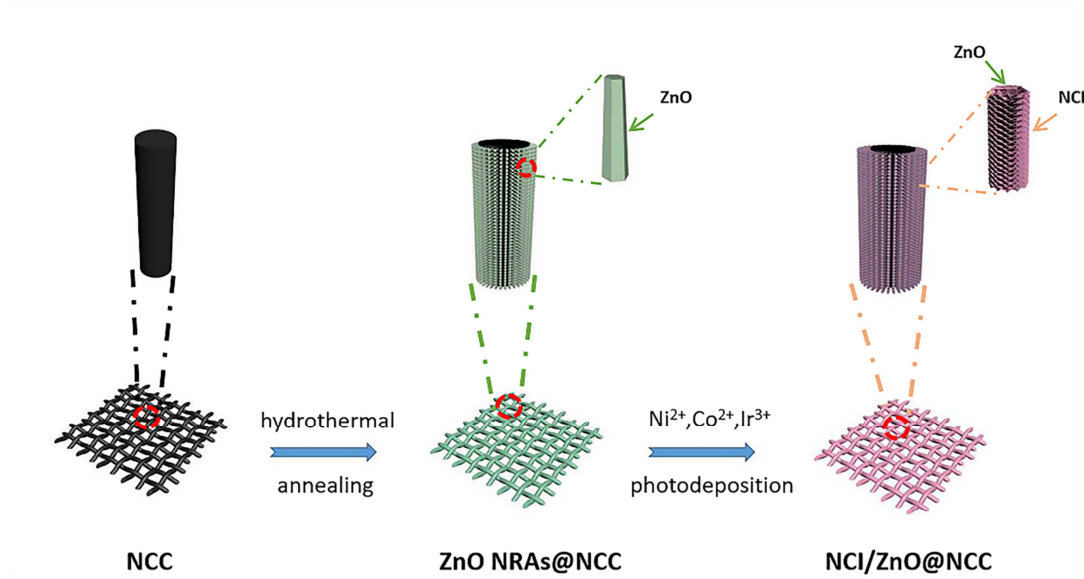
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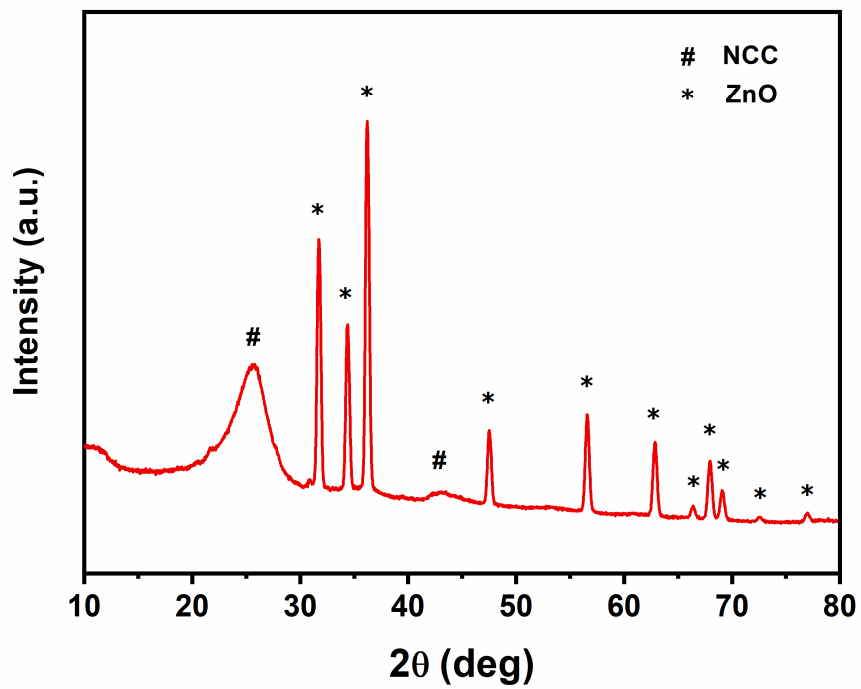
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604 **Fig. 1.** Schematic illustration for the fabrication of hierarchical layered Ir-doped
605 NiCoOOH/ZnO@NCC composite film.

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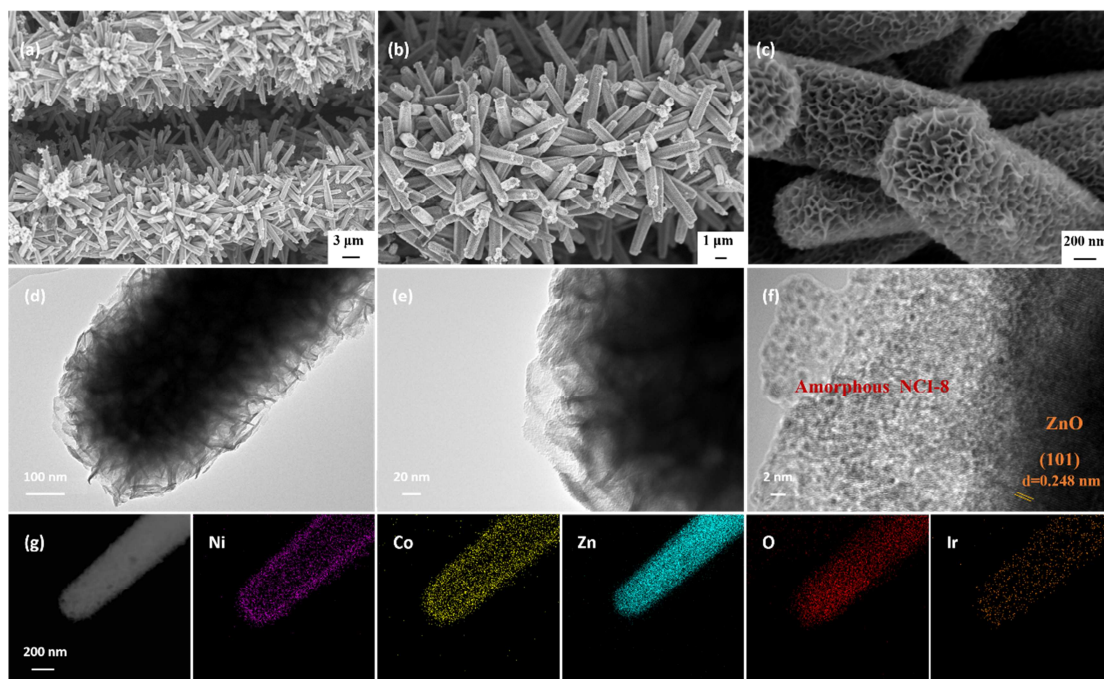


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608 **Fig. 2.** XRD pattern of the as-fabricated NCI-8.

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612 **Fig. 3.** (a-c) SEM, (d, e) TEM and (f) HRTEM images of NCI-8. (g) Elemental mappings of
613 Ni, Co, Zn, O and Ir in NCI-8.

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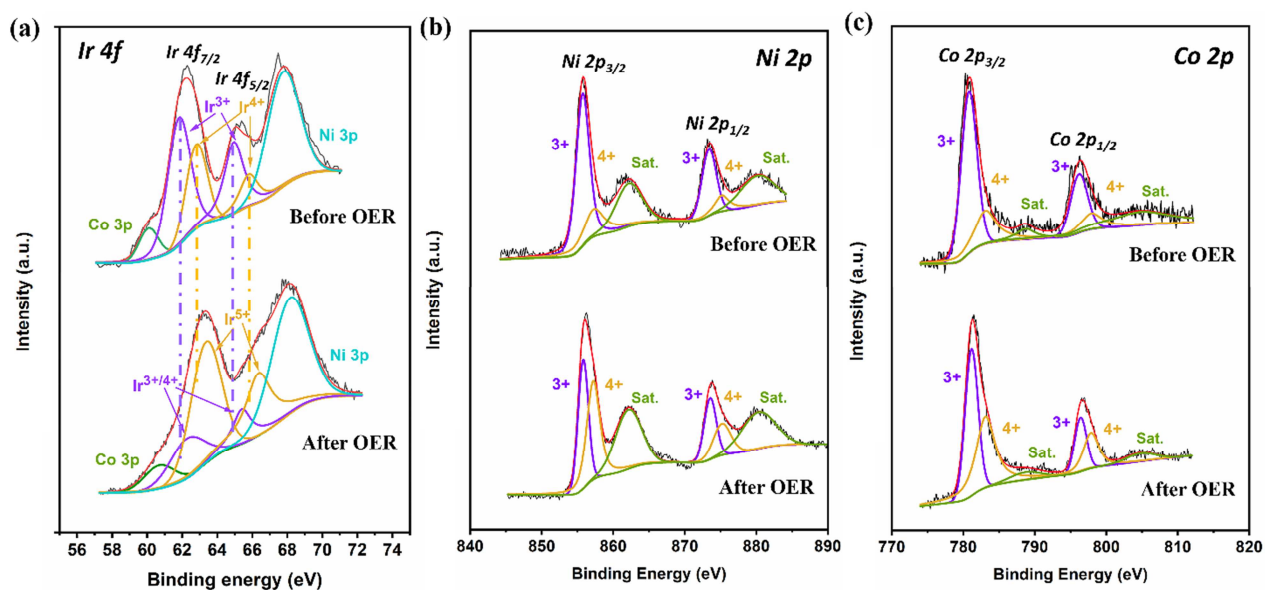
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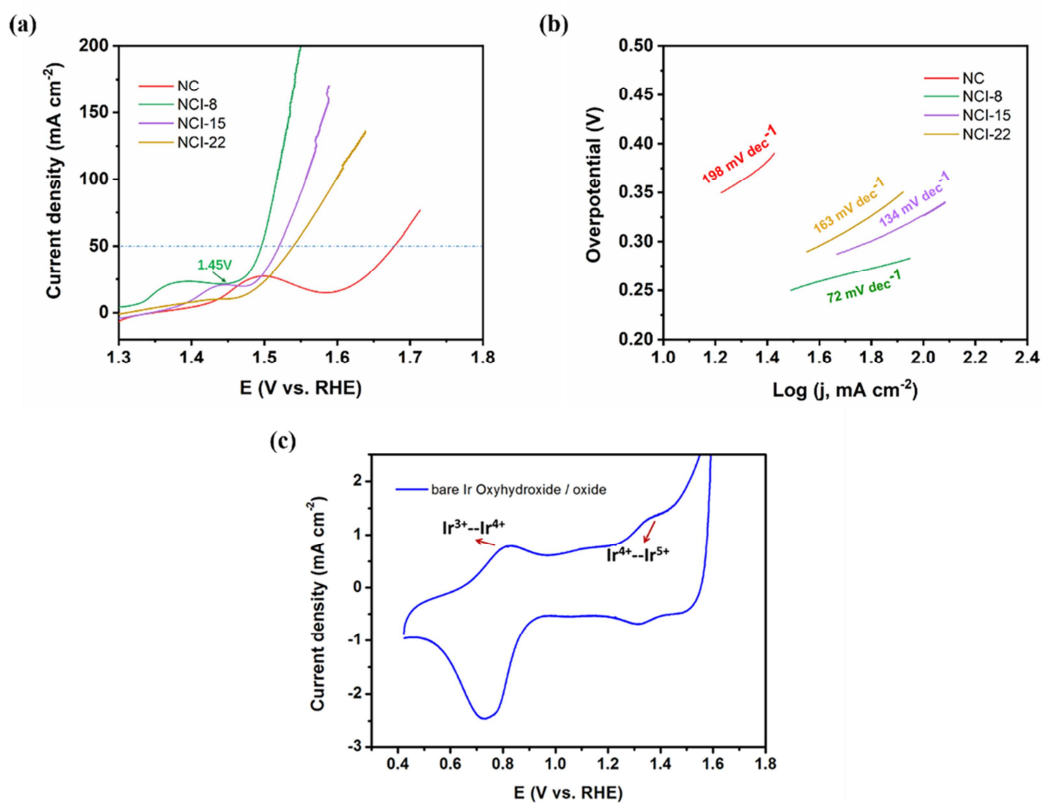
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626 **Fig. 4.** High-resolution XPS patterns of NCI-8 before and after OER. (a) Ir 4f, (b) Ni 2p and
627 (c) Co 2p.

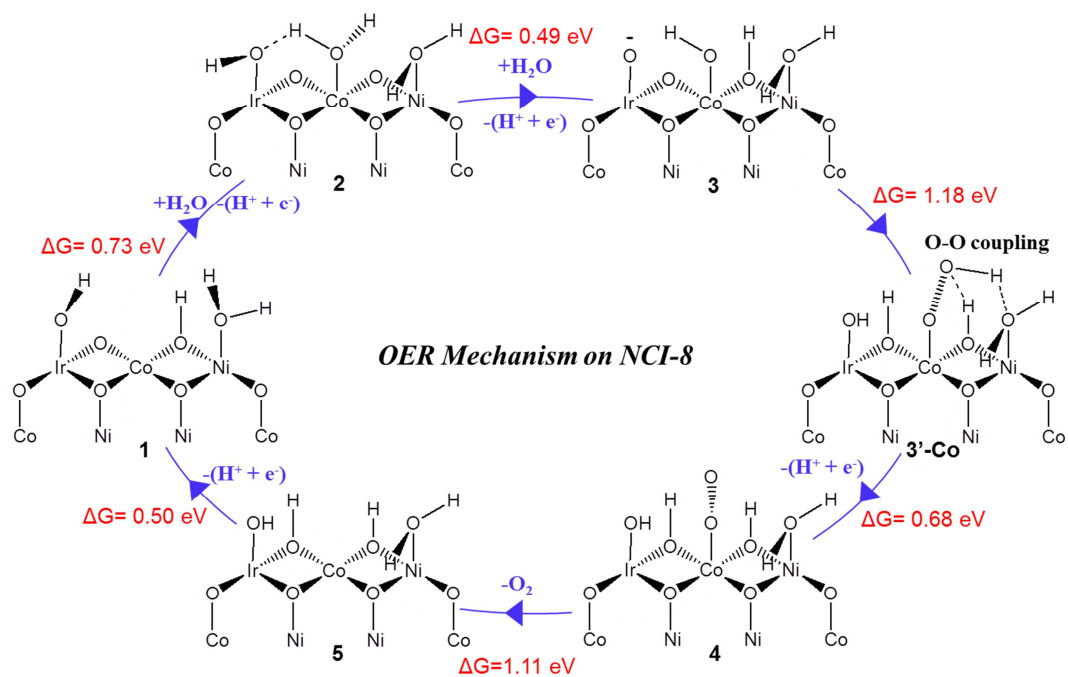
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630 **Fig. 5.** (a) LSV curves of NC, NCI-8, NCI-15 and NCI-22 for OER at a scan rate of 5 mV s⁻¹
 631 in 1 M KOH; (b) The corresponding Tafel plots; (c) the CV curve of bare Ir
 632 oxyhydroxide/oxide at 5 mV s⁻¹ in 1 M KOH.

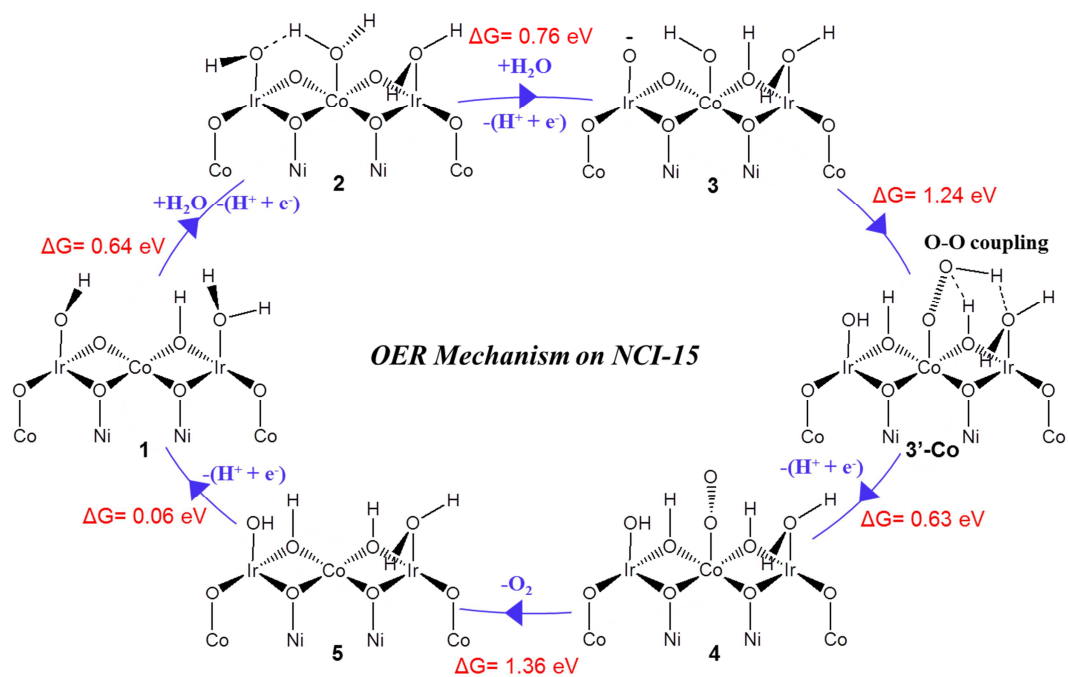
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635 **Fig. 6.** Mechanism for OER on NCI-8 catalyst.

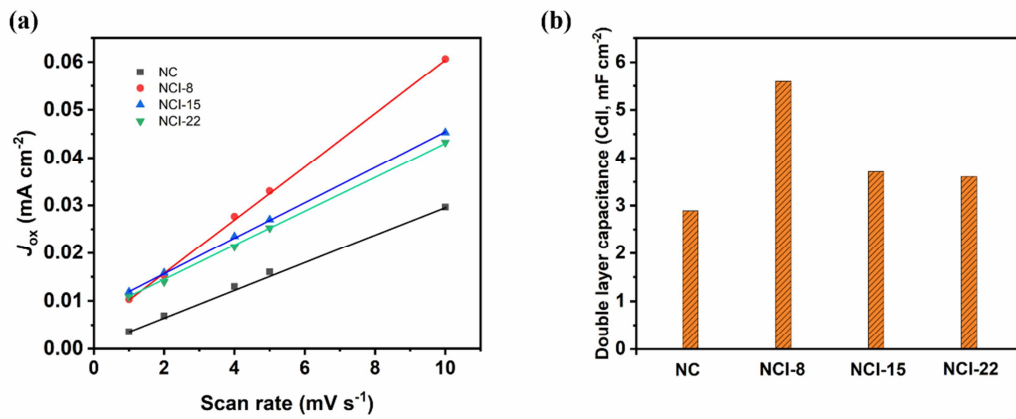
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638 **Fig. 7.** Mechanism for OER on NCI-15 catalyst.

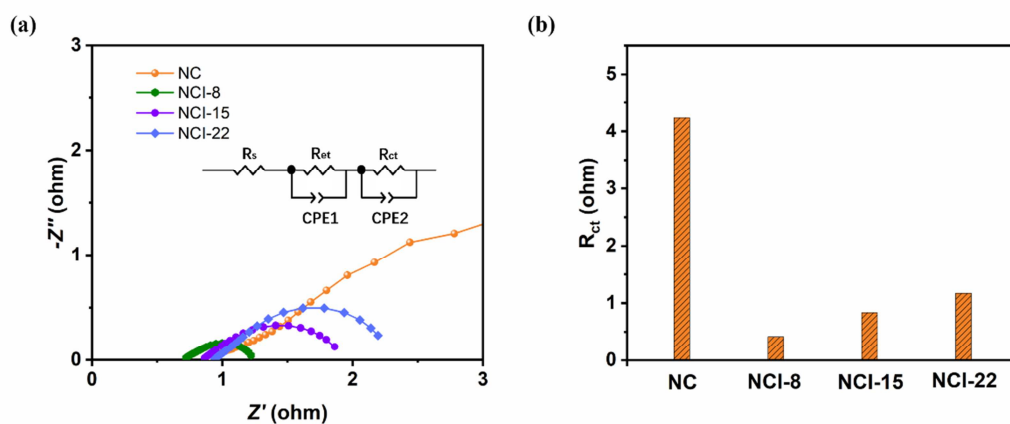
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641 **Fig. 8.** (a) The relationship between scan rate and charge current density of double layer
642 capacitor and (b) the corresponding electrochemical double-layer capacitances of NC, NCI-8,
643 NCI-15 and NCI-22.

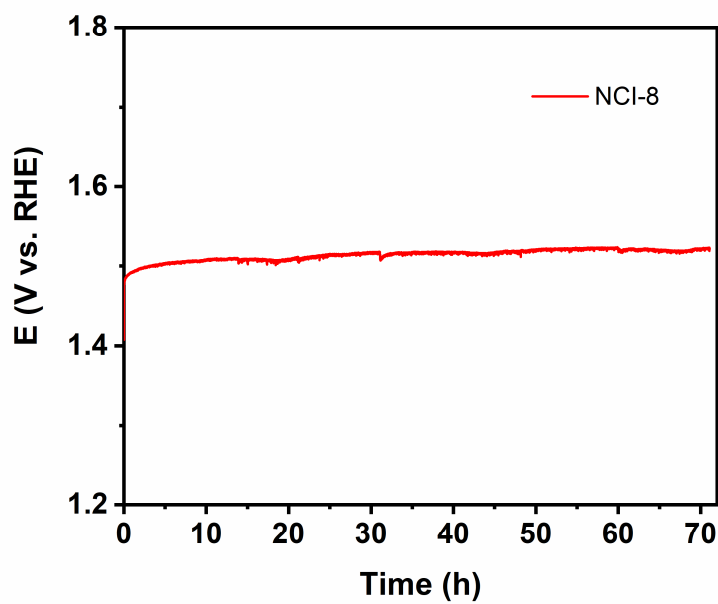
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646 **Fig. 9.** Nyquist plots (a) and charge transfer resistance (b) of NC, NCI-8, NCI-15 and NCI-
647 22 at 1.54 V vs. RHE. Inset in Fig. 9a is the corresponding equivalent circuit model for the
648 Nyquist plots.

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651 **Fig. 10.** Durability test at 20 mA cm^{-2} for 70 hours of NCI-8 electrode.

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Highlights

- Ir-doped NiCo oxyhydroxide nanosheets with 8% Ir and 46% Co (NCI-8) are synthesized using novel photodeposition method that producing uniform and large catalytic active area leading to enhanced catalytic activity.
- NCI-8 shows the best performance for the oxygen evolution reaction (OER) activity, leading to 50 mA cm^{-2} at an overpotential of 260 mV, which provides stable performance for 70 hours in alkaline media.
- Ir and Co play essential bifunctional roles in stabilizing the key O radical intermediate on Ir and promoting the O-O bond coupling on Co, respectively.

Declaration of interests

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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